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HAL Authorization

1 **Temperature-dependent performance prediction for cerium**
2 **oxynitride solid-state symmetric supercapacitor using Machine**
3 **Learning**

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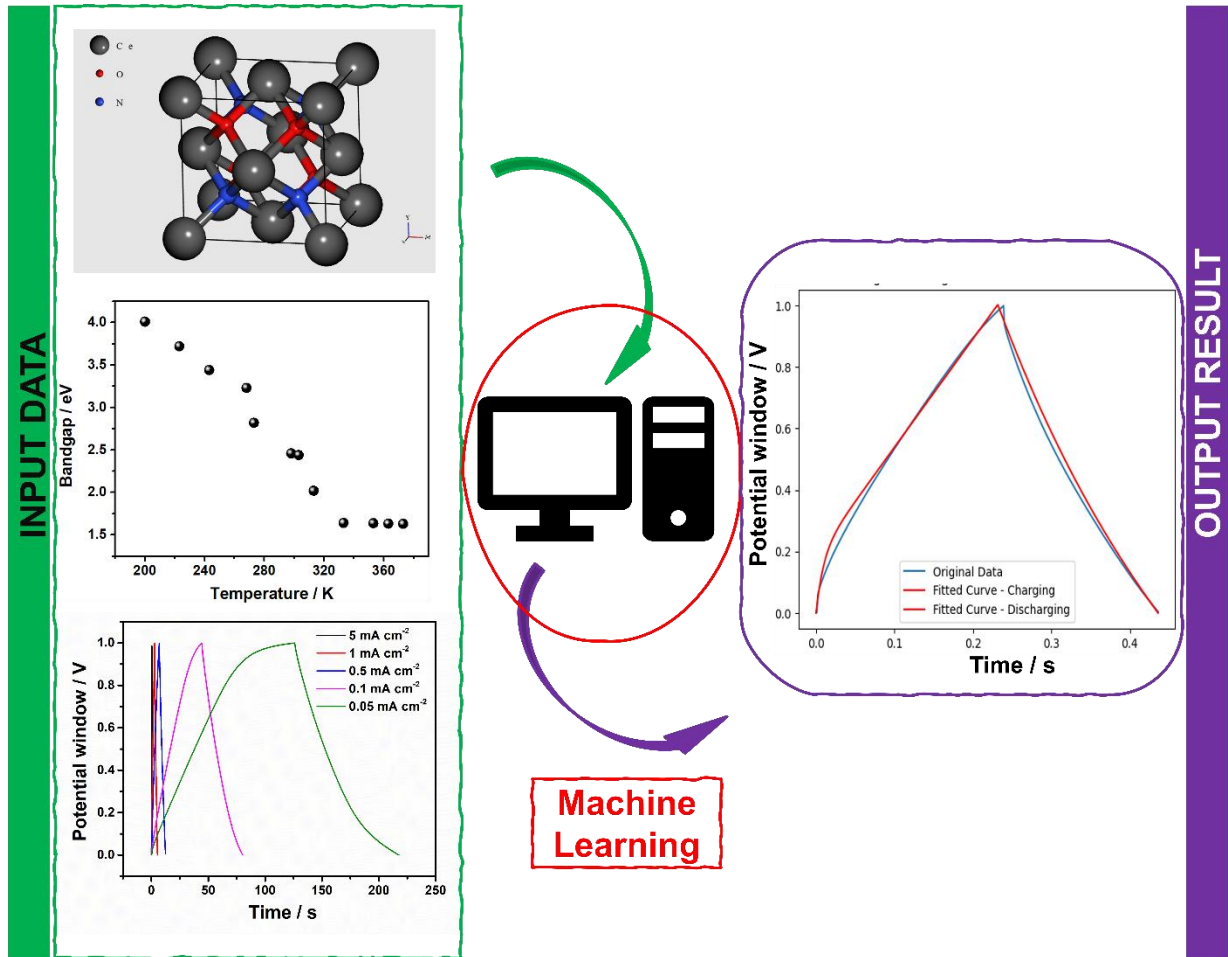
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27 Graphical abstract



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39 **Abstract**

40 The global demand for electricity and dwindling fossil fuel reserves necessitates advanced
41 energy storage solutions like next-generation supercapacitors. This study uses machine learning
42 (ML) to predict the temperature-dependent performance of cerium oxynitride-based solid-state
43 symmetric supercapacitors. Supercapacitors excel in power density and rapid charge-discharge
44 but face challenges under varying temperatures affecting ionic mobility and material stability.
45 Cerium oxynitride, with unique electronic and thermal properties, is analyzed using Density
46 Functional Theory (DFT) to study temperature-induced changes in the Density of States. ML
47 models trained on experimental and DFT data predict charge-discharge behaviour under thermal
48 and current variations. The supercapacitor exhibits good charge storage ($\sim 4.6 \text{ mVcm}^{-2}$) and low
49 ohmic resistance ($\sim 1.6 \Omega$) at room temperature. The study also models exponential and linear
50 relationships between time and potential window, highlighting the potential of integrating DFT,
51 experimental insights, and ML to tackle temperature variability in supercapacitors.

52 **Keywords:** temperature study; machine learning; supercapacitors; oxynitride; energy storage;
53 density functional theory

54 **1. Introduction**

55 The global push for a sustainable future has made energy storage essential in addressing climate
56 change, fossil fuel depletion, and pollution. While renewable energy sources like solar and wind
57 offer cleaner alternatives, their intermittency challenges grid stability, highlighting the need for
58 advanced storage solutions to ensure reliable energy capture, storage, and delivery [1,2]. In this
59 context, energy storage systems, including supercapacitors, batteries, and other emerging
60 technologies, play a pivotal role in enhancing the efficiency and feasibility of renewable energy
61 integration, facilitating a smoother transition to sustainable energy systems[3].

62 From an environmental perspective, developing efficient energy storage systems is vital for
63 minimizing the carbon footprint associated with energy production and consumption[4]. By
64 enabling more effective utilization of renewable resources, energy storage helps reduce reliance
65 on fossil fuels, lowering carbon dioxide emissions and other harmful pollutants. This transition
66 supports global efforts to meet international climate targets, such as those outlined in the Paris
67 Agreement, which aims to limit global warming to below 2 °C compared to pre-industrial
68 levels[5]. Energy storage enables decentralized energy systems, empowering local generation
69 and storage, enhancing energy independence, and reducing the environmental impact of long-
70 distance transmission.

