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Hyaluronan composite bioink preserves nucleus pulposus cell phenotype in a stiffness-dependent manner

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Abstract

Intervertebral disc degeneration is a major cause of neck and back pain, representing a significant global socioeconomic burden. The polysaccharide hyaluronan is key to maintaining disc physiology and mediating disc disease through its structural and biological roles in the nucleus pulposus, a component of the intervertebral disc highly susceptible to degeneration. In this study, we introduce a novel composite bioink designed for extrusion bioprinting of structures resembling the nucleus pulposus. Our bioink combines levels of hyaluronic acid and collagen that approach physiological concentrations and effectively mimics the disc's hydrated and mechanically resilient environment. We modulated the composite's mechanical properties through the tyramination of hyaluronic acid and subsequent photocrosslinking, influencing morphology and gene expression of embedded bovine nucleus pulposus cells. This allows us to replicate a range of properties from healthy to degenerated human nucleus pulposus, which would be challenging to achieve with traditional cell culture and in vivo models. Our results show that modulating hyaluronan physico-chemical properties influenced embedded cell phenotype. The outcomes of this study inform the future design of biomaterials for the modeling of disc disease and regeneration, and present a versatile platform that can be readily integrated with other biofabricated components to form engineered intervertebral disc-like structures.

Keywords: intervertebral disc, nucleus pulposus, bioprinting, in vitro models, hyaluronic acid

31 1. Introduction

32 Intervertebral disc (IVD) degeneration is the most common contributor to neck and back pain (DePalma
33 et al., 2011), which are leading causes of disability worldwide and represent a substantial socioeconomic
34 burden (Hartvigsen et al., 2018; Vos et al., 2012). Our understanding of its complex pathophysiology
35 remains limited despite extensive research, primarily because of reliance on animal tissue and simplistic
36 cell culture models (Mainardi et al., 2022; Wang et al., 2022). This limitation is reflected in the traditional
37 treatments, which escalate rapidly from pain management to invasive surgeries – such as discectomies,
38 spinal fusions, and synthetic replacements – yet often fail to restore the spine’s full range of functionality
39 (Romaniyanto et al., 2022).

40 Central to the IVD’s function is the nucleus pulposus (NP), a hyaluronic acid (HA)-rich gel-like yet
41 multiphasic core confined within an elastic annulus fibrosus ring that plays a crucial role in the spine’s
42 ability to transfer loads and maintain flexibility (Iatridis et al., 1996). NP’s unique properties are derived
43 from its composition: collagen type II and large quantities of proteoglycans, predominantly aggrecan.
44 Aggrecan, associated with HA, forms negatively charged complexes essential for maintaining the NP’s
45 hydration and resilience against compressive forces (Singh et al., 2009). Although the exact sequence of
46 events is yet to be fully elucidated, it is clear that disc degeneration alters this mechanical function
47 through compositional changes brought about by a combination of factors: cell senescence and reduced
48 regenerative potential (Le Maitre et al., 2007; Zhao et al., 2007), build-up of waste products and reduced
49 nutrient transport (Roberts et al., 1996; Shirazi-Adl et al., 2010), and an imbalance between the
50 production and degradation of the extracellular matrix (ECM) (Pattappa et al., 2012; Singh et al., 2009;
51 Sztrolovics et al., 1997). These factors ultimately result in a reduction of the NP’s swelling capacity and a
52 loss of the IVD’s structural integrity. The compromised structure further exacerbates the problem, leading
53 to tissue injury, vascularization, innervation, infiltration by immune cells, inflammation, and a rapid
54 progression of tissue breakdown (Freemont et al., 1997; Le Maitre et al., 2005; Phillips et al., 2013;
55 Vergroesen et al., 2015). This complex interplay of biology and mechanics in the disc degeneration
56 process illustrates the importance of studying both in parallel.

57 Current models of IVD degeneration have significant limitations. Animal models, essential due to the
58 scarcity of human IVD tissue, capture the full intricacy of developed native structures. However, despite
59 their critical role in advancing our understanding of disc degeneration, they are inherently limited due to
60 the biomechanical and biochemical differences from humans. For instance, animal tissue is adapted for
61 loads that are unrepresentative of the bipedal human anatomy (Alini et al., 2008). Furthermore, the
62 absence of notochordal cells in adult humans, which are present in several animal species, represents a
63 fundamental discrepancy in the IVD’s regenerative potential (Erwin et al., 2006; Pattappa et al., 2012;
64 Poletto et al., 2023; Wang et al., 2022). In contrast to animal models, traditional cell culture systems do
65 not mimic the IVD’s compositional, structural, and mechanical complexity. Whether in 2D or 3D
66 matrices, they lack the full range of the body’s mechanical cues (Lazaro-Pacheco et al., 2023). Such cues
67 play a vital role in maintaining IVD homeostasis (Gawri et al., 2014; Paul et al., 2012; Wuertz et al.,
68 2009) and as potential triggers of degeneration (Setton & Chen, 2004; Wuertz et al., 2009). For example,
69 IVD cell morphology and proteoglycan production are highly sensitive to the multi-axial movements in
70 the body and stiffness variations in the disc structure (Hsieh & Twomey, 2010; Setton & Chen, 2004). In
71 the nucleus pulposus and annulus fibrosus, shear stiffness can vary between 5-60 kPa and 100-400 kPa
72 respectively, depending on the severity of degeneration (Iatridis et al., 1997, 1999). Although organ-on-a-
73 chip systems have attempted to replicate dynamic tissue aspects like nutrient transport and mechanical
74 loads (Mainardi et al., 2022), they have focused on discrete aspects of cell-cell and cell-material

75 interactions, far from the scale and complexity of a whole organ. Therefore, a precise and comprehensive
76 in vitro representation of the native IVD environment remains a distant goal.

77 The limitations of existing IVD models underscore the need for tissue-engineered alternatives. Such
78 constructs should aim to more closely replicate the native IVD structure, while also leveraging the
79 benefits of a bottom-up approach. These benefits include utilizing clinically relevant human cell sources
80 and offering the flexibility to tailor the model to specific research inquiries, such as studying different
81 stages of disc degeneration. By processing and chemically modifying ECM components, we can employ
82 various biofabrication techniques, like printing and electrospinning, to accurately mimic both the structure
83 and composition of the IVD (Schmitz et al., 2020; Vernengo et al., 2020). However, recent engineering
84 efforts have primarily concentrated on regenerative applications, thereby overlooking this aspect of
85 biomimicry. This led to using materials that fall short of faithfully replicating the ECM's biochemical
86 signals and mechanical properties (Gloria et al., 2020; Gullbrand et al., 2018; Liu et al., 2022; Marshall et
87 al., 2021; Yang et al., 2018). Addressing these shortcomings in a novel biomaterial would enable the
88 development of more accurate and functional in vitro IVD models.

89 To address this gap, we introduce a polysaccharide-based biomaterial designed to create in vitro structures
90 that mimic the NP. We leverage the native ECM components HA and collagen at their natural
91 concentrations (approximately 18-36 mg/mL and 15-20 mg/mL, respectively) (Iatridis et al., 1996;
92 Martins et al., 2018; Singh et al., 2009), replicating the NP's hydrated environment with the negatively
93 charged HA and providing biochemical cues suitable for preserving the NP cell phenotype. HA's
94 versatility as a substrate for chemical modification allows us to tune its properties through a
95 photocrosslinking process. We demonstrate that minor adjustments in composition can alter the final
96 stiffness across the entire spectrum from healthy to degenerated human NP and investigate how these
97 changes influence cell behavior. We showcase our composite's compatibility with 3D bioprinting
98 approaches, which confers it with advantages over hydrogels designed for simpler casting approaches: it
99 enables a more streamlined future integration with biofabricated annulus fibrosus structures and facilitates
100 reproduction of the radial gradation in composition and stiffness, as the NP transitions into the annulus.
101 We achieve this printability by incorporating sacrificial gelatin microgels, which modulate the rheological
102 properties of the bioink independently of the crosslinking parameters and dissolve out of the produced
103 hydrogel constructs under standard cell culture conditions at 37 °C due to the inherent thermoresponsive
104 properties of gelatin (Ouyang et al., 2022; Soliman et al., 2023). In summary, our composite is a flexible
105 addition to the biomaterials repertoire, enabling the development of biomimetic IVD models and
106 facilitating future advances in disc degeneration research.

107 **2. Materials and methods**

108 **2.1. Materials**

109 Sodium hyaluronate from biotechnological origin (*Streptococcus equi*), with a molecular weight 150-350
110 kDa (SEC-MALS) and $\geq 90.0\%$ purity, was purchased from Contipro a.s. Collagen was isolated in-house
111 from Wistar rat tails. Type A gelatin from porcine skin with 300 g Bloom was purchased from Sigma-
112 Aldrich (70-90% protein by Biuret test). All other reagents were obtained from various commercial
113 suppliers (see individual section) and used as purchased without further purification.

114 **2.2. Functionalization of hyaluronic acid with tyramine**

115 The amidation of HA (Nutrihyl, Contipro, MW: 150-350 kDa) with tyramine (Tyr) was conducted using
116 4-(4,6-dimethoxy-1,3,5-triazin-2-yl)-4-methylmorpholinium chloride (DMTMM, $>95.0\%$ purity, TCI) as

117 a coupling agent. Initially, HA sodium salt was dissolved in ultrapure water (Milli-Q, EMD Millipore) at
118 1% w/v and then heated to 37 °C. DMTMM was added to a final concentration of 25 mM and allowed to
119 dissolve completely. Subsequently, Tyr HCl (Roth, ≥98%), pre-dissolved at 1 M in ultrapure water, was
120 introduced dropwise until a 25 mM concentration was reached. The reaction was maintained at 37 °C
121 with continuous stirring for 24 h. Next, saturated NaCl (Roth, ≥99.0%) was added to a final concentration
122 of 13.5% v/v, before the solution was cooled down to room temperature. HA-Tyr precipitation was
123 triggered by adding 3 volumes of 96% ethanol (Alcosuisse, A15, ≥99.8% stock) dropwise. The precipitate
124 was further washed under vacuum in a Gooch filter crucible (grade 2, DURAN), with gradually
125 increasing concentrations of ethanol (80% to 99.8%). An argentometric test with silver nitrate (Sigma-
126 Aldrich, ≥99.5%) confirmed the absence of chlorides. The product was first dried in a fume hood for 24 h,
127 followed by vacuum drying at room temperature for an additional 24 h and at 40 °C for 48 h. The degree
128 of functionalization was measured spectroscopically by absorbance at 275 nm (Infinite M Plex, TECAN)
129 and compared to Tyr HCl standards (Darr & Calabro, 2009). HA-Tyr with 6.2% degree of substitution
130 was used in all experiments.

