

In situ Raman monitoring of studtite formation under alpha radiolysis in ¹⁸**O-labeled water**

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¹ In situ Raman monitoring of studtite formation under alpha radiolysis 2 in ¹⁸O-labeled water

3

11 **Abstract**

12 Studtite $[(UO₂)(O₂)(H₂O)₂]\cdot 2(H₂O)$ is a secondary phase precipitating during the alteration of uranium-13 bearing materials. The mechanistic link between the formation of uranyl and peroxide bonds in the 14 solid and the nature of the oxidizing species produced by water radiolysis remains to be elucidated. In 15 order to improve our understanding of these mechanisms, an original experimental methodology in 16 the presence of 18 O isotopes and Raman spectroscopy has been developed. It appears that there is a 17 direct chemical relationship between the peroxo ligands inside studtite and the peroxide entities of 18 H_2O_2 molecule into the solution. The link between H_2O_2 formation mechanism in solution by the 19 radiolysis of water, the nature of the radiation and the isotopy of the peroxo ligands inside studtite has 20 been described thanks to a coherent set of experimental data. For the uranyl UO_2^{2+} ions, the 21 characterization of its isotopy allows to specify the mechanism of oxidation at the $UO₂$ /water 22 interface. The isotopic configurations observed for the uranyl ion inside studtite, can be explained by 23 assuming an oxidation mechanism of $UO₂$ involving both a simple transfer of electrons by interaction 24 with H_2O_2 and the incorporation of oxygen atoms from the solution into the fluorite structure via OH \degree 25 radicals.

26 **Keywords**: Nuclear fuel; Oxidative dissolution; Uranium dioxide; Irradiation; Hydrogen peroxide;

27 Spectroscopy

1. Introduction

30 Water in interaction with heterogeneous nuclear materials such as spent fuel (UO₂ or (U,Pu)O₂ based ceramic), fuel debris or molten nuclear fuel (corium) produces hydrogen peroxide by radiolysis [1-4]. Interaction between the surface and hydrogen peroxide induces oxidative dissolution of uranium oxide 33 and the precipitation of U(VI) secondary phases such as studtite $[(UO_2)(O_2)(H_2O)_2] \cdot 2(H_2O)$, a common uranium peroxide that generally forms under strongly oxidizing conditions. It has been the subject of important studies over the past 20 years as it can potentially incorporate radionuclides, play a protective role during spent fuel alteration or even control uranium solubility in solution [3-9]. Studtite 37 has also been observed in natural deposits such as barite solid solutions incorporating 226 Ra and associated with other uranyl minerals such as rutherfordine or uranophane [10]. The crystallographic structure of this uranium peroxide is well known [11]. It is composed of ribbons of U-coordination 40 polyhedra with shared edges, which are composed of O_2^2 bridging peroxo ligands.

 There are also numerous studies on the stability of uranium peroxides as a function of solution chemistry, irradiation conditions and temperature [12-16]. For example, with regard to solution 43 chemistry, the presence of carbonate ligand and H_2O_2 in groundwater increase studtite dissolution and 44 uranium (U) release [12]. The stability under γ and α irradiation of studtite, and more generally of uranyl peroxide-based solid materials, with various structural topologies, has also been studied by XRD 46 and Raman spectroscopy [13, 14]. Studtite is particularly affected by α irradiation which induces the

 formation of amorphous uranyl peroxide [14]. In addition, an increase in temperature leads to the loss 48 of water molecules and the formation of metastudtite $(UO_2)(O_2)(H_2O)_2$, or even that of an amorphous

compound [15, 16].

 Despite of all these studies, the link between the radiolytic products of water (i.e. radicals and molecular species) and the formation of the uranyl ion and peroxo ligands within studtite needs to be better understood, requiring further investigation. Recently, a study coupling Raman spectroscopy 53 with γ irradiation of ¹⁸O labelled water has shown the possibility of linking the formation of radiolysis products to the main structural entities of studtite, namely peroxo ligands and uranyl ion [17]. The goal 55 of the present work is to perform this study under α irradiation by monitoring studtite precipitation 56 using in-situ Raman spectroscopy and ^{18}O labelled water irradiated by 5 MeV He²⁺ particles. This can be achieved using the in-situ irradiation Raman set-up implemented at the CEMHTI cyclotron beamline, which has already been used in previous studies [18, 19]. The originality of this work lies in 59 the use of 18 O labelled water. Indeed, the gathering of Raman spectra coupled with solution analysis is expected to provide new insights to propose studtite formation mechanisms. The respective roles of hydrogen peroxide and radicals in the formation of peroxo and uranyl bonds will be particularly evaluated. The nature of isotopic exchanges under irradiation will provide a better understanding of the effects of Linear Energy Transfer (i.e. high or low LET effect) in studtite formation and, more broadly, in the mechanism of oxidative dissolution of uranium dioxide.

66 **2. Materials and methods**

67 **2.1. Sample preparation**

68 The target used for the alpha radiolysis experiment came from a sintered pellet of depleted $UO₂$, 69 manufactured at CEA Cadarache, and was obtained after wire saw cutting and mirror polishing with 70 diamond paste. The resulting $UO₂$ disk measured 8 mm in diameter and was 277 μ m thick, stored in 71 air within membrane storage boxes. Prior to leaching, three cycles of pre-leaching of the experimental 72 device (cell and pellet) were carried out in a water solution (MilliQ Water) in the presence of 10⁻³ mol.L⁻ 73 N NaHCO₃, in order to ensure the stoichiometry at the sample surface by removing the pre-oxidized 74 surface layer. The pre-leaching steps lasted 1 hour, with the exception of the last one, which was 75 extended to 14 hours. This pre-leaching was followed by three washing stages in the presence of the 76 leaching cell, two washing stages were performed with pure water (MilliQ Water) and the last one was 77 performed with 18 O-labelled water. 78 In order to carry out a complementary experiment to study the alteration of $UO₂$ without irradiation

79 but in the presence of ¹⁸O-labelled hydrogen peroxide, a second UO₂ sample was synthesized. This

80 sample was prepared according to the process described by Martinez et al., i.e. by precipitating 81 uranium (IV) in hydroxide form [20]. After washing, the precipitate obtained was deagglomerated then

82 heated at 700°C for 4h under $Ar/H₂$ atmosphere. Finally, the powder was pelletized and sintered at

83 1550°C for 6h under Ar/H₂. The resulting sintered UO₂ pellet was fragmented. One of the fragments 84 was annealed under $Ar/H₂$ atmosphere at 1100°C for 5h to ensure U/O stoichiometry. After synthesis

85 and prior to leaching, the sample was characterized by Raman spectroscopy. It then underwent a 4

86 stages pre-leaching cycle in the presence of 10^{-3} mol.L⁻¹ NaHCO₃. Pre-leaching lasted 1 hour, with the

87 exception of the last one, which was extended for 12 hours. These steps were followed by two washing 88 stages with deionized water and a last one with 18 O labelled water.