71 Supercapacitors, in particular, have become part of the broader energy storage landscape due to
72 their unique characteristics and ability to bridge the gap between traditional capacitors and
73 batteries[6–8]. Unlike conventional batteries that rely on slow chemical reactions,
74 supercapacitors store energy through electrostatic separation of charges, allowing them to deliver
75 high power density, rapid charge-discharge rates, and superior cycle life with millions of
76 cycles[9,10]. Supercapacitors are ideal for applications requiring quick power bursts, like
77 regenerative braking, backup power, portable electronics, and renewable energy load leveling.
78 However, their lower energy density limits prolonged energy supply. They are classified as
79 electrochemical double-layer capacitors (EDLCs), storing energy via ion adsorption, and
80 pseudocapacitors, utilizing fast redox reactions.[11,12]. A supercapacitor's key components are
81 electrodes for charge storage, electrolytes for ion exchange, separators to prevent short circuits
82 while enabling ion flow, and current collectors connecting electrodes to the circuit. Performance
83 depends largely on electrode materials, which impact capacitance, energy density, power density,
84 and thermal stability. Electrode materials range from carbon-based materials such as activated

85 carbon, graphene, and carbon nanotubes, known for their high surface area and excellent
86 conductivity, to transition metal oxides (e.g., RuO₂, MnO₂, NiO, CoFe₂O₄), conducting polymers
87 (e.g., polyaniline, polypyrrole), nitrides (e.g., TiN, VN), etc[13–17]. Novel materials like
88 oxynitrides, MOF-derived materials, bimetallic chalcogenides, etc, have been explored recently
89 in a similar context. These materials are studied to improve charge storage and optimize energy
90 and power density. Density Functional Theory (DFT) provides a theoretical framework to
91 analyze atomic-level interactions, including the Density of States (DOS), which influences
92 electrochemical performance. Elevated temperatures can shift the DOS, altering conductivity, ion
93 adsorption, and reaction kinetics, impacting charge storage capacity and cycling stability.

94 For instance, Yu-Ning Wu *et al* investigated the temperature effects on the band gap and optical
95 properties of rutile TiO₂ using ab initio methods[18]. They found that electron-phonon
96 interactions are the main factor affecting band gap changes with temperature, leading to a band
97 gap narrowing by about 138 meV at 1000 K. The findings also align with experimental data on
98 refractive index and dielectric constants, offering insights for developing high-temperature
99 materials for optical gas sensing applications. On the other hand, the temperature-dependent DFT
100 studies have been carried out by Ashish et al.) on VS₂ electrodes, and they found that doping VS₂
101 electrodes with non-metallic elements like nitrogen, phosphorus, and arsenic enhances their
102 quantum capacitance[19]. Among them, As-doped VS₂ exhibited the highest quantum
103 capacitance and maintained a stable capacitance at higher temperatures. Additionally, the band
104 structure of the doped VS₂ monolayers showed metallic properties, making them promising for
105 supercapacitor electrodes, with As-doped VS₂ achieving a high figure of merit due to its
106 excellent electrical conductivity. Bharti et al studied the density of states and quantum
107 capacitance of pure and cobalt-doped niobium nitrides using DFT[20]. NbN shows high quantum

108 capacitance, making it a strong candidate for supercapacitor electrodes. Cobalt doping further
109 enhances this property, and both pristine and doped niobium nitrides outperform graphene
110 regarding quantum capacitance near the Fermi level.

111 Temperature significantly affects supercapacitors, impacting ionic mobility, conductivity,
112 interface stability, and efficiency. Low temperatures hinder ion transport, reducing capacitance,
113 while high temperatures improve conductivity but risk material degradation, shorter cycle life,
114 and thermal runaway.[21–24]. Studying temperature-dependent supercapacitor performance is
115 vital for applications in extreme thermal environments, like automotive or aerospace. It aids in
116 designing resilient materials and systems for reliable, safe, and optimal performance under
117 varying conditions.

118 Machine learning (ML) is used here for temperature-dependent performance evaluation, offering
119 a powerful tool for analyzing data, identifying patterns, and making predictions. ML aids in
120 material property prediction, synthesis optimization, performance forecasting, and discovering
121 advanced materials for supercapacitors [25–29]. ML algorithms can be trained using
122 experimental measurements, DFT simulations, and electrochemical testing data to predict key
123 performance metrics such as capacitance, energy density, cycle stability, and thermal behaviour
124 [30–34]. Different data-driven algorithms used for different electrochemical energy storage
125 systems are tabulated in Table S1 for further study.

126 This work uses ML to predict supercapacitor performance at high temperatures, offering insights
127 into thermal resilience, optimizing materials, and identifying degradation pathways for better
128 performance in harsh environments. ML has been used in the literature to predict temperature-

129 related predictions for different photovoltaics applications and for air temperature inside
130 buildings[37,38].

131 This paper uses machine learning to predict the temperature-dependent performance of cerium
132 oxynitride-based solid-state symmetric supercapacitors, chosen for its high conductivity, thermal
133 stability, and unique electronic structure. It explores how ML models, trained on experimental
134 and DFT data, can predict the charge-discharge behavior of cerium oxynitride electrodes across
135 various temperatures. Integrating ML with DFT insights allows for a comprehensive
136 understanding of how electronic properties such as DOS evolve with temperature, directly
137 impacting the supercapacitor's performance[39–41]. By modelling these temperature-dependent
138 changes, this work provides a predictive framework that can be extended to other advanced
139 materials, driving the design of next-generation supercapacitors with improved thermal
140 performance.

141 **2. Experimental and Computational Methodology**

142 **2.1. Materials**

143 The supplementary material file provides the detailed materials synthesis process and
144 corresponding XRD and Scanning electron microscopy images. Further details are provided in
145 the literature [42,43]. The detailed device fabrication process is provided in the section below.

146 **2.2. Electrode and device preparation**

147 This active material is CeO_xN_y , and it is used to prepare the slurry along with acetylene black
148 and PVDF in a ratio of 85:10:5 in NMP solvent. Ni foam is the current collecting substrate on
149 which the active material is coated. The coated Ni foam is used to create a circular electrode of

150 18mm diameter with a 2 mg cm^{-2} mass loading of the mixed active material. The Ni foam is kept
151 in the heating oven at $80 \text{ }^{\circ}\text{C}$ (8 h duration) for drying.

152 Processing of the Ni foam before using it as the current collecting substrate: Washing of the Ni
153 foam is done in three continuous steps – (i) step 1: foams are dipped into 3 M HCl solution and
154 washed, (b) step 2: next DI water is used as the washing agent, and then (c) step 3: finally,
155 ethanol is used to wash the foam electrodes. Rigorous washing is crucial to remove the oxide
156 layer on the surface of the electrodes. After three step washing is completed, the foams are kept
157 at $60 \text{ }^{\circ}\text{C}$ for 6 h. This process will help dry the foams and make them ready for the coating
158 process mentioned above.

159 The solid-state symmetric supercapacitor is fabricated in a CR2032 coin cell. The coin cell is
160 studied for this work, and all the data are collected based on the two-electrode study using the
161 fabricated coin cell. The symmetric device contains CeO_xN_y -based slurry coated on the Ni foam,
162 used as both positive and negative electrodes (hence the term “symmetric” is used). PVA-KOH-
163 based gel electrolyte (prepared with 2 M KOH) is used along with a separator of cellulose filter
164 paper.