131 2.3. *Isolation and purification of collagen*

132 Type I collagen was extracted from the tails of young male Wistar rats. The tails were cleaned and rinsed
133 with 70% ethanol before dissection within a biosafety cabinet. The dissection involved gradually cutting
134 the tails from the tip in 1 cm segments to extract tendons, which were then washed with PBS (Sigma-
135 Aldrich) and dissolved in 50 mL of 0.5 M acetic acid (Carlo Erba) per tail for 24 h under gentle stirring.
136 Following dissolution, the collagen solution was centrifuged for 15 min at 3,000 g to remove debris and
137 purified by precipitation through the dropwise addition of 4 M NaCl until a final concentration of 0.7 M.
138 The precipitate was redissolved in fresh 0.5 M acetic acid at half the initial volume and dialyzed against
139 17 mM acetic acid to remove salts and adjust the pH to 4.5. The collagen solution was centrifuged at
140 30,000 g for 4 h at 10 °C to remove any remaining aggregates, and the supernatant was stored at 4 °C.
141 The collagen concentration was quantified using hydroxyproline titration, and its purity was confirmed by
142 SDS-PAGE electrophoresis (Mini-PROTEAN TGX, Bio-Rad). Finally, the collagen solution was
143 concentrated to 40 mg/mL by slow evaporation within the biosafety cabinet.

144 2.4. *Fragmentation of collagen*

145 40 mg/mL stock solution of collagen type I in 17 mM acetic acid was neutralized by the addition of 4.9%
146 v/v 1 M NaOH (Roth, ≥97.0%) and 11.7% v/v 10X PBS to achieve a final concentration of 1X PBS and
147 34.3 mg/mL collagen. This solution was transferred into 1 mL syringes (Omnifix, B. Braun) and
148 incubated at 37 °C for 30 min to form a stiff collagen hydrogel. The hydrogel was fragmented by repeated
149 extrusion, initially once without any needle attached, followed by three successive extrusions through
150 blunt-tip needles of sizes 20G, 25G, and 30G (Optimum, Nordson), respectively.

151 2.5. *Preparation of gelatin microgels*

152 Gelatin microgel preparation was adapted from literature, based on the method introduced by Hinton for
153 FRESH 3D printing (Hinton et al., 2015; Ouyang et al., 2022). Gelatin (300 g Bloom, type A, 70-90%,
154 Sigma-Aldrich) was dissolved in ultrapure water at a concentration of 4% w/v by stirring at 50 °C. An
155 equal volume of ethanol was added, followed by the addition of Pluronic F-127 (Sigma-Aldrich) and gum
156 Arabic (Thermo Scientific) at concentrations of 0.25% w/v and 0.1% w/v, respectively, relative to the
157 total volume. Upon complete dissolution, the mixture was cooled to room temperature under constant
158 stirring at 400 rpm for 16 h. The resulting microgel suspension was centrifuged at 300 g for 5 min, and
159 the supernatant was discarded. The microgels were washed and centrifuged three times with PBS. After

160 washing, the microgels were supplemented with 5 volumes of 99.8% ethanol while stirring at 400 rpm.
161 The suspension was filtered and washed with additional ethanol using a qualitative filter paper (grade 1,
162 Whatman) under vacuum to yield a dry powder. This powder was placed in an oven for further drying
163 under vacuum at room temperature for 24 h and subsequently stored at 4 °C for future use. To
164 characterize the produced microgels, they were rehydrated in PBS and imaged at 10x magnification using
165 phase contrast microscopy (Axio Vert.A1, Zeiss). The microgel images were segmented using Cellpose
166 2.0 with the human-in-the-loop approach (Pachitariu & Stringer, 2022), and their Feret diameter was
167 measured using ImageJ (National Institutes of Health).

168 2.6. Cell isolation and culture

169 NP cells were harvested from the tail discs of approximately one-year-old bovine males. The process
170 began with thoroughly washing the tails, followed by removing surrounding soft tissue using a scalpel.
171 NP tissue was extracted from the discs, starting from the largest intact disc and continuing until 5 to 10
172 discs were dissected, depending on the tail size. The collected tissue was rinsed in PBS containing 1% v/v
173 penicillin/streptomycin (P/S, Gibco). The tissue was finely minced into approximately 2 mm³ pieces
174 under sterile conditions in a laminar flow cabinet. These were then placed into a spinner flask to initiate
175 digestion, containing 0.2% w/v pronase (Roche) in Dulbecco's Modified Eagle Medium, with 4.5 g/L of
176 glucose and 110 mg/mL sodium pyruvate (DMEM HG, Gibco), supplemented with 10% v/v fetal bovine
177 serum (FBS, Corning) and 1% v/v P/S. After 1.5 h at 37 °C in a cell culture incubator, the pronase
178 solution was discarded, and the tissue was washed three times with PBS. The digestion was continued
179 with 200 U/mL of collagenase II (Worthington) in DMEM HG for another 14 h under stirring. The
180 resulting cell suspension was sequentially filtered through 100 µm and 40 µm cell strainers (Falcon) to
181 remove debris. Following filtration, the suspension was centrifuged, the supernatant was discarded, and
182 the pellet was resuspended in fresh DMEM HG. The cells were then seeded at 5,000 cells/cm² for
183 expansion. After reaching the desired confluency, the cells were frozen in liquid nitrogen and later utilized
184 at passage 2 for encapsulation in hydrogels.

185 All hydrogel constructs were cultured in DMEM HG containing 10% v/v FBS and 1% v/v P/S. Cell
186 cultures were maintained at 37 °C with a humidified atmosphere of 5% CO₂ and media was changed
187 every third day.

188 2.7. Bioink formulation and hydrogel preparation

189 All dry bioink components were exposed to UV light for 2 h in a laminar flow cabinet to reduce
190 contamination risk. All liquid components were filtered through a 0.45 µm sterile filter. Collagen
191 fragmentation was performed in a laminar flow cabinet. The fragmented collagen was mixed with HA-Tyr
192 and gelatin microgels to achieve final concentrations of 20 mg/mL collagen, 30 mg/mL HA-Tyr, and 50
193 mg/mL gelatin (**Figure 1**). PBS at 10X concentration was supplemented to achieve a final 1X
194 concentration, accounting for the PBS added during collagen fragmentation. Stock solutions of sodium
195 persulfate (SPS, Sigma-Aldrich, ≥98%) and tris(bipyridine)ruthenium(II) chloride (Ru, Combi-Blocks,
196 99%) were prepared in ultrapure water and incorporated into the bioink at 5% v/v each, resulting in final
197 concentrations of 0.1 mM Ru and variable SPS concentrations of 1, 2, or 4 mM, depending on the desired
198 hydrogel stiffness. The concentrations were determined in a preliminary rheological experiment to cover a
199 range from healthy to moderately degenerated human nucleus pulposus. Cells were thawed, expanded,
200 detached, and resuspended in pure FBS at a concentration of 50 x 10⁶ cells/mL for encapsulation. This
201 cell suspension was added to the bioink to achieve final concentrations of 5 x 10⁶ cells/mL and 10% v/v
202 FBS. The final volume of the bioink was adjusted with ultrapure water as needed. Hydrogels were formed
203 by extruding the bioink into a silicone mold (Sylgard 184, Dow) featuring cylindrical wells, each 5 mm in

204 diameter and 1.3 mm in height. To ensure consistency in the amount of shear cells were exposed to,
 205 conditions were kept constant for all experiments, using a 22G tapered tip (Optimum, Nordson) and 50
 206 kPa of pressure. HA-Tyr crosslinking was initiated through exposure from both the top and bottom of the
 207 mold to a 450 nm LED light source (SST-10-B, Luminus) at an intensity of 5 mW/cm² for 3 min.

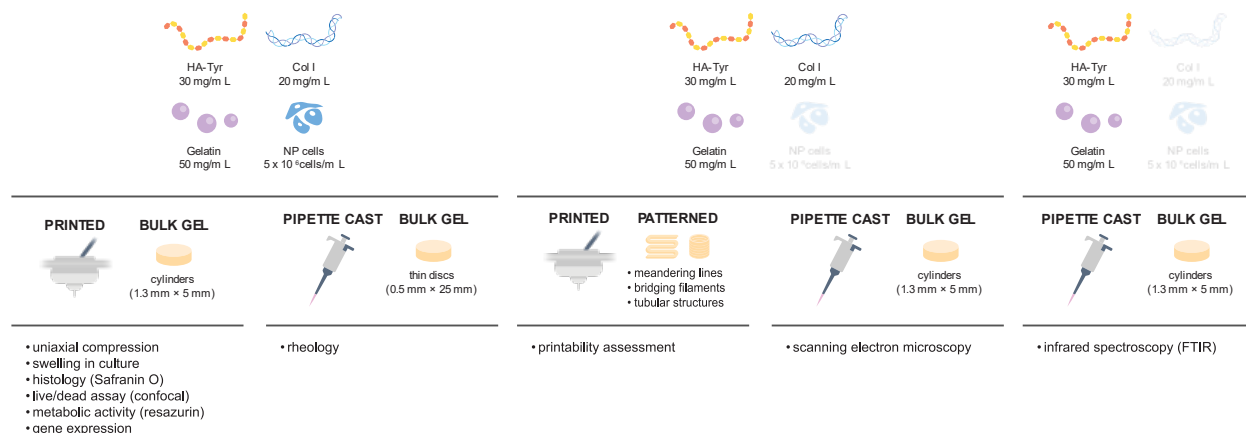


Figure 1. Overview of compositions and sample geometries. Combinations of tyramine-functionalized hyaluronic acid (HA-Tyr), collagen type I (Col I), and gelatin were tested both with and without encapsulated nucleus pulposus (NP) cells. The composites were either shaped into bulk gels, through extrusion with a 3D bioprinter or casting with a pipette, or patterned with a 3D bioprinter to assess printability.

208 2.8. Rheological testing

209 The evaluation of the bioink's viscoelastic properties was conducted using both shear oscillatory and
 210 rotational rheology (MCR 302, Anton Paar). The rheometer was equipped with a Peltier temperature
 211 control system, a parallel plate geometry with a glass lower plate, and a 450 nm LED light source (SST-
 212 10-B, Luminus). All experiments were carried out at 25 °C and a gap size of 0.5 mm. Oscillatory tests
 213 were performed using a fixed frequency of 1 Hz. An amplitude sweep between 0.1 and 100% shear strain
 214 was used to identify the linear viscoelastic region of the bioink. The limit of the linear viscoelastic region
 215 was determined as a 10% deviation from the initial plateau value. The flow point of the material, defined
 216 as the transition where the loss modulus exceeds the storage modulus, was determined. The bioink's
 217 elastic recovery was assessed by alternating between low strain within the linear viscoelastic region
 218 (0.1%) and high flow-inducing strain (100%). Mean storage moduli of the first and last low-strain cycles
 219 were compared as a measure of material deformation after repeated shearing. The bioink's viscosity was
 220 measured rotationally by altering the shear rate between 0.1 and 100 s⁻¹. Measured values were fit
 221 according to the Ostwald–de Waele relationship as $\eta = K\dot{\gamma}^{n-1}$, where η represents the viscosity, K the
 222 flow consistency index, $\dot{\gamma}$ the shear rate, and n the flow behavior index. The effect of light on gelation
 223 was examined in compositions with varying SPS concentrations of 1, 2, or 4 mM, exposing samples to 5
 224 mW/cm² of blue light for 3 min at a constant shear strain of 0.1% to assess the tunability of shear
 225 stiffness.

