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# QUANTITATIVE ANALYSIS OF CARBON BLACK AGGLOMERATES MORPHOLOGY IN ELASTOMER COMPOSITES BASED ON X-RAY TOMOGRAPHY BY MEANS OF NUMERICAL CLUSTERING

Jesbeer Kallungal, Laurent Chazeau, Jean-Marc Chenal, Jâérôme Adrien, Eric Maire, Claire Barrès

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1 **Quantitative analysis of carbon black**  
2 **agglomerate morphology in elastomer**  
3 **composites based on X-ray Tomography by**  
4 **means of numerical clustering**

5 Jesbeer. Kallungal<sup>1,2,3,\*</sup>, Laurent. Chazeau<sup>1,\*</sup>, Jean-Marc. Chenal<sup>1</sup>, Jérôme.  
6 Adrien<sup>1</sup>, Eric. Maire<sup>1</sup>

7 <sup>1</sup>*Univ Lyon, INSA Lyon, CNRS, MATEIS UMR5510, F-69621, Lyon, France*

8 Claire. Barrès

9 <sup>2</sup>*Univ Lyon, INSA Lyon, CNRS, IMP UMR5223, F-69621, Lyon, France*

10 Bernard. Cantaloube, Patrick. Heuillet

11 <sup>3</sup>*LRCCP Vitry-sur-Seine, France*

12 *Corresponding authors: [jesbeer.kallungal@insa-lyon.fr](mailto:jesbeer.kallungal@insa-lyon.fr) and [laurent.chazeau@insa-lyon.fr](mailto:laurent.chazeau@insa-lyon.fr)*

13 *Permanent adresse: Laboratoire MATEIS (PVMH), Bâtiment Blaise Pascal, 5<sup>o</sup> étage, 7 Avenue*

14 *Jean Capelle, 69621 Villeurbanne cedex.*

15 **Keywords:** defects, x-ray tomography, elastomers, microstructural analysis, carbon black  
16 **agglomerates.**

17

18 **ABSTRACT**

19 In this paper, a novel methodology for characterizing the morphology distribution of filler  
20 agglomerates in Elastomer composites is presented, based on laboratory sourced X-ray  
21 Tomography. Various feature extraction methods (via e.g. Image processing filters,  
22 segmentation) and selection tools (Spearman's rank correlation coefficient) combined with K-  
23 means unsupervised clustering algorithm were developed for identifying the distinct  
24 morphological classes in model materials (carbon filled ethylene propylene diene monomer  
25 rubber). The interest of this methodology was demonstrated by precisely differentiating the  
26 materials compounded with different processing parameters. For instance, in this example,  
27 thanks to this analysis, it was found that introducing the filler before the elastomer in internal  
28 mixer tends to favor more structured agglomerates.

## 29 **1 INTRODUCTION**

30 Elastomers are used in many applications such as load-bearing components, anti-vibration  
31 mounts, seals etc... The optimisation of properties for these applications requires the addition  
32 of many ingredients in the pristine polymers. The ingredients used are fillers (Carbon black CB,  
33 Silica), processing aids (paraffin wax), crosslinking agents (Sulphur, Peroxide), cure activators  
34 (zinc Oxide ZnO, Stearic Acid) and accelerators (to control the cure rate). A good dispersion of  
35 these ingredients, especially fillers, is achieved through their mixing, in an internal mixer and  
36 a two roll mill machine. Studies have shown that the choice of processing parameters such as  
37 fill factor, shear rate, ram pressure applied in the internal mixer, mixing energy etc... impacts  
38 the morphology (e.g size, shape factor), the spatial distribution and the volume fraction of the  
39 filler agglomerates (hereafter referred as  $CB_{aggl}$ )<sup>1-5</sup>. Apart from them, other microstructural  
40 heterogeneities can also exist in the rubber compounds such as unreacted ZnO inclusions, voids  
41 etc..<sup>6</sup> Studies have demonstrated the potential impact of all these so called “flaws” or “defects”  
42 on crack mechanisms during monotonic and dynamic solicitation of the material<sup>7-16</sup>. Different

43 research works emphasize the importance of quantifying their morphology. For instance, Gent  
44 *et al.* proved the dependence of the initiation of cavities in silicone elastomer on the size and  
45 spatial distribution of model flaws (using spherical glass beads)<sup>17</sup>. In addition, Chow *et al.* have  
46 shown theoretically the importance of the aspect ratio of rigid flaws on these mechanisms, as it  
47 impacts the stress concentration at their poles<sup>18</sup>.

48 In literature, the characteristics of filler agglomerates of carbon black ( $CB_{\text{aggl}}$ ) in an elastomer  
49 are obtained from the roughness measurement of their micro-tomed sections (ASTM D2663),  
50 or from the light reflexion analysis of a freshly cut sample surface with a Dispergrader  
51 technique or from high resolution electron microscopy such as SEM or TEM<sup>10,19</sup>. However, the  
52 main drawback of these techniques is that they can only provide 2D information and therefore  
53 a full description of the morphology or spatial distribution of  $CB_{\text{aggl}}$  cannot be ascertained. X-  
54 ray Tomography can be used to obtain 3D information on the polymer at the micron scale<sup>20-24</sup>.

55 Due to the similar attenuation contrast of filler agglomerates ( $CB_{\text{aggl}}$ ) and elastomer matrix, to  
56 the authors' knowledge, only one paper reports their detection and morphological  
57 characterization using non-synchrotron sourcing X-ray tomography in an elastomer  
58 composite<sup>25</sup>. To do so, the authors had to adapt the laboratory Tomography parameters.  
59 However, a methodology is missing to extract, from the obtained data, the pertinent descriptive  
60 features, which can help in relating the agglomerates morphology and spatial dispersion to the  
61 material mechanical properties. This is the motivation of the work presented here.

62 Such methodology could use different tools that have been developed for the classification of  
63 filler aggregates. Medalia *et al.*<sup>26-29</sup>, through a series of publications, used various Euclidean  
64 geometric features, such as bulkiness factor, anisometry, area and structuring factor, to analyse  
65 TEM images. Differently, Bourrat *et al.*<sup>30</sup> proposed to use mass fractal analysis, which also  
66 applied to the same type of images. This mass fractal approach was extended later by Rieker *et*  
67 *al.*<sup>31</sup> for N330 carbon black dispersed in a polymer matrix, characterized by small-angle X-ray

68 scattering (SAXS) which provides information in the bulk. It was also used by Baeza *et al.*<sup>32</sup>  
69 for analyzing the morphology of silica aggregates in an elastomer. Herd *et al*<sup>33</sup> have shown,  
70 when analyzing the morphology of nineteen different carbon blacks dispersed in an elastomer,  
71 that the results of both approaches, Fractal vs Euclidean geometry, are correlated<sup>22</sup>. However,  
72 Euclidean geometry approach appears to be the most pertinent for specific shape classification  
73 and for this reason, will be preferred in this paper.

74 The outline of this article is as follows. The proposed methodology for analyzing filler  
75 agglomerates in elastomers is developed using experimental data obtained from model  
76 materials. After the description of the model materials and testing methods in section 2, feature  
77 extraction and selection techniques are presented (section 3). Following the review of clustering  
78 techniques for morphological clustering (section 4), the results and discussion on the  
79 classifications are then presented in section 5 and 6 respectively. Such classification is  
80 performed for the agglomerates morphology in the different model materials. Our methodology  
81 enables to differentiate two materials with identical composition, not only by the volume  
82 fraction of  $CB_{aggl}$ , but also through various morphological features. It also provides a valuable  
83 tool to understand the impact of operating parameters during the compounding process on the  
84 morphology of  $CB_{aggl}$ .

## 85 **2 MATERIALS AND METHODS**

### 86 **2.1 Composition and materials processing**

87 Model elastomers have been fabricated using a non-crystallizing Ethylene Propylene Diene  
88 Monomer Rubber (EPDM), and an N300 series carbon black, N326 as reinforcing filler. Table  
89 I presents the compound composition, which was kept constant, and the compounding process  
90 conditions, which were varied. The crosslinking agent is unsupported Bis( $\alpha,\alpha$ -dimethylbenzyl)  
91 peroxide. The processing of the samples was performed in a Haake Rheomix 600 OS mixer

92 (chamber volume: 120 cm<sup>3</sup>) using Banbury rotors (42 cm<sup>3</sup>). All the components, except  
93 peroxide, were mixed in the internal mixer (IM) for 2 min, and then passed 10 times in the two  
94 roll-mill where the peroxide was added. The rotor speed of the internal mixer is specified for  
95 each material in Table I, as well as the sequence for introducing the ingredients. USD protocol  
96 refers to “upside down”, i.e. the elastomer was added after carbon black. The resulting model  
97 materials from this procedure are referenced as 30\_rpm\_usd and 60\_rpm\_usd. TD protocol  
98 refers to “top down”, i.e. carbon black was introduced after the elastomer in the internal mixer,  
99 and the resulting model materials are referenced as 30\_rpm\_td and 60\_rpm\_td. After the mixing  
100 steps, the samples were molded as 2mm thick films by hot pressing at 170°C under 190 MPa  
101 for 12 min. Cure time was estimated from the torque measurements, performed using a  
102 Monsanto rheometer analyzer.