165 **2.2. Electrochemical characterization and relevant formulas**

166 Post fabrication, the device's electrochemical performance is observed using the CHI608E
167 electrochemical workstation. The electrochemical studies are carried out in a potential window of
168 0 to 1 V. The study on temperature variations is done in the Weiss environmental-climate
169 chamber. The temperature is increased stepwise from 10 to $70 \text{ }^{\circ}\text{C}$ with a step size of $10 \text{ }^{\circ}\text{C}$. The
170 humidity inside the chamber is kept fixed at 50 % relative humidity (RH). The fabricated
171 supercapacitor device is housed inside the environmental chamber for 24 h duration to confirm

172 that the whole device is at the same temperature. This process ensured the reliability of the data
173 acquisition during the temperature-dependent electrochemical study. Conventional
174 electrochemical studies for supercapacitors, such as cyclic voltammetry analysis, electrochemical
175 analysis, and galvanostatic charge-discharge process, are done at room temperature. The charge-
176 discharge studies are done in different temperatures to aid the data-driven study on the charge-
177 discharge prediction at higher temperatures. The areal capacitance (AC) is calculated using the
178 equation given below:

$$179 \quad AC = \frac{\int_{V_1}^{V_2} I_s \times V dV}{ESA \times \nu \times (V_2 - V_1)} \quad (1)$$

180 In equation (1), ‘V2’ is upper limit of the potential window and ‘V1’ is the lower limit of the
181 potential window. Here the value of ‘V2 – V1’ is equal to 1 V. The scan rate is denoted with ‘ ν ’
182 and the unit is mVs^{-1} . ESA stands for electrode surface area. The ‘ I_s ’ signifies the current value
183 per area with a unit of $mAc m^{-2}$. The areal capacitance using a charge-discharge plot at a constant
184 current value (AC_G) is calculated using the equation below:

$$185 \quad AC_G = (I_s \times t_{dis}) / (ESA \times (V_2 - V_1)) \quad (2)$$

186 The t_{dis} stands for the discharge time in the galvanostatic charge-discharge plot.

187 **2.3. Density Functional Theory**

188 The DFT tool of the Vienna ab initio simulation package (VASP) is used to analyse the structural
189 and electronic properties of CeO_xN_y with projector-augmented wave (PAW) potentials along
190 with a plane-wave basis set of cutoff energy of 460 eV[44,45]. The generalized gradient
191 approximation (GGA-PBE) developed by the Perdew–Burke–Ernzerhof is used as exchange and
192 correlation potential[46]. An 11x11x11 Monkhorst Pack k-point mesh is used for the Brillouin

193 zone scheme[47]. Along with the periodic boundary state, the spin-polarized calculation is also
194 performed. In this work, the calculations are designed to optimise the lattice constant of CeO_xN_y
195 concerning total energy calculations for various temperatures. A stable configuration of the
196 CeO_xN_y compound upon temperature can be found by integrating the lattice distortion and the
197 thermal expansion of Ce/O(N) ions from equilibrium positions, even though the ab initio
198 calculations have been performed at 0K. This study has successfully predicted the electronic
199 structure of CeO_xN_y for the temperature range from 268 K to 353K using the above technique.

200 **2.4. Details of the Database**

201 A database is compiled using SCC experimental data under different temperatures. The database
202 contains features such as: a) Current (A), b) Temperature ($^{\circ}\text{C}$), c) Bandgap (eV), d) Time (s), and
203 e) Voltage (V). The range of temperature considered for this work is 10 - 70 $^{\circ}\text{C}$. The
204 galvanostatic charge-discharge data are collected against various current values ranging from 1
205 mA to 20 mA. Bandgap is a critical feature that determines the mobility of charge carriers
206 between the anode and cathode during charging and discharging. Also, the bandgap directly
207 depends on the temperature (a detailed computational study regarding band gap and temperature
208 is provided in section 3.1), where the experimentation is conducted. This is because, with
209 increased temperature, materials undergo expansion, resulting in changes in lattice parameters
210 and interatomic distances. This can influence the electronic structure of the materials, thus
211 affecting the bandgap.

212 Due to the relation of bandgap and temperature, the concern of multicollinearity arises.
213 Multicollinearity is a statistical phenomenon in regression analysis where two or more
214 independent variables in a model are highly correlated. In simpler terms, it means that there is a

215 linear relationship between two or more predictor variables. This correlation among independent
216 variables can pose challenges in estimating the individual coefficients and understanding the true
217 relationships between the independent and dependent variables. Multicollinearity becomes a
218 concern when these relationships are too strong. In this work, bandgap and temperature share a
219 correlation value of 1.0. To avoid multicollinearity, only temperature was fed into the model. The
220 objective was to predict the voltage curve for 70 °C, with temperatures up to 60 °C as the
221 training dataset.

222 **2.5. ML Algorithms**

223 Decision Trees and Random Forests are popular ML algorithms in supercapacitor research,
224 helping with performance prediction, material discovery, and optimization. Decision Trees split
225 data into branches based on input features, offering precise, easy-to-understand models showing
226 how material properties or operating conditions impact performance. This makes them especially
227 useful for predicting electrode performance under different temperatures or classifying materials
228 by electrochemical properties. However, Decision Trees can sometimes overfit noisy data,
229 making them less reliable for new datasets[48–50]. Random Forests tackle overfitting by
230 combining multiple decision trees and averaging their predictions, resulting in more accurate and
231 reliable models. This makes them ideal for capturing the complex, non-linear behavior of
232 supercapacitors. However, they are less interpretable than single decision trees and can be
233 computationally demanding, requiring more processing power and memory[51]. Decision trees
234 and random forests are potent tools for advancing supercapacitor research and balancing
235 interpretability, robustness, and computational efficiency. This work also utilizes support vector
236 regression and linear regression.