226 2.9. Compression testing

227 Hydrogel samples were incubated in DMEM HG at 37 °C for 24 h. After incubation, the dimensions of
 228 the swollen samples were measured using digital calipers (HeliosPreisser). Uniaxial compression tests
 229 were performed on an electrodynamic testing system (LTM1, ZwickRoell) equipped with a 50 N load cell

230 (U9C, HBM). The testing protocol consisted of an initial 0.01 N preload, followed by a stepwise stress-
231 relaxation test, wherein the samples were compressed in increments of 5% strain, up to a maximum of
232 25% strain, at a rate of 5%/s. A relaxation period of 15 min was allowed at each strain increment. Data
233 acquisition was performed at a sampling rate of 10 Hz. The stress relaxation data, obtained at each strain
234 level, were fit to a power law model (Cloyd et al., 2007) as $\sigma = A \exp(\beta\epsilon - 1)$, where σ and ϵ represent
235 the stress and strain, respectively, while A and β are material-specific constants. The linear region
236 equilibrium modulus, indicative of the material's stiffness, was subsequently calculated from the fitted
237 model as the derivative of stress with respect to strain at 10% compression. Stress relaxation at each strain
238 level was calculated as a proportion of the relaxed stress to the peak stress of the corresponding strain.

239 2.10. Scanning electron microscopy

240 Cell-free hydrogel samples were fixed in a 4% v/v solution of glutaraldehyde (Electron Microscopy
241 Sciences) at 4 °C for 24 h. Following fixation, they were washed three times in a buffer of 0.05 M sodium
242 cacodylate (Sigma-Aldrich) and 0.3 M sucrose (Sigma-Aldrich), adjusted to a pH of 7.4. After washing,
243 the samples were dehydrated through a series of ethanol baths, starting from 50% and increasing
244 incrementally to 100%, to ensure gradual removal of water. After dehydration, the samples were
245 processed using a supercritical CO₂ dryer (EM CPD300, Leica). For morphological analysis, the dried
246 samples were fractured to reveal their cross-sections. The fractured samples were mounted onto specimen
247 holders and sputter-coated with 15 nm of gold. Observations were carried out using a scanning electron
248 microscope (S-3400N, Hitachi) operating at 10 kV.

249 2.11. Fourier transform infrared spectroscopy (FTIR)

250 Hydrogels prepared using HA-Tyr and gelatin microgels were analyzed using attenuated total reflectance-
251 Fourier transform infrared spectroscopy (ATR-FTIR) to determine compositional changes in response to
252 gelatin's thermosensitive properties. Samples were prepared at three different degrees of crosslinking by
253 varying the SPS concentration (either 1, 2, or 4 mM) according to the steps outlined in section 2.7,
254 omitting collagen and cells to avoid confounding sample spectra. After crosslinking, the hydrogels were
255 incubated in PBS (pH 7.4) at 37 °C for either 0, 1, 3, or 7 days. This was followed by an overnight
256 freezing at -20 °C and subsequent freeze-drying. The dry polymer sponges were analyzed on a FT/IR 4X
257 (Jasco) spectrometer. Transmittance spectra were recorded between 4000 and 400 cm⁻¹ with a resolution
258 of 4 cm⁻¹ and 32 scans. The atmospheric background was subtracted, baseline corrected, and a 5-point
259 moving average smoothing performed. The spectra for each sodium persulfate/timepoint combination
260 were averaged and normalized to the 1035 cm⁻¹ peak, which was attributed to the glycosidic linkages in
261 the backbone of the hyaluronic acid (Nikonenko et al., 2000).

262 2.12. Printability

263 Printability was evaluated on a pneumatic extrusion bioprinter (3DDiscovery, REGENHU). Bioinks were
264 loaded into a 3 mL opaque black barrel (Optimum, Nordson), fitted with a 22G tapered tip (SmoothFlow,
265 Nordson) and a standard piston (White SmoothFlow, Nordson). The printing was performed at room
266 temperature, with an extrusion pressure of 50 kPa.

267 A series of simple patterns were extruded to evaluate the bioink printability quantitatively. The structures
268 were crosslinked at each extruded layer by 5 s of exposure to blue light at an intensity of 5 mW/cm².

269 Filament uniformity was first determined by extruding a single filament of length 20 mm on an etched
270 glass slide. The pattern was immediately imaged using a camera and later analyzed using ImageJ. The
271 diameter of the filament was measured at three equally spaced points.

272 To assess filament fusion, a continuous meandering pattern with gradually decreasing line spacing was
273 printed (Ribeiro et al., 2017). The closest two lines (1 mm) were used to determine the fused segment
274 length (L_F). This fused segment is observed at the transition from one line to the other, where the printing
275 direction is changed, and propagates from the corners as a function of surface tension and yield stress
276 (Schwab et al., 2020).

277 Next, a 10 mm by 10 mm square with a 3 x 3 grid was produced. The geometry of the internal
278 macropores was evaluated by measuring their perimeter (L) and area (A). A printability index (C ,
279 O'Connell et al., 2020) was determined as $P = L^2/16A$ with $P = 1$ indicating perfectly square
280 macropores, $P < 1$ overly circular macropores, and $P > 1$ an irregular macropore shape.

281 To evaluate the bioink's capability of bridging gaps, single filaments were extruded across a steel pillar
282 array with increasing gap distances of 2, 4, and 8 mm (Ribeiro et al., 2017). The filaments were assessed
283 by the amount of sagging at each of the three gap sizes by measuring their deflection angle (θ) from the
284 start of the gap to its middle, relative to the printing surface. Minimizing such sagging is crucial in
285 preserving vertical macroporosity.

286 Finally, a qualitative evaluation of the bioink's potential for 3D multi-layered structure fabrication was
287 performed by producing simple cylindrical structures.

288 2.13. Cell viability

289 The viability of bovine nucleus pulposus cells encapsulated in hydrogels was assessed using confocal
290 microscopy. Samples were imaged after printing and at subsequent time points up to 28 days of culture.
291 At each time point, samples were removed from the culture medium and washed for 20 min in serum-free
292 DMEM LG (low glucose, PAN-Biotech) supplemented with 20 mM HEPES buffer (Gibco). Next, the
293 samples were immersed in a fresh solution of DMEM supplemented with 4 $\mu\text{g}/\text{mL}$ each of calcein-AM
294 (Invitrogen) and ethidium homodimer-1 (Sigma-Aldrich) to stain the live and dead cells, and 8 $\mu\text{g}/\text{mL}$
295 Hoechst 33342 (Sigma-Aldrich) to stain the nuclei of the cells. This staining was followed by another
296 washing step. The washing and staining procedures were performed on a microplate shaker at 350 rpm
297 and 37 °C. Following the second washing, the samples were imaged using a confocal microscope (LSM
298 800, Zeiss). Z-stacks with a thickness of 200 μm were acquired at three representative regions of each
299 sample using a 10x/0.45 objective (Plan-Apochromat, Zeiss).

300 2.14. Cell metabolic activity

301 The metabolic activity of bovine nucleus pulposus cells encapsulated in hydrogels was quantified using
302 the CellTiter-Blue assay (Promega). Metabolic activity was measured after printing and at subsequent
303 time points up to 28 days of culture. At each time point, the samples were transferred into a solution
304 comprised of fresh culture medium and CellTiter-Blue reagent in a 5:1 ratio. The samples were incubated
305 for 4 h at 37 °C on a shaker set to 350 rpm. After incubation, 100 μl aliquots of the culture medium were
306 analyzed in duplicates using a microplate reader (Infinite M Plex, TECAN). Fluorescence was measured
307 using an excitation wavelength of 560 nm and an emission wavelength of 590 nm. Cell-free hydrogels
308 with identical composition and crosslinking density served as blanks. The gain value was established
309 based on the signal obtained on day 0 and used across all subsequent time points.

310 2.15. Safranin-O/Fast Green staining

311 Samples were collected at days 0 and 28 of cell culture. They were removed from the medium and
312 immediately placed into a 4% formaldehyde solution for 16 h at room temperature. This was followed by

313 immersion in 15% and 30% sucrose solutions for 3 h each, which was determined as sufficient for the
314 samples to sink. After equilibration in the sucrose solution, the samples were embedded in Tissue
315 Freezing Medium (Leica Biosystems) and snap-frozen in isopentane, cooled to approximately -160 °C
316 with liquid nitrogen.

317 Next, the samples were cut into 10 µm sections using a cryostat (CryoStar NX70, Thermo Fisher
318 Scientific). The Tissue Freezing Medium was removed, and the sections were rehydrated by rinsing in
319 distilled water for 10 min. The slides were first stained in Weigert's Iron Hematoxylin Solution (Sigma-
320 Aldrich) for 10 min in the dark, rinsed in lukewarm tap water for 10 min, and then rinsed in distilled
321 water for another 10 min.

322 The sections were then immersed in 0.02% Fast Green FCF (C.I. 42053, Sigma-Aldrich) for 6 min
323 followed by a 1% acetic acid solution for 20 s, and finally in Safranin O (C.I. 50240, Sigma-Aldrich) for
324 12 min. The slides were differentiated first in 70% ethanol until the red color was visible and the
325 black/blue colors faded (10 s to 6 min), and subsequently in 96% ethanol for 2-30 s. After dehydration in
326 absolute ethanol (2 x 4 min), the slides were immersed in xylene (2 x 2 min) and mounted with a cover
327 slip using Eukitt Quick-hardening mounting medium (Sigma-Aldrich). Brightfield images of the sections
328 were acquired with an Axioscan 7 slide scanner using a 20x/0.8 objective (Zeiss).

329 *2.16. Measure of swelling in culture*

330 The swelling and degradation properties of hydrogels were measured with and without encapsulated
331 bovine nucleus pulposus cells by varying the SPS concentration to achieve different degrees of
332 crosslinking. Samples were tracked throughout a 28-day culture period. At each time point, samples were
333 removed from the culture medium and gently wiped against a sterile glass slide to remove any excess
334 liquid. Subsequently, their mass was measured on a laboratory scale (ME103T/00, Mettler Toledo) under
335 sterile conditions, and they were imaged with a camera (ArduCam). Sample volumes were estimated
336 using ImageJ measurements of height and diameter.

337 *2.17. RNA isolation and RT-qPCR*

338 The gene expression of bovine nucleus pulposus cells cultured in hydrogels with varying degrees of
339 crosslinking was measured over a period of 28 days. Samples were collected at each time point by flash-
340 freezing in liquid nitrogen and subsequently stored at -70 °C for further processing.

