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# Electronic structure and electron-phonon interaction in aluminium hydrides

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**Résumé.** — Après implantation d'une forte dose d'hydrogène dans des échantillons d'aluminium, Lamoise *et al.* ont obtenu un accroissement de la température critique de passage à l'état supraconducteur. Puisque les observations expérimentales indiquent la possibilité d'un ordre de l'hydrogène, nous avons étudié la modification de la structure électronique et de l'interaction électron-phonon de Al pur par formation de AlH et AlH<sub>2</sub>. A cet effet, le paramètre électronique  $\eta$  défini par McMillan a été obtenu à partir de nos résultats APW de structure de bande, en utilisant le modèle de Gaspari-Gyorffy. Les bandes s-p du métal sont fortement affectées par l'interaction métal-hydrogène. Pour le mono- et le dihydrure, nous trouvons à basse énergie une bande d'états liants métal-hydrogène ; de plus, pour le dihydrure, une bande antiliante due à l'interaction hydrogène-hydrogène apparaît en-dessous du niveau de Fermi  $E_{\rm F}$ . Les deux hydrures sont métalliques ; leur densité d'états à  $E_{\rm F}$  est de l'ordre de 25 % plus élevée que celle du métal pur. Les principaux résultats obtenus pour  $\eta$  peuvent être résumés comme suit : alors qu'au site de l'aluminium, pour le mono- et le dihydrure,  $\eta_{\rm Al}$  décroît de 50 % à partir de sa valeur pour le métal pur, nous obtenons des valeurs élevées de  $\eta_{\rm H}$ , plus fortes que celle de  $\eta_{\rm H}$  dans PdH. Ceci est essentiellement dû à la valeur des densités d'état de symétrie s et p au site de l'hydrogène et fait de ces hydrures des métaux simples, de bons candidats à la supraconductivité.

Abstract. — The superconducting transition temperature of high dose H implanted Al samples has been found by Lamoise *et al.*, to be higher than that of pure Al. Since experimental observations indicate the possibility of H ordering, we have studied the changes in the electronic structure and the electron-phonon interaction of pure Al upon formation of AlH and AlH<sub>2</sub>. For this purpose, the McMillan parameter  $\eta$  was determined from our augmented plane wave band structure results, using the Gaspari-Gyorffy model. The metal s-p bands are rather strongly affected by the metal hydrogen interaction. For both the mono- and the dihydride, a metal-hydrogen bonding band is found at the low energy side; in addition, a hydrogen-hydrogen antibonding band appears in the dihydride, below the Fermi energy  $E_{\rm F}$ . Both hydrides are metallic; their density of states at  $E_{\rm F}$  is of the order of 25 % higher than in the pure metal. The essential results obtained for  $\eta$  can be summarized as follows : while at the Al site, for both the mono- and the dihydride,  $\eta_{\rm Al}$  decreases by about 50 % from its value in the pure metal, the values of  $\eta_{\rm H}$  are found to be large, even larger than those of  $\eta_{\rm H}$  in PdH; this feature which can be essentially ascribed to the magnitude of the partial s and p densities of states at the H site makes these simple metal hydrides good candidates for superconductivity.

1. Introduction. — The ion implantation technique can be used to load aluminium samples with hydrogen. An enhancement of the superconducting transition temperature of Al over its bulk value  $T_c = 1.2$  K has been observed by Lamoise *et al.* [1] at Orsay after ion implantation of H and D at low temperatures into Al thin films. A maximum value of  $T_c = 6.75$  K was reached for high hydrogen concentrations in Al samples. Resistivity annealing results [2] of Al after high dose H implantation suggested that H ordering or a phase transformation occurs. Since high dose implantation of hydrogen into palladium is known to lead to high values of  $T_c (T_c^{max} \simeq 8.8 \text{ K})$  [3], it could be speculated that the high value of  $T_c$  in the Pd-H and Al-H systems has a common origin. In the case of the Pd-H system, it is now established from theoretical models [4-6] and experimental observation [7-9] that the electron-optical phonon coupling is responsible for the high value of  $T_c$ . Alternative explanations for the rise in  $T_c$  in H-implanted Al should also be considered since Lamoise *et al.* [1, 10] have also observed an increase in the  $T_c$  of Al by

implantation of heavier ions such as He, C, O, Al  $(T_{\rm c}^{\rm max} = 4.2 \text{ K} \text{ for C implantation})$  which they ascribed to disordering in the Al lattice. Similar disorder effects on  $T_{\rm c}$  have also been obtained on films prepared by quenched condensation of Al with  $O_2$  [11]. The importance of lattice disorder on  $T_c$  has not yet been assessed in H-implanted Al samples; nevertheless, several experiments indicate that besides this possible effect, additional changes in the electronphonon interaction due to the formation of ordered compounds should be considered; Lamoise et al. [1] have shown for instance that the  $T_c$  of a sample of H-implanted aluminium ( $T_c^{max} = 6.75$  K) decreases to the limits found for Al-O and Al-He implantations when additional disorder is created by implantation of a low dose of Al. At the same time, the transition which was very sharp in the case of H-implanted Al samples becomes broad as in the case of disordered samples.

