

Durability Requirements for Reinforced Concrete Structures Placed in a Hostile Tropical Coastal Environment

Abel Castañeda Valdés, Francisco Corvo Pérez, Ildefonso Pech Pech, Rigoberto Marrero Águila, Emilio Bastidas-Arteaga

To cite this version:

Abel Castañeda Valdés, Francisco Corvo Pérez, Ildefonso Pech Pech, Rigoberto Marrero Águila, Emilio Bastidas-Arteaga. Durability Requirements for Reinforced Concrete Structures Placed in a Hostile Tropical Coastal Environment. Buildings, 2024, 14 (8), pp.2494. 10.3390/buildings14082494. hal-04672549

HAL Id: hal-04672549 <https://hal.science/hal-04672549v1>

Submitted on 19 Aug 2024

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

Article **Durability Requirements for Reinforced Concrete Structures Placed in a Hostile Tropical Coastal Environment**

Abel Castañeda Valdés 1,* [,](https://orcid.org/0000-0002-2531-9224) Francisco Corvo Pérez ² [,](https://orcid.org/0000-0001-7042-7382) Ildefonso Pech Pech ² , Rigoberto Marrero Águila ³ and Emilio Bastidas-Arteaga 4,[*](https://orcid.org/0000-0002-7370-5218)

- ¹ Material Protection Laboratory, Environment Division, National Center for Scientific Research, Havana 6412, Cuba
- ² Corrosion Research Center, Autonomous University of Campeche, San Francisco de Campeche, Campeche 24039, Mexico; frecorvo@uacam.mx (F.C.P.); ildepech@uacam.mx (I.P.P.)
- ³ Engineering Chemistry School, Technological University of Havana, Havana 19390, Cuba; rigo@quimica.cujae.edu.cu
- ⁴ Laboratory of Engineering Sciences for Environment (LaSIE) UMR CNRS 7356, La Rochelle University, Avenue Michel Crépeau, 17042 La Rochelle, France
- ***** Correspondence: abel.castaneda@cnic.cu (A.C.V.); ebastida@univ-lr.fr (E.B.-A.)

Abstract: In this work, a series of durability requirements are proposed for the construction of longservice-life reinforced concrete (RC) structures in a coastal environment with extreme atmospheric corrosivity. RC specimens were exposed in a coastal outdoor site in Cuba for three years. Carbon steel corrosion evaluation revealed an annual average atmospheric corrosion rate over the maximum limit established (ISO 9223:2012) for extreme (CX) atmospheric corrosivity. The service life of the RC structures, considered as the sum of the time-to-corrosion-initiation and time-to-corrosion-induced concrete cracking, was determined as a function of durability requirements. The most important durability requirements to achieve a long service life (>70 years) in RC structures subjected to a CX corrosivity category were defined as follows: water/cement ratio, compressive strength, percentage of effective capillary porosity, and concrete cover thickness. Under these hostile environments, the expansion of the corrosion products formed on the reinforcement steel and the induced cracking of the concrete could be attributed partially to the formation of the akaganeite phase in reinforcement steel, which revealed a different morphology compared to the akaganeite typically formed in bare carbon steels.

Keywords: durability requirements; service life; reinforced concrete; corrosivity categories; corrosion; chloride ingress; akaganeite

1. Introduction

Reinforced concrete (RC) dominates global construction due to its cost-effectiveness, streamlined manufacturing, and capacity to create a wide array of sections and structural elements $[1,2]$. Instances of a short service life (S_l) for these assets are frequently documented, particularly in coastal environments, where atmospheric corrosion of rebar stands as the primary factor influencing the premature degradation of these structures [3–8]. Since maintaining the aesthetic, functional, and safety aspects of RC structures in coastal environments are major challenges, establishing concrete durability prerequisites to ensure a suitable *S^l* for coastal RC structures becomes crucial to avoid direct and indirect economic losses [9,10].

Different research studies related to the durability of concrete and *S^l* of RC structures have been carried out under laboratory and on-site conditions during the few last years. Laboratory conditions generally simulate accelerated coastal environments. In [11–14], concrete and RC specimens were placed in salinity chambers and/or under total immersion in a NaCl solution simulating seawater. In addition, samples were exposed to the open

Citation: Castañeda Valdés, A.; Corvo Pérez, F.; Pech Pech, I.; Marrero Águila, R.; Bastidas-Arteaga, E. Durability Requirements for Reinforced Concrete Structures Placed in a Hostile Tropical Coastal Environment. *Buildings* **2024**, *14*, 2494. [https://doi.org/10.3390/](https://doi.org/10.3390/buildings14082494) [buildings14082494](https://doi.org/10.3390/buildings14082494)

Academic Editor: Grzegorz Ludwik Golewski

Received: 28 June 2024 Revised: 24 July 2024 Accepted: 7 August 2024 Published: 12 August 2024

Copyright: © 2024 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license [\(https://](https://creativecommons.org/licenses/by/4.0/) [creativecommons.org/licenses/by/](https://creativecommons.org/licenses/by/4.0/) $4.0/$).

atmosphere in rural and urban sites in Cuba. To simulate a short distance to the shoreline, a direct salt spray was applied [15,16]. Other research studies have been carried out in simulated concrete pore solutions [17–21]. To measure the corrosion rate of reinforcement steel at the time of exposure, different electrochemical techniques were used. In the research studies reported above, durability requirements in the concrete were not considered.

Regarding on-site diagnostic studies, durability requirements in the concrete are determined when the RC structures have reached a high level of deterioration due to the atmospheric corrosion of rebar in coastal environments [22–26].

Durability requirements are directly related to the concrete quality parameters determined before the construction of coastal RC structures as follows: water/cement (*w/c*) ratio, compressive strength (*fck*-MPa), ultrasonic pulse velocity *(UPS*-m s−¹), effective capillary porosity (*εe*-%), density (*D*-kg m−³), and concrete cover thickness (*CCth*-mm). After construction, the electrochemical corrosion rate (*Ic*-µA cm−²) is an indicator of the atmospheric corrosion of the rebar in RC.

According to Tuuti's model, the time-to-corrosion-initiation (*tⁱ*) and the time-tocorrosion-induced concrete cracking (t_{cc}) allow for the estimation of the S_l in RC structures [27,28]. These two stages constitute durability requirements that depend on concrete and exposure characteristics. Both stages can be assessed as a function of the atmospheric corrosion rate of reinforcement steel in RC. *tⁱ* is reached when a certain threshold of the free chloride ion concentration (% in mass of concrete or % in mass of cement) is reached on the reinforcement steel surface. However, the free chloride ion concentration capable of initiating the atmospheric corrosion of the rebar and the ingress of chloride ion salts into concrete from outdoor exposure conditions in coastal zones are controversial worldwide [5,29,30]. To guarantee the expected *S^l* in RC structures, each country has criteria to achieve durability targets. The *S^l* depends on the concrete quality and the corrosivity category of the atmosphere. A fundamental engineering tool in the establishment and confirmation of the durability requirements for concrete, as well as for the proper selection of primary and secondary protection systems based on mortars and paints, is the estimation and determination of the corrosivity categories of the atmosphere, mainly in coastal environments and particularly at short distances to the shoreline. At these sites, RC and metallic structures can be considered [31,32]. It is important to remark new trends in ISO standards, including paint protection systems for steel offshore and related structures [33]. This standard was approved in 2018 and is the first time that protective paint systems have been recommended particularly for very aggressive coastal and marine environments. The corrosivity categories of the atmosphere are defined according to the ISO 9223:2012 standard [34].

In order to predict the S_l in RC structures, different mathematical–statistical models have also been proposed. However, few models have been developed for studying the placement of RC specimens in zones with tropical coastal environments at a very short distance to the shoreline and for considering the durability requirements obtained for different concretes [23,24,35].

Few reports concerning the determination of durability requirements in concrete to guarantee a long *S^l* in RC structures have been carried out in Havana or in all Cuban coastal environments. This location is characterized by a very hostile environment where corrosion starts and propagates quickly. As a result, many RC structures built at a short distance to the shoreline in Havana and in other coastal environments in Cuba have presented a short *Sl* [36]. Therefore, the main objective of this paper is to define the durability requirements for concrete to guarantee an adequate S_l for RC structures placed in these hostile Cuban coastal environments.

RC specimens with *w/c* ratios varying between 0.4 and 0.6 and with cover thicknesses of 20 and 40 mm were placed in a coastal outdoor exposure site (COES) for three years. The COES was situated 10 m from the shoreline in Havana, Cuba. Concretes were prepared in a concrete production plant. During the first year of the study, we determined the corrosivity

category of the atmosphere. In addition, we also determined the following other concrete properties: f_{ck} -MPa, *UPS*-m s⁻¹, ε_e -%, and *D*-kg m⁻³. offer category of the atmosphere. In addition, we also determined

The *Ic*-μAcm⁻² measured was our indicator of the magnitude of atmospheric corrosion. t_i and t_{cc} were obtained according to the behavior and prediction of the atmospheric corrosion of the reinforcement steel at the time of exposure. Hence, the S_l as a function of the durability requirements of the concrete was determined and predicted for the RC structures in the Cuban coastal environment. The results were confirmed with visual observations. To further deepen the obtained results, microstructural analysis techniques

(DRX, SEM, and EDS) (DRX, SEM, and EDS) were used to characterize the rust layers in the third year of the study.

2. Experimental Procedure

2.1. Coastal Outdoor Exposure Site (COES)

RC specimens were exposed 10 m from the shoreline in Havana, Cuba, without a shielding effect (Figure 1a,b). The COES was located at a height of 11 m above sea level and placed at $23^{\circ}07'18.0''$ NL and $82^{\circ}25'45.0''$ WL.

Figure 1. Location of the COES in Havana, Cuba (a,b). MC means Meteorological Center in **b**. Dry plates and cellulose filters placed on a wooden rack (c). RC specimens (d). Ic measurement device (e).

During the first year of exposure, we determined the deposition of aggressive agents. During the first year of exposure, we determined the deposition of aggressive agents. We placed two dry plates (an absorbent cloth with dimensions of 320 mm \times 220 mm [37]) and two cellulose filter plates (porous filter paper with an alkaline surface with dimensions of 150 mm \times 100 mm [35]) on a wooden rack (Figure 1c). The dry plate device was used to assess the chloride deposition rate (Cl [−] DR -mg m^{−2}d). The cellulose filter plate device vice was used to measure the deposition rate of sulfur compounds (*SOx−DR*-mg m−2d). We was used to measure the deposition rate of sulfur compounds (*SO^x* [−]*DR*-mg m−2d). We positioned the four devices at a 45° angle from horizontal on the wooden rack, which was oriented toward the predominant wind direction to ensure the impaction of marine aerosols onto the surface. We placed these devices 3 m above the ground under a shed with $\sim 10^{11}$ a gabled roof, and they were protected from rain [37].