341 To isolate the RNA, the samples were transferred into TRI Reagent (MRC) at 4 °C while still frozen. For
342 homogenization, a 6 mm stainless steel ball was added to each sample. Then, the samples were processed
343 in a ball mill (MM 400, Retsch) at 30 Hz for 3 min. The supernatant was collected from the homogenate
344 and mixed with 1-bromo-3-chloropropane (BCP, Thermo Scientific) to trigger phase separation. The
345 aqueous phase was then collected and further purified using columns (RNeasy Mini, Qiagen) according to
346 the manufacturer's recommendations. Final elution was performed with RNase-free water, and the RNA
347 purity and concentration were evaluated using a spectrophotometer (NanoDrop One, Thermo Fisher).
348 RNA integrity was assessed, yielding sample RIN values above 9.0 (Bioanalyzer, Agilent). RNA was
349 stored at -70 °C for future use.

350 The isolated RNA was transcribed to complementary DNA (SuperScript VILO cDNA Synthesis Kit,
351 Invitrogen) following the manufacturer's recommendations. Subsequently, the relative gene expression
352 levels were determined using a real-time PCR system (QuantStudio 6 Pro, Applied Biosystems) with
353 TaqMan probes (TaqMan Universal PCR Master Mix, Applied Biosystems). A geometric mean of

354 GAPDH and RPLP0 was used as an internal control for normalizing RNA levels. The complete sequences
 355 and catalog numbers of all primers and probes used are listed in **Table 1**.

Table 1. Bovine oligonucleotide primers and probes used for RT-qPCR. Primers and probes with sequences were custom-designed and produced by Microsynth; primers and probes with catalogue numbers were purchased from Applied Biosystems.

Gene	Type	Sequence/catalog number
COL1A2	Primer fw (5'-3')	TGC AGT AAC TTC GTG CCT AGC A
	Primer rev (5'-3')	CGC GTG GTC CTC TAT CTC CA
	Probe (5'FAM/3'TAMRA)	CAT GCC AAT CCT TAC AAG AGG CAA CTG C
COL2A1	Primer fw (5'-3')	AAG AAA CAC ATC TGG TTT GGA GAA A
	Primer rev (5'-3')	TGG GAG CCA GGT TGT CAT C
	Probe (5'FAM/3'TAMRA)	CAA CGG TGG CTT CCA CTT CAG CTA TGG
ACAN	Primer fw (5'-3')	CCA ACG AAA CCT ATG ACG TGT ACT
	Primer rev (5'-3')	GCA CTC GTT GGC TGC CTC
	Probe (5'FAM/3'TAMRA)	ATG TTG CAT AGA AGA CCT CGC CCT CCA T
FOXF1	Primer fw (5'-3')	TGC ACC AGA ACA GTC ACA ACA C
	Primer rev (5'-3')	GAC GAA CTC CTT TCG GTC ACA
	Probe (5'FAM/3'TAMRA)	CTG CAA GGC ATC CCG CGG TAT C
SOX9	Primer fw (5'-3')	AAC GCC GAG CTC AGC AAG
	Primer rev (5'-3')	ACG AAC GGC CGC TTC TC
	Probe (5'FAM/3'TAMRA)	TTC AGC AGT CTC CAG AGC TTG CCC A
CDH2	Assay on demand	Bt04298958 m1
GAPDH	Primer fw (5'-3')	GGC TGC TTT TAA TTC TGG CAA A
	Primer rev (5'-3')	AAT CAT ACT GGA ACA TGT AGA CCA TGT A
	Probe (5'FAM/3'TAMRA)	TGG ACA TCG TCG CCA TCA ATG ACC
RPLP0	Assay on demand	Bt03218086 m1

356 2.18. Statistical analysis

357 All measurements are presented as means of three replicates and standard deviation SD. The normal
 358 distribution of measured values was assessed using the Shapiro-Wilk test. Statistical comparison in the
 359 rheological test of elastic recovery was performed with a repeated measures t-test between sequential
 360 cycles, followed by a Bonferroni correction to account for multiple comparisons. Due to the small sample
 361 sizes and challenges in meeting the normality assumptions of parametric tests, the more robust Kruskal-
 362 Wallis test was used for comparisons between groups of different crosslinking densities. This was
 363 followed by Dunn's test with the Bonferroni correction for post-hoc paired comparisons. Notably, in
 364 analyzing the metabolic activity of embedded cells, the statistical comparison was first performed with
 365 timepoint as an independent factor, then separately with concentration. In-depth pairwise comparisons
 366 were not performed due to poor statistical power and the conservative nature of the utilized tests. This
 367 similarly applies to the gene expression and swelling experiments. All statistical analyses were performed
 368 in Python using the scipy (Virtanen et al., 2020) and statsmodels (Seabold & Perktold, 2010) libraries.

369 3. Results

370 3.1. Viscoelastic properties of the bioink

371 The HA/collagen/gelatin bioink was evaluated rheologically to determine its suitability for printing, and
372 compared to gelatin-free compositions to evaluate gelatin microgels' effectiveness as a rheological
373 enhancer. Before photocrosslinking, the cell-laden composition displayed elastic prevalence ($G' > G''$),
374 with a storage modulus of approximately 800 Pa in all formulations. The modulus was reduced by almost
375 an entire order of magnitude to approximately 90 Pa when gelatin was absent. These properties were
376 consistent even in the absence of cells (data not shown). An amplitude sweep was performed to determine
377 the limit of the linear viscoelastic region (**Figure 2a**), which was observed at a mean of 2.7% shear strain
378 (SD = 0.3%), corresponding to a shear stress of only 26 Pa (SD = 3 Pa). Flow in the material occurred
379 early at a low stress of only 74 Pa (SD = 7 Pa), corresponding to a shear strain of 11% (SD = 1%). These
380 values were reduced considerably in the absence of gelatin, with a limit of the linear viscoelastic region at
381 1.3% (SD < 0.1%), corresponding to a shear stress of only 1 Pa (SD = 0.1 Pa), and a flow point at 7% (SD
382 = 1%), with a shear stress of 4 Pa (SD = 0.7 Pa).

383 Next, the material's elastic recovery was evaluated. When subjected to high shear, the bioink displays a
384 prevalence of loss over the storage modulus, indicative of good flow when pushed through the nozzle.
385 Upon cessation of the shear stimulus, storage modulus prevalence over the loss modulus was re-
386 established, which is indicative of shape retention after extrusion. However, under our experimental
387 conditions, a drop in storage modulus was observed after high shear, with a change of 19% (SD = 3%)
388 after three high shear cycles (**Figure 2b**). Most of this change occurred during the first cycle (15%, SD =
389 3%). All changes in storage moduli between sequential cycles were deemed statistically significant ($p <$
390 0.020). When gelatin microgels were removed from the composite, the samples could no longer recover
391 elastically. With a decrease of 58% (SD = 1%) in the storage modulus, dropping it below the loss
392 modulus, a predominantly liquid-like behavior was indicated even after shear was reduced.

393 In a rotational measurement of viscosity, the bioink demonstrated shear-thinning behavior (**Figure 2c**).
394 After fitting the data to a power-law fluid model, a mean flow behavior index of 0.45 ($R^2 = 0.999$) was
395 established. This was comparable in gelatin-free samples with an index of 0.47 ($R^2 = 0.986$).

396 Formulations with varying concentrations of sodium persulfate were exposed to blue light while
397 undergoing a time sweep test. Starting from a mean storage modulus of 800 Pa (SD = 100 Pa), 3 min of
398 light exposure resulted in a modulus increase to 7, 11, and 17 kPa (SD = 0.2, 0.3, and 0.6 kPa),
399 respectively, for SPS concentrations of 1, 2, and 4 mM (**Figure 2d**). Only the difference between 1 mM
400 and 4 mM SPS was deemed statistically significant ($p = 0.022$). However, there was a strong linear
401 correlation between the SPS concentration and storage modulus ($R^2 = 0.990$, $p < 0.001$). All formulations
402 showed a rapid increase in storage modulus immediately after light exposure, reaching half the final
403 storage modulus within 43 s (SD = 5 s).

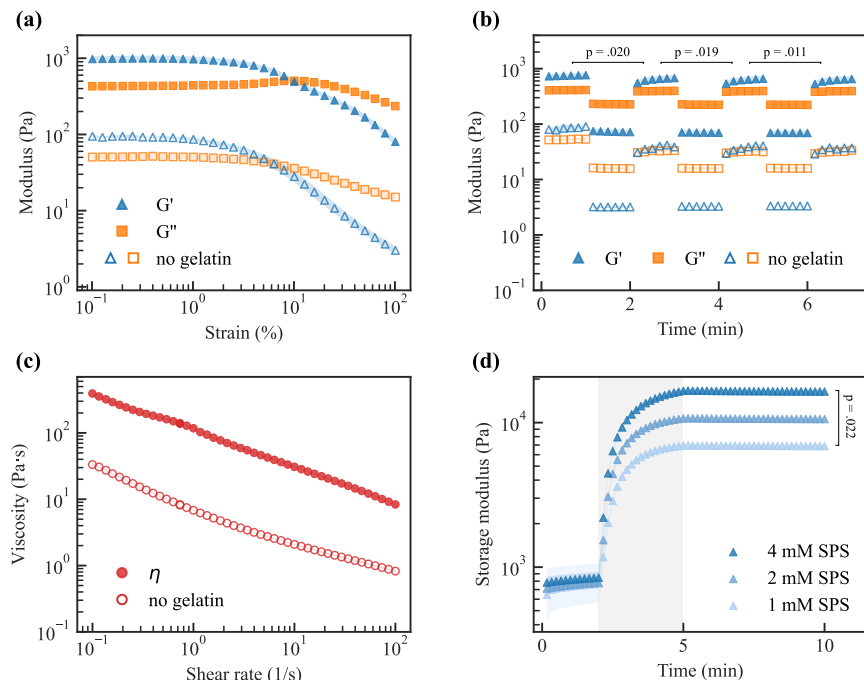


Figure 2. Rheological characterization of the NP bioink laden with 5×10^6 bovine NP cells/mL. Presented as means of three technical replicates, with bands indicating the standard deviation. Shear properties prior to crosslinking (a-c) were compared with gelatin-free controls. (a) An amplitude sweep from 0.1% to 100% shear strain reveals a short linear viscoelastic range and an early flow point, indicating that low shear stresses are required during printing. The presence of gelatin results in considerable higher storage and loss moduli. (b) Upon repeated cycling between low (0.1%) and high (100%) shear strain, the bioink exhibits a permanent drop in storage modulus. However, it flows under high shear and retains its elasticity when the shear is reduced. This elastic recovery is compromised in the absence of gelatin. Significant differences determined with repeated measures t-tests indicated. (c) The bioink's viscosity decreases with an increase in shear rate from 0.1 to 100 1/s, indicating shear-thinning behavior. Gelatin contributes to an overall higher viscosity in the composites. (d) The material's shear stiffness can be tuned through exposure to blue light (shaded area). By varying the concentration of sodium persulfate (SPS), moduli ranging from 7 to 17 kPa can be achieved. Significant difference determined with Dunn's test indicated.