Motivated by the experimental results briefly summarized above, we attempted to approach the problem from a different point of view. Assuming, as suggested by the experimental results [2], that ordered compounds are formed, we have studied the modifications in the electronic structure brought about by the presence of high concentration of hydrogen in an Al f.c.c. lattice. We have then evaluated how these differences in the electronic structure affect the electronic contribution to the electron-phonon interaction.

The lattice location of hydrogen implanted in f.c.c. metals has been investigated [12, 13] by nuclear reaction analysis and channelling techniques. The results indicate that above 35 K the hydrogen occupies the tetrahedral interstices of the f.c.c. metal lattice and that this site occupation seems to be associated with the formation of monovacancies created by the implantation process. However, Bugeat and Ligeon [13] proposed that by analogy with other f.c.c. metals (Ag, Cu, Pd, Pt), the octahedral position for hydrogen in Al should be more stable at temperatures below 35 K. It is to be noted that the relative stability of the tetrahedral versus octahedral interstitial position of hydrogen in an aluminium lattice has not yet been fully ruled out by the theoretical models presently available [14, 15]. Thus, in this work, we considered the two possible site occupancies by studying the monohydride AlH with a rocksalt structure (in which H occupies the octahedral interstices) and the dihydride AlH<sub>2</sub> with the fluorite structure (in which H occupies the tetrahedral interstices). The remainder of this paper is organized as follows : in section 2, we describe the modification of the electronic structure of Al upon hydrogenation by a comparison of the energy bands and densities of states of AlH and AlH<sub>2</sub> obtained by the augmented-plane-wave (APW) method, to the corresponding results for Al. The changes in the characteristics of the states at the Fermi energy  $E_{\rm F}$  are duly emphasized by means of a site and angular momentum analysis of the density of states (DOS)

at  $E_{\rm F}$ . In section 3 the band structure results are used to calculate the McMillan [16] electronic parameter  $\eta$  with the Gaspari and Gyorffy [17] model, based upon the rigid muffin-tin approximation.

2. Electronic structure of AlH and AlH<sub>2</sub>. -2.1AlH. — Since the octahedral site occupancy is generally observed at low temperatures for f.c.c. metals and has also been found from theoretical models, we have investigated the electronic structure of rocksalt structure AlH assuming a lattice constant a = 8.056 a.u. which corresponds to a ~ 5.3 % linear increase in the lattice parameter of pure f.c.c. Al (a = 7.6527 a.u.). The choice of this linear dilatation was guided by the experimental results for f.c.c. Ni and Pd hydrides. The APW method with warped muffin-tin potential was used as described previously for other metal hydrides [18, 19], to generate the energy eigenvalues at 89 k points in the 1/48th irreducible Brillouin zone (BZ). The energy bands of AlH along several high symmetry directions are plotted in figure 1. In figure 2, we plotted the energy bands of



Fig. 1. — The energy bands of AlH along several high symmetry directions. Energies are in Rydberg.



Fig. 2. — The energy bands of Al along several high symmetry directions. Energies are in Rydberg.

			Al site				H si	te	
k	Energy (Ry)	$z_{s}^{\dagger k}(\ell = 0)$	$z_p^{\stackrel{\rightarrow}{k}}(\ell = 1)$	$z_{d}^{\vec{k}}(l = 2)$	$z_{f}^{k}(l = 3)$	$z_{s}^{\dagger}(l = 0)$	$z_p^{\mathbf{k}}(\ell = 1)$	$z_d^{\downarrow}(\ell = 2)$	$z_{f}^{\dagger}(l = 3)$
Г	2921	0.4094	0.0	0.0	0.0	0.2472	0.0	0.0	0.0
	1.0753	0.2398	0.0	0.0	0.0	0.6234	0.0	0.0	0.0
	1.3954	0.0	0.6851	0.0	0.0309	0.0	0.1072	0.0	0.0026
x	0.1471	0.1483	0.0	0.1105	0.0	0.4952	0.0	0.0011	0.0
	0.3914	0.0	0.4821	0.0	0.0047	0.0	0.0637	0.0	0.0
	0.8384	0.6834	0.0	0.0162	0.0	0.1709	0.0	0.0145	0.0
	0.9000	0.0	0.5561	0.0	0.0328	0.0	0.0904	0.0	0.0011
W	0.1814	0.0	0.0998	0.0907	0.0059	0.5565	0.0	0.0002	0.0001
	0.5314	0.0	0.4351	0.0560	0.0029	0.0	0.0611	0.0026	0.0
	0.5801	0.7102	0.0	0.0156	0.0144	0.0	0.0384	0.0075	0.0002
к	0.1593	0.0188	0.1134	0.0818	0.0045	0.5241	0.0003	0.0004	0.0009
	0.4614	0.0	0.4566	0.0304	0.0029	0.0	0.0623	0.0014	0.0
	0.4912	0.4976	0.0901	0.0452	0.0081	0.0007	0.0483	0.0045	0.0001
	0.9684	0.0	0.5222	0.0427	0.0249	0.0	0.0861	0.0027	0.0009
L	0.0351	0.0	0.2726	0.0	0.0074	0.4105	0.0	0.0011	0.0
	0.2677	0.5402	0.0	0.0599	0.0	0.0	0.0551	0.0	0.0001
	1.0859	0.0	0.2360	0.0	0.0695	0.3713	0.0	0.0092	0.0
	1.2857	0.0	0.7288	0.0	0.0263	0.0	0.0	0.0252	0.0
	1					1			