We exposed three RC specimens, each with a different *w/c* ratio, in the same open atmospheric conditions for three years: from 2008 to 2010 (Figure 1d). atmospheric conditions for three years: from 2008 to 2010 (Figure 1d).

2.2. Characterization of the Atmospheric Corrosivity Categories

The characterization of the atmospheric environment was carried out from October 2007 to September 2008 (first year). Two values of *Cl*−*DR* and *SO^x* [−]*DR* were determined monthly. The average values of the depositions of the two aggressive agents were monitored monthly. Monthly averages of meteorological parameters such as the relative humidity, *RH*, in %, air temperature, *T*, in °C, and wind speed, *WS*, in m s⁻¹, were provided by the Meteorological Center (MC) in Havana, located 500 m from the COES (Figure 1b).

We used dose–response functions established in the ISO 9223:2012 standard to determine the annual atmospheric corrosion rate (*rcorr*-µm y−¹) and to therefore establish the atmosphere corrosivity category for both exposed zinc and carbon steel at the COES [34]. The annual average values of *T*, *RH*, *Cl*−*DR*, and *SO^x* [−]*DR* were also monitored.

Regarding the dose–response functions, a wet candle device was used to determine the annual average *Cl*−*DR*. Thus, we recalculated the annual average *Cl*−*DR* data with the wet candle device (*Swc*) based on the annual average *Cl*−*DR* obtained from the dry plate device (*Sdp*) following the guidelines of the ISO 9225:2013 standard [34]:

$$
S_{wc} = 2.4 S_{dp} \tag{1}
$$

2.3. Preparation of Concrete and Reinforced Concrete (RC) Specimens

We used calcareous sand (modulus of fineness of 3.62) and ordinary Portland cement to produce the concrete. It also contained hard limestone gravel (nominal size of 19 mm) as its fine and coarse aggregates. Table 1 gives the chemical compositions of these materials.

Table 1. Chemical composition of materials.

Table 2 presents the three concrete mixtures considered in this study. Each mixture was characterized by a different water-to-cement ratio. All mixtures contained 365 kg/m^3 of cement, 1030 kg/m³ of fine aggregate (sand from a river), and 750 kg/m³ of coarse aggregate. These formulations aimed at reducing the voids between the fine and coarse aggregates, thus providing fluid consistency and ensuring good compaction.

Table 2. Concrete mixtures considered in this study (cement content was 365 kg/m³ for all mixtures).

w/c	Water (kg m $^{-3}$)	Superplasticiser (kg m^{-3})	$Slump$ (cm)	
0.6 ₂	222			
0.5	186	ל. ا		
0.4	148	17	15	

We prepared all the specimens according to Cuban standards [38]. Figure 2 shows the geometry of the specimens used in this study. A total of six specimens (two for each considered *w/c* ratio) were cast using carbon steel molds. Each specimen had rebars with different cover depths (*CCTh* of 20 and 40 mm). We placed three RC specimens with different w/c ratios at the COES. The remaining specimens were stored in the laboratory as references.

Figure 2. Geometry of RC specimens considered in this study. **Figure 2.** Geometry of RC specimens considered in this study.

We used carbon steel bars as the reinforcements (200 mm in length and 12.5 mm in We used carbon steel bars as the reinforcements (200 mm in length and 12.5 mm in diameter). Before immersion in the fresh concrete mixture, the rebars were subjected to diameter). Before immersion in the fresh concrete mixture, the rebars were subjected to chemical cleaning. To remove poor corrosion products from the rebar surface, we used a solution consisting of 500 mL of concentrated hydrochloric acid ($\rho = 1.19 \text{ g m}$ l⁻¹) and of hexamethylenetetramine diluted in distilled water up to 1000 mL. Afterwards, we 3.5 g of hexamethylenetetramine diluted in distilled water up to 1000 mL. Afterwards, we cleaned the rebars with a high-pressure water jet, dried them with a cloth, and placed them in an oven at 50 °C for 45 min. Subsequently, we stored the rebars in desiccators [38]. The rebars were immersed in the concrete mixture. We left 40 mm of each rebar outside the specimens to be connected to electrochemical instruments. We protected the exposed parts of the rebars with adhesive tape.

We also prepared six concrete specimens, two for each considered *w/c* ratio, with the We also prepared six concrete specimens, two for each considered *w/c* ratio, with the same form and dimensions as those in Figure 2 but without reinforcement steel. Additionally, we prepared 36 cylindrical concrete specimens (height = 300 mm and diameter = 150 mm), twelve for each *w/c* ratio, using carbon steel molds.

We used a vibrating table to compact both the RC and concrete specimens, ensuring We used a vibrating table to compact both the RC and concrete specimens, ensuring the removal of entrapped air in the concrete mixture for each cubic and cylindrical mold. the removal of entrapped air in the concrete mixture for each cubic and cylindrical mold. All molds were moistened before pouring in the concrete mixture. All molds were moistened before pouring in the concrete mixture.

The casted specimens remained in the molds for 24 h. Subsequently, they were cured
28 h. ~ 22.6 Feel for 28 days, with the average water temperature maintained at 23° C [39].

$2.47 \t\t 1.1 \t\t 1.1$ *2.4. Concrete Quality Analysis*

for ensuring concrete quality assessment. They refer to recommendations for the compres- $\frac{1}{2}$ for extending concrete $\frac{1}{2}$ referred to $\frac{1}{2}$ and effective capillary porosity compressive strength, density, and effective capital pulse velocity, and effective capital α We considered the procedures recommended by the DURAR net (CyTED program [40]) sive strength, density, ultrasonic pulse velocity, and effective capillary porosity.

2.4.1. Compressive Strength (*fck*)

We used cylindrical concrete specimens and a testing machine with a maximum axial compression force of 2000 kN mm⁻² to obtain 12 values of f_{ck} for all the considered *w/c* ratios using the method from [41] with a curing time of 28 days.

2.4.2. Density (*D*)

We determined 12 values of *D* in the fresh concrete for each *w/c* ratio. We measured the weight of each cubic carbon steel mold before pouring in the concrete mixtures. An AVERY balance, with a measuring range of from 200 g to 50 kg, was used to determine the density.

2.4.3. Ultrasonic Pulse Velocity (*UPS*)

We determined twelve *UPS* values (for each *w/c* ratio) using the three cubic unreinforced concrete specimens at a curing time of 28 days. We used the direct transmission method, obtaining four values between each pair of opposite sides [42]. We applied a thin layer of grease (petrolatum) to ensure good coupling and electric wave transmission between the transducers. We used a Tico Proceq Testing Instrument to perform these measurements. This equipment has a range of from 15 μs to 6550 μs (10 $^{-6}$ s) and a bandwidth of 54 Hz. Before starting the measurements, we calibrated the instrument using concrete reference bars.

2.4.4. Effective Capillary Porosity (*εe*-%)

We extracted three cylindrical concrete cores from the unreinforced cubic concrete specimens. We cut these cores using tungsten to obtain twelve cylindrical concrete specimens (diameter = 62 mm and thickness = 20 mm). This allowed us to determine twelve values of *ε^e* (for each *w/c* ratio) using the Göran Fagerlund methodology based on determining the capillary water absorption [43].

2.5. Electrochemical Characterization

We measured six electrochemical corrosion rate values (*Ic* in µA cm−²) for each rebar, i.e., for the two concrete cover depths (20 and 40 mm) and for the three exposed RC specimens, every year. These measures served as indicators of atmospheric corrosion. We measured the electrochemical corrosion rate for the reference specimens (stored in the laboratory) only during the first year.

We used a GECOR-8™ brand GEOCISA corrosimeter to measure the electrochemical corrosion rate. To enhance the electric conductivity of the measurement system, we placed a dampened cloth between the surfaces of the RC specimens and the instrument's sensor. We positioned the sensor on the surface of the RC specimens by pressing it by hand (Figure 1e). The polarized surface area introduced to the instrument was 65.25 cm^2 . The potential range was ± 20 mV with a sweep speed of 12 mV s⁻¹. The instrument enabled the determination of the electrochemical corrosion rate using the Stern–Geary equation.

$$
I_c = \frac{B}{A * R_p} \tag{2}
$$

where *A* is the rebar polarized surface (critical) area, *B* is a constant equal to 26 mV, and R_p is the polarization resistance. DURAR net (CyTED program) [40] provides various criteria regarding the electrochemical corrosion rate magnitude for the service life of RC structures.

To determine the RC structures' durability requirements, we studied twelve values of *Ic* obtained during the third exposure year when considering twelve values of *D*, *UPS*, *fck*, and *ε^e* for all the contemplated concrete covers and *w/c* ratios.

2.6. Statistical Analysis

To determine which of the parameters defining the concrete quality were the most influential in terms of *Ic* for both concrete covers during the entire study, two statistical methods were used in this research work: multiple regression fitting and a comparison from multiple range tests.

For both statistical analyses, *Ic* (atmospheric corrosion indicator) was taken as the dependent variable. Parameters defining the concrete quality (*w/c*, *fck*-MPa, *εe*-%, *D*-kg m−³ , *UPS*-m s−¹) were taken as the independent variables.

The multiple regression used for the fit $(n = 36)$ can be written as follows:

$$
Ic = a \pm b(w/c) \pm c(f_{ck}) \pm d(\varepsilon_e) \pm e(UPV) \pm f(D),
$$
\n(3)

where *a*, *b*, *c*, *d*, *e*, and *f* are coefficients that measure the increase or decrease in the effect of the independent variables on the dependent variable.

Regarding to the comparison from the multiple range tests (DUNCAN), homogeneous groups between the durability requirements (*Ic*-µA-cm−² , *w/c* ratio, *εe*-%, *fck*-MPa, *D*-kg m^{−3}, and *UPV*-m s^{−1}) were obtained according to the alignment on X homogeneous groups for both concrete covers during the entire duration of the study (*n* = 36). In order to confirm the result obtained, statistically significant differences between each pair of means were estimated (pairs: *Ic-w/c, Ic-f_{ck}, Ic-ε_e, Ic-UPV, Ic-D, w/c-f_{ck}, w/c-ε_e, w/c-UPV, w/c-D, f_{ck}-ε_e, fck-UPV*, *fck-D*, *εe-UPV*, and *εe-D*, *UPV-D*) for a limit of ±24.86 in both values of *CCTh*.

2.7. Prediction of Service Life (S^l)

To predict the S_l in the RC structures based on the estimation of t_i and t_{cc} , *Ic* was determined in the RC with a *w/c* ratio = 0.4 and values of *CCTh =* 20 and 40 mm, fitting the following linear power regression:

$$
I_c = r_{corr}t^k \tag{4}
$$

where *t* (years) is the exposure time (from 1 to 66 years), *Ic* (µAcm−²) is the electrochemical corrosion rate (rebar atmospheric corrosion indicator), and *rcorr* is the average *Ic* corresponding to the first year of study for the RC specimens with a *w/c* ratio of 0.4 and *CCTh* values of 20 and 40 mm exposed at the COES. According to ISO-9224:2012 [44], *k* is a time exponent equal to 0.52.

