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1 **Printability assessment of cement-based materials based on rheology,** 2 **hydration kinetics, and viscoelastic properties**

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6 **Highlights:**

- 7 ▪ Workability, Extrudability, and Buildability (WEB) requirements were linked to material
8 properties.
- 9 ▪ Restructuring parameters were suggested to assess the extrudability of cement-based materi-
10 als.
- 11 ▪ The influence of mix-design parameters on WEB aspects was captured.
- 12 ▪ Advanced rheology, hydration evolution, and viscoelastic properties combined to predict the
13 printability of cement-based materials.
- 14 ▪
- 15 ▪
- 16 ▪

17 **Abstract**

18 Designing printable cement-based materials aims to achieve targeted properties to fulfill the specific
19 performance criteria. At fresh state, the material should satisfy particular rheological requirements
20 to achieve a tradeoff between its workability (W), extrudability (E), and buildability (B). Given the
21 printability criteria to be fulfilled, the selection of suitable mixture design is of the utmost im-
22 portance to ensure successful 3D printing process and good structural performance, which is linked
23 to the material's strength and stiffness properties (E, G) of the print elements. In this study, the
24 printability of different mixtures was evaluated. The properties of the investigated mixtures were
25 varied by mean of paste content and superplasticizer dosages (SP), viscosity-modifying admixture
26 (VMA), and supplementary cementitious materials (SCMs). The isothermal calorimetry, static yield
27 stress evolution, strain sweep, restructuration indices, and ultrasonic wave responses of the investi-
28 gated mixtures were assessed and used to evaluate their printability. Comparison between selected
29 material properties and printing data allowed establishing useful recommendations to select the
30 mixture with targeted properties for successful printing and adequate structural performance.

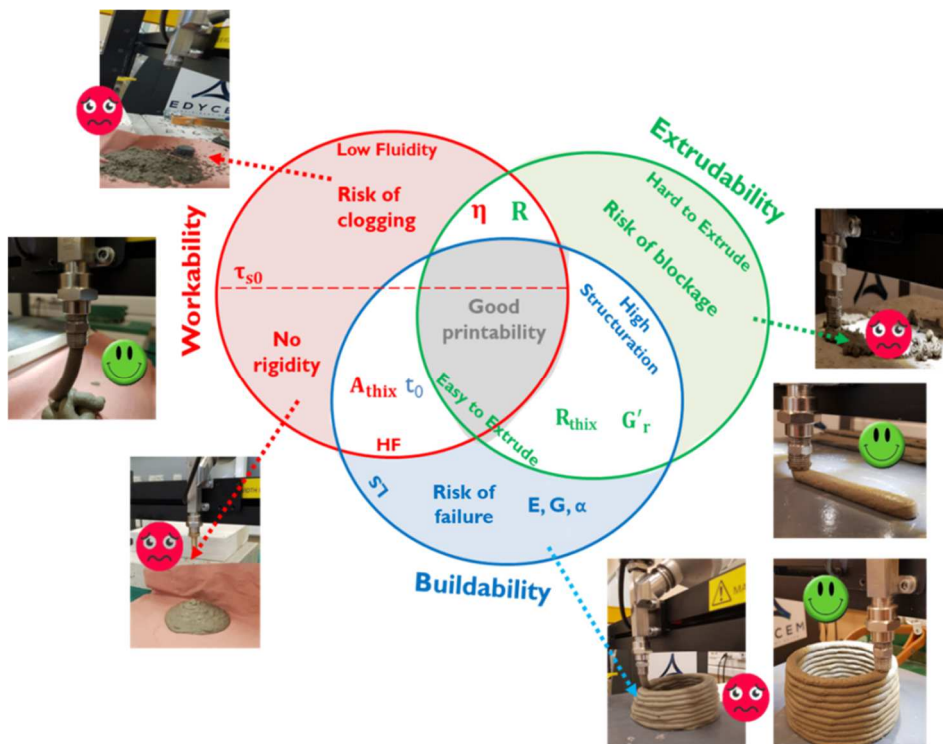
31 **Keywords:** Printable cement-based materials, Workability (W), Extrudability (E), Buildability (B),
32 paste volume, viscosity-modifying admixture (VMA), supplementary cementitious materials
33 (SCMs), rheology, viscoelastic properties, structuration kinetics.

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34 **1. Introduction**

35 Across the board, the covid-19 pandemic has had far-reaching consequences on the world of work.
36 For the construction field, many unscheduled closures of jobsites, supply chain issues, lack of mate-
37 rials, and projects put on hold occurred in early 2020. These issues have had great impact on
38 productivity, workforce, and stunted cash flow for many businesses [1]. Meanwhile, the construc-
39 tion digitalization grasped great interest due to the advantages that 3D printing has shown in coping
40 with this health crisis [1]. Indeed, the use of 3D printing technology in construction industry can
41 reduce the number of workers construction's sites and ensure continuous project processing with
42 remote control via Building Information Modeling (BIM) tools. Moreover, this technology allows
43 more architectural freedom and enhances construction sustainability by minimizing wastes. Also, it
44 reduces the construction time and costs by operating at a constant rate [2, 3], hence improving
45 productivity. However, there are some drawbacks that should be tackled before considering its use
46 at large scale. The major challenge is to design mixtures with adequate properties for successful 3D
47 printing processing. Various types of construction materials with adapted properties for each 3D
48 printing step have been experimented. However, selecting the required properties for each step is a
49 very complicated task, given the different compromises to be achieved. Printable cement-based ma-
50 terials should be fluid to facilitate extrusion, but stiff enough immediately after printing to ensure
51 good buildability. Thus, mastering and controlling rheological properties at each processing step are
52 required to ensure successful printing. This necessitates tailoring very different and sometimes con-
53 trasting rheological requirements depending on the printing step. In order to take into consideration
54 the overall printability aspects, a new WEB approach, reflecting Workability, Extrudability, and
55 Buildability, is suggested in this study to design cement-based materials to be used in the layered
56 extrusion printing process. Unlike the research studies reported in the literature [4–6], where the
57 chosen parameters related to workability and buildability are independent, WEB parameters are
58 chosen in a way to consider the interaction between the three aforementioned aspects. Workability
59 will, therefore, be assessed in terms of the initial static yield stress (τ_{s0}), its evolution rate with time
60 (A_{thix}), as defined by Roussel [7], and the apparent viscosity (η_{app}). The initial static yield stress is
61 a key parameter for 3D printing process, especially during the extrusion and deposition phases in
62 which the material can undergo deformations and collapse, if the generated shear stress during
63 printing exceeds the shear threshold value at that time [8]. Combining the 3 parameters will allow
64 us to evaluate the workability of the mixture while taking into account the rate of structu-
65 ration A_{thix} also providing a useful insight on the buildability aspect, while the apparent viscosity
66 can capture the microstructural changes with time due to flocculation.

67 During 3D printing process, the cementitious mixture undergoes considerable agitation during ex-
 68 trusion, i.e. before being deposited as layers, hence resulting in internal structural and rheological
 69 changes. These changes and their influence on the shape stability of layers must be taken into ac-
 70 count [9]. It is therefore essential to control these variations overtime at the various stages of print-
 71 ing process. The influence of extrusion phase, during which the material can undergo significant
 72 deformations, depends on the pumping pressure, diameter of the nozzle, and the extrusion device.
 73 The rheological changes of print material, induced during the extrusion phase was never considered
 74 while designing printable materials. In this study, the effect of extrusion on the rheological proper-
 75 ties of cement-based materials was evaluated by assessing its restructuration after extrusion, which
 76 defines its **Extrudability**. The recovery of the apparent viscosity (\mathbf{R}), restructuration rate of the
 77 storage modulus (\mathbf{G}'_r), and the rate of re-flocculation (\mathbf{R}_{thix}) as defined by Kruger et al. [10] will
 78 be evaluated to predict the effect of extrusion of the printable material. Immediately after extrusion,
 79 the strength and rigidity of the printable material become paramount. The evolution of the elastic
 80 (\mathbf{E}) and stiffness moduli (\mathbf{G}) with the hydration kinetics will be investigated and exploited to assess
 81 the **Buildability** aspect of the mixtures. On the other hand, the workability period will be estimated
 82 by means of \mathbf{t}_0 which represents the open time, i.e. time period before the beginning of the accel-
 83 eration stage of the cement hydration reaction. The possible interconnections between these pa-
 84 rameters are illustrated in **Fig. 1**. It is worth mentioning that a given parameter may be representa-
 85 tive of several printability aspects at once.



86
 87 Fig. 1. Key Workability, Extrudability, and Buildability (WEB) parameters

88 The above-mentioned process requirements can be adapted given the printing process by tailoring
 89 proper mixture parameters and design approach to achieve adequate properties. In the present study,
 90 different mixture proportioning approaches have been adopted to design cement-based materials for
 91 3D printing. The printability of the investigated mixtures was evaluated according to the WEB con-
 92 cept requirements, as defined earlier. In the first part of this study, the combined effect of paste vol-
 93 ume and SP variation on printability aspects was evaluated. Then, two different approaches to im-
 94 prove the printability of mortar were investigated in the second part. In the first approach, the effect
 95 of viscosity-modifying admixture (VMA) was analysed. The effect of supplementary-cementitious
 96 materials (SCMs) addition was assessed in the second approach. The use of VMA and SCMs aims
 97 to cover wide ranges of rheological properties, thus allowing investigating the relevance of different
 98 printability criteria.

99 **2. Materials, sample preparation and printing process**

100 **2.1 Materials**

101 All the investigated mixtures were portioned using a portland CEM I 52.5 N cement and a compati-
 102 ble polycarboxylate superplasticizer (SP). The chemical compositions and physical properties of
 103 CEM I cement are summarized in Table 1. Fontainebleau sand having a specific gravity of 2.65 was
 104 used to proportion the mortar mixtures. In general, a maximum ratio of 1/10 is often used between
 105 the maximum particle diameter of the sand and the size of the nozzle [9, 10]. However, in order to
 106 prevent any risk of blockage, the maximum particle diameter of the sand was limited to 0.4mm. In
 107 addition to cement, class F fly ash (FA), silica fume (SF), limestone filler (LF), and kaolinite clay
 108 (K) were used. The particle-size distributions of the powder and sand are summarized in Fig. 2. A
 109 viscosity modifying admixture (VMA) was used to modify the rheological properties of the mixture
 110 and enhance its stability after extrusion [13].