## 103 **2.2 X-ray tomography**

104 The X-ray Tomography device is EasyTom from RX solutions. The Tungsten target current is  
105 45 μA and a LaB<sub>6</sub> cathode filament is used, along with a CCD detector. Two sample specimens  
106 extracted from each material is used for the experiments (only one was used for morphological  
107 clustering analysis). Cutting of the specimen is performed with a scalpel. Specimen length and  
108 width is adapted for monitoring within the field of view of X-ray Tomography: The volume  
109 analyzed is in the order of 4-6mm<sup>3</sup>. The sample is rotated over 360 degrees in front of the X-  
110 ray beam. Image acquisition is performed every 0.09° (4000 projections in total). Each  
111 projection step lasted 0.45s resulting in a total time per scan of 30 minutes. The voxel resolution  
112 in these experiments is 1μm<sup>3</sup>. Imaging was performed at a tube voltage of 40kV for an  
113 optimised attenuation contrast between the elastomer and agglomerates as discussed in a  
114 previous paper<sup>25</sup>.

## 115 **3 FEATURE EXTRACTION**

116 3D images obtained from X-ray Tomography show different flaws like carbon black  
117 agglomerates, voids, etc... The feature extraction methodology involves applying different  
118 algorithms of image processing to selectively display  $CB_{\text{aggl}}$  from the acquired tomography  
119 scans. Thereby, various statistical information related to their morphology can be obtained. A  
120 methodology to extract the  $CB_{\text{aggl}}$  spatial distribution was also applied on the same samples. It  
121 is described in Appendix A along with the obtained results.

### 122 **3.1 Image processing**

123 Various functions such as geometry offset, ring filter, and phase contrast are applied to the raw  
124 images to reduce artefacts and to improve the contrast between the matrix and the filler  
125 particles. Image stacks are then built using a filtered back projection reconstruction algorithm.  
126 Several Python modules are then used for image processing on the 3D reconstructed volume.  
127 The general approach of feature segmentation by segregating the greyscale cannot be applied  
128 in our case, since there is no significant difference in the attenuation contrast between the matrix  
129 and the  $CB_{\text{aggl}}$ . Therefore, various image filters are applied again on the image stack (Figure 1a)  
130 to optimise the contrast and to remove additional noise. Typical denoising filters such as the  
131 median filter are not effective. For this reason, a different workflow of image processing is  
132 devised, which is inspired from the works of Emmanuelle Guillard<sup>34,35</sup>. After intensity  
133 rescaling (to improve the brightness and contrast) (Figure 1b), bilateral (Figure 1c) and non-  
134 local means (Figure 1d) filters are applied on the image stack. In bilateral filter, the intensity of  
135 each pixel in the image is replaced with a weighted average intensity of nearby pixels for noise  
136 reduction and edge preservation. In non-local means filter (effective for images with fine-  
137 grained texture), each target pixel value is replaced by an average of all the neighbourhood  
138 pixels value, weighted by how similar these pixels are to the target pixel<sup>36</sup>. Subsequently,  
139 segmentation of the image is carried out, which is a crucial step for obtaining pertinent

140 information. Here we are using a supervised segmentation algorithm called Random Walker for  
141 which was developed by Grady to reduce manual intervention and user bias<sup>36,37</sup>. The  
142 segmentation is directly carried out on 3D images. The different image treatments have been  
143 implemented through the scikit-image module in Python<sup>38</sup>. The output of the image processing  
144 and segmentation is presented in Figure 1e. The final process of removal of the particles  
145 touching the edges of the stack, as these may intervene in the morphological classification of  
146  $CB_{aggl}$ , is performed using ImageJ © free software. A sample sub volume of the reconstructed  
147 3D image is shown in Figure 2 and Appendix B.

### 148 **3.2 Morphological features extracted**

149 Each agglomerate is labelled in the binary image after its segmentation from the matrix. The  
150 X ray tomography resolution being  $1\mu\text{m}$  (i.e one voxel), and given the possible error in the  
151 identification of agglomerates border (at least 1 voxel), agglomerates with volume below  $5\times 5\times 5$   
152 (i.e. 125 voxels) have not been considered. This way, we reduce the errors in the calculation of  
153 the morphological parameters of the small agglomerates (note also that this volume roughly  
154 corresponds to the minimum cutoff volume in the D2663 ASTM method for evaluating carbon  
155 black dispersion in rubber). Geometric features such as Eccentricity  $E$ , Anisometry  $Q$ , Bulk  
156 factor  $B$ , Volume  $V$ , Equivalent Diameter  $Eq_{diam}$ , Length of major principle axis  $a$ , and Length  
157 of minor principle axis  $b$  of each  $CB_{aggl}$  are extracted using the *regionprops* module in Python<sup>38</sup>.  
158 The volume  $V$  of a  $CB_{aggl}$  is calculated by counting the number of voxels it contains. The  
159 Equivalent diameter ( $Eq_{diam}$ ) represents a diameter of a fictive spherical inclusion having the  
160 same volume as the analyzed  $CB_{aggl}$ . The  $Eq_{diam}$  of  $CB_{aggl}$  with a volume of  $125\mu\text{m}^3$  (1 voxel  
161  $=1\mu\text{m}^3$ ) is equal to  $6.2\mu\text{m}$ . Length of major ( $a$ ) and minor principal axis ( $b$ ) are those of an  
162 *ellipsoid* that has similar normalized 2<sup>nd</sup> central moment<sup>36</sup> as the  $CB_{aggl}$ . Eccentricity ( $E$ ) of an  
163 ellipse is defined as the ratio between its distance of two foci and the major principal axis length.



164 Here,  $E$  is the eccentricity of the centered elliptic section of the ellipsoid, perpendicular to the  
165  $c$  axis (with  $a > c > b$ ):

$$166 \quad E = \sqrt{1 - b^2/a^2} \quad (1)$$

167 For  $E=0$ , the morphology is a sphere and for  $E$  close to 1 the morphology get closer to that of  
168 a 1D or 2D object. Anisometry ( $Q$ ) is the ratio of radii of gyration and corresponds to  
169 anisometry of the agglomerates as calculated by Medalia *et al.* <sup>26</sup>:

$$170 \quad Q = b/a \quad (2)$$

171 Bulk factor ( $B$ ) corresponds to the ratio of the free space (volume in this case) between the  
172 agglomerates and the convex hull constructed around the agglomerate ( $\Delta V$ ) to the total volume  
173 of the convex hull ( $V_{ch}$ ), as shown in Figure 3:

$$174 \quad B(\Delta V) = 1 - V/V_{ch} \quad (3)$$

175 This definition is slightly different from the approach used by Medalia *et al.* <sup>26</sup> who proposed  
176 to express bulkiness as a ratio of the area of radius-equivalent ellipse drawn around the object  
177 to its projected area.

### 178 **3.3 Feature selection**

179 Using all the features  $V$ ,  $Eq_{diam}$ ,  $B$ ,  $E$ ,  $b$ ,  $a$ , and  $Q$  for clustering presents some disadvantages.  
180 Firstly, if the two relevant features chosen are strongly correlated to each other, they add more  
181 weight to the classification process and thereby generate biased clusters. Secondly, the  
182 computation complexity of clustering increases as the number of features increases. Therefore,  
183 feature selection techniques have to be applied to evaluate the relevance of each one. Such step  
184 is very critical when defining the morphology classes by unsupervised pattern recognition, since  
185 we lack prior knowledge of each morphology class features. First, the relevance of each feature  
186 was checked using Laplacian Score technique (described in Appendix C). Then Spearman's  
187 rank correlation coefficient ( $\rho$ ) has been used to discard the redundant feature.  $\rho$  characterises  
188 the monotonic relations (linear or not) between two features. It varies between -1 to 1. A value

189 close to  $|1|$  implies that they are correlated. Here, if two features have a coefficient value below  
190  $|0.7|$ , they will be considered to be distinct. All the calculated  $\rho$  values are reported in Figure 4  
191 along with their distribution. All data from all studied materials have been used for the analysis.  
192 They show that  $a$ ,  $b$ , and  $Eq_{diam}$  are correlated to each other and also to  $V$ . As expected,  $E$  is  
193 negatively correlated to  $Q$  (aspect ratio).  $B$  is only correlated to  $a$  and feature  $a$  is correlated to  
194 other features ( $V$ ,  $Eq_{diam}$  and  $b$ ). Feature  $B$  has been retained because it can provide information  
195 on the fact whether the agglomerates are structured or not and the two other relevant retained  
196 features are  $V$  (size) and  $E$  (Eccentricity).

197 The distributions of  $V$ ,  $E$  and  $B$  for the 4 model materials are plotted in Figure 5.  $V$   
198 distribution is similar for all the materials, with a slight difference in the domain  $1.25 \cdot 10^3$ -  
199  $2.25 \cdot 10^3 \mu\text{m}^3$ . The relative frequency is lower for the materials processed with the upside-down  
200 protocol (USD) in this domain (cf Figure 5a, b). The distribution curves of  $B$  (structure factor)  
201 have a similar shape for all the materials (Figure 5c). The curves also indicate that the  
202 agglomerates of the 60\_rpm\_td are the less structured, and those in the 30\_rpm\_usd are the  
203 most structured. The  $E$  distributions of the 4 materials (cf Figure 5d) have similar Gaussian  
204 shape with mean values around 0.75 for all the materials, indicating that their agglomerates are  
205 not spherical.

