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A real time neutron scattering experiment for the study of polymer dynamics

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Résumé. — On étudie la dynamique d'un polymère fondu (polystyrène amorphe) de longues chaînes $(M_w = 650\ 000)$ en observant à des instants successifs le facteur de forme statique de la chaîne pendant la relaxation, à longueur fixée, d'un échantillon préalablement étiré uniaxialement. Cette lettre présente des données de diffusion de neutrons aux petits angles pour une série de valeurs de la durée de la relaxation qui sont environ égales, ou supérieures au temps de Rouse de la chaîne. Les données de cette expérience couvrent tout le domaine submoléculaire du vecteur de diffusion, grâce à la combinaison de trois configurations différentes des spectromètres. Une discussion plus complète de la dépendance en temps du facteur de forme est maintenant possible. Une contribution à cette discussion est apportée dans la lettre suivante.

Abstract. — The dynamics of an amorphous polystyrene ($M_w = 650\ 000$) melt is studied by successive observations of the static coil form factor during stress relaxation following a uniaxial deformation. This paper presents neutron scattering data for a duration of relaxation of the order of, or bigger than the Rouse time of the chains. The data cover the entire submolecular range for the scattering vector, as the result of the successive use of three different spectrometer configurations. A more complete discussion of the time dependence of the static form factor is now possible. A contribution to this discussion is given in the next letter.

1. Introduction. — By a real time experiment we mean successive instantaneous measurements of observables in a time dependent system.

The measured quantity is the static coil form factor obtained by small angle neutron scattering in the static approximation. Because the time required for such a measurement is long compared to the characteristic time of the system, we quench the sample for each observation. The sample is a strip of polystyrene (atactic, blend of normal polystyrene, $M_{\rm wH} = 760\,000$, and perdeuterated polystyrene, $M_{\rm w} = 680\,000$, $M_{\rm w}/M_{\rm n} \lesssim 1.3$) [2], suddenly stretched [3] by a factor 3 at a temperature $T_s > T_g$ (glass transition temperature) and maintained at a constant elongation. The stress relaxes while the sample returns to equilibrium, and it is observed at different states of relaxation defined by the duration of relaxation $t_{\rm R}$ following the deformation. The strip is quenched at $t_{\rm R}$ and, subsequently, brought in front of the neutron beam of a small angle scattering spectrometer of the high flux reactor of the Institute von Laue Langevin, Grenoble (France). We assume that the configurations of the coils are not modified by quenching [4], only frozen.

The control of the experiment is difficult but results are now reproducible. This is discussed in reference [1], together with the variation of the transverse radius of gyration and the variation of the form factor in the submolecular range of the scattering vector **q**. The range of scattering vectors in which the static coil form factor has been measured was however limited to the interval $[7 \times 10^{-3}, 2 \times 10^{-2} \text{ Å}^{-1}]$. Since that time, by using successively three different configurations for the spectrometers at the Institute von Laue Langevin, this form factor has been measured in the largest possible scattering vector interval covering the submolecular domain, $[7 \times 10^{-3}, 1.5 \times 10^{-1} \text{ Å}^{-1}]$.

This interval, controlling the scattering experiment, is defined with respect to the characteristic lengths associated with the polystyrene coils. We consider here data corresponding to the orientations of **q** parallel (subscript \parallel) and perpendicular (subscript \perp) to the uniaxial direction.

2. Order of magnitude of polymer inverse characteristic lengths. — A typical length is the radius of gyration R_{σ} . In the isotropic, totally relaxed, state, the radius is for our sample

$$R_{\rm gian} \simeq 250 \,\rm{\AA}$$

The affine value for the quantity corresponding to the elongation 3, are

$$R_{g\perp} = 250/\sqrt{3}$$

 $R_{g\parallel} = 250 \times 3$.

We have checked in reference [1] the validity of the affine deformation immediately after deformation using data in perpendicular direction. Another characteristic length is often proposed in the discussion of polymer melts, although its existence is very questionable. It is the diameter Dof the fixed tube in which a coil is constrained to move during a time interval equal to T_{Rouse} . It is often written :

$$D = R_{g_{iee}} \sqrt{(N_e/N)} \tag{1}$$

where N is the number of statistical units of the chains and N_e is a number of units such as chains with $N > N_e$ show, for various physical quantities observed by various techniques, a behaviour different from that of smaller chains. Depending on the technique, one obtains [7]:

$$100 < N_{a} < 1000$$
 which implies $25 \text{ Å} < D < 90 \text{ Å}$.

