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STUDY OF SURFACES COVERED BY THIN FILMS, BY SPEC AND RAMAN SPECTROSCOPY

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Résumé - On décrit ici deux techniques originales, adaptées à l'étude et à l'identification in-situ de films solides fins, apparaissant au cours de traitements électrochimiques : la spectrophotoélectrochimie (SPEC), c'est-à-dire l'analyse spectrale des courants électrochimiques photogénérés dans le film, d'une part, et d'autre part la spectroscopie Raman, utilisée au cours de l'électrolyse. Ces techniques sont appliquées à l'étude de la passivité du fer.

Abstract - Two new techniques, well-suited to the in-situ study and identification of the thin films grown during electrochemical processes are described here: on the one hand the spectrophotoelectrochemistry (SPEC) i.e., the spectral analysis of the photogenerated electrochemical currents in the film, and, on the other hand, Raman spectroscopy during electrolysis. These techniques are applied to the study of the passivity of iron.

The optical techniques are particularly well-suited to the study of the electrochemical systems, since they can be used in-situ during electrolysis. But, when the system consists of a thin solid film covering a metallic electrode, one encounters a failure of most of the usual techniques: for instance, in the particularly complex case of the passivity of iron, where several oxihydroxide or oxide modifications are possible, a knowledge of only the optical indices is insufficient to give a complete chemical identification of the film.

Of course, this chemical identification could be done, without any ambiguity, by Raman spectroscopy, but, with the small amount of material present, one is at the limit of the sensitivity for this technique. Therefore it is useful to have in parallel, spectrophotoelectrochemical measurements (SPEC), giving the shape of the response to the light excitation for a film polarised in an electrolytic solution.

I - EXPERIMENTAL

Raman spectroscopy. We have measured spectra with DILOR RT 30 spectrometer with an interface specially designed for an APPLE 2 computer, which allows the accumulation of spectra and the substraction of background. This capability seemed to us useful for substracting the spectrum of the electrolytic solution. The sample was placed flat in the electrolyse cell, it received illumination at grazing incidence by a laser beam; the Raman light collected at right angle entered the spectrometer by a microscope passing through one mm solution. The output of laser power is generally 400 mW, and we use 4880 Å radiation, the results being the same with the 5145 Å.

We observed no noticeable change in the spectra between TE or TM modes of propagation. The spectral range available for the study of iron oxides was between 200 and 750 cm⁻¹

 $\underline{\mathsf{SPEC}}$. When a semiconducting material is illuminated with a light energy higher than the edge of its optical absorption, and when it is polarized by an electrolyte, one observes a sharp increase of the current, i_{Ta} , followed by a decrease, i_{ta} , and a stabilisation to a steady-state value, $\mathsf{i}_{\mathsf{SSa}}$, which is different from the value of the dc electrolyse current. When the sample is darkened, one observes another transient current, i_{tC} , having an opposite sign and a value comparable to the value of i_{ta} . One can say, to a first approximation, that i_{Ta} gives a picture of all the sites where an e-h pair can be created, i.e. it represents the optical absorption in the semiconducing material, coming from a charge transfer phenomenon; the currents ita and itc give a picture of the sites where the trapping of one of the carriers can occur. The spectral distribution of these different currents, and the time constants of the transients, can give valuable information about the material, and can lead to its identification.

However, these photocurrents are generally weak, over all in the present case (thin film) where the amount of material is low. To overcome this difficulty, we have built a special differential amplifier which allows to separate the photocurrent having an amplitude of about 0.5 nA from the very much larger dc currents. Under these conditions, and the illumination of a 450 W Xenon lamp, we are able to follow the passivity of iron overall its range, and to work with films having thickness between a monolayer and 50 Å.

<u>Electrochemistry</u>. The iron samples were polarized in different electrolytes, and for instance in the sulfuric acid solution, the passive range extends from 0 to 1V/SSE (reference electrode at saturated sulfate).

Standards. These two methods required standards: we have used single crystals of hematite, $\alpha\text{-Fe}_20_3$, and magnetite, Fe $_30_4$. Unfortunately, we had no $\alpha\text{-Fe}_30_4$. Which is supposed to play an important role in the passivation processes. We have also studied the thermally grown oxides on iron to compare the results of the two techniques: when the oxidation temperature is above 400°C , one obtains the hematite, and when it is below 200°C one obtains the magnetite. A range of temperature exists where one produces mixture of these two modifications /1/. These spectrum allows us to explain results found for the case of passive films. For Raman and for SPEC, the reference samples were polarized under the same conditions as the passive iron.