206 To sum up, there are only slight differences between the 4 materials, regarding their  $E$ ,  $B$ ,  
207 and  $V$  distributions. Moreover, as they can only be discussed separately, it is impossible to know  
208 for instance, if the largest number fraction of agglomerates with a low structure factor in the 60  
209 rpm\_td is related to the smallest or the biggest agglomerates. In other words, these distributions  
210 do not provide a clear picture of the differences in the agglomerates morphology. This motivates  
211 the use of the clustering methodology described in the following.

## 212 4 CLUSTERING

213 Manual clustering of all agglomerates using all the relevant features is a cumbersome task.  
214 Some authors have recently used unsupervised clustering techniques (Machine learning  
215 algorithms) such as hierarchical clustering, or principle component analysis<sup>39</sup> to solve this  
216 problem<sup>39-41</sup>. For practical reasons, we chose K-means unsupervised clustering algorithm.  
217 Classification using other clustering algorithms (agglomerative clustering, hierarchical  
218 clustering etc...) did not provide interpretable results. K-means algorithm was implemented  
219 using scikit-learn machine learning module in Python<sup>42</sup>.

### 220 4.1 K-means unsupervised clustering

221 K-means clustering algorithm attributes a set of data points  $x_i$  ( $x_i$  is defined by p coordinates)  
222 describing n agglomerates ( $1 \leq i \leq n$ ) by p features ( $V, B, E$ ) into  $k$  clusters  $C = \{C_1, C_2, \dots, C_k\}$ .  
223 Each cluster can be described by the mean  $\mu_j$  of the observations in the cluster, commonly called  
224 as cluster centroid. The algorithm tries to find a centroid that minimizes the inertia or the  
225 variance of the squared Euclidean distance within the cluster,  $\|x_i - \mu_j\|^2$

$$226 \sum_{j=1}^k \sum_{x \in C_j} \|x_i - \mu_j\|^2 \quad (4)$$

227 As the number of clusters  $k$  is not known *a priori*, 2 validation criteria were used to determine  
228 the optimal number of clusters.

- 229 • Calinski Harabasz (*CAH*): this ratio is sometimes called variance ratio criterion.

$$230 CAH = (SS_B / SS_W) \times (N - k) / (k - 1) \quad (5)$$

231 Where  $SS_B$  is the overall “between-cluster” variance,  $SS_W$  is the overall “within cluster”  
232 variance,  $k$  is the number of clusters and  $N$  is the total number of data points (i.e.  
233 agglomerates). *CAH* metric works very well for clustering algorithm based on squared  
234 Euclidean distances such as K-means<sup>43</sup>. The ideal number of clusters corresponds to the  
235 highest *CAH* score.

236 • The Davies and Bouldin index (*DBI*) can be used to infer the average similarity of a cluster  
237 with its most similar cluster<sup>44</sup>. In this context, it is given by the equation as follows:

$$238 \quad DBI = \frac{1}{k} \sum_{i=1}^k \max_{j \neq i} \left( \frac{s_i + s_j}{d_{ij}} \right) \quad (6)$$

239  $s_i$  is the average distance between each data points in cluster  $i$  and its centroid (similarly  
240 for  $s_j$ ),  $d_{ij}$  is the average distance between the clusters centroids  $i$  and  $j$ . *DBI* index is the  
241 lowest for the ideal number of clusters.

242 The workflow deployed in this article to have a generalized morphological cluster analysis is  
243 schematized in Figure 6. Across-over sampling between model materials 30\_rpm\_usd,  
244 60\_rpm\_usd, 30\_rpm\_td and 60\_rpm\_td has been applied. 30% of the overall data points  
245 (163907 data points) have been randomly chosen to train the clustering algorithm. This step is  
246 repeated 10 times to remove the effect of sampling on the determination of the number of  
247 clusters. The trained model is later on applied to all the data of each material to build their final  
248 clustering.

## 249 4.2 Statistical analysis of geometric features

250 As shown in the relative frequency distribution of  $V$  (cf. Figure 5), the 75% quartile and  
251 mean values indicate that most of the data lie close to the lower side of the distribution. In the  
252 case of  $B$  and  $E$ , they are more uniformly distributed. Since these features have a different order  
253 of magnitude, clustering on the raw data of  $V$  might bring biased clusters. For this reason, we  
254 applied standardization calculation to each feature  $f$  as follows:

$$255 \quad f_{standard} = \frac{f(x) - f_{mean}}{\sigma} \quad (8)$$

256 which scales the data to unit variance and 0 mean.  $f_{mean}$  and  $\sigma$  are the mean and standard  
257 deviation of the distribution of the observations of the feature  $f$ . Other standardization  
258 techniques exist, such as rescaling the data to  $[0, 1]$ , scaling the data according to the quantile  
259 range, or transforming data from various distributions to a normal distribution. These

260 techniques were discarded as it was found that they do not provide relevant results. The  
261 normalized distribution for each feature is presented in Appendix Figure D.

### 262 **4.3 Number of clusters**

263 The number of clusters is chosen from the calculations of the metrics *DBI* and *CAH*. As said  
264 previously, they have been calculated from 10 random samplings of data. These samplings have  
265 a very small influence on the found metric values. Their average is given in Table II for *DBI*,  
266 and reported in Figure 7 for *CAH*. Note that a third criteria, Silhouette coefficient ( $S_{avg}$ ) has  
267 also been used to determine the optimal number of clusters<sup>45</sup>. However, this metric does not  
268 seem relevant for the choice of the cluster number (Appendix E). For a cluster number of 9,  
269 *DBI* and *CAH* metrics show a local minimum and maximum respectively, suggesting that 9 is  
270 the optimal number of clusters.

## 271 **5 RESULTS**

### 272 **5.1 Classification of the agglomerates**

273 Morphology features of the centroids ( $V_C$ ,  $E_C$ ,  $B_C$ ) of the 9 classes are presented in Table  
274 III. They can also be visualised in ( $V$ ,  $B$ ) or ( $E$ ,  $B$ ) plots (cf. Figure 8).

275 The agglomerates in classes #1, #2, #3 are the smallest and most similar in size (Appendix  
276 Figure Fa and Figure 8a). They differ from each other by their  $E_C$  values and  $E$  distributions  
277 (Figure 9a). These distributions are well separated from each other with small overlap. The  
278 class #3 agglomerates have the lowest eccentricity (most spherical) and structure (lowest bulk  
279 factor), whereas class #2 has the highest eccentricity ( $E_C = 0.804$ ). The representative  $B_C$  and  $B$   
280 distributions (Appendix Figure Fb and Figure 8b) for the 3 first classes suggest that they contain  
281 low structured agglomerates (low  $B_C$  values). Class #1 and #3 have similar  $B$  distribution,  
282 whereas  $B$  values for class #2 is higher (meaning its  $CB_{aggl}$  are more structured).

283 The agglomerates belonging to classes #2, #4 and #6 have an intermediate size (even  
284 though the class #6 agglomerates are significantly bigger than those from classes #2 and #4)  
285 (Appendix Figure Ga). Their eccentricity is the largest suggesting that they are the most  
286 elongated in one or both directions (cf Table III, Appendix Figure Gb). The main difference  
287 between these three classes lies in their structure factor  $B$ , as shown in Figure 9b, the class #6  
288 agglomerates are the most structured, followed by the class #4 agglomerates.

289 The agglomerates belonging to the #7, #8 and #9 classes are the biggest of all  
290 agglomerates. The differentiating parameter between these classes is the size of the  
291 agglomerates: the higher the class, the bigger the agglomerates. Their eccentricity distribution  
292 is very similar. Moreover, the bigger the agglomerates, the more structured they are, as shown  
293 on Figure 8a. Finally, the class #5 agglomerates have an intermediate size and a  $B$  distribution  
294 close to that of class #4 agglomerates (cf. Figure 9b), and an  $E$  distribution close to that of  
295 agglomerates in class #1 (cf. Appendix Figure Gc).

## 296 **5.2 Application of the methodology to the model materials**

297 Model materials presented in Materials and Testing section have been processed differently.  
298 Two rotor speeds (30 rpm vs 60 rpm) along with 2 ways of introducing the fillers and rubber  
299 into the internal mixer (Upside-down and Top-down approach) have been tested. The analysis  
300 has been performed on a volume which actually could have been eight times lower than the  
301 used volume, as shown by the spatial clustering analysis described in Appendix A. Note also  
302 that the same experiment and volume fraction analysis have been made on a second sample,  
303 and gave the same results ( $\mp 0.3\%$ ). Cumulated volume fraction  $\phi_{\text{cumul}}$  of  $\text{CB}_{\text{aggl}}$  in the model  
304 mixes is clearly correlated to the rotor speed (30\_rpm\_usd: 10.4%, 30\_rpm\_td: 11.6%,  
305 60\_rpm\_usd: 6.0%, 60\_rpm\_td: 5.6%). Note that the cumulated volume fraction of the  
306 agglomerates of size below 125 voxels, which have not been taken into account in the  
307 morphological analysis, is around 0.4% for all the materials. From  $\phi_{\text{cumul}}$ , we can estimate a

308 dispersion index<sup>46</sup> ( $\phi_{\text{cumul}}$  divided by the CB volume fraction, without any factor accounting for  
309 the possible elastomer content in the agglomerates) equal to 0.54, 0.6, 0.31, 0.29 for  
310 30\_rpm\_usd, 30\_rpm\_td, 60\_rpm\_usd and 60\_rpm\_td respectively. Deeper analysis of the  
311 agglomerates morphology and spatial distribution is required to understand the differences  
312 induced by the USD and TD approaches. Actually, as described in Appendix A, the  
313 consequences of changing from TD to USD protocol on the  $\text{CB}_{\text{aggl}}$  spatial distribution is small.  
314 An analysis of the morphology classes of the agglomerates appears to be much more insightful  
315 and is presented in the following.

