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# **To cite this version:**

J.-C. Orlianges, O. Allegret, E.-N. Sirjita, A. Masson, A. Boulle, et al.. Electrical activation of insulator-to-metal transition in vanadium dioxide single-crystal nanobeam and their high-frequency switching performances. Journal of Applied Physics,  $2024$ ,  $136$  (6), pp.064502.  $10.1063/5.0221152$ . hal-04671425

# **HAL Id: hal-04671425 <https://hal.science/hal-04671425v1>**

Submitted on 19 Aug 2024

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# Electrical activation of insulator-to-metal transition in vanadium dioxide single-crystal nanobeam and their high-frequency switching performances

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(Dated: 12 July 2024)

We demonstrate the integration of vanadium dioxide single-crystal nanobeams fabricated by modified vapor–liquid–solid method (VLS) as electrical switching elements into a radio-frequency transmission line and evaluate the performances of the overall device in modulating the transmission of the conveyed RF electromagnetic waves. The switching capability of the RF device is based on the metal-insulator transition of  $\rm VO_2$  nanobeams, with on/off electrical switching ratio of  $10^4$  i.e. resistance modulation from more than  $10^6 \Omega$  when the wires are in the insulating state to only  $\approx 20\Omega$  when they are in the metal-like state. The thermal and electrical activation of the VO<sub>2</sub> wires between the two dissimilar states is resulting RF switching performances characterized by more than 15*dB* change in the transmission coefficient of the device over the 100*MHz*−24*GHz* frequency domain.

## I. INTRODUCTION

Among the strongly correlated electron materials exhibiting insulating to metal transition,<sup>1,2</sup> vanadium dioxide  $(VO_2)$ is undoubtedly the one which has attracted the most research interest both from a fundamental point of view, $3$  but also for its integration potential into electronic,<sup>4</sup> photonic<sup>5</sup> and neuromorphic devices.<sup>6</sup> The particular attention on  $VO<sub>2</sub>$ comes from the large modification of its electrical and optical properties during the onset of the metal-insulator transition (MIT) occurring at relatively low temperatures (insulating behavior below  $\approx 340K$  and metal-like characteristics above) and for the possibility to additionally trigger the state change using conveniently implemented electrical or optical stimuli.<sup>5–11</sup> Most of the applications reported in the scientific literature describe the use of the material in the form of thin films. Whether  $VO<sub>2</sub>$  thin films are epitaxial or polycrystalline, their insulating to metal transition (IMT) is characterized by the simultaneous coexistence of different phases and its percolative insulator-metal nature.<sup>7,8</sup> These peculiar properties triggered the demonstration of electrical and high frequency devices with interesting performance.<sup>10,11</sup> Thus, due to their broadband electrical response, vanadium dioxide thin films were integrated as localized switching or tuning elements in devices spanning large operational frequency domains, from microwaves and millimeter waves up to terahertz waves.10–12 Among currently employed deposition techniques for growing high quality  $VO<sub>2</sub>$  thin films are pulsed laser deposition,  $3,8,13$  magnetron sputtering  $11,14$  or reactive electron-beam evaporation.<sup>15</sup> Chemical vapor deposition (CVD) or Vapor-Liquid-Solid (VLS) techniques were recently applied to fabricate, in a simple way,  $\rm VO_{2}$  nano / microobjects like wires, plates, beams and nanoparticles which have the particularity of being single-crystalline.<sup>16</sup> This specificity of their crystalline structure makes it possible to overcome the random, percolative domain structures occurring in thin films, hardly compatible with production of reliable and reproducible devices.<sup>17</sup> Initial reports on  $VO<sub>2</sub>$  wires or nanobeams integration in electrical devices were centered on top down approaches starting from  $\rm VO_{2}$  films and their nanometer scale structuring using electron-beam lithography.<sup>18</sup> Still, the obtained  $\text{VO}_2$  beams keep the multi-domain granularity of the initial thin films resulting in strain associated with grain boundaries, dislocations or stoichiometry fluctuations.<sup>19</sup>