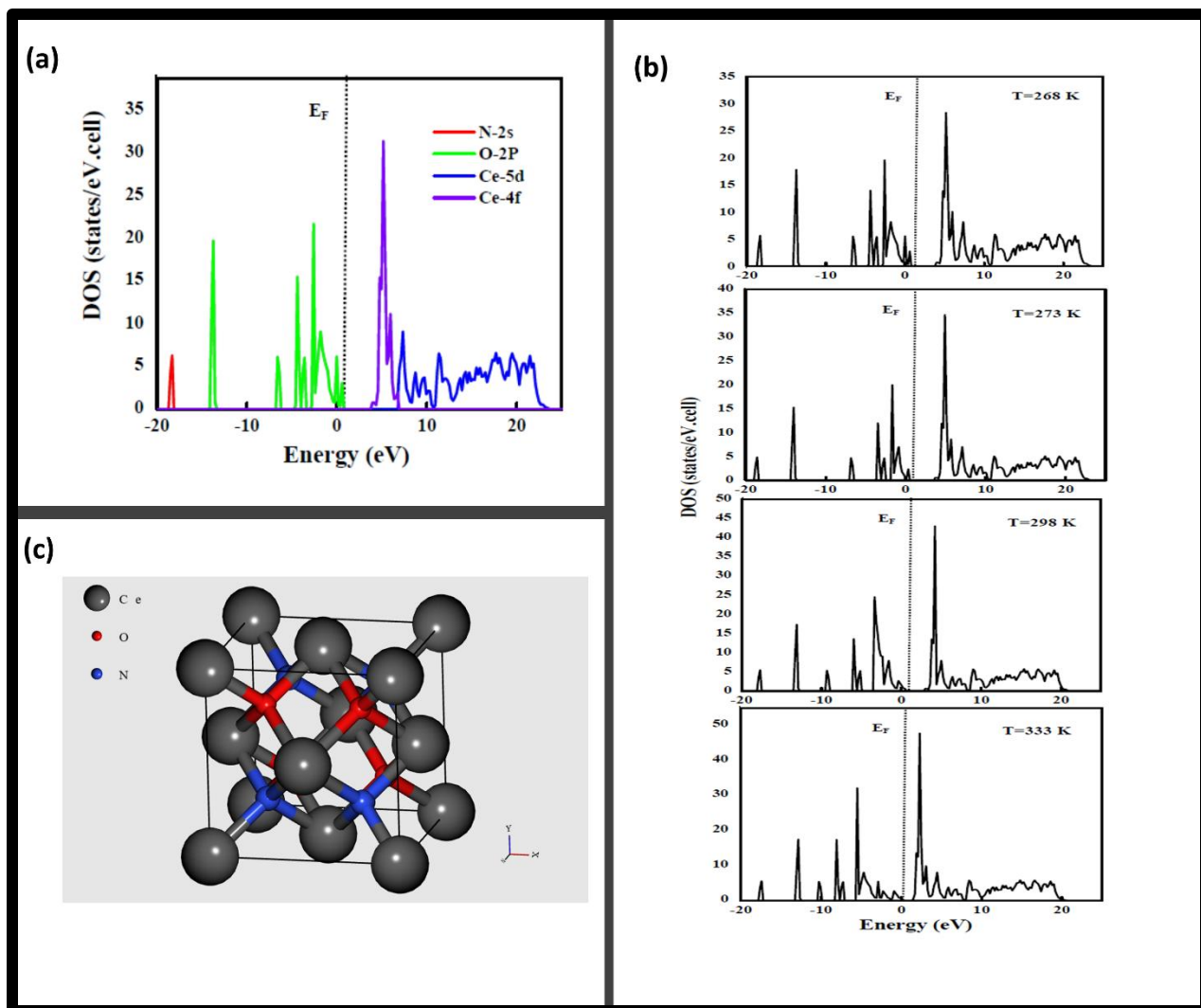
237 **3. Results and Discussions**

238 **3.1. DFT analysis**

239 The electronic properties need to be controlled as decisively as possible to accomplish superb
240 execution of the materials for modern applications, especially oxides and oxynitrides. In this
241 way, it is crucial to know precisely the reliance of the energy bandgap of oxides and oxynitrides
242 on the arrangement and temperature. Numerous studies have shown that as the temperature rises,
243 the energy gaps of oxides and oxynitrides decrease [52]. This temperature dependence is very
244 close to that seen in crystals and is generally explained by thermal lattice expansion and electron-
245 phonon interactions [53]. On the other hand, from an atomistic standpoint, little is known about
246 the mechanism underlying the temperature dependence of the energy gaps of the materials
247 mentioned above. Consequently, it is critical to explain this component, considering the
248 structural fluctuation expected to increase as miniaturization progresses. Due to the high
249 computational costs involved, mainly when first-principles methods are used, few electronic
250 structure calculations have been performed on the temperature dependence of energy gaps.

251 The lattice parameter of CeO_xN_y increases monotonically as the temperature rises, as seen in
252 graph X. By comparing the value at 353 K to the value at 0 K, it is observed that the thermal
253 expansion along an x-axis increases by 0.6 per cent. The band gap variance contributed by lattice
254 thermal expansion can be determined using the corresponding lattice constants, as shown in Fig.
255 1. The bandgap shift indicates a gradual decrease from 0 to 200K, then a steady behaviour for
256 temperatures above 200 K. Compared to 0 K, the band gap is reduced by 1.18 eV at 333K. Like
257 numerous different semiconductors, the fundamental computational estimations on CeO_xN_y
258 show comparable patterns of band gap narrowing with expanding temperature[54].

259 The total and partial density of states (DOS) were computed to analyze the electronic properties
260 of CeO_xN_y at different temperatures. The outcomes are shown in **Figure 1**. The dotted flat line
261 indicates the Fermi energy level (E_F). **Figure 1** shows that the CeO_xN_y is a wide band gap
262 semiconductor ($E_g = 2.82$ eV) at 0 °C. The N-2s and N-2p peaks are sharp and narrow, which lie
263 in the range of ~ -18 eV. These are very different from the peaks near the Fermi level, dominated
264 by O-2p states, found near ~ -10 eV. The peaks form below E_F due to O-2p and N-2p state
265 electrons and a relatively minor contribution from Ce-5d state electrons. Above E_F , there is a
266 hybridization between Ce-4f and the O-2p states. The highest peaks in the conduction state of
267 CeO_xN_y are due to Ce-4f, with a minor contribution from Ce-5d and O-2p state electrons
268 [**Figure 1**].



269

270 **Figure 1.** (a) The partial density of states (PDOS) of CeO_xN_y (b) the total density of states of
 271 CeO_xN_y at different temperatures, (c) the crystal structure of CeO_xN_y. Fermi energy and band
 272 gap decrease upon reducing temperature.

273 The DOS of CeO_xN_y shows a significant change in the conduction state because of the
 274 expansion in temperature. The Fermi level of CeO_xN_y also decreases and shifts downwards from
 275 0 °C, as shown in **Figure 1**. With rising temperature, the lattice expands, increasing the
 276 interatomic distances. This reduces the overlap of atomic orbitals, which decreases the band
 277 dispersion and shifts electronic states to lower energy. Moreover, the hybridization strength,

278 particularly between Ce-4f and O-2p states in CeO_xN_y , weakens, reducing the conduction band's
279 width and lowering the Fermi level. The dependence of the electronic structure on thermal lattice
280 expansion has been studied in oxide materials by Lucovsky et al. [55], and they discussed how
281 thermal expansion leads to electronic band shifts in transition metal oxides, which correlates with
282 the findings here. Additionally, increased temperatures may enhance lattice vibrations (phonons),
283 which interact with electronic states and redistribute energy levels. This interaction causes a shift
284 of electronic states to lower energy regions. It narrows the effective bandgap, as observed in
285 oxides and semiconductors like Si and GaAs, as reported by Zollner et al. [56]. These
286 mechanisms extend to wide bandgap materials such as oxides and oxynitrides and agree with the
287 results in this work. Furthermore, the narrowing of the peaks occurs because thermal vibrations
288 introduce disorder into the system. This reduces the coherence of the electronic states, leading to
289 a less uniform distribution of states and sharper, narrower peaks at specific energy levels.
290 Theoretical studies by Giustino et al. [57] on the electron-phonon coupling in semiconductors
291 align with this observation. They demonstrate that thermal disorder reduces the density of
292 available states, leading to more localized features. Thermal expansion and phonon interactions
293 explain the temperature-dependent shifts and narrowing of DOS peaks observed in materials like
294 CeO_xN_y .

295 As a result of the increased temperature, the width of the conduction state in CeO_xN_y is reduced,
296 and the height of the peak is increased. The decrease in the band gap is primarily due to a change
297 in the conduction state and a decrease in the Fermi level value, which predominantly comes from
298 the relatively minor hybridisation between Ce-4f and O-2p state electrons. The relation between
299 lattice constant, Fermi level (E_F), and bandgap is provided in Table 1. The experimental relevant
300 outcomes are the following - (i) Lattice parameters increase with temperature rise, (ii) Fermi

301 energy and band gap decrease upon reducing temperature, (iii) Peaks are shifted towards the
302 lower energy region and narrowing, (iv) The highest peak is due to Ce-4f state orbital with little
303 contribution from O and N-2p states (indicates the hybridisation), and (v) The majority peaks in
304 the valence and conduction bands are due to O-2p and Ce-5d states, respectively.