A third characteristic length is here the size of the monomer

$$a = 2.5 \text{ Å}$$
 .

Table I shows the characteristics of the three configurations of the two small angle neutron scattering spectrometers used in our experiment to cover the whole range $[7 \times 10^{-3} \text{ Å}^{-1}]$, $2 \times 10^{-1} \text{ Å}^{-1}]$.

From figure 1 it appears that the experiment is suitable to obtain data in the asymptotic range defined by

$$q_{\parallel} R_{g\parallel} \geq 4$$
.

Table I.

Scattering vector range (Å ⁻¹)	Spectrometer	Sample detector distance	Wavelength	Wavelength resolution	Angular collimation (rad.)
$7 \times 10^{-3}, 2 \times 10^{-2}$	D11	10 m	10 Å	9%	5×10^{-3}
$10^{-2}, 6 \times 10^{-2}$	D17	3 m	13 Å	10 %	7×10^{-3}
$6 \times 10^{-2}, 2 \times 10^{-1}$	D11	2 m	5.4 Å	9%	10^{-3}

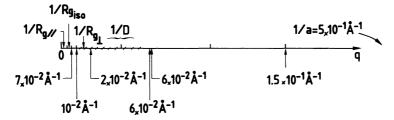


Fig. 1. — Representation of the reciprocal space intervals explored in the three spectrometer configurations used in our experiment (see Table I). The inverse characteristic lengths of the coils, discussed in this letter, are also shown.

Further, we notice that, if the ratio N/N_e is sufficiently large, this asymptotic range can be divided in two subranges qD < 1 and qD > 1. Therefore, if the quantity D should have an effect on the observable, then this effect could be detected.

3. Order of magnitude of times. — Data are gathered at several durations of relaxation $t_{\rm R}$. This value $t_{\rm R}$ is in fact controlled by the time interval between deformation and quenching. It will be convenient to compare $t_{\rm R}$ to characteristic times of the system. Theoretical quantities are proposed such as Rouse time, $T_{\rm Rouse}$, and reptation time [5, 6], $T_{\rm rep}$. The times are not presently measured experimentally; we use an evaluation [7] of $T_{\rm Rouse}$ from viscosity measurements at low molecular weights to evaluate $T_{\rm Rouse}(\alpha N^2)$ for $M_{\rm w} = 650\,000$. We evaluate $T_{\rm rep}$ from the experimental values of terminal relaxation time at this molecular weight [7]. These evaluations are displayed and discussed in reference [1] (written as $T_{\rm eq}$ for $T_{\rm Rouse}$ and $T_{\rm dis}$ for $T_{\rm rep}$). They are strongly dependent upon temperature, T. We assume a Williams Landel Ferry temperature dependence [7] :

$$T_{\text{Rouse}}(\mathbb{T}) = \frac{a_{\mathsf{T}}}{a_{\mathsf{T}_0}} T_{\text{Rouse}}(\mathbb{T}_0)$$
(2)

with

$$\log_{10} a_{\rm T} = \frac{1744 \, ({\rm T} - 100)}{50 + {\rm T} - 100}.$$
(3)

The temperatures used for stretching the melt in our experiment are 113 °C, 117 °C, 122 °C, 128 °C and 134 °C. But we discuss here data obtained at 128 °C and 134 °C. These temperatures are chosen because they allow exploration of a time domain in which

$$1 \leq t_{\rm R}/T_{\rm Rouse} \leq T_{\rm rep}/T_{\rm Rouse}$$
 (4)