404 3.2. Hydrogel compressive mechanics

405 Uniaxial compression was performed on hydrogels crosslinked to varying degrees of stiffness. The
 406 relaxation data fit well to a power law model, with R^2 of 0.992, 0.990, and 0.982 for the 1 mM, 2 mM,
 407 and 4 mM SPS formulations, respectively. The equilibrium moduli (i.e., determined after equilibration in
 408 stepwise compression) demonstrated a strong dependence on crosslinking density and were tunable
 409 between 4 kPa (SD = 1 kPa) for the softest hydrogels to 23 kPa (SD = 6 kPa) for the stiffest hydrogels
 410 (**Figure 3a**). The difference in moduli between 1 mM and 4 mM SPS were deemed significant ($p =$
 411 0.022), with a moderately strong linear correlation between the concentration and equilibrium modulus
 412 ($R^2 = 0.838$, $p < 0.001$).

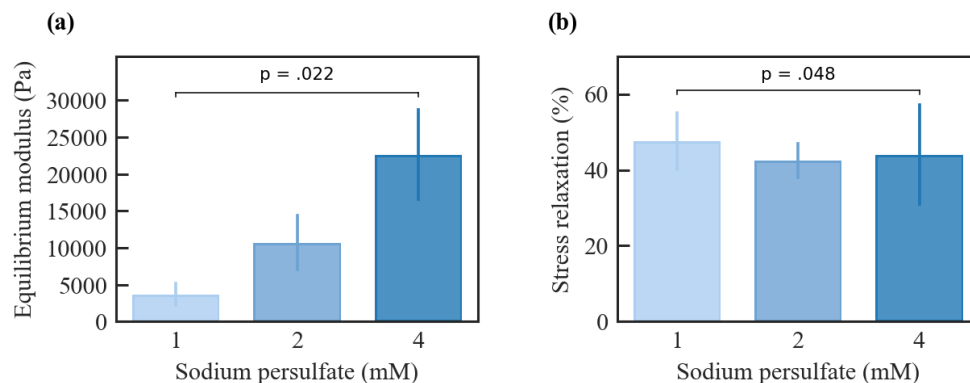


Figure 3. Uniaxial compression of hydrogels laden with 5×10^6 bovine NP cells/ml. Presented as means of three technical replicates, with error bars indicating the standard deviation. Significant differences determined with Dunn's test indicated. (a) The equilibrium modulus can be tuned between 4 and 23 kPa by varying the sodium persulfate concentration between 1 and 4 mM, as a result of modulating the crosslinking density. (b) Despite varying hydrogel stiffness, stress relaxation behavior remains consistent across all sodium persulfate concentrations, at approximately 40% relaxation from the peak.

413 Sample relaxation was consistent across strain levels and crosslinking densities, at more than 40%
 414 relaxation from peak stress after a 15 min relaxation period (**Figure 3b**). As with the modulus, a
 415 statistically significant difference was only observed between the extreme concentrations ($p = 0.048$), and
 416 there appeared to be no correlation between the concentration and degree of relaxation.

417 3.3. Morphology of the hydrogels

418 Using light and electron microscopy, we evaluated the morphological features of the hydrogels. Phase
 419 contrast imaging followed by software analysis of hydrated gelatin microgels revealed a Feret diameter
 420 distribution (**Figure 4a**) centered at $51 \mu\text{m}$ ($\text{SD} = 7 \mu\text{m}$).

421 Scanning electron microscopy of fractured hydrogels revealed morphological features of all three
 422 components, with an evident collagen fibril network scattered across sheet-like structures of HA-Tyr
 423 (**Figure 4b**). Additionally, gelatin microgels were embedded throughout the hydrogel (**Figure 4c**). The
 424 morphologies remained consistent across all crosslinking densities.

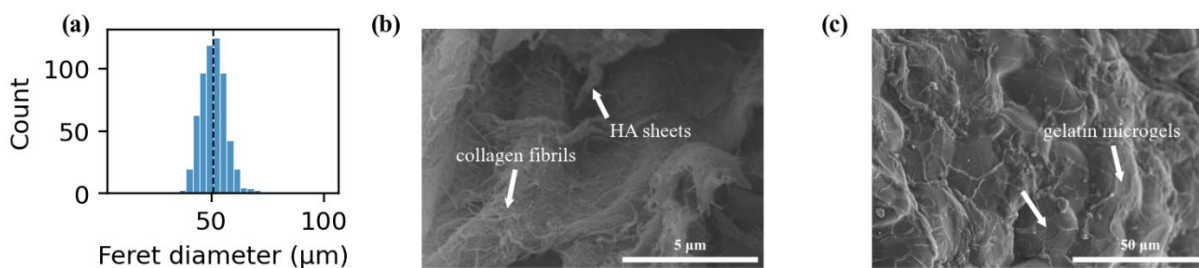


Figure 4. (a) Measurements of the gelatin microgel Feret diameter from phase-contrast microscopy images resulted in a distribution with a mean of $51 \mu\text{m}$. A total of 610 representative microgels used for the measurement. (b-c) Representative scanning electron microscopy images of dried and fractured hydrogels prior to cell culture, reveal the presence of all three major morphological features: collagen fibril networks and HA-Tyr sheet-like structures in a 4 mM SPS sample (b), and gelatin microgels embedded in a 2 mM SPS sample (c). Scale bars represent $5 \mu\text{m}$ and $50 \mu\text{m}$, respectively.

425 3.4. Compositional changes

426 We evaluated the change in HA-Tyr/gelatin microgel composites over the course of 7 days in PBS at 37
427 °C via ATR-FTIR analysis. Spectra were normalized to the 1035 cm^{-1} peak, which can be attributed to C-
428 O stretching vibrations. Since gelatin and HA-Tyr bear similar functional groups (amide, hydroxy,
429 functional groups in tyramine moieties) spectra do now show obvious distinctive features of one
430 component or the other. A notable trend was detectable at 860 cm^{-1} (**Figure 5**), where at the lowest degree
431 of crosslinking (**Figure 5a**), the formation of a peak at day 1 was seen, with further increase at day 3,
432 when a plateau was reached. In the medium crosslinked hydrogels (**Figure 5b**), changes were only
433 observed starting at day 3. No obvious increases in the 860 cm^{-1} peak were observed at the highest degree
434 of crosslinking or in the absence of gelatin (**Figure 5c-d**). The peak at 860 cm^{-1} may be attributed to
435 aromatic compounds, such as tyramine and tyrosine, both present in our samples. An increase at 860 cm^{-1}
436 associated with a decrease in other characteristic wavelengths (e.g., 3200-3400 cm^{-1}) may indicate gradual
437 degradation or rearrangement of the polymers. Given that this increase was not observed in the absence of
438 gelatin, we attributed this change to the gradual degradation of gelatin and subsequent exposure of the
439 tyramine grafted on the hyaluronic acid.

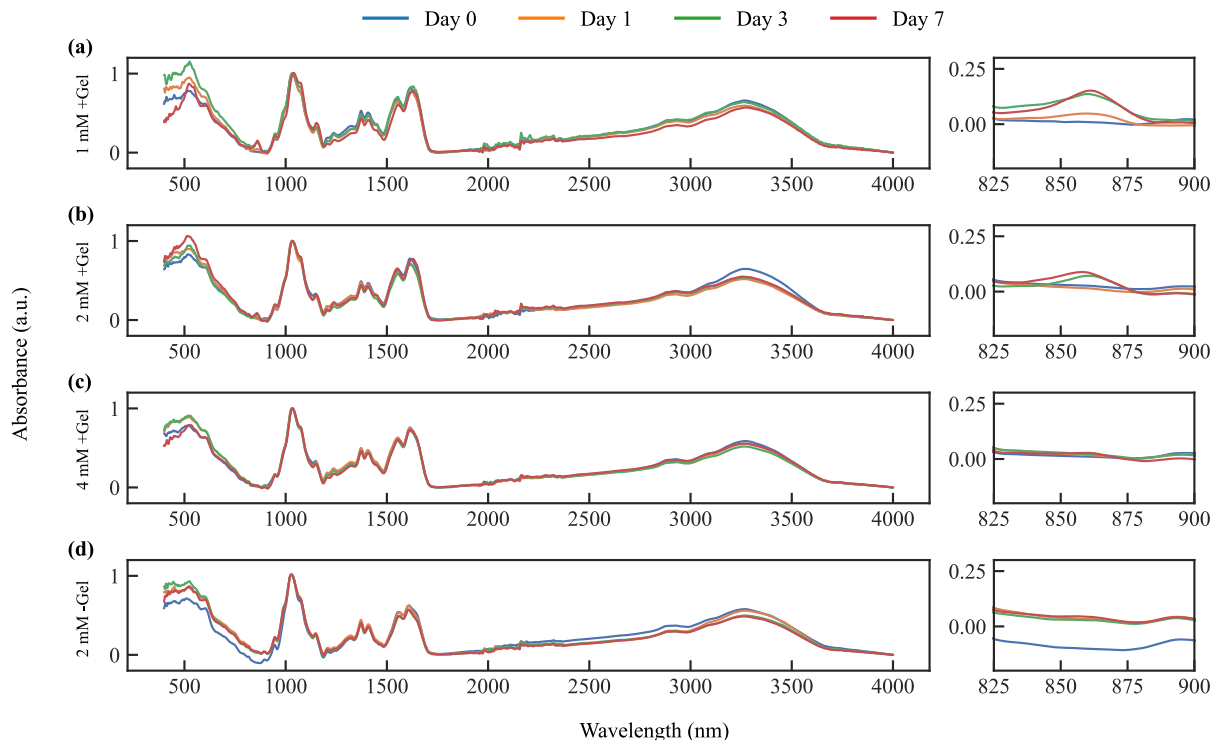


Figure 5. ATR-FTIR absorbance spectra of HA-Tyr/gelatin microgel samples prepared at different degrees of crosslinking. Panels on the right match the spectra on the left, zoomed in to the 825-900 cm^{-1} region. Presented as means of three technical replicates. (a) When samples are weakly crosslinked, an increase of the peak at 860 cm^{-1} corresponding to the length of incubation at 37 °C is observed. (b) The changes are less pronounced when the degree of crosslinking is increased, and only observed after 3 days of incubation. (c) No clear changes are observed in the strongly crosslinked samples. (d) Furthermore, there is no 860 cm^{-1} peak formation in the absence of gelatin.

440 **3.5. Printability**

441 We evaluated the bioink printability using a series of extrusion patterns. Straight extruded filaments were
442 uniform with a mean diameter of 445 μm (SD = 25 μm). Filaments extruded 1 mm apart demonstrated a
443 fused segment length of 0.9 mm (SD = 0.1 mm, **Figure 6a**). In producing grid structures, a printability
444 index of 0.91 (SD = 0.02, **Figure 6b**) was achieved, indicating only a slight rounding of the square
445 macropores. All gaps were bridged successfully in the pillar array test, with no sagging observed at 2 and
446 4 mm, and a 2.6° (SD = 0.4°, **Figure 6c**) deflection angle at 8 mm. Cylinders with a height of 7.5 mm and
447 a single layer wall thickness were readily produced (**Figure 6d**).

