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# Effect of Nitrogen and Phosphorus Doping of Reduced Graphene Oxide in the Hydrogen Evolution Catalytic Activity of Supported Ru Nanoparticles

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## 1. Figures and Tables



**Figure S1.** TEM images and corresponding size distribution histograms of (a-c) **Ru@rGO**, (d-f) **Ru@NH<sub>2</sub>-rGO** and (g-i) **Ru@P-rGO**. Figures 1a-f are reproduced with permission from ref. 1. Copyright 2019 Elsevier Inc.



**Figure S2.** HRTEM micrographs of (a) **Ru@rGO**, (b) **Ru@NH<sub>2</sub>-rGO** and (c) **Ru@P-rGO**. HRTEM image of **Ru@P-rGO** (left, right bottom) and Fourier Transform analysis with the planar reflections (right, top), which displays reflections to the (100) and (002) atomic planes with a lattice fringe spacing of 2.289 Å and 2.165 Å, respectively. Figure S2b is reproduced with permission from ref. 1. Copyright 2019 Elsevier Inc.



Figure S3. Raman spectra of (a) P-rGO and (b) Ru@P-rGO.



**Figure S4.** XPS spectra of C 1s signal of (a) rGO, (b)  $NH_2$ -rGO and (c) P-rGO. Figure S4b is reproduced with permission from ref. 1. Copyright 2019 Elsevier Inc.



**Figure S5.** XPS spectra of (a) P 2p signal of P-rGO, and the Ru 3p signal of **Ru@P-rGO** (b) before and (c) after reduction conditions (180 °C under  $H_2$  atmosphere for 5 h).



**Figure S6.** XPS spectra of Ru 3p signal of **Ru@rGO** (a) before and (b) after reduction (180 °C under H<sub>2</sub> atmosphere for 5 h), and **Ru@NH<sub>2</sub>-rGO** (c) before and (d) after reduction conditions (180 °C under H<sub>2</sub> atmosphere for 5 h). Figures S6c-d are reproduced with permission from ref. 1. Copyright 2019 Elsevier Inc.

Material	Before red	luction (%)	After reduction (%)		
	Ru(0)	RuO <sub>2</sub>	Ru(0)	RuO <sub>2</sub>	
Ru@rGO	42.6	57.4	48.4	51.6	
Ru@NH <sub>2</sub> -rGO	40.7	59.3	72.6	27.4	
Ru@P-rGO	31.6	68.3	87.3	12.6	

**Table S1.** Quantification of Ru(0) and  $RuO_2$  by XPS of **Ru@rGO**, **Ru@NH<sub>2</sub>-rGO** and **Ru@P-rGO** before (left) and after (right) reduction (180 °C under H<sub>2</sub> atmosphere for 5 h).



Figure S7. <sup>31</sup>P MAS NMR spectrum of P-rGO.



**Figure S8.** TEM images and corresponding size histograms of (a-c) **Ru@rGO**, (d-f) **Ru@NH<sub>2</sub>-rGO** and (g-i) **Ru@P-rGO** after 2 h under catalytic conditions ( $j = -10 \text{ mA} \cdot \text{cm}^{-2}$ ).



**Figure S9.** LSV of (a, green) **Ru-r@rGO**, (b, red) **Ru-r@NH<sub>2</sub>-rGO** and (c, blue) **Ru-r@P-rGO** before (light color) and after (dark color) a 12-h chronopotentiometry experiment at a j= -10 mA·cm<sup>-2</sup>.



**Figure S10.**  $H_2$ -monitored current-controlled bulk electrolysis of (a) **Ru@rGO**, (b) **Ru@NH<sub>2</sub>-rGO** and (c) **Ru@P-rGO** in 1 M  $H_2SO_4$ . The production of  $H_2$  was quantified in the gas phase by using a Clark-type electrode.



**Figure S11.** Determination of cathodic transfer coefficient (α) for **Ru-r@rGO** (green), **Ru-r@NH<sub>2</sub>rGO** (blue) and **Ru-r@P-rGO** (red).



**Figure S12.** Representative multi-CV experiments at different scan rates for  $C_{DL}$  determination of (a) **Ru@rGO**, (c) **Ru@NH<sub>2</sub>-rGO** and (e) **Ru@P-rGO**. Plot of current values at 0.25 V *vs.* SCE for the different scan rates in 1 M H<sub>2</sub>SO<sub>4</sub> for (b) **Ru@rGO**, (d) **Ru@NH<sub>2</sub>-rGO** and (f) **Ru@P-rGO**.

