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Engineering 2D material exciton lineshape with graphene/*h*-BN encapsulation

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Control over the optical properties of atomically thin two-dimensional (2D) layers, including those of transition metal dichalcogenides (TMDs), is needed for future optoelectronic applications. Remarkable advances have been achieved through alloying [1, 2], chemical [3–5] and electrical [6, 7] doping, and applied strain [8–10]. However, the integration of TMDs with other 2D materials in van der Waals heterostructures (vdWHs) to tailor novel functionalities remains largely unexplored. Here, the near-field coupling between TMDs and graphene/graphite is used to engineer the exciton lineshape and charge state. Fano-like asymmetric spectral features are produced in WS₂, MoSe₂ and WSe₂ vdWHs combined with graphene, graphite, or jointly with hexagonal boron nitride (*h*-BN) as supporting or encapsulating layers. Furthermore, trion emission is suppressed in *h*-BN encapsulated WSe₂/graphene with a neutral exciton redshift (44 meV) and binding energy reduction (30 meV). The response of these systems to electron-beam and light probes is well-described in terms of 2D optical conductivities of the involved materials. Beyond fundamental insights into the interaction of TMD excitons with structured environments, this study opens an unexplored avenue toward shaping the spectral profile of narrow optical modes for application in nanophotonic devices.

Interlayer near-field coupling in vdWHs plays a determinant role on the performance of such 2D nanostructures. For example, the characteristics of graphene-based field-effect transistors were dramatically improved in devices encapsulated in *h*-BN [11–14], due to the atomic flatness and low trap density of *h*-BN [13, 14]. Indeed, encapsulation in graphite (Gr) or *h*-BN ensures monolayer flatness down to tens of picometers [15, 16], as well as improved interfacial cleanliness and homogeneous dielectric environment [17]. In addition to these passive environment and dielectric disorder effects, electromagnetic coupling to substrates has also been used to modify exciton transition and binding energies [18]. However, by tuning the nature of the substrate, one could envision even better control over the transitions, as commonly done in nanophotonics. A variety of coupling regimes can be obtained in plasmonic systems. For example, strong coupling between two plasmons having similar energies and linewidths produces two new energy-split

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states [19]. Coupling of a spectrally broad plasmonic resonance to a sharp excitonic or phononic mode induces a Fano-like resonance, with a modified and possibly asymmetric lineshape [20]. In the same line, the engineering of Fano-resonant ultrathin optical multilayers leads to new optical properties with applications in, for example, photovoltaics [21]. Given the ability to realize atomic-layer control of vdWHs, detailed monitoring of their optical properties at high spatial and spectral resolution is indispensable.

In this work, different electromagnetic environments for TMD monolayers have been designed and precisely fabricated to produce a dramatic modification of their optical response. Optical absorption and spatially resolved electron energy-loss spectroscopy (EELS) spectra show asymmetric lineshapes, similar to Fano profiles at energies close to the excitonic transitions in TMD monolayers when encapsulated or supported on graphene or thin graphite (<10 nm thickness). EELS measurements were performed at temperatures of $T \approx 110$ K in which the transition linewidth is not altered compared to the response of the same monolayers encapsulated in *h*-BN. A simple model based on the 2D optical conductivity [22], including retardation corrections for the interaction with the electron beam and combining a TMD monolayer with a conductive or dielectric environment, explains the observed lineshapes. This indicates that: i) coupling mainly has an electromagnetic origin, without significant modifications to the TMD electronic structure; and ii) dissipation and charge transfer to the graphene or graphite layers do not modify the exciton line broadening within the 10 meV energy resolution in the EELS measurements.

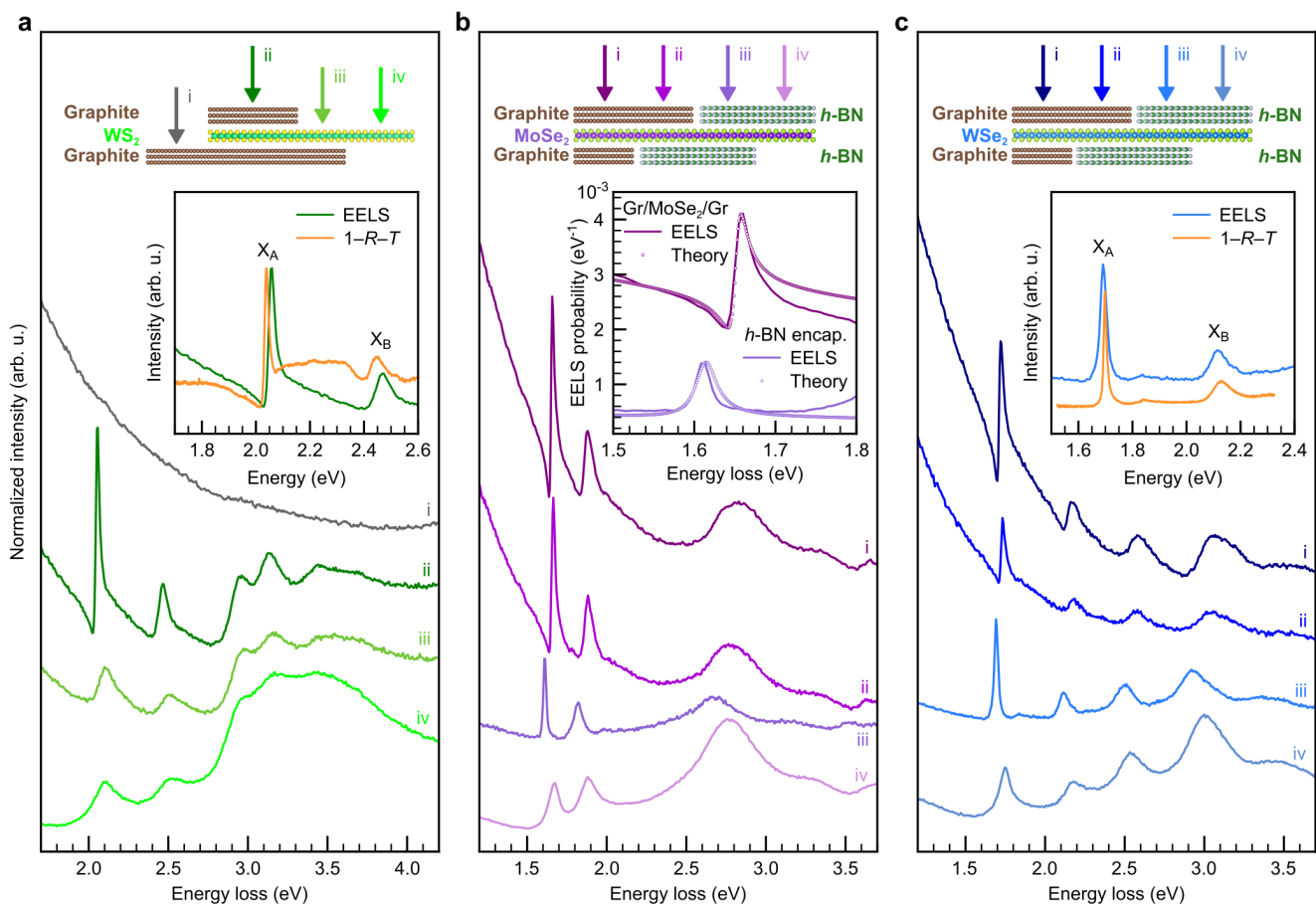


FIG. 1. **Spectra of TMD monolayer with graphite and/or *h*-BN encapsulation:** EELS spectra of (a) WS_2 , (b) MoSe_2 , and (c) WSe_2 monolayers in different configurations of freestanding, supported or encapsulated with *h*-BN and/or thin graphite measured at $T = 110$ K. The configuration of each spectrum is color-coded by the arrows and roman numerals on the sketches in the upper part of the panels. All spectra are normalized with respect to the total intensity of the elastic (zero-loss) peak (ZLP) and vertically offset for clarity. Insets in (a) and (c) show the optical absorption spectrum (orange curves) on similarly produced $\text{Gr}/\text{WS}_2/\text{Gr}$ and $\text{h-BN}/\text{WSe}_2/\text{h-BN}$ heterostructures at 5 K and 150 K, respectively, compared to their respective EELS spectrum. The inset in (b) shows the comparison of experimentally measured (lines) and modelled (dots) EELS spectra for a MoSe_2 monolayer encapsulated in thin graphite (asymmetric lineshape) and *h*-BN (Lorentzian lineshape).

For each of the TMD monolayers considered here (WS_2 , MoSe_2 , and WSe_2), vdWHs were designed and fabricated with a structure varying along both the in-plane and out-of-plane directions, as represented in the upper sketches of

Figs. 1(a–c) and S11 in the Supplementary Information (SI). This ensures that a single TMD monolayer is studied under the same conditions but with varying dielectric environment. The sub-nm sized electron probe used in the EELS experiment is much smaller than the typical lateral extension of different stack configurations, allowing site selectivity when studying them individually and at their interfaces [such as shown in Fig. S14(a)].