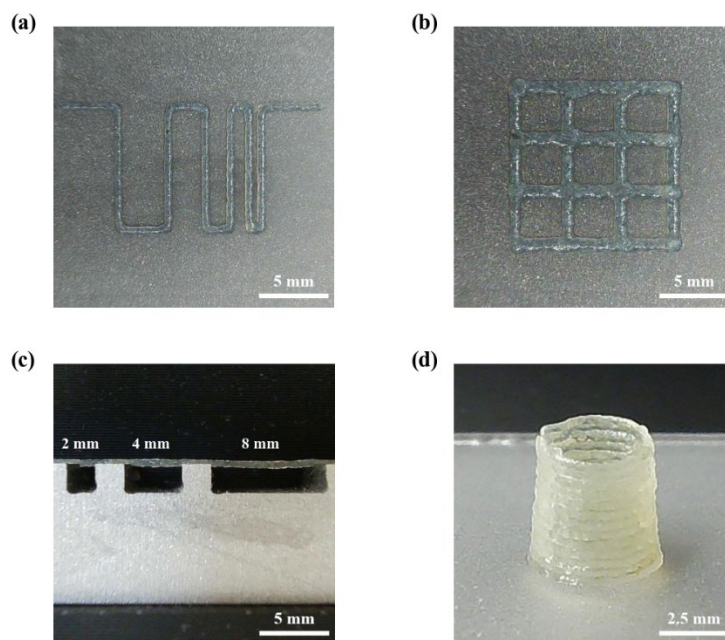


Figure 6. Representative samples of the bioink printability tests. (a) A spacing of slightly more than double the nozzle diameter (22G, 1 mm) is achieved without fusion, with good filament uniformity. (b) Square macropores are maintained in grid-like structures, with only minor rounding where perpendicular filaments meet. (c) The filament bridges gaps of up to 8 mm with minimal sagging. (d) Multi-layered constructs can be readily achieved. Depicted is a hollow cylinder approximately 7.5 mm tall.

448 **3.6. Cell viability**

449 The viability of nucleus pulposus cells embedded in hydrogels was assessed throughout 28 days of culture
450 using confocal fluorescent microscopy. On the day of printing, cells maintained excellent viability in all
451 three hydrogel stiffness compositions, exhibiting a round morphology (**Figure 7**, Day 0). During the first
452 week of culture, cells in the softer hydrogels formulated with 1 mM and 2 mM SPS began to proliferate
453 and exhibited spherical clustering, corresponding to the shape and size of gelatin microgels (**Figure 7**,
454 Day 7). Cells in the stiff (4 mM SPS) formulation remained similarly distributed as after printing and did
455 not exhibit such clustering, instead only showing minor spreading compared to earlier timepoints. By the
456 end of the 28-day culture period, all three formulations exhibited higher cell densities indicative of
457 proliferation. In softer hydrogels (1 and 2 mM SPS), large cell clusters formed, exhibiting an elongated
458 morphology (**Figure 7**, Day 28). Cells in the stiffer hydrogel (4 mM SPS) demonstrated spherical
459 clustering similar to the softer gels after the first week of culture. The cells remained viable throughout
460 the entire culture period across all formulations.

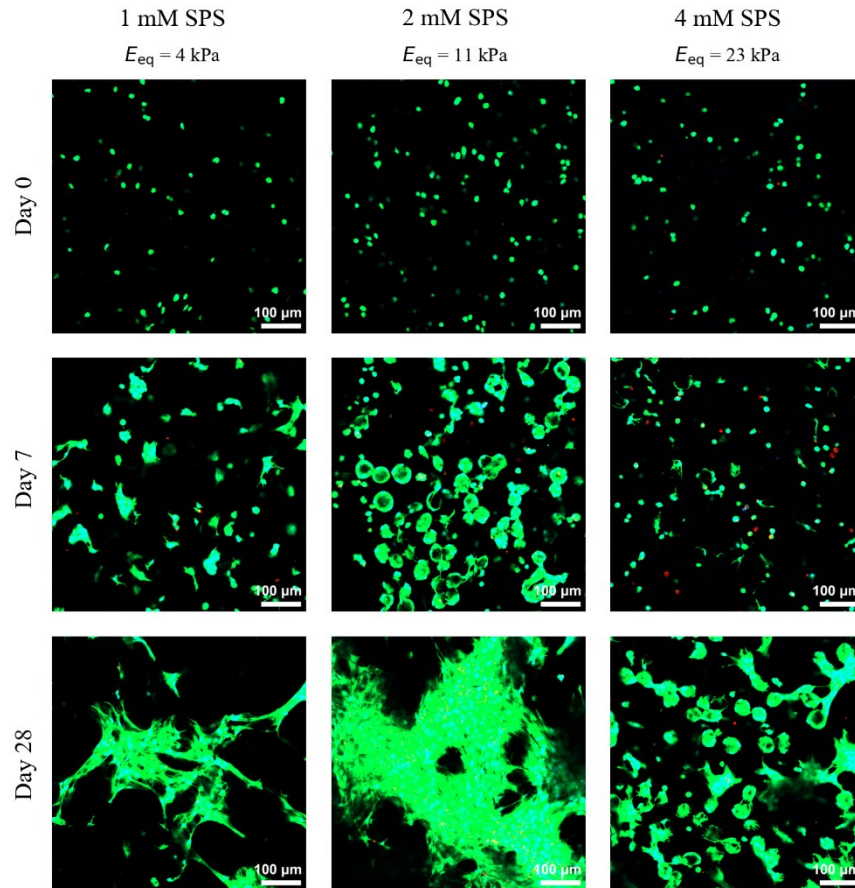


Figure 7. Confocal fluorescent characterization of nucleus pulposus cell viability in hydrogel-embedded cultures. Presented as representative slices out of 200 μm imaged stacks. Live cell cytoplasm was stained with calcein-AM (green), dead cell nuclei were stained with ethidium homodimer-1 (red), and all cell nuclei were stained with Hoechst 33342 (blue) to facilitate the identification of individual cells at high cell densities. Excellent cell viability was observed in the hydrogels on the day of printing and maintained up to 28 days of culture, with cell proliferation evident in all formulations. In softer hydrogels crosslinked with 1 and 2 mM of SPS, early clustering of cells in spherical structures resembling gelatin microgels was observed, suggesting infiltration of the hydrogel's porous structure. These structures were later replaced by larger clusters of elongated cells. The stiffer hydrogels crosslinked with 4 mM SPS exhibited a delay in cell clustering; however, these clusters were maintained by day 28 of culture. Scale bars represent a length of 100 μm .

461 3.7. *Metabolic activity*

462 The metabolic activity of encapsulated nucleus pulposus cells was quantified throughout 28 days of
 463 culture using the resazurin-based assay CellTiter-Blue. The measured values are consistent with
 464 fluorescent microscopy observations, showing that metabolic activity gradually increased over the 28
 465 days in all formulations ($p < 0.001$, **Figure 8**). Despite starting with the same cell densities, the activity in
 466 softer hydrogels (1 and 2 mM SPS) was nearly double that observed in the stiffer hydrogels (4 mM SPS),
 467 with a statistically significant difference in metabolic activity across concentrations ($p = 0.048$). Notably,
 468 there was a considerable increase in metabolic activity between days 3 and 7 of culture across all
 469 formulations, although a pairwise statistical comparison of groups was not performed.

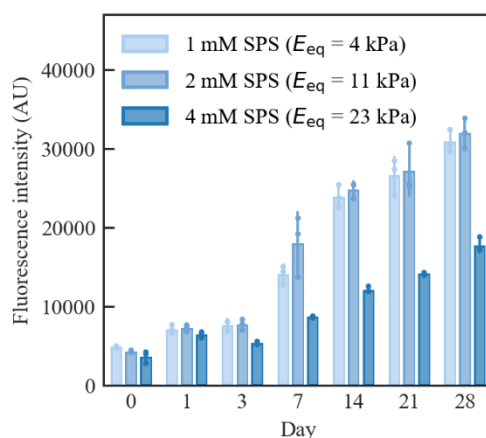


Figure 8. Quantification of the metabolic activity during 28 days of culture of nucleus pulposus cells embedded in hydrogels. Presented as means of three biological and two technical replicates each, with error bars indicating the standard deviation. The CellTiter-Blue assay demonstrated an increase in metabolic activity throughout the culture period for all formulations ($p < 0.001$). Differences in metabolic activity across concentrations were statistically significant ($p = 0.048$), with the softer hydrogels (1 and 2 mM SPS) exhibiting nearly double the activity of the stiffer hydrogel (4 mM SPS).

470 3.8. Matrix production

471 Composites with encapsulated bovine NP cells were evaluated for their capacity to support new matrix
 472 production. Safranin-O staining of glycosaminoglycans was strong on day 0 due to the high concentration
 473 of hyaluronic acid in the samples (**Figure 9**). A microporous HA-Tyr network with gelatin microgels
 474 (**Figure 9A**) and collagen fragments (**Figure 9B**) embedded was visible at all three compositions (1 mM,
 475 2 mM, and 4 mM SPS), with no obvious differences dependent on the degree of crosslinking. At day 28,
 476 staining intensity varied considerably between groups. The least crosslinked samples exhibited a weak
 477 matrix with poor Safranin-O staining. An improved matrix integrity was observed in the 2 mM SPS
 478 group, though with no conclusive evidence of new matrix production. The strongly crosslinked group (4
 479 mM SPS) exhibited a biphasic matrix. Large regions of dense matrix were visible in the samples (**Figure**
 480 **9C**), separated by regions resembling the hyaluronic acid network on day 0 (**Figure 9D**). Presence of
 481 gelatin microgels could not be clearly established in any of the groups by day 28.