**Table S2.** Comparison of the most relevant graphene-derived and Ru/graphene-based HER nanoelectrocatalysts with our materials under acidic conditions. Commercial Pt/C and Ru@C electrodes have also been added for the sake of comparison. Parameters: mean diameter ( $\emptyset$ ), onset overpotential ( $\eta_0$ , mV), overpotential at |j| =10 mA·cm<sup>-2</sup> ( $\eta_{10}$ , mV), Tafel slope (b, mV·dec<sup>-1</sup>) and exchange current density ( $|j_0|$ , mA·cm<sup>-2</sup>).

Entry	Catalyst	Ø	$\eta_0$	$\eta_{\scriptscriptstyle 10}$	b	j <sub>0</sub>	Rof	
	Catalyst	(nm)	(mV)	(mV)	(mV·dec⁻¹)	(mA∙cm⁻²)	Kel.	
1	Ru-r@rGO	2.0	0	71	65 0.85		This work	
2	Ru-r@NH₂-rGO	1.5	≈ 0	30	56	2.87	This work	
3	Ru-r@P-rGO	1.5	≈ 0	2	51	10.88	This work	
4	Ru/G-r	1.9	≈ 0	29	48	2.50	2	
5	Ru/P-G-r	1.5	≈ 0	15	49	4.97	1	
6	GCE-S-GNs-1000- CB-Bu	30	≈ 60	80	61 (Tafel) 71 (FIS)	0.541 0.431	3	
7	Ru-GLC	2-5	3	35	46	-	Δ	
8	Ru <sub>2</sub> P/RGO	<7	≈ 0	22	29	2.2	5	
9	Ru <sub>2</sub> P@PNC	18	≈ 0	15	28	-	6	
10	Ru@GnP	2	≈ 0	13	30	-	7	
11	N-G	-	≈ 250	490	116	-	8	
12	P-G	-	≈ 300	550	133	_	8	
13	N,P-G	-	≈ 240	420	91	0.00024	8	
14	N,B-CN	-	≈ 410	710	198	-	9	
15	N-CN	-	≈ 400	620	159	-	9	
16	N,P-CN	-	≈ 340	550	139	-	9	
17	N,S-CN	-	≈ 100	290	77	-	9	
18	N-G	-	499	-	405	86	10	
19	N,P-G	-	399	-	565 265		10	
20	P-G	-	536	-	348 76		10	
21	P,N-G	-	247	380	126	21	10	
22	VG	-	≈ 375	-	158	-	11	
23	N-VG	-	≈ 200	290	121	-	11	
24	Ru/NG-750	3-7	≈ 0	53	44	-	12	
25	Ru@CN	2.37	≈ 70	126	-	-	13	
26	Ru-NGA	3.5	≈ 15	55	32	-	14	
27	Ru@NC	2.1	≈ 10	62	40	-	15	
28	Commercial Ru-b	-	70	150	65	0.14	16	
29	Commercial Pt/C	-	0	27	32	1.4	16	

Legend: Ru/G-r, Ru NPs supported on alginate-derived graphene after a reductive treatment; Ru/P-G-r, Ru NPs supported on alginate-derived P-doped graphene after a reductive treatment; GCE-S-GNs-1000, glassy carbon modified sulphur-doped graphene nanosheets heat treated at 1000 °C; GLC, graphene layered carbon; RGO, reduced graphene oxide; PNC, P,N-doped C nanofibers; GnP, graphene nanoplatelets; N-G, N-doped graphene; P-G, P-doped graphene; N,P-G, N,P-doped graphene; N,B-CN, N,B-doped carbon nanosheets; N,P-CN, N,P-doped ca

N,S-CN, N,S-doped carbon nanosheets; P,N-G, P,N-doped graphene; VG, vertical graphene; N-VG, N-doped vertical graphene; Ru/NG-750, Ru nanoclusters on N-doped graphene prepared at 750 °C; Ru@CN, Ru NPs over N-doped carbon; Ru-NGA, Ru-modified N-doped graphene aerogel, Ru@NC, Ru NPs embedded in N-doped carbon; Ru-black.

## 2. DFT computational details

## 2.1. Design of the structural models

This subsection is organized as follows. Firstly, results based on experimental data are shown, related to the computational modelling of the rGO,  $NH_2$ -rGO, and P-rGO supports from pristine graphene. Then, the anchoring of the hydrogenated ruthenium nanoparticle (Ru NP) is discussed, paying special attention to the metal-support interaction and structural changes derived from the coordination of the metallic system on the different carbon materials.