305 Table 1: Relation between temperature, Lattice constant, Fermi Level and Bandgap of CeO_xN_y.

T (K)	Lattice constant	E _F (eV)	E _g (eV)
200	5.249	1.510	4.01
223	5.286	1.466	3.72
243	5.301	1.379	3.44
268	5.360	1.366	3.23
273	5.401	1.101	2.82
298	5.416	0.799	2.46
303	5.420	0.710	2.44
313	5.424	0.502	2.020
333	5.428	0.120	1.642
353	5.431	0.118	1.638
363	5.439	0.116	1.634
373	5.442	0.112	1.630

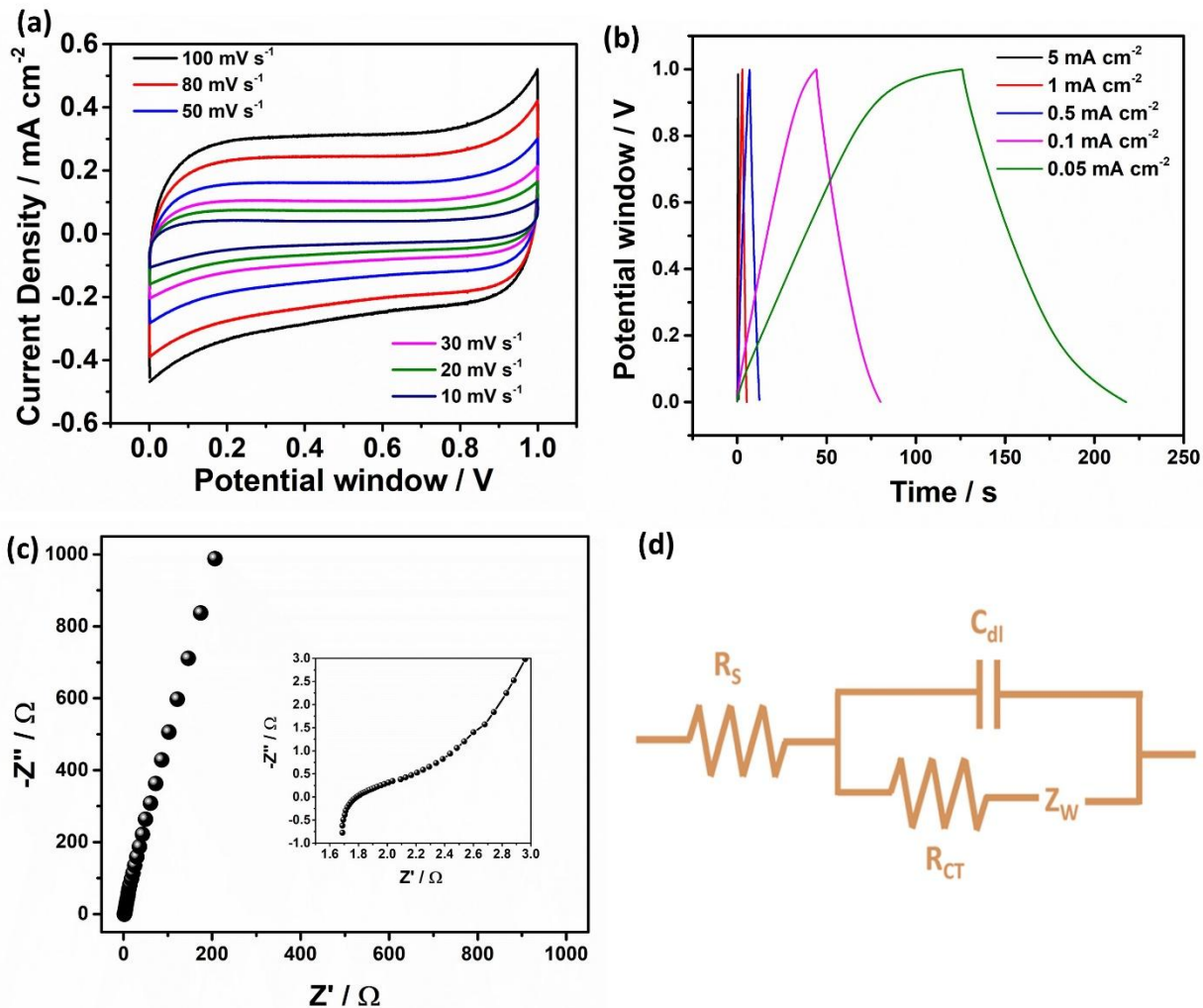
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307 3.2. Analysis of Electrochemical Characterization

308 The electrochemical properties of the symmetric coin cell (SCC) are evaluated using cyclic
309 voltammetry (CV), galvanostatic charge-discharge (GCD), and electrochemical impedance
310 spectroscopy (EIS). As depicted in **Figure 2a**, the CV profiles for SCC exhibit a quasi-
311 rectangular shape characteristic of typical pseudocapacitor behaviour within a potential range of

312 0 to 1 V at a scan rate of 10 to 100 mVs⁻¹. These electrodes' overall working potential windows
313 define the device's effective cell voltage, resulting in a symmetric solid-state supercapacitor with
314 a cell voltage of 1 V.

315 The CV curves for the SCC device is studied at a 10 to 100 mV/s scan rate range. The curves
316 exhibit characteristics typical of pseudocapacitors with quasi-rectangular shapes[58,59]. The
317 SCC device demonstrates an areal capacitance of 4.11 mF cm⁻² at a scan rate of 10 mV s⁻¹ and
318 2.91 mF cm⁻² at 100 mV s⁻¹. Notably, it retains ~70.8% of its initial capacitance at higher scan
319 rates, indicating robust rate capability even under rapid scanning conditions. The reduced charge
320 storage capability in higher scan rates is a well-documented phenomenon in supercapacitor
321 devices, primarily due to less interaction of electrolyte ions with the active electrode
322 materials[60,61]. Typical governing equations for the CV study is added in the supplementary
323 file.



324

325 **Figure 2.** Electrochemical characterization study conducted at room temperature of the
 326 symmetric coin cell fabricated using the cerium oxynitride electrode. (a) cyclic voltammety
 327 study for various scan rates, (b) charge-discharge plot at constant current at various current
 328 densities, (c) Nyquist plot derived from electrochemical impedance spectroscopy study, and (d)
 329 representative equivalent circuit of the device using the series resistance (R_s), charge-transfer
 330 resistance (R_{CT}), double layer capacitance (C_{dl}), and Warburg impedance (W).

331

332 The GCD curves of the SCC device at a current density range of 0.05 to 5 mA cm⁻², are

333 illustrated in **Figure 2b**. The curves exhibit a characteristic triangular shape, indicative of
334 symmetric supercapacitor behaviour. The device achieves a maximum areal capacitance of ~ 4.6
335 mF cm^{-2} at a current density of 0.05 mA cm^{-2} . In **Figure 2c**, the Nyquist plot of the SCC device is
336 displayed over a frequency range from 0.01 Hz to 100 kHz. The inset highlights a depressed
337 semicircle in the high-frequency region, followed by a linear segment in the low-frequency
338 region, corresponding to charge transfer and diffusion resistance, respectively. The depressed
339 semicircle suggests the presence of interfacial charge-transfer resistance, while the transition
340 between the semicircle and the linear portion indicates diffusion-controlled processes. The linear
341 region reflects the capacitive nature of the device. This demonstrates that the SCC device utilizes
342 three distinct charge storage mechanisms: (a) interfacial surface redox reactions, (b) diffusion-
343 controlled processes, and (c) capacitive charge storage[62–64]. The coin cell shows a solution
344 resistance (ESR) of $\sim 1.6 \Omega$. Figure 2d shows the representative equivalent circuit of the device
345 based on the electrochemical impedance spectroscopy. The low series resistance value highlights
346 the oxynitride-based device's rapid kinetics and excellent rate capability, contributing to its high
347 cyclic stability.