### 316 **5.2.1 Differentiation of model mixes using $\text{CB}_{\text{aggl}}$ morphology clustering**

317  $\text{CB}_{\text{aggl}}$  volume fraction in each morphological class for the 4 model mixes is presented in  
318 Figure 10. For a given filler addition protocol, in most of the classes, the agglomerate volume  
319 fraction decreases when the rotor speed is increased, especially for classes with the biggest  
320 agglomerates and/or the most structured ones. Interestingly, the contrary is found for classes #1  
321 and #3, which correspond to relatively small and low structured agglomerates (that have a larger  
322 eccentricity than in class #2).

323 As mentioned earlier,  $\phi_{\text{cumul}}$  of  $\text{CB}_{\text{aggl}}$  in the model mixes is not significantly changed when the  
324 filler addition protocol changes. Compared to TD protocol, Upside Down protocol on model  
325 mixes processed at 30\_rpm leads to larger volume fractions of agglomerates in the classes #4  
326 and #6, which correspond to the agglomerates with the largest eccentricity. In the other  
327 morphological classes, the agglomerates volume fraction is lower when the USD protocol was  
328 applied. The impact of changing from USD to TD protocol is different for the materials  
329 processed at 60\_rpm. With this rotor speed, the USD protocol leads to a higher volume fraction  
330 of  $\text{CB}_{\text{aggl}}$  in the classes #4 and above, corresponding to agglomerates with  $V$  and  $B$  distribution  
331 which spread over a domain of values larger than in classes #1, #2 and #3.

332 Knowing the distribution of each feature in each morphological class, the differentiation  
333 between the model mixes can be analyzed even more accurately. One can use for instance, the  
334 cumulative frequency distribution of the features *B* and *E*, shown in Figure 11. Increasing the  
335 rotor speed produces agglomerates with lower structure (with a shift in the cumulative  
336 probability distribution of *B* towards the lowest values). This effect is prominent for the  
337 morphological classes #7 and #8 (Figure 11a, b) (as seen in Appendix Figure H, this effect is  
338 negligible for the other classes), which were the classes for which the differentiating parameter  
339 was the size. Similarly, there is a decrease in *E* of the agglomerates in class #8 (which  
340 corresponds to the biggest agglomerates) due to an increase in the rotor speed, as shown in  
341 Figure 11c.

## 342 **6 DISCUSSION**

343 As shown above, the methodology we developed, provides insightful information (and  
344 because of the use of only 3 features, they were easy to visualize, in Figure 8) to evaluate the  
345 influence of the processing parameters on the morphology of the  $CB_{\text{aggl}}$ . By doubling the rotor  
346 speed, the total shear deformation undergone by the compound in the internal mixer was also  
347 doubled. This promotes more rupture and erosion of agglomerates<sup>47,48</sup>. As expected, this leads  
348 to a lower cumulative volume fraction of agglomerates with *B* parameter above 0.2 (10.4% vs  
349 6.0%), corresponding to the classes #4 and above, i.e. this decreases the volume of the large  
350 and more structured agglomerates. Moreover, the increase in the volume fraction of the less  
351 structured small agglomerates (classes #1 and #3) with the smallest eccentricity (compared to  
352 class #2) suggests that the increase in the rotor speed promotes the creation of more spherical  
353 agglomerates from the erosion of the big ones. This is confirmed by the evolution of the *E* and  
354 *B* distributions within the classes #7 and #8.

355 In the comparison of TD vs USD protocol, interesting differences can be found through  
356 the clustering methodology. The consequences of the carbon black addition protocol seems to



357 depend on the rotor speed. At 30 rpm, when changing from TD to USD protocol, one observes  
358 in particular that the volume fraction of the big agglomerates (classes #7, #8 and #9) decreases.  
359 These agglomerates seem to be broken down to lower size agglomerates with large eccentricity  
360 corresponding to an increase in the CB volume fraction in the classes #4 to #6. At 60 rpm, the  
361 trend seems different with a slight increase in the volume fraction of the biggest agglomerates  
362 (classes #7 and #8). In general, TD protocol leads to a more rapid incorporation of CB in  
363 comparison to USD approach, as shown by the higher fraction of remaining CB, not  
364 incorporated after the IM step (Appendix Table I), i.e. prior to the roll mill step. This is actually  
365 in contradiction with the previous observations for the model materials mixed at 30 rpm. This  
366 means that the final CB agglomerates dispersion and morphology are the result of a very  
367 complex interplay between the CB dispersion at the end of the IM step with the following roll  
368 mill step. Only supplementary studies, including a precise CB dispersion characterization as a  
369 function of time during the different mixing steps may help to understand our model material  
370 microstructure. For this, the use of the methodology developed in this paper is required.

## 371 **7 CONCLUSION**

372 Very few studies have focused on the morphologies of carbon black agglomerates in  
373 industrial rubber compounds. This is mainly due to the limitations in the characterization  
374 techniques available at laboratory scale. Here, we demonstrated that even when using laboratory  
375 sourced X-ray Tomography, a good attenuation contrast between agglomerates and matrix is  
376 attainable provided that a LaB<sub>6</sub> X ray source operated at 40 kV is used. The attenuation contrast  
377 between phases can also be increased with the help of various image processing algorithms,  
378 thereby the extraction of various features becomes efficient. Like with carbon black aggregates,  
379 various morphological features can be used to characterize CB<sub>aggl</sub>. Out of the 7 features  
380 extracted from image analysis, only 3 were retained using feature selection technique  
381 (Spearman's rank correlation coefficient). The relevant features appeared to be the volume, the

382 bulk factor and the eccentricity. Morphological classification based on these features was  
383 performed using K-means unsupervised clustering algorithm.

384 Thus, we developed a complete methodology to obtain a morphological classification on CB<sub>aggl.</sub>  
385 This provides many insights on the peculiarity of the morphology of agglomerates in the studied  
386 model materials, and enables to distinguish samples with the same measured agglomerates  
387 volume fraction. The main advantage of our approach is that it enables to understand the impact  
388 of small changes in the materials processing protocol. Notably in the current study, the impact  
389 of two processing parameters, (influence of rotor speed and filler addition protocol) can be  
390 precisely characterized. This paves the way for a better understanding of the incorporation and  
391 desagglomeration mechanisms of carbon blacks during the process. This can also provide useful  
392 information to identify in the final materials, the critical morphology of carbon black  
393 agglomerates for crack initiation and propagation. More generally, the above presented  
394 methodology and the obtained results bring out new questions and prospects for further works.  
395 Not only we may study the influence of different processing parameters, e.g. mixing time,  
396 milling steps in roll mill etc., on the carbon blacks dispersion, but - as long as we obtain  
397 sufficiently contrasted tomography images - we may also apply the methodology to various  
398 type of elastomer filled with various types of fillers (e.g. glass beads, clay, calcium carbonate,  
399 recovered carbon black from tire pyrolysis, silica etc.).

## 400 **8 ACKNOWLEDGEMENTS**

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## 402 **9 REFERENCES**

403 <sup>1</sup>G. R. Cotton. *Rubber Chem. Technol.* **57**, 118 (1984). DOI: 10.5254/1.3535988.

404 <sup>2</sup>M. Astruc, Thesis, *l'École des Mines de Paris*, (2008).

405 <sup>3</sup>E. S. Dizon. *Rubber Chem. Technol.* **49**, 12–27 (1976). DOI: 10.5254/1.3534941.

406 <sup>4</sup> W. M. Hess, R. A. Swor, E. J. Micek. *Rubber Chem. Technol.* **57**, 959 (1984). DOI:  
407 10.5254/1.3536052.

408 <sup>5</sup>V. Collin, I. Boudimbou, E. Peuvrel-Disdier. *J. Polym. Sci.* **127**, 2121 (2013). DOI:  
409 10.1002/app.37769.

410 <sup>6</sup>S. Horiuchi, H. Dohi. *Langmuir.* **22**, 4607 (2006). DOI: 10.1021/la052308f.

411 <sup>7</sup>B. Huneau, I. Masquelier, Y. Marco, V. Le Saux, S. Noizet, C. Schiel, P. Charrier. *Rubber*  
412 *Chem. Technol.* **89**, 126 (2016). DOI: 10.5254/rct.15.84809.

413 <sup>8</sup>G. Weng, G. Huang, H. Lei, L. Qu, Y. Nie, J. Wu. *Polym. Degrad. Stab.* **96**, 2221 (2011). DOI:  
414 10.1016/j.polymdegradstab.2011.09.004.

415 <sup>9</sup>R. Liu, E. Sancaktar. *Int. J. Fatigue.* **111**, 144 (2018). DOI: 10.1016/j.ijfatigue.2018.01.033.

416 <sup>10</sup>E. S. Dizon, A. E. Hicks, V. E. Chirico. *Rubber Chem. Technol.* **47**, 231-249 (1974). DOI:  
417 10.5254/1.3540429.