**2.1.1. Design of the supports.** The first step in the development of supported metal catalysts is the design of the supports. For this, graphene was used as a model carbon support, given its excellent properties in supported catalysis.<sup>17</sup> Lattice imperfections are inevitably introduced during its growth or processing. SW defects, single and multiple vacancies (SVs and MVs), carbon, foreign adatoms, and substitutional impurities are typical stable point defects that have been predicted and experimentally observed. Among these, the V<sub>2</sub>(555–777) defect, consisting of a double vacancy reconstructed with three pentagonal and three heptagonal cells of carbon atoms in the graphene lattice, appears as one of the most favored and commonly observed vacancy defects in electron microscopy analyses.<sup>18</sup> Therefore, the introduction of these V<sub>2</sub>(555–777) defects was first considered on pristine graphene, causing no deformation in the surface plane.

The next step was the functionalization of the carbon surface. From the elemental analyses and XPS data previously reported for rGO and NH2-rGO, Error! Bookmark not defined. as well as those shown in Figures S4, S5 and S7 for P–rGO, the molar ratio and density of functional groups, respectively, could be deduced for each support. In order to consider each of these functional groups detected and quantified experimentally, the modelling of the different moieties was carried out through a sequence of serial optimizations on the defective graphene, similarly to the procedure previously reported in our group.<sup>19</sup> The addition/creation of functional groups is carried out homogeneously, with those found within the carbon lattice being introduced first, followed by increasingly larger groups. The final results can be seen in the right part of Figure S13. In the case of rGO, the order was: carboxyl  $\rightarrow$  hydroxyl  $\rightarrow$  epoxide. The moderate concentration of oxygenated groups induces a loss of planarity in the graphitic system, giving rise to the formation of remarkable superficial corrugations. Regarding  $NH_2$ -rGO, the order was: graphitic–N + pyridinic–N + pyrrolic–N  $\rightarrow$  carboxyl  $\rightarrow$  amino  $\rightarrow$  hydroxyl  $\rightarrow$  epoxide. The N-doped support presents a higher density of functionalities and, therefore, a higher deformation of the carbon surface. It is interesting to mention the increase in vacancy defects on the carbon lattice due to the presence of pyridine and pyrrole rings, including some protonated ones. Finally, the order in the P-rGO was: graphitic-P + graphitic-(P=O)  $\rightarrow$  carboxyl  $\rightarrow$  hydroxyl. This support presents the least of the functionalization, partly due to the low phosphorus incorporation in the graphene oxide shell. This leads to a higher concentration of pristine graphitic areas and therefore less rippling of the carbon system.

The thermal stability of previously optimized rGO (clean and doped with N and P)-based supports was investigated through AIMD simulations at a temperature of 450 K (according to the reaction conditions). First, the simulation analysis evidenced in all three cases a relatively fast total energy convergence over time, as shown in Figure S14. The supports reached dynamic equilibrium at approximately 5 ps, after which only a constant oscillation of energy is noticeable. Structural changes were detected, especially in rGO and NH<sub>2</sub>–rGO. In the former, an entire –

COOH group was released, while in the latter,  $CO_2$  and  $H_2O$  were formed as a result of condensation between a carboxyl and a hydroxyl group. P–rGO showed only rather conformational changes derived from out-of-plane vibrations of reduced phosphorus (unlike oxidized phosphorus, P=O). The final geometry of the simulations was relaxed (right part of Figure S13), presenting only small rotations (of carboxyl and amino groups) and deformations (out of plane in the graphitic region).



**Figure S13.** Top and side views of the initial and final (**a-b**) rGO, (**c-d**) NH<sub>2</sub>–rGO, and (**e-f**) P–rGO model carbon supports. Atomic color code: P (Orange); O (Red); N (Blue); C (Grey); H (White).



**Figure S14.** AIMD of the (a) rGO, (b)  $NH_2$ -rGO, and (c) P-rGO model carbon supports: total energy as a function of time.