II - RESULTS

Raman Spectroscopy

- 1. Role of the solution. Figure 1 gives the in-situ Raman spectrum (the sum of 4 scans) of a film having a thickness of about 150-200 Å. The technique for increasing the thickness of a film has been shown by Arvia and Pallotta (unpublished results), and involved the use of many very fast oxidation-reduction cycles (0.R.C.). The thickness of films is estimated form the interference in the optical spectrum of oxidized iron. The presence of Fe304 is clearly seen in the figure 1 with the two peaks at 550 cm⁻¹ and 676 cm⁻¹ /1,2/. The presence of electrolyte prevents any observation below 350 cm⁻¹; but after this value, the KOH solution, as the most of passivating solutions, reproduces the water spectrum; its faint broad maximum at 450 cm⁻¹ is even not detectable here. On the contrary, the study in-situ of the passivation in sulfuric acid solution is impossible because of the number and magnitude of the Raman peaks in the range.
- 2. Identification of a passive film. Figure 2 gives the Raman spectrum of a film grown potentiostatically at 0.6~V/SSE in a 0.5 M sulfuric acid solution, and obtained after removing sample out of the electrolyte. This sample has been polarized exactly under the passivation conditions but its thickness was thicker than the "actual" passive film by maintaining the polarization for 60 hours: it has reach a

thickness comparable to these of the precedent sample (figure 1). Here, the two first peaks at 300 cm⁻¹ and 412 cm⁻¹ /3/ are characteristic of Fe₂0₃ and one also sees the two peaks from Fe₃0₄: therefore one has a mixture of magnetite and hematite. This spectrum has some points in common with the spectrum of α -Fe00H published in the literature /4/ but the main difference is in the position of the main peak, which is at 385 cm⁻¹ in the oxihydroxide, and at 413 cm⁻¹ either in pure hematite, or as seen on the thermal oxides, in samples mixed with magnetite. This preliminary study allows us then to ascertain that one has the mixture of oxides, and not the oxihydroxide.

The Raman spectroscopy has now proved for the first time that magnetite is always present, in films grown under conditions near that of the passivation. This is of great significance, because the majority of the theories used in electrochemistry, assign the passivation phenomenon to the crystallization of purely trivalent species.

 $\underline{\mathsf{SPEC}}$. Figure 3 gives the photocurrent behaviour of a Fe_20_3 single crystal. It reproduces the optical spectrum of hematite, except for the transitions below 3 eV which are crystal-field and ligand transitions. /5/. In this case, steady-state and transients currents give exactly the same spectral distribution. The response of a single crystal of magnetite is quite different /6/. The currents i_{Ta} , i_{ta} and i_{tc} have the same spectral distribution, but a steady-state current begins at 3.65 eV, which is the absorption edge of Fe0. This steady-state current is assigned to the presence of Fe2+ ions. This behavior is seen on the figure 4, which is relative to a passive film, obtained at 0.4 V/SSE in a buffer borate solution. A more detailed analysis of the SPEC in single crystals and passive films is done in ref. 6. The important point here is the identification of Fe_20_4.

In the first part of passive range the currents i_{ta} and i_{tc} disappear, and one has only a steady state current with the "Fe²+" structure. One assumes that one has probably a poorly-crystallized oxide, with Fe²+-OH bindings.

III - CONCLUSIONS

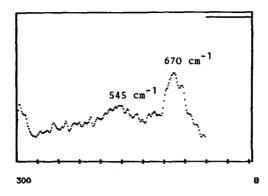
These two approaches have led us point out the presence of divalent iron in the film. The SPEC is an indirect method, which requires a preliminary measurement of standards, and a certain knowledge of the e-h creation and recombination mechanisms in the material. It does not allow us to discriminate between the two modifications having the same Fe-O bindings. Its main advantage is to give a response independent of the crystallinity of the compound, and for thin film as monolayers.

The Raman spectroscopy, even by summing the spectra for 10 scans has not given a response for our passive film eventhough their thickeness may be only 5 times less than the films shown figures 1 and 2. Maybe, the effect of ageing of the sample shown figure 2 is also caused by a higher degree of crystallinity. One has probably a "sandwich" layer, the effect of ageing being to slowly reoxidise the inner layer of Fe $_30_4$, but also to recrystallize the film. One can then considered that these two techniques have an excellent complementarity.

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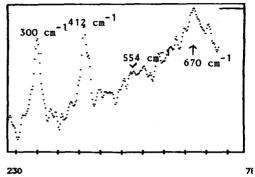
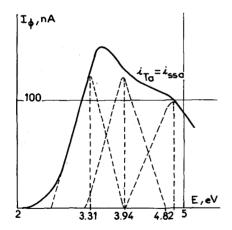
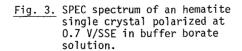


Fig. 2. Raman spectrum of a 200 Å thick passive film after
potentiostatic polarization
at 0.6 V/SSE in 1M H₂SO₄
during 60 h.





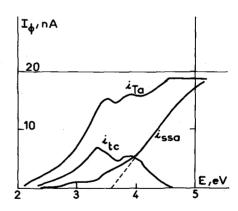


Fig. 4. SPEC spectrum of a passive film polarized at 0.4 V/SSE in buffer borate solution.