Table I. — Energy eigenvalues of AlH and angular momentum analysis of the charge  $(z_i^k)$  inside the Al and H muffin-tin spheres for some high symmetry points.

pure f.c.c. Al. If we compare the two sets of results, the first remarkable feature is the presence in the case of AlH of a split-off band at the low energy side of figure 1. The composition of the states forming this low-lying band is shown in table I where we give, for some high symmetry points, an analysis of the charge into its angular momentum components inside the muffin-tin Al and H spheres. Note that the corresponding wavefunctions are normalized in the unit cell volume; the analysis given in table I does not include the charge in the interstitial region. The results listed in table I reveal the importance of the hydrogen-s and metal states in the split-off band. Similar analysis is given in table II for pure f.c.c. Al. The  $\Gamma_1$  state of pure s symmetry in f.c.c. Al is found to be a bonding combination of H-s and Al-s states in AlH. The admixture of metal and H states in the low-lying band is also very clear for other high symmetry points analysed in table I; their energies have been lowered by the metal-hydrogen interaction.

The existence of low-lying metal-hydrogen bonding states in rocksalt structure transition metal hydride PdH was first emphasized by Switendick [20] and also in subsequent work [18, 21]; the photoelectron spectroscopy data of Eastman *et al.* [22] for PdH and more recent data on other transition [23] and rare earth [24] metal hydrides consistently show the formation of such H-derived states. The present calculation Table II. — Energy eigenvalues of Al and angular momentum analysis of the charge  $(z_i^k)$  inside the Al muffin-tin sphere for some high symmetry points.

ŧ.	Energy (Ry)	$z_{s}^{\vec{k}}(\ell = 0)$	$z_{p}^{\overrightarrow{k}}(l = 1)$	$z_{d}^{\vec{k}}(\ell = 2)$
Г.	225	0.7133	0.0	0.0
1 X'	0.374	0.0	0.7005	0.0
x <sub>1</sub>	0.465	0.5366	0.0	0.1976
x' <sub>5</sub>	0.976	0.0	0.7227	0.0003
W <sub>3</sub>	0.527	0.0	0.5968	0.1142
W <sub>1</sub>	0.614	0.0	0.3671	0.2385
W2	0.630	0.8150	0.0	0.0275
к <sub>3</sub>	0.450	0.0	0.6463	0.0610
кı	0.490	0.1450	0.4364	0.1230
кı	0.545	0.5236	0.0601	0.1579
r <sup>1</sup>	0.257	0.0	0.6534	0.0
L'_2	0.258	0.6535	0.0	0.1318
_				

shows that similar metal-hydrogen bonding band exists also in s-p metal hydrides like AlH.

Two structures are observed in the DOS of the low

lying bands of AlH around E = 0.035 Ry and E = 0.147 Ry as can be seen in figure 3. They correspond respectively to a flat band around  $L'_2$  and to a flat band between  $X_1$  and  $K_1$ . As observed for other



Fig. 3. — The total DOS of AlH, full line curve, left-hand side scale ; units are states of both spin/Ry unit cell. The total number of electrons, dashed line and right-hand side scale.

metal hydrides, the position of the low-lying band depends upon the strength of the metal-hydrogen interaction which is directly related to the metalhydrogen distance. With the present choice of lattice constant, the low-lying band of 6.44 eV width is separated by a gap of 1.17 eV from the next band complex. A partial wave analysis of the DOS inside the Al and H muffin-tin spheres is given respectively in figures 4 and 5. We can see that the low-lying band is globally dominated by the hydrogen's contribution; nevertheless, the Al-s component is particularly important at the bottom of the band, where it is even larger than the H-s contribution; at higher energies, the Al-p component and to a lesser extent, the Al-d contribution are also important. It can be seen clearly that the Al-p partial DOS gives an important contribution to the peak in the total DOS observed around  $L'_2$  at E = 0.035 Ry, while the structure in the total DOS found around  $K_1$  and  $X_1$  at  $E \sim 0.15$  Ry is mostly due to H-s states with a much smaller contribution from the metal site. From a rapid glance at figure 3, we can notice that the overall parabolic shape of the total DOS of pure Al is very much altered in AlH.

If we examine the partial wave DOS analysis we can see from figures 4 and 5 that the bands above the energy gap E > 0.268 Ry are mostly composed of metal s and p states; the magnitude of the H-s and p components is however important in this energy range since it is comparable to the metal-d contribution. This feature is specific to the simple metal hydrides;



Fig. 4. — The angular momentum DOS analysis  $n_l$  of AlH inside the Al muffin-tin sphere  $R_{Al} = 2.459$  6 a.u.; l = 0 (full line); l = 1 (dashed line); l = 2 (dotted line). Units are states of both spin/Ry unit cell.