2.1.2. Grafting of Ru<sub>55</sub> NPs on G-derived supports. Once the pristine graphene has been oxidized into rGO and the corresponding N- and P-doped systems, they can be used as supports for metal deposition. Thus, a hcp-based 55-atoms Ru NP model was deposited on the functionalized surfaces. Such nanoparticles are representative of highly dispersed Ru catalysts (see Figure S1 for the TEM images and size distribution histograms of the Ru@rGO, Ru@NH2-rGO, and Ru@P**rGO** catalysts). Additionally, since the experiments were carried out under  $H_2$  pressure and dihydrogen easily dissociates on the surface of the NP, it is expected to be basically covered with surface hydrides. Consistent with what was previously reported for hydride Ru NPs,<sup>20</sup> a hydrogen-to-surface-metal ratio of approximately 1.2:1 and 1.6:1 has been used, so the final surface composition was found to be Ru<sub>55</sub>H<sub>53</sub> and Ru<sub>55</sub>H<sub>70</sub>, respectively, as illustrated in Figure S15. In all carbon models, the NP is deposited in such a way that the metal-support interaction is established preferentially through an extended (111) facet. As it can be seen, however, the anchoring of the metal species strongly depends on the nature of the support. In the case of rGO, coordination occurs exclusively on a graphitic region, since there are no point vacancies or substitutional heteroatoms that enhance accommodation. Regarding NH<sub>2</sub>-rGO, the interaction is established mainly on a pyridinic– $N_3V_1$  motif, that is, three pyridine rings including a single vacancy in the center. Lastly, according to P-rGO, the adsorption of the NP takes place by the interaction with the graphitic region containing the two species of phosphorus detected experimentally, in addition to two nearby carboxyl motifs. On average for the two hydrogen surface compositions, the Ru-C bonds are in the range between 2.1 and 2.5 Å, while the few Ru–N, Ru–O, and Ru–P bonds turn out to be around 2.2, 2.2, and 2.4 Å, respectively. Regardless of the catalyst, 35 of the 55 Ru atoms are accessible, so the theoretical dispersion of the metal is around 64%. Finally, concerning the nature of the hydrogen atoms, between 60 and 62% of the H atoms showed a preference for bridging adsorption sites (coordinated to two Ru atoms) for the hydrogen coverages studied in this work. In contrast, as the H/Ru ratio increased, the hollow sites (coordinated to three Ru atoms) decreased from 32 to 26% while the top ones (coordinated to a single Ru atom) increased from 6 to 14%, mainly due to the evident crowding of the metal surface for the ratio Ru<sub>55</sub>H<sub>70</sub> with respect to Ru<sub>55</sub>H<sub>53</sub>.

Structure-property analysis of the supported  $Ru_{55}H_{53}$  and  $Ru_{55}H_{70}$  NPs using nearest-neighbor (NN) search revealed average Ru–Ru and Ru–H bond lengths of 2.56 and 1.75 Å, respectively, with no significant differences between the supports and stoichiometry studied in this work. It is found that the average NN coordination number (CN) of Ru–Ru and Ru–H is 7.96 and 2.15, respectively, in the case of the former with differentiated values depending on the Ru atoms located in the core ( $CN_{max}$ =12) and on the surface ( $CN_{min}$ =5) of the NP. In addition, the mean interatomic distance (MIAD) was calculated, which reflects the effective radius of a cluster, obtaining a value of 0.58 nm, which is equivalent to a diameter of approximately 1.2 nm, very close to the 1.5 ± 0.2 nm or 1.5 ± 0.3 nm measured by TEM in the case of Ru NPs grafted on the nitrogen- and phosphorus-doped rGO supports, and slightly under the observed diameter for the Ru-containing rGO supports (2.0 ± 0.8 nm). These numbers highlight the absence of major structural changes involved by the support.



**Figure S15.** Top views of  $Ru_{55}H_{53}$  and  $Ru_{55}H_{70}$  NPs on the (**a-b**) rGO, (**c-d**) NH<sub>2</sub>–rGO, and (**e-f**) P–rGO functionalized supports. Atomic color scheme: Ru (Celadon Green); P (Orange); O (Red); N (Blue); C (Grey); H (White); Hydrides (Yellow).

#### 2.2. Volcano plots and DFT calculations of H adsorption energies

Volcano plots were calculated using the Yang, Saidi and coll. electrochemical model,<sup>21,22</sup> which is an improvement of the seminal Nørskov model.<sup>23</sup> Both provide valuable theoretical frameworks for understanding the kinetics and mechanisms of the HER. As such, they can be used to rationalize the behavior of electrocatalysts, but also to design and optimize new catalysts. Both rely on the hydrogen adsorption free energy descriptor,  $\Delta G_{H^*}$ : the plot of the logarithm of the experimental exchange current density, log  $j_0$ , as a function of the DFTcalculated  $\Delta G_{H^*}$  shows that optimum catalysts have a moderate adsorption strength, *i.e.*  $\Delta G_{H^*} \approx$ 0. This is in line with the Sabatier principle.

The  $\Delta G_{H^*}$  descriptor is actually the dissociative adsorption energy of H<sub>2</sub> on the surface, *i.e.* the energy of the reaction H<sub>2</sub>(g) + 2\*  $\rightarrow$  2H\* (\* = active site of the surface of the electrochemical catalyst). The general formula for the average adsorption energy of n hydrides is given by:  $\Delta E_{H^*} = [E(nH^*) - (n/2)E(H_2) - E(NP)]/n$ . Given that the explicit calculation of the Gibbs free energy correction factor for all models considered in this study would be a computationally expensive task, we decided to apply a universal correction:  $\Delta G_{H^*} = \Delta E_{H^*} + 5.5$  kcal/mol, where the 5.5 kcal/mol can be considered a representative value that accounts for the vibrational contribution to *G* energy for all metals.<sup>23</sup>