Measured EELS and optical absorption spectra of the vdWHs are shown in Fig. 1 and contain a series of excitonic transitions [23, 24]. The sharpest feature in the spectra is the lowest-energy exciton, named X_A , which occurs at the lowest-lying optically active transition at the K/K' points in reciprocal space. For the analyzed samples, the evolution from freestanding to h -BN supported and encapsulated monolayer behaves as reported in the literature [13, 16, 23, 24]: the ~ 100 – 150 meV broad peak sharpens, attaining a Lorentzian lineshape when encapsulated in h -BN [Fig. 1(b–c)] with a full-width at half-maximum (FWHM) in the 20–40 meV range at 110 K. Encapsulation between h -BN and/or graphite ensures reduced monolayer roughness and fewer adsorbates on TMD monolayers [16], as shown for the case of the Gr/WS₂/Gr heterostructure in Fig. S12 in the SI.

For TMD monolayers encapsulated in thin graphite (Gr/TMD/Gr), the lineshape is markedly different from those encapsulated in h -BN, especially when comparing to spectra from EELS or optical absorption in h -BN/WSe₂/ h -BN [Fig. 1(c) inset]. Asymmetric lineshapes characteristic of Fano profiles [25] are observed at slightly redshifted energies compared to the excitonic transitions of the freestanding TMD, as most evident in the two lowest-energy excitons X_A and X_B in the insets. The asymmetric lineshape appears in addition to the known continuous absorption of graphite [Fig. 1(a) grey EELS spectrum for bare and thin graphite]. The redshifts of the excitons in Gr/WS₂/Gr relative to freestanding WS₂ arise from dielectric screening, and in fact 1–2 layered graphene has been used previously to tune the electronic gap and exciton binding energy [26]. A Gr/WS₂/Gr heterostructure supported on a sapphire substrate shows a similar feature at the X_A and X_B exciton energies in the optical absorption spectrum measured at $T = 5$ K [inset of Fig. 1(a)]. Furthermore, the persistence of the asymmetric lineshape at room temperature in EELS for the Gr/WS₂/Gr heterostructure [shown in Sec. S4 and Fig. S14] suggests a different origin than previously reported, where the continuum stems from the trion state [27]. Coupling occurs for TMD layer thicknesses beyond monolayers as well, leading to similarly asymmetric lineshapes [see results for graphite-encapsulated WS₂ bilayer shown in Fig. S15].

The observations so far suggest that the resulting lineshape and its asymmetry depends on the thickness of the encapsulating graphene/graphite layers [see details in Sec. S4 and Fig. S17(a)]. For a WS₂ monolayer supported on graphite, a broad peak (larger than 100 meV FWHM) is observed, similar to that of the suspended monolayer [Fig. 1(a)], but with a strongly asymmetric lineshape towards lower energy. The broadened linewidth most likely occurs due to the poor optical response of non-encapsulated layers caused by residues and adsorbates on the remaining free surface [16]. The consequence of heterostructure interfacial cleanliness on the visibility of the asymmetric Fano lineshape due to linewidth broadening is demonstrated in Gr/TMD/ h -BN heterostructures of MoSe₂ and WSe₂ shown in Fig. 1(b) and (c), respectively. Full encapsulation ensures locally clean interfaces that can be probed selectively by positioning the electron beam, for which the lineshape asymmetry is also observed, although with a lower contrast for TMDs encapsulated only on one side by thin graphite. The FWHM of the Fano-like lineshapes, given by the interval between the minimum and maximum of the asymmetric profile, for graphite encapsulation is similar to those with h -BN-encapsulation, indicating that the line broadening is not largely influenced by the damping from the graphene/graphite layers.

To summarize the experimental observations in Fig. 1, the lineshapes around the excitonic transitions of X_A and X_B strongly depend on the material and thickness of the encapsulating layers. The observed spectral behavior can be understood by starting with a theoretical description of the optical response of 2D heterostructures within the framework of classical electrodynamics. It is assumed that each j -th layer forming the heterostructure is so thin that finite-thickness effects can be neglected. Optical properties of an individual layer are described by a frequency-dependent conductivity $\sigma_j(\omega) = i\omega t_j / (4\pi) [1 - \epsilon_j(\omega)]$, where t_j is the layer thickness, and ϵ_j is its dielectric function depending on optical frequency ω . Stacking the layers together results in the overall optical conductivity of the vdWH modeled as $\sigma(\omega) = \sum_j \sigma_j(\omega)$ [28].

The interaction of electromagnetic waves with the vdWH described by the total optical conductivity is encompassed in the Fresnel coefficients [29]. Incidentally, the Fresnel reflection coefficient for p-polarized light, $r_p(k_{\parallel}, \omega) = 1 / [1 + \omega / (2\pi\sigma \sqrt{(\omega/v)^2 - k_{\parallel}^2})]$, where k_{\parallel} is the in-plane (with respect to the layer planes) wave vector, also enters the electron energy-loss probability for an electron beam passing through the thin vdWH

$$\Gamma_{\text{EELS}}(\omega) = \frac{4e^2}{\pi\hbar v^2} \int_0^{\infty} \frac{k_{\parallel}^3 dk_{\parallel}}{[k_{\parallel}^2 + (\omega/v\gamma)^2]^2} \text{Re} \left\{ \frac{1}{\sqrt{(\omega/v)^2 - k_{\parallel}^2}} r_p(k_{\parallel}, \omega) \right\}, \quad (1)$$

where e is the elementary charge, \hbar the reduced Planck constant, v the electron velocity, c is the speed of light in vacuum, and $\gamma = 1/\sqrt{1 - v^2/c^2}$. Incidentally, a non-retarded expression for the Fresnel coefficient is used as

an accurate description for layers of small thickness compared with c/ω . The fully non-retarded expressions can be obtained straightforwardly by setting $c \rightarrow \infty$ and $\gamma = 1$ [22]. Although this theoretical description completely neglects microscopic electronic interaction (e.g. orbital hybridization) between the individual atomic planes, such as those captured by first-principle calculations, it can still nicely describe the experimental observations in Fig. 1. Focusing on modelling the spectra around the region of the X_A excitonic transition, for which the Lorentz-Drude model of the optical conductivity is applied: $\sigma_{\text{TMD}} = i\omega t_{\text{TMD}}/(4\pi)[-f/(\omega_A^2 - \omega^2 - i\gamma\omega)]$, where f represents the transition strength, ω_A is its angular frequency and γ stands for a phenomenological damping. The dielectric response of the encapsulating materials can be approximated by constant values in the considered energy region: $\epsilon_{\text{Gr}} \approx 6 + i10$, and $\epsilon_{h\text{-BN}} \approx 4 + i0.5$ [30, 31]. The TMD and encapsulating layer thicknesses are further set according to the experimentally estimated values [see Table I in the SI].

The inset in Fig. 1(b) shows the correspondence between experimentally measured spectra and the model for MoSe_2 , where values of f , ω_A and γ are fitted to take into account small energy shifts and differences in broadening originating from microscopic effects. The fitted values (summarized in Table I), however, show only small differences for the three modelled spectra, which confirms the validity of this approach. The model reproduces the symmetric Lorentzian peak observed for MoSe_2 encapsulated in $h\text{-BN}$, while it follows the asymmetric lineshape for the graphite encapsulation. A comparison of all experimentally measured spectra with their modelled counterparts is summarized in Figs. SI6 and SI7(a) in the SI.

The main reason for the asymmetry of the lineshape for graphite encapsulation *versus* the symmetric spectral features with $h\text{-BN}$ encapsulation stems from the different dielectric response of the two materials. In contrast to $h\text{-BN}$, which in the spectral region of interest exhibits only negligible values of $\text{Im}[\epsilon_{h\text{-BN}}]$, graphite is a quite lossy material. Fig. 2(a,b) further elaborates on the lineshape of the EELS probability as a function of graphite and $h\text{-BN}$ thickness, respectively. By increasing the thickness of the graphite substrate (or encapsulation) attached to a TMD monolayer ($t_{\text{TMD}} = 0.6$ nm) featuring a prototypical excitonic transition characterized by $\hbar\omega_{\text{TMD}} = 1.7$ eV, $\hbar\gamma = 0.02$ eV and $f = 2$ eV², there is an increase in the asymmetry (and a decrease in contrast) with respect to a slowly varying background that gradually converges to a smooth spectrum of a freestanding graphite layer (dashed lines). On the contrary, EELS lineshapes for the same TMD layer in the vicinity of $h\text{-BN}$ are nearly symmetric for all $h\text{-BN}$ thicknesses.

To obtain more systematic insight into the dependence of the spectral asymmetry on the optical properties of the layers surrounding the TMD monolayer, EELS spectra are calculated for a large range of constant values of $\text{Re}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$ and $\text{Im}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$ representing the real and imaginary parts of the dielectric function of the substrate multiplied by its thickness. The resulting spectra can then be fitted to an empirical model function

$$\mathcal{F}(\omega) = \frac{1}{\omega} \left[b + a \frac{(q + \Omega)^2}{1 + \Omega^2} \right], \quad (2)$$

where $\Omega = (\omega - \omega_{\text{TMD}})/\gamma$, b represents a constant background, the parameter a is related to the contrast and q to the asymmetry (one recognizes the original Fano lineshape [25] in the fraction). Fig. 2(c) corroborates that the largest asymmetries indeed emerge for large values of $\text{Im}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$. The substrate damping thus clearly opens a coupling channel resulting in the Fano-like asymmetric spectral profile. The increase in asymmetry is however accompanied by the decrease in the contrast as confirmed in Fig. 2(c) [see Fig. SI8 in the SI for all parameters extracted from the fitting].