418 <sup>11</sup>G. J. Lake, P. B. Lindley. *J. Polym. Sci.* **8**, 707 (1964). DOI: 10.1002/app.1964.070080212.

419 <sup>12</sup>F. Abraham. In *Constitutive models for rubber VII*. S. Jerrams, N. Murphy, Eds. CRC Press  
420 Inc, London, **pp 331** (2012).

421 <sup>13</sup>J. -B. Le Cam, B. Huneau, E. Verron. *Int. J. Fatigue* **52**, 82 (2013). DOI:  
422 10.1016/j.ijfatigue.2013.02.022.

423 <sup>14</sup>A. N. Gent, P. B. Lindley. *Proc. Math. Phys. Eng.* **249**, 195 (1959). DOI:  
424 10.1098/rspa.1959.0016.

425 <sup>15</sup>A. N. Gent, P. B. Lindley, A. G. Thomas. *J. Polym. Sci.* **8**, 455 (1964). DOI:  
426 10.1002/app.1964.070080129.

427 <sup>16</sup>G. J. Lake, P. B. Lindley. *J. Polym. Sci.* **9**, 1233 (1965). DOI: 10.1002/app.1965.070090405.

428 <sup>17</sup>A. N. Gent, B. Park. *J. Mater. Sci.* **19** 1947 (1984). DOI: 10.1007/BF00550265.

429 <sup>18</sup>T. S. Chow. *J. Mater. Sci.* **15**, 1873 (1980). DOI: 10.1007/BF00550613.

430 <sup>19</sup>S. Otto, O. Randl, O. Goncalves, B. Cantaloube. *KGK Kautschuk Gummi Kunststoffe.* **58**, 390

431 (2005).

432 <sup>20</sup>S. C. Garcea, Y. Wang, P. J. Withers. *Compos. Sci. Technol.* **156**, 305 (2018). DOI:  
433 10.1016/j.compscitech.2017.10.023.

434 <sup>21</sup>E. Maire, P. J. Withers. *Int. Mater. Rev.* **59**, 1 (2014). DOI:  
435 10.1179/1743280413Y.0000000023.

436 <sup>22</sup>S. Robin, T. Alshuth. In *Constitutive Models for Rubber VIII*. N. Gil-Negrete, A. Alonso.  
437 Eds.; Taylor & Francis, (2013).

438 <sup>23</sup>M. El Yaagoubi, D. Juhre, J. Meier, N. Kröger, T. Alshuth, U. Giese. *Int. J. Fatigue.* **112**, 341  
439 (2018) DOI: 10.1016/j.ijfatigue.2018.03.024.

440 <sup>24</sup>O. Gehrman, M. El Yaagoubi, H. El Maanaoui, J. Meier. *Polym. Test* **75**, 229 (2019) DOI:  
441 10.1016/j.polymertesting.2019.02.025.

442 <sup>25</sup>J. Kallungal, L. Chazeau, J. -M. Chenal, J. Adrien, E. Maire, C. Barres, B. Cantaloube, P.  
443 Heuillet. In *Constitutive Models for Rubber XI*; CRC Press, pp 77 (2019). DOI:  
444 10.1201/9780429324710-14.

445 <sup>26</sup>A. I. Medalia, F. A. Heckman. *Carbon.* **7**, 567 (1969). DOI: 10.1016/0008-6223(69)90029-3.

446 <sup>27</sup>A. I. Medalia, F. A. Heckman. *J. Colloid Interface Sci.* **36**, 173 (1971) DOI: 10.1016/0021-  
447 9797(71)90162-7.

448 <sup>28</sup>A.I. Medalia. *J. Colloid Interface Sci.* **32**, 115 (1970). DOI: 10.1016/0021-9797(70)90108-6.

449 <sup>29</sup>A.I. Medalia. *J. Colloid Interface Sci.* **24**, 393 (1967). DOI: 10.1016/0021-9797(67)90267-6.

450 <sup>30</sup>X. Bourrat, A. Oberlin, H. Van Damme, C. Gateau, R. Bachelar. *Carbon.* **26**, 100 (1988).  
451 DOI: 10.1016/0008-6223(88)90016-4.

452 <sup>31</sup>T. P. Rieker, M. Hindermann-Bischoff, F. Ehrburger-Dolle. *Langmuir* **16**, 5588 (2000). DOI:  
453 10.1021/la991636a.

454 <sup>32</sup>G. P. Baeza, A.- C. Genix, C. Degrandcourt, L. Petitjean, J. Gummel, M. Couty. J. Oberdisse.  
455 *Macromolecules.* **46**, 317 (2013). DOI: 10.1021/ma302248p.

456 <sup>33</sup>C. R. Herd, G. C: McDonald, W. M. Hess. *Rubber Chem. Technol.* **65**, 107 (1992). DOI:  
457 10.5254/1.3538594.

458 <sup>34</sup>E. Gouillart, J. Nunez-Iglesias, S. van der Walt. *Adv. Struct. Chem.* **2**, 18 (2017). DOI:  
459 10.1186/s40679-016-0031-0.

460 <sup>35</sup>E. Gouillart *Github*. [https://github.com/emmanuelle/image-processing-](https://github.com/emmanuelle/image-processing-tutorials/blob/master/tomo_segmentation.ipynb)  
461 [tutorials/blob/master/tomo\\_segmentation.ipynb](https://github.com/emmanuelle/image-processing-tutorials/blob/master/tomo_segmentation.ipynb).

462 <sup>36</sup>E. Gouillart, M. J. Toplis, J. Grynberg, M. -H. Chopinet, E. Sondergard, L. Salvo, M. Suéry,  
463 M. Di Michiel, G. Varoquaux, *J. Am. Ceram. Soc.* **95**, 1504 (2012). DOI: 10.1111/j.1551-  
464 2916.2012.05151.x.

465 <sup>37</sup>L. Grady. Random walks for image segmentation *IEEE Trans. Pattern Anal. Mach. Intell.* **28**,  
466 11 (2006). DOI: 10.1109/TPAMI.2006.233.

467 <sup>38</sup>S. van der Walt, J. L. Schönberger, J. Nunez-Iglesias, F. Boulogne, J. D. Warner, N. Yager,  
468 E. Gouillart, T. Yu. *PeerJ* **2**, e453 (2014). DOI: 10.7717/peerj.453.

469 <sup>39</sup>E. P. Denis, C. Barat, D. Jeulin, C. Ducottet. *Mater. Charact.* **59**, 338, (2008).  
470 DOI:10.1016/j.matchar.2007.01.012.

471 <sup>40</sup>M. del M. Fernández-Arjona, J. M. Grondona, P. Granados-Durán, P. Fernández-Llebrez, M.  
472 D. López-Ávalos. *Front. Cell. Neurosci.* **11**, 235 (2017). DOI: 10.3389/fncel.2017.00235.

473 <sup>41</sup>A. Madra, J. Adrien, P. Breilkopf, E. Maire, F. Trochu. *Compos. Part A Appl. Sci.* **102**, 184  
474 (2017). DOI: 10.1016/j.compositesa.2017.07.028.

475 <sup>42</sup>F. Pedregosa, G. Varoquaux, A. Gramfort, V. Michel, B. Thirion, O. Grisel, M. Blondel, P.  
476 Prettenhofer, R. Weiss, V. Dubourg, J. Vanderplas, A. Passos, D. Cournapeau, M. Brucher, M.  
477 Perrot, É. Duchesnay. *J Mach Learn Res* **12**, 2825 (2011). DOI: 10.48550/arXiv.1201.0490.

478 <sup>43</sup>J. Hämmäläinen, S. Jauhiainen, T. Kärkkäinen. *Algorithms.* **10**, 105 (2017). DOI:  
479 10.3390/a10030105.

480 <sup>44</sup>D. L. Davies, D. W. Bouldin. *IEEE Trans. Pattern Anal. Mach. Intell. PAMI-1*, vol **PAMI-**

481 **1**, no. 2, (1979). DOI: 10.1109/tpami.1979.4766909.

482 <sup>45</sup>P. J. Rousseeuw. *J. Comput. Appl. Math.* **20**, 53 (1987). DOI: 10.1016/0377-0427(87)90125-  
483 7.

484 <sup>46</sup>T. Villmow, P. Pötschke, S. Pegel, L. Häussler, B. Kretschmar. *Polymer.* **49**, 3500 (2008).  
485 DOI: 10.1016/j.polymer.2008.06.010.

486 <sup>47</sup>S. P. Rwei, I. Manas-Zloczower, D. L. Feke. *Polym Eng Sci.* **31**, 558 (1991). DOI:  
487 10.1002/pen.760310804.

488 <sup>48</sup>V. Collin. *Thesis*,. *École Nationale Supérieure des Mines de Paris.* (2009).