Fig. 5. — The angular momentum DOS analysis  $n_l$  of AlH inside the H muffin-tin sphere  $R_{\rm H} = 1.5683$  a.u.; l = 0 (full line); l = 1 (dashed line); l = 2 (dotted line). Units are states of both spin/Ry unit cell.

it is very different from what we obtained for the transition metal (TM) hydrides of the beginning or of the middle of the TM series for which the partial DOS of s type at the hydrogen site was found to be very small. As we shall see in section 3, the presence of sizeable hydrogen s and p components plays an important role in the strength of the electron-phonon matrix elements. The reader is referred to table I for an analysis of the charge at the points of high symmetry above the energy gap. The  $\Gamma_1$  state at E = 1.075 Ry is an antibonding combination of metal s and hydrogen s functions. In the bands of pure Al, this state lies much above  $\Gamma'_{25}$  and  $\Gamma_{15}$ . A comparison of the bands of AlH and pure Al in figures 1 and 2 shows that the metal s-p bands have been quite strongly affected by the hydrogen interaction. This is in contrast to what is observed in transition metal hydrides where the middle and upper part of the metal d bands are not substantially modified. This difference is due to the larger extension of the nearly free electron functions of Al, as compared to the more tightly bound d states, which results in an increased interaction with the hydrogen site functions.

The sharp peak in the total DOS around  $E \sim 0.45$ Ry corresponds to flat bands around K and U in the  $\Sigma$ and S directions as can be seen in figure 1. The partial DOS analysis of figures 4 and 5 shows that this structure in the DOS is due essentially to the large values of the s and p metal state contributions.

As the low-lying hydrogen-metal bonding band can accomodate 2 electrons, the two remaining electrons fill the bottom of the metal bands hybridized with H-s and p states. The Fermi level falls at 0.924 Ry above the  $\Gamma_1$  point which marks the bottom of the metal-hydrogen band; this is to be compared to the value of 0.819 Ry for pure Al. The higher position of the Fermi level relative to the bottom of the bands should not however be understood in terms of a rigid band model applied to pure Al since we have emphasized the importance of the modification of the bands due to the presence of H in the lattice; the expansion of the lattice plays also an important role. As we can see in figure 1, 3 bands cut the Fermi level, and AlH is found to be metallic. Bands 2, 3 and 4 accomodate respectively 81.1 %, 18.3 %, 0.6 % of the two electrons which partially fill them. The value of the DOS at  $E_{\rm F}$ is 6.4 states of both spin/Ry unit cell. This value is of the order of 25 % larger than that of pure Al which is of the order of 5.0 states (Leonard [25] obtained a value of 5.4 states while Papaconstantopoulos et al. [26] found 4.78 states at  $E_{\rm F}$ ).

The partial DOS analysis at  $E_F$  is given in table III. The p and s metal contributions are dominant at  $E_F$ ; however, the H-s component is very important since it is only 50 % smaller than the metal s and p terms and it is larger than the metal d component. The hydrogen-p partial DOS is about 50 % smaller than the H-s component. The hybridization with hydrogen s and p states at the Fermi level is thus found to be important in the case of AlH, in contrast to the results found for trivalent transition and rare earth metal hydrides. We give in detail in table III the contributions to the DOS from each band; we clearly see that the H-s partial DOS at  $E_F$  is provided essentially by the contribution of the second band. This result is not surprising in view of the presence of metal-hydrogen antibonding states in this band, which we mentioned above.

2.2 AlH<sub>2</sub>. — Since channelling experiments indicate the presence of hydrogen in the tetrahedral interstices we have also studied the fluorite structure aluminium dihydride. We will be concerned here with an ideal perfect lattice without vacancies. Our choice of the lattice parameter was guided by the linear increase of nearly 8 % found experimentally in the formation of rare earth dihydrides. Figure 6 shows that like for fluorite structure transition and rare earth metal dihydrides, two bands are found at low



Fig. 6. — The energy bands of  $AlH_2$  along several high symmetry directions. Energies are in Rydberg.

Table III. — Total DOS  $N(E_F)$  and site and angular momentum analysis  $n_1$  of the DOS at the Fermi energy  $E_F$  for AlH and AlH<sub>2</sub>. The contribution of the different bands at  $E_F$  is also given. Units are states of both spin/Ry unit-cell.

			Al site		H	l site (or site	es)	
		n <sub>s</sub>	$n_{\rm p}$	n <sub>d</sub>	n <sub>s</sub>	$n_{\rm p}$	n <sub>d</sub>	$N(E_{\rm F})$
					<u> </u>	<del></del>		
	band 2	0.488	0.225	0.150	0.520	0.104	0.002	
	band 3	0.579	1.186	0.266	0.130	0.207	0.018	
AIH	band 4	0.174	0.103	0.019	0.001	0.204	0.003	
	total	1.241	1.514	0.435	0.651	0.335	0.023	6.4
	band 3	1.202	0.482	0.099	0.230	0.240	0.014	
AlH,	band 4	0.165	0.556	0.148	0.062	0.190	0.005	
-	total	1.357	1.038	0.247	0.292	0.430	0.019	6.3

Table IV. — Energy eigenvalues of  $AlH_2$  and angular momentum analysis of the charge  $(z_1^k)$  inside the Al and the two hydrogen muffin-tin spheres, for some high symmetry points.

