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▶ To cite this version:

Paul Eyméoud, Christian Turquat, Cédric Pardanaud, Christine Leroux, Alexandre Merlen. Raman spectroscopic detection of vacancies in LaFeO 3. Materials Letters, 2023, 330, pp.133296. 10.1016/j.matlet.2022.133296 . hal-04534658

HAL Id: hal-04534658 https://hal.science/hal-04534658

Submitted on 8 Apr 2024

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Raman spectroscopic detection of vacancies in LaFeO₃

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ARTICLE INFO

Keywords: Lanthanum orthoferrite Vacancies Raman spectroscopy Density Functional Theory

ABSTRACT

By coupling Raman measurements and first-principles investigations on bulk lanthanum orthoferrite (LaFeO₃), we have demonstrated that the Raman peak around 620 cm⁻¹ usually observed in these samples is a signature of vacancies within the material. This Raman peak can be used as a fast and non-destructive detection test, to determine whether a LaFeO3 sample contains a non-negligible quantity of vacancies. Finally, using first-principles calculations, we propose an infrared spectroscopic method to differentiate the chemical nature of vacancies (oxygen, iron, or lanthanum).

1. Introduction

Lanthanum orthoferrite $LaFeO_3$ constitutes a first-choice perovskite material for various industrial applications, because of its moderate cost, non-toxicity, and wide range of physico-chemical properties such as methane combustion catalysis [1], water-splitting photo-catalysis for hydrogen production [2], high-temperature refrigeration [3], superconductivity [4].

However, depending on its synthesis process, $LaFeO_3$ can spontaneously present a defective structure, with oxygen vacancies that are complementary to iron [5] or lanthanum [6] vacancies in order to ensure electroneutrality. Particularly, oxygen vacancies have major consequences on $LaFeO_3$ properties, among which polar displacement of Fe ions [7], catalytic power improvement [8, 9, 10] and modification of optical absorption [11]. Thus, an experimentalist working with such a sample needs to know whether that sample is lacunary or not.

In this Letter, we propose a way to detect vacancies in LaFeO₃ by means of Raman spectroscopy, a technique of particular interest seeing its numerous advantages (simple to implement, fast, non-destructive, micrometric resolution). We propose also a complementary infrared protocol to differentiate the chemical nature of vacancies (La, Fe, or O).

This work is organized as follows. In a first part, we will compare our Raman measurements with experimental data from literature. In a second part, we will present first-principles calculations from which Raman spectra could be deduced, which lead to a vacancy identification test. In a third part, we will propose, from first-principles calculations, an infrared (IR) identification criterion to determine the type of vacancies (La-, Fe-, or O-type vacancy).

2. Experimental acquisition of Raman spectrum

Our experimental investigations were performed on a polycrystalline bulk $LaFeO_3$ ceramic (previously purchased in Kurt J. Lesker Company [12]), using a RENISHAW Raman spectrometer with 633 nm laser (10s of time exposure, and 50 long-working-distance objective). Fig. 1 presents three Raman spectra recorded from three different grains. These spectra, exhibiting characteristic Raman modes in ranges 100-200 cm⁻¹, 250-350 cm⁻¹, 400-450 cm⁻¹, 600-650 cm⁻¹, are fully consistent with literature data [13, 14].

However, one can notice two important facts. First, considering solely the four spectra extracted from the literature, one can see that the peak around 620 cm^{-1} present serious intensity variations depending on the sample considered: it appears very intense in ref. [13, 14], moderately intense in ref. [15], and is quasi-inexistent in ref. [16]. Second, considering solely our three experimental spectra, one can note that the intensity of the 620 cm^{-1} peak varies depending on the observed grain.

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Figure 1: Raman spectra measured on three different grains on the LaFeO₃ target (red lines), in comparison with data from experimental literature (blue [15, 16] and black [13, 14] lines).

To attempt to solve this inconsistency on LaFeO₃ Raman peak intensity around 620 cm^{-1} , we examined the works of Rusevich et al. [17] and Tenne et al. [18] who both studied different type of perovskites. Using first-principles calculations, Rusevich et al. [17] showed that introducing an oxygen vacancy induces major changes in BaTiO₃ Raman spectrum, switching from a spectrum constituted of modes in range 0-550 cm⁻¹ for the pure bulk to a spectrum exhibiting a series of modes in range 600-700 cm⁻¹. On the other hand, in their experimental investigation, Tenne et al. [18] proved that oxygen vacancies in SrTiO₃ cause the appearance of a Raman peak around 700 cm⁻¹. Considering both results, it is very likely that the Raman peak towards 620 cm^{-1} detected experimentally in several LaFeO₃ samples is related to oxygen vacancies. Such assumption is compatible with Raman measurements by Blanck et al. on LaFeO₃ [19], showing that the intensity of the 620 cm^{-1} peak is decreasing during temperature-controlled hydration at 500°C. In order to verify this conjecture that vacancies are the origin of the 620 cm^{-1} Raman peak, we will then compare the perfect and lacunary LaFeO₃ spectra using first-principles calculations.