111 Table 1. Physico-chemical composition of CEM I 52.5 N cement

<i>Chemical analysis (% by weight)</i>					
CaO	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	SO ₃	MgO
63.7	19.6	4.5	2.3	2.6	3.9
<i>Main compounds of clinker (% by weight)</i>					
C ₃ S	C ₂ S	C ₃ A	C ₄ AF		
69	9	9	7		
<i>Physical properties</i>					
Normal compressive strength (MPa)		Blaine surface area (cm ² /g)		Density (g/cm ³)	
59		3900		3.13	

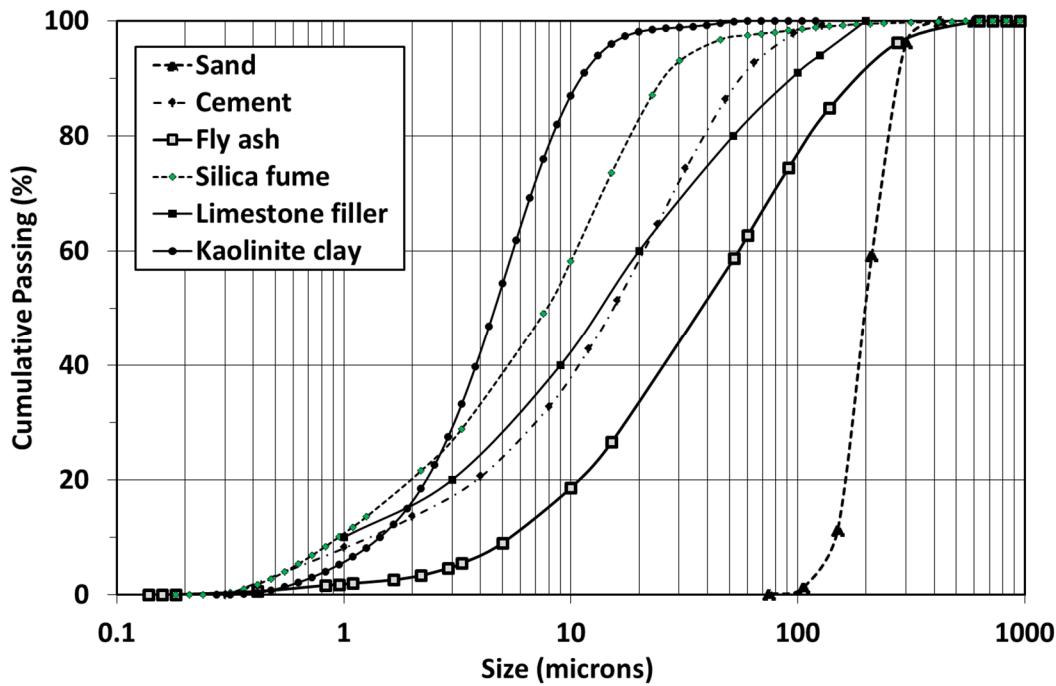


Fig. 2. Particle-size distribution of powders and sand

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113

114 2.2 Mixture proportions and mixing procedure

115 The mixture proportions of the mixtures intended for 3D printing are summarized in Table 2. The
 116 mixtures were proportioned with a water-to-binder (w/b) ratio of 0.28. First, the combined effect of
 117 paste volume and their corresponding SP dosages on printability of mortar mixtures was evaluated.
 118 These mortar mixtures are identified by M100, M80, and M63, indicating their paste contents. For
 119 example, M100 mixture corresponds to plain paste mixture, i.e. mortar made with 100% of paste. In
 120 the second phase of this study, the M63 mortar is then used to evaluate the effect of VMA dosages
 121 on its rheological properties. These mixtures are designed by M-x, where x is the dosage of VMA.
 122 In addition, three additional mixtures were considered to evaluate the effect of replacing fly ash and
 123 silica fume by equal volumes of limestone filler and kaolinite clay. In this part, the M63 mixture
 124 will be noted FA-SF for the sake of simplicity. This aims to investigate the effect of SCMs in im-
 125 proving the printability using ternary binders.

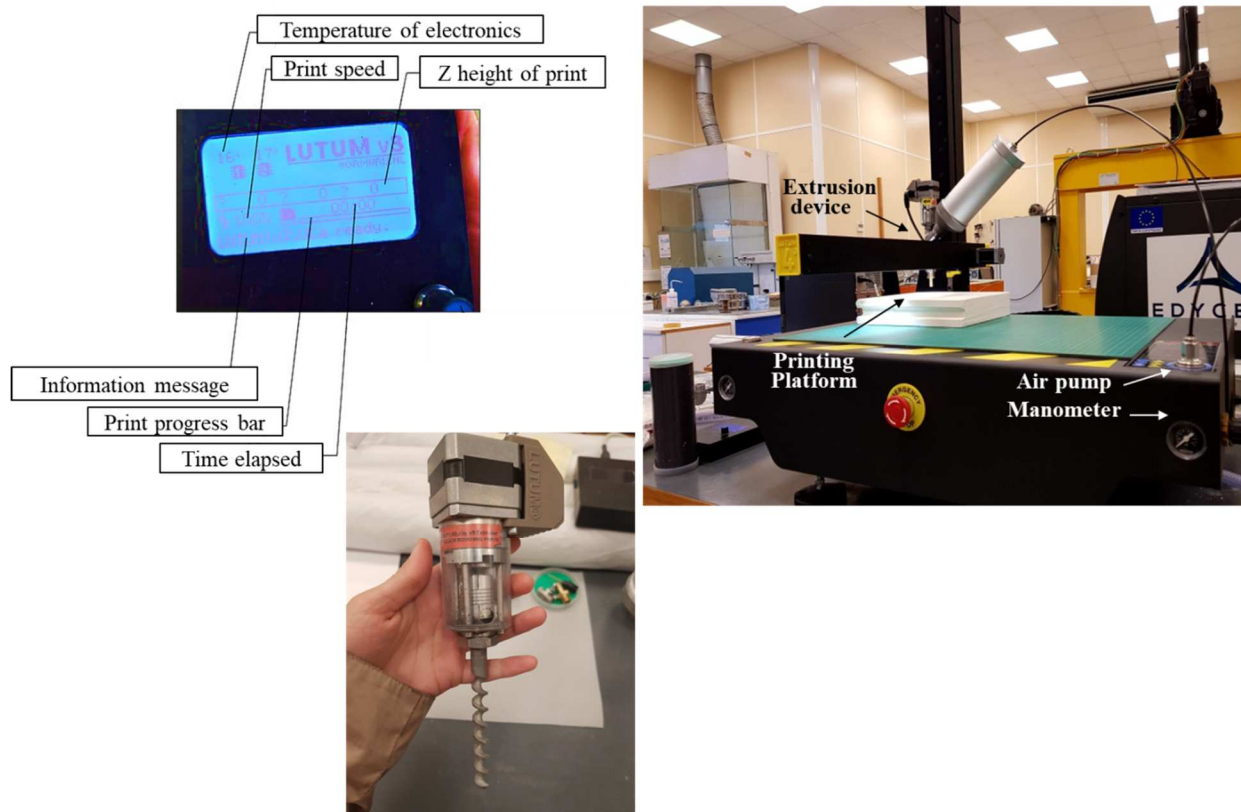
126 The mixtures were prepared using a conventional mortar mixer according to the following protocol.
 127 At first, the mixing sequence consisted in introducing dry materials into the mixer and mixing for 1
 128 min at a low speed (140 rpm). Then, the 2/3 of mixing water was slowly introduced while the mixer
 129 was rotating. After 1 min of mixing, the remaining mixing water along with admixtures were added
 130 and the mixture received another 2 min of mixing. Finally, the edges of the bowl were scraped, and
 131 the speed of mixing was increased (285 rpm) for an additional mixing of 2 min, thus allowing a to-
 132 tal mixing time of 6 min.

133 Table 2. Mixture proportions of the investigated mixtures

Mixtures		Variation of paste volume with SP dosage			Variation of VMA concentration				Effect of LF and K additions		
		M100	M80	M63 FA - SF	M-0.2	M-0.5	M-0.8	M-1.5	FA - K	LF - K	LF - SF
Components (kg/m ³)	Sand	0	530	981	981	981	981	981	981	981	981
	CEM I	1030	825	650	650	650	650	650	650	650	650
	FA	294	235	185	185	185	185	185	185	-	-
	SF	148	118	93	93	93	93	93	-	-	93
	LF	-	-	-	-	-	-	-	-	200	200
	K	-	-	-	-	-	-	-	93	93	-
	Total binder	1472	1178	928	928	928	928	928	928	943	943
water	413	330	260	260	260	260	260	260	260	260	
Admixtures (%, by weight of cement)	SP	0.3	0.5	1.2	1.2	1.2	1.2	1.2	1.2	1.2	1.2
	VMA	-	-	-	0.2	0.5	0.8	1.5	0	0	0
Water to binder ratio (w/b)		0.28									
Paste volume (l/m ³)		100	80	63	63				63		

134 **2.3 Printing process**

135 The printing tests were conducted using a layered extrusion system LUTUM v4.x from *VormVrij*
136 *Company*. This printer is equipped with a 10-mm nozzle diameter functioning under controlled air-
137 pressures and flow rates. First, the mixture was loaded in a cylindrical barrel located above the head
138 of printer. During printing, the pressure generated by compressed air was adjusted given the fluidi-
139 ty of the mixture using a manometer. Extrusion was carried out by means of the worm screw which
140 exerts an additional force on the mixture to push it out of the nozzle. The extrusion device is con-
141 nected to motor and torque limiter to control the extrusion flow rate and the printing speed. These
142 two parameters can be varied from a control box, as shown in Fig. 3. The whole device is mounted
143 on a gantry steering system that allows movement in X, Y, and Z directions (Fig. 3).



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Fig. 3. 3D printer components on the right, control box and the extrusion device on the left

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The printability of the investigated mixtures was evaluated by printing circular hollow cylinders. This was evaluated in terms of ease of extrusion and shape stability of the printed cylinders. These aspects were evaluated according to the pumping pressure and number of printed layers. Several series of side-view images of the printed cylinders were captured to compare the textures of different print mixtures. For each mixture, three printing trials were carried out, and an average extrusion pressure and number of printed layers were recorded. The printing parameters are summarized in Table 3.

Table 3. Printing parameters

Printing velocity (mm/s)	Layer height (mm)	Layer thickness (mm)	Pumping pressure (bar)	Flow rate (mm ³ /s)
20	5	10	variable	275.71

154

3. Experimental program

155

156

In addition to the rheological properties, the heat of hydration and viscoelastic properties of the printable mixtures were investigated.

157

3.1 Rheological Measurements

158

159

The rheological measurements were performed using Discovery hybrid rheometer from *TA Instruments Company*. The vane-in-cup geometry was used to assess the rheological properties of paste

160 and mortar mixtures, as shown in Fig. 4. The single-batch method was used for each rheometric
 161 test. The diameters of the cup and bob are 30.36 mm and 26.20 mm, respectively, hence providing a
 162 shear gap size of 4 mm. The measuring system and tested samples were protected with an enclosure
 163 to prevent water evaporation during measurements. On the other hand, all the measurements were
 164 performed at a constant temperature of 20°C. Three different rheometry protocols were carried out
 165 to assess the static yield stress and its evolution with time, strain sweep, and restructuration tests, as
 166 follow.