	$T = 128^{\circ}C$ $T = 134^{\circ}C$						T İ	
value of q	$t_{R} = 30_{s}$ 42	t _R = 60 _s 44	t _R = 480 _s 43	t _R = 1800 _s 46	t _R = 60 _s 49	t _R = 960 _s 50	isotrope	$\frac{\Delta s_{t_{R}}}{s_{t_{R}}}$
$\begin{array}{c} 5.15 \pm -03\\ 5.77 \pm -03\\ 5.77 \pm -03\\ 7.64 \pm -03\\ 1.02 \pm -02\\ 1.52 \pm -02\\ 1.54 \pm -02\\ 1.55 \pm -02\\$	$\begin{array}{c} 1.88 \pm 03\\ 1.68 \pm 03\\ 1.46 \pm 03\\ 1.57 \pm 03\\ 1.57 \pm 03\\ 1.57 \pm 03\\ 1.58 \pm 03\\ 1.27 \pm 03\\ 1.27 \pm 03\\ 1.27 \pm 03\\ 1.11 \pm 03\\ 9.75 \pm 02\\ 9.61 \pm 03\\ 9.75 \pm 02\\ 7.76 \pm 02\\ 7.71 \pm 02\\ 7.71 \pm 02\\ 6.01 \pm 02\\ 5.54 \pm 02\\ 5.47 \pm 02\\ 5.47 \pm 02\\ \end{array}$	$\begin{array}{c} 1.83 \\ 6.83 \\ 6.83 \\ 6.93 \\ 1.657 \\ 6.93 \\ 1.571 \\ 6.93 \\ 1.571 \\ 6.93 \\ 1.571 \\ 6.93 \\ 1.571 \\ 6.93 \\ 1.571 \\ 6.93 \\ 1.280 \\ 0.33 \\ 0.33 \\ 1.280 \\ 0.33 \\$	$\begin{array}{c} 9.7 \\ 0.3 \\ 1.75 \\ 0.3 \\ 1.55 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 1.65 \\ 0.3 \\ 0$	$\begin{array}{c} 1.926 & 03\\ 1.716 & 03\\ 1.506 & 03\\ 1.596 & 03\\ 1.596 & 03\\ 1.596 & 03\\ 1.526 & 03\\ 1.416 & 03\\ 1.306 & 03\\ 1.306 & 03\\ 1.306 & 03\\ 1.306 & 03\\ 1.366 & 03\\ 1.366 & 03\\ 1.166 & 03\\ 1.166 & 03\\ 1.166 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.666 & 03\\ 1.166 & 03\\ 1.116 & 03\\$	$\begin{array}{c} .04e & 03\\ 1,76e & 03\\ 1,56e & 03\\ 1,56e & 03\\ 1,57e & 03\\ 1,57e & 03\\ 1,57e & 03\\ 1,37e & 03\\ 1,37e & 03\\ 1,27e & 03\\ $	$\begin{array}{c} 2.03e & 03\\ 1.75e & 03\\ 1.75e & 03\\ 1.55e & 03\\ 1.55e & 03\\ 1.55e & 03\\ 1.39e & 03\\ 1.27e & 03\\$	$\begin{array}{c} 1.81e & 03\\ 1.54e & 03\\ 1.26e & 03\\ 1.27e & 03\\ 1.20e & 03\\ 9.59e & 02\\ 8.21e & 02\\ 8.21e & 02\\ 7.55e & 02\\ 6.81e & 02\\ 7.55e & 02\\ 4.69e & 02\\ 4.69e & 02\\ 4.21e & 02\\ 4.03e & 02\\ 3.63e & 02\\ 3.63e & 02\\ 3.63e & 02\\ 2.72e & 02\\ 0.72e & 02\\$	13x
		1.116 03 9.926 02 9.926 02 5.666 02 5.666 02 5.666 02 4.716 02 5.666 02 4.716 02 2.476 02 2.476 02 2.476 02 2.066 02 1.816 02 1.350 02 1.3	1.18= 03 1.05= 03 1.05= 02 7.82= 02 7.05= 02 4.82= 02 4.82= 02 4.82= 02 2.51= 02 4.82= 02 1.55=	11000000000000000000000000000000000000	5.65996 02 6.65996 02 2.9908 02 2.9918 02 2.9918 02 2.9918 02 1.97108 02 1.97426 02 1.97426 02 1.4286 02 1.4286 02 1.4286 02 1.4286 02 1.4286 02 1.4286 01 8.6096 01 8.6096 01 7.00	$\begin{array}{c} 1.3 \\ e & 0.2 \\ 9.5 \\ e & 0.2 \\ 0.2$	$\begin{array}{c} 8 \\ 8 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\$	167