3. First-principles calculations: Raman detection of vacancies

Our calculations have been performed in the framework of Density Functional Theory (DFT) with CRYSTAL17 code [20] (details of implementation are given in Supplementary Material). We have computed the LaFeO3 Raman spectrum in the perfect bulk case, and in the lacunary cases 8.3% (resp. 12.5%) vacancies in O (resp. in Fe or La), which is realistic with respect to the compositions found in literature (for instance, LaFe_{0.82}O_{3- δ} in Ref. [5], determined by X-energy dispersive spectroscopy).

Our DFT spectra are represented on Fig. 2 with experimental data for easy comparison. The pure bulk spectrum computed in DFT (without vacancies) matches well with the experimental data from Ref. [16]. For a perfect crystal, the DFT Raman modes in range 590-700 cm⁻¹ are only detectable in logarithmic scale. However, DFT supercells with either an oxygen vacancy (either in LaO or FeO₂ plane), an iron vacancy, or a lanthanum vacancy, induce high intensity peaks in the range 590-700 cm⁻¹. It is clear that our experimental spectra are a combination of the undefective and lacunary DFT spectra. Contrary to the sample studied by Chou et al. that is clearly non-lacunary, our sample exhibits all the characteristics of a lacunary sample.



Figure 2: Raman spectra computed in DFT (perfect crystal, O vacancy in LaO plane, O vacancy in FeO2 plane, Fe vacancy, La vacancy) compared to experimental results (zone A of our target, and data from Ref. [16]).

Consequently, Raman spectroscopy is a rapid and non-destructive way to detect vacancies in LaFeO₃. Starting from the considered LaFeO₃ sample, one can acquire the Raman spectrum, then look at the 600-700 cm⁻¹ zone: if a peak is present, this means that the sample presents vacancies. However, this technique does not allow to clearly differentiate the nature of those vacancies (La-, Fe- or O-vacancies). We will thus propose differentiation criteria, based on IR spectra analysis.

4. First-principles calculations: infrared differentiation of vacancies

Our DFT calculations, presented on Fig. 3, revealed that: (*i*) the computed perfect and oxygen-deficient IR spectra are in good agreement with the experimental results from Ref. [21] (obtained by Fourier-transform infrared spectroscopy), (*ii*) contrary to the perfect and O-deficient spectra, the La-deficient and Fe-deficient spectra exhibit characteristic intense modes beyond 610 cm^{-1} , respectively two intense peaks in range $600-700 \text{ cm}^{-1}$ for Fe-deficient case and one intense peak in range $750-800 \text{ cm}^{-1}$ for La-deficient case.

5. Conclusion

To conclude, from the previous considerations, one can draw the following identification protocol of vacancies for any LaFeO₃ sample. First, one needs to acquire the Raman spectrum, and consider the zone between 600 and 700cm⁻¹: a peak in this zone means that the sample is lacunary. Second, to distinguish the chemical nature of vacancies, one can acquire the IR spectrum in 600-800 cm⁻¹ region using mid-IR techniques: (*i*) if there is no intense peak in range 600-800 cm⁻¹, there are O vacancies only, (*ii*) else, there are also Fe vacancies if two intense peaks are detected in range 600-700 cm⁻¹, and/or La vacancies if one intense peak is detected in range 750-800 cm⁻¹.

Similar Raman-based procedures of vacancies detection can be developed for others materials, such as $SrTiO_3$ [22], $PbTiO_3$ [23], ZnO [24], or CeO_2 [25].



Figure 3: IR spectra computed in DFT (perfect crystal, O vacancy in LaO plane, O vacancy in FeO2 plane, Fe vacancy, La vacancy) compared to experimental data from Ref. [21].

Acknowledgments, Data availability statement

We thank Sofiane Meradji, Michel Rérat, and Agnieszka Kopia. Centre de Calcul Intensif d'Aix-Marseille and Centre de Calcul Jean Zay are acknowledged for granting access to their high performance computing resources.

The data that support the findings of this study are available from the corresponding author upon request.