The Yang and Saidi electrochemical model for HER combines exchange current densities  $j_0$ , volcano plots, DFT adsorption energies  $\Delta G_{H^*}$  and Volmer-Heyrovsky pathways, according to which the overall HER proceeds through  $H^+ + e^- \rightarrow \frac{1}{2}H_2(g)$ . The HER electrochemical exchange current density model is cast as a function of the rate-determining step (rds) of the Volmer, Heyrovsky and Tafel reactions:

- Volmer regime, *i.e.* rds is  $H^+ + e^- \rightarrow H^*$ , which corresponds to  $\Delta G_{H^*} > 0$  of the Nørskov model

$$j_0 = ek_0 C_{\text{tot}}(1-\theta) \exp\left(-\alpha \frac{\Delta G_{\text{H*}}}{k_{\text{B}}T}\right)$$
(1)

- Heyrovsky regime, *i.e.* rds is  $H^* + e^- + H^+ \rightarrow H_2$ , which corresponds to  $\Delta G_{H^*} < 0$  of the Nørskov model

$$j_0 = ek_0 C_{\text{tot}}(1-\theta) \exp\left(-(1-\alpha)\frac{\Delta G_{\text{H*}}}{k_{\text{B}}T}\right)$$
(2)

- the Tafel reaction is the rds at  $\Delta G_{\mathrm{H}^*} = 0$ 

In these equations,  $\Delta G_{H^*}$  is the dissociative hydrogen Gibbs free energy previously introduced,  $\alpha$  is the so-called Butler-Volmer transfer coefficient for the electrode reaction,  $\theta$  is the fraction of occupied sites,  $C_{tot}$  is the total concentration of adsorbed hydrides, H<sup>\*</sup>, and of protons, H<sup>+</sup> ( $C_{tot} = C_{H^+} + C_{H^*}$ ), and  $k_0$  is the rate constant. Using the Langmuir model,  $\theta$  can be expressed as a function of the equilibrium constant K as  $\theta = K/(1+K)$ , where

$$K = \exp\left(-\frac{\Delta G_{\mathrm{H}*}}{k_{\mathrm{B}}T}\right) (3)$$

Yang and Saidi hypothesized that  $k_0$  is metal-independent. They found  $k_0 = 126 \text{ s}^{-1}$  by fitting the theoretical model to a database that contains  $j_0$  and  $\alpha$  experimental data and DFT adsorption energies,  $\Delta G_{H^*}$ . Volcano curves are plotted in Figure S16 as a function of  $\alpha$ . The lower the  $\alpha$  transfer coefficient, the smoother the slope, and the higher the exchange current density for a given  $\Delta G_{H^*}$  value.



Figure S16. Volcano curves computed using eq. 1 and 2, with  $C_{tot} = 1$  and  $k_0 = 126 \text{ s}^{-1}$ .

As a consequence, the experimental knowledge of  $\alpha$  is key to establish a quantitative correspondence between experimental exchange current densities  $j_0$  and their theoretical counterpart,  $\hat{j}_0$  (where  $\wedge$  stands for "predicted"). This is clearly shown in Figure S17, where the predicted  $log(\hat{j}_0)$  values obtained with three different models are reported as a function of the selection of experimental data of ref. 22. A linear regression has been systematically been done, as well as the calculation of the Mean Average Error (MAE) between predicted and experimental  $\log(j_0)$  values. Figure S17a shows the accuracy of the seminal Nørskov model, with  $\alpha$  fixed to 1. There is a systematic underestimation of the efficiency of the catalysts, which increases for bad catalysts. The MAE is found to be 2.806, which is quite large. There is a significant enhancement of the prediction when  $\alpha$  is considered in the model, as shown in Figure S17b. The coefficient of determination R<sup>2</sup> improves (0.907 vs. 0.877), and the MAE decreases (1.292). There is however a significant deterioration of the model in the low  $log(j_0)$  range, *i.e.* for the least efficient catalysts. The electrochemical model used in the present work does better in this range, and is also accurate for higher  $\log(i_0)$  values, *i.e.* to predict exchange current densities of efficient catalysts that lie close to the top of the HER volcano plot (see Figure S17c). Whilst the coefficient of determination  $R^2$  is almost the same as that of the  $\alpha$ -dependent Nørskov model (0.903 vs. 0.907), the MAE decreases (1.011).