Different vdWH configurations involving graphene or other metals with $h\text{-BN}$ -encapsulated TMDs were also investigated, where the interlayer distance between the TMD and graphene/metal can be varied by using $h\text{-BN}$ as a spacer [see Fig. 3 and Fig. SI5(b)]. EELS spectra comparing graphene/ WSe_2 heterostructure and the same WSe_2 monolayer only [both $h\text{-BN}$ -encapsulated, as seen in the sample optical micrograph of Fig. 3(a)] in Fig. 3(b) show X_A linewidths spanning the same range (20–40 meV FWHM) and a redshift (47 meV) in the X_A^{1s} resonance in the presence of graphene. The first excited states ($n = 2$) of the two lowest-energy ground-state excitons ($n = 1$), namely X_A^{2s} and X_B^{2s} , are also prominently visible in contrast to the graphite-encapsulated WSe_2 in Fig. 1(c). The energy separation (Δ_{12}) between the $1s$ and $2s$ states of an exciton within its Rydberg series is proportional to the exciton binding energy [26, 32], and can be used to approximate the electronic gap. The energy separation for the $h\text{-BN}$ -encapsulated WSe_2 measures $\Delta_{12} = 135$ meV, whereas it is reduced to $\Delta_{12} = 120$ meV for the graphene/ WSe_2 heterostructure, indicative of a reduction in exciton binding energy by 20–30 meV for the latter if considering $E_B \approx 1.3\Delta_{12} - 2\Delta_{12}$ for the non-hydrogenic Rydberg series for WSe_2 [33, 34].

The integration of graphene with $h\text{-BN}$ encapsulation also allows for the opportunity to carry out combined electron spectroscopies of EELS plus cathodoluminescence (CL) on identical regions with subwavelength resolution. The CL spectrum from $h\text{-BN}$ -encapsulated WSe_2 monolayer region is shown in Fig. 3(c, bottom). It contains two emission lines that are assigned to the neutral exciton X_A^0 (1.718 eV) and trion X^* (1.691 eV), respectively. The Stokes shift of 4.8 meV for the WSe_2 monolayer is consistent with optical measurements [35]. Severe quenching in the CL signal was observed in graphene/ WSe_2 [Fig. 3(c, top)], but the remaining single emission line, which is redshifted by 44 meV, can

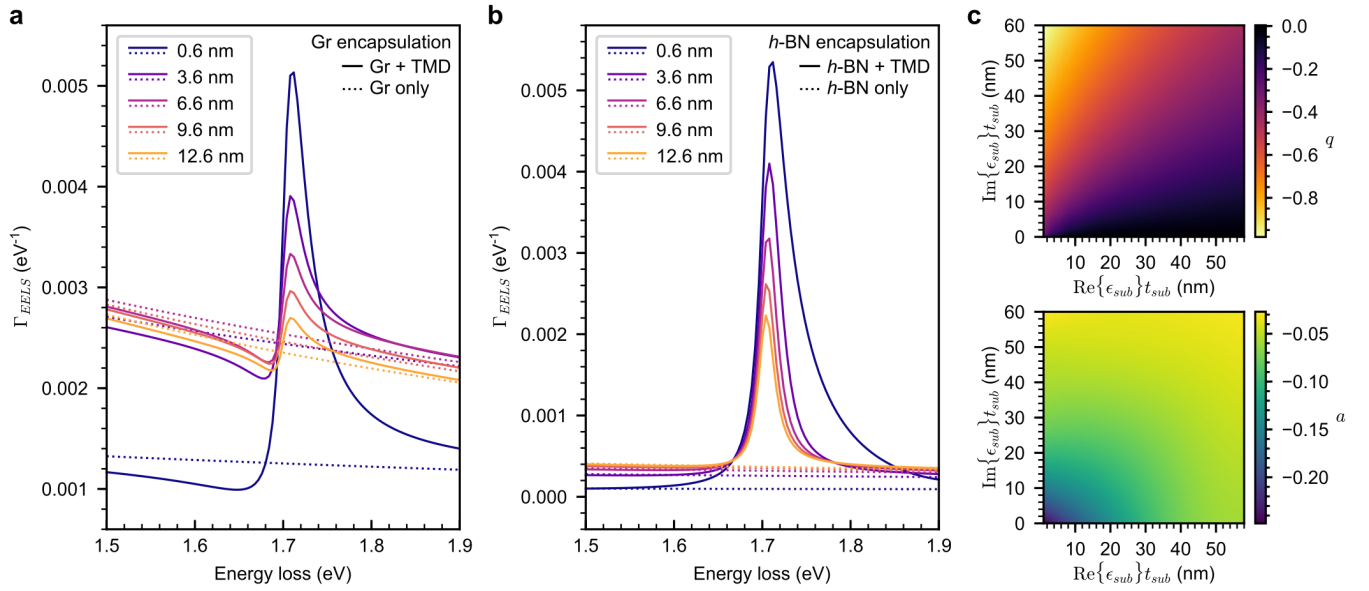


FIG. 2. **Modelled spectra versus parameters of the encapsulating layers:** Evolution of the 2D-modelled EELS spectral shape as a function of (a) graphite and (b) *h*-BN encapsulation thickness. Solid lines represent TMD monolayer + encapsulation, while dashed lines are calculated for encapsulation material only. (c) Fitted parameters q and a of the modified Fano-like profile [Eq. (2)] as a function of the encapsulation properties, specifically represented by the real versus imaginary part of the dielectric function multiplied by the thickness of the corresponding encapsulating layer, $\text{Re}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$ and $\text{Im}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$, respectively.

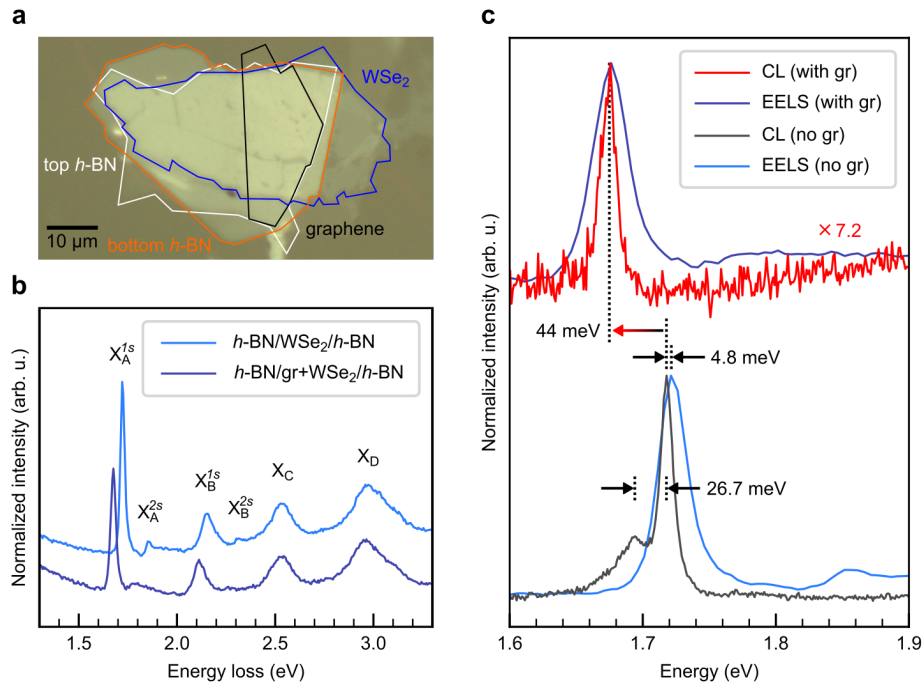


FIG. 3. **Near-field coupling of *h*-BN-encapsulated TMD monolayers with graphene:** (a) Optical micrograph of a *h*-BN/graphene+WSe₂/*h*-BN heterostructure with the different constituents outlined. (b) EELS spectra of *h*-BN encapsulated WSe₂ and graphene/WSe₂ heterostructures, highlighting the clear redshift in the presence of graphene (gr). (c) CL and EELS spectra from identical nanometric regions allows for a clear assignment of the emission peaks, including a prominent low-energy trion emission in the absence of graphene, as well as a clear quenching of the neutral exciton ($\sim 7\times$ lower) but no obvious trion emission.

be rightfully assigned to X_A^0 when compared to the EELS absorption with no obvious low-energy X^* shoulder. This observation is in agreement with low-temperature photoluminescence (PL) from TMD/graphene heterostructures that has shown such a single emission line from only X_A^0 for MoS_2 , WS_2 , MoSe_2 , and WSe_2 [36].

The neutralization effect was attributed to the competition of several relaxation pathways and their lifetime differences, between the neutral (2.3 ps for MoSe_2) and charged exciton (30 ps for MoSe_2) relative to the non-radiative transfer to the graphene [36, 37]. In addition to this, the binding energy of the neutral exciton is expected to decrease in the presence of graphene due to additional Coulomb screening [26], which increases the radiative lifetime [36]. For heterostructures with the TMD monolayers in contact with nm-thick graphite, no CL emission was observed, contrary to structures incorporating h -BN-encapsulation [24, 38]. Altogether, the comparison of the effect produced by graphene versus graphite can be rationalized in the following way: in the case of graphite, a larger non-radiative damping is at the origin of both the emergence of Fano-like lineshapes and the disappearance of radiative emission; with the influence of the graphene being weaker, radiative emission is still possible but weaker and charge-state selective, and logically the Lorentzian lineshape is preserved.

