Supplementary material

Silicon mediated twin formation in laser direct energy deposited 316L stainless steel

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Figure S1. (a, c, e and g) EBSD IPFs along building direction and (b, d, f and h) ∑3 twin boundary maps of as-built (a-b and e-f) 316L-Si and (c-d and g-h) 316L samples, in which (a-d) and (e-h) are the areas from middle and bottom of polished surface, respectively. PFs and length fractions of ∑3 twin boundaries are inserted. The scaler bar for (a-h) is shown in (h).

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Figure S2. Schematic (a) of L-DED printed single line; (b1 and c1) EBSD IPF and (b2 and c2) ∑3 twin boundary maps of middle and end section of the line.



Figure S3. EBSD IPF maps of grains (a) before and (b) after surface removal with thickness of 5 μ m in 316L-Si sample.

Grains a1-6 and Grains b1-4 in Figure S3a possess similar orientations. After surface removal with thickness of 5 μ m, Grain a3 is divided into Grains a3', a3", a3", a3", a3", a3", a3", Grains a4-5 merge into Grain a4', and Grains b1-4 merge into b1', as shown in Figure S3b. Therefore, the dashed grains are entangled with each other in 316L-Si samples, which can increase the resistance of crack propagation during deformation.



Figure S4. Highlighted TRDs showing five-fold symmetry axes, occupying 3.1% of the observed area.



Figure S5. BSE images of etched microstructures of L-DED (a) 316L and (b) 316L-Si.