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Exploring the Substitution of Fe(III) by Gd(III) in Nanomagnetite

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Gadolinium-based contrast agents (GBCAs) have been
widely used in medical magnetic resonance imaging
(MBI) since the 1080s⁻¹ This conception of Gd complexes is (MRI) since the 1980s.¹ This generation of Gd complexes is highly stable and inert regarding human biochemistry. However, their environmental impact might not be negligible as originally thought. Increasing levels of Gd have been reported in aquatic systems near major urban centers where several hospitals are present.2[−]⁴ The San Francisco Bay water experienced an ∼70% increase in Gd concentration over an ∼8-year time period.⁵ The amount of Gd released in the environment for the United States and in the EU was estimated to be ∼21 tons/year and ∼19 tons/ year, respectively.⁶ It is expected that, due to the exponential growth of medical applications, the concentration of anthropogenic Gd in aquatic systems will keep increasing.^{7,8}

Magnetite is widely used as a powerful remediation agent due to its low production cost, redox-active behavior, magnetic properties, and capability to bind contaminants efficiently. $9,10$ Previous studies have shown that the incorporation of trace metals, such as $Co(II)$, $Zn(II)$, $Cr(III)$, within the crystal lattice of magnetite is possible.^{11,12} A very limited number of studies examined the processes leading to the incorporation of Gd into nanomagnetite. $12,13$ We still ignore whether Gd can be incorporated into the magnetite structure or directly adsorbed on its surface. This crucial information is required for appropriately designing sustainable and efficient solutions preventing the release of Gd into the environment. Additionally, the influence of Gd substitution within magnetite must be refined, because upon oxidation magnetite can be transformed into maghemite. This iron oxide mineral is less interesting for industrial applications, as its reactivity and magnetic susceptibility are lower compared to magnetite.^{14,15}

Our objectives were (i) to synthesize Gd-doped magnetite/ maghemite nanoparticles and substitute Fe(III) by Gd(III) across a range of loadings (1 to 12% of Gd), (ii) to characterize all the experimental products with an array of different techniques, and (iii) to assess the influence of Gd on nanomagnetite.

The magnetite oxidation into maghemite results from the acidic conditions that favor the release of Fe(II) into solution (eq 1). $16,17$ An inverse linear relationship is reported for Fe concentration in log scale versus pH for values ranging from 5 to 10 ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S1).

$$
Fe_3O_4 + 2H^+ \rightarrow Fe_2O_3 + Fe^{2+} + H_2O
$$
 (1)

Assuming that eq 1 is the only process releasing $Fe(II)$ in solution, the percentage of magnetite transformed into maghemite can be determined. For pH values \geq 7, the magnetite conversion is stalled, and the amount of released $Fe(II)$ remained below 0.07 mM throughout the experiment. At pH 6 (human blood typically has a pH between 7.35 and 7.45), only $~\sim$ 6% of maghemite was produced; but for lower pH (\leq 5), the majority of the magnetite was transformed into maghemite, up

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to ∼82% at pH 3 ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S1). The specific surface areas (SSAs) measured directly after synthesizing the nanomagnetites at pH 6 and 8 ranged from 62 to 70 $\mathrm{m}^2/\mathrm{g}.$ At pH 4, the SSA increased to 92 m²/g.

For all doped nanomagnetites, no detectable levels of Gd were found in the supernatants after solid separation. The concentrations of Gd in nanomagnetites were found to be close to the predicted values $(\pm 10\%)$, confirming that Gd was completely fixed by the nanomagnetites.

All nanomagnetite particles exhibit a large distribution in size, ranging from 5 to 50 nm (Figure 1). By applying selected area

Figure 1. TEM bright field images of nanomagnetites doped with 3% (a), 5% (b), 7% (c), and 9% (d) Gd(III). Panels (e) and (f) show magnetite stabilized at $pH = 4$ and $pH = 8$, respectively.

electron diffraction and high-resolution transmission electron microscopy analyses, we confirmed the presence of magnetite ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S2). No differences were observed in the size distribution and shape of nanomagnetites stabilized at pH 4 and 8 and for nanomagnetites coprecipitated with Gd(III). The only noticeable difference was found for particles with 7 and 9% of Gd(III), where some eleongated crystals of goethite measuring between 50 and 100 nm were observed.

Gadolinium was present and evenly distributed in all nanomagnetites (Figure 2). Additionally, XEDS measurements confirm the Gd concentration within all nanomagnetites increased with the percentage of Gd(III) added during the synthesis [\(Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S3).