Finally, it is emphasized that the here-described Fano effect is different from previously observed Fano effects in the context of TMD spectra. In contrast to previously reported Fano-like lineshapes in TMDs [27], the continuum is extrinsic to the TMDs, allowing full control on the effect. As compared to Fano-like coupling in plasmonic systems, not only are the continuum and discrete state more rigorously defined, but also the (non-)radiative nature of these two states are reversed between TMD vdWHs and plasmons in nanoparticles.

In summary, the near-field coupling between mono- and few-layered TMDs and graphene/graphite with/without h -BN was experimentally shown to enable a customization of the TMD exciton lineshapes, in excellent agreement with theory. Asymmetric exciton lineshapes along with narrow linewidths were reported for vdWHs with thin graphite or graphene encapsulation using both cryogenic optical absorption and EELS. Such a response can be explained through a simple 2D optical conductivity model of the heterostructure. Interfacing graphene (or other metals) in combination with h -BN encapsulation under different vdWH configurations offers additional flexibility in the control of interlayer separation. This further enabled complementary electron spectroscopies of EELS and CL, wherein for WSe_2 monolayer in the presence of graphene, suppressed charged exciton CL emission and a reduction in exciton binding energy (and hence the quasi-particle gap) of a few tens of meV were observed. Understanding and exploiting the coupling between TMDs and graphene enables the engineering of the exciton lineshape, as well as the electronic bandgap and exciton binding energy, thus opening the agenda for the development of disruptive excitonic applications..

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- [1] G. Wang, C. Robert, A. Suslu, B. Chen, S. Yang, S. Alamdari, I. C. Gerber, T. Amand, X. Marie, S. Tongay, and B. Urbaszek, *Nat. Commun.* **6**, 10110 (2015).
 - [2] A.-Y. Lu, H. Zhu, J. Xiao, C.-P. Chuu, Y. Han, M.-H. Chiu, C.-C. Cheng, C.-W. Yang, K.-H. Wei, Y. Yang, Y. Wang, D. Sokaras, D. Nordlund, P. Yang, D. A. Muller, M.-Y. Chou, X. Zhang, and L.-J. Li, *Nat. Nanotech.* **12**, 744 (2017).
 - [3] S. Mouri, Y. Miyauchi, and K. Matsuda, *Nano Lett.* **13**, 5944 (2013).
 - [4] A. R. Kim, Y. Kim, J. Nam, H.-S. Chung, D. J. Kim, J.-D. Kwon, S. W. Park, J. Park, S. Y. Choi, B. H. Lee, J. H. Park, K. H. Lee, D.-H. Kim, S. M. Choi, P. M. Ajayan, M. G. Hahm, and B. Cho, *Nano Lett.* **16**, 1890 (2016).

- [5] C.-H. Ho, W.-H. Chen, K. K. Tiong, K.-Y. Lee, A. Gloter, A. Zobelli, O. Stéphan, and L. H. G. Tizei, *ACS Nano* **11**, 11162 (2017).
- [6] K. F. Mak, K. He, C. Lee, G. H. Lee, J. Hone, T. F. Heinz, and J. Shan, *Nat. Mater.* **12**, 207 (2013).
- [7] A. Chernikov, A. M. van der Zande, H. M. Hill, A. F. Rigosi, A. Velauthapillai, J. Hone, and T. F. Heinz, *Phys. Rev. Lett.* **115**, 126802 (2015).
- [8] K. He, C. Poole, K. F. Mak, and J. Shan, *Nano Lett.* **13**, 2931 (2013).
- [9] R. Schmidt, I. Niehues, R. Schneider, M. Drüppel, T. Deilmann, M. Rohlfing, S. Michaelis de Vasconcellos, A. Castellanos-Gomez, and R. Bratschitsch, *2D Mater.* **3**, 021011 (2016).
- [10] M. G. Harats, J. N. Kirchhof, M. Qiao, K. Greben, and K. I. Bolotin, *Nat. Photon.* **14**, 324 (2020).
- [11] C. R. Dean, A. F. Young, I. Meric, C. Lee, L. Wang, S. Sorgenfrei, K. Watanabe, T. Taniguchi, P. Kim, K. L. Shepard, and J. Hone, *Nat. Nanotech.* **5**, 722 (2010).
- [12] R. Decker, Y. Wang, V. W. Brar, W. Regan, H.-Z. Tsai, Q. Wu, W. Gannett, A. Zettl, and M. F. Crommie, *Nano Lett.* **11**, 2291 (2011).
- [13] F. Cadiz, E. Courtade, C. Robert, G. Wang, Y. Shen, H. Cai, T. Taniguchi, K. Watanabe, H. Carrere, D. Lagarde, M. Manca, T. Amand, P. Renucci, S. Tongay, X. Marie, and B. Urbaszek, *Phys. Rev. X* **7**, 021026 (2017).
- [14] J. Wierzbowski, J. Klein, F. Sigger, C. Straubinger, M. Kremser, T. Taniguchi, K. Watanabe, U. Wurstbauer, A. W. Holleitner, M. Kaniber, K. Müller, and J. J. Finley, *Sci. Rep.* **7**, 12383 (2017).
- [15] J. C. Meyer, A. K. Geim, M. I. Katsnelson, K. S. Novoselov, T. J. Booth, and S. Roth, *Nature* **446**, 60 (2007).
- [16] F. Shao, S. Y. Woo, N. Wu, R. Schneider, A. J. Mayne, S. M. de Vasconcellos, A. Arora, B. J. Carey, J. A. Preuß, N. Bonnet, *et al.*, *Phys. Rev. Mater.* **6**, 074005 (2022).
- [17] D. Rhodes, S. H. Chae, R. Ribeiro-Palau, and J. Hone, *Nat. Mater.* **18**, 541 (2019).
- [18] S. Borghardt, J. S. Tu, F. Winkler, J. Schubert, W. Zander, K. Leosson, and B. E. Kardynał, *Phys. Rev. Mater.* **1**, 054001 (2017).
- [19] P. Nordlander, C. Oubre, E. Prodan, K. Li, and M. Stockman, *Nano Lett.* **4**, 899 (2004).
- [20] A. E. Schlather, N. Large, A. S. Urban, P. Nordlander, and N. J. Halas, *Nano Lett.* **13**, 3281 (2013).
- [21] M. ELKabbash, T. Letsou, S. A. Jalil, N. Hoffman, J. Zhang, J. Rutledge, A. R. Lininger, C.-H. Fann, M. Hinczewski, G. Strangi, and C. Guo, *Nat. Nanotech.* **16**, 440 (2021).
- [22] F. J. García de Abajo, *ACS Nano* **7**, 11409 (2013).
- [23] A. Arora, N. K. Wessling, T. Deilmann, T. Reichenauer, P. Steeger, P. Kossacki, M. Potemski, S. Michaelis de Vasconcellos, M. Rohlfing, and R. Bratschitsch, *Phys. Rev. B* **101**, 241413 (2020).
- [24] N. Bonnet, H. Y. Lee, F. Shao, S. Y. Woo, J.-D. Blazit, K. Watanabe, T. Taniguchi, A. Zobelli, O. Stéphan, M. Kociak, *et al.*, *Nano Lett.* **21**, 10178 (2021).
- [25] U. Fano, *Phys. Rev.* **124**, 1866 (1961).
- [26] A. Raja, A. Chaves, J. Yu, G. Arefe, H. M. Hill, A. F. Rigosi, T. C. Berkelbach, P. Nagler, C. Schüller, T. Korn, C. Nuckolls, J. Hone, L. E. Brus, T. Heinz, D. R. Reichman, and A. Chernikov, *Nat. Commun.* **8**, 15251 (2017).
- [27] A. Arora, K. Nogajewski, M. Molas, M. Koperski, and M. Potemski, *Nanoscale* **7**, 20769 (2015).
- [28] Z. M. Abd El-Fattah, V. Mkhitarian, J. Brede, L. Fernández, C. Li, Q. Guo, A. Ghosh, A. R. Echarri, D. Naveh, F. Xia, *et al.*, *ACS Nano* **13**, 7771 (2019).
- [29] F. J. García de Abajo, *ACS Photon.* **1**, 135 (2014).
- [30] A. B. Djurišić and E. H. Li, *J. Appl. Phys.* **85**, 7404 (1999).
- [31] S.-Y. Lee, T.-Y. Jeong, S. Jung, and K.-J. Yee, *physica status solidi (b)* **256**, 1800417 (2019).
- [32] A. Chernikov, T. C. Berkelbach, H. M. Hill, A. Rigosi, Y. Li, O. B. Aslan, D. R. Reichman, M. S. Hybertsen, and T. F. Heinz, *Phys. Rev. Lett.* **113**, 076802 (2014).
- [33] K. He, N. Kumar, L. Zhao, Z. Wang, K. F. Mak, H. Zhao, and J. Shan, *Phys. Rev. Lett.* **113**, 026803 (2014).
- [34] A. V. Stier, N. P. Wilson, K. A. Velizhanin, J. Kono, X. Xu, and S. A. Crooker, *Phys. Rev. Lett.* **120**, 057405 (2018).
- [35] I. Niehues, P. Marauhn, T. Deilmann, D. Wigger, R. Schmidt, A. Arora, S. Michaelis de Vasconcellos, M. Rohlfing, and R. Bratschitsch, *Nanoscale* **12**, 20786 (2020).
- [36] E. Lorchat, L. E. P. López, C. Robert, D. Lagarde, G. Froehlicher, T. Taniguchi, K. Watanabe, X. Marie, and S. Berciaud, *Nat. Nanotech.* **15**, 283 (2020).
- [37] P. T. K. Loan, W. Zhang, C.-T. Lin, K.-H. Wei, L.-J. Li, and C.-H. Chen, *Adv. Mater.* **26**, 4838 (2014).
- [38] S. Zheng, J.-K. So, F. Liu, Z. Liu, N. Zheludev, and H. J. Fan, *Nano Lett.* **17**, 6475 (2017).
- [39] P. J. Zomer, M. H. D. Guimarães, J. C. Brant, N. Tombros, and B. J. van Wees, *Appl. Phys. Lett.* **105**, 013101 (2014).
- [40] S. B. Desai, S. R. Madhvapathy, M. Amani, D. Kiriya, M. Hettick, M. Tosun, Y. Zhou, M. Dubey, J. W. Ager III, D. Chrzan, and A. Javey, *Adv. Mater.* **28**, 4053 (2016).
- [41] T. Taniguchi and K. Watanabe, *J. Crystal Growth* **303**, 525 (2007).
- [42] R. F. Egerton, *Electron Energy-Loss Spectroscopy in the Electron Microscope*, 3rd ed. (Springer, New York, 2011) pp. 124, 194, 265, 271.
- [43] T. Malis, S. C. Cheng, and R. F. Egerton, *J. Elec. Microsc. Tech.* **8**, 193 (1988).
- [44] A. Arora, T. Deilmann, T. Reichenauer, J. Kern, S. Michaelis de Vasconcellos, M. Rohlfing, and R. Bratschitsch, *Phys. Rev. Lett.* **123**, 167401 (2019).