	Al site					2 H sites			
ř.	Energy (Ry)	$z_{s}^{\dagger}(l = 0)$	$z_p^{\mathbf{k}}(\ell = 1)$	$\vec{k}_{d}(\ell = 2)$	$z_{f}^{\vec{k}}(\ell = 3)$	$\frac{1}{z_s}^k (\ell = 0)$	$z_p^{\vec{k}}(\ell = 1)$	$\vec{z_d}^k (\ell = 2)$	$\vec{x}_{f}^{k}(\ell = 3)$
Г	3729	0.2810	0.0	0.0	0.0	0.2935	0.0	0.0	0.0
	0.6428	0.0	0.0	0.0	0.0607	0.7311	0.0	0.0	0.0
	1.3149	0.0	0.0	0.3302	0.0	0.0	0.1526	0.0007	0.0008
	1.3831	0.0	0.6574	0.0	0.0040	0.0	0.0105	0.0227	0.0004
x	0.0055	0.0	0.2023	0.0	0.0049	0.4208	0.0	0.0013	0.0
	0.3311	0.0	0.0	0.1260	0.0	0.5772	0.0	0.0017	0.0
	0.4234	0.4223	0.0	0.0426	0.0	0.0	0.0713	0.0005	0.0001
	0.8482	0.0	0.4183	0.0	0.0123	0.0	0.0988	0.0003	0.0004
	1.1350	0.2882	0.0	0.1369	0.0	0.0	0.0004	0.0199	0.0
W	0.1247	0.0	0.1546	0.0357	0.0030	0.4415	0.0047	0.0011	0.0001
	0.5219	0.5787	0.0	0.0064	0.0054	0.0	0.0700	0.0029	0.0001
	0.6008	0.0	0.2171	0.0656	0.0007	0.0	0.0739	0.0006	0.0
ĸ	0.0667	0.0	0.1804	0.0169	0.0041	0.4285	0.0030	0.0011	0.0001
	0.1581	0.0529	0.1036	0.0528	0.0014	0.4180	0.0106	0.0008	0.0001
	0.4907	0.3859	0.0479	0.0338	0.0017	0.0037	0.0704	0.0014	0.0
	0.8466	0.1553	0.0756	0.0928	0.0083	0.1363	0.0557	0.0044	0.0001
	0.9120	0.0	0.4092	0.0191	0.0103	0.0	0.0952	0.0020	0.0004
L	0030	0.2451	0.0	0.0415	0.0	0.3349	0.0135	0.0004	0.0
	0.0176	0.0	0.2076	0.0	0.0046	0.3530	0.0125	0.0001	0.0001
	1.0437	0.4156	0.0	0.0628	0.0	0.2913	0.0355	0.0052	0.0005
	1.1534	0.0	0.0509	0.0	0.0582	0.3080	0.0369	0.0029	0.0002
	1.1778	0.0	0.5474	0.0	0.0128	0.0	0.0776	0.0103	0.0003

energy which are formed of metal-hydrogen bonding states and hydrogen-hydrogen antibonding combinations. From the results listed in table IV we can see that the lowest  $\Gamma_1$  point is a bonding combination of metal-s and the two hydrogen s states while  $\Gamma'_2$  is an antibonding combination of the two H atom s states, with a residual metal f contribution. The two lowlying bands of width 13.8 eV overlap the hybridized metal s-p bands located above the X<sub>1</sub> point (see Fig. 6). The position of the two low-lying bands and their width which is given by the energy difference between the bonding  $\Gamma_1$  and antibonding  $\Gamma'_2$  states, depend sensitively upon the metal-hydrogen and the hydrogen-hydrogen distances.

As already mentioned for AlH the total DOS of AlH<sub>2</sub> plotted in figure 7 shows strong deviations from the parabolic shape of the pure metal DOS. Several structures are observed : the large value of the DOS around E = 0.0 Ry corresponds to flat bands around L<sub>1</sub> and X'<sub>4</sub> and also L'<sub>2</sub> as we can see in figure 6; another peak structure around E = 0.12 Ry arises from flat bands around W<sub>3</sub> and from the crossing of the  $\Sigma_1$  and  $\Sigma_3$  branches in this energy range. The value of the total DOS of the two low-lying bands then drops



Fig. 7. — The total DOS of  $AlH_2$ , full line curve, left-hand side scale ; units are states of both spin/Ry unit cell. The total number of electrons, dashed line and right-hand side scale.

as E increases and starts rising again for E > 0.423 Ry, above the X<sub>1</sub> point which marks the bottom of the third band. The partial DOS analysis inside the muffin-

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tin Al sphere (see Fig. 8) shows the importance of the metal s contribution in the first low-lying metalhydrogens bonding band, while the metal p states play an important role in the energy range spanned by the second band. The two structures previously described in the total DOS appear clearly in the partial Al-p DOS plotted in figure 8. A metal d contribution is also found



Fig. 8. — The angular momentum DOS analysis  $n_l$  of AlH<sub>2</sub> inside the Al muffin-tin sphere  $R_{A1} = 2.2255$  a.u.; l = 0 (full line); l = 1 (dashed line); l = 2 (dotted line). Units are states of both spin/Ry unit cell.

in the DOS of band 2 essentially. Figure 9 shows that the s component at the H sites gives the major contribution to the total DOS of the two low-lying bands. The contribution of H-p and H-d components to the first two bands is extremely small.