348 **3.3. Machine Learning-based Study**

349 This work deals with the effect of temperature on the charging-discharging of the coin cell. With
350 increasing temperature, faster charging is observed, and it also agrees with the literature. From
351 the work of Jiao-ling Hong et al., it is shown that the increased temperatures lead to a reduction

352 in electrolyte viscosity resistance. This facilitates accelerated ion movement and diffusion within
353 the supercapacitor. This enhanced mobility enables a greater influx of ions into the porous
354 electrode, resulting in the supercapacitor's expanded capacitive storage capacity. This increase in
355 ion mobility helps increase current flow while charging and improves performance at higher
356 temperatures.

357 This work uses Decision Trees and Random Forests for regression tasks. Decision Trees predict
358 continuous values by splitting data into decision nodes based on features, creating a simple,
359 interpretable tree structure that captures both linear and non-linear relationships. Random Forest,
360 an ensemble method, builds multiple decision trees using random subsets of data and features,
361 then averages their outputs for a robust prediction. Chosen for its ability to handle non-linear
362 relationships and mitigate overfitting, Random Forest is particularly suited to experimental data,
363 where noise can obscure underlying patterns. Additionally, linear regression and support vector
364 machines were also tested. Linear regression assumes a linear relationship between variables,
365 while support vector machines find an optimal hyperplane to separate or fit data with maximum
366 margin in a high-dimensional space. This work aims to predict SCC charge storage performance
367 at higher temperatures using lower-temperature electrochemical data. The step-by-step
368 procedures and data-driven insights are outlined below.

369 From the charge-discharge plots of the supercapacitor, observations indicate that the time period
370 for each charge-discharge cycle increases with an escalation in temperature and decreases with a
371 fall in the current value (**Figure 3**). Hence, the initial step is to address this pattern to predict the
372 time period for a temperature of 70 °C at various current values. A new feature (or variable),
373 named 'Time period', is introduced to calculate the total time for a specific temperature and
374 current. **Fig. 3** demonstrates a proportionality-based relation between the time period and

375 temperature. On the other hand, the time period is observed to decrease with the current,
376 particularly in the logarithmic scale. This is a known phenomenon in energy storage materials
377 examined under different constant current values for charge-discharge analysis.

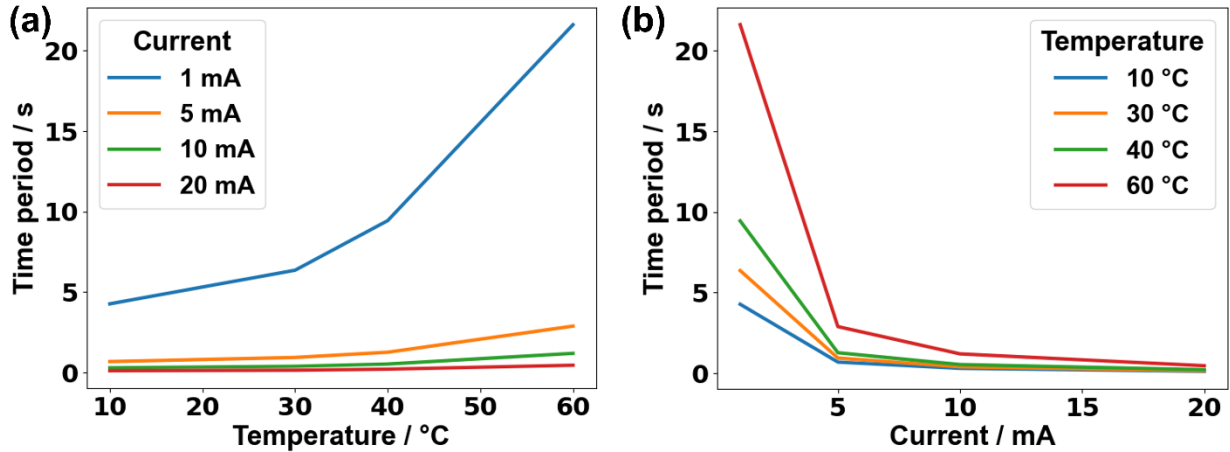
378 Subsequently, machine learning algorithms such as linear regression, support vector regression,
379 decision tree and random forests are constructed with temperature and current as independent
380 variables and time period as the dependent variable. For decision trees and random forests,
381 temperature and current are treated as input features. The ratio of temperature and current is
382 considered the input feature for linear regression and support vector regressors with linear
383 kernels. This is done because linear kernels are suitable for linear regression and support vector
384 regression algorithms due to their computational efficiency and performance for linear
385 relationships. Their effect becomes optimum when data can be separated by a plane, hyperplane,
386 or a straight line. In support of this opted methodology, a work by Nariswari et al. can be referred
387 to, where it is observed that the support vector machine with a linear kernel outperforms other
388 models while evaluating non-linear data[65]. The machine learning models underwent training
389 and arranged along with their mean squared error values in Table 2.

390 Table 2: Comparison of different Machine learning models in terms of mean squared error.

Machine learning algorithm	Mean squared error
Linear regression	0.48
Support vector regression	4.80

Decision tree	0.06
Random forest	7.02

391



392

393 **Figure 3.** Variation of time period with temperature and current. (a) depicts an increasing time
 394 period with rising temperature at different current values and (b) a decreasing time period with
 395 increasing current values at different temperatures.

396 The decision tree performed much better than the other models, as shown in Table 2. Further, to
 397 optimize the model's performance, hyperparameter tuning is conducted on parameters such as
 398 'max_depth', 'min_samples_split', and 'criterion'. 'max_depth' represents the maximum depth
 399 of the tree. It controls the maximum number of levels from the root node to the deepest leaf
 400 node. 'min_samples_split' is the minimum number of samples required to split an internal node.
 401 If the number of samples at a node is below this threshold, the node will not split, becoming a
 402 leaf node. The 'criterion' parameter determines the function used to measure the quality of a
 403 split. This tuning is accomplished through GridSearchCV, a technique for hyperparameter
 404 optimization in machine learning[66]. GridSearchCV, part of the scikit-learn library in Python,

405 executes an exhaustive search over a specified parameter grid to identify the best combination of
 406 hyperparameter values for a given model. GridSearchCV also helps avoid overfitting, as it
 407 performs cross-validation, up to 5 iterations, on each combination. The resulting model exhibited
 408 exceptional performance, achieving an R^2 score of 99.92% and a mean squared error of 0.05. The
 409 R^2 score, also known as the coefficient of determination, is a statistical metric illustrating the
 410 proportion of the variance in the dependent variable explained by the independent variables in a
 411 regression model. The formula for the R^2 score is defined as:

$$412 \quad R^2 = 1 - \frac{\text{Sum of squared residuals}}{\text{Total sum of squares}} \quad (3)$$