SUPPLEMENTARY INFORMATION

S1. Methods

1. Sample Preparation

The samples were prepared using a (bisphenol A polycarbonate) polymer-assisted viscoelastic stamping method to pick up the individual layers making up each vdWH, then dropped onto TEM holey carbon support grids or sapphire substrates directly from the polymer stamp [39]. An example of the fabrication and overview of a sample of a single TMD monolayer with mixed encapsulation material at select steps in the procedure is shown in Fig. S11. Large-area MoSe_2 and WSe_2 monolayers were exfoliated from bulk crystals following a gold-mediated exfoliation method onto SiO_2/Si substrates [40], while WS_2 monolayers were grown by chemical vapor deposition. Encapsulating graphite/graphene and h -BN flakes were produced from conventional tape exfoliation of bulk crystals sourced from NGS Naturgraphit GmbH and synthesized by the high-pressure high-temperature method [41], respectively.

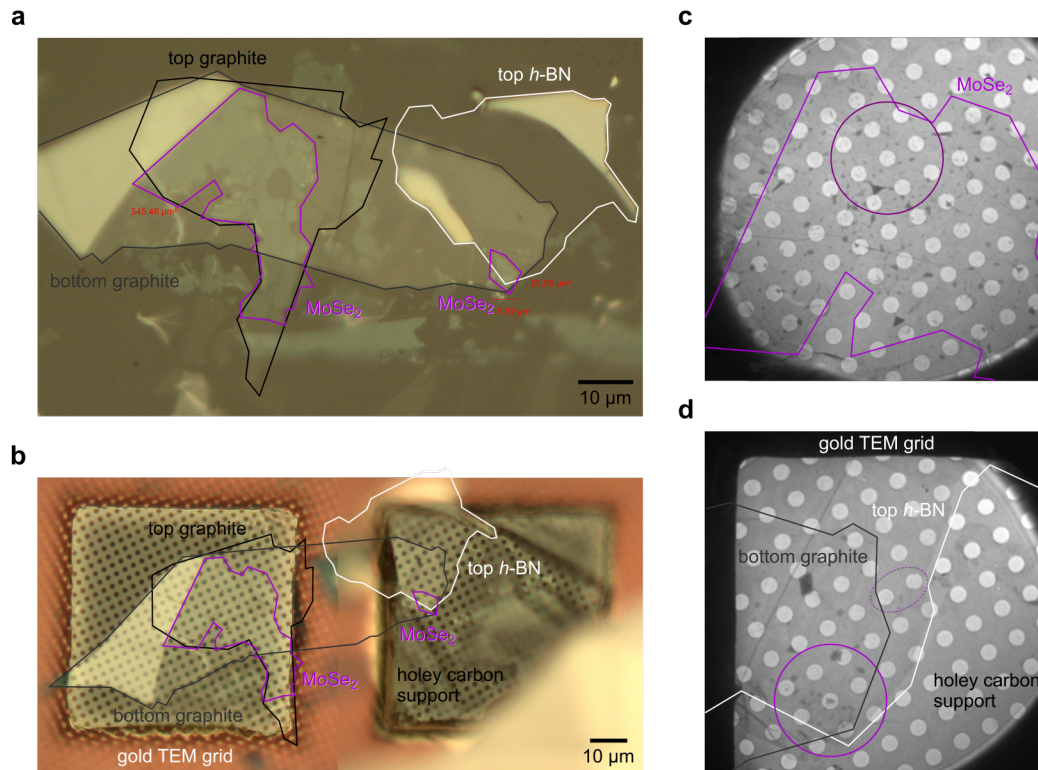


FIG. S11. **Fabrication of the mixed encapsulation sample of MoSe_2 monolayer.** (a) Optical micrograph of a h -BN/graphite + MoSe_2 + graphite heterostructure imaged on the polymer stamp with the different layer constituents outlined. (b) Optical micrograph of the same heterostructure after dropping onto a holey carbon support TEM grid and washing off any polymer residues. Micrographs in (a,b) are each spliced from two separate images due to differences in focus in the top graphite and top h -BN areas. (c) Bright-field STEM image of the heterostructure with the Gr/ MoSe_2 /Gr measurement regions circled. (d) Bright-field STEM image of the h -BN/ MoSe_2 /Gr region of the vdWH stack, with the h -BN/ MoSe_2 /Gr and h -BN/ MoSe_2 measurement areas circled in solid and dotted lines, respectively. Holes in the TEM grid carbon support are 1.2 μm in size. Folds, trapped dirt and bubbles of few hundreds of nm size can be observed in the images of (c,d) as dark contrast.

2. Electron Spectroscopy and Electron Diffraction

The monochromated EELS, CL, and electron diffraction were performed on a modified Nion HERMES-S200 (also known as ChromaTEM) operated at 60 keV with the sample at liquid nitrogen temperatures ($T \approx 110$ K) unless otherwise indicated at room temperature (RT). For EELS and CL experiments, 10 – 15 mrad convergence angle was used and 1 mrad for nano-diffraction experiments. The EELS spectra were recorded onto a Quantum Detectors

MerlinEELS Medipix3 direct electron detector with 256×256 pixel-sized chips in a 4 by 1 geometry. The CL was collected using an Attolight Mönch system fitted with a 150 groves/mm diffraction grating to give a wavelength resolution of 0.34 nm (~ 0.8 meV at 1.72 eV or 720 nm wavelength), recorded onto a Princeton Instruments ProEM EMCCD camera. Different stacking configurations in measured regions of hundreds of nm^2 were determined roughly from the optical micrographs from sample preparation and confirmed locally with nm-specificity by a combination of electron diffraction and core-loss EELS. Graphite and *h*-BN encapsulation layer thicknesses [t_{sub}] were determined experimentally using EELS log-ratio method [42] and calculated values for their inelastic mean free path [43] with the input parameters (incident electron energy, EELS detector collection semi-angles) set according to the experimental conditions while accounting for contributions from the TMD monolayer within the measured effective mean free path.

S2. Monolayer Roughness with Graphite Encapsulation

Nano-beam electron diffraction spots at high sample tilt-angle show no discernible broadening in the Gr/WS₂/Gr heterostructure [Fig. SI2(b)] as compared to the freestanding WS₂ monolayer [Fig. SI2(d)]. This evidences that thin graphite layers have comparable capabilities to reduce corrugation in atomically-thin layers as *h*-BN layers as encapsulation material [16].