Vanadium dioxide micro and nanowires fabricated using the bottom-up VLS-CVD method were since thoroughly investigated not only for their peculiar fundamental properties (investigation of the metal-insulator transition and the associated physics at nanoscale in pure, single-crystal objects) but also for their applications as extremely localized temperature or strain sensors,<sup>20,21</sup> photo-and thermoelectric nanoscale switches,<sup>22</sup> solid-state thermal memories<sup>23</sup> or memristors with self-heating capabilities.<sup>24</sup> Since best quality  $\rm VO_2$  films (in terms of the magnitude of their electrical conductivity variation subsequent to the insulator to-metal transition) are obtained on sapphire substrates at relatively high temperatures (superior to 500◦*C*), this makes their CMOS (complementary metal oxide semiconductors) monolithic integration within high-frequency devices difficult. Instead,  $VO<sub>2</sub>$  micro and nanowires can be fabricated by the CVD method on  $SiO<sub>2</sub>$  layers (or substrates) and integrated afterward on a plethora of different substrates, including polymer ones.<sup>25</sup> As RF switches are key elements in reconfigurable telecommunications systems or consumer electronics, many efforts have focused lately on emerging technologies based on micro-and nanoscale devices with, potentially, improved performances than current, more conventional solutions based on CMOS or RF-MEMS technologies.<sup>26</sup> In the past, several reports studied the feasibility of using inorganic nanowires<sup>27</sup> or nanoscale memristors based on conductive filaments<sup>28</sup> as active elements for radio frequency and microwaves switches but no studies have been yet reported on  $\rm VO_{2}$  nanobeams performances at these frequency domains. Here we report for the first time to the best of our knowledge on the evaluation of

high-frequency characteristics of  $VO_2$  nanobeams and the demonstration of simple switching devices in the microwave domain based on the thermal and electrical activation of the metal-insulator transition in vanadium dioxide single-crystal beams.

#### II. METHODS

#### A. Nanowires synthesis

The synthesis was performed by a vapor transport method using commercial  $V_2O_5$  powder employed as precursor for the synthesis of  $\rm VO_{2}$  crystals, as already reported by Guiton et al.<sup>16</sup> Nucleation and growth of nanowires takes place on the surface of a substrate of  $Si(100)$  with a thermal oxide layer  $(2\mu m)$  thick amorphous  $SiO_2$ ). The substrate and powder were placed in an alumina boat. The part of the boat containing the powder was introduced precisely in the center zone of a high temperature tube furnace. After a primary vacuum step, argon was introduced into the tube at a flow rate of 200*sccm* under a pressure of 2 Torr while the furnace was heated at 850◦*C* with a speed rate of 6◦*C*/*min*. After a one-hour hold at this temperature, the sample was naturally cooled to room temperature in the same pressure conditions. The local differences in the flow of precursor at the substrate surface lead to an inhomogeneity of the morphology and the density of  $\text{VO}_2$  nanostructures on the same sample.<sup>29</sup> As shown in Fig. 1 (a) and (b), synthesis can lead to the formation of nano- and micro-scale objects with shapes ranging from micro/nanowires to microplates.

#### B. Structural characterization

XRD experiments have been performed on a Bruker D8 Discover diffractometer equipped with a parabolic multilayer mirror, a two-reflection asymmetrically cut Ge(220) monochromator (Cu  $K_{\alpha 1}$  radiation,  $\lambda = 1.5406\text{\AA}$ ) as primary optics and linear position sensitive detector. They were performed in a  $\theta - 2\theta$  configuration between 15 and 80 $\degree$  with a 0.01<sup>°</sup> angular resolution.

Micro-diffraction Laue pattern of a  $\rm VO_{2}$  nanobeam have been obtained at the BM32-IF beamline of the European Synchrotron Radiation Facility (ESRF) using the diffraction experimental setup in which the focused electrons beam has energies in the 5−22*keV* range and a spot size of 500*x*500*nm*<sup>2</sup> .

#### C. DC and high-frequency measurements

To study its dc and high-frequency thermal and electrical switching properties, a single  $VO<sub>2</sub>$  nanobeam  $(14\mu m)$  in length, width of 1.5µ*m* and height of 600*nm*) was integrated between the two discontinuous parts  $(10\mu m)$  apart) of the central signal line of a coplanar waveguide (CPW, microwave transmission line), as shown in Fig. 1(d), where the nanobeam act as a tunable impedance in the CPW. The device designed

on the  $SiO<sub>2</sub>/Si$  substrate has a center signal conductor (S) and two outer ground conductors (G) supporting travelling electromagnetic (EM) waves, with dimensions (width of the signal line and gap between the central line and grounds) adapted to  $50\Omega$ . The device was fabricated by conventional photolithography techniques using 500*nm* thick molybdenum (Mo) electrodes obtained by DC magnetron sputtering and patterned by photolithography and wet etching.