2.8. Microstructural Characterization

2.8.1. Scanning Electron Microscopy (SEM)

We obtained a sample of the rust layer from a rebar in the RC specimen with a *w/c* ratio of 0.6 and a *CCTh* of 20 mm after three years of exposure. The sample was observed by scanning electron microscopy and analyzed using an energy-dispersive X-ray detector to determine the chemical composition.

2.8.2. X-ray Diffraction (XRD)

We scraped off a powder sample from the same rust layer. We identified the crystalline phases using X-ray diffraction (XRD) with a Bragg–Brentano geometry and $CuK\alpha1$ radiation ($\lambda = 1.54056$ Å). We obtained the X-ray diffractogram by scanning from 5 to 80 $^{\circ}$ (2*θ*) in 0.02° steps with a time per step of 12 s.

3. Results

3.1. Description of the Typical Atmospheric Environment in the COES

3.1.1. Monthly Mean Values of *Cl*−*DR* and *SO^x* [−]*DR*

A difference in the behavior of the monthly mean values of *Cl*−*DR* and *SO^x* [−]*DR* between the cold (dry) and summer (wet) seasons was observed at the COES located at a 10 m distance from the shoreline during the first year of the study (Figure 3). Due to the entry of frontal systems (cold fronts) coming from the north, higher values were obtained during the cold season (November–April). Therefore, the marine aerosols in the coastal environment of Havana, mainly at a short distance from the shoreline, carried both aggressive agents.

Figure 3. Behavior of monthly mean values of Cl^-DR and SOx^-DR between the cold and summer seasons during the first year of study. seasons during the first year of study.

A fundamental factor in the early deterioration of metallic structures built using the A fundamental factor in the early deterioration of metallic structures built using the most common metallic materials in the construction industry (carbon and galvanized most common metallic materials in the construction industry (carbon and galvanized steels, copper, and aluminum) caused by atmospheric corrosion in the Cuban coastal environment is the combination of Cl^-DR and SO_x^-DR [43]. High (C4) and very high (C5) corrosivity categories of the atmosphere were determined for each metallic material [45]. categories of the atmosphere were determined for each metallic material [45].

Despite the fact that during the summer season (May–October) the entry of frontal Despite the fact that during the summer season (May–October) the entry of frontal systems (cold fronts) coming from the north does not occur, the monthly values of *Cl−DR* systems (cold fronts) coming from the north does not occur, the monthly values of *Cl*−*DR* and *SO_x*[−]*DR* can be considered high (Figure 3). In this way, existing and new RC structures could be exposed to high (C4), very high (C5), extreme (CX), and maybe higherthan-extreme (>CX) atmospheric corrosivity categories almost permanently. In order to ensure a long S_l in RC structures in Cuban coastal environments and worldwide, it is very necessary to consider the corrosivity categories of the atmosphere in terms of the durability requirements.

3.1.2. Monthly *RH*, *T*, and *WS* 3.1.2. Monthly *RH*, *T*, and *WS*

The monthly average *RH* can be considered high due to the strong permanent impact in the strong permanent impact of marine aerosols in the COES. The monthly average *RH* was higher in the summer season (May–October) due to the abundant rainfall (Figure 4). The annual average *RH* and *T* $\frac{1}{2}$ values were 78%, and 25.5°C, respectively. Very similar averages have been botained in the last five years in the coastal city of Havana [46,47]. Thus, chloride ion salts coming from
the coast wears and discovering comeals were densited and the PC structure on saling from the sea, transported by marine aerosols, were deposited onto the RC structures as a solution and not in the form of dry salt crystals. This situation was also reported in the coastal city of Recife, Brazil [48,49]. In this way, a higher penetration of chloride ion salts into concrete should occur, mainly in RC structures made of low-quality concrete, which salts into concrete should occur, mainly in RC structures made of low-quality concrete, present a high percentage level of the effective capillary porosity (*εe*). values were 78%, and 25.5◦C, respectively. Very similar averages have been obtained in the the sea, transported by marine aerosols, were deposited onto the RC structures as a saline

which present a high percentage level of the effective capillary porosity (*εe*). On the other hand, a higher monthly average *WS* and a lower monthly average *T* were determined in the cold season (November–April). Hence, the highest monthly average *Cl*−*DR*, and *SO^x* [−]*DR* were determined in the cold season due to the entry of cold fronts coming from the north (Figure 4).

A total of 17 cold fronts penetrated Havana City from October to April during the first year of the study, according to data from the Meteorological Centre of Havana (MC in Figure 1). Seven of these cold fronts were moderate. The monthly maximum sustained wind speed exceeded 9 m/s from November to March. Additionally, the monthly average *Cl[−]DR* was over 1000 mg m^{−2} d^{−1} (Figure 3).

The monthly average behavior of the above meteorological parameters during the first year of the study was consistent with the determination of high (C4), very high (C5), extreme (CX), and maybe higher-than-extreme (>CX) atmospheric corrosivity categories for RC structures.

Figure 4. Monthly behavior of *RH*, *T*, and *WS* during the first year of study. **Figure 4.** Monthly behavior of *RH*, *T*, and *WS* during the first year of study.

3.2. Corrosivity Categories of the Atmosphere

On the other hand, a higher monthly average *WS* and a lower monthly average *T* According to the *r_{corr}* determined by the dose/response function, an extreme (CX) at-*CORPIETR COLOSIVITY CALCOBY* WAS CHASSINED TOT CALPON SECT and *SALVANIZED SECT* (TADLE 0). In the case of galvanized steel, this involves the atmospheric corrosion of zinc. To build In the case of galvanized steel, this involves the almospheric corrosion of zinc. To band
RC structures in coastal zones worldwide, galvanized steel (zinc coating thickness higher than 21 μm) has been used as reinforcement steel [50–52]. Therefore, RC structures in the $f(x)$ and $f(x)$ and seen used us remore ement of $f(x)$ and $f(x)$, the structures in the coastal city of Havana, mainly at a short distance from the shoreline, are exposed to an extreme (CX) atmospheric corrosivity category. Hence, the phenomenon of the atmospheric extreme (CX) exceeding (ext) annot protect to y and the monthly average y. The monthly are protected to the annot protect corrosion of rebar has been or will be developed very intensively in RC structures built **COLOGETS OF DRIM** THE COLL OF THE COLLEGE 1000 methods is area of the corrosivity in this area, mainly those built with very-low-quality concrete. Therefore, the corrosivity category of the atmosphere can be taken as a durability requirement. first year of the study was consistent with the determination of $\frac{1}{2}$, very high (C4), very high (C5), v mospheric corrosivity category was classified for carbon steel and galvanized steel (Table 3).

Table 3. Atmospheric corrosivity categories for carbon steel and zinc.

mospheric corrosivity category was classified for carbon steel and galvanized steel (Table 3.3. Electrochemical Corrosion Rate (Ic) and Durability Requirements

The behavior of the annual average *Ic* plotted versus the time of exposure determined for the six types of RC specimens (*w/c* ratio 0.4, 0.5, and 0.6 and *CCTh* of 20 mm and 40 mm) is presented in Figure 5. For a *CCTh* value of 20 mm, the increase was more marked (Figure 5a,b). The influence of the w/c ratio and *CCTh* on the electrochemical corrosion rate as an indicator of the atmospheric corrosion of the rebar in the RC specimens under the real exposure conditions and those exposed to the extreme (CX) corrosivity category of the atmosphere in the coastal environment at a short distance from the shoreline can be thickness. These are the same factors that influence the R_p as well as the S_l of RC structures. clearly observed. *Ic* increased for large values of the *w/c* ratio and for a small concrete cover

structures.
Structures.

Water/cement ratio (w/c)

Figure 5. Behavior of annual average values of the twelve *Ic* values versus time of exposure (R = reference, 1 = first year, 2 = second year, and 3 = third year). $CCTh = 20$ mm (a). $CCTh = 40$ mm (b).

Water/cement ratio (w/c)

Based on the criteria established in the DURAR net (CyTED program) about the *Ic* Based on the criteria established in the DURAR net (CyTED program) about the *Ic* magnitude for the S_l of RC structures, the t_i and t_{cc} were obtained. The t_i was considered when a medium level ($0.1 < Ic < 0.5$) was reached, and the t_{cc} was considered when high $(0.5 < Ic < 1)$ and very high levels $(Ic > 1)$ were reached for the annual average values of in the RC specimens. According to Tuuti's model, the sum of both times allows for the *Ic* in the RC specimens. According to Tuuti's model, the sum of both times allows for the estimation of the S_l in RC structures as a function of the concrete quality.

In this way, for the RC specimen with a *w/c* ratio 0.4 and with a *CCTh* of 20 mm, *ti* In this way, for the RC specimen with a *w/c* ratio 0.4 and with a *CCTh* of 20 mm, *tⁱ* was reached in the second year of exposure (t_i = 2 years). This is a very low value of the t_i . However, the *tcc* was not reached during the three years of the study (Figure 5a). To predict However, the *tcc* was not reached during the three years of the study (Figure 5a). To predict the S_l when building RC structures, t_{cc} can be determined from the prediction of *Ic* as an indicator of atmospheric corrosion using the model (Equation (4)). According to the model indicator of atmospheric corrosion using the model (Equation (4)). According to the model obtained, the t_{cc} could be reached between six (t_{cc} = 6 years) and ten years (t_{cc} = 10 years) exposure (0.5 < *Ic* < 1) (Figure 6). Costly repair works must be initiated between these of exposure (0.5 < *Ic* < 1) (Figure 6). Costly repair works must be initiated between these values of the t_{cc} . Hence, the S_l of RC structures made from concrete with a w/c ratio of 0.4 and a *CCTh* of 20 mm exposed to an extreme (CX) atmospheric corrosivity category should be between eight and twelve years. A long *S*_{*l*} in RC structures is not ensured.

In the RC specimens with *w/c* ratios of 0.5 and 0.6, the t_i was reached in the first year $(t_i = 1$ year). The t_{cc} was reached in the third $(t_{cc} = 3$ years) and second year $(t_{cc} = 2$ years), respectively (Figure 5a). Therefore, the *Sl* of RC structures made from concrete with *w/c* respectively (Figure 5a). Therefore, the *S^l* of RC structures made from concrete with *w/c* ratios of 0.5 (S_l = 4 years) and 0.6 (S_l = 3 years) with a CCTh of 20 mm and exposed to an extreme corrosivity category (CX) does not exceed five years.