## 489 **10 APPENDIX**

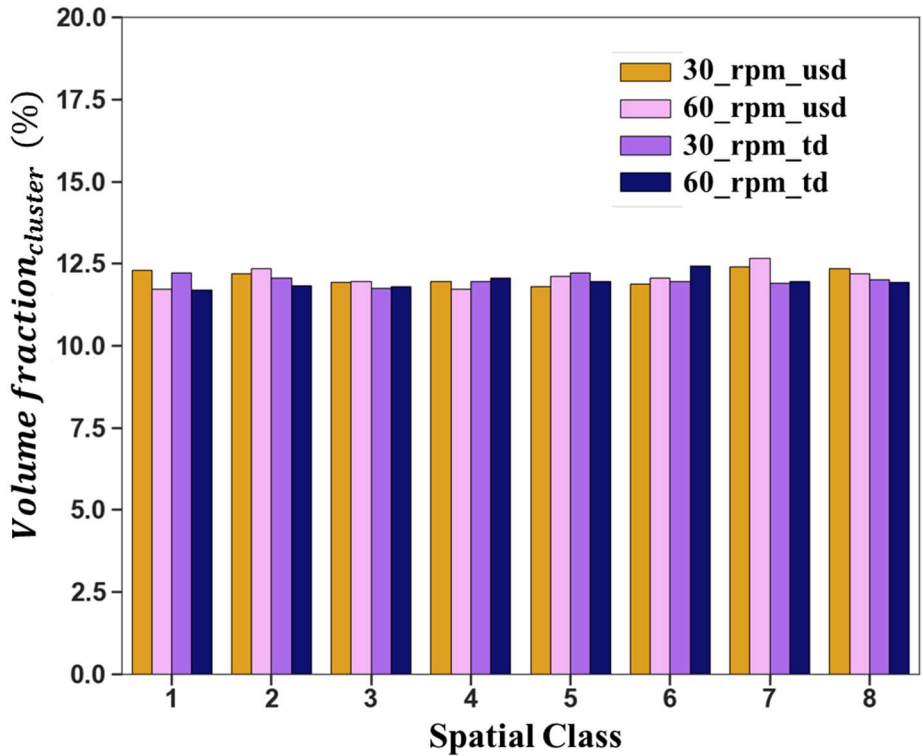
### 490 **Appendix A: spatial distribution**

491 In addition to the agglomerates morphologies, the spatial distribution of agglomerates in a mix  
492 is another important information to characterise the quality of the filler dispersion. To  
493 understand the aggregation of agglomerates at the micron level, it is interesting to calculate the  
494 distances to the nearest neighbours of each  $CB_{\text{aggl}}$ . In order to do it, a matrix with the distances  
495 between each agglomerate and all other agglomerates in the model material has to be evaluated,  
496 based on their position. Since the number of agglomerates in each model material is very high,  
497 it requires huge computation power and memory storage. To reduce this computation and the  
498 memory requirement, spatial clustering was performed on each model material using the same  
499 K-means algorithm applied for morphological clustering, except that the data points are the  
500 spatial coordinates of the agglomerates. All the agglomerates that are close to each other  
501 (centroids of the agglomerates) in the model material are clustered into one spatial class. Here  
502 8 classes have been identified as the optimal number, which translates to around 12% of the  
503 volume fraction for each spatial class in the stack (Appendix Figure A1). The distribution of  
504 the spatial distance of each agglomerate in each class to the class centroid  $D_{\text{centroid}}$  ( $D_{\text{centroid}} =$   
505  $(x_i - \mu_k)^{0.5}$ , where  $x_i$  is the spatial coordinates of agglomerate center and  $\mu_k$  is the centroid of the

506  $k^{\text{th}}$  spatial class ( $1 \leq k \leq 8$ ) gives insight on the agglomerates spatial distribution. The  
507 distributions found for the 8 spatial classes are very close to each other (Appendix Figure A2a)  
508 indicating that each class can be considered as a representative volume of the model mix. As  
509 mentioned earlier, we are more interested in the dispersion at lower scale. The clustering  
510 (equivalent to a reduction of the sample size) being completed, the distances between the 20  
511 nearest neighbours for each agglomerate in one class can therefore be calculated. 20 has been  
512 arbitrarily chosen, knowing that a parallelepiped in which an agglomerate can be bounded,  
513 counts 26 neighbors (in contact through the faces, vertices and edges). These distances  $d$  are  
514 the minimal ones between the surfaces of 2 agglomerates, as depicted in Appendix Figure A2b,  
515 and are calculated as follows:

$$516 \quad d_i = \sqrt{(x_i - x_j)^2 + (y_i - y_j)^2 + (z_i - z_j)^2} - (r_i + r_j) \quad (1 \leq j \leq 20)(1 \leq i \leq M) \quad (9)$$

517 Where  $x, y, z$  are the spatial coordinates of the agglomerate centre,  $r$  its radius calculated from  
518 its  $Eq_{diam}$ , the index  $i$  is for the agglomerate considered,  $j$  for one of its 20 closest neighbours  
519 and  $M$  is the total number of agglomerates in the spatial class considered for the analysis . In  
520 Appendix Figure A3, we have chosen the spatial class n°1 for all the model materials.



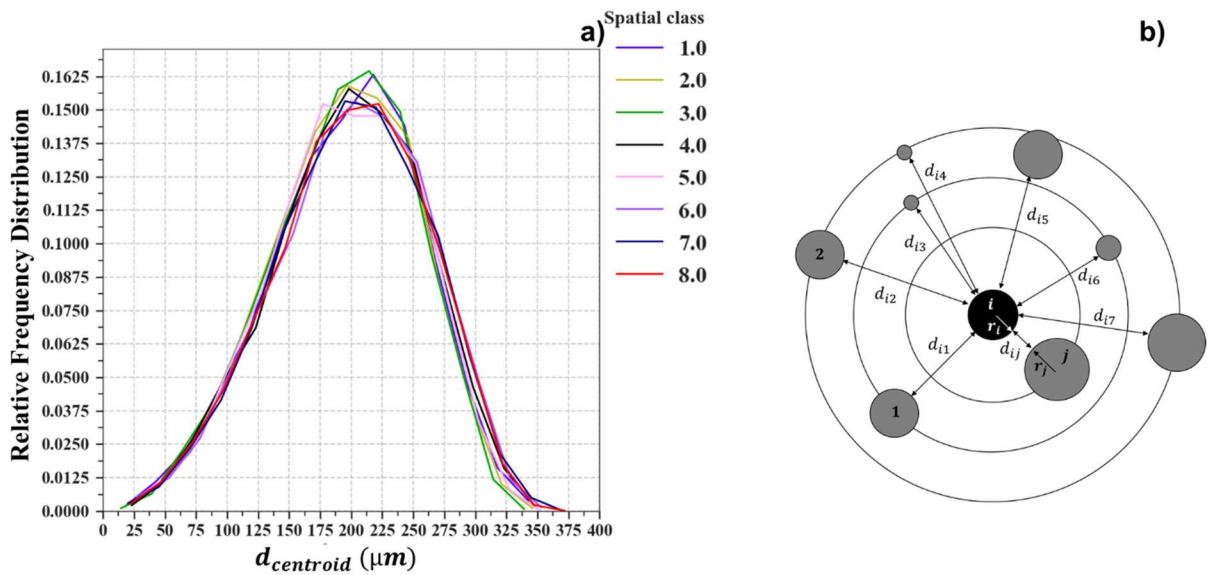
521

522 Appendix Figure A1: Volume fraction of each spatial class for 4 model materials.

523

524

525



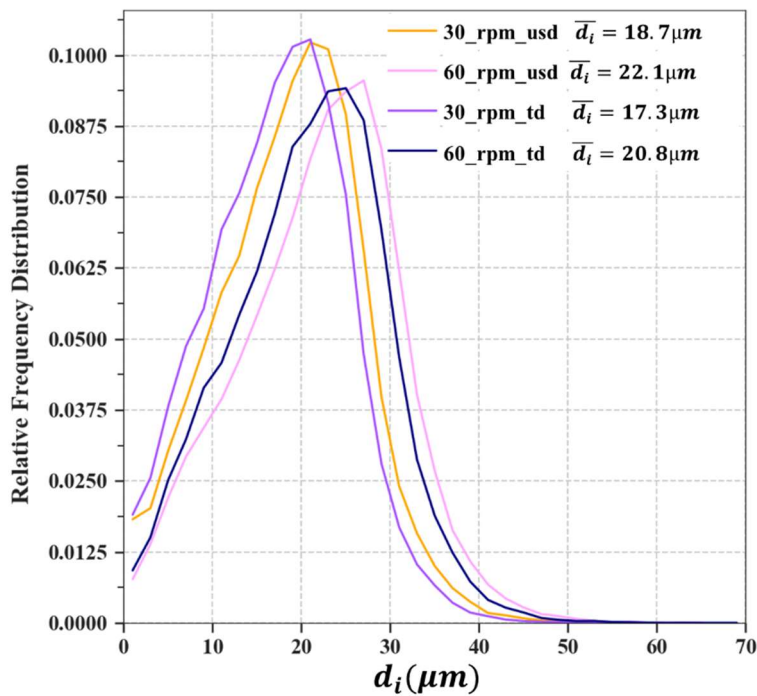
526

527 Appendix Figure A2: a) Distribution of the distances of each agglomerate to its class

528 centroid. b) Illustration of the nearest-neighbour distance



529 **Result: Spatial Distribution of CB<sub>aggl</sub>**



530

531 Appendix Figure A3: Distribution of nearest-neighbour distance for various model mixes

532 along with the mean value for each mix.  $\bar{d}_i$  corresponds to the mean of the distribution

Model mixes	WMH ( $\mu m$ )
30_rpm_usd	18.9
60_rpm_usd	20.9
30_rpm_td	18.1
60_rpm_td	20.1