167
 168 Fig. 4. The rheometer device and the vane-in-cup used geometry

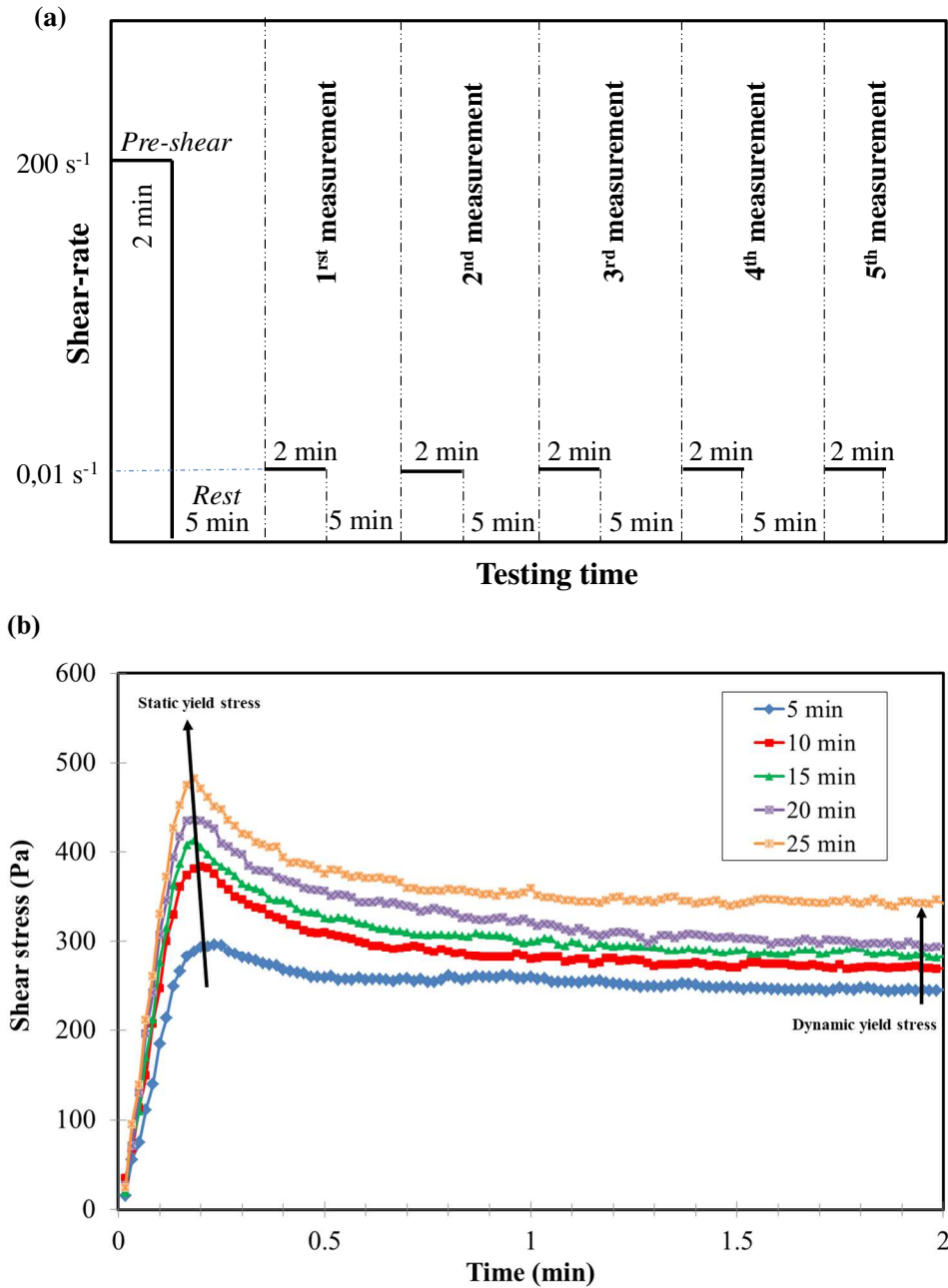
169 **3.1.1 Static yield stress measurements**

170 The static yield stress of the investigated mixtures was measured by subjecting the sample to a con-
 171 stant shear-rate of 0.01 s^{-1} during 2 min. The peak value of shear stress corresponds to static yield
 172 stress, while the shear stress after the thixotropic breakdown corresponds to dynamic yield stress.
 173 Initially, the samples were pre-sheared for 2 min using a rotational shear-rate of 200 s^{-1} to achieve a
 174 well-defined reference state for all the samples. The sample was then allowed a rest period of 5 min
 175 before monitoring the initial static yield stress (τ_{s0}) and thereafter before each measurement (Fig.
 176 5a). The evolution of static yield stress was determined up to 25 min, as shown in Fig. 5b. Since the
 177 printing tests were carried out within the first hour after mixing, the subsequent exponential evolu-
 178 tion of yield stress reported by Perrot et al. [14] was not assessed in this study. The re-flocculation
 179 and linear structuration processes, as well as their corresponding kinetics (R_{thix} , A_{thix}) as defined
 180 by Kruger et al. [10] and Roussel [15], respectively, were evaluated. These parameters have been
 181 calculated using the equations (1) and (2), as follow:

182
$$\tau_s(t) = A_{\text{thix}}t + \tau_{s,\text{trf}} \quad (1)$$

183
$$\tau_s(t) = R_{\text{thix}}t + \tau_{D0} \quad (2)$$

184 where $\tau_s(t)$ corresponds to static yield stress of the material at time t after the pre-shear (Pa), and
 185 τ_{D0} is its initial dynamic yield stress. On the other hand, $\tau_{s,trf}$ corresponds to static yield stress (Pa)
 186 of the material after re-flocculation, and is calculated by subtracting from the initial static yield
 187 stress τ_{s0} , the structuration that took place during the re-flocculation period ($A_{thix,trf}$), as follows:
 188 $\tau_{s,trf} = \tau_{s0} - A_{thix,trf}$ (Fig. 6).



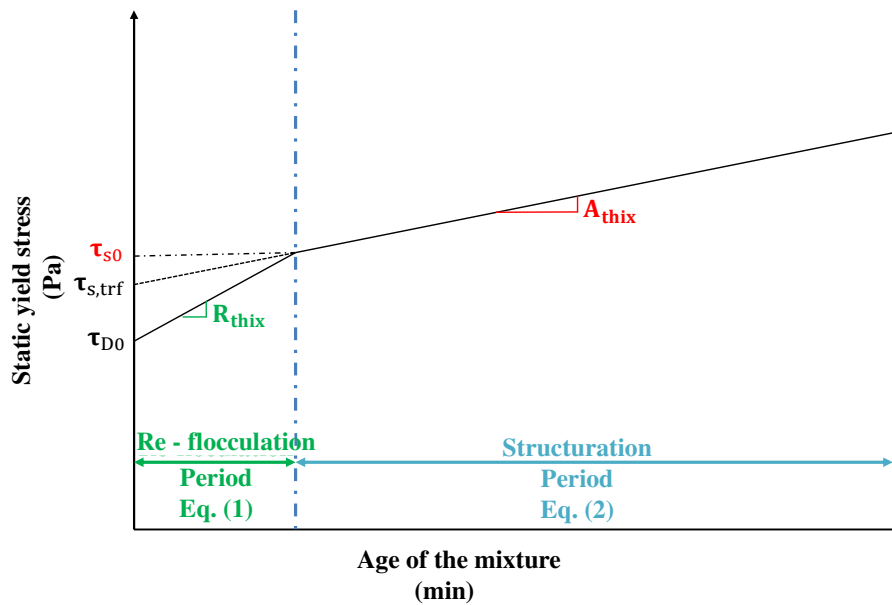
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192

Fig. 5 . (a) Test protocol for yield stress measurement and (b) typical flow onset curve results



193
194 Fig. 6. Evolution of static yield stress with time in both re-flocculation and linear structuration
195 processes [10]

196 3.1.2 Strain sweep

197 Strain sweep measurements were performed on printable mixtures to determine their linear
198 viscoelastic regions. This allowed identifying the critical strain to allow carrying out the
199 restructuration tests under undisturbed shear regime. The determination of the critical strain can also
200 provide useful insight into the fresh-state microstructure of the investigated mixtures.

201 3.1.3 Restructuration kinetics

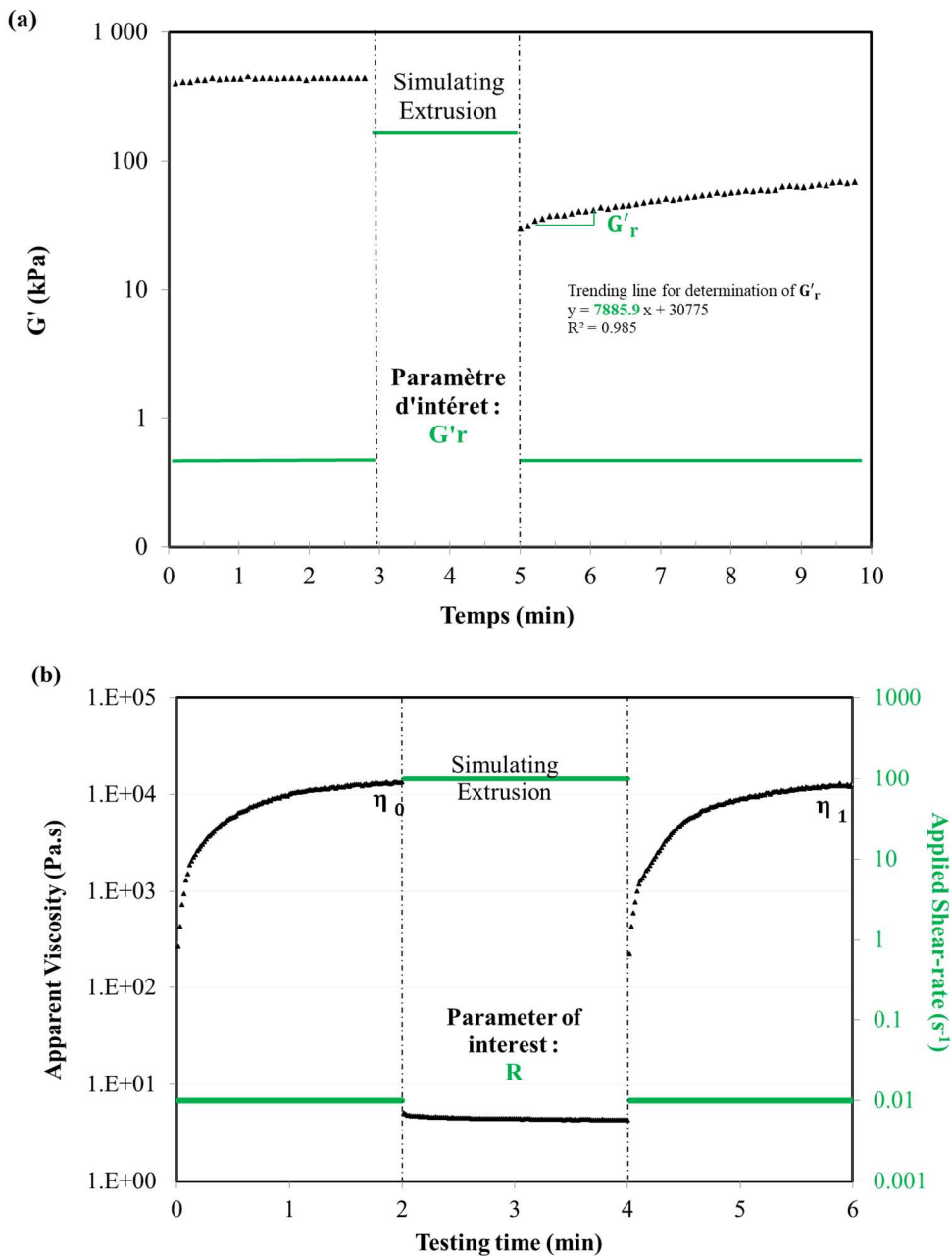
202 Two different approaches were used to quantify the restructuration of the investigated mixtures and
203 evaluate the variations of their rheological properties after extrusion (i.e. viscosity and storage
204 modulus), including the restructuration and recovery rates. These approaches are a new way to
205 evaluate the extrudability of cement-based materials. The first approach, known as three-interval
206 thixotropy test (3ITT), consists in first applying a constant frequency within the Linear Viscoelastic
207 Domain (LVD) to identify a well-defined reference state of the sample in non-destructive manner.
208 Afterwards, higher frequency is applied to breakdown the sample microstructure, hence simulating
209 the extrusion process. The final step is similar to the first one in which the reversible restructuration
210 of the sample is assessed. A strain of 0.001% and 10 rad/s frequency were used for the first and
211 third intervals (Fig. 7a). The restructuration parameter (G'_r) represents the slope of the evolution of
212 G' with time.