Here, we used Cs corrected STEM to enhance the localization of Gd in coprecipitated samples. The images captured in pure nanomagnetites reveal contrasting features between columns with varying iron density, such as distinct brightly colored lines formed by iron in octahedral sites on crystals oriented to expose

Figure 2. STEM-HAADF and STEM-XEDS elemental mapping of magnetite crystals with 3%; 5%, 7%, and 9% Gd(III).

the (111) crystallographic planes. This technique facilitates clear differentiation between Gd and Fe atoms based on their distinct atomic numbers. Whether Gd is situated on the surface or within the structure becomes evident. Providing Gd would be adsorbed, the Gd atoms exhibit contrast spots, more visible near the edge or even at the last surface plane of the particles. The contrast of individual Gd atoms is stronger since they are located, by projection, on the thinnest part of the particle. Conversely, if Gd is present within the structure, it should be observable throughout the particle, often with a weaker contrast. Figure 3 [\(Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S7 as well) allows for a comparative analysis of

Figure 3. HAADF STEM images of (a) doped nanomagnetite containing 4% Gd (III) and (b) Gd adsorbed on pure nanomagnetite.

the coprecipitated and adsorbed samples, revealing the identification of individual Gd atoms for both cases. As expected, the adsorbed sample predominantly displays Gd atoms at the particle edge, while in the doped sample, Gd atoms are distributed throughout the particle, seemingly aligned with sites corresponding to an octahedral Fe column XRD patterns of a dry, immediately synthesized nanomagnetite show only peaks typical corresponding to pure magnetite phase ([Figures](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S2 and [S4](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf)) face-centered cubic unit cell parameter *a* = 8.391(1)− 8.397(1) Å). The crystallite sizes estimated from the Scherrer equation give values between 12 and 17 nm ([Table](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S1). The width of the diffraction peaks of the three stabilized samples for pH 3, 5, and 8 are similar [\(Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S4), excluding a decrease in the magnetite crystal size, in agreement with prior studies.¹⁸ For a 16 nm spherical particle with 20% unmodified nanomagnetite, we estimated an inner diameter of 9.4 nm and an oxidized layer around the magnetite core of ∼ 3 nm thickness. However, the TEM images (Figure 1f) show a larger size distribution (∼5−50 nm). Therefore, it is possible that some small particles may be completely oxidized, while others may have only a thin layer of oxidation at the surface.

The particle size of the Gd-nanomagnetite samples showed no significant difference compared with pure nanomagnetites, supporting our TEM observations. Additionally, no shifted diffraction peak was observed nor was a peak corresponding to a Gd hydroxide phase detected in the diffraction patterns. However, in those samples containing more than 5% Gd(III), the solids showed crystallized phases of goethite. Magnetite doped with 8% Gd seems to contain ∼5% of a green rust phase with a pyroaurite structure. For nanomagnetites coprecipitated with 7% and 9% Gd(III), the presence of less than ∼3% of green rust was less obvious because the XRD detection limit is ∼ 1%.

For magnetite to be formed, the $Fe(II)/Fe(II)+Fe(III)$ ratio must be equal to 0.66 (pH range: 8.5−12; ionic strength: 0.5− $3M^{16}$). Our nanomagnetite synthesis involved a high concentration of OH[−], which maintained the solution pH close to 12 for all preparations. For solutions containing 1 to 5% Gd(III), the Fe(II)/Fe(II)+Fe(III) ratio varied between 0.655 and 0.667 ([Figure.](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S6). For concentrations between 6 and 12%, the $Fe(II)/Fe(II) + Fe(III)$ ratio ranged from 0.638 to 0.653 (values < 0.66), which is below the stoichiometric value required for the formation of magnetite. As a result, goethite and green rust formed through the dissolution−crystallization process.

A typical Mössbauer spectrum for magnetite at room temperature consists of two sextets: (i) Fe(III) in tetrahedral positions and (ii) Fe(III) and Fe(II) in octahedral positions, that appear as an average $\text{Fe}^{2.5+}$ resulting from fast electron exchange. Magnetite spectra collected at $T = 77$ K, below the Verwey transition (∼119 K) where electron hopping is absent, show the expected structural change from a cubic to a monoclinic system. 19

For the Gd free nanomagnetite sample ($pH = 8$; T = 300 and 77 K), typical spectra expected for magnetite were collected. The sextet lines are wider than expected for crystalline magnetite, and the broadening was attributed to the presence of superparamagnetic relaxation effects due to the small size and distribution of particles. The mean value of the isomer shift is typical of the exclusive presence of Fe(III) and, therefore, consistent with the presence of nanomagnetites

For the nanomagnetite samples doped with Gd(III) ranging from 1 to 5%, the spectra at 300 K show broader lines than those for Gd free nanomagnetite, while the hyperfine structures at 77 K are rather similar. These features can be explained both by the

presence of superparamagnetic nanoparticles and the additive role of Gd(III) ions introducing internal and/or surface structural disorder.

The magnetite-to-maghemite conversion ratios obtained from Mössbauer results and ICP-AES measurements are similar (see [Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S1) and confirm prior findings.¹⁰ For Gd-doped nanomagnetites (1 to 5%), the mean isomer shift values are consistent with the oxidation of some Fe species, allowing an estimate of the mean composition of the nanoparticles ([Table](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) [S2](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf)), supported by the XRD data. Although, there were differences observed in the spectra between 300 and 77 K, the cause of this discrepancy remains unknown.