**Figure S17**. Predicted exchange current densities,  $\log(\hat{j}_0)$ , with  $\hat{j}_0$  in A·cm<sup>-2</sup>, as a function of experimental data taken from the database compiled in ref. 22 (see also Table S3). (a) Seminal Nørskov model, with  $\alpha = 1$ ; (b) Nørskov model, with experimental  $\alpha$  values; (c) Yang and Saidi model, which is expressed as a function of experimental  $\alpha$  values. Plain claret line: hypothetical exact prediction. Dashed claret line: linear regression between  $\log(\hat{j}_0)$  and  $\log(j_{0,exp})$ ; the slopes, y-intercept,  $R^2$  and MAE are also reported on the graphs (Figs (a) and (c) are similar to the plots reported in Figure S2 of ref. 22).

	electrode	DGH/eV	logj0(YS2022)	logj0(Norskov)	logj0(Norskov,α)	logj0(exp)	exp.alpha	auth-year	doi
0	Pt/C	-0.09	-2.35	-2.920	-2.228	-1.80	0.5	J. Zheng et al. (2016)	10.1126/sciadv.1501602
1	Pt/C	-0.09	-2.35	-2.920	-2.228	-0.92	0.5	J. Durst et al. (2014)	10.1149/2.0981501jes
2	lr	-0.14	-2.72	-3.700	-2.609	<mark>-1.4</mark> 4	0.5	J. Durst et al. (2014)	10.1149/2.0981501jes
3	lr	-0.14	-3.44	-3.700	-3.296	<mark>-1</mark> .89	0.8	J. Zheng et al. (2016)	10.1126/sciadv.1501602
4	Pd	-0.18	-2.75	-4.340	-2.641	-2.52	0.4	A. Safavi et al. (2014)	10.1016/j.fuel.2013.10.016
5	Pd	-0.18	-3.35	-4.340	-3.230	-3.08	0.6	J. Zheng et al. (2016)	10.1126/sciadv.1501602
6	Rh/C	-0.20	-3.89	-4.680	-3.766	-2.28	0.7	J. Durst et al. (2014)	10.1149/2.0981501jes
7	Rh/C	-0.20	-3.55	-4.680	-3.430	-2.17	0.6	J. Zheng et al. (2016)	10.1126/sciadv.1501602
8	Ru	-0.25	-3.59	-5.450	-3.514	-2.35	0.5	A. Kuhn et al. (1970)	10.1016/S0022-0728(70)80194-2
9	Cu	0.30	-5.81	-6.469	-5.455	-6.84	0.8	N. Pentland et al. (1957)	10.1149/1.2428530
10	Cu	0.25	-5.81	-6.440	-4.779	-6.84	0.8	N. Pentland et al. (1957)	10.1149/1.2428530
11	Co	-0.37	-3.91	-7.400	-3.901	-5.44	0.4	K. Kawashima et al. (2020)	10.1021/acsaem.0c00321
12	Ni	-0.33	-3.68	-6.810	-3.632	-5.59	0.4	E. Navarro-Flores et al. (2005)	10.1016/j.molcata.2004.10.029
13	Au(111)	0.50	-6.78	-10.790	-6.046	-6.60	0.5	J. Perez et al. (1998)	10.1021/jp9831987
14	Au(111)	0.50	-6.78	-10.790	-6.046	-6.47	0.5	S. Štrbac et al. (2018)	10.1149/2.0441815jes
15	PolyAu	0.50	-6.78	-10.790	-6.046	-6.85	0.5	J. Perez et al. (1998)	10.1021/jp9831987
16	Re	-0.33	-4.30	-6.850	-4.189	-5.90	0.9	J.G. Rivera et al. (2019)	10.1016/j.ijhydene.2019.08.212
17	Re	-0.33	-4.30	-6.850	-4.189	-6.00	0.9	R. Garcia-Garcia et al. (2015)	10.1007/s12678-014-0240-z
18	Bi	1.07	-13.80	-20.830	-12.250	-14.50	0.6	Y.M. Wu et al. (2005)	10.1016/j.jpowsour.2004.10.019
19	Cd	1.03	-9.56	-19.810	-8.362	-10.77	0.4	J.O.M. Bockris et al (1964)	10.1016/0013-4686(64)80003-7
20	In	0.91	-10.39	-17.890	-9.089	-10.82	0.5	J.N. Butler et al. (1965)	10.1149/1.2423502

**Table S3**. Reproduction of the database used in ref. 22 to compare the predictive performance of electrochemical models with respect to experimental values (values used in the in-house Jupyter Notebook workflow developed for this study). logj0(YS2022): Yang and Saidi electrochemical model; logj0(Nørskov): seminal Nørskov model with  $\alpha = 1$ ; logj0 (Nørskov, $\alpha$ ): seminal Nørskov model with experimental  $\alpha$ . DGH/eV is  $\Delta G_{H^*}$ . The logj0 (Nørskov,  $\alpha$ ) predicted values were not reported in Table S3 of ref. 22.