89 **2.2. Leaching experiments**

2.2.1. Leaching of UO² sample under α irradiation in presence of H² ¹⁸ 90 **O**

91 **Irradiation conditions.** An external helium irradiation beam was used to irradiate the $UO₂/H₂¹⁸O$ 92 system. This irradiation was performed at the CEMHTI cyclotron, using 45 MeV $4He^{2+}$ ions. The He²⁺ ion 93 beam was collimated to obtain a 6 mm diameter centred on the $UO₂$ target. The 277 μ m thickness of 94 the UO₂ target was chosen according to the results of the SRIM software [21], which evaluated the 95 trajectory and the dissipated energy of ions passing through a material. The SRIM-calculated energy of 96 4 He²⁺ ions at the UO₂ disk exit was 7.7 ± 0.9 MeV, with UO₂ density of 10.74 g.cm⁻³ (98 % theoretical 97 density) and taking into account the presence of three 26.8 μ m-thick titanium foils at the beam 98 controller. Atom displacement energies have been set at 20 eV for oxygen and 40 eV for uranium, 99 based on calculation of Soullard et al. [22]. The calculated projected range of these 7.7 MeV ${}^{4}He^{2+}$ ions, 100 which induce water radiolysis in H₂¹⁸O, was 71.8 ± 0.8 µm. The irradiation lasted 6 hours at an intensity 101 of 10 nA, only during the first 35 minutes, then the intensity was decreased to 5 nA to limit the 102 appearance of dihydrogen bubbles. The corresponding fluxes were 5.5×10^{10} α .cm⁻².s⁻¹ for 5 nA and 103 1.1×10^{11} α.cm⁻².s⁻¹ for 10 nA, to reach a final fluence of 1.3 \times 10¹⁵ α.cm⁻². Based on fluence and SRIM 104 simulation data [21], the displacement per atom (dpa) within the uranium dioxide layer probed by 105 Raman spectroscopy (γ 1 µm) was calculated at 4.0 \times 10⁻⁴ dpa. This value corresponds to a measure of 106 the number of atoms ejected from their initial position in the crystal lattice, following ballistic 107 collisions. It provides us a quantitative means of assessing the damage inflicted on the material by 108 irradiation. However, it does not take into account defect recombination.

109 **Experimental set-up in the irradiation chamber.** This set-up allowing in-situ Raman monitoring of the 110 UO₂/H₂¹⁸O interface during and after irradiation is presented in Figure 1. It consists of a leaching cell, 111 manufactured in a polymer material resistant to radiolytic species attacks (PCTFE) in which the $UO₂$ 112 target is placed and held in place by a system of screwed rings to ensure watertightness. The leaching

- 113 solution used was ¹⁸O-enriched deionized water (97.17% ¹⁸O, 0.34% ¹⁷O, 2.49% ¹⁶O, INNOVA-CHEM,
- 114 France). During irradiation, the α beam is transmitted through the UO₂ specimen inducing water
- 115 radiolysis. Finally, a 40 mm diameter quartz window located on the opposite side of the UO₂ sample
- 116 provides optical access for in-situ Raman monitoring of the $UO_2/H_2^{18}O$ interface.

 Figure 1. In situ experimental set-up combining a leaching cell under - irradiation with Raman monitoring: a. picture of the instrumental set-up; b. schematic diagram of the set-up, inspired from [18].

 In-situ Raman spectroscopy. Raman measurements were performed continuously on the direction 122 parallel to the ⁴He²⁺ beam direction (**Figure 1**), using a Renishaw RA100 spectrometer equipped with a He-Ne excitation laser (633 nm). The laser beam was delivered to the irradiation cell via a multimode optical fiber and a Renishaw's RP10 probe, equipped with a Mitutoyo long working distance (30.5 mm) objective which offers a 0.28 numerical aperture and with a 20x magnification. The 7 mW laser power 126 measured at the sample surface allows a very good compromise between $UO₂$ disk laser damage resistance and Raman efficiency. The spectra were recorded with a 1800 groove/mm holographic 128 grating, allowing 4 cm⁻¹ spectral resolution and offering simultaneous observation from 310 to 1244 cm⁻¹, and an accumulation time of 2 minutes for each spectrum. Raman monitoring was carried out during a 21-hour leaching period, enabling the continuous observation of studtite precipitation, including post-irradiation precipitation induced by radiolytic products remaining in solution after a 6- hour irradiation period.

- When the leaching test was stopped, the solution was drained, followed by 3 successive washes of the cell with pure water. The pellet was then taken off and carefully dabbed with Kimtex© paper to remove
- water and stop surface alteration, ensuring no rubbing occurred during the process.

136 2.2.2. Leaching of UO₂ in presence of H₂¹⁸O₂

 To understand the role of oxidizing species during water radiolysis, an additional leaching experiment 138 was carried out by replacing the presence of these radiolytic species produced in 18 O-labelled water 139 with a $H_2^{18}O_2$ solution. The 10⁻³ mol.L⁻¹ $H_2^{18}O_2$ solution was prepared from a commercial solution 140 (Sigma-Aldrich Hydrogen Peroxide - $^{18}O_2$ Solution - 2-3% in H₂O, 90 atom % of ^{18}O) diluted with H₂¹⁸O 141 water. The H₂O₂ concentration was verified in the resulting solution by the Ghormley method [23]. The 142 sample was weighed and placed in a 10 mL leaching tube for 48 hours in the presence of 5 mL of 10^{-3} 143 mol.L⁻¹ H₂¹⁸O₂ solution. After leaching, the samples were characterized by Raman spectrometry, while the leachates were filtered to 3 kDa by centrifugation before being analysed by ICP - OES.