On the other hand, for the concrete with a *w/c* ratio of 0.4 and a *CCTh* of 40 mm, it seems that the reinforcement steels remained in a passive state during the three years seems that the reinforcement steels remained in a passive state during the three years of of the study. An increase in the *Ic* was not obtained. The annual average *Ic* remained the study. An increase in the *Ic* was not obtained. The annual average *Ic* remained at a at a negligible level (*Ic* < 0.1), as happened in the RC specimens taken as references and at a hegapiere level ($E \sim 0.1$), as happened in the *NC* specimens taken as references and kept in the laboratory (Figure 5b). However, the S_l could be predicted using the model (Equation (4)). According to the model obtained, the t_i could be reached after 42 years of exposure $(t_i = 42 \text{ years})$ (Figure 6). Maintenance works, although less costly than repair works, could be initiated from that t_i in RC structures. Therefore, a long S_l higher than On the other hand, for the concrete with a *w/c* ratio of 0.4 and a *CCTh* of 40 mm, it 70 years can be ensured in RC structures exposed to a CX corrosivity category in the Cuban coastal environment.

For the RC specimens with w/c ratios of 0.5 and 0.6, t_i was reached in the first $(t_i = 1$ year) and second year $(t_i = 2$ year), respectively. For both RC specimens, the t_{cc} was reached in the third (t_i = 3 year) and second year (t_i = 2 year) (Figure 5b). Therefore, the S_l for RC structures with w/c ratios of 0.5 (S_l = 5 years) and 0.6 (S_l = 3 years) and with a value of *CCTh* of 40 mm exposed to extreme (CX) corrosivity does not exceed five years. A similar result was obtained for a *CCTh* value of 20 mm.

environment.
Environment

Figure 6. Prediction of *Ic* for concrete with w/c ratio of 0.4 versus exposure time according to the fitted model. fitted model.

respond to the most used design conditions for Cuban RC structures [53,54]. A short *S*_{*l*}, respond to the most used design conditions for Cuban RC structures [53,54]. A short *S*_{*l*} recipend to the atmospheric corrosion of rebar, has been determined for many RC structures in due to the atmospheric corrosion of rebar, has been determined for many RC structures in all Cuban coastal environments [55]. Hence, to ensure a long S_l in RC structures built in structures with *w/c* ratios of 0.5 (*Sl* = 5 years) and 0.6 (*Sl* = 3 years) and with a value of *CCTh* Cuban and other tropical coastal environments worldwide, it is very necessary to define appropriate concrete durability requirements. Concretes elaborated with *w/c* ratios = 0.5 and 0.6 and *CCTh =* 20 and 40 mm cor-

Based on the behavior of the *Ic* determined in the third year of exposure for the RC specimens as a function of the w/c ratios, CCTh, and the corresponding values of f_{ck} , ε_e , UPS, and *D*, durability requirements were obtained.

The Ic decreased as function of f_{ck} , UPS, and D, while the opposite happened for the ε_e (Figure 7). It was observed that for a *CCTh* value of 40 mm and a w/c ratio of 0.4, the *Ic* was lower than 0.1 μ A cm⁻²; this is an indicator of the atmospheric corrosion of the rebar in the RC specimens exposed for the three years (Figure 7).

Figure 7. *Cont*.

Figure 7. Behavior of the twelve *Ic* values determined in the third year of exposure in the RC specimens as a function of the w/c ratios, CCTh, and the twelve values of f_{ck} , ε_e , UPS, and D. Changes in *Ic* vs. compressive strength (a), percentage of effective capillary porosity (b), ultrasonic pulse (**c**), and density (**d**) for different *w/c* ratios and *CCTh*. velocity (**c**), and density (**d**) for different *w/c* ratios and *CCTh*.

Therefore, it was confirmed that the rebars remained in a passive state for the differ-Therefore, it was confirmed that the rebars remained in a passive state for the different values of f_{ck} , ε_e , UPS, and D. To ensure a long S_l in RC structures that are currently under construction in Cuban coastal environments exposed to high (C4), very high (C5), under construction in Cuban coastal environments exposed to high (C4), very high (C5), extreme (CX), and above-extreme (>CX) atmospheric corrosivity categories, Table 4 pro-extreme (CX), and above-extreme (>CX) atmospheric corrosivity categories, Table 4 provides the ranges of these parameters directly linked with concrete quality that should be taken as durability requirements. In addition, *t*_{*i*} = 42 years must be taken as another bility requirement. durability requirement.

Table 4. Durability requirements directly linked with concrete quality parameters for CX corrosivity **Table 4.** Durability requirements directly linked with concrete quality parameters for CX corrosivity category. category.

w/c	Concrete Cover Thickness (mm)	f_{ck} (MPa)	ε_e (%)	UPV $(m s^{-1})$	໋ $(kg m^{-3})$	ıc $(\mu A \text{ cm}^{-2})$	ı. (Years)
0.4	40	[23, 24, 35]	$[5-9]$	$[4220 - 4300]$	$[2340 - 2360]$	$[0.001 - 0.002]$	42

3.4. Macroscopic Analysis

3.4. Macroscopic Analysis High induced cracking caused by the expansion of the corrosion products of the rebar was observed in the RC specimens when the *w/c* ratios increased for both *CCTh* values used in the third year of the study (Figure 8). Concrete cracking, as expected, was more noticeable for a *CCTh* of 20 mm. However, cracking began after two years in the RC specimens with *w/c* ratios of 0.5 and 0.6 for both values of *CCTh*. The appearance of cracks coincided with the *tcc* values obtained in the RC specimens with *w/c* ratios of 0.5 and 0.6 and for both *CCTh* values. The influence of the CX corrosivity category on the RC structures (with *w/c* of 0.5 and 0.6 and *CCTh* values of 20 mm and 40 mm) was confirmed.

Figure 8. Visual observation of RC specimens placed at COES during the three years of study. **Figure 8.** Visual observation of RC specimens placed at COES during the three years of study.

A higher penetration of chloride ions was caused by the induced cracking. The RC A higher penetration of chloride ions was caused by the induced cracking. The RC specimens with *w/c* ratios of 0.5 and 0.6 and with *CCTh* values of 20 mm and 40 mm did specimens with *w/c* ratios of 0.5 and 0.6 and with *CCTh* values of 20 mm and 40 mm did not ensure suitable primary protection as a physical barrier for the protection of the rebars. not ensure suitable primary protection as a physical barrier for the protection of the rebars. In this way, the RC structures did not maintain their initial design conditions over time in In this way, the RC structures did not maintain their initial design conditions over time in the entire Cuban coastal environment. the entire Cuban coastal environment.

The induced cracking confirms that the S_l for RC structures already built and exposed to extreme (CX) atmospheric corrosivity categories does not exceed five years. Therefore, to extreme (CX) atmospheric corrosivity categories does not exceed five years. Therefore, concrete that does not meet the defined durability requirements cannot ensure a long S_l for RC structures in the Cuban coastal environment.

4. Discussion 4. Discussion

4.1. Corrosivity of the Atmosphere 4.1. Corrosivity of the Atmosphere

According to research works carried out in all Cuban coastal environments during last century, an extreme (CX) atmospheric corrosivity category was obtained for carbon the last century, an extreme (CX) atmospheric corrosivity category was obtained for car-steel (*rcorr* = 443.72 µm y−¹ , for CX: 200 < *rcorr* < 700) in the coastal city of Havana. The annual average value of *Cl[−]DR* determined by the dry plate device was 919.3 mg m^{−2} d^{−1}. and COES was placed at a 10 m distance from the shoroline and was yery close to the COES The COES was placed at a 10 m distance from the shoreline and was very close to the COES
used in this study [56.57] According to research works carried out in all Cuban coastal environments during the used in this study [56,57].

Taking into account some of the research work carried out in the last 20 years and over the last century (ISOCORRAG, ICP/UNECE, and MICAT Databases) worldwide, an extreme (CX) atmospheric corrosivity category for carbon steel was obtained [58]. The annual average values of *Cl[−]DR* were determined using a wet candle device in different coastal cities. It should be noted that this annual value of the atmospheric corrosion rate for carbon steel (443.72 µm y⁻¹, for CX: 200 < r_{corr} < 700) is only lower than those determined in Iran and Saudi Arabia (Table 5). *rannoning than* than the set of the set

Table 5. Extreme atmospheric corrosivity category (CX: 200 < *rcorr* < 700) estimated for carbon steel for various coastal cities.

The monthly *Cl*−*DR* has been determined using a dry plate device in all Cuban coastal environments [56,57]. Regarding to the behavior of the monthly mean values of *Cl*−*DR*, the annual average was determined to be 851.04 mg m⁻² d⁻¹ (Figure 3). Despite the fact that a dry plate device was used, both the annual average values of *Cl*−*DR* determined in the COES placed in the coastal city of Havana (919.3 mg m⁻²d⁻¹ and 851.04 mg m⁻²d⁻¹) can be considered very high. The annual *Cl*−*DR* average was lower than that determined in the coastal environments of Mexico [59] (Table 5).

According to the relationship established in the ISO standard, i.e., Equation (1), the annual average values of *Cl[−]DR* determined by the wet candle device were 2206.32 mg m⁻²d⁻¹ (919.3 mg m⁻²d⁻¹) and 2042.49 mg m⁻²d⁻¹ (851.04 mg m⁻²d⁻¹). No reports have been found where the annual average values of *Cl*−*DR* have turned out to be greater than those determined in this work. Annual average values of *Cl*−*DR* closer to those determined in this study have been reported in the coastal cities of Sherman, Panama (1909.11 mg m $^{-2}$ d $^{-1}$), and Cabo Villano, Spain (1906.0 mg m⁻²d⁻¹) [60]. However, the wet candle devices remained for one month at the two COESs for a time of exposure of six months.

On the other hand, a research study about the atmospheric corrosion of carbon steel was carried out over the entire coastal environment of Havana for one year (April/2018– March/2019). Ten COESs were placed at a distance of 10–20 m from the shoreline. The annual average atmospheric corrosion rate determined for two carbon steel specimens placed at one of the COESs was 849.23 μ m y⁻¹ (200 < CX < 700). A value over the maximum limit established in the ISO 9223:2012 standard for the classification of an extreme (CX) atmospheric corrosivity category for carbon steel was obtained. A new corrosivity category of a higher-than-extreme atmosphere (>CX) or an increase in the range of CX corrosivity could be considered. Visual observation of the two specimens allowed for confirming the very high annual average corrosion rate determined. Intense deterioration due to atmospheric corrosion was observed (Figure 9). In addition, due to the occurrence of pitting corrosion, some of the carbon steel specimens were pierced. These results indicated that the coastal city of Havana is among the most corrosive sites worldwide, mainly in the coastal environments at a short distance from the shoreline. Many RC structures have presented a short *S^l* due to atmospheric corrosion of rebar. To ensure a long *S^l* in RC structures exposed to a CX corrosivity category, the definition of durability requirements becomes paramount.