The application of the electrochemical model to the rGO systems leads to the volcano curves plotted in Figure S18. If the electrochemical model is reliable, and provided that DFT adsorption energies are accurate enough and that the structural models developed in this study (*vide infra*) are close to the actual experimental systems, these plots show that in average  $\Delta G_{H^*}$  is expected

to be found at  $\approx$  -1, -3 and -5 kcal·mol<sup>-1</sup> for **Ru-r@P-rGO**, **Ru-r@NH<sub>2</sub>-rGO** and **Ru-r@rGO**, respectively. This is obviously a real challenge given the complexity of the system and the standard accuracy of DFT, so it would be satisfying to find the same trend.



**Figure S18**. Volcano plots predicted after the Yang, Saidi and coll. electrochemical model,<sup>21,22</sup> using  $\alpha$  transfer coefficients found in the present study (see Table 1 in the MS), and with  $k_0$ : 126 s<sup>-1</sup> and  $C_{\text{tot}}$ : 1. The experimental log( $j_0$ ) values are also reported.

#### 2.3. Evaluation of the range effect of the supports on H adsorption energies

Instead of calculating the adsorption energy of a few hydrides selected here and there on the surface, a procedure was defined to evaluate the effect of the radius of action of the support on the hydride's adsorption energy. The methodology is summarized in Figure S19a, where R is the distance between the support and the atoms of the NP. It primarily consists in discretizing the interaction distance, R, by creating a set of contiguous intervals that span their range of values between 0 and  $R_{max}$ , where the support is the origin of the R scale. The m hydrides H\* that lie in a given  $[R_i, R_{i+1}]$  range are removed in order to calculate the corresponding  $\Delta E_{H*}$  value:  $\Delta E_{H*} = [E(NH*) - E((N-M)H*) - (M/2)E(H_2)]/M$ .



**Figure S19.** Range effect of the support on the H adsorption energies for the Ru-r@rGO case: (a) discretization of the distance *R* between the support and the atoms of the nanoparticle for this Ru<sub>55</sub>H<sub>53</sub>@rGO model. Cyan atoms are hydrides that lie in a [ $R_{i}$ ,  $R_{i+1}$ ] interval. The close-packed planes (cpp) are itemized from bottom to top; (b) example of log  $\hat{j}_0$  plotted as a function of *R*: hexagonal markers = value calculated for a given [ $R_i$ ,  $R_{i+1}$ ] interval (their size is proportional to  $n_{\rm H}(R)$ , the number of hydrides in the corresponding interval; error bars are calculated considering a possible arbitrary error of ±0.05 in the determination of the  $\alpha$  value), thick dashed grey line = weighted average of log  $\hat{j}_0$  between 0 and a given distance  $R_i$  (see text) - the area surrounding the weighted average value of log  $\hat{j}_0$  takes into account any experimental determination ±0.05 error on  $\alpha$ , horizontal dash dotted grey line = weighted average of log  $\hat{j}_0$  between 0 and  $R_{\rm max}$ , horizontal dashed claret line = experimental value.

The log  $\hat{f}_0$  value is then easily calculated for each range, as described in section 2.2. It can be plotted for each interval, as shown in Figure S19b (hexagonal markers). A weighted average of log  $\hat{f}_0$  can also be calculated as

$$\overline{\log \hat{j}_0(R_I)} = \frac{\sum_{R=0}^{R_I} n_{\rm H}(R) \log \hat{j}_0(R)}{\sum_{R=0}^{R_I} n_{\rm H}(R)}$$
(4)

which is plotted as a thick dotted grey line in Figure S19b. It is then possible to report the  $\log \hat{j}_0$  values calculated on all  $[R_i, R_{i+1}]$  intervals on a volcano plot defined from the previously described Yang and Saidi model, using the experimental  $\alpha$  values. This is what is reported in Figure S20 (blue diamonds). This figure supports various analysis, as explained in the figure itself as well as in its caption. The difference between the horizontal blue and light blue dashed lines can be interpreted as the direct short-range effect of the support on the HER performance. On the

other hand, on an unadsorbed model, hydrides would be better distributed on the surface. The influence of the H---H closer interaction on  $log \hat{i}_0$  in  $Ru_{55}H_{53}$ @NH<sub>2</sub>-rGO with respect to the free  $Ru_{55}H_{53}$  model is also evaluated by considering two systems:  $Ru_{55}H_{53}$  after its supported counterpart and  $Ru_{55}H_{53}$  with hydrides distributed on the whole metal surface of the  $Ru_{55}$  metal part, including the bottom of the  $Ru_{55}$  model (named  $Ru_{55}H_{53}$  eq. in the caption of Figure S20).