2.3. Post-mortem characterization

 ESEM. ESEM (Environmental Scanning Electron Microscopy) characterization was carried out to observe the morphology of studtite precipitated on the pellet surface, using a ThermoFischer Scientific

- Quattro S SEM. Micrographs were recorded using a voltage varying between 1 and 10 kV and a 149 pressure in the chamber between 10^{-3} and 10^{-4} Pa.
- Following an initial observation of the pellet's surface, an analysis of its cross-section was conducted.
- Prior to this second ESEM observation, the irradiated sample was vacuum-coated in resin, cut with a
- Buehler IsoMet 1000 milling cutter, then polished with a Buehler Beta polishing machine using
- diamond paste 1 µm to achieve a mirror-polished surface. To prevent charge accumulation onto the
- surface, the cross-section was coated with carbon prior to SEM observation.
- **X-Rays Diffraction.** Structural analysis of the solids was carried out by X-ray diffraction (XRD) using a 156 Bruker D8 ADVANCE diffractometer (Cu-K α_1 radiation, λ =1.5406 Å, 40 kV, 40 mA, LinxEye detector). XRD patterns were collected at room temperature over the 5-100° (2θ) range, using a step size of
- 0.009° and a duration per step equal to 1 s.
- **ICP-OES/ICP-MS**. Concentrations of uranium released in the leachate were first determined using a iCAP 7000 series optical plasma emission spectroscopy (ThermoFischer Scientific). The analysis was 161 performed considering the results obtained at λ = 385.958 nm. In a second step, solutions with uranium 162 concentrations below the limit of quantification by ICP-OES (LOQ (U) \sim 100 ppb) were analysed by ICP-163 MS (LOQ (U) \sim 50 ppt). The elemental concentrations of uranium was calculated as the average of three replicates. The instrument used was a ThermoFischer Scientific iCAP RQ. For all elemental analyses, a calibration curve was produced using a range of diluted samples prepared from a certified solution, framing the concentration range under consideration. All samples were diluted with 0.2 167 mol.L⁻¹ HNO₃ prior to analysis, to limit uranium adsorption onto the walls of the vials.
- **Raman spectroscopy.** Post-mortem µ-Raman analyses were carried out using the LabRam ARAMIS 169 spectrometer (Horiba Jobin-Yvon) to study the alteration of UO₂ sample leached in H₂¹⁸O₂ solution. The instrument is equipped with He-Ne excitation laser (632.8 nm) and a Nd:YAG laser (532 nm). The laser 171 beam was focused onto the sample surface using an Olympus BX 41 microscope equipped with a 100 \times 172 SLMP objective. The resulting spot size was of the order of 1 μ m². During spectra acquisitions, the laser (532 nm) was filtered down to 0.3 mW (measured at the sample surface) to avoid any sample 174 modification. The spectra were recorded using the 1800 grooves. $mm⁻¹$ holographic grating and taking into account accumulation time of 4 minutes per spectrum. These parameters were chosen to ensure a good signal-to-noise ratio. For the sake of reproducibility, different remote areas were analysed to 177 confirm the homogeneity of the sample surface. Once the spectra have been recorded, they were processed using OriginPro (Version 2023b. OriginLab Corporation, Northampton, MA, USA) software. A baseline correction was performed and the spectra were then normalized to the band of maximum intensity.
-

182 **3. Results**

183 **3.1. In-situ Raman monitoring of studtite precipitation under alpha irradiation**

184 Uranium dioxide presents a fluorite-type crystal structure (CaF₂), characterized by a space group 185 $Fm\bar{3}m$ (O_h^5) . In this structure, the U⁴⁺ cations adopt a face-centred cubic (f.c.c.) arrangement, while 186 the anions (O^2) occupy the tetrahedral positions [24]. In accordance with group theory, a triply 187 degenerate Raman-active vibrational mode at 445 $cm⁻¹$ is observed in the initial Raman spectrum. This 188 band, called T_{2g} , results from the antiphase oscillation of stretching oxygen atoms around uranium [25]. The initial spectrum admits other bands, including defect bands between 500 and 700 cm-1 189 (**Figure** 190 **2**). A broad band is also observed at 1150 cm⁻¹. This band has been assigned to a second-order 191 longitudinal optic phonon (2LO) [24]. Finally, a band at 911 cm⁻¹ was also identified, generally 192 attributed to the 2TO_R mode [24, 26]. All spectra recorded during the leaching test were normalized in 193 intensity with T_{2g} and then used to acquire a temporal Raman map (Figure 2). During the first few 194 minutes of irradiation, dihydrogen was produced by radiolysis of water within the cell. This led us to 195 make short interruptions of the irradiation at 10 nA to evacuate the bubbles present and to lower the 196 beam intensity. During these stops, the recording of Raman spectra were stopped. These stops 197 correspond to the grey lines observed in **Figure 2**.

199 *Figure 2. 2D time-resolved Raman spectra alteration with the pristine UO² signal (bottom) and the final spectrum showing* 200 *studtite precipitation (top).*

201 The characteristic bands of the UO₂ compound were observed in all the Raman spectra recorded during 202 the entire experiment. The final spectrum was still composed of the $UO₂$ fingerprint, in which no 203 position and FWHM variation was detected in the T_{2g} and 2LO bands compared to the initial spectrum. 204 No significant evolution of the FWHM was also observed by Gutierrez et al. for a dpa lower than 0.03 205 [27]. However, a significant evolution of the band associated with defects was observed between 500 206 and 700 cm⁻¹. The intensity of this defect band composed of three contributions: U_1 , U_2 and U_3 is 207 reported in **Figure 3a**. The increasingly pronounced appearance of these three defect bands during 208 irradiation is clearly evidenced. Their growth was due to the ballistic impact of the He $^{2+}$ ions in the 209 fluorite structure. The latter is composed of 3 defect bands: Firstly, the U_1 band is present at 530 cm⁻¹, 210 which assignment is still discussed [18, 28]. Indeed, it is sometimes associated with the distortion of 211 the cubic lattice by interstitial oxygen. Secondly, the U₂ band at 577 cm⁻¹, which is assigned to the first-212 order longitudinal optical phonon (1LO) mode, active in IR and observable in Raman in the presence of 213 defects in the fluorine structure, due to loss of symmetry. This loss of symmetry resulted from the 214 appearance of oxygen clusters and the change in ionic radius between U (IV) and U (V) during oxidation 215 [29, 30]. Lastly, the U₃ band at 634 cm⁻¹ which is characteristic of the presence of oxygen cuboctahedra, 216 is observed [28, 29, 31]. When irradiation was stopped, the increase of bands intensity stopped and 217 the phenomenon was not reversible. No saturation appeared to have been reached before the end of 218 irradiation. This observation is in good agreement with the results obtained by G. Guimbretière et al. 219 Indeed, saturation was not yet observed for fluence values of 1.3×10^{15} α.cm⁻² [32]. The displacement 220 per atom (dpa) previously calculated tells us that alpha particles are transmitted with a very low impact 221 on UO₂, revealing the extreme sensitivity of defect bands to irradiation-induced damages. In addition, 222 in Figure 2, the growth of 3 bands at 774, 790 and 817 cm⁻¹ can be noted. These bands correspond to 223 the $[(UO₂)(O₂)(H₂O)₂]·2(H₂O)$ studtite compound and were already detected on ¹⁸O labelled studtite 224 [11, 17]. The observation of this phase under our conditions is consistent with the results obtained by 225 Corbel et al. [33]. The intensity of the studtite characteristic bands is plotted against time in **Figure 3b**. 226 The increase in their intensity clearly shows the formation of this phase from the very beginning of the 227 experiment, and its subsequent growth during the experiment. The evolution of the Raman response 228 of this phase allowed us to establish that studtite precipitation occurred according to a linear kinetic. 229 It was observed that the linear evolution of the Raman response continues while the pellet is no longer 230 irradiated, until around 1000 minutes. This result is in good agreement with the observations by R. 231 Mohun who has reported that studtite kept growing despite the stop of irradiation [34]. After 1000 232 minutes dissolution experiment, precipitation slowed down.