The incorporation of structural Gd is possible if Gd(III) replaces Fe(III) atoms in an octahedral position. Such substitution is possible because the charge and size are similar (0.938 and 0.645 Å for Gd and Fe, respectively). However, $Gd(III)$ displays a much larger ionic radius than $Fe(III)$ that may hinder or condition at low concentrations its incorporation into the magnetite lattice.²⁰ This assumption is supported by the analysis of coprecipitated nanomagnetites with 7 and 9% of Gd. The Mössbauer spectra differ significantly from the spectra of Gd free magnetite (pH = 8) (Figure 4). Significant new quadrupolar doublets were observed in the central part (see red curve), suggesting the presence of some Fe-containing impurities. Indeed, the X-ray patterns indicate the presence of other mineral phases, such as goethite and green rust. The first phase generally contributes to a magnetic sextet with broadened lines and a specific value of quadrupolar shift (∼0.20 mm/s) and/or quadrupolar doublet depending on the size of the particles. The complexity of the hyperfine structure prevents the identification and estimation of any goethite content. But the red curve (quadrupolar doublets) could be attributed to some Fe species located in structurally and chemically disturbed green rusts.

Conversely, the hyperfine spectral structures for nanomagnetites at $pH = 3$ and 5 differ significantly from the previously discussed spectra for Gd-rich magnetites (Figure 4). These differences result from magnetic components with broadened and asymmetric lines that can be described via a linearly correlated distribution of hyperfine fields and isomer shifts. The most relevant parameter is given by the average values of the isomeric shifts, which are intermediate between those typically observed in oxide phases containing Fe(II) and Fe(III) species. Such description is consistent with a mixture of stoichiometric magnetite and maghemite phases [\(Table](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) S2 and [Figure](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf) $S1$;^{21,22}) and confirms the absence of maghemite in Gdrich particles.

We synthesized nanomagnetites with a stoichiometric ferrous-to-ferric ions ratio close to 0.5 and a size ranging from 5 to 50 nm. Our results show that the maghemite level remained below ∼10%, and that Gd(III) is uniformly distributed throughout the nanoparticles in lieu of Fe(III) in octahedral sites. Gd-doped nanomagnetite samples form true $Fe_{(3-x)}Gd_xO_4$ solid solutions, $0.02 < x < 0.1$, with Fe(III) being substituted by Gd(III) within the magnetite superparamagnetic iron oxide nanoparticles (SPION). Such trapping (or substitution) within the mineral structure prevents the release of $Gd(III)$ and unlocks its potential as a theragnostic agent within the cytoplasm or nucleus of the targeted cells. These nanoparticles clearly present a strong potential for medical applications, combining the advantages of SPION nanoparticles and Gd-concentrates particles for imaging, therapeutic, or theragnostic applications. Moreover, this composite is environmentally friendly as it could

Figure 4. Mössbauer spectra measured at 300 K (a) and 77 K (b) for Gd free nanomagnetite, nanomagnetite equilibrated at pH 3 and 5, and nanomagnetite doped with 1−9% of Gd (III) fitted with different components containing $\text{Fe}^{2.5+}$ (red line) and $\text{Fe}(\text{III})$ (blue line) species.

be readily eliminated through conventional wastewater treatment methods.

■ **ASSOCIATED CONTENT**

s3 Supporting Information

The Supporting Information is available free of charge at [https://pubs.acs.org/doi/10.1021/acsnanoscienceau.4c00032.](https://pubs.acs.org/doi/10.1021/acsnanoscienceau.4c00032?goto=supporting-info)

Materials and methods, calculated magnetite to maghemite data conversion from ICP-AES and Mössbauer spectroscopy, additional magnetite crystal analysis of TEM, SAED, HRTEM and STEM; XRD patterns data

from all magnetite synthesis, calculated parameters after Rietveld refinements and a representative Rietveld analysis pattern of powder diffraction data; ratios of iron and OH concentrations used to prepare Gd-doped magnetites and the refined values of the hyperfine parameters of Mössbauer [\(PDF](https://pubs.acs.org/doi/suppl/10.1021/acsnanoscienceau.4c00032/suppl_file/ng4c00032_si_001.pdf))

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Author Contributions

Carolina Guida: Conceptualization, Writing − original draft, Writing - Review & Editing, Anthony Chappaz: Conceptualization, Funding acquisition, Supervision, Validation, Writing - Review & Editing. Agnieszka Poulain: Formal analysis, Writing − original draft, Jean-Marc Greneche: Conceptualization, Methodology, Alexandre Gloter: Formal analysis, Nicolas Menguy: Formal analysis, Nathaniel Findling: Formal analysis, Laurent Charlet: Conceptualization, Funding acquisition, Supervision, Validation, Writing - Review & Editing.

Notes

The authors declare no competing financial interest.

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