The high-frequency performances of the CPW device were evaluated in the transmission mode using a pair of Ground-Signal-Ground (GSG) high-frequency probes (Cascade, with 125*mm* spacing between the G-S-G individual probes) connected to a vector network analyzer (VNA) (ZVA from Rohde & Schwarz).

#### III. RESULTS AND DISCUSSION

Fig. 1 (c) displays the XRD data recorded for as-grown VO<sub>2</sub> nanowires on SiO<sub>2</sub>/Si substrates. In the  $\theta - 2\theta$  scan, only 011 reflections from monoclinic  $\rm VO_{2}$  M1 phase and 400 reflections from Si are observed, which demonstrates that the  $\rm VO_{2}$  nanostructures grow with the (011) planes parallel to the surface of the sample. These XRD macroscale results were confirmed by structural analysis of individual nanobeams using Laue micro-diffraction measurements, which are showing the presence of both  $VO<sub>2</sub> M1 (011)$  and  $VO<sub>2</sub> M1 (0-11)$  twinning orientations as illustrated in Fig. 2).



FIG. 1. (a) Optical microscopy image of as grown  $\text{VO}_2$  nanostructures (nanowires, nanobeams, microwires. . . ) deposited at 850◦*C* on  $SiO<sub>2</sub>/Si$  substrate (b) SEM micrograph of a cluster of nanowires illustrating their rectangular section (c) X-ray diffraction data of the  $VO<sub>2</sub>$  NWs sample (d) 3D sketch and optical microscopy image of the measuring device integrating  $\rm VO_{2}$  nanobeam

Fig. 3 displays the temperature dependence of electrical resistivity of a  $VO<sub>2</sub>$  nanobeam measured using the fabricated device. When the temperature rises, a first jump of resistivity is observed towards 60◦*C*, corresponding to the transition



FIG. 2. (a) Micro-diffraction Laue pattern of a  $VO_2$  nanobeam on  $SiO<sub>2</sub>/Si$  substrate obtained at the BM32-IF beamline of the ESRF (b) Set of diffraction spots from both orientations of M1  $\mathrm{VO}_2$  twins (and from Si substrate) (c) Schematic of the micro diffraction experimental setup in which the focused electrons beam has energies in the 5−22*keV* range and a spot size of 500*x*500*nm*<sup>2</sup>

between the M1 and M2 monoclinic insulating states. $30$  Then the resistivity gradually decreases until  $100\degree C$  where it drops to the metallic resistivity value of the rutile R phase. For the cooling process, the path is simpler, the material is suddenly transitioning from the metallic to the insulating state M1 at a temperature of 70◦*C*. The large hysteresis width of the temperature-dependent resistance and the remarkably high insulator to metal transition temperature during heating are attributed to the defect-free, single-crystalline and single domain structure of the measured nanowire.<sup>31</sup>



FIG. 3. Resistivity variation with temperature of the  $\rm VO_{2}$  nanobeam integrated in the CPW device during a heating-cooling cycle between 20 and 110◦*C*

Fig. 4 (a) and (b) show a typical current-voltage (I-V) curve of the device measured at  $50\degree C$ , as the applied bias voltage is increased from 0 to 10*V* then decreased to -10V and then finally brought back to 0V. As illustrated on the inset in the Fig. 4 (a), a 1*k*Ω resistor is used to limit the current flowing through the nanowire when switching into the metallic state. When the voltage increases beyond a threshold volt3

age,  $V_t = 9V$ , we can observe the sudden switching from the insulating state to the metallic state of the vanadium dioxide. Subsequently, when the voltage decreases, the metallic state of the nanowire persists down to a voltage of approximately 1.2*V*. This behavior is perfectly symmetrical for applied negative voltages. These I(V) measurements were carried out at different temperatures below the material's transition temperature. From these results, it is possible to estimate the necessary voltage drop on the  $VO_2$  nanobeam  $(V_{MIT})$  for the onset of its insulator-metal transition. Fig. 4 (c) illustrates the strong dependence of this activation voltage on the temperature. The linearity of the power dissipated in the nanowire (proportional to the square of *VMIT* ) as a function of the temperature indicate that the Joule heating mechanism explains the voltageinduced MIT transition.<sup>32,33</sup> Indeed, when the transition is driven by Joule heating, the power *PMIT* required to induce the activation of the nanowire depends on the temperature according to a relationship of the form:  $33$ 

$$
P_{MIT} = \frac{V_{MIT}^2}{R_I(T)} = G_{therm}(T_{MIT} - T)
$$
\n(1)

with *G*<sub>therm</sub> is the thermal conductance of the overall device,  $R_I(T)$  is the resistance of the device for a nanowire in the insulating state at the temperature T and *TMIT* is the transition temperature.

