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ON THE ATOMIC STRUCTURE OF LIQUID GaAs

C. Bergman, C. Bichara, P. Chieux* and J.P. Gaspard**

Centre de Thermodynamique et de Microcalorimétrie du CNRS, 26 rue du 141e R.I.A., 13003 Marseille, France *Institut Laue Langevin, B.P. 156X, 38042 Grenoble, France **Université de Liège, Institut de Physique (B5), 4000 Sart Tilman, Belgium

<u>Résumé</u> - Comme pour Si et Ge, les propriétés de GaAs changent fortement à la <u>fusion</u> : la phase liquide possède une conduction métallique et sa densitéest supérieure à la phase cristalline semiconductrice. De plus, l'entropie de fusion est anormalement élevée. Les propriétés de la phase liquide peuvent être expliquées par sa structure. Nous avons mesuré le facteur de structure total de GaAs liquide juste au-dessus de son point de fusion (T = 1250 °C) par diffraction de neutrons sur le diffractomètre D4B à l'ILL de Grenoble. Le facteur de structure et la fonction de corrélation de paires sont semblables à ceux du silicium et du germanium. Cependant, le nombre de coordination mesuré est $5,5 \pm 0,5$, inférieur à la valeur correspondante (6,4) de Si et Ge. Différents modèles de structure sont discutés.

<u>Abstract</u> - Like Si and Ge, the properties of GaAs change drastically upon melting : the molten phase shows a metallic conduction and its density is larger than in semiconducting crystalline phase. In addition, the entropy of melting is anomalously large. The liquid phase properties of GaAs can be explained by its structure. We have measured the total structure factor of liquid GaAs just above its melting temperature (T = 1250 °C) by neutron diffraction on the D4B diffractometer at the ILL (Grenoble). The structure factor and pair correlation function are similar to those of Si and Ge. However the coordination number is 5.5 ± 0.5, lower than the corresponding value (6.4) for Si and Ge. Different structural models are discussed.

I - INTRODUCTION

The crystalline structures of light group IV elements (Si, Ge, α -Sn) and their isoelectronic compounds (III-V and II-VI) are fourfold coordinated at low temperature and low pressure. These structures are traditionally explained by the formation of sp³ hybrids that just point along the direction of the bonds in the diamond or würtzite structures. However, when pressure or temperature is increased, the coordination number increases from 4 to 6 (β -Sn structure) and the sixfold coordinated phase shows a metallic conduction. In addition, when Si or Ge melts, their coordination number increases to 6.4 /1,2/ and the liquid phase is also metallic. In this study we analyse the structure of molten GaAs (isoelectronic to Si). It is known that melting of GaAs is accompanied by a 11 % increase of density /3/ . This results from a large increase in the coordination number (4 \rightarrow 5.5) although the bond length is also increased.

II - EXPERIMENTS

GaAs alloys at 50 at % were prepared from pure metals and filled into cylindrical containers which were made from a \emptyset 6-8.2 mm quartz tube. Before the diffraction experiment, we verified that GaAs didn't react with the quartz at the experimental temperature (1250 °C).

The measurements were carried out on the neutron diffraction spectrometer D4B located at the Institut Max Von Laue-Paul Langevin, Grenoble, France, operating at a wave-length $\lambda = 0.703$ Å. The data were corrected for the quartz container, the 0.1 mm thick vanadium foil heater and the sample self absorption in the usual way /4/. An unexplained time evolution of the sample scattering pattern rendered more difficult the analysis. However, thanks to the fact that we had performed three consecutive scans, we could exploit a large part of the data with a fair degree of confidence after a small correction (3 to 4 % over a full scan) for a linear decrease of the intensities with time. The data were then corrected, as usual, for inelasticity effects, multiple scattering and incoherent scattering /4/. The highest angular values 14 < k < 16 Å⁻¹ which were structureless but very slightly affected by an apparent systematic error (\sim 1 %) were discarded.

III - EXPERIMENTAL RESULTS

i) Structure factor S(k)

The structure factor is presented in Fig. 1. One clearly sees damped oscillations up to 14 Å⁻¹. The first peak is split into two peaks at $k_1 = 2.5$ Å⁻¹ and $k'_1 \ge 3.1$ Å⁻¹; the second peak is situated at $k_2 = 5.3$ Å⁻¹. Let us notice that the ratio k_2/k_1 is slightly larger than 2, the observed value in Si and Ge, which is a rather unusual value; most systems indeed present a ratio k_2/k_1 , close to or smaller than 1.8. According to /5/, in the case of amorphous semiconductors, this ratio seems to be sensitive to the medium range order (in that case, the percentage of edd membered rings in the structure).

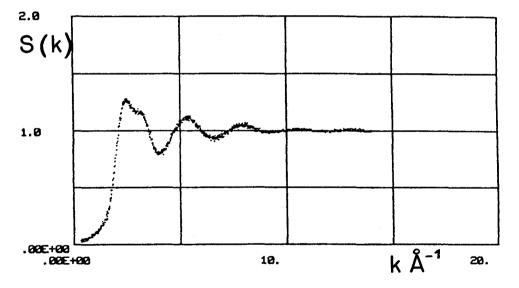


Fig. 1 - Total structure factor, S(k), of $Ga_{0.50}As_{0.50}$ in the liquid state.

ii) Thermodynamic limit

The long wavelength limit $(k \rightarrow 0)$ of the structure factor is related to the macroscopic thermodynamic properties of the system. In the case of a binary alloy, this involves three quantities /6/, the isothermal compressibility, the difference in partial molar volumes and the second derivative of the free energy versus concentration.