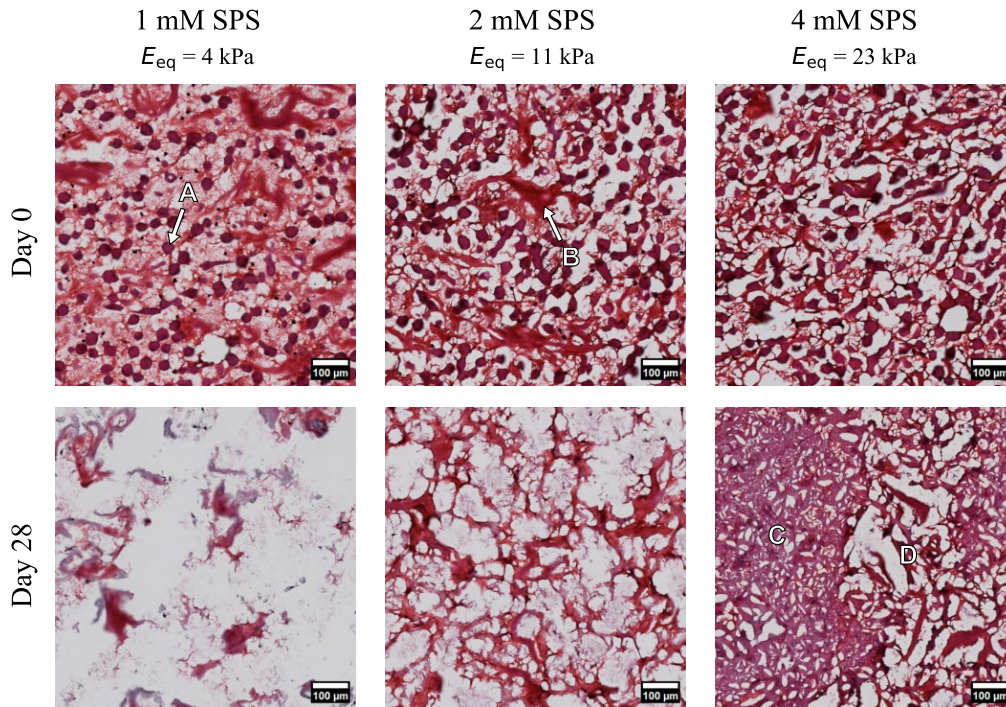


Figure 9. Representative images of NP cell-laden hydrogels stained with Safranin-O at days 0 and 28 of culture. Embedded gelatin microgels (A) and collagen fragments (B) are clearly visible at day 0 in all three compositions (1, 2, and 4 mM SPS). At day 28, the softer hydrogels exhibited a weak matrix with little or no new matrix production. In contrast, the stiffest hydrogels exhibited dense regions of nascent matrix (C) dispersed throughout the well-preserved hyaluronic acid structure (D).

482 3.9. Gene expression

483 The expression of nucleus pulposus markers was quantified using RT-qPCR (**Figure 10**). The selection of
 484 markers was based on their potential to differentiate between the nucleus pulposus, annulus fibrosus, and
 485 articular cartilage (Lv et al., 2014). Common anabolic markers such as COL1A2, COL2A1, and ACAN
 486 were selected due to the compositional differences of collagen type I and II, and aggrecan in the three
 487 tissue matrices. FOXF1 was selected due to its strong expression in the nucleus pulposus compared to the
 488 annulus fibrosus. SOX9 was similarly selected as a standard chondrogenic marker, and CDH2 as
 489 specifically a clear marker of healthy nucleus pulposus.

490 COL1A2 was upregulated significantly in the softest hydrogels and only slightly in the stiffer hydrogels,
 491 where it peaked at day 21 of culture. In contrast, COL2A1 was slightly downregulated in all three
 492 formulations. ACAN was somewhat upregulated in all formulations: in the 1 mM and 2 mM SPS
 493 formulations, the expression was transient with a peak at day 3, whereas the expression in the stiffer 4
 494 mM peaked at day 21 and was sustained throughout the entire culture period.

495 Both FOXF1 and SOX9 remained consistent throughout 28 days of culture and did not appear to be
 496 affected by variations in material stiffness, whereas CDH2 was upregulated in all three formulations in
 497 the early days of culture and then decreased in the softer hydrogels (1 mM and 2 mM SPS). High levels of
 498 CDH2 expression were measured in the stiffer hydrogels (4 mM SPS) throughout all 28 days of culture.
 499 Significant differences in expression over time were observed for all genes ($p < 0.016$), whereas the
 500 differences across concentrations were significant only for COL1A2 ($p = 0.002$), FOXF1 ($p < 0.001$),
 501 SOX9 ($p < 0.001$), and CDH2 ($p < 0.001$).

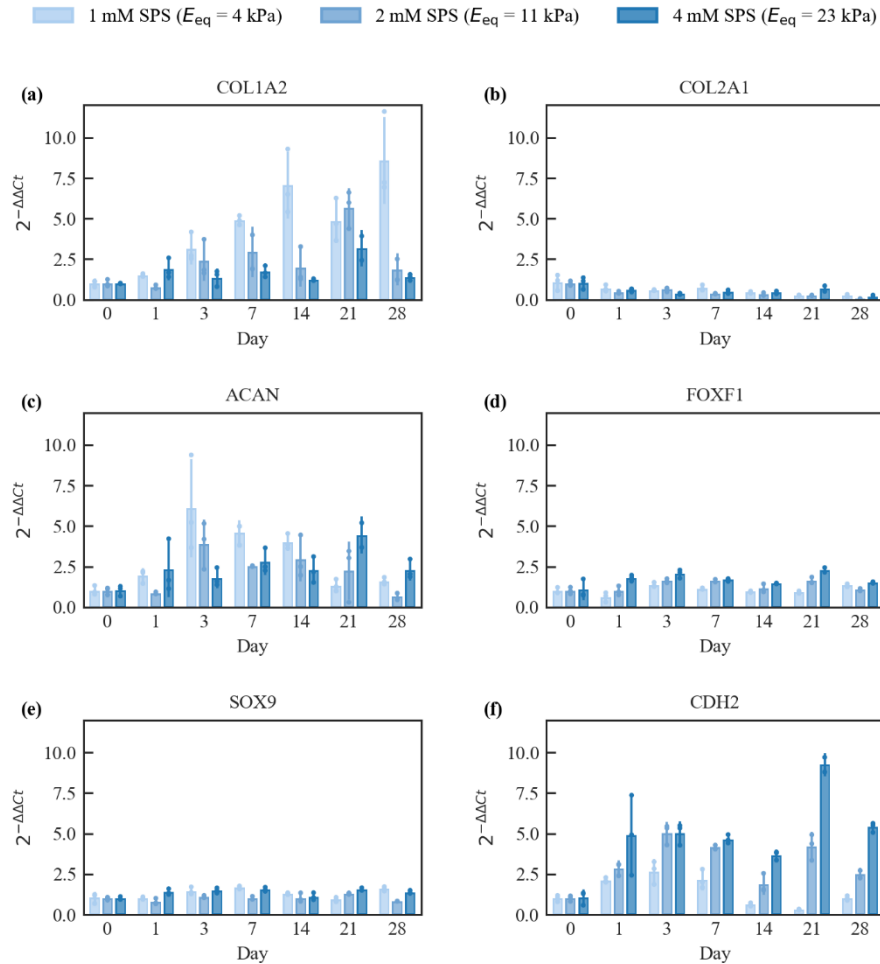


Figure 10. Expression of standard nucleus pulposus and chondrogenic markers by cells cultured for 28 days in hydrogels with varying stiffnesses. Presented as means of three biological replicates and two technical replicates each, with error bars indicating the standard deviation. A notable upregulation of COL1A2 in softer hydrogels (1 mM SPS) and CDH2 in stiffer hydrogels (4 mM SPS) was observed. FOXF1 and SOX9 remained relatively consistent throughout the 28-day culture period. COL2A1 exhibited a slight downward trend in expression by day 28. ACAN expression peaked in all three formulations by day 3 and decreased again by day 28 of the culture period. Significant differences in expression over time were observed for all genes ($p < 0.016$), whereas the differences across concentrations were significant only for COL1A2 ($p = 0.002$), FOXF1 ($p < 0.001$), SOX9 ($p < 0.001$), and CDH2 ($p < 0.001$).

502 3.10. Swelling and degradation

503 Samples were observed macroscopically throughout 28 days of culture, both with and without embedded
 504 cells. The initial swelling increased mass, as observed in all formulations after one day in culture (**Figure**
 505 **11**). As expected, the swelling was inversely proportional to the degree of crosslinking ($p < 0.001$), with
 506 the softest hydrogels (1 mM SPS) swelling to almost 300% of their initial mass and volume, and the
 507 stiffer hydrogels (4 mM SPS) only to approximately 150%. No significant differences between hydrogels
 508 with and without embedded bovine NP cells or between time points within the 28-day culture period
 509 could be observed. All formulations remained stable in mass and volume after the first day of culture.

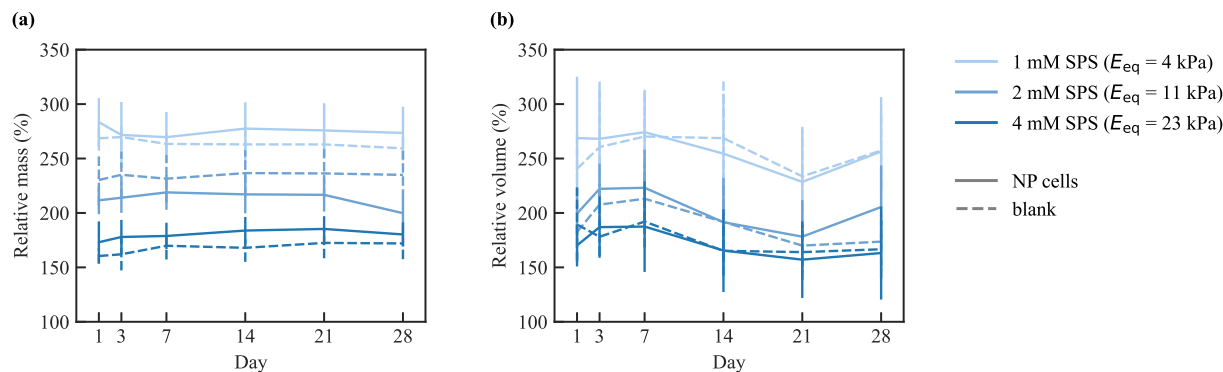


Figure 11. Relative mass (a) and volume (b) in hydrogels of various stiffnesses, with and without cells, were measured over a culture period of 28 days. Presented as means of three biological replicates, with error bars indicating the standard deviation. Rapid swelling, inversely proportional to the degree of crosslinking ($p < 0.001$), was observed in all formulations after a day of culture. No statistically significant differences were observed between hydrogels with or without cells, or across time points. All hydrogels remained stable after the initial swelling for the entire 28-day culture period.

510 4. Discussion

511 Advancements in our understanding of IVD degeneration hinge on our ability to faithfully replicate the
 512 disc's structure and function. Our study focused on the NP, which is crucial for maintaining the disc's
 513 hydration and mechanical resilience and is a key failure point during degeneration. We aimed to develop a
 514 composite hydrogel bioink that more accurately mimics the in vitro NP environment, offering broader
 515 relevance as a model for studying disc degeneration.