The high-frequency performances of the CPW device integrating the  $VO_2$  nanobeam were evaluated in the  $100MHz$  – 24*GHz* frequency domain in the transmission mode. The transmission ( $S_{12}$  and  $S_{21}$ ) and reflection ( $S_{11}$  and  $S_{22}$ ) parameters at the two-ports of the symmetrical CPW device  $(S_{12} = S_{21}$  and  $S_{11} = S_{22}$ ) were recorded while heating the device through the  $\text{VO}_2$  nanobeam transition temperature. Alternatively, the  $VO<sub>2</sub>$  material was electrically polarized at fixed, room temperature, by superposing on the RF signal a DC voltage between the two parts of the signal line using DC bias Ts.

The transmission *S*<sup>12</sup> parameters of the CPW device are shown on Fig. 5 (a) when thermally cycled between room temperature (24◦*C*) and 105◦*C*, through the insulator-to metal transition temperature of the  $VO<sub>2</sub>$  single crystal. At room temperature, the device is significantly blocking the transmission of the RF signal (Off state), with an isolation (attenuation of the transmitted signal) higher than 20 dB through the entire measured frequency domain. As the temperature of the device is increased over  $100^{\circ}$ C, the VO<sub>2</sub> beam becomes conductor and allow the RF waves to be transmitted through the coplanar waveguide (On state).

The hysteretic behavior of the transmission coefficient with temperature is highlighted in fig. 5 (b), (c) and (d) for frequencies of 5*MHz*, 10*GHz* and 20*GHz* respectively. These hysteresis characteristics are similar to those obtained when measuring the resistance variation of the nanowire with temperature shown in fig. 3.

The transmission losses in the metal-like state of the  $VO<sub>2</sub>$ nanobeam (better than 4 dB up to 24 GHz) can be perceived as relatively high, but this can be explained by the impedance mismatch within the gap of the CPW device and the resistance



FIG. 4. (a) Current voltage characteristic of the single domain VO<sub>2</sub> wire integrated in the CPW device at 50<sup>°</sup>C, illustrating the insulator –to-metal threshold voltage switching at ≈ 9*V* and the hysteretic behavior of the I-V curve; (b) Same graphical representation in logarithm scale for positive voltage values (c)

Plot of  $V_{MIT}^2$  (threshold voltage to induce MIT in the nanobeam) versus the temperature.

in the conductive state of the active  $\rm VO_{2}$  area, in the order of  $2\Omega/\mu$ *m*. These losses can be further reduced to acceptable values (< 1dB) through more adapted CPW designs, by decreasing the distance between the two parts of the RF signal line or by stacking in parallel several  $VO<sub>2</sub>$  wires in order to decrease their equivalent linear resistance in the metallic state under  $1 \Omega / \mu$ *m*. The device can be modeled by a simple electrical circuit consisting of a capacitor  $(C_{OFF})$  in parallel with a high-value resistance  $(R_{OFF})$  when the VO<sub>2</sub> beam is in the insulating state and by a simple resistive impedance with a low value  $(R_{ON})$  when the  $VO<sub>2</sub>$  material is in the conducting state.

The high frequency performances of RF switching technologies can be described by their figure of merit, expressed as:

$$
FOM = R_{ON} \times C_{OFF} \tag{2}
$$

The lower the FOM, the better the RF switching performances of a specific switching device. *RON* can be extracted from the experimental insertion loss (IL) data of the device  $(S_{12}$  at high temperature,  $VO_2$  in the conductive state), while *COFF* can be evaluated from the measured device isolation (*S*<sup>12</sup> parameter measured below the transition temperature, when  $\text{VO}_2$  is in the insulating state), according to the expressions below:<sup>38</sup>

$$
IL(dB) = 20 \log \left( 1 + \frac{R_{ON}}{2 Z_0} \right) \tag{3}
$$

$$
Isolation(dB) = 10 log \left( 1 + \left( \frac{1}{2 Z_0 \omega C_{OFF}} \right)^2 \right) \quad (4)
$$