213 The second approach consists in assessing the rate of recovery of apparent viscosity after breaking
214 down the structure, as shown in Fig. 7b. First, a very low shear-rate was applied to identify the ref-
215 erence viscosity. A higher shear-rate is then applied to simulate the extrusion process. At the end of
216 shearing regime, the sample was subjected to a low shear-rate of 0.01 s^{-1} to simulate rest condi-

217 tions, which corresponds to the deposit phase after printing. The recovery apparent viscosity η_1 was
 218 then determined and used to calculate the recovery parameter R using Eq. (3), as follow:

219
$$R = \frac{\eta_1}{\eta_0} \quad (3)$$

220 where η_1 and η_0 are the initial and recovery apparent viscosities, respectively.



221

222
 223

Fig. 7. Protocol of restructuring tests and typical results

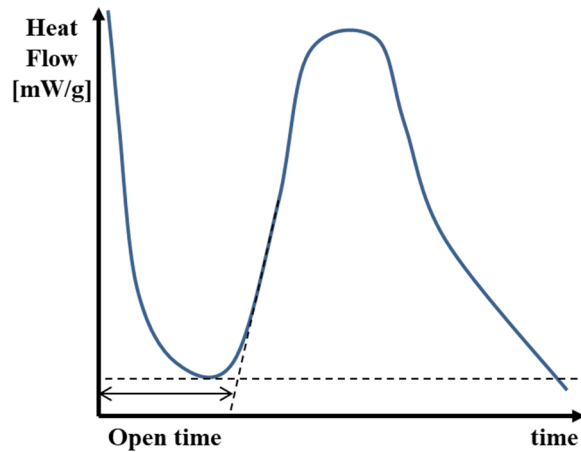
224 **3.2 Calorimeter measurements**

225 A TAM Air isothermal micro-calorimeter was used to determine the hydration degree of the inves-
 226 tigated mixtures. The measurements were performed using 100-g samples at a controlled tempera-
 227 ture of 20°C. The tested samples were prepared using the same mixing procedures described in sec-

228 tion 2.2, and immediately transferred to the calorimeter. The degree of hydration α was calculated
 229 according to Eq. (4), as follow:

$$230 \quad \alpha = \frac{Q(t)}{Q_{\infty}} \quad (4)$$

231 The workability period was estimated using the open time t_o that represents the induction period
 232 (see Fig.8). The influence of mineral additions and VMA dosages on the hydration was evaluated
 233 using the mortar M63 as reference mixture. As for the paste content effect, M100 mixture was taken
 234 as reference. The normalized t_o is used to investigate the hydration process (delayed if t_o is higher
 235 than 1 and accelerated if t_o is lower than 1).



236
 237 Fig. 8. Determination of the open time t_o from the isothermal calorimetry measurement

238 3.3 Elastic properties of the investigated mixtures

239 The *FreshCon* device was used to monitor the structuration kinetic beyond the setting time of the
 240 mixtures. The test was conducted according to the procedures described in the reference [16]. All
 241 the measurements were performed in controlled room temperature of $20 \pm 2^\circ\text{C}$. The monitoring of
 242 the viscoelastic properties can provide useful information regarding the stiffness and stability of the
 243 material after printing. The elastic (\mathbf{E}) and shear (\mathbf{G}) moduli were determined from the velocities of
 244 compression and shear waves according to Eq. (5) and (6), as follow.

$$245 \quad E = \left(\frac{3.v_p^2 - 4.v_s^2}{v_p^2 - v_s^2} \right) \cdot v_s^2 \cdot \rho_c \quad (5)$$

$$246 \quad G = v_s^2 \cdot \rho_c \quad (6)$$

247 **4. Results and discussions**

248 **4.1 Workability and extrudability parameters**

249 *4.1.1 Workability parameters*

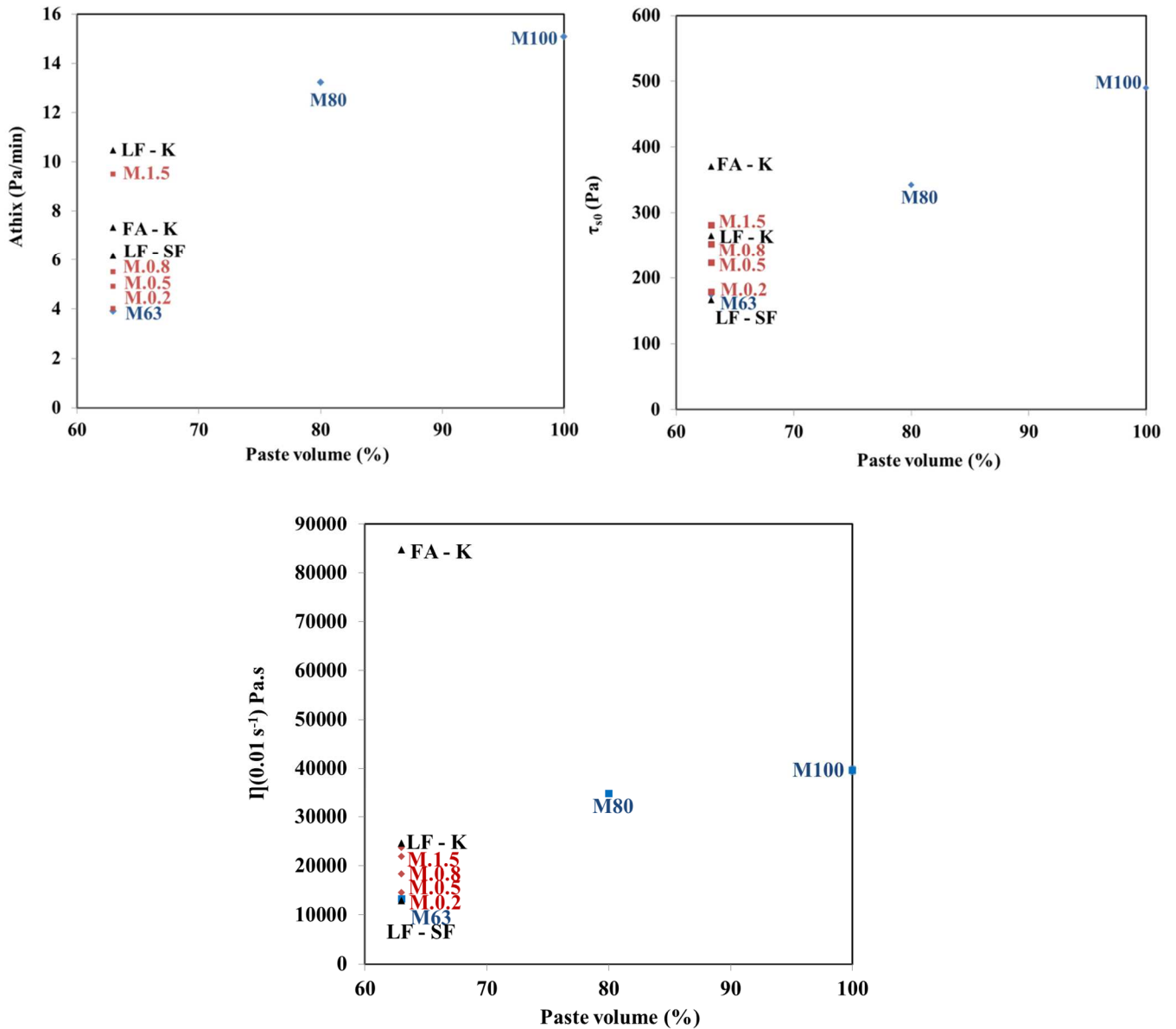
250 The workability parameters of the investigated mixtures are summarized in Fig. 9. As can be ob-
251 served, the reference mixture (i.e. M100) exhibited the highest viscosity, which is due to its low SP
252 dosage, hence resulting in higher colloidal interactions. This accelerated the structuration process,
253 reflected by higher A_{thix} value. In contrast, the M63 mixture containing higher amount of SP exhib-
254 ited lower viscosity and A_{thix} than M100 mixture, which is due to greater dispersing effect of SP.
255 This is likely to facilitate its pumping but can negatively affect its buildability aspect. On the other
256 hand, the incorporation of VMA significantly increased the viscosity and A_{thix} parameters. This can
257 be mainly due to the long chains and high molecular weight of VMA that absorb the free mixing-
258 water and increase the entanglement effect [17]. It is well established that VMAs modify signifi-
259 cantly the rheology of cement-based materials, which to a certain extent increases the macroscopic
260 yield stress [13-16]. There are very few studies in the literature that investigated the effect of VMA
261 on the printability of cementitious materials. Chen et al. [5] evaluated the competitive effects be-
262 tween the dispersive effect of SP and viscous aspect of VMA. This competition can deteriorate the
263 printability of the mixture depending on the resulting balance between the two opposite effects,
264 which is function of admixture dosages and their compatibility. Furthermore, the synergy between
265 the cement, SP, and VMA becomes very critical in this context. The type and chemical composition
266 of cement-SP-VMA system can greatly modify both the fresh and hardened properties of the mix-
267 ture.

268 Besides, the type and content of SCMs can significantly change the workability of the mixture. For
269 example, the mixtures containing limestone filler (LF) showed higher fluidity than those incorporat-
270 ing fly ash (FA). Thus, replacing fly ash with limestone filler tends to improve workability of the
271 mixture, reflected by lower viscosity and initial yield stress values. On the other hand, the use of
272 limestone filler induced higher structuration rate. This may be due to the physical effect of lime-
273 stone filler in modifying the particle-size distribution, packing density, and increasing the heteroge-
274 neous nucleation of the system [21]. The higher packing density combined with the low-water de-
275 mand of LF contribute in improving the fluidity of the mixture, whereas the modification of parti-
276 cle-size distribution and heterogeneous nucleation contribute in increasing A_{thix} .

277 On the other hand, the replacement of silica fume by kaolinite increased both the initial yield stress
278 and rate of structuration. Due to its high water-demand [22], kaolinite can absorb a significant part
279 of the mixing water, otherwise would normally contribute in lubricating the system and facilitate
280 particles sliding during shear, which decreases the flowability of the mixture. This explains the sig-

281 nificant increase in viscosity, which is more obvious in the case of FA – K combination than LF –
 282 K.

283 It is worthy to mention that the ranges of workability parameters obtained in this study are compa-
 284 rable with those reported in literature [20–22]. The workability characteristics should be adapted to
 285 the printing parameters. For example, stiffer materials can be used for larger scale and different
 286 printing processes, as reported in literature [8, 10].