Figure 9. Visual observation of two specimens of carbon steel. Intense deterioration due to atmospheric corrosion is observed. The metal has been perforated by atmospheric corrosion. spheric corrosion is observed. The metal has been perforated by atmospheric corrosion. **Figure 9.** Visual observation of two specimens of carbon steel. Intense deterioration due to atmo-

4.2. Durability Requirements

The model used (Equation (4)) to predict the S_l in the RC structures based on their behavior is established in the ISO 9224:2012 standard [44], which is designed for the atmospheric corrosion of metals. In our case, the *Ic* in addition to the requirement for durability is used as an indicator of the atmospheric corrosion of rebar. According to the theoretical *Ic* values predicted from the model, atmospheric corrosivity categories were predicted for times of exposure higher than one year for metallic materials that are most often used in the construction industry (carbon steel and galvanized steel). High (C4), very high (C5), and extreme (CX) atmospheric corrosivity categories were considered for both metallic materials as durability requirements for metallic structures in Cuban coastal environments and in Ecuador [61,62]. The annual atmospheric corrosion rate of carbon steel and galvanized steel specimens was obtained during the first year of the study. The COESs were placed at a short distance from the shoreline.

Considering the experimental values obtained for the *Ic* as an indicator of the atmospheric corrosion of the reinforcement steel in the RC specimens with $w/c = 0.4$ and with *CCTh* values of 20 mm (r_{corr} = 0.065 μ A cm⁻²) and 40 mm (r_{corr} = 0.002 μ A cm⁻²) that were exposed during the first year, a prediction was made for the RC structures using the same model, i.e., Equation (4) (Figure 6). In this way, the t_{cc} was obtained for a *CCTh* of 20 mm, and the t_i was obtained for a CCTh of 40 mm. According to the specifications established in the ISO 9224-2012 standard, the value "n" in the model was established to be equal to the average time exponents reported in the regression analysis of the carbon steel specimens exposed for a long time in the ISO CORRAG project [58].

Worldwide, few studies have been carried out to establish durability requirements that guarantee a long S_l in RC structures based on exposure to the atmosphere in real coastal conditions. A comparative study of different electrochemical techniques used to measure the corrosion rate of reinforcement steel in RC specimens placed in marine environments is reported [63]. The different electrochemical techniques were as follows: galvanostatic pulse, electrical resistivity, and half-cell potential. Exposure to the atmosphere in coastal environments was not considered. Corrosion of the reinforcement steel under splash conditions was higher than under tide conditions. The methodology used was not very accurate for the determination of the *tⁱ* . The evaluation was carried out only for one *CCTh* value. A clear relationship between the *tⁱ* determined in the tidal environment and in the splash environment was not observed. However, the *tⁱ* decreased when the *w/c* ratio increased from 0.35 to 0.5. The t_{cc} was not determined, and the S_l of the RC structures exposed to the marine environment was not presented. Durability requirements directly related to the parameters that characterize the concrete quality were not considered.

On the other hand, an international research project involving 11 Ibero-American countries was conducted to study the effect of coastal environments on RC structures over one year of exposure. RC specimens were prepared using locally available materials in each country [64]. A description of the typical atmospheric environments of the eleven COESs located in coastal cities was given. Two types of RC specimens (*w/c* = 0.45 and 0.65, $CCTh = 10$, 20, and 30 mm) were exposed at each site, with the atmospheric corrosivity categories estimated as high (C4) and very high (C5). The *fck*, elastic modulus, permeability to free chloride ion concentrations, and *ε^e* were measured for the concrete quality characterization. The time taken to initiate depassivation, considered as *tⁱ* , was determined only from the behavior of the electrochemical corrosion rate of the reinforcement steels in the RC specimens with a *w/c* ratio of 0.65 at the time of exposure in two countries (Portugal and Venezuela). However, the *tcc* was not determined. In this way, the *S^l* of RC structures exposed to coastal environments was not obtained.

To predict the S_l for RC structures built in coastal environments worldwide, nine models were obtained via multiple regression fitting [65]. The *Ic* was measured using different electrochemical techniques under laboratory and on-site conditions. The *Ic* determined in the reinforcement steel was highly sensitive to three parameters related to concrete quality: the corrosion time, concrete resistivity, and concrete chloride content. To predict the S_l in RC structures, the *tⁱ* , and *tcc* determined from the behavior and the prediction of the *Ic* could be included in the corrosion time, depending on the concrete quality.

According to the behavior observed during the third year of this study (Figure 7), the data were fitted to a multiple regression model (Table 6). The *w/c* ratio, *fck*, and *ε^e* were the most significant parameters related to the *Ic* measured in the reinforcement steel for both values of *CCTh* during the three years of this study.

Multiple regression fitting has been widely used to define factors that are more statistically significant in the atmospheric corrosion of metallic structures in the tropical coastal climate of Cuba and worldwide; *Cl*−*DR*, a high *RH*, and *T* are the factors that most often influence atmospheric corrosion [66].

Table 6. Multiple regression fitting for the data collected during the three years of this study (Equation (3)).

Regarding the comparison from the multiple range test method, following the same behavior shown during the third year of this study (Figure 7), two homogeneous groups were identified according to the alignment of the X column (Table 7). Thus, there were no significant differences between the groups sharing the same column. The *w/c* ratio and *ε^e* for both concrete cover thicknesses fell into the first homogeneous group (X column). The f_{ck} and ε_e fell into the second homogeneous group (Table 7).

	$CCTh$ of 20 mm					CCTh of 40 mm					
Durability Requirements			Homogeneity Group					Homogeneity Group			
	n	Average		X Column			Average			X Column	
<i>Ic</i> (μ A-cm ⁻²)	36	1.260	X				0.486	X			
w/c ratio	36	0.5	$\boldsymbol{\chi}$				0.5	X			
ε_e (%)	36	13.7	$\boldsymbol{\chi}$	X			13.7	X	X		
f_{ck} (MPa)	36	30		X			30		X		
D (kg m ⁻³)	36	2333			X		2333			л	
UPV (m s ⁻¹)	36	4097				X	4097				л

Table 7. Comparison from the multiple range test method.

Table 8 provides the estimated differences between each pair of means. The asterisks next to the 11 pairs indicate that these pairs showed statistically significant differences at the 95.0% confidence level for a limit of ±24.863 in both values of *CCTh*. No report was found where the comparison from the multiple range test method has been applied to determining which durability requirements are the most sensitive to the *Ic* in reinforcement steel.

Table 8. Estimated differences between each pair of means.

Due to the fact that the sample sizes were equal $(n = 36)$ and the treatments presented an ordinal relationship (Figure 7), the multiple range test method was performed. According to the two statistical methods applied, the w/c ratio, f_{ck} , and ε_e were the parameters that were the most sensitive to the *Ic* as an indicator of atmospheric corrosion in the reinforcement steel for both values of *CCTh*.

The nine models chosen did not consider these last two parameters [65]. The *fck* was only considered in the predictive models proposed by Lu and Vu in their research works about the modelling of corrosion propagation for the *S^l* prediction of chloride-contaminated RC structures and in measuring the structural reliability of concrete bridges, including improved chloride-induced corrosion models, respectively [67,68]. Therefore, to ensure a long *S^l* in RC structures in the Cuban coastal environment exposed to extreme (CX) and above-extreme (>CX) atmospheric corrosivity categories, the most important durability requirements are a w/c ratio of 0.4, f_{ck} , and ε_e , with a CCTh of 40 mm. In this way, a free chloride ion concentration of less than 0.05 (% m.c.) could be achieved. The same free chloride ion concentration (% in m.c.) was considered in a case study of the deterioration of RC docks in a marine environment in Portugal [69].

4.3. Microstructure of Rust Layers 4.3. Microstructure of Rust Layers

Different crystalline phases such as lepidocrocite ($γ$ – $FeOOH$), goethite ($α$ – $FeOOH$), magnetite (Fe_3O_4) , and akaganeite $(\beta - FeOOH)$ were identified in the rust layers of the reinforcement steel embedded in the RC specimen with a *w/c* ratio of 0.6 and a CCTh value of 20 mm placed at the COES until the third year of exposure (Figure 10).

Figure 10. Different crystalline phases identified in the rust layers of reinforcement steel embedded **Figure 10.** Different crystalline phases identified in the rust layers of reinforcement steel embedded in the RC specimen with a *w/c* ratio of 0.6 and a *CCTh* value of 20 mm placed at the COES until the in the RC specimen with a *w/c* ratio of 0.6 and a *CCTh* value of 20 mm placed at the COES until the third year of exposure. third year of exposure.

Similar crystalline phases have been identified in carbon steel specimens freely exposed to high (C4) and very high (C5) atmospheric corrosivity categories in the entire Cuban coastal environment and worldwide [70,71]. Induced cracking in the *CCTh* due to the atmospheric corrosion of the reinforcement steel tends to further accelerate this process. atmospheric corrosion of the reinforcement steel tends to further accelerate this process.

The morphology of the crystalline phases identified in the oxide layer formed on the The morphology of the crystalline phases identified in the oxide layer formed on the reinforcement steel was observed using the scanning electron microscopy microstructural analysis technique (Figure 11a–d). analysis technique (Figure 11a–d).

Lepidocrocite in a foam form, flower-shaped goethite, and ring-shaped magnetite Lepidocrocite in a foam form, flower-shaped goethite, and ring-shaped magnetite showed a similar morphology compared to the oxide layers identified in the carbon steel $\frac{1}{2}$ specimens freely exposed to high (C4) and very high (C5) atmospheric corrosivity categories
 $\frac{11}{6}$ all Cuban coastal environments and worldwide [70,71]. The EDS analysis carried out on the
annualization coastal environments and worldwide [70,71]. The EDS analysis carried out on the on the morphologies associated with lepidocrocite, goethite, and magnetite (Figure 11b) high percentages of iron (>90 wt.%) and low concentrations of oxygen (<5 wt.%) as well as the absence of the chlorine element (Cl). Therefore, these results corroborate the fact that the absence of the chlorine element (Cl). Therefore, these results corroborate the fact that the these morphologies identified were associated with different iron phases and that none of fact that these morphologies identified were associated with different iron phases and that these phases were associated with the akaganeite phase. morphologies associated with lepidocrocite, goethite, and magnetite (Figure 11b) revealed

nese phases were associated with the anaganeite phase.
Due to the high concentration of chloride ion sand and the lack of oxygen, akaganeite Due to the high concentration of chloride ion sand and the lack of oxygen, akaganeite is formed in reinforcement steel/oxide/concrete interfaces, as happened in the carbon is formed in reinforcement steel/oxide/concrete interfaces, as happened in the carbon steel steel specimens freely exposed to high (C4) and very high (C5) atmospheric corrosivity exposed to engot (C4) and very high (C4) and very high (C4) atmospheric corresponding categories in the entire Cuban coastal environment and worldwide. In this sense, the akaganeite formed in the reinforcement steel showed Cl percentages of between 4.3 and 6.4 wt.% (Figure 11e,f). The Cl content varied according to the formation conditions, and

percentages of up to 47 wt.% have been found in carbon steel exposed to tropical marine environments [72]; however, the theoretical maximal concentration of chloride ions that this phase can contain in its structure is approximately 7 wt.%, meaning that excess chloride ions are often associated with the absorption of these ions onto the akaganeite surface, while concentrations lower than 7 wt.% are associated with a framework containing tunnels partially filled with chloride anions [73]. tially filled with chloride anions [73].

formed in the reinforcement steel showed Cl percentages of between 4.3 and 6.4 wt.%

Therefore, the high Cl percentages (up to 6.4 wt% in Figure 11f) found in akaganeite Therefore, the high Cl percentages (up to 6.4 wt% in Figure 11f) found in akaganeite suggest a high occupancy of Cl sites within the tunnels in the framework of this phase or suggest a high occupancy of Cl sites within the tunnels in the framework of this phase or the presence of some unoccupied vacancy sites or the fact that some vacant Cl sites are the presence of some unoccupied vacancy sites or the fact that some vacant Cl sites are occupied by impurities or oxygenated species such as hydroxyl groups or water [73–75]. occupied by impurities or oxygenated species such as hydroxyl groups or water [73–75]. These results are consistent with previous works, where Cl percentages between 2 and mol% have been found for this phase [76]. 7 mol% have been found for this phase [76].