234 *Figure 3. Monitoring the time-dependent Raman response of bands normalized with T2g during the leaching time. Evolution* 235 *of the area of defect bands (a) and evolution of band area associated with studtite (b).*

236 **3.2. Analysis of the solutions and post-mortem characterization of the studtite layer**

- 237 The study of UO₂ alteration by water radiolysis is well documented in the literature [4, 17, 33, 35]. 238 Consequently, studtite precipitation under these experimental conditions can be anticipated and 239 should occur as soon as saturation of this phase in solution is reached. To correlate uranium release 240 with studtite precipitation, the concentration of uranium in solution during the leaching test was 241 measured. First, the pre-leaching steps ensured that the oxidized surface layer of the UO₂ pellet was
- 242 removed. Thus, the effectiveness of the washing steps carried out after this pre-leaching stage was
- 243 monitored, verifying that all the uranium present in the leach cell was recovered. This then enabled a
- 244 balance to be drawn up for the uranium in solution. For all these reasons, the uranium concentration
- 245 was determined after each stage (**Table 1**).
- 246 A low concentration of uranium released during the pre-leaching stage was noted, indicating a weak 247 pre-existing oxidized layer of U(VI). This observation seems to be in agreement with the initial Raman 248 response of the pellet, which did not show significant defect bands (in particular the U_3 band).
- 249 After the leaching test, the uranium concentration of the leaching solution was measured at (1.1 ± 0.1) 250 \times 10⁻⁴ mol.L⁻¹, from which a quantity of uranium released into solution equal to (5.9 ± 0.5) \times 10⁻⁷ mol 251 was deduced. These release values are in agreement with the release previously obtained by Corbel et
- 252 al. [33].

 A speciation calculation was carried out using CHESS software [36] and the ThermoChimie database [37]. At the end of the leaching stage, the saturation index in uranium was determined to be 2.58 at a pH of 3. This value confirms that the conditions in the solution were oversaturated with respect to 256 studtite during leaching, resulting in the precipitation of this phase.

- 257 The thickness of the studtite layer was estimated using the data reported by Corbel et al. under similar 258 conditions [33]. Knowing the output energy of the alpha particles (7.7 MeV) and the fluence at the end 259 of the experiment (1.3 × 10¹⁵ α.cm⁻²), it can be calculated that the total energy deposited in the solution 260 is 456 J. Corbel et al. have established a relationship between this deposited energy and H_2O_2 261 concentration, allowing us to estimate that the H_2O_2 concentration present at the end of irradiation 262 was $C_{H_2O_2}$ ≈ 6 × 10⁻³ mol⋅L⁻¹ [33]. From the H₂O₂ concentration and a second relationship 263 established by Corbel et al, a layer growth rate of 100 nm.h⁻¹ was able to be estimated, enabling the 264 determination of the precipitated studtite layer's thickness at 2.1 μ m. With the knowledge of studtite's 265 density (3.64 g.cm⁻³) [38], it can be deduced that the mass of studtite that was precipitated on the 266 pellet's surface amounts was approximately 1 mg.
- 267 *Table 1. Uranium concentration determination in solution during pre-leaching, washing and leaching stages when leaching* **268 D2** *UO***₂ irradiated with** α **particles**

270 Once the radiolysis experiment was completed, the UO₂ disk was first characterized by X-ray diffraction 271 (XRD) to identify crystal phases. The X-ray pattern obtained (**Figure 4**) in reflectivity mode with normal 272 incidence, revealed the coexistence of two different phases: $UO₂$ and studtite

^{}* 269 *6 hours of irradiation*

273 $[(UO₂)(O₂)(H₂O)₂]·2(H₂O)$. This compound crystallizes in a monoclinic structure with space group C2/c [11].

 The recorded diffractogram is compared with the studtite diffractogram reported by Burns et al. [11]. Thus, the XRD diffractogram revealed a strong anisotropy in the intensity of the lines, marked by a 277 crystal growth surface favoured along the (-202) plane, associated with a position of 21.12° (20). The ratio of peak amplitudes (1,1,0)/(-2,0,2) is ∼5.34, whereas it is ∼0.13 for our sample [11]. Thus, it can be deduced, as shown in **Figure 4**, that studtite chains are oriented parallel to the axis of needle growth, this axis corresponding to the c-axis. The orientation of studtite within a needle is illustrated in **Figure 4a**, where the (-202) plane (**Figure 4b**) is arranged perpendicular to the growth direction while the (200) plane (**Figure 4c**) is aligned parallel along the growth axis. The work of Corbel et al. did not identify any preferential orientation in the formation of studtite under alpha irradiation [33], however, our results are consistent with the observations of Schlegel and Jegou who reported the presence of a preferential orientation of crystal growth along (-202) in the case of precipitation without irradiation [39]. Two hypotheses may be put forward to explain this contradiction. On the one hand, if irradiation 287 does indeed inhibit studtite formation with preferentially oriented textures, our finding could result from the fact that, in our experiments, studtite growth mainly occurred in the absence of irradiation, during the last part of the experiment, when the beam was off. This could lead to a diffractogram that is predominantly representative of studtite that grew without the influence of irradiation. On the other hand, it's important to point out that studtite growth duration reported by Corbel et al. was significantly shorter than that observed in our study, with a maximum of 6h for continuous irradiation. However, research by Schlegel and Jegou has shown that the orientation effect along the (-202) plane is only perceptible after six hours of precipitation [39]. This time difference could therefore play a crucial role in whether or not orientation effects are observed, providing a second hypothesis to explain the observed contradiction.