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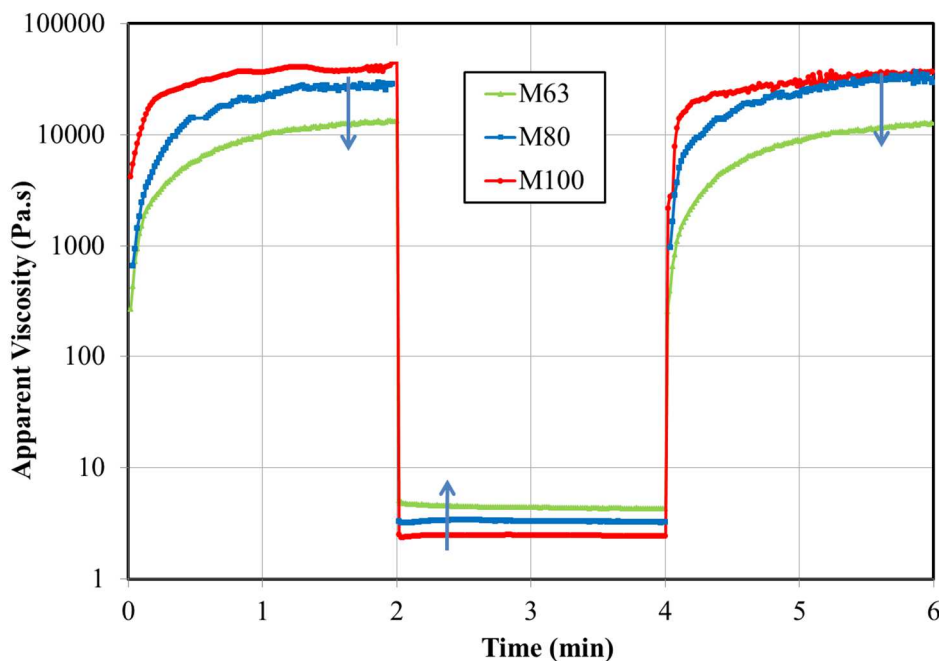
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Fig. 9. Workability parameters of the investigated mixtures

290 4.1.2 Extrudability parameters

291 Although the previously measured workability parameters provide an insight on the printability of
 292 the investigated mixtures, several changes of the material's properties can be expected during
 293 pumping and extrusion [26]. As mentioned earlier, a low viscosity is required during the pumping
 294 and extrusion processes, but a higher value is needed after extrusion. Eventhough these

295 requirements appear to be contradictory, they can be easily achieved by exploiting the thixotropy
296 and shear-thinning behaviors of the mixture, as shown in Fig. 10.



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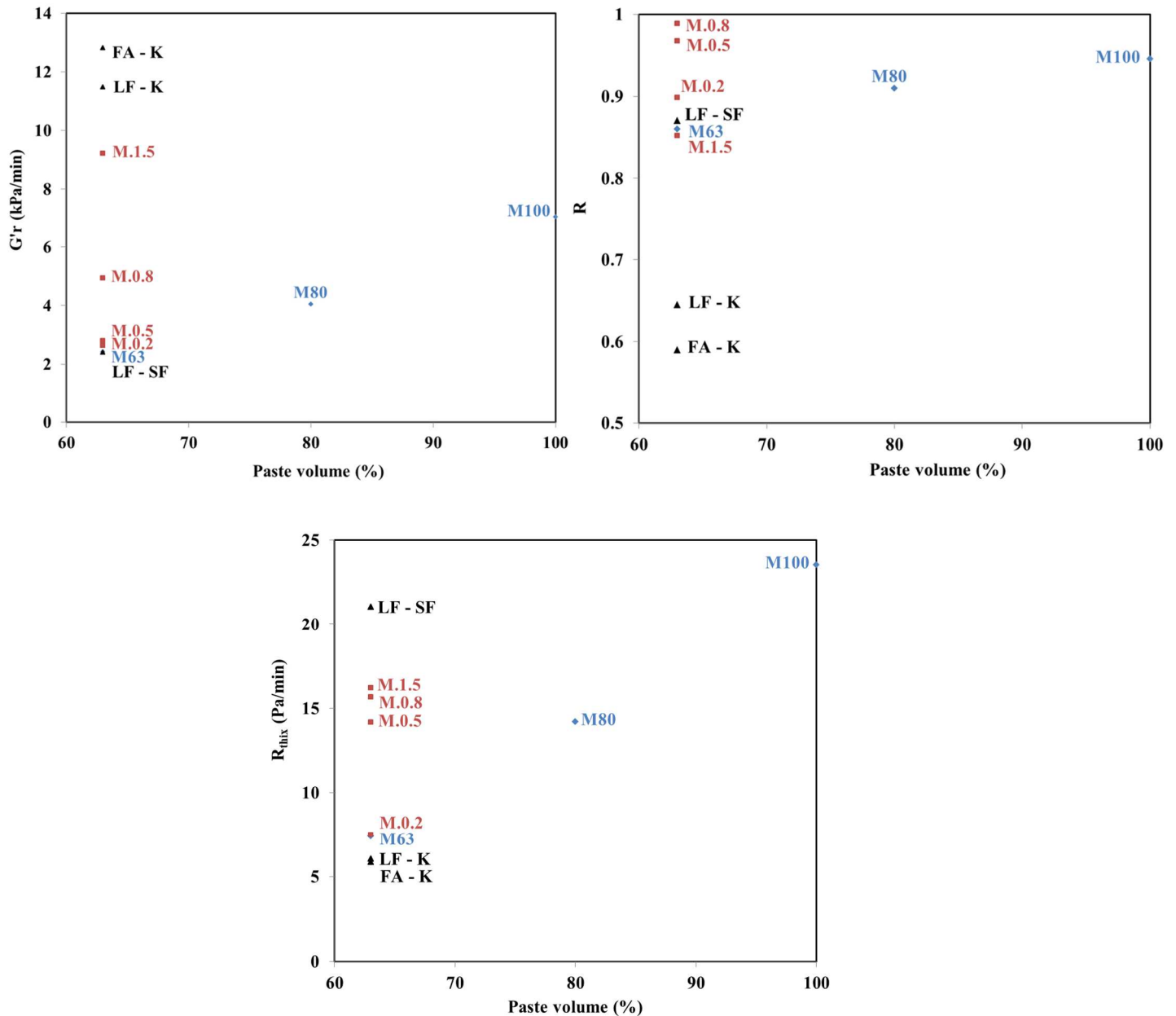
Fig. 10. Variation of apparent viscosity with time during the restructuring tests

299 The M100 mixture (i.e. no sand) perfectly fulfilled the extrusion requirements, hence justify its
300 common use in the 3D printing applications. This mixture exhibited the lowest apparent viscosity
301 during extrusion, but the highest value at rest.

302 The extrudability parameters of the investigated mixtures are summarized in Fig. 11. As expected,
303 the M100 mixture showed an increasing recovery rate of R_{thix} , R , and G'_r parameters due to the
304 rapid flocculation of cement particles. The inclusion of sand in cement-based materials inhibits ce-
305 ment particles flocculation, hence reducing the flocculation-induced structuration [27], as reflected
306 by the low R_{thix} value of M63 mixture. This can also be due to the relatively high dosage of SP of
307 the mixture. Indeed, the presence of SP chains induces repulsive interparticle forces and reduce the
308 overall attraction between cement particles and re-flocculation rate [28].

309 The time required for a printable mortar to build-up its structure increases with SP content due to
310 the reduction in attractive forces [29]. This can be overcome by using accelerators or VMA admix-
311 tures to increase the rigidification rate. Indeed, as shown in Fig. 11, the restructuring and re-
312 flocculation rates increased with VMA concentration. On the other hand, the viscosity recovery rate
313 R increased up to a certain limit. This suggests the existence of an optimum VMA dosage beyond
314 which the recovery of the initial apparent viscosity is limited, which may deteriorate the printability
315 of the mixture. The initial increase can be due to the thickening response of the long chains of
316 VMA, while the lowest recovery rate can be due to the repulsive dispersion induced by SP. This
317 behavior has been also reported in literature [17, 27].

318 Replacing fly ash with limestone fillers improved the re-flocculation rate. This is mainly due to the
 319 physico-chemical effect of limestone fillers, including better packing density and higher nucleation
 320 sites, as explained in the previous section. However, the replacement of silica fume by kaolinite
 321 promoted the granular stacking and, therefore, reduced the flocculation-induced restructuration rate
 322 of the mixture, but accelerated its rigidification by nucleation, reflected by higher G'_r values of LF-
 323 K and FA-K systems.



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325
 326

Fig. 11. Extrudability parameters of the investigated mixtures

327 At this stage, the determined parameters are relevant to assess the printability of the investigated
328 mixtures, especially in the case of short-printing-period trials. Taken into consideration the time-
329 dependent behavior of cement-based materials, monitoring their physico-chemical structuration
330 may be necessary to assess the long-term evolution, which may not be possible to assess using the
331 rheological parameters.

332 **4.2 Buildability parameters of the mixtures**

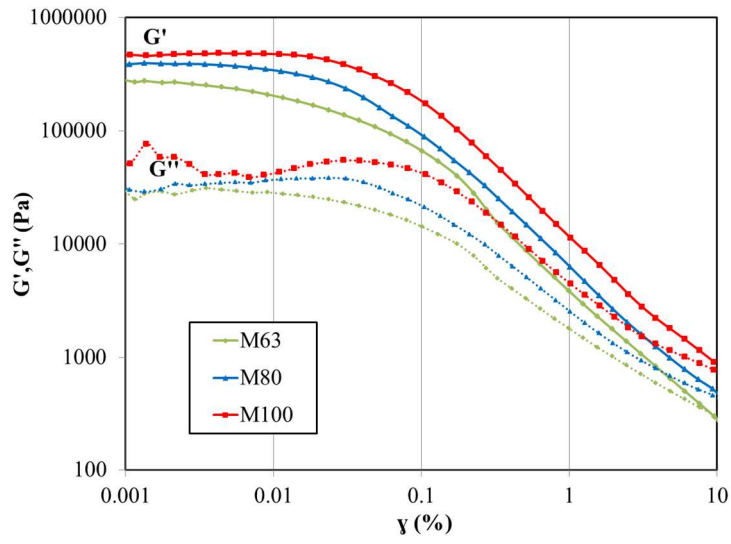
333 Results of the strain sweep measurements conducted on the investigated mixtures are shown in Fig.
334 12. On the other hand, the values of G' and critical strain are summarized in Table 4. As can be ob-
335 served, the M100 mixture exhibited the highest storage modulus and critical strain compared to
336 M80 and M63 mixtures. This can be due to cement particles interaction and higher packing density
337 of the system. The presence of sand particles and SP can improve the dispersion state of cement
338 particles, hence reducing attraction and increasing interparticle distances. The M100 mixture
339 showed also a higher increase of G'' before the drop compared to other mixtures. This may be relat-
340 ed to its high apparent viscosity value due to denser microstructure and higher interparticle friction
341 before flowing.

342 The addition of VMA up to 0.8%, by weight of cement, increased the storage modulus and critical
343 strain to a certain limit. The use of higher dosage reduced the critical strain, hence reflecting the
344 complex competition between SP and VMA due to their opposite effects.

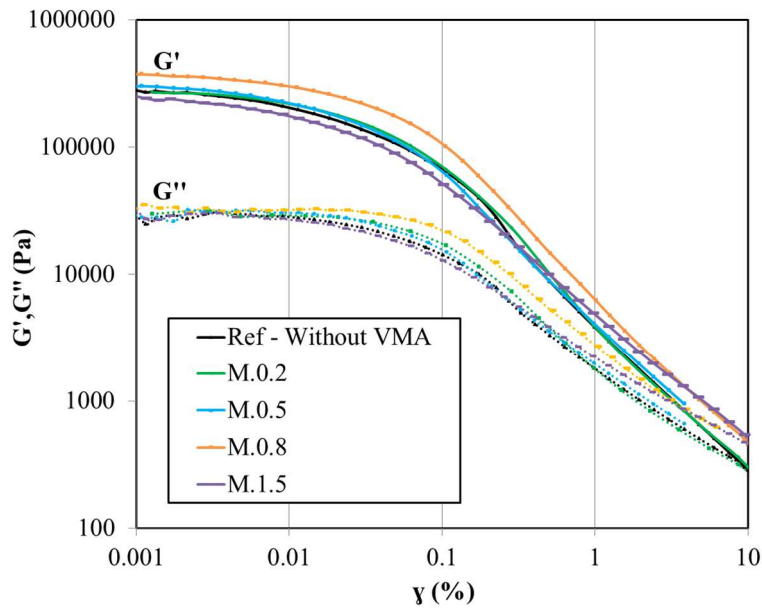
345 The mixtures containing kaolinite exhibited a wider viscoelastic domain than those containing silica
346 fume, hence reflecting better dispersion of the mixtures containing kaolinite. As mentioned earlier,
347 this is due to the contribution of its particle-size distribution in improving the skeleton of the mix-
348 ture. The LF-K and FA-K mixtures exhibited similar G'' evolution than M100, which is consistent
349 with high viscosity values. On the other hand, the incorporation of limestone filler resulted in lower
350 critical strain compared to that observed with FA. This is mainly due to the dilution effect of LF,
351 hence contributing in a smaller deformation needed to initiate the flow.