On the other hand, akaganeite formed in freely exposed carbon steel specimens On the other hand, akaganeite formed in freely exposed carbon steel specimens (−) typically shows cotton ball, cigar, and rosette morphologies [77,78]; how-(*β* − *FeOOH*) typically shows cotton ball, cigar, and rosette morphologies [77,78]; however, Morcillo and collaborators reported the formation of akaganeite constituted of aggregates of elongated cylindrical crystals with lengths and widths of between 100-200 nm and and 60–80 nm, respectively [79]; they suggested that this cylindrical-type morphology 60–80 nm, respectively [79]; they suggested that this cylindrical-type morphology could be a typical morphology of akaganeite formed under marine conditions because it differs from the spindle-type morphology that akaganeite shows when synthesized under laboratory conditions. Interestingly, our results obtained by SEM revealed the formation of highly agglomerated akaganeite crystals with an elongated morphology and right-angled edges (Figure $11a$, c ,d). The growth of akaganeite on carbon steel with a cylindrical or rod-type morphology has also been documented by Juan C. Guerra and collaborators $[62]$; however, morphology has also been documented by Juan C. Guerra and collaborators [62]; however,
the morphology similar to bars with right-angled edges found in this work has not been reported previously. The change in morphology could be associated with different factors such as pH and the chloride ion concentration.

In this sense, a more alkaline environment should be present in specimens protected by concrete compared to the pH typically found in freely exposed specimens [80]. Also, surface-absorbed chloride ions on freely exposed carbon steel specimens can be solubilized by rainfall events [80]; consequently, it is expected that the effect of rainfall on the washing of chloride ions will be lower due to the barrier effect of the concrete. Therefore, a higher pH and a more constant chloride concentration should be expected at the steel/concrete interface compared with freely exposed samples.

On the other hand, Emily and collaborators found that akaganeite structures can be modified by the crystallization microenvironment [81]; they reported that the akaganeite length, the presence of twinned forms, and the number of arms are affected by factors such as the content of chloride ions and the pH. Therefore, the above works suggest that the concentration and stability of chloride ions, as well as the pH in the crystallization microenvironment, could contribute to the formation of this type of morphology, i.e., elongated with right-angled edges. On the other hand, concrete is a product rich in silicon (Si). One of the main components of cement is tricalcium silicate, which produces a gel when it is hydrated. In the akaganeite structure, the release of chloride ions and the absorption of Si-based compounds are favored at an alkaline pH [81]. Also, the formation of akaganeite as a three-dimensional assembly was reported when this structure came into contact with a Si-based hydrogel, in contrast to the discrete rods formed in solution [82]. Therefore, the different phases of Si that can contain concrete could modify the morphology of akaganeite; however, more detailed studies should be carried out to corroborate this hypothesis. On the other hand, due to the highly agglomerated morphology of this crystalline phase, it is possible that it is one of the reasons for the corrosion product expansion of concrete, as happened in the freely exposed carbon steel specimens, which subsequently led to cracking, with *CCTh* values of 20 and 40 mm in the RC with *w/c* ratios = 0.5 and 0.6.

5. Conclusions

A comprehensive research study was carried out to obtain durability requirements for RC structures in hostile tropical coastal environments. The following conclusions were drawn:

- According to the annual average chloride deposition rate and atmospheric corrosion rate for the metallic materials most commonly used in the construction industry (mainly for carbon steel), the coastal city of Havana can be considered among the coastal cities with the highest atmospheric corrosivity categories worldwide. An annual average atmospheric corrosion rate over the maximum limit established in the ISO 9223:2012 for the extreme (CX) atmospheric corrosivity category for carbon steel was determined. An increase in the range established for the extreme (CX) atmospheric corrosivity category in a coastal environment at a short distance from the shoreline could be considered. This result was confirmed from the visual observation of the carbon steel specimens.
- According to the behavior and prediction of the annual average *Ic*, considered as an indicator of the atmospheric corrosion rate of reinforcement steel in RC versus the time of exposure, the S_l from the sum of the time-to-corrosion-initiation (t_i) and the time-to-corrosion-induced concrete cracking (*tcc*) was determined and predicted as a function of the durability requirements. The model established in the ISO 9224-2012 to predict the corrosivity category of the atmosphere for the metallic materials that are most often used in the construction industry (carbon steel, galvanized steel) for times of exposure over one year can be used to predict the S_l in RC structures in the coastal city of Havana and in all Cuban coastal environments.
- The two statistical methods used allowed for the demonstration of the most important durability requirements to ensure a long *S^l* of more than 70 years in RC structures in the Cuban coastal environment exposed to extreme (CX) and above-extreme (>CX) corrosivity categories. This requirement is as follows: *w/c* ratio of 0.4, *fck*, and *ε^e* , with a *CCTh* of 40 mm.
- It was confirmed that the occurrence of cracking in the RC specimens with *w/c* ratios of 0.5 and 0.6 for both *CCTh* values tested is an indicator of the end of the *S^l* , in less than five years, for RC structures exposed to very high (C5) and extreme (CX) atmospheric corrosivity categories in the coastal city of Havana. The results were confirmed by the visual observation of the RC specimens placed at the COES during the three years of this study.
- A significant difference between the chemical composition of the oxide layers developed in the freely exposed carbon steel specimens and the reinforcement steel embedded in the RC was not observed, but the presence of akaganeite with different morphologies was reported for the reinforced steel. It seems that crystalline-phase akaganeite could be one of the reasons for the corrosion product expansion of the reinforcement steel, which subsequently led to cracking in the specimens with a *CCTh* of 20 and 40 mm in the RC prepared with *w/c* ratios of 0.5 and 0.6.
- Since only one type of cement was considered in this work, the proposed protocol could be applied in future research to study the durability of RC containing supplementary cementitious materials or made using green formulations, recycled concrete aggregates, and/or sand to provide a large variety of solutions to ensure the sustainability and durability of constructions in these tropical environments.

Author Contributions: Conceptualization, A.C.V., F.C.P., and I.P.P.; methodology, A.C.V., F.C.P., and R.M.Á.; software, I.P.P.; validation, A.C.V.; formal analysis, F.C.P. and E.B.-A.; investigation, A.C.V. and R.M.Á.; resources, A.C.V., F.C.P., and R.M.Á.; data curation, A.C.V.; writing—original draft preparation, A.C.V., F.C.P., I.P.P., and E.B.-A.; writing—review and editing, E.B.-A.; visualization, E.B.-A.; supervision, F.C.P. and E.B.-A.; project administration, A.C.V.; funding acquisition, A.C.V. All authors have read and agreed to the published version of the manuscript.

Funding: This research was supported by the Mortars and Concretes Branch Program of the Ministry of Construction of Cuba.

Data Availability Statement: The data provided in this study did not lead to the development of patents. Therefore, they are not subject to any kind of privacy. They can be used publicly.

Acknowledgments: The authors wish to thank the Ministry of Construction of Cuba for its financial support and for guaranteeing the necessary materials and resources during the execution of this research. The author Ildefonso Pech Pech thanks the LANCIC Project LN315853 and LN 314846 for SEM/EDS analysis.

Conflicts of Interest: The authors declare no conflicts of interest.