516 We formulated our bioink using a combination of native disc components, HA and collagen. Although
 517 previous studies have explored these components individually, few have combined them to leverage their
 518 individual advantages (Gloria et al., 2020). Using tyraminated HA enables the formation of tunable
 519 hydrogel networks with high swelling potential and varied mechanical properties. In combination with
 520 collagen, it provides essential biochemical cues, such as those experienced by cells in tissue (Guerrero et
 521 al., 2021). Our composite bioink further differentiates itself with biopolymer concentrations
 522 approximating physiological levels in human NP (3% HA and 2% collagen) (Singh et al., 2009). Such
 523 concentrations can present challenges when scaling up to physiological disc size in the centimeter range
 524 (G. D. O'Connell et al., 2007). A further challenge of high polymer concentrations involves preparing
 525 stable formulations of HA and collagen, which may form polyionic complexes, as has been discussed in
 526 previous literature (De Oliveira et al., 2023). In this initial study, we investigated isotropic cylindrical
 527 structures of 5 x 1.3 mm as a model to elucidate the effects of the viscoelastic properties on printability
 528 and cell response. It is important to note that, although NP is avascular, physiological nutrient diffusion
 529 may differ due to the size differences (C. Daly et al., 2016). In future models of the disc, these differences
 530 could be minimized through tuning of macro- and micro-porosity, which our material facilitates through
 531 its printability and gelatin component. To achieve a homogeneous composition at these high
 532 concentrations, we embedded pre-gelled collagen fragments within a bulk HA network. Although the use
 533 of mechanically fragmented hydrogels in injectable and extrudable formulations is well established (A. C.
 534 Daly, 2023; Muir et al., 2023), to our knowledge, this is the first time it has been utilized to ensure
 535 homogeneous distribution of a collagen fibrillar network throughout a bulk HA construct. An important
 536 consideration here is our use of type I collagen, rather than the type II present in native tissue. Both are
 537 fibril-forming collagens with similar biochemical properties. The choice of type I was driven primarily by

538 its availability and possibility of achieving relatively high concentrations. Although literature differs
539 somewhat with respect to potential downsides of this substitution, it suggests that type I should
540 nonetheless support discogenic differentiation (Chen et al., 2005; Rutgers et al., 2013). The bioink
541 introduced in this study is not meant for tissue replacement.

542 Designing our hydrogels such that they could be fabricated using accessible and versatile approaches such
543 as extrusion bioprinting, enables future integration in larger tissue-like constructs. Several studies have
544 fabricated annulus fibrosus-like structures using techniques such as 3D printing, electrospinning, or melt
545 electrowriting (Bhunja et al., 2021; Liu et al., 2022; Marshall et al., 2021; Nerurkar et al., 2007). Using
546 multi-material or multimodal fabrication approaches, our bioink could be readily combined with such
547 fibrous structures to form full IVD constructs. We facilitate this through the incorporation of sacrificial
548 gelatin microgels, which modulate the rheological properties of our bioink and enable its extrusion under
549 mild cell-compatible conditions, with pressures within a range that is not expected to impact cell viability
550 (Nair et al., 2009). Additionally, microgel-based materials have actually been shown to shield cells from
551 stresses during extrusion (Sinclair et al., 2018). In contrast to enzymatic crosslinking, where high cell
552 densities can disrupt the gelation process (Petta et al., 2018), we found that sacrificial microgels robustly
553 maintain viscoelasticity even with encapsulated cells. Previous literature has reported that such microgels
554 are dissolvable under standard cell culture conditions and can enhance the rheological properties in a
555 variety of bulk hydrogels, independent of the post-printing crosslinking parameters (Ouyang et al., 2022;
556 Soliman et al., 2023). Our findings agree with this, having observed considerable rheological
557 improvements upon incorporation of the microgels, as well as signs of gradual degradation, indicated by
558 time- and crosslinking-dependent changes in sample FTIR spectra. Due to the presence of tyrosine in the
559 gelatin structure, the precursor to tyramine, we expect our microgels to crosslink proportionally to the
560 bulk gel, thereby reducing their degradation rate. Similar systems in previous work have delayed gelatin
561 degradation by up to two weeks at high degrees of crosslinking, well in excess of concentrations used
562 here (Soliman et al., 2023).

563 We demonstrated the versatility of our bottom-up, in vitro engineered model by showcasing the tunable
564 properties of our material. Functionalizing HA with tyramine facilitates crosslinking through exposure to
565 blue light. Through shear and compression mechanical tests, we have established that minor variations in
566 crosslinker concentration (1-4 mM of SPS) enable us to achieve stiffness values ranging from
567 approximately 5 to 20 kPa. The disc's stiffness increases drastically with compositional changes,
568 particularly its hydration level. The stiffness of a healthy human NP starts at approximately 5 kPa in both
569 compression and shear, and can increase up to tenfold in the most extreme cases of degeneration (Cloyd
570 et al., 2007; Iatridis et al., 1997). This study covered a crucial subset of this range, from healthy to
571 moderately degenerated discs, enabling us to mimic the mechanical changes associated with tissue
572 degradation. As we have not exhaustively explored the limits of the photocrosslinking system in our
573 composite, it may very well be applicable to a broader range of IVD states, including further increases in
574 stiffness for the modelling of severely degenerated NP. Furthermore, changes in crosslinking density also
575 modulate swelling behavior: the stiffer hydrogels exhibit proportionally less swelling. Free swelling of the
576 native nucleus pulposus when removed from the annulus fibrosus can range from 200% in healthy to 60%
577 in degenerated discs (Charnley, 1952; Iatridis et al., 1997). All our formulations maintain considerable
578 swelling capabilities, with mass increases of 150-300% due to swelling observed. This high degree of
579 hydration, attributable to the high HA content, is crucial in replicating the healthy NP mechanics as it
580 supports the tissue's mechanical resilience (Iatridis et al., 1996). In order to explore degenerated NP states
581 further, adaptations such as higher degrees of crosslinking, changes in polymer molecular weight and
582 degree of functionalization, and addition of hydrophobic components might enable additional tunability
583 of the swelling behavior. Future research should also investigate the behavior of this composite under
584 confined conditions, including an annulus fibrosus ring, which limits swelling of the NP in its

585 physiological state. In addition to its flexibility, our crosslinking approach is rapid and has shown no
586 adverse effects on encapsulated cells, with all compositions demonstrating excellent cell viability
587 combined with an apparent cell proliferation. The cell compatibility may be attributed to the use of non-
588 damaging blue visible light, in contrast to commonly employed crosslinking processes with ultraviolet
589 light (Bryant et al., 2000).

590 Our observations of cell morphology and gene expression have shed light on how stiffness influences NP
591 cells. Contrary to our initial hypothesis, the stiffest hydrogels - designed to mimic a degenerated disc –
592 emerged as the most promising environment for NP cell culture. Within these stiffer matrices, cells
593 preserved their rounded phenotype, mirroring native tissue characteristics (Risbud et al., 2015). They
594 displayed a modest increase in aggrecan expression over the four-week culture while maintaining a low
595 collagen I expression. In contrast, cells cultured in the softer formulations underwent considerable
596 elongation, which correlated with increased collagen type I expression and decreased aggrecan expression
597 after a brief initial increase on day 3, suggesting a shift toward a more pathological fibrotic phenotype.
598 Notably, matrix stiffness also influenced the expression of the NP-specific marker CDH2, with
599 upregulation observed in stiffer matrices corresponding with the native cell morphology and phenotype.
600 These alterations may stem from the restricted cell mobility and matrix remodeling capabilities in stiffer
601 hydrogels, which correspond to a denser polymer network (Ehrbar et al., 2011). Previous work has
602 reported the importance of CDH2 expression in NP phenotype preservation: in degenerate human NP
603 cells, which were CDH2 negative, cell clustering promoted re-expression of CDH2 and healthy matrix
604 synthesis (Hwang et al., 2016). It is possible that the combination of a denser polymer network, which
605 limits cell migration, and the presence of distributed spherical pores due to the use of sacrificial gelatin
606 microgels, have together contributed to the preferential clustering of encapsulated cells and formation of
607 CDH2-positive cell clusters. This was in agreement with our qualitative analysis of the sample matrix,
608 where soft hydrogels exhibited a weak matrix by day 28 of culture and no sign of nascent matrix
609 deposition, in contrast to the well-preserved structures and regions of nascent glycosaminoglycans in the
610 stiffest hydrogels. Importantly, these modifications were obtained without supplementation of growth
611 factors such as TGF- β and GDF-5, which could have masked the effects of the material. They are,
612 therefore, inherent to the hydrogel's biochemical composition and mechanical properties. It is not clear if
613 the absence of supplementation with ascorbic acid, an essential cofactor in the collagen synthesis process
614 (Stone & Meister, 1962), is also responsible for the collagen type II downregulation or whether the
615 observed gene expression profile is the result of a monolayer expansion, as observed with phenotypically
616 similar cartilage chondrocytes (Tekari et al., 2014) (Okoro et al., 2023). The cells used in this study were
617 used at passage 2, and nucleus pulposus cells may have the capacity to recover their original potential
618 when transferred in a 3D environment (Gan et al., 2003). However, further work will be required to
619 identify the optimal cell source and handling procedure for use with the developed biomaterials. Although
620 our study focused on juvenile bovine NP cells, which were selected in order to validate the material's
621 capacity to preserve the NP phenotype, the method could be adapted by using differentiating
622 mesenchymal stromal cells (MSCs) into NP cells within HA/collagen hydrogels (Okoro et al., 2023), or
623 induced-pluripotent stem cells. In the future, our composite system could therefore be expanded to MSCs
624 or annulus fibrosus cells to broaden the relevance and applicability of our findings. Further investigating
625 the stiffness-dependent behavior of these cells could offer more detailed insights into cell-specific
626 responses, potentially expanding our bioink's clinical relevance beyond modelling to applications in
627 regenerative medicine.

628 **5. Conclusions**

629 We introduce a composite bioink tailored for the biofabrication of NP constructs, formulated with 3% HA
630 and 2% collagen, to approximate the composition of native NP tissue. By varying the degree of
631 crosslinking, we achieved a wide range of swelling behaviors, compressive moduli, and shear moduli,
632 encompassing both healthy and degenerated human IVD. Our bioink formulations demonstrate potential
633 in supporting the NP phenotype without supplementation with growth factors. Furthermore, the varying
634 crosslinking degrees enabled us to modulate cell morphologies, nascent matrix production, and gene
635 expression profiles. Interestingly and somewhat paradoxically, softer substrates allowed increased cell
636 spreading, with a tendency to de-differentiate into a fibroblast-like phenotype, whereas in stiffer
637 substrates, cells maintained a rounder morphology, which better corresponds to NP. These observations
638 were possible thanks to the absence of growth factors, providing valuable insight on how material cues
639 influence biological properties, which could inform future model design and IVD research. This work
640 expands the range of bioinks with high biopolymer concentrations available for the biofabrication of IVD-
641 like constructs.

642 **6. CRediT authorship contribution statement**

643 GM: Conceptualization, data curation, formal analysis, investigation, methodology, project
644 administration, validation, visualization, writing - original draft, writing – review & editing; SDO:
645 investigation, methodology; MS: investigation, methodology; CLV: funding acquisition, methodology,
646 project administration, resources, supervision, writing – review & editing; CH: funding acquisition,
647 methodology, project administration, resources, supervision, writing – review & editing; SJF: funding
648 acquisition, methodology, project administration, resources, supervision, writing – review & editing; MD:
649 conceptualization, data curation, funding acquisition, methodology, project administration, resources,
650 supervision, writing – review & editing.

651 **7. Declaration of competing interest**

652 The authors declare no conflict of interest.

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658 **9. Data availability**

659 Data supporting this study's findings are available from the authors upon reasonable request.

660 **10. References**

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