$\begin{array}{c} 3.89e-02\\ 4.54e-02\\ 5.18e-02\\ 5.88e-02\\ 6.48e-02\\ 7.12e-02\\ 7.12e-02\\ 7.62e-02\\ 9.09e-02\\ 9.09e-02\\ 9.09e-02\\ 9.09e-02\\ 9.09e-02\\ 1.10e-01\\ 1.16e-01\\ 1.29e-01\\ 1.36e-01\\ 1.36e-01\\ 1.58e-01\\ 1.58e-01\\ 1.69e-01\\ 1.69e-01\\ 1.69e-01\\ 1.69e-01\\ 1.69e-01\\ 1.69e-01\\ 1.89e-01\\ 1.89e-01\\ 1.95e-01\\ 1.95e-01\\ 2.81e-01\\ 1.89e-01\\	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	1.33e 02 1.00e 02 7.48e 01 5.59e 01 4.46e 01 3.62e 01 2.98e 01 2.98e 01 2.41e 01 2.06 \in 01 1.71e 01	$\begin{array}{c} 1.34 \varepsilon & 02\\ 9.92 \varepsilon & 01\\ 7.33 \varepsilon & 01\\ 5.45 \varepsilon & 01\\ 4.34 \varepsilon & 01\\ 2.83 \varepsilon & 01\\ 2.30 \varepsilon & 01\\ 1.91 \varepsilon & 01\\ 1.91 \varepsilon & 01\\ 1.36 \varepsilon & 01\\ 1.36 \varepsilon & 01\\ 1.36 \varepsilon & 01\\ 1.32 \varepsilon & 00\\ 7.74 \varepsilon & 00\\ 6.32 \varepsilon & 00\\ 5.52 \varepsilon & 00\\ 4.99 \varepsilon & 00\\ 4.99 \varepsilon & 00\\ 4.99 \varepsilon & 00\\ 3.77 \varepsilon & 00\\ 3.41 \varepsilon & 00\\ 3.05 \varepsilon & 00\\ 2.55 \varepsilon & 00\\ 2.55 \varepsilon & 00\\ 2.55 \varepsilon & 00\\ \end{array}$	$\begin{array}{c} 1.22 \in 02\\ 8.95 e 01\\ 6.71 e 01\\ 4.94 e 01\\ 3.89 e 01\\ 3.11 e 01\\ 2.52 e 01\\ 1.75 e 01\\ 1.45 e 01\\ 1.25 e 01\\ 1.45 e 01\\ 1.25 e 01\\ 1.25 e 01\\ 1.45 e 00\\ 5.75 e 00\\ 5.79 e 00\\ 5.79 e 00\\ 5.79 e 00\\ 5.79 e 00\\ 3.84 e 00\\ 4.24 e 00\\ 3.58 e 00\\ 3.26 e 00\\ 3.26 e 00\\ 2.62 e 00\\ 2.62 e 00\\ 2.54 e 00\\ 3.54	$\begin{array}{c} 1.35e & 02\\ 9.87e & 01\\ 7.32e & 01\\ 3.47e & 01\\ 4.37e & 01\\ 2.38e & 01\\ 2.38e & 01\\ 1.99e & 01\\ 1.62e & 01\\ 1.40e & 01\\ 1.22e & 01\\ 1.62e & 01\\ 1.22e & 01\\$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	7.670 01 5.730 01 3.380 01 2.780 01 1.910 01 1.910 01 1.410 01 1.410 01 1.410 01 1.410 01 1.650 01 9.270 00 6.470 00 6.470 00 6.470 00 6.450 00 3.350 00 3.490 00 3.400 00 3.400	657