References

- 1. Abosrra, L.; Ashour, A.F.; Youseffi, M. Corrosion of steel reinforcement in concrete of different compressive strengths. *Constr. Build. Mater.* **2011**, *25*, 3915–3925. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2011.04.023)
- 2. Zhao, Y.; Yu, J.; Wu, Y.; Jin, W. Critical thickness of rust layer at inner and out surface cracking of concrete cover in reinforced concrete structures. *Corr. Sci.* **2012**, *59*, 316–323. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2012.03.018)
- 3. Shi, X.; Xie, N.; Fortune, K.; Gong, J. Durability of steel reinforced concrete in chloride environments: An overview. *Constr. Build. Mater.* **2012**, *37*, 125–138. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2011.12.038)
- 4. Pradhan, B.; Bhattacharjee, B. Rebar corrosion in chloride environment. *Constr. Build. Mater.* **2011**, *25*, 2565–2575. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2010.11.099)
- 5. Ann, K.Y.; Ahn, J.H.; Ryou, J.S. The importance of chloride content at the concrete surface in assessing the time to corrosion of steel in concrete structures. *Constr. Build. Mater.* **2009**, *23*, 239–245. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2007.12.014)
- 6. Narasimhan, H.; Chew, M.Y. Integration of durability with structural design: An optimal life cycle cost based design procedure for reinforced concrete structures. *Constr. Build. Mater.* **2009**, *23*, 918–929. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2008.04.016)
- 7. Liu, P.; Yu, Z.; Lu, Z.; Chen, Y.; Liu, X. Predictive convection environment depth of chloride in concrete under chloride environment. *Cem. Concr. Compos.* **2016**, *72*, 257–267. [\[CrossRef\]](https://doi.org/10.1016/j.cemconcomp.2016.06.011)
- 8. Cheewaket, T.; Jaturapitakkul, C.; Chalee, W. Initial corrosion presented by chloride threshold penetration of concrete up to 10 year-results under marine site. *Constr. Build. Mater.* **2012**, *37*, 693–698. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2012.07.061)
- 9. Castañeda, A.; Rodríguez, M. Las pérdidas económicas causadas por el fenómeno de la corrosión atmosférica del acero de refuerzo embebido en el hormigón armado. Resultadospreliminares. *Rev. CENIC. Chem. Sci.* **2014**, *45*, 52–59.
- 10. Rincon, L.F.; Moscoso, Y.M.; Hamami, A.E.A.; Matos, J.C.; Bastidas-Arteaga, E. Degradation Models and Maintenance Strategies for Reinforced Concrete Structures in Coastal Environments under Climate Change: A Review. *Buildings* **2024**, *14*, 562. [\[CrossRef\]](https://doi.org/10.3390/buildings14030562)
- 11. Muralidharan, S.; Vedalakshmi, R.; Saraswathi, V.; Joseph, J.; Palaniswamy, N. Studies on the aspects of chloride ion determination in different types of concrete under macro-cell corrosion conditions. *Build. Environ.* **2005**, *40*, 1275–1281. [\[CrossRef\]](https://doi.org/10.1016/j.buildenv.2004.10.005)
- 12. Güneyisi, E.; Özturan, T.; Gesoglu, M. Effect of initial curing on chloride ingress and corrosion resistance characteristics of concretes made with plain and blended cements. *Build. Environ.* **2007**, *42*, 2676–2685. [\[CrossRef\]](https://doi.org/10.1016/j.buildenv.2006.07.008)
- 13. Yiğiter, H.; Yazıcı, H.; Aydın, S. Effects of cement type, water/cement ratio and cement content on sea water resistance of concrete. *Build. Environ.* **2007**, *42*, 1770–1776. [\[CrossRef\]](https://doi.org/10.1016/j.buildenv.2006.01.008)
- 14. Mohamed, N.; Boulfiza, M.; Evitts, R. Corrosion of Carbon Steel and Corrosion-Resistant Rebars in Concrete Structures Under Chloride Ion Attack. *J. Mater. Eng. Perform.* **2012**, *22*, 787–795. [\[CrossRef\]](https://doi.org/10.1007/s11665-012-0314-0)
- 15. Castañeda, A.; Corvo, F.; O'Reilly, V. Comparación entre el pronóstico de corrosiónbasado en la medición de potenciales y la determinación de la velocidad de corrosión de la barra de refuerzomediantetécnicaselectroquímicas. *Mater. Construcción* **2003**, *53*, 155–164.
- 16. Castañeda, A.; Corvo, F.; González, J. Estudiocomparativo de la corrosióndelacero de refuerzo en el hormigónarmadoa partir de técnicaselectroquímicas y convencionales. *Rev. CENIC. Chem. Sci.* **2010**, *41*, 1–9.
- 17. Behera, P.K.; Misra, S.; Mondal, K. Corrosion Behavior of Strained Rebar in Simulated Concrete Pore Solution. *J. Mater. Eng. Perform.* **2020**, *33*, 1939–1954. [\[CrossRef\]](https://doi.org/10.1007/s11665-020-04708-x)
- 18. Poursaee, A. Corrosion of Steel Bars in Saturated Ca(OH)² and Concrete Pore Solution. *Concr. Res. Lett.* **2010**, *3*, 90–97.
- 19. Mundra, S.; Criado, M.; Bernal, S.A.; Provis, J.L. Chloride-Induced Corrosion of Steel Rebars in Simulated Pore Solutions of Alkali Activated Concretes. *Cem. Concr. Res.* **2017**, *100*, 385–397. [\[CrossRef\]](https://doi.org/10.1016/j.cemconres.2017.08.006)
- 20. Hussain, R.R.; Singh, J.K.; Alhozaimy, A.; Al-Negheimish, A.; Bhattacharya, C.; Pathania, R.S.; Singh, D.D.N. Effect of Reinforcing Bar Microstructure on Passive Film Exposed to Simulated Concrete Pore Solution. *ACI Mater. J.* **2018**, *115*, 181. [\[CrossRef\]](https://doi.org/10.14359/51701237)
- 21. Williamson, J.; Isgor, O.B. The Effect of Simulated Concrete Pore Solution Composition and Chlorides on the Electronic Properties of Passive Films on Carbon Steel Rebar. *Corros. Sci.* **2016**, *106*, 82–95. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2016.01.027)
- 22. De Rincón, O.T.; Castro, P.; Moreno, E.I.; Torres-Acosta, A.A.; De Bravo, O.M.; Arrieta, I.; Martínez-Madrid, M. Chlorideprofiles in two marine structures—Meaning and somepredictions. *Build. Environ.* **2004**, *39*, 1065–1070. [\[CrossRef\]](https://doi.org/10.1016/j.buildenv.2004.01.036)
- 23. Pang, L.; Li, Q. Service life prediction of RC structures in marine environment using long term chloride ingress data: Comparison between exposure trials and real structure surveys. *Constr. Build. Mater.* **2016**, *113*, 979–987. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2016.03.156)
- 24. Cui, Z.; Alipour, A. Concrete cover cracking and service life prediction of reinforced concrete structures in corrosive environments. *Constr. Build. Mater.* **2018**, *159*, 652–671. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2017.03.224)
- 25. Vieira, D.R.; Moreira, A.L.R.; Calmon, J.L.; Dominicini, W.K. Service life modeling of a bridge in a tropical marine environment for durable design. *Constr. Build. Mater.* **2018**, *163*, 315–325. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2017.12.080)
- 26. Castañeda, A.; Corvo, F.; Marrero, R.; Fernández, A.; Del Angel-Meraz, E. The service life of reinforced concrete structures in an extremely aggressive coastal city. Influence of concrete quality. *Mater. Struct.* **2023**, *56*, 12. [\[CrossRef\]](https://doi.org/10.1617/s11527-023-02100-4)
- 27. De Vera, G.; Anton, C.; Lopez, M.P.; Climent, M.A. Depassivation time estimation in reinforced concrete structures exposed to chloride ingress: A probabilistic approach. *Cem. Concr. Compos.* **2017**, *79*, 21–33. [\[CrossRef\]](https://doi.org/10.1016/j.cemconcomp.2016.12.012)
- 28. Samindi, S.M.; Samarakoon, M.; Jan Sælensminde, K. Condition assessment of reinforced concrete structures subject to chloride ingress: A case study of updating the model prediction considering inspection data. *Cem. Concr. Compos.* **2015**, *60*, 92–98.
- 29. Angst, U.; Elsener, B.; Larsen, C.K.; Vennesland, F. Critical chloride content in reinforced concrete—A review. *Cem. Concr. Res.* **2009**, *39*, 1122–1138. [\[CrossRef\]](https://doi.org/10.1016/j.cemconres.2009.08.006)
- 30. Zacchei, E.; Bastidas-Arteaga, E. Multifactorial Chloride Ingress Model for Reinforced Concrete Structures Subjected to Unsaturated Conditions. Collection Advanced Concrete Structures in Civil Engineering. *Buildings* **2022**, *12*, 107. [\[CrossRef\]](https://doi.org/10.3390/buildings12020107)
- 31. Anwar Hossain, K.M.; Easa, S.M.; Lachemi, M. Evaluation of the effect of marine salts on urban built infrastructure. *Build. Environ.* **2009**, *44*, 713–722. [\[CrossRef\]](https://doi.org/10.1016/j.buildenv.2008.06.004)
- 32. Trocónis, O.; Duracon Collaboration. Durability of concrete structures: DURACON, an Iberoamerican project. Preliminary results. *Build. Environ.* **2006**, *41*, 952–962. [\[CrossRef\]](https://doi.org/10.1016/j.buildenv.2005.04.005)
- 33. *ISO 12944-9:2018*; Paints and Varnishes-Corrosion and Protection of Steel Structures by Protective Paint Systems. Part 9. Protective Paint Systems and Laboratory Performance Test Methods for Offshore and Related Structures. Beuth: Berlin, Germany, 2018.
- 34. *ISO 9223:2012*; Corrosion of Metals and Alloys—Corrosivity of Atmospheres—Classification, Determination and Estimation. ISO: Geneva, Switzerland, 2012.
- 35. Jiang, J.H.; Yuan, Y.S. Development and prediction strategy of steel corrosion rate in concrete under natural climate. *Constr. Build. Mater.* **2013**, *44*, 287–292. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2013.03.033)
- 36. Castañeda, A.; Rivero, C.; Corvo, F. Evaluación de sistemas de protección contra la corrosión en la rehabilitación de estructurasconstruidas en sitios de elevadaagresividadcorrosiva en Cuba. *Rev. Construcción* **2012**, *11*, 49–61. [\[CrossRef\]](https://doi.org/10.4067/S0718-915X2012000300005)
- 37. *ISO 9225: 2012*; Corrosion of Metals and Alloys—Corrosivity of Atmosphere—Measurement of Environmental Parameters Affecting Corrosivity of Atmosphere. International Organization for Standardization: Geneva, Switzerland, 2012.
- 38. *Cuban Standard. 120:2014*; Reinforced Concrete. Specifications. Cuban National Bureau of Standards: La Habana, Cuba, 2014.
- 39. *ISO 8407: 2009*; Corrosion of Metals and Alloys—Removal of Corrosion Products from Corrosion Tests Specimens. International Organization for Standardization: Geneva, Switzerland, 2009.
- 40. Troconis de Rincon, O. *Manual for Inspecting, Evaluating and Diagnosing Corrosion in Reinforced Concrete Structures*; NACE: Houston, TX, USA, 2000.
- 41. *Cuban Standard 724:2009*; Test on Concrete. Resistance of Hardened Concrete. Cuban National Bureau of Standards: La Habana, Cuba, 2009.
- 42. *Cuban Standard 231:2012*; Test on Concrete. Determination, Interpretation and Application of Ultrasonic Pulse Velocity in Concrete. Cuban National Bureau of Standards: La Habana, Cuba, 2012.
- 43. Fagerlund, G. On the Capillarity of Concrete. *Nord. Concr. Res.* **1986**, *1*.
- 44. *ISO 9224:2012*; Corrosion of Metals and Alloys—Corrosivity of Atmospheres—Guiding Values for the Corrosivity Categories. ISO: Geneva, Switzerland, 2012.