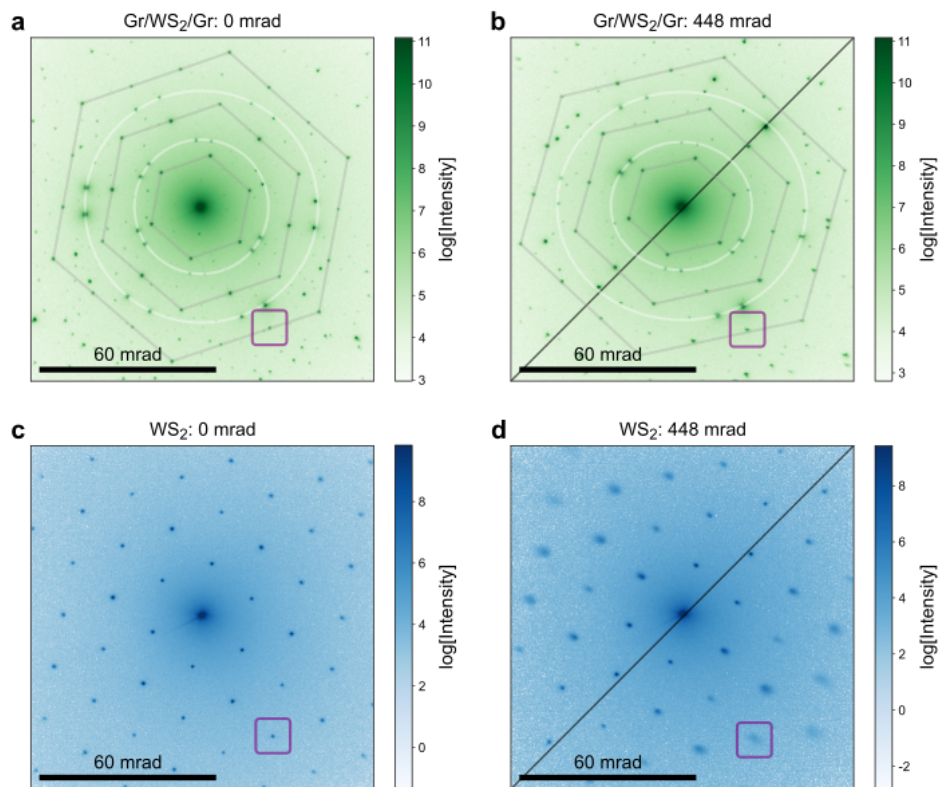


FIG. SI2. Nano-beam electron diffraction patterns of WS₂ monolayer in (a,b) Gr/WS₂/Gr and (c,d) freestanding WS₂ configurations at sample tilt-angles of 0 and 448 mrad. The diffraction spots boxed in purple are of the same index order used to compare the roughness of the WS₂ monolayer. Diffraction spots highlighted by grey hexagons and white circles are from WS₂ and graphite, respectively.

S3. Optical Spectroscopy

To retrieve the absorption spectrum $A(\lambda)$, reflectance $R(\lambda)$ and transmittance $T(\lambda)$ spectra are measured on the same location of the graphite or *h*-BN encapsulated TMD sample transferred onto sapphire substrates under the same experimental conditions. The optical absorption of the sample is thus calculated as $A(\lambda) = 1 - R(\lambda) - T(\lambda)$. The advantage is that the optical absorption spectral lineshapes obtained by this method can be analyzed directly without taking into account interference effects that arise from multiple interfaces in such samples [44]. These interference

effects remain visible within the $R(\lambda)$ and $T(\lambda)$ spectra, as shown in the low-temperature ($T = 5$ K) comparison alongside the $A(\lambda)$ spectrum in Fig. S13. Most notably, any asymmetric lineshape is not apparent in the $R(\lambda)$ and $T(\lambda)$ spectra, however, the asymmetry is unambiguously present in the $A(\lambda)$ spectrum on the X_A peak.

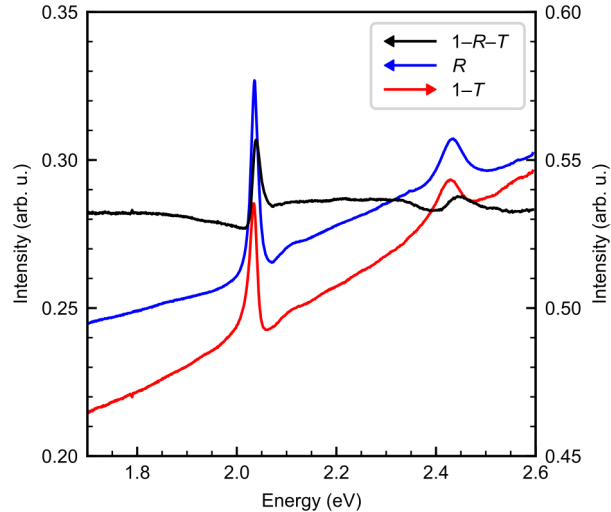


FIG. S13. Optical spectroscopy of graphite-encapsulated WS_2 monolayer on sapphire substrate measured at 5 K, highlighting the evident asymmetric lineshape in the absorption spectrum ($1 - R - T$) in black, and less evident in both the reflectance (R) and transmittance ($1 - T$) spectra in blue and red, respectively.

S4. Effect of Temperature

Fano-type resonance at the X_A transition in MoSe_2 monolayers with a negative asymmetry parameter (dip at higher energy) by reflectance contrast at $T = 5$ K has been previously reported [27]. The progressive transition towards a Lorentzian lineshape and disappearance of the charged exciton (trion, X^*) peak with increasing temperature proposes the unusual lineshape is caused by the interaction of the ground-state $1s X_A^0$ with the quasi-continuum of the trion X^* excited states. The binding energy of trions are of the order of ~ 35 meV for WS_2 , and therefore diminishes in oscillator strength towards room-temperature [44].

Room-temperature EELS measurement on identical areas of the $\text{Gr}/\text{WS}_2/\text{Gr}$ heterostructure from Fig. 1(a) continues to exhibit the asymmetric Fano-lineshape as presented in Fig. S14(b), albeit with significantly lower oscillator strength. A comparison of WS_2/Gr and freestanding WS_2 monolayer between low- and room-temperature also identifies the same decrease in EELS signal intensity, in addition to the expected exciton linewidth broadening, shown in Fig. S14(c) and (d). The persistence of the asymmetric lineshape at room temperature for the $\text{Gr}/\text{WS}_2/\text{Gr}$ heterostructure suggests that the observed asymmetric lineshapes do not originate from coupling to trions.

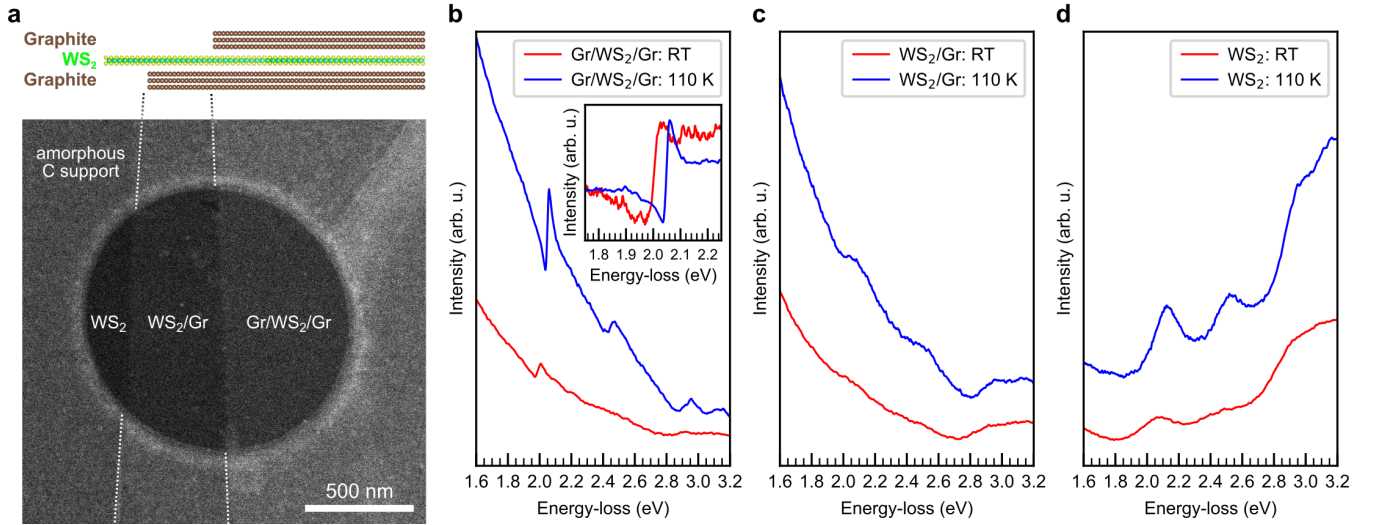


FIG. SI4. An area including WS_2 , WS_2/Gr , $Gr/WS_2/Gr$ measured at two different temperatures. (a) annular dark-field (ADF) image of the region with its partition on top; (b) the spectra of $Gr/WS_2/Gr$, the inset is background subtracted by the Lorentzian tail of graphite; (c) The spectra of WS_2/Gr ; (d) the spectra of freestanding WS_2 monolayer.

S5. Effect of TMD Layer Thickness

Discrete excitonic transitions in few-layered TMDs encapsulated in thin graphite also exhibit identical asymmetric lineshapes. For example, the X_A and X_B exciton peaks from the $Gr/WS_2/Gr$ heterostructure with 1–2 layered WS_2 [shown in Fig. SI5(a)] are clearly asymmetric for both WS_2 layer thicknesses. The graphite encapsulation thickness is the same at both monolayer (1L) and bilayer (2L) regions, as such, their asymmetric lineshape does not change significantly as a function of TMD layer thickness.