352 The open-time values of the investigated mixtures are shown in Table 5. In terms of the workability
353 and extrudability parameters, it is expected that M63 mixture would exhibit longer open time com-
354 pared to the other mixtures due to its relatively high SP dosage, which delayed the hydration reac-
355 tion [28]. On the other hand, the induction period decreased with the VMA concentration up to
356 0.8%, by weight of cement. Beyond this dosage, the trend showed a turning point.

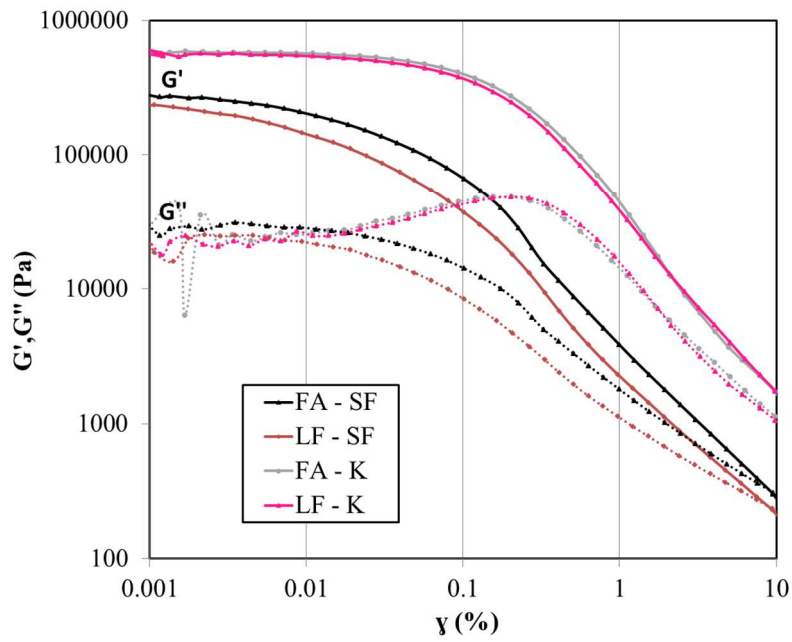
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Fig. 12. Stain sweep results of the investigated mixtures

361 Table 4. The storage modulus and critical strain values of the investigated mixtures

	G' (10⁵ Pa)	γ_c (10⁻¹)
M100	4.51	17.7
M80	3.48	8.89
M63 (FA-SF)	2.51	3.52
M.0.2	2.47	4.86
M.0.5	2.52	5.64
M.0.8	3.05	8.95
M.1.5	2.1	4.47
FA - K	5.52	22.1
LF - SF	1.96	2.25
LF - K	5.15	17.5

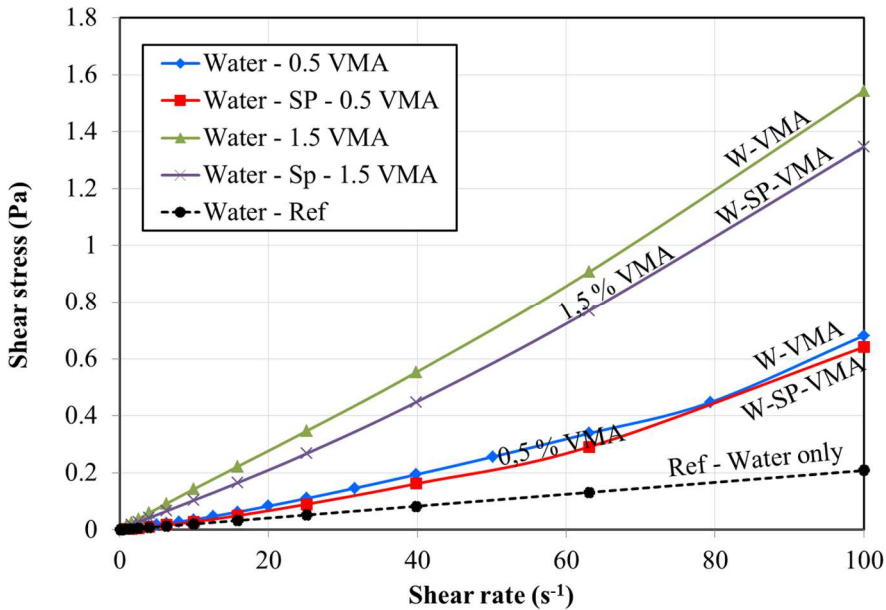
362

363 Table 5. Open time values of the investigated mixtures

Mix	Open time (h)	
	Measured value	Normalised Value
M100	3.26	1.00
M80	6.00	1.84
M63	8.00	1.33
Ref - M63	8.00	1.00
M-0.2	5.50	0.69
M-0.5	5.00	0.63
M-0.8	4.22	0.53
M-1.5	6.50	0.81
FA-SF	8.00	1.00
FA-K	3.54	0.32
LF-SF	5.36	0.67
LF-K	2.56	0.44

364 As stated earlier, the competition between SP and VMA may be the origin of open-time variation.
 365 In fact, at low VMA concentration, the available long chains can adsorb onto cement particles with-
 366 out hindering the hydration kinetics [31]. In the case of higher dosage, only very few cementitious
 367 particles are exposed to water, which slow down the hydration. The adsorption of SP molecules can
 368 also contribute in delaying the hydration. In this context, the flow curves of (water – SP – VMA)
 369 system were compared with that of (water - VMA) combination for different dosages of VMA to
 370 evaluate the effect of admixtures on the mixing water flow behavior. As can be observed in Fig. 13,
 371 the viscosity of water is greatly affected by incorporating SP and VMA. On the other hand, the flu-
 372 idification efficiency of SP is higher in the case of high VMA dosages mixtures. This reflects the
 373 SP-VMA interaction, which confirms what has been reported in literature [17], [32].

374 The incorporation of supplementary cementitious materials also affected the hydration kinetics. In-
 375 deed, the results presented in Table 5 revealed that replacing silica fume by kaolinite reduced by
 376 three times workability period (open time). This is mainly due to the relatively higher water demand
 377 of kaolinite compared to that of silica fume. This is consistent with the previous workability param-
 378 eters. The accelerating effect of limestone fillers was also observed when it is combined with either
 379 silica fume or kaolinite. This may be attributable to its tri-fold effects, including (1) better packing
 380 density, (2) increased surface area per unit cement available for nucleation and growth of hydration
 381 products, and (3) the reactivity of the limestone fillers itself [31–33].

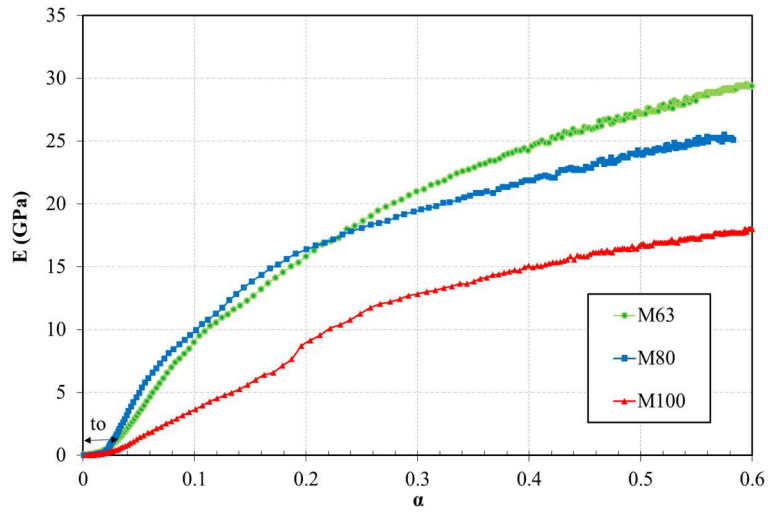


382
 383 Fig. 13. Flow curves of water mixtures, with Sp and VMA at low and high dosages

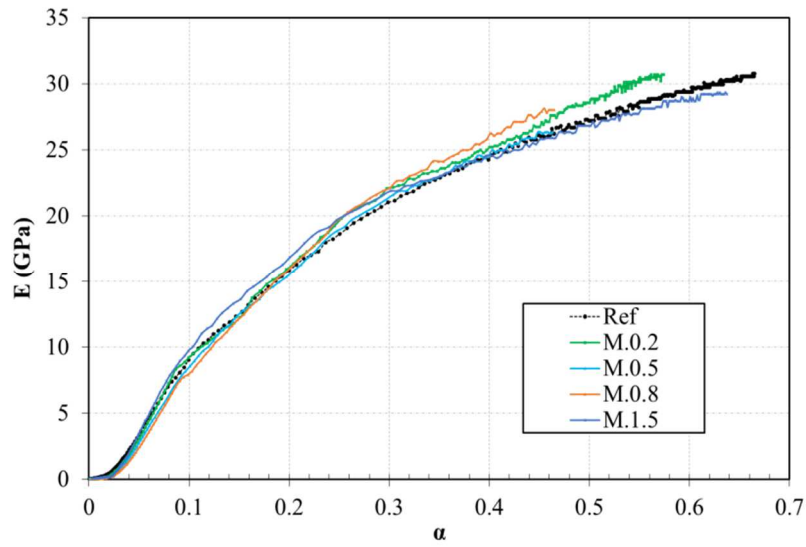
384 In general, depending on the targeted printing time and strength development, adding VMA or
 385 combining different SCMs can be effective in achieving suitable parameters for successful printing
 386 process. The evolution of the elastic modulus with degree of hydration of the investigated mixtures
 387 is shown in Fig.14. The obtained results showed that the shear modulus followed the same trend of
 388 the Young modulus.

389 It is worthy to mention that the addition of VMA did not have a significant effect on the elastic
 390 modulus of the investigated mixtures. Indeed, its effect was limited at early-age during the induc-
 391 tion period. On the other hand, combining different SCMs influenced the elastic behavior of the
 392 investigated mixtures at the hardened state. However, the strength development during the open-
 393 time was comparable, regardless of the type of SCMs. The mixture containing silica fume exhibited
 394 the highest elastic modulus at early age compared to other mixtures. This may be attributed to its
 395 higher reactivity compared to other SCM types. On the other hand, the accelerated hydration and
 396 improved early-age strength development observed with limestone fillers may be due to the stabili-
 397 zation of ettringite that increases the volume of hydration products, hence improving the compact-

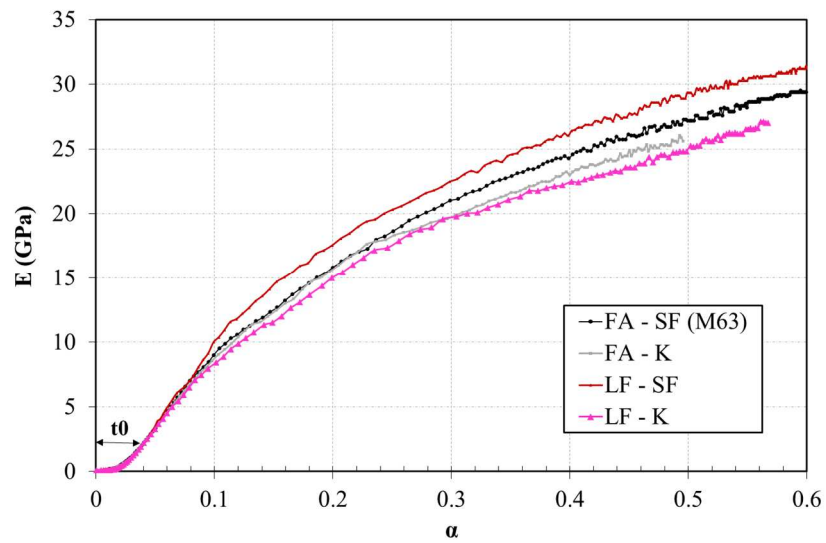
398 ness and strength of the matrix [36]. These results suggest that the LF-SF mixture exhibited the best
399 buildability aspect in terms of open time and strength, while M100 and M80 mixture showed the
400 lowest constructability parameters.