-
- **300** *Figure 4. XRD diagram and structural analysis of studtite growth: a. XRD pattern of UO₂ pellet after leaching, showing the 301 <i>formation of studtite at the solid/solution interface (left) and schematic illust formation of studtite at the solid/solution interface (left) and schematic illustration of the structure of studtite during crystal growth (right); uranium atoms being at the center of the coordination polyhedra; b. View of the plans (−202) and (−404); c. View of the plan (200)*
- ESEM observations of the UO² disk reported in **Figure 5**, confirmed the formation of a homogeneous 305 crystalline layer characteristic of the studtite phase formed during the oxidative dissolution of $UO₂$ in presence of H2O² [33]. The area that was not in contact with the solution (**Figure 5a**) does not show any secondary phase or alteration marks on the surface. Only marks due to the polishing process such as scratches and grain tears are observed, whereas a layer covered with needle-like studtite crystals was formed on the area in contact with the solution (**Figure 5b**).
- In addition, the observation of the cross-section of the pellet (**Figure 5c**) allowed us to image the alteration layer and to determine the morphology of the crystal growth. A single homogeneous layer 312 of studtite, formed by needles grown perpendicularly to the UO₂ surface, with an average height of
- about 1.4 µm, was observed. The studtite thickness measured is in accordance with that which has
- 314 been previously estimated by leachate analysis (i.e. $2.1 \mu m$).
- The SEM observations of the studtite layer clearly indicate that the needles have grown according to a
- preferential orientation. This result is in good accordance with XRD data, where a texture marked by
- preferential orientation of crystal growth was observed (**Figure 4**).

319
320

320 *Figure 5.* ESEM micrographs of UO₂ leached pellet: a. Non-leached peripheral zone; b. Altered surface covered with a layer of studtite needle-like crystals; c. Cross-section of the pellet. of studtite needle-like crystals; c. Cross-section of the pellet.

322 **3.3. Analysis of studtite isotopic enrichment**

323 The use of 18 O enriched water and Raman spectroscopy offer a precise method to determine the origin 324 of the oxygen present in studtite and to deduce its formation mechanisms. In order to analyse the 325 enrichment of the different bonds in studtite, it was first necessary to determine the different Raman 326 band positions of ¹⁸O enriched studtite. To evaluate the theoretical vibration frequencies and deduce 327 the wavenumber associated with each ^{18}O enriched studtite bond, it was considered that an 328 intramolecular stretching vibration can be described by a harmonic oscillator model in which the 329 binding force is described by the Hooke's law [40].

$$
\tilde{\nu} = \frac{1}{2 \pi c} \sqrt{\frac{k}{\mu}} \tag{1}
$$

330

331 where \tilde{v} is the wavenumber (in cm⁻¹), c is the speed of light (in cm.s⁻¹), k is the binding force constant 332 (in kg.s⁻²) and μ corresponds to the reduced mass (in kg). There is a direct relationship between mass and bond vibration frequency. Isotopic labelling thus induces a shift in the Raman bands by modifying the mass of the original atoms. To determine the position of the bands after isotopic enrichment, the experimental position of the isotopically unlabelled bands was used and then the calculated shift was applied [17].

 In the case of studtite, its response to isotopic labelling is characterized by a behaviour similar to that of a free molecule, where each possible isotopic configuration is associated with a specific Raman response. This is in contrast to solids, which generally show a single band per bond, integrating the different contributions [17]. The calculated vibration frequencies associated with each of the different oxygen configurations within the peroxide and uranyl bonds of studtite are presented in **Table 2**.

342

*Table 2. Calculated Raman band positions associated with the contribution of the various bonds for O¹⁸ enriched H₂O₂<i>, studtite and UO₂²⁺ studtite and UO*₂²⁺

| Bond | Compound | Configuration | Experimental position $(cm-1)$ [17] | Calculated Raman band position for ¹⁸ O enriched compound $(cm-1)$ |
|-------------|----------|-----------------------|--|---|
| Peroxyde | H_2O_2 | $16O-16O$ | 876(1) | |
| | | $16O-18O$ | | 851 |
| | | $^{18}O^{-18}O$ | | 826 |
| | Studtite | $160 - 160$ | 863(1) | |
| | | $16O-18O$ | | 838 |
| | | $^{18}O^{-18}O$ | | 814 |
| Uranyl | $UO22+$ | $^{16}O = U = ^{16}O$ | 870(3) | |
| | | $160 = U = 180$ | | 845 |
| | | $18Q = U = 18Q$ | | 820 |
| | Studtite | $^{16}O = U = ^{16}O$ | 817(2) | |
| | | $160 = U = 180$ | | 793 |
| | | $^{18}O = U = ^{18}O$ | | 770 |

345

 Figure 6 shows the Raman spectra of studtite from three leaching experiments together with the reference spectrum of studtite. Pseudo-Voigt function fits of the Raman spectra presented in **Figure 6** 348 were performed on the bands present between 700 and 900 cm $^{-1}$ associated with studtite, in order to assign the corresponding bonds to each band (**Table 2**) and thus deduce the isotopic enrichment of studtite.

351 As mentioned above, the Raman spectrum of pellet subjected to leaching by alpha irradiated labelled 352 water shows a cluster of 3 bands between 700 and 900 cm⁻¹ (Figure 6a). The first band at 774 cm⁻¹ is

353 associated with the uranyl bond composed of two 18 O oxygen (18 O = U = 18 O). The second band at 790 354 cm⁻¹ is associated with the uranyl bond composed of one 18 O and one 16 O (16 O = U = 18 O). Finally, the 355 third band at 817 cm⁻¹ is attributed to the sum of the contributions of the uranyl bond composed of 356 two ¹⁶O oxygen (¹⁶O = U = ¹⁶O) and the peroxide bond composed of two ¹⁸O oxygen (¹⁸O – ¹⁸O). No 357 bands at 840 cm⁻¹ and 860 cm⁻¹ attributed respectively to the peroxide bond ^{18}O - ^{16}O and ^{16}O - ^{16}O 358 were observed despite the presence of ^{16}O (3 atom %) present in water.