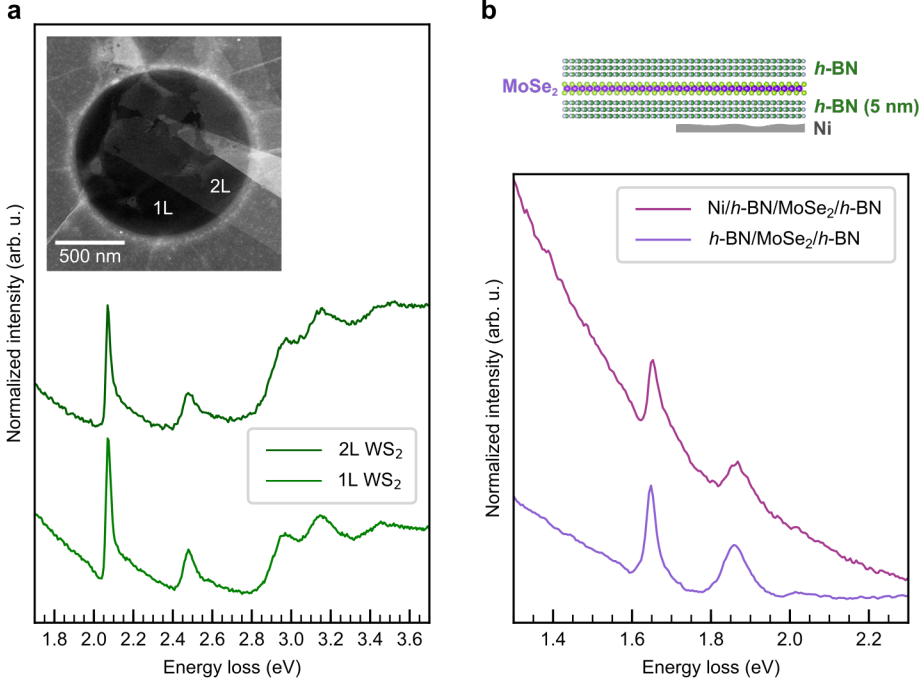


FIG. SI5. **Near-field coupling of graphite-encapsulated TMD few-layers and *h*-BN encapsulated TMD monolayers with other metals.** (a) EELS spectra of monolayer (1L) and bilayer (2L) WS₂ encapsulated between graphite flakes, inset with the annular dark-field (ADF) image of the measurement regions marked; (b) EELS spectra of *h*-BN encapsulated MoSe₂ monolayer with and without Ni metal on the 5 nm-thick *h*-BN side also showing the same asymmetric lineshape.

S6. Coupling between TMD and Other Metals

Bulk metallic thin films, such as nickel (Ni), deposited onto a continuous silicon nitride support can be used to mimic typical electrical contacts in optoelectronic devices while allowing variable TMD/metal interlayer separation from the *h*-BN spacer thickness. This configuration is also optimal for maintaining the TMD atomic flatness by the use of the *h*-BN [16]. *h*-BN encapsulated MoSe₂ monolayer was deposited onto a 4 nm-thick Ni thin film, with the bottom *h*-BN spacer of 5 nm. The EELS spectrum in the presence of Ni also exhibits clear asymmetric lineshape in Fig. SI5(b).

S7. Effect of Graphite Thickness

TABLE I. **Encapsulation layer thicknesses and parameters of the dielectric function of TMDs and graphite for various vdWH measured.** Total encapsulation layer thicknesses [t_{sub}] are determined experimentally (using EELS log-ratio method [42] and calculated values for mean free path [43] in the case of mixed material encapsulation, both graphite/ h -BN thickness are listed separately. Values of $\epsilon_{\text{graphite}}$ are represented by the in-plane response of graphite taken from literature [30] and approximated by a constant in a given energy range, the rest of the parameters are found via fitting to the form of a Lorentz oscillator.

vdWH configuration	t_{sub} (nm)	ω_0 (eV)	f_{TMD} (eV^2)	γ (meV)	$\epsilon_{\text{graphite}}$	t_{TMD} (nm)
graphite/WS ₂ /graphite	17.5	2.05	1.6	30	5.34 + 8.74i	0.6
graphite/WS ₂ /graphite	6.1	2.05	2.8	15	5.34 + 8.74i	0.6
graphene/WS ₂ /graphene	0.69	2.06	0.6	30	5.34 + 8.74i	0.6
graphite/MoSe ₂ /graphite	8.3	1.65	3.0	10	5.91 + 10.13i	0.6
graphite/MoSe ₂ / h -BN	4.1, 9.6	1.66	3.0	20	5.91 + 10.13i	0.6
h -BN/MoSe ₂ / h -BN	28.6	1.61	3.0	30	-	0.6
graphite/WSe ₂ /graphite	20.2	1.71	10	20	5.84 + 9.92i	0.6
graphite/WSe ₂ / h -BN	20.9, 45.7	1.72	10	20	5.84 + 9.92i	0.6
h -BN/WSe ₂ / h -BN	32.7	1.69	5.7	40	-	0.6

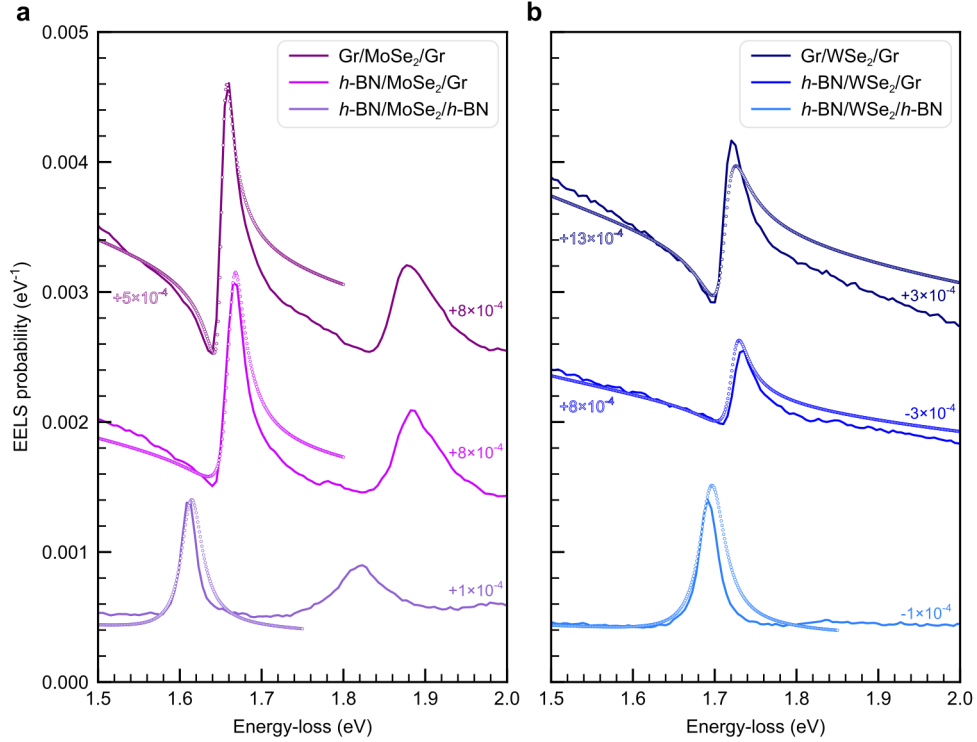


FIG. SI6. **Comparison of experimental EELS spectra with the theoretical model.** EELS spectra of mixed graphite and/or h -BN encapsulation of (a) MoSe₂ and (b) WSe₂ monolayers from experiments (solid lines) and the 2D optical conductivity model (dots) obtained by using Eq. (1). The experimental data is plotted as EELS probability $\Gamma_{\text{EELS}}(\omega_i) = I(\omega_i) / \int_0^{\omega_i} I(\omega) d\omega$, where $I(\omega)$ is the measured EELS intensity at energy ω . All experimental spectra are reproduced from Fig. 1(b) and (c) following the same color-code. Vertical offsets for selected experimental and modelled curves have been added for clarity, and are noted adjacent their curves in solid and open color-coded text, respectively.