Thus, at 24 GHz, we obtain for the single nanobeam  $VO<sub>2</sub>$ switching device a  $FOM \approx 180$  *fs*. This value is similar or



FIG. 5. (a) Transmission (*S*<sup>12</sup> parameter) of the CPW device during heating from room temperature to  $105\degree C$  (b), (c) and (d) Evolution of the transmission coefficient as a function of temperature for a rise (in red) and a decrease (in blue) at frequencies of 5MHz, 10GHz and 20GHz respectively.

better than the state-of-the art of current semiconductor-based switching technologies.<sup>39</sup>

In order to electrically actuate the  $VO<sub>2</sub>$  nanobeam within the CPW, we connected the device (via bias Ts) to a simple electrical polarization circuit (presented in the inset of Fig. 6) allowing to apply a voltage between the two segments of the

Reference	Activation mode	Switching amplitude at 20GHz	$VO2$ synthesis method	Substrate
Li et al. $34$	Thermal activation	$11-12dB$	Screen printing of $VO2$ ink	Kapton
Yang et al. $35$	Thermal activation	$12-13dB$	Screen printing of $VO2$ ink	sapphire
Crunteanu et al. <sup>9</sup>	Thermal and electrical activations	20dB	e-beam evaporation	sapphire
Jiang et al. $36$	Thermal activation	27dB	RF sputtering	SiN/SiO <sub>2</sub> /Alumina
Lee et al. $37$	Thermal activation	$13-15dB$	Pulsed laser deposition	$SnO_2/TiO_2(001)$
This work	Thermal and electrical activations	15dB	VLS nanowires	SiO <sub>2</sub> /Si

TABLE I. List of  $\rm VO_2$ -based RF switch specifications extracted from relevant reports in the literature

CPW signal line separated by the  $\text{VO}_2$  beam.

The measured transmission  $S_{12}$ -parameters at 58<sup>°</sup>*C* of the device submitted to a periodic square-type voltage signal (5*V* amplitude, 2*Hz*) is shown on Fig.6. As the material is changing between the two states following the electrical activation, the impedance of the  $VO<sub>2</sub>$  beam is rapidly and periodically modified between its extreme values, reflected in the periodic modulation of the *S*<sup>12</sup> parameter. The switching dynamics of the device is relatively important, keeping in mind that the active part of the material is composed of only a  $10 \mu m$  long VO<sub>2</sub> nanobeam. As in the case of thermal activation, this dynamics can be further improved by using shorter  $\rm VO_2$  beams and decreasing the gap between the two parts of the signal line or by integrating several beams in parallel within the CPW gap for decreasing the overall *RON* of the device. We compare, in Table I, the performances of our device with similar RF devices integrating  $VO<sub>2</sub>$  thin films, extracted from relevant reports in the literature. As mentioned above, even if the design of our proposed device can be further improved to consider the limited size of the  $\rm VO_{2}$  nanowire, its switching performances compare rather well with similar  $\text{VO}_2$ -based thin films RF switches.



FIG. 6. Dynamic variation of the  $S_{21}$  parameter (transmission of the CPW line) for the series  $\text{VO}_2$ -wire switch (inset) submitted to a square-shaped electrical activation waveform periodically activating the IMT in the  $VO<sub>2</sub>$  material

### IV. CONCLUSION

In conclusion, we presented the proof of concept and experimental demonstration of microwave switching using thermal and electrical activation of the insulator-to-metal transition in  $\rm VO_2$  single-crystal beams. Compared with  $\rm VO_2$  films, vanadium dioxide nanobeams have higher insulator-to-metal activation temperatures implying more stable electrical performances for larger temperature domains, up to 100◦*C*. They present convenient reversible and repetitive thermal and electrical switching behavior with more than 15 dB switching amplitude between the two states in the 100*MHz* − 24*GHz* domain. While the performances of the devices presented here can be further improved by conveniently adjusting their topologies (beam parallelization, dimensions), the demonstration of RF signals switching using  $\rm VO_{2}$  nanobeams initiates interesting opportunities for micro and nanoscale microwave devices for efficient switching and modulation of electromagnetic signals in the microwaves and millimeter waves domains.

### ACKNOWLEDGMENTS

This work was financially supported by the Agence Nationale de la Recherche (ANR) under grant no. ANR-22- CE24-0016, CIRANO project.

## DATA AVAILABILITY STATEMENT

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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