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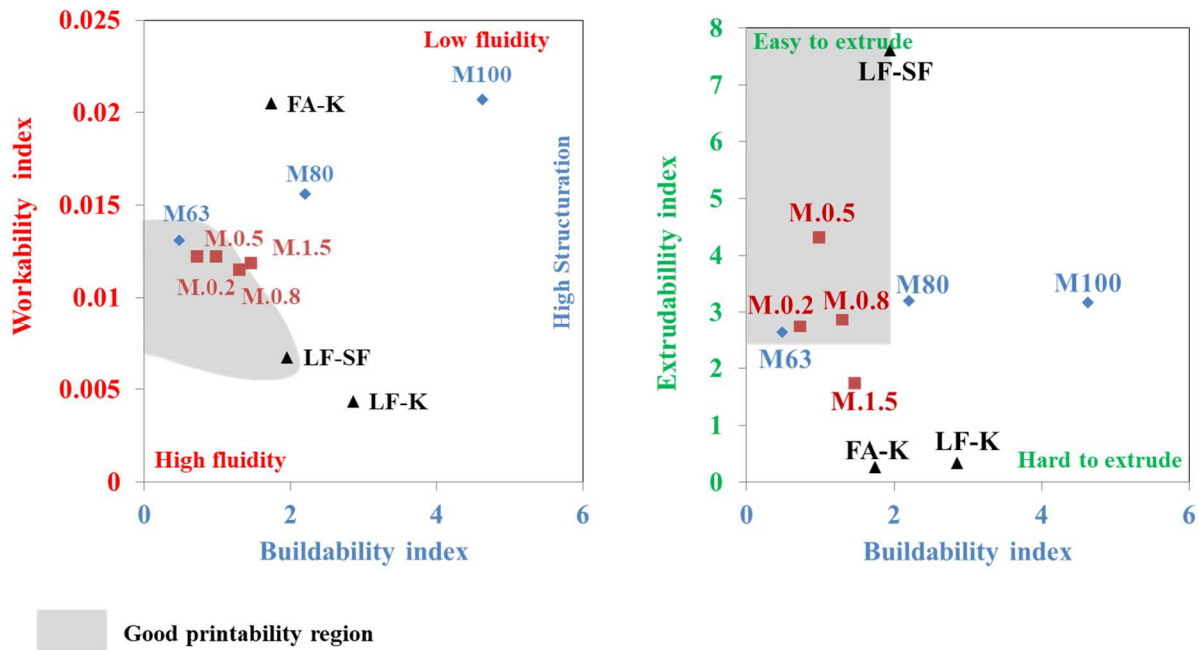
Fig. 14. Evolution of the elastic modulus with degree of hydration of the investigated mixtures

405 **4.3 Printability validation**

406 For comparison purposes, three indices reflecting workability, extrudability, and buildability aspects of the investigated mixtures, were defined, as follow:

408
$$W = \frac{\tau_{s0}}{\eta_{app}} , \quad E = \frac{R_{thix}}{G'_r} \cdot R , \quad B = \frac{A_{thix}}{t_0}$$

409 where W, E, and B represent the workability, extrudability, and buildability indices, respectively. It
 410 is worth mentioning that these parameters are based on the previously identified parameters. Since
 411 the printing-time-period is short, the evolution of viscoelastic properties at latter age has not been
 412 taken into consideration in the buildability index. The determined parameters W, E, and B of the
 413 investigated mixtures are summarized in Fig.14. These printability WEB indices can be useful in
 414 facilitating the selection of optimum design to achieve successful printing quality, and predicting in
 415 advance the challenges that may occur during the printing process, as shown in Fig.1 (e.g. blockage,
 416 failure).



418 Fig. 15. W.E.B indices of the investigated mixtures

419 Various printing trials were conducted to validate the printability aspects of the investigated mix-
 420 tures, as shown in Table 6. Based on the obtained results, it can be observed that mixtures with the
 421 lowest workability and extrudability and the highest buildability indices resulted in the lowest quali-
 422 ty of filament surfaces. This was evaluated in terms of air voids, cracks, and deformations appearing
 423 during the printing process (M100, LF-K, FA-K, and M.1.5 mixtures). The M63 mixture showed
 424 very good extrusion quality and good interlayer adhesion, but moderate rate of constructability.
 425 However, the addition of VMA dosage less than 0.8%, by weight of cement, improved the rate of
 426 constructability. The use of higher dosage affected the quality of the extrusion, as observed with
 427 M.1.5 mixture. The extrudability index obtained with M.1.5 (E=1.7) is lower than that of M63 made

428 without any VMA ($E = 2.6$). On the other hand, the LF-SF mixture exhibited very good buildability
429 rate even with quite low viscosity and initial yield stress values. This can be due to its high rate of
430 re-flocculation and structuration, thus increasing its extrusion ease and shape retention. A targeted
431 W , E , and B corresponding to $0.006 < W < 0.015$, $E > 2.5$, and $B < 2$ (Fig. 14) can ensure good
432 printability. Since a single batch approach was used for the printing, increasing the number of print
433 layers resulted in decreasing the extrudability of mortar mixtures, especially those having
434 high A_{thix} and G'_r , corresponding to mixtures with high values of E and B . This confirms the fact
435 that the initial rheological properties are not sufficient to assess the printability of cement-based
436 materials, hence the extrudability parameters are found to be more relevant in this regard. On the
437 other hand, as expected the pumping pressure was found to increase with the apparent viscosity of
438 the printed mixtures.

439 **4.4 Recommendations**

440 Based on the results discussed above, the following useful recommendations can be established to
441 design printable cement-based materials for 3D printing:

- 442 - The trade-off between paste volume and SP content aims at facilitating the extrusion while tak-
443 ing into account the rate of constructability.
- 444 - The use of adequate VMA dosage (up to 0.8%, by weight of cement, in the case of this study)
445 showed great potential in increasing the buildability cement mortars in 3D-printing, without af-
446 fecting their strength development in the hardened state. However, higher dosage can deterio-
447 rate the extrusion quality. On the other hand, good compatibility between cement, SP, and
448 VMA is essential to achieve adapted WEB parameters for successful printing.
- 449 - The effects of supplementary cementitious materials (SCMs), used as partial replacement of
450 cement, on the open time and strength development depend on their type and content. For ex-
451 ample, the combination of silica fume and limestone filler was shown to be favorable for 3D
452 printing applications. On the other hand, the use of kaolinite has deteriorated both the workabil-
453 ity and extrudability aspects of the investigated mixtures but improved the buildability rate.
454 This can be due to the relatively high content (10%) used in this study.
- 455 - In addition to improve the printability aspects, the use of the SCMs reduces the carbon footprint
456 of cement-based materials.

457 **Conclusions**

458 A new approach to design printable cement-based materials has been described in this paper. The
459 printability was evaluated in terms of workability, extrudability, and buildability (WEB) aspects.
460 The investigated mixtures were proportioned to cover wide ranges of rheological properties, by var-

461 ying paste volume, superplasticizer content, incorporating VMA or using different SCMs. Based on
462 the results obtained in the present study, the following conclusions can be drawn:

- 463 - The printability of the investigated mixtures is successfully correlated to their rheological param-
464 eters, including apparent viscosity (η), and static yield stress (τ_{s0}) as well as its evolution rate
465 (R_{thix}, A_{thix}). This reflects the importance of structure formed during the flocculation process
466 for early-age 3D printing process.
- 467 - The initial rheological properties are not sufficient to adequately evaluate the printability of ce-
468 ment-based materials. Their combination with the restructuration parameters (R, G'_r) is im-
469 portant to follow up the changes that can occur during the printing process. The printability of
470 the mixtures may require adjustments given the printing speed, flow rate, and the height of print
471 layers.
- 472 - The selection of the mixture design approach depends on the printing time and the targeted
473 strength. The use of VMA is shown to be efficient on improving the rate of constructability,
474 while supplementary cementitious materials are effective in modifying strength development,
475 open time, and microstructure. The combination of both approaches can help to improve the
476 printability of cement-based materials.

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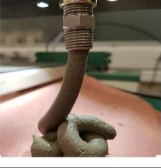












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Table 6. Results of the printing validation of the investigated mixtures

Printing trials	Mix	Average extrusion pressure (bar)	Texture of the filament	Average number of printed layers	Printability comments
	M100	0.7		20	The filament is homogeneous. The buildability rate is high. The cohesion between the layers is not good enough considering the high rates of reflocculation and structuration comparing to the printing speed. The printability time is very limited, and air voids have been observed during the printing.
	M80	0.5		18	The extrusion quality is quite good. The bonding between the layers is fine. The constructability rate is close to that of the paste, and fewer air voids were seen on the filaments.
	M63	0.2		15	The mixture is very easy to extrude. Its fluidity allowed a better adhesion between the layers at a moderate rate of buildability. No air voids were noticed.
	M0.2	0.3		18	
	M0.5	0.5		23	By increasing the VMA dosage, the constructability rate increased due to the stabilising effect of the colloidal agent. On the other hand, the extrusion quality was deteriorated with the increase in VMA concentration. Microcracks and air voids reappeared on the surface of the filaments, particularly pronounced for M1.5. Sagging of some layers was also observed in M1.5 which reflects the heterogeneous nature of the mixture.
	M0.8	0.8		26	
	M1.5	1		25	
	LF - SF	0.5		27	The mixture presents a good texture. Only slight deformation were observed. The printing time is large.
	LF - K	1.3		23	The geometry of the printed cylinder is very deficient. The texture of the filaments is heterogeneous but the shape stability is conserved.
	FA - K	1.5		23	The shape of the filament is perfect. The geometry is conserved but some layers show slight deformation. The printability time is restricted and the rate of buildability is high.

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494 Nantes.