359 For the stoichiometric UO₂ pellet submitted to a leaching by a 1mM H₂¹⁸O₂ + H₂¹⁸O solution (Figure 360 **6b**), the Raman spectrum showed a band at 772 cm⁻¹, corresponding to the uranyl $^{18}O = U = ^{18}O$ bond. 361 A second band was seen at 788 cm⁻¹, corresponding to the uranyl ¹⁶O = U = ¹⁸O bond. Finally, a last 362 band located at 811 cm⁻¹ was associated with the contribution of the uranyl ¹⁶O = U = ¹⁶O bond, but 363 also with that of the peroxide 18 O – 18 O bond. A slight band at 860 cm⁻¹ attributed to the peroxide 16 O 364 $-$ - ¹⁶O bond seemed to be present. This band is confused with the noise and the 2TO_R band at 925 cm⁻¹ 365 and was expected in these proportions due to the presence of ¹⁶O (10 atom% ¹⁶O) in the H₂¹⁸O₂ 366 enrichment. Sarrasin had noted for a pellet having undergone leaching by a gamma-irradiated labelled 367 water (Figure 6c) the presence of a significant band associated with the ¹⁶O - ¹⁶O peroxide bond while 368 the band at 817 cm⁻¹ was weak [17]. Finally, the reference spectrum of studtite without isotopic 369 labelling (Figure 6d) showed two bands at 817 cm⁻¹ and 860 cm⁻¹ corresponding to the uranyl (¹⁶O = U 370 $=$ 16 O) bond and the peroxide (16 O - 16 O) bond, respectively.

372 *Figure 6.* Results of deconvolution of studtite Raman spectra recorded for UO₂ after 1 day of α radiolysis (a), UO₂ after 2 days of leaching with H₂¹⁸O₂ (b) UO₂ after 1 day of y radiolysis, based on t **373** *<i>days of leaching with H*₂¹⁸O₂ (b) UO₂ after 1 day of γ radiolysis, based on the results of [17] (c) and studtite without **374** *any isotopic labeling (d)* any *isotopic labeling (d)*

371

375 In order to quantify the isotopic proportion of each band, the method proposed by Sarrasin was carried 376 out [17], which results are presented **Figure 7**. The model proposed by Sarrasin is a binomial 377 distribution in which x corresponds to the isotopic abundance of ¹⁶O while $1 - x$ corresponds to the 378 isotopic abundance of ^{18}O . Thus, for the uranyl bond, the respective probabilities of the different

379 configurations are as follows: x^2 for ¹⁶O=U=¹⁶O, $2x(1-x)$ for ¹⁶O=U=¹⁸O and $(1-x)^2$ for ¹⁸O=U=¹⁸O. The intensity of the three uranyl peaks is directly proportional to the probability of a given isotopic configuration. By knowing the probabilities of the configurations, the isotopic abundance can be determined from the intensity ratios of the uranyl bands. Applying this model at the end of the leaching 383 test of the UO₂ sample under α irradiation in H₂¹⁸O, an isotopic fraction of 79% ¹⁸O in the uranyl bond 384 and 97% 18 O in the peroxide bond was obtained. Refinement of the recorded spectrum of studtite 385 precipitated on stoichiometric UO₂ in the presence of $H_2^{18}O_2$, gave an isotopic fraction of 85% ¹⁸O in 386 the uranyl bond and 97% ¹⁸O in the peroxide bond. No ¹⁶O - ¹⁸O band was found at 840 cm⁻¹. For these experiments, the process of studtite formation seems to have produced only oxygen groups of identical isotopy, and this feature has persisted over time.

 Figure 7. Isotopic labeling fit of studtite bands by lorentzian function: a. UOx after 1 day with α radiolysis; b. UO² after 2 days of leaching with H₂¹⁸O₂. The isotopic enrichment of uranyl (blue) and peroxide (red) bonds is shown on each graph.

 During leaching, a significant change in the isotopic ratio of the uranyl bond was observed, as shown 394 in Figure 8. During irradiation, a band corresponding to $^{18}O=U=^{16}O$ was observed, with an intensity comparable to that of the band associated with the ¹⁸O=U=¹⁸ O bond (**Figure 8a**). When irradiation was 396 stopped, a change in the isotopic response occurred: the $^{18}O=U=^{16}O$ band became progressively 397 weaker compared to that of the ¹⁸O=U=¹⁸O band (Figure 8b). The observation of an isotopic change when the beam is stopped suggests an evolution in studtite formation mechanism, with or without irradiation, which will be discussed later. The shape of the curve showing a gradual isotopic transition following the end of the irradiation can be attributed to the fact that the studtite layer as a whole influences the Raman response. As a result, the isotopic impact of studtite formed under irradiation remains despite the beam being stopped. The result is a direct link between the curve profile and the

- kinetics of studtite growth after the beam is stopped (**Figure 3b**). Thus, the inflexion observed in the
- curve after 1000 minutes can either be interpreted as the attainment of a spectrum characteristic of
- studtite formed without irradiation, or as the result of a slowdown in precipitation.
- Finally, it was not possible to obtain a satisfactory fit with the binomial distribution model, suggesting
- the existence of a mechanism for the formation of the specific uranyl bond during alpha irradiation.
- 408 This mechanism seems to favour the formation of the $^{18}O=U=^{16}O$ bond during irradiation. The peroxide
- bond, on the other hand, is not affected by these changes during leaching.

411 *Figure 8. Evolution of uranyl bond isotopy during leaching experiment a. spectrum recorded at the end of irradiation; b.**intensity ratio of 790 cm^{-1 (18}O=U=¹⁶O) to 774 cm⁻¹ (¹⁸O=U=¹⁸O) bands. intensity ratio of 790 cm-1 (¹⁸O=U=¹⁶O) to 774 cm-1 (¹⁸O=U= ¹⁸O) bands.*

413 **4. Discussion**

414 In light of the results obtained, the first open question is to understand the chemical relationship 415 between the peroxide bond of the H₂O₂ radiolytic product and the bridging peroxo ligand inside 416 studtite.