Table II. — $S_{t_{\mathbf{P}}}(q)$, perpendicular direction.

.

T = 128°C			T = 134°C					
value of q	t _R = 30 _s 42	t _R =60 _s 44	t _R = 480 _s 43	t _R = 1800 _S 46	t _R = 60 _s 49	t _R =960 _s 50	Isotrope	$\frac{\Delta S_{t_R}}{S_{t_R}}$
$\begin{array}{c} 5.16e-03\\ 5.79e-03\\ 6.45e-03\\ 7.08e-03\\ 7.71e-03\\ 9.01e-03\\ 9.01e-03\\ 9.01e-03\\ 9.01e-03\\ 1.03e-02\\ 1.28e-02\\ 1.28e-02\\ 1.28e-02\\ 1.28e-02\\ 1.38e-02\\ 1.38e-02\\ 1.47e-02\\ 1.54e-02\\ 1.55e-02\\	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	6.91e 02 5.47e 02 3.78e 02 2.78e 02 2.78e 02 2.17e 02 1.97e 02 1.53e 02 1.53e 02 1.44e 02 1.23e 02 1.34e 02 1.23e 02 1.07e 02 9.76e 01 8.77e 01 7.22e 01 7.34e 01 6.28e 01	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{c} 2.18e & 03\\ 1.85e & 03\\ 1.35e & 03\\ 1.35e & 03\\ 1.35e & 03\\ 1.07e & 03\\ 2.97e & 02\\ 3.71e & 02\\ 3.71e & 02\\ 4.96e & 02\\ 6.57e & 02\\ 6.57e & 02\\ 5.61e & 02\\ 5.61e & 02\\ 5.61e & 02\\ 4.79e & 02\\ 4.18e & 02\\ 3.87e & 02\\ 3.20e & 02\\ 3.19e & 02\\ 2.96e & 02\\ 2.66e & 02\\ \end{array}$	13%
$\begin{array}{l} 932\\ =& -902\\ =& -902\\ =& -902\\ 2022\\ =& -9022\\ 2022\\ =& -9022\\ 2022\\ =& -9022\\ 2022\\ =& -9022\\ 2022\\ =& -9022\\ 2022\\ 2022\\ =& -9022\\ 2022\\ 2022\\ =& -9022\\ 2022\\$		$\begin{array}{c} 179 \in \ 02\\ 142 \in \ 02\\ 112 \in \ 02\\ 942 \in \ 01\\ 782 \in \ 01\\ 512 \in \ 01\\ 512 \in \ 01\\ 512 \in \ 01\\ 474 \in \ 01\\ 423 \in \ 01\\ 423 \in \ 01\\ 364 \in \ 01\\ 364 \in \ 01\\ 298 \in \ 01\\ 193 \in \ 01\\ 193 \in \ 01\\ 172 \in \ 01\\ 162 \in \ 01\\ 168 \in \$	$\begin{array}{c} 1.92 \\ \bullet 92 \\ \bullet 02 \\ \bullet 01 \\ \bullet 02 \\ \bullet 01 \\ \bullet 02 \\ \bullet 01 \\ \bullet $	$\begin{array}{c} 2.12 \\ 0.12 \\ 0.75 \\ 0.24 \\ 0.$	$\begin{array}{c} 2.5 \\ = & 0.2 \\ 1.75 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ = & 0.2 \\ 1.20 \\ 1.20 \\ = & 0.2 \\ 1.20 $	$\begin{array}{c} 2.95 \\ = 02\\ 2.95 \\ = 02\\ 1.55 \\ = 02\\ 1.55 \\ = 02\\ 1.55 \\ = 02\\ 1.217 \\ = 022\\ 1.007 \\ = 022\\ 1.007 \\ = 022\\ 1.007 \\ = 022\\ 1.007 \\ = 022\\ 1.007 \\ = 022\\ 1.007 \\ = 022\\ 1.007 \\ = 022\\ 1.007 \\ = 012\\ 1.007 \\ =$	$\begin{array}{c} 0.2\\ 0.2\\ 0.2\\ 0.2\\ 0.2\\ 0.2\\ 0.2\\ 0.2\\$	16%
$\begin{array}{c} 3, 6 \leq e \\ 4, 5 \geq e \\ 2, 2 \geq e \\ 5, 2 \geq e \\ 6, 5 \geq e \\ 14 = -02 \\ 2, 7 \leq e \\ 2, 7$	$\begin{array}{c} 2.386 & 01\\ 1.936 & 01\\ 1.576 & 01\\ 1.576 & 01\\ 1.336 & 01\\ 1.166 & 01\\ 9.996 & 00\\ 8.906 & 00\\ 7.036 & 00\\ 6.396 & 00\\ 5.816 & 00\\ 5.816 & 00\\ 5.816 & 00\\ 5.816 & 00\\ 5.816 & 00\\ 3.756 & 00\\ 3.756 & 00\\ 2.756 & 00\\ 2.556 & 00\\ 2.556 & 00\\ 2.566 & 00\\ 2.276 & 00\\ 1.886 & 00\\ 1.886 & 00\\ \end{array}$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{c} 3.03 \\ c 01 \\ 2.48 \\ c 01 \\ 1.98 \\ c 01 \\ 1.66 \\ c 01 \\ 1.41 \\ c 01 \\ 1.23 \\ c $	$\begin{array}{c} 3.64 \\ e & 01 \\ 2.87 \\ e & 01 \\ 1.94 \\ e & 01 \\ 1.63 \\ e & 01 \\ 1.20 \\ e & 00 \\ 1.35 \\ e & 00 \\ 1.42 \\ e & 00 \\ 3.41 \\ e & 00 \\ 1.45 \\ e & 00 \\ 2.45 \\ e & 00 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ $	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{c} 4.98 \\ \bullet \\ 3.86 \\ \bullet \\ 01\\ 2.99 \\ \bullet \\ 01\\ 2.94 \\ \bullet \\ 01\\ 1.45 \\ \bullet \\ 01\\ 1.45 \\ \bullet \\ 01\\ 1.45 \\ \bullet \\ 01\\ 1.28 \\ \bullet \\ 01\\ 1.45 \\ \bullet \\ 00\\ 2.74 \\ \bullet \\ 00\\ 2.74 \\ \bullet \\ 00\\ 2.74 \\ \bullet \\ 00\\ 2.38 \\ \bullet \\ 00\\ 00\\ 2.38 \\ \bullet \\ 00\\ 00\\ 00\\ 00\\ 00\\ 00\\ 00\\ 00\\ 0$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	657