Fig. 9. — The angular momentum DOS analysis  $n_l$  of AlH<sub>2</sub> inside the H muffin-tin sphere  $R_{\rm H} = 1.349$  9 a.u.; l = 0 (full line, lefthand side scale); l = 1 (dashed line, right-hand side scale); l = 2 (dotted line, left-hand side scale). Units are states of both spin/Ry unit cell.

Above the point  $X_1$ , the total DOS is dominated by the metal-s states. For energies larger than E = 0.6 Ry which marks the bottom of band 4, the metal-p contribution is found to rise very sharply and, although it remains smaller than the metal s component, it gives a very important contribution at the Fermi energy. The metal d states have a much smaller contribution. At the H site, the partial DOS decreases sharply in bands 3 and 4 from its large value in the two low-lying bands. On the contrary, band 3 and band 4 are marked by a sharp increase of the H-p contribution to the total DOS.

As can be seen in figure 7, the two metal-hydrogen low-lying bands are filled by 4 electrons and the remaining electron fills the bottom of bands 3 and 4 which accomodate respectively 87 % and 13 % of an electronic charge. Like the monohydride, the dihydride is metallic; the Fermi level lies at 1.066 Ry above the bottom of the bands; this is to be compared to the values 0.924 Ry and 0.819 Ry respectively obtained for AlH and pure Al. The total value of the DOS at  $E_{\rm F}$ for AlH<sub>2</sub> is 6.3 states of both spin/Ry unit cell which is very close to the value obtained for AlH and of the order of 25 % higher than the DOS of pure Al.

The values of the partial DOS at  $E_F$  resulting from an analysis inside the muffin-tin Al and the two hydrogen spheres are listed in table III. As for AlH and pure Al the metal s and p components are dominant at  $E_F$ . Nevertheless, the hydrogen-s and p contributions at  $E_F$  are not negligible since they are larger than the metal d component. From the results listed in table III we can see that for AlH<sub>2</sub> the partial DOS of s type at the H sites is ~ 50 % smaller than for AlH. In the case of AlH<sub>2</sub> the two hydrogen atoms-p partial DOS is slightly larger than the two hydrogen-s contribution ; for AlH, on the contrary, the s component is larger than the p term at the H site.

We also give in table III the detailed contribution to the DOS from the two bands crossing the Fermi level. The H-s contribution is provided mostly by the third band; however, this band has a dominant metal s character (with a smaller metal p contribution); it is very different from band 2 of AlH which was formed of antibonding metal-hydrogen states and thus was giving a much larger H-s character at  $E_{\rm F}$ . In the case of AlH<sub>2</sub>, band 2 which at  $\Gamma'_2$  is formed of antibonding s states of the two H atoms in the unit cell, has a larger s character; this can be seen clearly in figure 9, in the energy range spanned by band 2. Since before starting the calculation, we did not possess any experimental information on the lattice parameter of Al hydrides, one could surmise that, with a smaller lattice expansion of the Al lattice, the  $\Gamma'_2$  point could lie above  $E_{\rm F}$  and in that case, band 2 could give a larger H-s contribution at  $E_{\rm F}$ . This remark should be kept in mind for the discussion of the value of the electron-phonon interaction in Al hydrides which is given in the next section.

(1)

3. Electron-phonon interaction. — The electronphonon coupling constant can be expressed in the following way [16] in terms of the electron-phonon spectral function  $\alpha^2(\omega) F(\omega)$ 

$$\hat{\alpha} = 2 \int \alpha^2(\omega) F(\omega) \, \omega^{-1} \, \mathrm{d}\omega$$

where :

$$\alpha^{2}(\omega) F(\omega) = \frac{1}{(2\pi)^{3}} \int d\mathbf{k} \int d\mathbf{k}' \times \\ \times \sum_{\nu} \frac{|\langle \psi_{k} | \boldsymbol{\epsilon}_{\mathbf{k}-\mathbf{k}',\nu} \cdot \nabla V | \psi_{k'} \rangle|^{2}}{2 M \omega_{\mathbf{k}-\mathbf{k}',\nu}} \delta(\omega - \omega_{\mathbf{k}-\mathbf{k}',\nu}) \\ \times \delta(\boldsymbol{\epsilon}_{k} - \boldsymbol{\epsilon}_{\mathrm{F}}) \delta(\boldsymbol{\epsilon}_{k'} - \boldsymbol{\epsilon}_{\mathrm{F}}) / \int d\mathbf{k} \, \delta(\boldsymbol{\epsilon}_{k} - \boldsymbol{\epsilon}_{\mathrm{F}}) .$$
(2)