417 Firstly, the reference leaching experiment of the UO₂ pellet in the presence of H₂¹⁸O₂ shows that there 418 is isotope exchange with the solution and secondly that the isotopy of the bridging peroxo ligand in 419 studtite is directly linked to that of H2O² in solution. Indeed, **Figure 7b** shows that in the presence of 420 \pm H₂¹⁸O₂, the bridging peroxo ligand in studtite is predominantly ¹⁸O-¹⁸O. In other words, the stability of 421 the peroxo ion is important and the peroxide bond is not broken when studtite precipitation occurs. 422 Based on this observation, it is interesting to discuss the isotopy of the peroxide bond in studtite in 423 relation to the radiolytic processes leading to the formation of H_2O_2 in solution under high and low LET 424 radiation. **Figure 6** clearly shows that under alpha irradiation (high LET) the peroxide bond is 425 predominantly ¹⁸O-¹⁸O, whereas it is mainly ¹⁶O-¹⁶O under gamma irradiation (low LET). Under water 426 radiolysis, primary hydrogen peroxide is produced mainly by recombination of OH° radicals during the 427 heterogeneous chemistry step (< 10^{-6} s) in the spurs generated by the energy deposition of alpha 428 particles and gamma rays. However, under gamma irradiation, over longer periods and during the 429 homogeneous chemistry stage (> 10⁻⁶ s), dissolved oxygen (¹⁶O₂) in solution controls the final H₂O₂ 430 concentration at steady state and is strongly involved in its formation processes [41-45]. Under gamma 431 irradiation, dissolved oxygen leads to the formation of $H_2^{16}O_2$ by scavenging the H° radical and the 432 aqueous electron according to the following reaction scheme:

$$
H^{\circ} + {}^{16}O_2 \rightarrow H^{16}O_2^{\circ}
$$
 (2)

$$
e_{aq} + {}^{16}O_2 \rightarrow {}^{16}O_2^{\circ}
$$
 (3)

$$
H^{16}O_2^{\circ} + {^{16}O_2}^{\circ} \cdot (+H_2O) \rightarrow H_2^{16}O_2 + {^{16}O_2} + OH^{\circ}
$$
 (4)

$$
H^{16}O_2^{\degree} + H^{16}O_2^{\degree} \to H_2^{16}O_2 + {}^{16}O_2 \tag{5}
$$

 This process, related to the presence of dissolved oxygen, has been the subject of several comparisons between radiolysis calculations and experimental measurements and is now well known and reported in the literature [41-45]. Under alpha irradiation, the primary production of H° radicals and aqueous 440 electron is much lower than under gamma irradiation, therefore the production of H_2O_2 is not very 441 sensitive to the presence of dissolved oxygen in the homogeneous solution. The main process of H_2O_2 $\,$ formation in H₂¹⁸O labelled water remains the recombination of two ¹⁸OH° radicals according to:

$$
^{18}\text{OH}^\circ + ^{18}\text{OH}^\circ \rightarrow \text{H}_2{}^{18}\text{O}_2 \tag{6}
$$

 The isotopic signatures of the peroxo ligands in studtite for the experiments carried out under alpha and gamma irradiation are therefore fully consistent with the formation processes of hydrogen 446 peroxide in solution. A ¹⁶O-¹⁶O chemical bond is observed under gamma irradiation, whereas an ¹⁸O- $\mathrm{^{18}O}$ bond is favoured under alpha irradiation. The presence of a peroxo $\mathrm{^{18}O\text{-}^{18}O}$ bond under alpha irradiation also confirms the stability of water with respect to isotopic exchange over the duration of the experiment, which is consistent with the long-term studies conducted by Bonales [46].

 The second objective is to obtain some information on the oxidative dissolution mechanism from the 451 use of isotopes. For the uranyl bond, the isotopic composition of the oxygen atoms inside UO₂²⁺ ions can provide information on the oxidative dissolution mechanisms of uranium dioxide. However, in order to ensure that the actinyl species keeps the memory of the dissolution mechanism, the residence

- 454 time of the oxygen atoms on the UO₂²⁺ ions must be long enough since its release from the surface to 455 the studtite to be analysed by Raman spectroscopy. The isotopic exchange of oxygen atoms within the
- 456 uranyl ion has been studied since the early 1960's. These studies [47, 48] have shown that the uranyl
- 457 ion does not exchange its oxygen atoms with the solution for a period of 2000 hours, much longer than
- 458 the duration of the experiments presented in this work. Similarly, studies of the nature of isotopic
- 459 exchange between studtite and ^{18}O -enriched water have shown the absence of exchange with the
- 460 peroxo and uranyl bonds within studtite [47]. These bonds are strongs and not very sensitive to
- 461 isotopic exchange. The only fast exchanges are with the water molecules of studtite. Therefore, it is
- 462 reasonable to assume that the oxygen isotope of the uranyl ion provides relevant information about
- 463 the nature of the oxidative dissolution mechanism at the $UO₂$ / water interface.
- 464 The reference leaching experiment of the UO₂ pellet in the presence of $H_2^{18}O_2$ clearly shows that the 465 uranyl bond contains a significant amount of ^{16}O (15%) while the peroxo bond mainly contains ^{18}O 466 (97%) (Figure 6). The uranyl bond well integrates oxygen ¹⁸O coming from the labelled water (H₂¹⁸O + 467 $H_2^{18}O_2$ media), but also ¹⁶O. The latter certainly comes from the solid and reflects an oxidation 468 mechanism at the surface of the pellet. Indeed, the hypothesis of an initial release of the U^{4+} ion from 469 the surface into the solution, followed by oxidation in the solution ($H_2^{18}O$ + $H_2^{18}O_2$ system), would have 470 led to an ¹⁸O enrichment close to 100%. Furthermore, ${}^{16}O_2$ coming from the air is 5 times less 471 concentrated than $H_2^{18}O_2$ and its kinetic oxidation constant is 200 times lower [49], which does not 472 support its participation in the uranium oxidation process.
- 473 With regard to the surface oxidation process in the presence of H_2O_2 , it is generally described in two 474 stages, with the formation of $U(V)$ first, followed by the formation of the uranyl ion. A plausible 475 mechanism proposed by Ekeroth and Jonsson [50] for the reaction between UO₂ and H₂O₂ is a primary 476 slow one-electron transfer stage producing U(V) and OH° radicals according to the following reaction:
- $UO_2 + H_2O_2 \rightarrow UO_2^+$ surf + OH^o+ OH^o (7)
- 478 Which becomes by taking into account the isotopy of the studied system:
- $U^{16}O_2 + H_2^{18}O_2 \rightarrow U^{16}O_2^*$ surf $+{}^{18}OH^{\circ} + {}^{18}OH^{\circ}$ (8)

480 According to such a process, the formation of $U(V)$ from H_2O_2 would not necessarily involve the 481 incorporation of oxygen atoms into the fluorite structure. The formation of the uranyl ion can then be 482 written [50]:

483 $U^{16}O_2^+{}_{surf} + U^{16}O_2^+{}_{surf} \rightarrow U^{16}O_2^{2+}{}_{surf} + U^{16}O_2$ (9)

484 As the ¹⁸ OH° radicals formed, they can contribute to catalytic decomposition of H_2O_2 according to the 485 following mechanism [51-53]:

- $H_2^{18}O_2 + M \rightarrow 18OH + 18OH^{\circ}{}_{ads} + M^{\dagger}$ (10)
- $^{18} \text{OH}^{\circ}_{\text{ads}} + \text{H}_2^{18} \text{O}_2 \rightarrow \text{H}^{18} \text{O}_2^{\circ} + \text{H}_2^{18} \text{O}$ (11)
- $M^+ + H^{18}O_2^{\circ} \rightarrow H^{+} + {}^{18}O_2 + M$ (12)
- 489 $2 H_2^{18}O_2 \rightarrow {}^{18}O_2 + 2 H_2^{18}O$ (13)

490 They can even oxidize the $UO₂$ surface. The relative contribution of these two processes (catalytic 491 decomposition versus oxidation) depends in particular on the amount of OH° radicals scavenged by 492 hydrogen peroxide [52]. Sunder [54] has proposed an interaction process between OH° and the surface 493 involving the incorporation of oxygen atoms into the fluorite structure. OH° radicals, produced during 494 the first oxidation step with H₂O₂ are smaller in size than the H₂O₂ molecule and can also oxidize the 495 surface. The $UO₂$ surface can incorporate oxygen atoms in an interstitial position according to the 496 following process [54]:

$$
U(IV) + {}^{18}OH^{\circ} \to H^+ + {}^{18}O_1{}^{2-} + U(V) \tag{14}
$$

498 The interstitial ¹⁸O oxygen atom can therefore also contribute to the formation of UO₂⁺_{surf} and UO₂²⁺surf 499 ions. More recently, studies on the interaction between water plasma and a UO₂ surface mention the 500 incorporation of oxygen atoms into the fluorite structure, resulting in a decrease of the U4f/O1s 501 surface area ratio measured by XPS [55]. Thus, the formation of the uranyl ion in the presence of H₂¹⁸O₂ 502 alone ($H_2^{18}O$ + $H_2^{18}O_2$ media), with both ¹⁸O and ¹⁶O isotopic contributions, is fully compatible with an 503 oxidation mechanism involving both electronic transfer and incorporation of interstitial oxygen from 504 the solution.

505 Therefore, it is interesting to discuss the isotopy of the uranyl bond for the alpha and gamma 506 irradiation experiments. For the gamma irradiation experiment, the uranyl bond is highly enriched up 507 to 95% in ¹⁸O. The primary yields of OH° radicals for gamma radiation (low LET radiation) are high [35] 508 and these highly reactive species can rapidly interact with the UO₂ surface. The high reactivity of these 509 18 OH° radicals produced by radiolysis of labelled water (H₂¹⁸O) would lead to a massive incorporation 510 of ¹⁸O at the surface and explain the low ¹⁶O content of the uranyl ion. On the other hand, under alpha 511 irradiation, H_2O_2 peroxide remains a major product of water radiolysis and the primary production of 512 OH° radicals is 10 times lower than under gamma irradiation. Furthermore, the 18 O enrichments 513 obtained for the uranyl bond for the reference experiment with simple addition of $H_2^{18}O_2$ and for the 514 experiment under alpha irradiation with ¹⁸O labelled water are very similar (figure 6). This observation 515 clearly argues for a predominant role of $H_2^{18}O_2$, especially after the beam has been cut. It is then 516 consistent to find the two isotopes (18) and (16) of oxygen in the uranyl bond if processes involving 517 both electronic transfer and surface incorporation occur simultaneously under alpha irradiation.

It should be noted that under alpha irradiation and before the beam is cut off, the 18 O enrichment in 519 the uranyl bond is lower and the 16 O content is higher than after the beam is cut. This point may reflect 520 an increased mobility of oxygen atoms (^{16}O) in the solid due to the continuous creation of defects, but 521 will require additional development in the future. Another hypothesis is that the high local content of 522 hydrogen peroxide in the first seventy microns under alpha irradiation induces strong scavenging of 523 ¹⁸OH° radicals.

5. Conclusion

526 An original experimental approach combining Raman spectroscopy to the use of H_2^{18} O labelled water 527 was developed to better understand the oxidizing dissolution of $UO₂$ and studtite precipitation mechanisms under alpha radiolysis of water. Two static leaching experiments were performed: the 529 first involved the alteration of a UO₂ pellet under alpha radiolysis in the presence of ¹⁸O-labelled water 530 with in-situ Raman monitoring of the altered layer, and the second one involved the alteration of UO₂ 531 in the presence of ¹⁸O-labelled hydrogen peroxide. In these experiments, the precipitation of a studtite layer on the surface of the pellet was observed. It is clear that the bridging peroxo ligands in studtite originate from the peroxo species within the hydrogen peroxide molecule, which is itself produced by the radiolysis of water. Under alpha irradiation, the isotopic composition of the peroxide bond in 535 studtite is essentially ¹⁸O-¹⁸O and originates from the H_2 ¹⁸O₂ molecule formed by recombination of 536 OH $^{\circ}$ radicals in solution.

537 For the uranyl bond of the UO₂²⁺ ions, the presence of both ¹⁶O and ¹⁸O isotopes is observed. This can

be explained by allowing an oxidation process involving both a simple electron transfer by interaction

- 539 with H_2O_2 and the incorporation of oxygen atoms coming from the solution into the fluorite structure 540 due to oxidation by OH $^{\circ}$ radicals. The production of 18 OH $^{\circ}$ radicals, particularly by catalytic
- 541 decomposition of hydrogen peroxide $H_2^{18}O_2$ on the surface of UO₂, remains a preferred route leading
- to the insertion of oxygen atoms into the material structure at the surface.
- Moreover, these results are consistent with previous studies, this time carried out under gamma 544 irradiation and in aerated conditions. The chemical relationship between the peroxide bond of H_2O_2 and the O-O bridging bond in studtite is also confirmed, as is the important role of OH° radicals in the insertion of oxygen atoms into the fluorite structure.
- In the future, it will be useful to further investigate the role of atomic defects in the dynamics of 548 isotopic exchange at the reaction interface. In this context, the use of a solid doped with 18 O, which allows the study of exchange dynamics within the solid under irradiation, would be relevant and is currently under extensive investigation.
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