533 Appendix Table A : Value of the Width at MidHeight of the peak of the nearest neighbor

534 distance distribution for the model mixes

535 The distributions of the distances to the nearest neighbours are presented in Appendix Figure

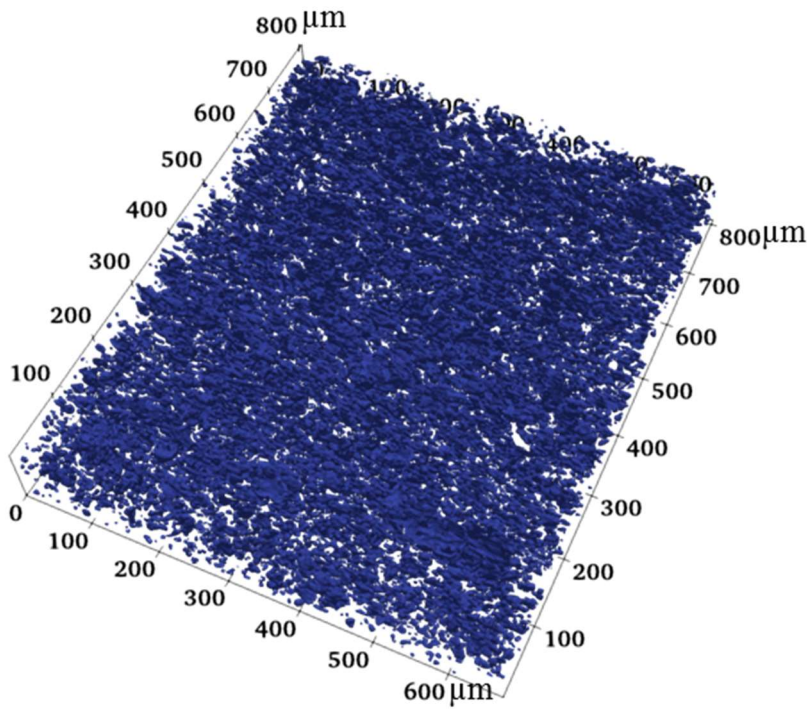
536 A3 for all the model mixes. There are slight differences between the different materials.

537 Increasing the rotor speed leads to a shift in the distribution towards larger values. The Width

538 of the peak distribution at its MidHeight (cf Appendix Table A1) is also slightly increased. In

539 addition, we know that the agglomerates volume fraction is reduced. Thus, the increase in the  
540 rotor speed improves the CB deagglomeration, reduces the size and volume fraction of the  
541 agglomerates, and logically increases the distance between them.

542 **Appendix Figure B: Sub-section volume of the model material 30\_rpm\_usd**



543  
544 Appendix Figure B: Sub-section volume of the model material 30\_rpm\_usd obtained using 3D  
545 Tomography visualized using ParaView © software

546 **Appendix C: laplacian score**

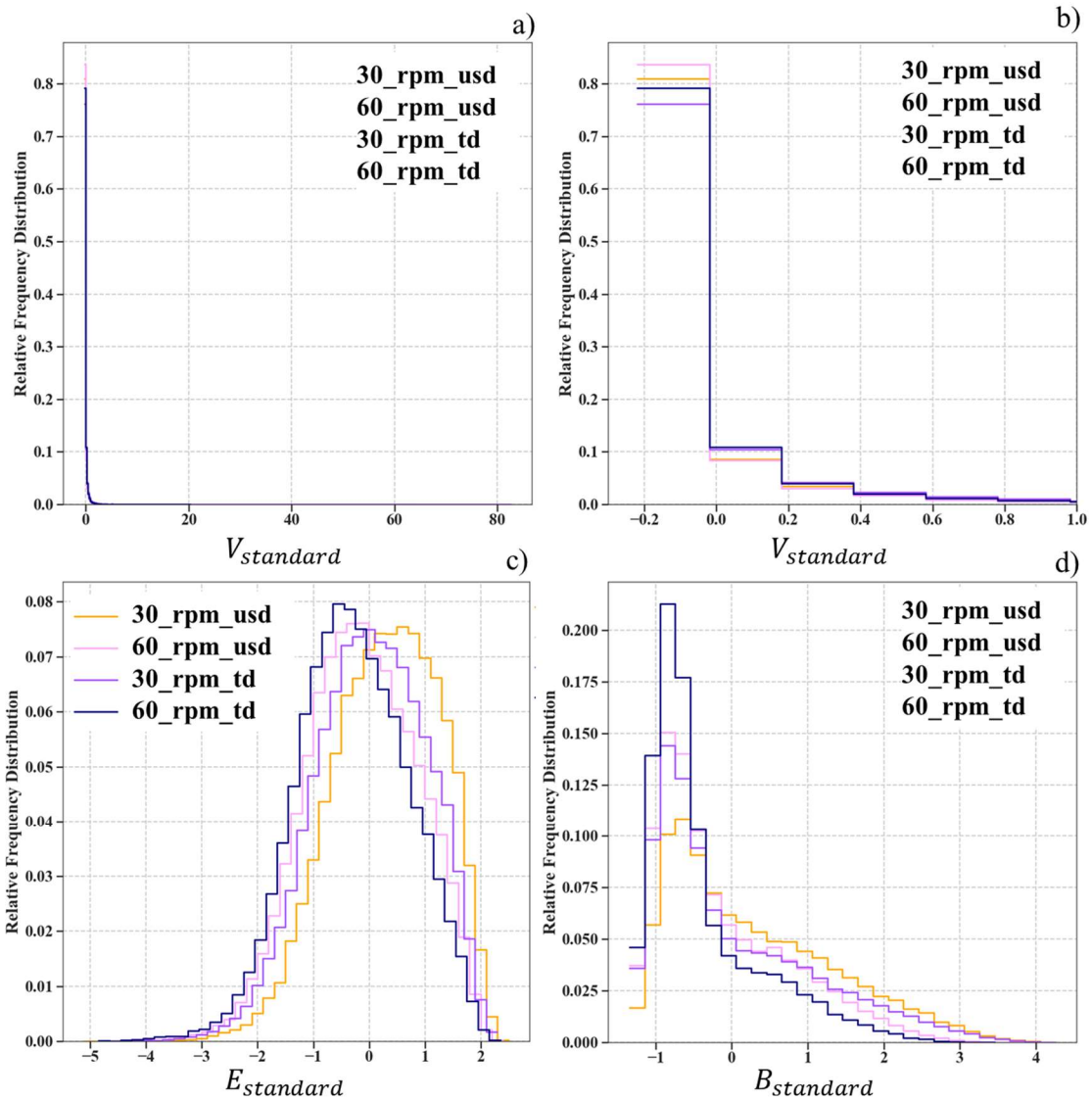
547 Laplacian Score (LS) works on the principle that data from the same class tend to be close to  
548 each other and each feature is evaluated based on its “locality preserving power”. LS ranges  
549 from 0 to 1, where score value close to 1 corresponds to a relevant parameter. The algorithm  
550 developed by He *et al.* <sup>45</sup> has been used to calculate the Laplacian score for each feature  
551 (Appendix Table C). Laplacian score shows that each feature can be used to distinguish  
552 different clusters in the data as all the features have a LS above 0.9, except *B* (0.85).

553

$V$	$E_{qdiam}$	$B$	$E$	$b$	$a$	$Q$
1.000	1.000	0.851±0.034	0.920±0.009	0.972±0.004	0.970±0.003	0.925±0.010

554 Appendix Table C: Laplacian score for each feature

555 **Appendix Figure D**



556

557 Appendix Figure D: Statistics on each standardized geometric features(using equation 8),

558  $V_{standard}$  ((a), (b)),  $E_{standard}$  (c) and  $B_{standard}$  (d)- which have been retained as descriptors of the

559 morphology of  $CB_{aggl}$  for all the model mixes (30\_rpm\_usd, 60\_rpm\_usd, 30\_rpm\_td,

560 60\_rpm\_td) - is presented

561 **Appendix E: silhouette coefficient**

562 The Silhouette coefficient ( $S$ )<sup>42</sup> is a validation metric to check the consistency within  
 563 clusters of data. It calculates the cohesion of an observation to its own cluster compared to other  
 564 clusters using the following equation.

565 
$$S(i) = \frac{b(i)-a(i)}{\max\{a(i),b(i)\}} \quad (7)$$

566 Where  $b(i)$  is the mean distance of  $i^{th}$  data point to all other data points in the closest  
 567 neighboring cluster to which  $i$  is not linked, i.e. between an agglomerate and agglomerates from  
 568 other clusters (mean nearest-cluster distance) and  $a(i)$  is the average distance of  $i^{th}$  observation  
 569 to all other observations in the same cluster (mean intra-cluster distance).  $S(i)$  varies between  
 570 -1 and 1, where 1 corresponds to a good partition of a cluster, score near 0 indicates overlapping  
 571 of clusters space and negative value signifies that observations have been partitioned wrongly.  
 572 We will use  $S_{avg}$  which is the average of all the  $S(i)$  values (Appendix Table E).