Table II (suite). — $S_{t_{R}}(q)$, parallel direction.

•

To verify (4), one can use the values of T_{Rouse} and T_{rep} tabulated in reference [1] for 117 °C. Using (2), (3) we obtain :

at $\underline{128 \text{ °C}}$: 150 s < T_{Rouse} < 1 500 s, 15 000 s < T_{rep} < 150 000 s at $\overline{134 \text{ °C}}$: 60 s < T_{Rouse} < 600 s, 6 000 s < T_{rep} < 60 000 s

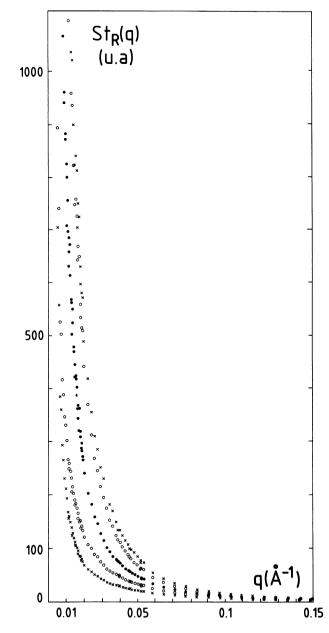


Fig. 2. — Representation of the measured static form factors $S_{t_R}(q)$ versus the scattering vector q, for samples 50 (O), 43 (×) and the isotropic sample (\bullet), in both directions perpendicular and parallel to the stretching. For each anisotropic sample, values of S_{t_R} measured in the perpendicular (parallel) direction are above (below) the values corresponding to the isotropic sample.

values which can be compared to the duration of relaxation of our samples :

at $\frac{128 \text{ °C}}{134 \text{ °C}}$: $t_{\rm R} = 30 \text{ s}$, 60 s, 480 s, 1 800 s at $\frac{134 \text{ °C}}{134 \text{ °C}}$: $t_{\rm R} = 60 \text{ s}$, 960 s.

4. Neutron scattering data. — We now consider the reciprocal space dependence of the static coil form factors $S_{t_R}(q_{\parallel})$ and $S_{t_R}(q_{\perp})$. These quantities are tabulated in table II for all samples stretched at 128 °C and 134 °C. They are obtained in the following way :

— first, the direct neutron scattering signal for each sample is decreased by the amount of incoherent contribution, and the result is normalized to a unique sample thickness (1 mm). These two operations require a precise measurement of the thickness of the sample, which is the most important source of uncertainty. Table II gives an evaluation of the uncertainty in S_{t_R} ;

— second, the data collected from three different spectrometers are normalized in such a way that continuity is obtained between adjacent scattering vector ranges.

Figure 2 shows a plot of such data, versus q, for the isotropic sample and two stretched samples. As $t_{\rm R}$ is increasing, the form factor is relaxing back to that of the isotropic sample. From the time dependence of these form factors, the theories for the coil dynamics in the melt can now be tested. A comparison between data and theories is discussed in the following letter of this issue.

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