- 45. Castañeda, A.; Corvo, F.; Fernández, D.; Valdés, C. Outdoor-indoor atmospheric corrosion in a coastal wind farm located in a tropical island. *Eng. J.* **2017**, *21*, 43–62. [\[CrossRef\]](https://doi.org/10.4186/ej.2017.21.2.43)
- 46. Corvo, F.; Pérez, T.; Dzib, L.; Martín, Y.; Castañeda, A.; González, E.; Pérez, J. Outdoor-Indoor corrosion of metals in tropical coastal atmospheres. *Corros. Sci.* **2008**, *50*, 220–230. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2007.06.011)
- 47. Castañeda, A.; Corvo, F.; Howland, J.J.; Marrero, R. Penetration of marine aerosol in a tropical coastal City: Havana. *Atmósfera* **2018**, *31*, 87–104. [\[CrossRef\]](https://doi.org/10.20937/ATM.2018.31.01.06)
- 48. Meira, G.R.; Andrade, C.; Alonso, C.; Padaratz, I.J.; Borba, J.C. Modelling sea-salt transport and deposition in marine atmosphere environment—A tool for corrosion studies. *Corros. Sci.* **2008**, *50*, 2724–2731. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2008.06.028)
- 49. Meira, G.R.; Andrade, C.; Alonso, C.; Borba, J.C., Jr.; Padilha, M., Jr. Durability of concrete structure in marine atmosphere environment. The use of chloride deposition rate on wet candle as an environment indicator. *Cem. Concr. Compos.* **2010**, *32*, 427–435. [\[CrossRef\]](https://doi.org/10.1016/j.cemconcomp.2010.03.002)
- 50. Molina, L.; Alonso, F.J.; Alonso, M.C.; Sánchez, M.; Jarabo, R. Corrosion protection of galvanized rebars in ternary binder concrete exposed to chloride penetration. *Constr. Build. Mater.* **2017**, *156*, 468–475. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2017.09.002)
- 51. Zheng, H.; Dai, J.G.; Li, W.; Poon, C.S. Influence of chloride ion on depassivation of passive film on galvanized steel bars in concrete pore solution. *Constr. Build. Mater.* **2018**, *156*, 572–580. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2018.01.174)
- 52. Wang, Y.; Kong, G. Corrosion inhibition of galvanized steel by MnO₄⁻ ion as a soluble inhibitor in simulated fresh concrete environment. *Constr. Build. Mater.* **2020**, *257*, 119532. [\[CrossRef\]](https://doi.org/10.1016/j.conbuildmat.2020.119532)
- 53. Castañeda, A.; Howland, J.J.; Corvo, F.; Marrero, R. Concrete quality assessment before building structures submitting to environmental exposure conditions. *Rev. Constr.* **2017**, *16*, 374–387.
- 54. Castañeda, A.; Howland, J.J.; Corvo, F.; Marrero, R. Concrete quality assessment before submitting to environmental exposure conditions in Cuba. *Rev. Acta Microscópica* **2017**, *26*. [\[CrossRef\]](https://doi.org/10.7764/RDLC.16.3.374)
- 55. Carvajal, M.; Vera, R.; Corvo, F.; Castañeda, A. Diagnosis and rehabilitation of real reinforced concrete structures in coastal areas. *Corros. Eng. Sci. Technol.* **2011**, *47*, 70–77. [\[CrossRef\]](https://doi.org/10.1179/1743278211Y.0000000018)
- 56. Corvo, F.; Pérez, T.; Martin, Y.; Reyes, J.; Dzib, L.R.; González, J.A.; Castañeda, A. Corrosion research frontiers. Atmospheric corrosion in tropical climate. On the concept of Wetness and its interaction with contaminants deposition. In *Electroanalytical Chemistry: New Research*; Nova Science Publishers: New York, NY, USA, 2008; Chapter 2.
- 57. Corvo, F.; Pérez, T.; Martin, Y.; Reyes, J.; Dzib, L.R.; González-Sánchez, J.; Castañeda, A. *Atmosphericcorrosion in tropical humidclimates. Environmental Degradation of Infrastructure and Cultural Heritage in Coastal Tropical Climate*; Transworld Research Network: Kerala, India, 2009.
- 58. Chico, B.; De la Fuente, D.; Díaz, I.; Simancas, J.; Morcillo, M. Annual Atmospheric Corrosion of Carbon Steel Worldwide. An Integration of ISOCORRAG, ICP/UNECE and MICAT Databases. *Materials* **2017**, *10*, 601. [\[CrossRef\]](https://doi.org/10.3390/ma10060601) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/28772966)
- 59. Mariaca, L.; Menchaca, C.; Sarmiento, E.; Sarmiento, O.; Ramírez, J.L.; Uruchurtu, J. Atmospheric Corrsion Dose/Response Functions from Statistical Data Analysis for Different Sites of Mexico. *Innov. Corros. Mater. Sci.* **2014**, *4*, 11–20.
- 60. Alcántara, J.; Chico, B.; Diaz, I.; De la Fuente, D.; Morcillo, M. Airborne chloride deposit and its effect on marine atmospheric corrosion of mild steel. *Corros. Sci.* **2015**, *97*, 74–88. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2015.04.015)
- 61. Castañeda, A.; Valdés, C.; Corvo, F. Atmospheric corrosion study in a harbor located in a tropical island. *Mater. Corros.* **2018**, *69*, 1462–1477. [\[CrossRef\]](https://doi.org/10.1002/maco.201810161)
- 62. Guerra, J.C.; Castañeda, A.; Corvo, F.; Howland, J.J.; Rodríguez, J. Atmospheric corrosion of low carbon steel in a coastal environment of Ecuador: Anomalous behavior of chloride deposition versus distance from the sea. *Mater. Corros.* **2018**, *69*, 444–460.
- 63. Valipour, M.; Shekarchi, M.; Ghods, P. Comparative studies of experimental and numerical techniques in measurement of corrosion rate and time-to-corrosion-initiation of rebar in concrete in marine environments. *Cem. Concr. Compos.* **2014**, *48*, 98–107. [\[CrossRef\]](https://doi.org/10.1016/j.cemconcomp.2013.11.001)
- 64. Trocónis, O.; Duracon Collaboration. Effect of the marine environment on reinforced concrete durability in Iberoamerican countries: DURACON project/CYTED. *Corros. Sci.* **2007**, *49*, 2832–2843. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2007.02.009)
- 65. Siamphukdee, K.; Collins, F.; Zou, R. Sensitivity Analysis of Corrosion Rate Prediction Models Utilized for Reinforced Concrete Affected by Chloride. *J. Mater. Eng. Perform.* **2012**, *22*, 1530–1540. [\[CrossRef\]](https://doi.org/10.1007/s11665-012-0447-1)
- 66. Corvo, F.; Pérez, T.; Martin, Y.; Reyes, J.; Dzib, L.R.; González-Sánchez, J.; Castañeda, A. Time of wetness in tropical climate: Considerations on the estimation of TOW according to ISO 9223 standard. *Corros. Sci.* **2008**, *50*, 206–219. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2007.06.012)
- 67. Lu, Z.-H.; Zhao, Y.-G.; Yu, K. Stochastic Modeling of Corrosion Propagation for Service Life Prediction of Chloride Contaminated RC Structures. In Proceedings of the 1st International Symposium on Life-Cycle Civil Engineering, Lake Como, Italy, 11–14 June 2008; pp. 195–201.
- 68. Vu, K.A.T.; Stewart, M.G. Structural reliability of concrete bridges including improved chloride-induced corrosion models. *Struct. Saf.* **2000**, *22*, 313–333. [\[CrossRef\]](https://doi.org/10.1016/S0167-4730(00)00018-7)
- 69. Costa, A.; Appleton, J. Case studies of concrete deterioration in a marine environment in Portugal. *Cem. Concr. Compos.* **2002**, *24*, 169–179. [\[CrossRef\]](https://doi.org/10.1016/S0958-9465(01)00037-3)
- 70. Morcillo, M.; Chico, B.; De la Fuente, D.; Simancas, J. Looking Back on Contributions in the Field of Atmospheric Corrosion Offered by the MICAT Ibero-American Testing Network. *Int. J. Corros.* **2012**, *2012*, 824365. [\[CrossRef\]](https://doi.org/10.1155/2012/824365)
- 71. Castañeda, A.; Corvo, F.; Pech, I.; Valdés, C.; Marrero, R.; Del Angel-Meraz, E. Atmospheric Corrosion in an Oil Refinery Located on a Tropical Island under New Pollutant Situation. *J. Mater. Eng. Perform.* **2021**, *30*, 4529–4542. [\[CrossRef\]](https://doi.org/10.1007/s11665-021-05746-9)
- 72. Li, S.; Hihara, L.H. A micro-Raman spectroscopic study of marine atmospheric corrosion of carbon steel: The effect of Akaganeite. *J. Electrochem. Soc.* **2015**, *162*, C495. [\[CrossRef\]](https://doi.org/10.1149/2.0881509jes)
- 73. Reguer, S.; Mirambet, F.; Dooryhee, E.; Hodeau, J.L.; Dillmann, P.; Lagarde, P. Structural evidence for the desalination of Akaganeite in the preservation of iron archaeological objects, using synchrotron X-ray powder diffraction and absorption spectroscopy. *Corros. Sci.* **2009**, *51*, 2795–2802. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2009.07.012)
- 74. Ståhl, K.; Nielsen, K.; Jiang, J.; Lebech, B.; Hanson, J.C.; Norby, P.; Van Lanschot, J. On the akaganéite crystal structure, phase transformations and possible role in post-excavational corrosion of iron artifacts. *Corros. Sci.* **2003**, *45*, 2563–2575. [\[CrossRef\]](https://doi.org/10.1016/S0010-938X(03)00078-7)
- 75. Mazeina, L.; Deore, S.; Navrotsky, A. Energetics of bulk and nano-Akaganeite, β-FeOOH: Enthalpy of formation, surface enthalpy, and enthalpy of water adsorption. *Chem. Mater.* **2006**, *18*, 1830–1838. [\[CrossRef\]](https://doi.org/10.1021/cm052543j)
- 76. Cornell, R.M.; Schwertmann, U. *The Iron Oxides: Structure, Properties, Reactions, Occurrence and Uses*; VCH: Weinheim, Germany, 1996.
- 77. de la Fuente, D.; Díaz, I.; Simancas, J.; Chico, B.; Morcillo, M.J. Long-term atmospheric corrosion of mild steel. *Corros. Sci.* **2011**, *53*, 604–617. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2010.10.007)
- 78. De la Fuente, D.; Alcántara, J.; Chico, B.; Díaz, I.; Jiménez, J.A.; Morcillo, M. Characterisation of rust surfaces formed on mild steel exposed to marine atmospheres using XRD and SEM/Micro-Raman techniques. *Corros. Sci.* **2016**, *110*, 253–264. [\[CrossRef\]](https://doi.org/10.1016/j.corsci.2016.04.034)
- 79. Morcillo, M.; González-Calbet, J.M.; Jiménez, J.A.; Díaz, I.; Alcántara, J.; Chico, B.; De la Fuente, D. Environmental conditions for Akaganeite formation in marine atmosphere mild steel corrosion products and its characterization. *Corrosion* **2015**, *71*, 872–886. [\[CrossRef\]](https://doi.org/10.5006/1672) [\[PubMed\]](https://www.ncbi.nlm.nih.gov/pubmed/39125548)
- 80. Huet, B.; L'Hostis, V.; Miserque, F.; Idrissi, H. Electrochemical behavior of mild steel in concrete: Influence of pH and carbonate content of concrete pore solution. *Electrochim. Acta* **2005**, *51*, 172–180. [\[CrossRef\]](https://doi.org/10.1016/j.electacta.2005.04.014)
- 81. Asenath-Smith, E.; Lara, E. Role of Akaganeite (β-FeOOH) in the growth of hematite (α-Fe₂O₃) in an inorganic silica hydrogel. *Cryst. Growth Des.* **2015**, *15*, 3388–3398. [\[CrossRef\]](https://doi.org/10.1021/acs.cgd.5b00475)
- 82. Naren, G.; Ohashi, H.; Okaue, Y.; Yokoyama, T. Adsorption kinetics of silicic acid on Akaganeite. *J. Colloid Interface Sci.* **2013**, *399*, 87–91. [\[CrossRef\]](https://doi.org/10.1016/j.jcis.2013.02.032)

Disclaimer/Publisher's Note: The statements, opinions and data contained in all publications are solely those of the individual author(s) and contributor(s) and not of MDPI and/or the editor(s). MDPI and/or the editor(s) disclaim responsibility for any injury to people or property resulting from any ideas, methods, instructions or products referred to in the content.