**Figure S20.** Volcano plot for the Ru-r@NH<sub>2</sub>-rGO, after the Yang and Saidi model and using the experimental  $\alpha$  value and the "universal"  $k_0$  rate constant calculated in ref. 22. The experimental log  $j_0$  is also shown as a horizontal dotted claret line. Blue diamonds: log  $\hat{j}_0$  values calculated for the Ru<sub>55</sub>H<sub>53</sub>@NH<sub>2</sub>-rGO model in each [ $R_i$ ,  $R_{i+1}$ ] interval, between  $R_0 = 0$  and  $R_i = 3$  Å. Light blue diamonds: log  $\hat{j}_0$  values calculated for the same intervals, but on the unsupported Ru<sub>55</sub>H<sub>53</sub> model. The corresponding log $\hat{j}_0(3)$  averaged value is also reported in both cases, as horizontal light blue plain (Ru<sub>55</sub>H<sub>53</sub>@NH<sub>2</sub>-rGO) and dashed (Ru<sub>55</sub>H<sub>53</sub>) lines. Since this value is calculated in both cases with the experimental  $\alpha$  value found for Ru-r@NH<sub>2</sub>-rGO, their difference can be analyzed in terms of range effect of the support. The lower dash dotted horizontal line is plotted after the log $\hat{j}_0(3)$  averaged value calculated for the original Ru<sub>55</sub>H<sub>53</sub> model developed in ref. 20. In this model, hydrides are distributed on the whole surface of the Ru<sub>55</sub> model, whereas in its adsorbed counterpart some hydrides at the bottom ones were moved to favor a strong coordination between the support and the NP. The meaning of the red and blue double arrows is explained on the figure.

#### 2.4. DOS and pCOHP analysis of the support - Ru-r interactions in $Ru_{55}H_{70}$

pCOHP focuses on the strength and nature of atom-atom or orbital-orbital interactions over the energy spectrum, whereas pDOS provides a broader context by showing how specific orbitals contribute to the electronic states across the materials. Together, they deliver a comprehensive understanding of the material electronic structure and bonding. If needed, readers can familiarize themselves with such analysis by referring to ref. 40, which offers a detailed examination of the same Ru<sub>55</sub> core, stabilized with hydrides and/or carbon monoxide ligands.

The DOS and pCOHP profiles of  $Ru_{55}H_{70}@X$  (X = rGO, NH<sub>2</sub>-rGO, P-rGO models) are reported in Fig. S21. At first glance, they look alike, in agreement with the global similarity of the rGO supports. Some interesting numbers however arise from these two properties:

1) While the d-band center of the surface atoms is similar across all cases (approximately -3.03 eV), the d-band center of the cpp1 layer—the layer grafted onto the surface—is significantly higher in the  $Ru_{55}H_{70}$ @P-rGO system (-2.84 eV vs. -2.97 eV). Although other factors, such as local electron density, can also play a role, this in line with its superior activity towards HER.

2) The COHP Ru-surface interaction energy is strong in all cases (17.8 eV, 15.6 eV, 15.3 eV). There is no contradiction with the numbers reported in Figure 2 of the manuscript, given that this energy clue must not be confused with a bond dissociation energy (BDE), but rather considered as an intrinsic interaction energy index, not dependent of the geometry relaxation of systems when BDEs are calculated. Interestingly, it shows that an individual Ru-P interaction is significantly stronger than a Ru-C and Ru-N index (2.7 eV vs. 0.8 eV and 1.7 eV). Together with atomic charges and d-band center values, all these results suggest that a variation of the number of C-Ru, C-O and C-N interactions sites could modulate the activity of the Ru-r@X systems towards HER. As also revealed by the COHP analysis, the intrinsic stability of the NP is not altered by the support, each Ru-Ru bond COHP index being close to 1 eV (i.e. *ca.* 225 eV for Ru<sub>55</sub>), a value similar to index energies in the Ru<sub>55</sub>H<sub>70</sub> model (results taken from the SI of a previous study, see ref. 20; energies are given here in eV for the sake of comparison with this previous study).



**Figure S21.** DOS and pCOHP analysis for the  $Ru_{55}H_{70}$  Ru NP model grafted on the three supports considered in the present study (rGO,  $NH_2$ -rGO, P-rGO). The Fermi energy (brown horizontal line) and the d-band center of surface Ru atoms (red horizontal line, energy in eV) as well as the d-band center of the cpp1 and cpp2 layers of Ru atoms (in eV, cyan characters, see definition in Fig. S19) are also given. The Ru-C, Ru-N, Ru-O and Ru-P COHP profiles are not shown as they are not sufficiently intense to stand out, even when compared to the C-O interactions, which are both more numerous and stronger. A discussion is however provided in the text.

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