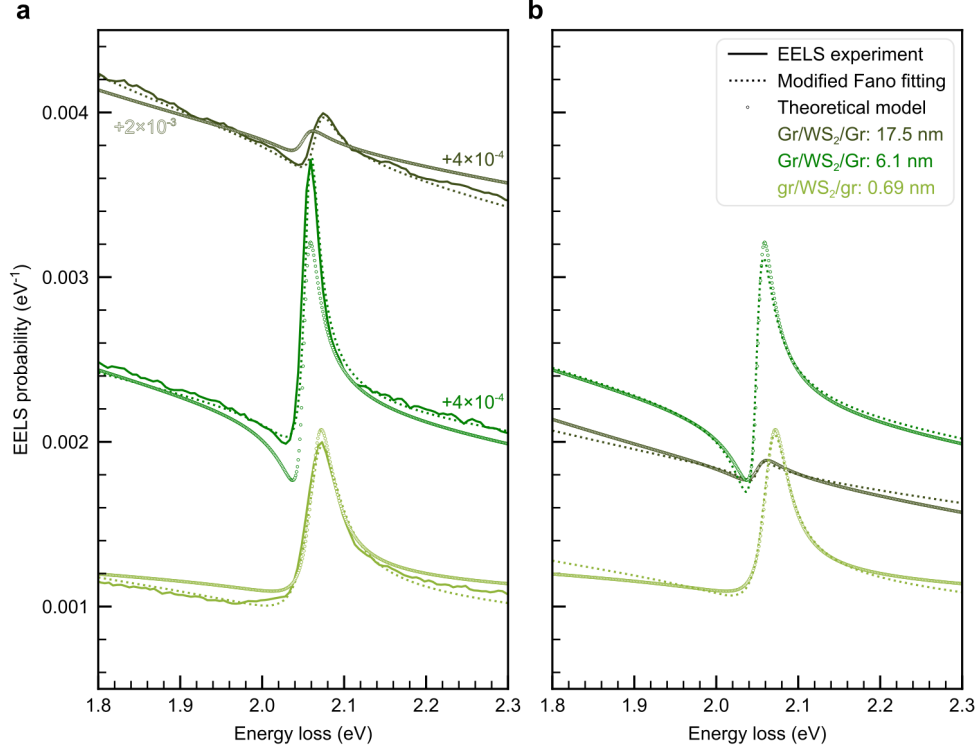


FIG. SI7. **Comparison between experimental and modelled EELS spectra of monolayer WS₂ of different graphite thickness with their respective fitting with the modified Fano function.** (a) EELS spectra of WS₂ monolayer encapsulated between two graphite layers of total thickness of 17.5 nm (thick), 6.1 nm (thin), and two graphene layers (0.69 nm) from experiments (solid lines), the 2D optical conductivity model (dots) obtained using Eq. (1), and fitting of the experimental data with the modified Fano-like function (dotted lines) in Eq. (2). (b) Modelled EELS spectra (dots), reproduced from (a) obtained using Eq. (1) and its fitting with the modified Fano-like function given in Eq. (2) (dotted lines). The experimental data is plotted as EELS probability $\Gamma_{\text{EELS}}(\omega_i) = I(\omega_i) / \int_0^{\omega_i} I(\omega) d\omega$, where $I(\omega)$ is the measured EELS intensity at energy ω . The spectrum of the 6.1 nm (thin) Gr-encapsulation thickness is reproduced from Fig. 1(a). The fitted parameters used in the modified Fano-like function curves are listed in Table II. Vertical offsets for selected modelled curves and experimental/fitted curve pairings have been added for clarity in (a), and are noted adjacent to their curves in open and solid color-coded text, respectively.

TABLE II. **Graphite thicknesses and fitting parameters of the modified Fano function of graphite-encapsulated WS₂.** Total graphite/graphene encapsulation layer thicknesses [t_{sub}] are determined experimentally (using EELS log-ratio method [42] and calculated values for mean free path [43], and fitting parameters of the experimental and modelled EELS spectra using the modified Fano-like function from Eq. (2).

Configuration	t_{sub} (nm)	Exp.	Theory	Exp.	Theory	Exp.	Theory	Exp.	Theory	Exp.	Theory
		ω_0 (eV)		γ (meV)		a		q		b	
Gr/WS ₂ /Gr	17.5	2.066	2.05	14.906	10.0	0.013	0.004	-0.677	1.0	0.202	0.098
Gr/WS ₂ /Gr	6.1	2.056	2.05	10.488	10.350	0.086	0.052	-0.365	-0.752	0.187	0.175
gr/WS ₂ /gr	0.69	2.066	2.065	23.098	18.627	0.051	0.050	-0.356	-0.325	0.112	0.115

S8. Modified Fano-like function fitted to numerical calculations

To get insight into the behavior of the numerical model [Eq. (1)], the modified Fano-like function [Eq. (2)] has been fitted to calculations for a dielectric (within its energy band gap, $\text{Re}\{\epsilon(\omega)\} > 0$ and $\text{Im}\{\epsilon(\omega)\} > 0$) and a metal ($\text{Re}\{\epsilon(\omega)\} > 0$ and $\text{Im}\{\epsilon(\omega)\} < 0$). The extracted fitted parameters and the least squares error are plotted in Figs. SI8–SI9.

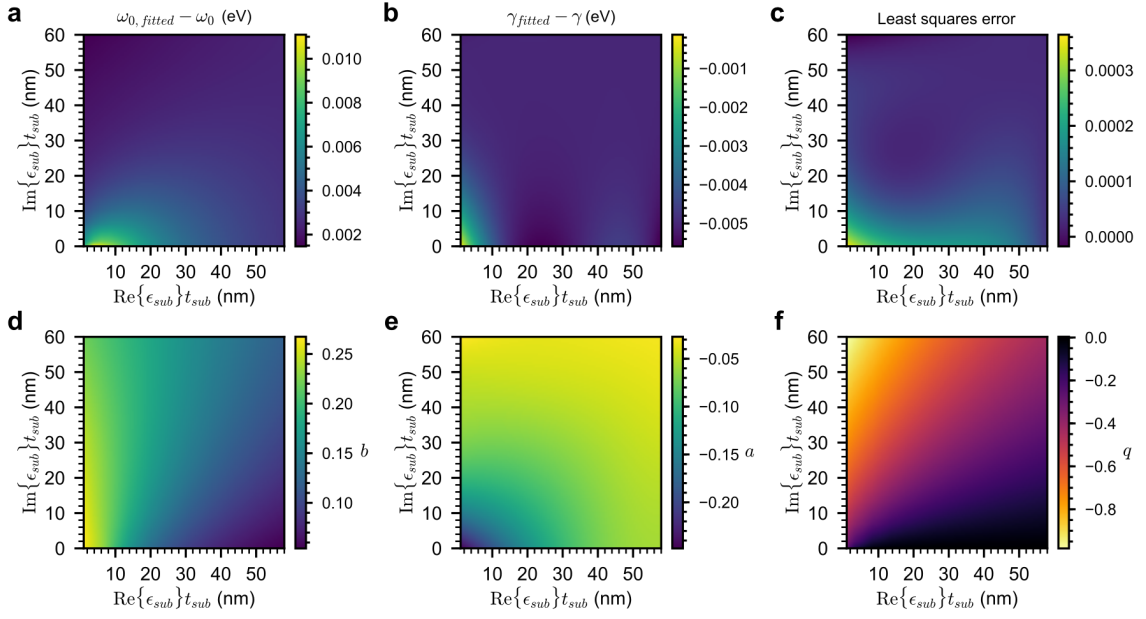


FIG. S18. **Modified Fano-like function fitting of the 2D model:** Evolution of fitted parameters (a) $\omega_{\text{fitted}} - \omega$, (b) $\gamma_{\text{fitted}} - \gamma$, (c) the least squares error, (d) b , (e) q , and (f) a of the modified Fano-like function [Eq. (2)] dependent on the substrate response, specifically by the real versus imaginary part of the dielectric function multiplied by its thickness, $\text{Re}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$ and $\text{Im}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$, respectively.

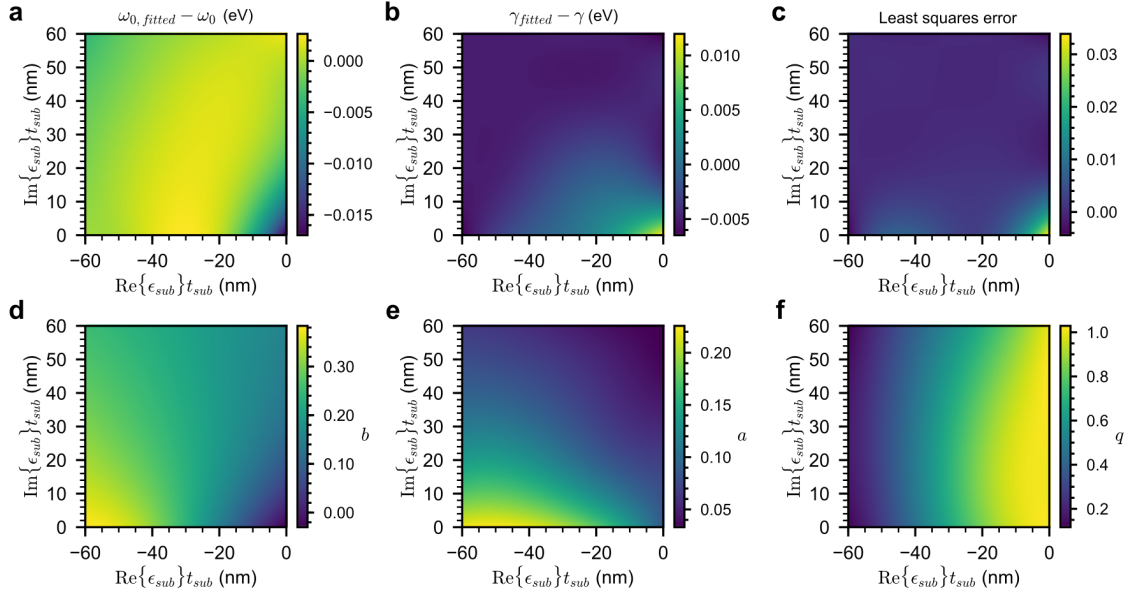


FIG. S19. **Modified Fano-like function fitting of the 2D model for a metallic substrate:** Evolution of fitted parameters (a) $\omega_{\text{fitted}} - \omega$, (b) $\gamma_{\text{fitted}} - \gamma$, (c) the least squares error, (d) b , (e) q , and (f) a of the modified Fano-like function [Eq. (2)] dependent on the substrate response, specifically by the real versus imaginary part of the dielectric function multiplied by its thickness, $\text{Re}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$ and $\text{Im}\{\epsilon_{\text{sub}}\}t_{\text{sub}}$, respectively.