Because of the interest in semiconducting industry, the thermodynamic properties of

the GaAs system have been widely studied in the solid state but no experimental study in the liquid state has been reported in the literature. A recent critical analysis /7/ based on experimental determination of the phase diagram, the thermodynamic properties of the solid compound and vapour pressure measurements presents an analytical expression for the excess free energy of formation of the liquid phase $GaAs_{1-x}$

$$\Delta G^{E} = 376.2 - 5.983 \text{ T cal mol}^{-1}$$

This representation implies a regular behaviour for the melt ; from these coefficients, a value of 0.010 is obtained for S(0).

On the other hand, the experimental determination of S(0) from the structure factor data yields a value of 0.03 at the lowest after a supplementary rough isotropic correction for the double scattering between sample and container. This discrepancy can be explained either by an underestimate of the neutron incoherent cross sections or an overestimate of the deviation from ideality in the thermodynamic properties. According to the literature, both reasons can be considered and new experimental information is needed. Concerning the thermodynamic behaviour, a small value of the excess functions of mixing suggests a weak chemical short range order.

iii) Pair correlation function

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The pair correlation function G(r) or $4 \pi r \rho_0(g(r) - 1)$ shows an asymmetric peak at 2.56 Å with a shoulder at about 2.96 Å, above which the structure is very weak except for a peak at about 6 Å (see Fig. 2).

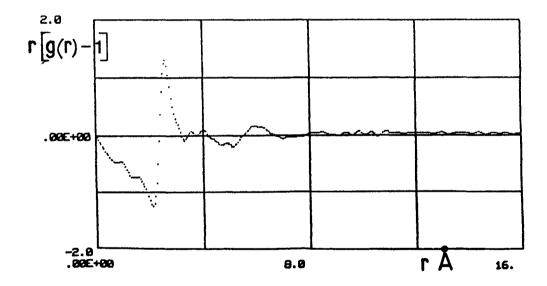


Fig. 2 - Pair correlation function of $Ga_{0.50}As_{0.50}$ in the liquid state.

The absence of oscillations above the first nearest neighbours is quite understandable in the weakly compact structures. By contrast to compact structures (e.g. normal liquid metals) where the atoms are ordered shell by shell so that g(r) oscillates up to 4 or 5 nearest neighbour distances, in liquid GaAs, as there are only \sim 6 nearest neighbours in the first coordination shell, it results to empty interstitial space and disruption of the multilayer shell ordering. Consequently g(r)does not show well defined oscillations above the nearest neighbour separation. The increase in nearest neighbour distance upon melting (crystal : 2.448 Å, liquid : 2.56 Å) is 4.5 % for GaAs (a value to compare with 6 % for Si and 15 % for Ge). This is consistent with quantum mechanical calculations that predict that the interatomic distance gets larger when the coordination number increases.

As usual in liquids, the coordination number is subject to some uncertainty in its determination. We fit the first peak of g(r) with two gaussian curves, the area under which is determined. The fairly narrow gaussian curve centered at 2.56 Å is largely insensitive to the k cutoff value ; it gives a fraction of the coordination number which is 3 ± 0.15. By contrast, the second gaussian curve is poorly identified and in particular, its parameters are strongly affected by the data between 10.5 and 16 $Å^{-1}$. At this stage of the analysis, we conclude that the total coordination number is between 5.3 and 6. Let us notice that a brute force integration of G(r) up to 3.04 Å gives 5.3.

IV - MODELS

To the authors' knowledge, there is no microscopic models of molten Si or Ge (e.g. similar to the Polk model for tetracoordinated amorphous Si). Many authors have discussed the peculiar structure of g(r) for Si and Ge. They divide into two groups : some authors assume that two types of Si(or Ge) atoms are present in the melt : covalent units with coordination 4 and "metallic" Si atoms with a large coordination (~ 12) so that 6.4 is a weighted average of both types of atoms. We reject this hypothesis as long as they are no physical measurements showing the existence of these two species of Si. Other authors argue that liquid Si should be related to a distorted simple cubic (s.c) or β -Sn structure. This is supported by the fact that the p electrons play a central role in the structure. As they have 6 lobes pointing at right angle, the maximum stability corresponds to a coordination number of six in a s.c. or β -Sn structure. We believe that the GaAs structure is also related to the β -Sn structure rather than the s.c. This is supported by the fact that the first peak of g(r) is asymmetric and can be analysed into two peaks (2.56 and 2.96 Å) which would correspond to the two nearest neighbour distances (with coordination numbers 4 and 2 respectively) of the β -Sn structure.

The electronic properties of molten Si and Ge are quite consistent with an hexacoordinated structure indeed calculations have shown /8/ that the s.c. and β-Sn structure give a metallic type cohesion with a finite density of states at the Fermi energy.

Let us endly recall that the entropy of melting is anomalously high in the group IV elements and the III-V (and to some extend the group II-VI) compounds. For example for Si and Ge, $S_f = 7.2$ cal/at.g. whilst for GaAs, $S_f = 8.6$ cal/at.g. The value is reduced for ZnTe to 5.8 cal/at.g.

The leading contribution to the entropy of melting of Si originates in the electronic delocalization in the metallic molten phase /9/. The electronegativity difference should decrease the delocalization of the electrons (they tend to remain near the most electronegative element) as it does for ZnTe, but the effect does not seem to be important for GaAs.

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