In eq. (2),  $\varepsilon_{q,v}$  and  $\omega_{q,v}$ , are respectively the eigenvector and frequency of a phonon of wavevector **q** and branch index v;  $\nabla V$  is the gradient of the screened potential and  $\psi_k$  the electron Bloch function where k stands for both the wavevector **k** and the band index. Direct calculation of  $\lambda$  using eq. (2) is a formidable task since they involve a double integration on the Fermi surface (FS) which requires a detailed knowledge of the phonon dispersion curves and electronic wavefunctions. McMillan [16] proposed an approximate expression for  $\lambda$ :

$$\lambda \simeq \frac{N_{\uparrow}(E_{\rm F}) \langle I^2 \rangle}{M \langle \omega^2 \rangle} = \frac{\eta}{M \langle \omega^2 \rangle}$$
(3)

in which the mean square phonon frequency is defined by :

$$<\omega^{2}> = \frac{\int \alpha^{2}(\omega) F(\omega) \omega d\omega}{\int \alpha^{2}(\omega) F(\omega) \omega^{-1} d\omega}$$
 (4)

 $< I^2 >$  is the square of the electron-phonon matrix element averaged over the FS and  $N_{\uparrow}$  is the DOS per spin at  $E_{\rm F}$ . In eq. (3)  $\lambda$  is defined as the ratio of an electronic contribution  $\eta$  to an effective phonon force constant. The approximation  $\alpha^2 =$  constant is often used in eq. (4) to evaluate  $< \omega^2 >$ . Using the rigid muffin-tin approximation which ignores the screening of the crystal potential associated with a lattice vibration, Gaspari and Gyorffy [17] have shown that  $\eta$  can be easily calculated using the results of band structure calculations :

$$\eta = \frac{1}{N_{\uparrow}(E_{\rm F})} \sum_{K} \frac{E_{\rm F}}{\pi^2} \sum_{l} 2(l+1) \times \\ \times \sin^2 \left( \delta_{l+1}^{K} - \delta_{l}^{K} \right) \frac{n_{l}^{K}(E_{\rm F}) n_{l+1}^{K}(E_{\rm F})}{n_{l}^{K(1)}(E_{\rm F}) n_{l+1}^{K(1)}(E_{\rm F})}.$$
 (5)

.

In eq. (5) the summation on K runs on all the atoms in the unit cell;  $\delta_i^K$  is the phase shift of the muffin-tin potential for site K, calculated at the Fermi energy;  $n_l^K$  is the partial DOS with angular momentum l at site K and  $n_l^{K(1)}$  is the single site DOS defined by :

$$n_l^{K(1)} = \frac{\sqrt{E_F}}{\pi} \left(2 \ l + 1\right) \int_0^{R_{MT}^K} \Re_l^{K^2}(r, E_F) \ r^2 \ dr \qquad (6)$$

where  $\Re_l$  is the radial wavefunction with angular momentum *l*. The integration is carried out inside the muffin-tin sphere of radius  $R_{MT}$  at site *K*. Important contributions to  $\eta$  from site *K* are expected when the difference between the *l* and *l* + 1 phase shifts is close to a resonance, provided that the ratios of the *l* and *l* + 1 partial DOS to their single site values are not small.

As we can see from the results listed in table V, the s phase shift at the H site is large in both AlH and AlH<sub>2</sub>; it is close to a resonance while the phase shifts of higher

Table V. — Value of several parameters entering the calculation of  $\lambda$ . The phase shifts of the scattering muffin-tin potentials  $\delta_1$  are given in radians; the McMillan electronic parameter  $\eta$  is given in  $eV/Å^2$ : (<sup>a</sup>) from Papaconstantopoulos et al. [26]; (<sup>b</sup>) from Gupta and Burger [27].

		$\delta_0$	$\delta_1$	$\delta_2$	$\eta (eV/Å^2)$
				_	
A 1T T	Al site	0.336 5	0.401 4	0.052 8	0.294
AIH	H site	1.185 6	0.040 4	0.001 2	2.292
A 11 T	Al site	0.253 6	0.371 4	0.055 6	0.224
AIH <sub>2</sub>	H site	1.173 8	0.037 9	0.001	0.744
Al ( <sup>a</sup> )		0.344	0.436	0.065	0.558
	Pd site	- 0.511 5	- 0.109 4	2.806 6	0.886
PaH (°)	H site	1.193 1	0.028 0	0.000 6	0.641

angular momentum are much smaller. The strong scattering of the conduction electrons by the hydrogen potential is also found in the case of transition and rare earth metal hydrides such as PdH, NiH, TiH<sub>2</sub>, LaH<sub>2</sub>[27]. Thus, the strength of the H site contribution to the electron-phonon matrix elements in hydrides is expected to be essentially governed by the magnitude of the s and p type partial DOS values. As we mentioned previously in section 2, these values are relatively important in the case of the simple metal monoand dihydrides presently studied since in both cases the Fermi level falls in a portion of the metal s-p bands where the hydridization with H-s and p states is quite important. Particularly, we mentioned that for AlH, the antibonding metal-hydrogen band which crosses the Fermi energy is responsible for the enhanced value of the H-s partial DOS at  $E_{\rm F}$ . In view of these remarks, we expect  $\eta_{\rm H}$  to be quite large for the Al hydrides. We also give in table V, for comparison, the values of the parameters entering the calculation of  $\eta$  in the case of PdH. For AlH<sub>2</sub>, the value of  $\eta_{\rm H}$  for each hydrogen atom is slightly larger than that found in PdH. At the H site, the s-p scattering is dominant since it gives 98 % of the total contribution while the p-d mechanism gives