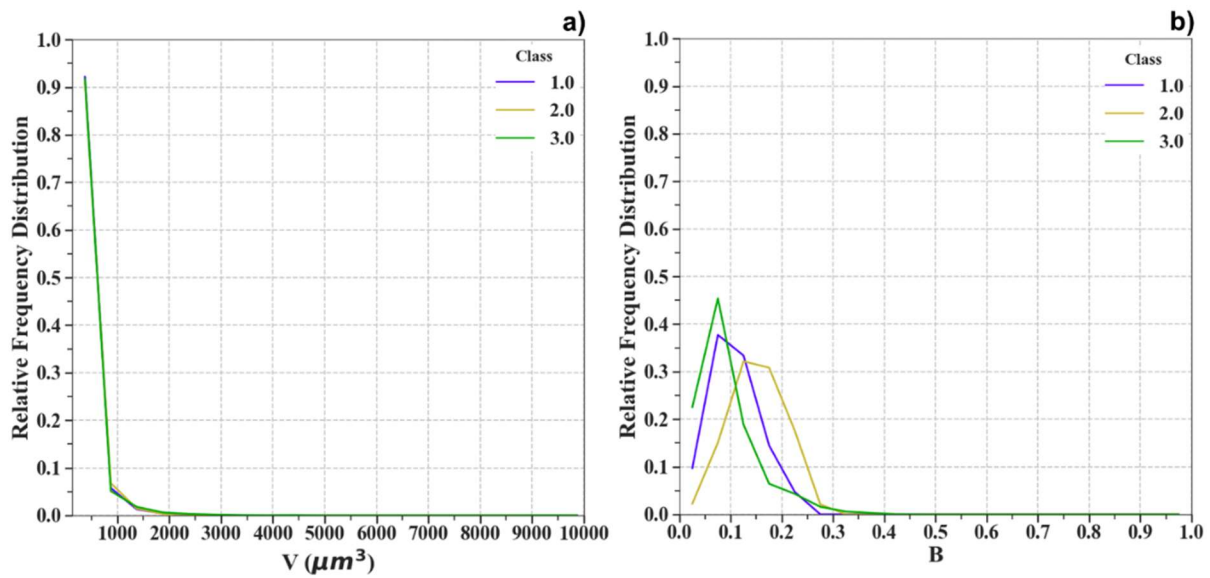
$k$ number of clusters	$S_{avg}$
2	0.448±0
3	0.450±0
4	0.371±0
5	0.365±0
6	0.347±0
7	0.358±0
8	0.351±0
9	0.344±0
10	0.327±0
11	0.322±0
12	0.326±0
13	0.323±0
14	0.315±0

573 Appendix Table E:  $S_{avg}$  metric evaluation for each number of clusters, deduced from 10 random  
 574 samplings

575

576

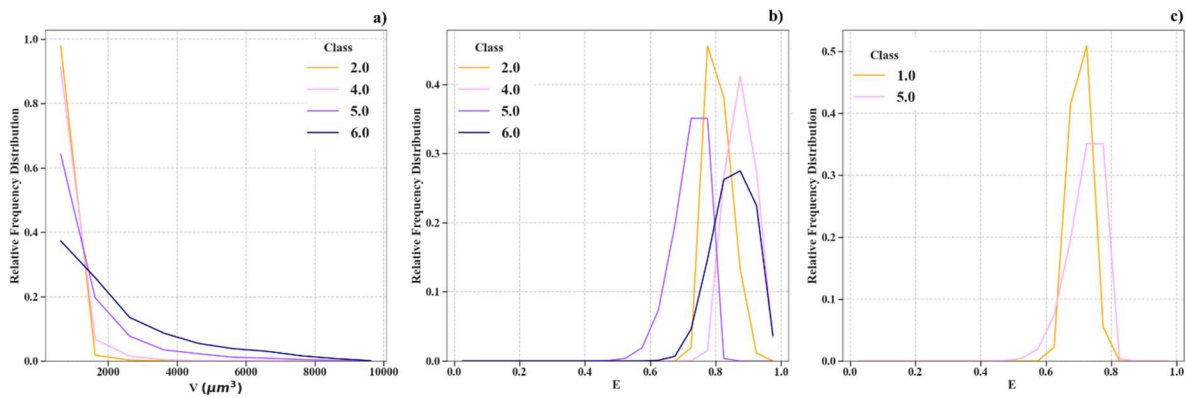
577 **Appendix Figure F**



578

579 Appendix Figure F: a) and b) Show the relative frequency distribution plot of  $V$  and  $B$  for  
 580 model material 30\_rpm\_usd corresponding to morphological classes #1, #2 and #3.

581 **Appendix Figure G**



582

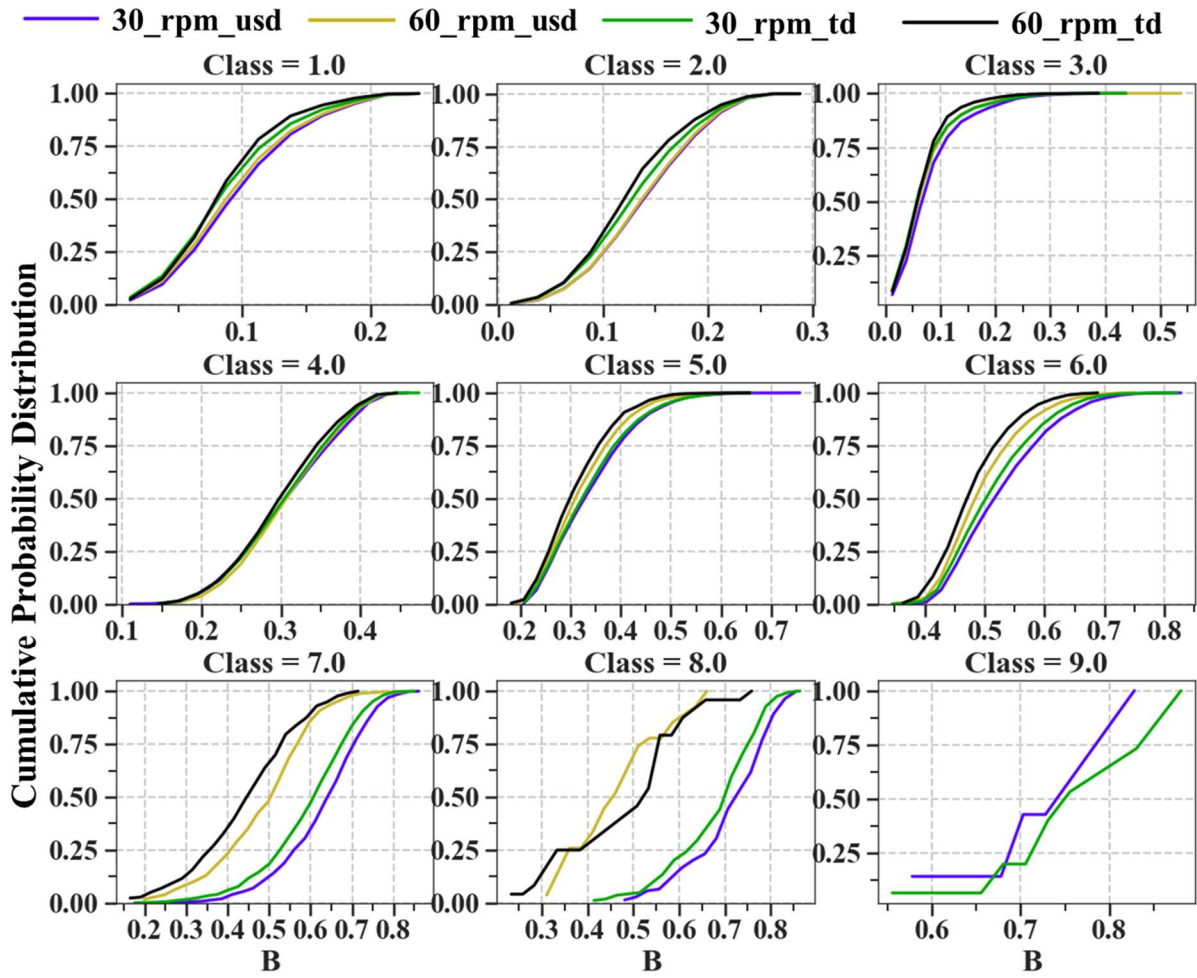
583 Appendix Figure G: a) and b) Show the relative frequency distribution plot of  $V$  and  $E$   
 584 respectively corresponding to morphological classes #2, #4, #5 and #6, c) shows the relative  
 585 frequency distribution plot of  $E$  corresponding to morphological classes #1 and #5 . The  
 586 agglomerates considered for plotting is from model material 30\_rpm\_usd

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590 Appendix Figure H



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592 Appendix Figure H: Cumulative probability distribution of B within each morphological class  
593 corresponding to each model material.

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602 **Appendix Table I**

<b>Protocol</b>	<b>Time (min)</b>	<b>Total rotation (rad)</b>	<b>Cohesive (gm)</b>	<b>Loose Carbon Black (gm)</b>	<b>Total mass (gm)</b>	<b>% of Carbon Black loose</b>
USD @ 60 rpm	2	753.6	53.1	1.8	54.9	3.3
TD @ 60 rpm	2	753.6	54.4	0.2	54.6	0.4
USD @ 30 rpm	2	376.8	52	2.392	54.392	4.4
TD @ 30 rpm	2	376.8	51	1.761	52.76	3.3

603 Appendix Table I: Amount of Carbon Black incorporated depending on the filler addition  
 604 protocol and rotor speed of the internal mixer. (This information was obtained for a mix  
 605 fabricated using the filler addition protocol suggested in Materials and Testing). This table  
 606 demonstrates the differences in the CB incorporation in the mix depending on the time spent in  
 607 the internal mixer.

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