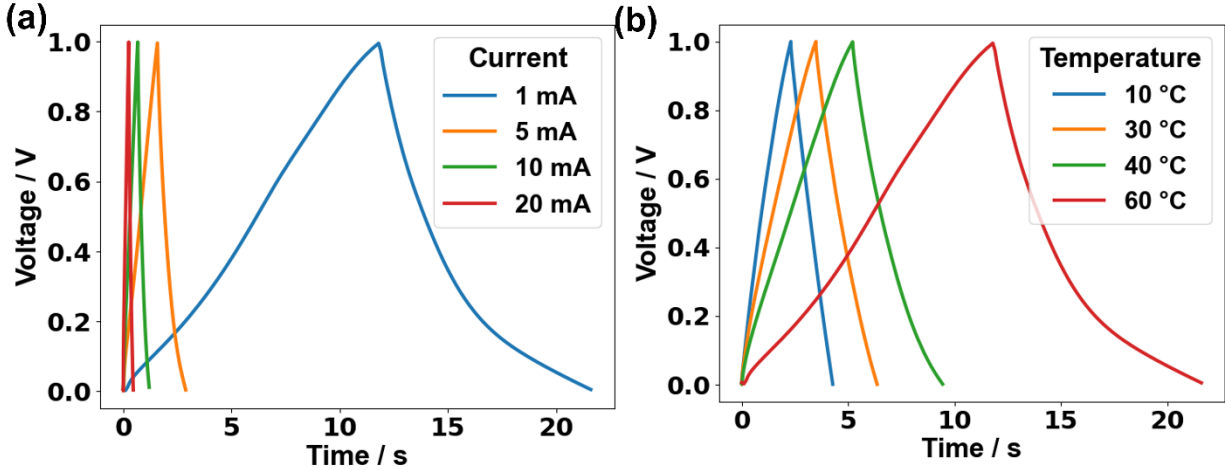
413 Table 3: Performance of the model to predict the time period at 70 °C.

Current (mA)	Calculated time period (s)	Predicted time period (s)	Difference (%)
1	23.26	21.60	7.13
5	3.04	2.89	4.93
10	1.22	1.20	1.63
20	0.48	0.46	4.16

414

415 Table 3 shows that the predicted time period value is considerably close to the calculated value,
 416 with a maximum difference of ~7.1 % and a minimum difference of ~1.6%. This low difference
 417 highlights the ML model's good performance in predicting the time period. The next task is to
 418 plot the charge and discharge cycle and observe the visible difference in the alignment of the
 419 charge-discharge plots. Therefore, the subsequent objective involved the construction of the

420 charge-discharge cycle utilizing the calculated and predicted time periods. The temperature
 421 dependency (with fixed current value) and current dependency (with fixed temperature value) are
 422 given in **Figure 4**.



423
 424 **Figure 4.** Galvanostatic charge-discharge curves for (a) different current values at a fixed
 425 temperature of 60 °C and (b) different temperatures at a fixed current value of 1 mA.

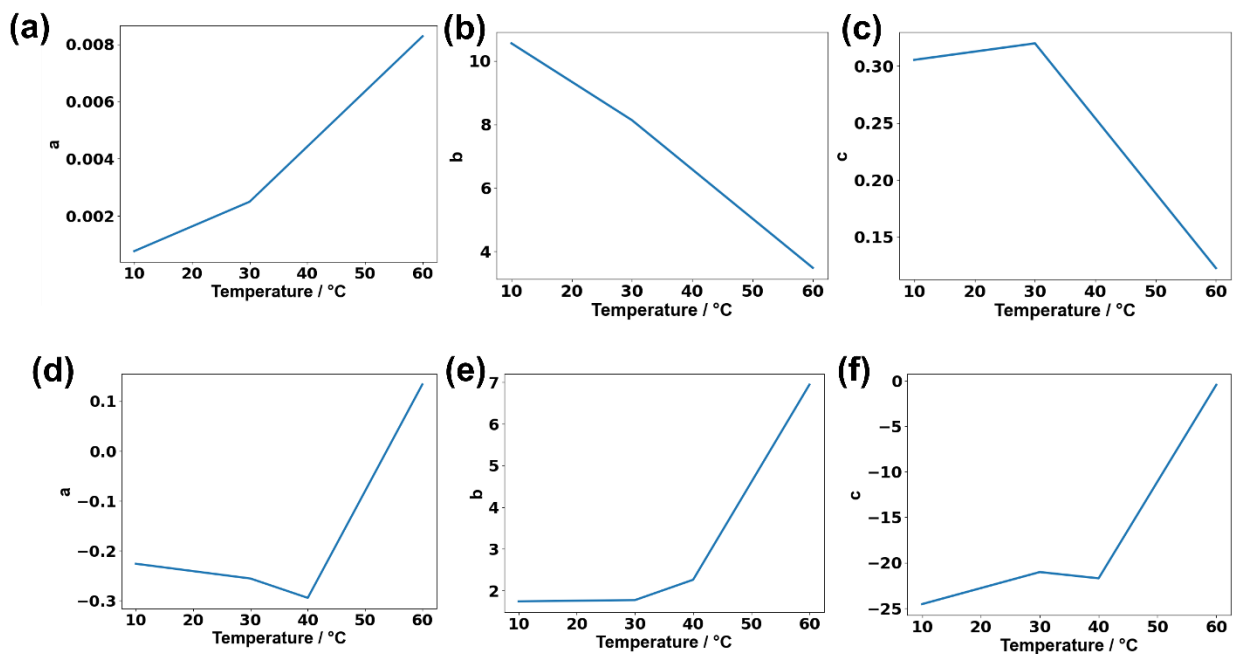
426 The voltage-time curve exhibited both exponential and linear behaviours. Furthermore,
 427 observations revealed that variations in temperature and current influenced the curve's behaviour,
 428 with an increase in temperature and decrease in current leading to a more exponential response,
 429 and conversely, a decrease in temperature and increase in current resulting in a more linear
 430 behaviour. As a result, the voltage is conceptualised as a function of time, integrating both
 431 exponential and each term. The formulation for charging looks as follows:

432
$$V(t) = b \times (1 - e^{-\frac{t}{a}}) + c \times t \quad (4)$$

433 The formulation for discharging looks as follows:

434
$$V(t) = b \times (e^{-\frac{t}{a}}) + c \times t \quad (5)$$

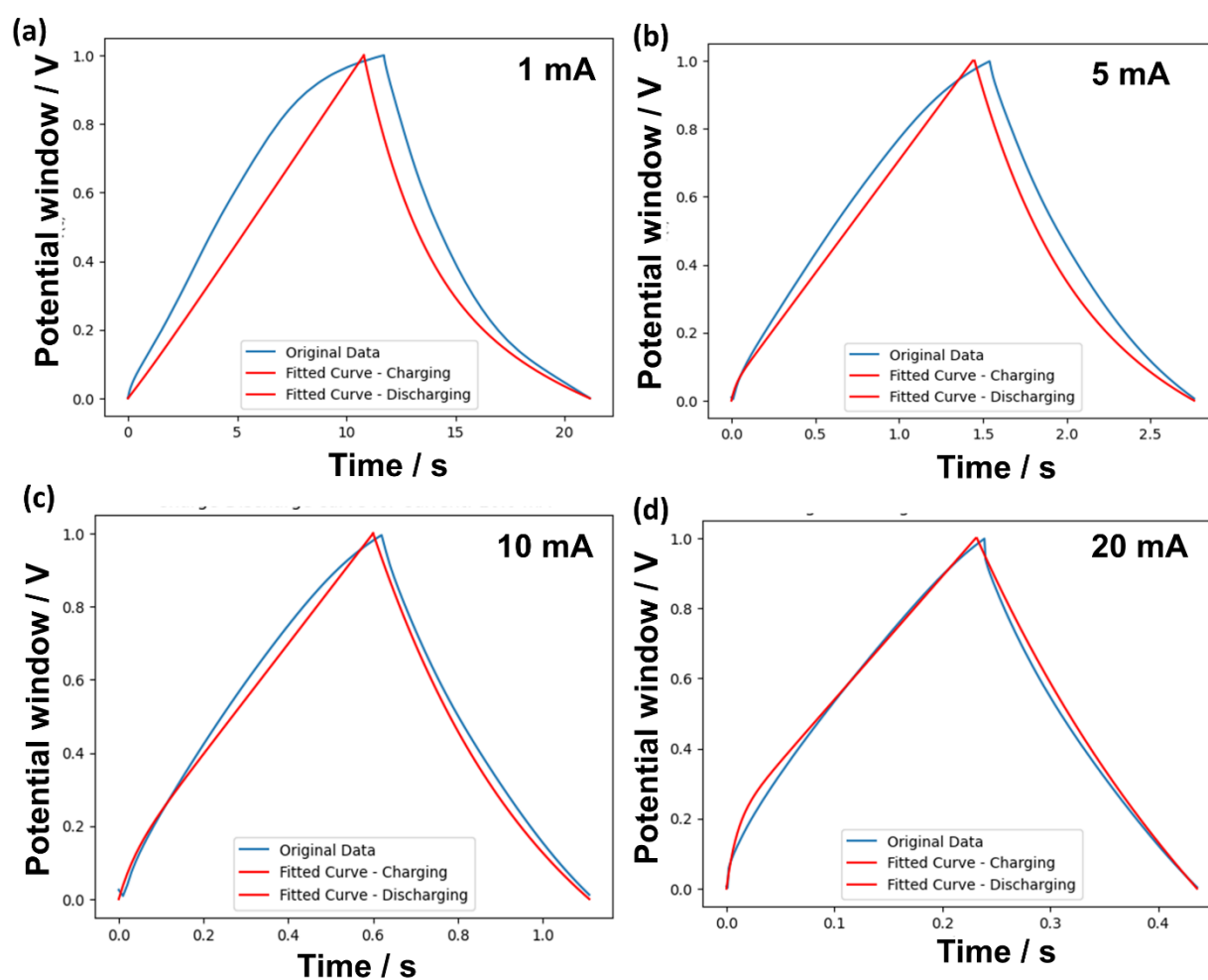
435 The parameters 'a', 'b', and 'c' in both equations underwent optimization and tuning. The
 436 devised function is specifically designed to align with the existing time and voltage data for each
 437 time period, employing a maximum of 25,000 iterations. The parameters exhibited variations in
 438 response to temperature, and the formulated functions aimed to capture these bi-natured
 439 variations (linear and exponential). The variation of the parameters with temperature is depicted
 440 in **Figure 5** for the charging and discharging scenario.



441
 442 **Figure 5.** Variation of 'a', 'b', and 'c' parameters against increasing temperature at a constant
 443 current value while charging for (a) parameter 'a', (b) parameter 'b', (c) parameter 'c', and while
 444 discharging for (d) parameter 'a', (e) parameter 'b', (f) parameter 'c'.

445 Subsequently, distinct Random Forest models were constructed, each with different parameters
 446 mentioned above as the dependent variable and the time period as the independent variable.
 447 Analogous to the earlier prediction of the time period, these models underwent training with time
 448 periods associated with temperatures up to 60 °C. Subsequently, they are utilized to forecast time

449 periods for a temperature of 70 °C. This iterative process is replicated for both the charging and
450 discharging segments of the cycle. The predicted parameters were then employed to formulate
451 the voltage function plotted against time. The experimental data is confined to charging up to 1V
452 (this is due to the electrochemical potential window of 1 V as discussed in section 3.2), so the
453 predicted function is appropriately scaled before visualization. The comparative performance of
454 the predicted and original curves is demonstrated below (**Figure 6**).



455
456 **Figure 6.** Machine learning derived predicted galvanostatic charge-discharge curve at 70 °C for
457 different current values, (a) 1 mA, (b) 5 mA, (c) 10 mA, and (d) 20 mA.

458 This ML-based study demonstrates the potential to predict supercapacitors' high-temperature
459 charge storage performance using lower-temperature electrochemical data. Heating, caused by
460 internal resistance during high-frequency charge-discharge cycles, impacts supercapacitor
461 performance in buffer circuits, leading to efficiency loss and potential electrode and electrolyte
462 damage. Hence, predicting the charge-storage performance of the supercapacitor electrode
463 materials at a higher temperature bears contemporary importance in advanced energy storage
464 solutions [67–69].

465 The ML model developed in this study accurately predicts supercapacitor performance across
466 varying temperatures, aiding material and device optimization under operational conditions.
467 These insights are crucial for scaling the technology, identifying temperature limits, performance
468 bottlenecks, and energy-efficiency trade-offs, and supporting prototype testing (TRL 6) and
469 system integration (TRL 7). Cerium oxynitride shows promise as an electrode material, with
470 future work on large-scale synthesis and real-world testing poised to advance it to higher TRLs.

471 **4. Conclusion**

472 This work focuses on the application of ML in the temperature-dependent performance of
473 supercapacitors. Combining machine learning with DFT insights, this article provides a
474 comprehensive approach to optimizing thermal resilience in supercapacitor electrodes. Cerium
475 oxynitride offers a reasonable charge storage capability of $\sim 4.6 \text{ mF cm}^{-2}$ and low ohmic
476 resistance of $\sim 1.6 \text{ } \Omega$ in a symmetric coin cell setup. Consequently, the temperature-dependent
477 data are taken by running electrochemical tests inside an environmental chamber by increasing
478 the temperature stepwise. Finally, the Random Forest algorithm is used to predict the charge-
479 discharge performance at higher temperatures and different applied current values (1 to 20 mA).
480 Data-driven study is utilized to model to capture both exponential and linear relation of the

481 potential window with respect to time, during both charging and discharging. The value of the
482 relevant parameters associated with time and potential window is iterated over 25,000 times
483 before finalizing the model.

484 Energy storage is vital to achieving a sustainable and resilient energy future. Advancing this field
485 requires integrating high-performance materials with data-driven predictive models to overcome
486 limitations and unlock potential. This work underscores the transformative impact of combining
487 data-driven methodologies with electrochemistry and materials science, paving the way for
488 innovative strategies essential for a greener future.

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494 **Data Availability**

495 Data can be obtained upon request from the corresponding author.

496 **Declaration of Competing Interest**

497 The authors declare that they have no known competing financial interests or personal
498 relationships that could have appeared to influence the work reported in this paper.

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