only 2  $\%_{0}$ . In the case of AlH, the value of  $\eta_{\rm H}$ , which is also dominated by the s-p scattering term is found to be very large; it is a factor of 3.5 larger than  $\eta_{\rm H}$  in PdH. The magnitude of this term is due to the large value of the H-s type partial DOS at  $E_{\rm F}$  and to the magnitude of the H-p partial DOS which is only 50 % smaller than  $n_{\rm s}$ ; this is in contrast to the case of PdH where the p-type DOS at the H site is an order of magnitude smaller than  $n_{\rm s}$ . For AlH<sub>2</sub>, as we can see in table III, the values of  $n_{\rm s}$  and  $n_{\rm p}$  per H site are smaller than in AlH and lead to a smaller value of  $\eta_{\rm H}$ .

The contribution to  $\eta$  provided by the scattering at the metal site is found to be small for both AlH and AlH<sub>2</sub>. The results listed in table V show that, at the metal site, the p and s phase shifts largely dominate the d phase shift, as it is the case for simple metals. The values for pure f.c.c. Al from the work of Papaconstantopoulos et al. [26] are listed in table V for comparison. Since at the Al site the p and s phase shifts are of the same order of magnitude we can see from eq. (5) that the s-p scattering term is expected to be small; the dominant term at the Al site is given by the p-d scattering in both the mono- and the dihydride, where it represents respectively ~ 92 % and ~ 79 % of the values of  $\eta_{Al}$ , the values of  $\eta_{Al}$  found for AlH and AlH<sub>2</sub> are about 50 % smaller than  $\eta_{Al}$  obtained for pure f.c.c. Al using the same approximations. It should however be noted that for simple metals, the expressions of  $\hat{\lambda}$  given by eq. (3) in terms of the ratio of an electronic to a phonon contribution underestimates  $\hat{\lambda}$  as has been shown by Allen and Cohen [28], and that for simple metals, the screening is expected to be more effective than in transition metals and consequently the rigid muffin-tin approximation is not expected to be as good. The values of  $\eta_{A1}$  obtained by means of Gaspari and Gyorffy's model are too small [26]. Nevertheless, the decrease in  $\eta_{A1}$  from the pure metals [26] to the metal hydrides is meaningful since the same model has been used consistently. This decrease suggests that the metal lattice may not be responsible for the increase of  $T_c$ , unless the decrease of the electronic contribution  $\eta_{A1}$  is accompanied by an even larger decrease in the phonon contribution; in spite of the lattice expansion of the hydrides, such a drastic decrease in the effective phonon force constant seems to be doubtful [29]. Our results point out the importance of the electron-phonon matrix elements at the H sites in the aluminium hydrides since the total value of  $\eta_{\rm H}$  is 1.49 eV/Å<sup>2</sup> for AlH<sub>2</sub> and 2.29 eV/Å<sup>2</sup> for AlH compared to 0.64  $eV/Å^2$  for PdH. If the optic

modes are not too hard, the large values presently found for  $\eta_{\rm H}$  should result in a substantial contribution to the electron-optical phonon coupling constant  $\lambda_{\rm H}$ . Sufficient information on the phonon DOS of the aluminium hydrides is not yet available and precludes further estimate of  $\lambda$ , and calculation of  $T_{\rm c}$ . Tunnelling data have been obtained for D implanted Al samples [29] which seem to indicate that the optic modes do not play a role for energies smaller than 50 meV. Further investigations of the tunnelling characteristics at higher energies, now underway at Orsay will provide further insight on the origin of the high value of  $T_{\rm c}$  in the aluminium hydrides.

4. Conclusion. — The study of the electronic structure of the mono-and dihydrides of aluminium shows that upon hydrogenation substantial deviations from the overall parabolic shape of the DOS of the pure f.c.c. metal, occur. As for transition and rare earth metal mono- and dihydrides, we observe the formation of a low-lying metal-hydrogen bonding band and in the case of  $AlH_2$ , of an additional band due to the hydrogen-hydrogen interaction. The new feature is the importance of the admixture of hydrogen s and p states in the metal bands. This hybridization, which is particularly large in the case of AlH at the Fermi energy, is responsible for the large value of the electron-phonon matrix elements evaluated at the H sites : this feature makes these compounds good candidates for superconductivity and should lead to an important electron-optical phonon coupling provided that the optic modes do not have a too high energy. Further conclusion concerning the increase of  $T_{\rm c}$  over its value in pure Al cannot be drawn prior a better knowledge of the vibrational properties of the Alhydrides. As mentioned previously, the rise in  $T_{\rm c}$  could also be partially due to the lowering of the acoustic phonons due to disorder in H-implanted Al samples. Tunnelling experiments now in progress at Orsay should provide a further clue to this interesting problem. The thrust of the present paper was to emphasize the changes in the electronic structure and the electron phonon matrix elements of Al, upon formation of AlH and AlH<sub>2</sub>.

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