495 **References**

- 496 [1] H. October 2020, « Digitalisation how covid-19 is driving change in construction », *Building*.
497 [https://www.building.co.uk/sponsored-content/digitalisation-how-covid-19-is-driving-change-in-](https://www.building.co.uk/sponsored-content/digitalisation-how-covid-19-is-driving-change-in-construction/5108588.article)
498 [construction/5108588.article](https://www.building.co.uk/sponsored-content/digitalisation-how-covid-19-is-driving-change-in-construction/5108588.article) (consulté le 5 décembre 2020).
- 499 [2] J. G. Sanjayan et B. Nematollahi, « 3D Concrete Printing for Construction Applications », in *3D*
500 *Concrete Printing Technology*, Elsevier, 2019, p. 1-11. doi: 10.1016/B978-0-12-815481-
501 6.00001-4.
- 502 [3] F. Bos, R. Wolfs, Z. Ahmed, et T. Salet, « Additive manufacturing of concrete in construction:
503 potentials and challenges of 3D concrete printing », *Virtual Phys. Prototyp.*, vol. 11, n° 3, p.
504 209-225, juill. 2016, doi: 10.1080/17452759.2016.1209867.
- 505 [4] P. Shakor, J. Renneberg, S. Nejadi, et G. Paul, « Optimisation of Different Concrete Mix Designs
506 for 3D Printing by Utilising 6DOF Industrial Robot », juin 2017. doi:
507 10.22260/ISARC2017/0036.
- 508 [5] Y. Chen *et al.*, « Improving printability of limestone-calcined clay-based cementitious materials
509 by using viscosity-modifying admixture », *Cem. Concr. Res.*, vol. 132, p. 106040, juin 2020, doi:
510 10.1016/j.cemconres.2020.106040.
- 511 [6] T. S. Rushing *et al.*, « Investigation of concrete mixtures for additive construction », *Rapid Pro-*
512 *totyp. J.*, vol. 23, n° 1, p. 74-80, janv. 2017, doi: 10.1108/RPJ-09-2015-0124.
- 513 [7] N. Roussel, « Rheological requirements for printable concretes », *Cem. Concr. Res.*, vol. 112, p.
514 76-85, oct. 2018, doi: 10.1016/j.cemconres.2018.04.005.
- 515 [8] L. Reiter, T. Wangler, N. Roussel, et R. J. Flatt, « The role of early age structural build-up in dig-
516 ital fabrication with concrete », *Cem. Concr. Res.*, vol. 112, p. 86-95, oct. 2018, doi:
517 10.1016/j.cemconres.2018.05.011.
- 518 [9] A. Kazemian, X. Yuan, R. Meier, et B. Khoshnevis, « Chapter 2 - Performance-Based Testing of
519 Portland Cement Concrete for Construction-Scale 3D Printing », in *3D Concrete Printing Tech-*
520 *nology*, J. G. Sanjayan, A. Nazari, et B. Nematollahi, Éd. Butterworth-Heinemann, 2019, p.
521 13-35. doi: 10.1016/B978-0-12-815481-6.00002-6.
- 522 [10] J. Kruger, S. Zeranka, et G. van Zijl, « An ab initio approach for thixotropy characterisation of
523 (nanoparticle-infused) 3D printable concrete », *Constr. Build. Mater.*, vol. 224, p. 372-386, nov.
524 2019, doi: 10.1016/j.conbuildmat.2019.07.078.
- 525 [11] F. Sanchez, « Tools of the Future - Printing Cement-Based Materials in 3D ». Consulté le: 17
526 janvier 2022. [En ligne]. Disponible sur:
527 <https://onlinepubs.trb.org/onlinepubs/webinars/201117.pdf>
- 528 [12] F. Lyu, D. Zhao, X. Hou, L. Sun, et Q. Zhang, « Overview of the Development of 3D-Printing
529 Concrete: A Review », *Appl. Sci.*, vol. 11, n° 21, p. 9822, janv. 2021, doi: 10.3390/app11219822.
- 530 [13] « Agent de viscosité pour béton | Master Builders Solutions », *France*. [https://www.master-](https://www.master-builders-solutions.com/fr-fr/products/mastermatrix/mastermatrix-sdc-100)
531 [builders-solutions.com/fr-fr/products/mastermatrix/mastermatrix-sdc-100](https://www.master-builders-solutions.com/fr-fr/products/mastermatrix/mastermatrix-sdc-100) (consulté le 21 dé-
532 cembre 2020).
- 533 [14] A. Perrot, D. Rangeard, et A. Pierre, « Structural built-up of cement-based materials used for
534 3D-printing extrusion techniques », *Mater. Struct.*, vol. 49, n° 4, p. 1213-1220, avr. 2016, doi:
535 10.1617/s11527-015-0571-0.

- 536 [15] N. Roussel, G. Ovarlez, S. Garrault, et C. Brumaud, « The origins of thixotropy of fresh cement
537 pastes », *Cem. Concr. Res.*, vol. 42, n° 1, p. 148-157, janv. 2012, doi:
538 10.1016/j.cemconres.2011.09.004.
- 539 [16] R. Cortas, E. Rozière, S. Staquet, A. Hamami, A. Loukili, et M.-P. Delplancke-Ogletree, « Ef-
540 fect of the water saturation of aggregates on the shrinkage induced cracking risk of concrete at
541 early age », *Cem. Concr. Compos.*, vol. 50, p. 1-9, juill. 2014, doi:
542 10.1016/j.cemconcomp.2014.02.006.
- 543 [17] « Effect of Welan Gum-High-Range Water Reducer Combinations on Rheology of Cement
544 Grout », *ACI Mater. J.*, vol. 94, n° 5, 1997, doi: 10.14359/321.
- 545 [18] H. Bessaies-Bey, R. Baumann, M. Schmitz, M. Radler, et N. Roussel, « Effect of polyacryla-
546 mide on rheology of fresh cement pastes », *Cem. Concr. Res.*, vol. 76, p. 98-106, oct. 2015, doi:
547 10.1016/j.cemconres.2015.05.012.
- 548 [19] M. Sonebi et A. Perrot, « Effect of mix proportions on rheology and permeability of cement
549 grouts containing viscosity modifying admixture », *Constr. Build. Mater.*, vol. 212, p. 687-697,
550 juill. 2019, doi: 10.1016/j.conbuildmat.2019.04.022.
- 551 [20] M. Benaïcha, X. Roguiez, O. Jalbaud, Y. Burtschell, et A. H. Alaoui, « Influence of silica fume
552 and viscosity modifying agent on the mechanical and rheological behavior of self compacting
553 concrete », *Constr. Build. Mater.*, vol. 84, p. 103-110, juin 2015, doi:
554 10.1016/j.conbuildmat.2015.03.061.
- 555 [21] D. K. Panesar et R. Zhang, « Performance comparison of cement replacing materials in concrete:
556 Limestone fillers and supplementary cementing materials – A review », *Constr. Build. Mater.*,
557 vol. 251, p. 118866, août 2020, doi: 10.1016/j.conbuildmat.2020.118866.
- 558 [22] Y. Fan, S. Zhang, S. Kawashima, et S. P. Shah, « Influence of kaolinite clay on the chloride dif-
559 fusion property of cement-based materials », *Cem. Concr. Compos.*, vol. 45, p. 117-124, janv.
560 2014, doi: 10.1016/j.cemconcomp.2013.09.021.
- 561 [23] Y. Zhang, Y. Zhang, W. She, L. Yang, G. Liu, et Y. Yang, « Rheological and harden properties
562 of the high-thixotropy 3D printing concrete », *Constr. Build. Mater.*, vol. 201, p. 278-285, mars
563 2019, doi: 10.1016/j.conbuildmat.2018.12.061.
- 564 [24] D. P. Bentz, S. Z. Jones, I. R. Bentz, et M. A. Peltz, « Towards the formulation of robust and
565 sustainable cementitious binders for 3-D additive construction by extrusion », *Constr. Build. Ma-
566 ter.*, vol. 175, p. 215-224, juin 2018, doi: 10.1016/j.conbuildmat.2018.04.167.
- 567 [25] O. A. Mendoza Reales, P. Duda, E. C. C. M. Silva, M. D. M. Paiva, et R. D. T. Filho, « Nanosil-
568 ica particles as structural buildup agents for 3D printing with Portland cement pastes », *Constr.
569 Build. Mater.*, vol. 219, p. 91-100, sept. 2019, doi: 10.1016/j.conbuildmat.2019.05.174.
- 570 [26] H. Jeong, S.-J. Han, S.-H. Choi, Y. Lee, S. Yi, et K. Kim, « Rheological Property Criteria for
571 Buildable 3D Printing Concrete », *Materials*, vol. 12, n° 4, p. 657, févr. 2019, doi:
572 10.3390/ma12040657.
- 573 [27] J. T. Kolawole, R. Combrinck, et W. P. Boshoff, « Rheo-viscoelastic behaviour of fresh cement-
574 based materials: Cement paste, mortar and concrete », *Constr. Build. Mater.*, vol. 248, p. 118667,
575 juill. 2020, doi: 10.1016/j.conbuildmat.2020.118667.
- 576 [28] M. T. Souza, I. M. Ferreira, E. Guzi de Moraes, L. Senff, et A. P. Novaes de Oliveira, « 3D
577 printed concrete for large-scale buildings: An overview of rheology, printing parameters, chemi-
578 cal admixtures, reinforcements, and economic and environmental prospects », *J. Build. Eng.*, vol.
579 32, p. 101833, nov. 2020, doi: 10.1016/j.jobbe.2020.101833.
- 580 [29] D. Marchon, S. Kawashima, H. Bessaies-Bey, S. Mantellato, et S. Ng, « Hydration and rheology
581 control of concrete for digital fabrication: Potential admixtures and cement chemistry », *Cem.
582 Concr. Res.*, vol. 112, p. 96-110, oct. 2018, doi: 10.1016/j.cemconres.2018.05.014.
- 583 [30] R. Bouras, A. Kaci, et M. Chaouche, « Influence of viscosity modifying admixtures on the rheo-
584 logical behavior of cement and mortar pastes », *Korea-Aust. Rheol. J.*, vol. 24, n° 1, p. 35-44,
585 mars 2012, doi: 10.1007/s13367-012-0004-3.
- 586 [31] Ó. H. Wallevik et I. Nielsson, *PRO 33: 3rd International RILEM Symposium on Self-
587 Compacting Concrete*. RILEM Publications, 2003.

- 588 [32] T. H. Phan, M. Chaouche, et M. Moranville, « Influence of organic admixtures on the rheologi-
589 cal behaviour of cement pastes », *Cem. Concr. Res.*, vol. 36, n° 10, p. 1807-1813, oct. 2006, doi:
590 10.1016/j.cemconres.2006.05.028.
- 591 [33] L. R. Gurney, D. P. Bentz, T. Sato, et W. J. Weiss, « Reducing Set Retardation in High-Volume
592 Fly Ash Mixtures with the Use of Limestone: Improving Constructability for Sustainability »,
593 *Transp. Res. Rec. J. Transp. Res. Board*, vol. 2290, n° 1, p. 139-146, janv. 2012, doi:
594 10.3141/2290-18.
- 595 [34] M. Aqel et D. K. Panesar, « Hydration kinetics and compressive strength of steam-cured cement
596 pastes and mortars containing limestone filler », *Constr. Build. Mater.*, vol. 113, p. 359-368, juin
597 2016, doi: 10.1016/j.conbuildmat.2016.03.031.
- 598 [35] T. Oey, A. Kumar, J. W. Bullard, N. Neithalath, et G. Sant, « The Filler Effect: The Influence of
599 Filler Content and Surface Area on Cementitious Reaction Rates », *J. Am. Ceram. Soc.*, vol. 96,
600 n° 6, p. 1978-1990, 2013, doi: <https://doi.org/10.1111/jace.12264>.
- 601 [36] A. Schöler, B. Lothenbach, F. Winnefeld, M. B. Haha, M. Zajac, et H.-M. Ludwig, « Early hy-
602 dration of SCM-blended Portland cements: A pore solution and isothermal calorimetry study »,
603 *Cem. Concr. Res.*, vol. 93, p. 71-82, mars 2017, doi: 10.1016/j.cemconres.2016.11.013.
604