References

- P. Ciambelli, S. Cimino, L. Lisi, M. Faticanti, G. Minelli, I. Pettiti, and P. Porta *Applied Catalysis B : Environmental*, vol. 33, p. 193, 2001. https://doi.org/10.1016/S0926-3373(01)00163-1.
- [2] S. Tijare, M. Joshi, P. Padole, P. A. Mangrulkar, S. Rayalu, and N. Labhsetwar International Journal of Hydrogen Energy, vol. 37, p. 10451, 2012. https://doi.org/10.1016/j.ijhydene.2012.01.120.
- [3] N. Tahiri, S. Dahbi, I. Dani, O. E. Bounagui, and H. Ez-Zahraouy Computational and Theoretical Chemistry, vol. 1204, p. 113421, 2021. https://doi.org/10.1016/j.comptc.2021.113421.
- [4] Y. Song, Z. Chen, Q. Zhang, H. Xu, X. Lou, X. Chen, X. Xu, X. Zhu, R. Tao, T. Yu, H. Ru, Y. Wang, T. Zhang, J. Guo, L. Gu, Y. Xie, R. Peng, and D. Feng *Nature Communications*, vol. 12, p. 5926, 2021. https://doi.org/10.1038/s41467-021-26201-2.
- [5] M. Jedrusik, C. Turquat, L. Cieniek, A. Kopia, and C. Leroux European Physical Journal Applied Physics, vol. 96, p. 30301, 2021. https://doi.org/10.1051/epjap/2021210195.
- [6] J. Mizusaki, T. Sasamoto, W. R. Cannon, and H. K. Bowen Journal of the American Ceramic Society, vol. 65, p. 363, 1982. https://doi.org/10.1111/j.1151-2916.1982.tb10485.x.
- [7] R. Mishra, Y. Kim, J. Salafranca, S. K. Kim, S. Chang, A. Bhattacharya, D. Fong, S. Pennycook, S. Pantelides, and A. Borisevich *Nanoletters*, vol. 14, p. 2694, 2014. https://doi.org/10.1021/nl500601d.
- [8] X. Peng, J. Zheng, Y. Zhang, and Z. Wang Materials Letters, vol. 309, p. 131317, 2022. https://doi.org/10.1016/j.matlet.2021.131317.
- [9] M. Wu, S. Chen, and W. Xiang Chemical Engineering Journal, vol. 387, p. 124101, 2020. https://doi.org/10.1016/j.cej.2020.124101.
- [10] R. Gao, Q. Chen, W. Zhang, D. Zhou, D. Ning, G. Schumacher, D. Smirnov, L. Sun, and X. Liu Journal of Catalysis, vol. 384, p. 199, 2020. https://doi.org/10.1016/j.jcat.2020.02.024.
- [11] T. Jia, Z. Zeng, X. Zhang, P. Ohodnicki, B. Chorpening, G. Hackett, J. Lekse, and Y. Duan *Physical Chemistry Chemical Physics*, vol. 21, p. 20454, 2019. https://doi.org/10.1039/C9CP03883C.
- [12] M. Jedrusik. Doctoral dissertation, Université de Toulon, 2019. https://www.theses.fr/2019TOUL0013.
- [13] S. Chanda, S. Saha, A. Dutta, B. Irfan, R. Chatterjee, and T. Sinha Journal of Alloys and Compounds, vol. 649, p. 1260, 2015. http://dx.doi.org/10.1016/j.jallcom.2015.07.215.
- [14] A. S. Mahapatra, A. Mitra, A. Mallick, M. Ghosh, and P. Chakrabarti *Materials Letters*, vol. 169, p. 160, 2016. http://dx.doi.org/10.1016/j.matlet.2016.01.117.

- [15] C. Jagadeeshwaran, A. B. Selvadurai, V. Pazhanivelu, and R. Murugaraj, "Proceedings of the international conference on advances in new materials," 2014. https://doi.org/10.13140/2.1.1506.4966.
- [16] A. B. Selvadurai, V. Pazhanivelu, C. Jagadeeshwaran, R. Murugaraj, I. P. Muthuselvam, and F. Chou Journal of Alloys and Compounds, vol. 646, p. 924, 2015. http://dx.doi.org/10.1016/j.jallcom.2015.05.213.
- [17] L. Rusevich, E. Kotomin, G. Zvejnieks, and A. Popov Low Temperature Physics, vol. 46, p. 1185, 2020. https://doi.org/10.1063/10.0002472.
- [18] D. Tenne, I. Gonenli, A. Soukiassian, D. Schlom, S. Nakhmanson, K. Rabe, and X. Xi Physical Review B, vol. 76, p. 024303, 2007. https://doi.org/10.1103/PhysRevB.76.024303.
- [19] D. Blanck, A. Schön, A. Maede, C. Dujardin, J. Dacquin, P. Granger, J. Paul, and E. Berrier *Catalysis Today*, vol. 283, p. 151, 2017. https://doi.org/10.1016/j.cattod.2016.03.045.
- [20] R. Dovesi, A. Erba, R. Orlando, C. M. Zicovich-Wilson, B. Civalleri, L. Maschio, M. Rerat, S. Casassa, J. Baima, S. Salustro, and B. Kirtman WIREs Comput. Mol. Sci., vol. 8, p. e1360, 2018. https://doi.org/10.1002/wcms.1360.
- [21] E. Haye, E. André, F. Capon, S. Barrat, M. D. L. Pierre, R. Dovesi, and C. Carteret *Journal of Physical Chemistry C*, vol. 122, p. 10519, 2018. https://doi.org/10.1021/acs.jpcc.8b01433.
- [22] D. Chapron, F. Cordero, and M. Fontana Journal of Applied Physics, vol. 126, p. 154101, 2019. https://doi.org/10.1063/1.5115106.
- [23] K. Nishida, M. Osada, J. Sakai, N. Ito, T. Katoda, R. Ikariyama, H. Funakubo, H. Moriwake, and T. Yamamoto Journal of the Ceramic Society of Japan, vol. 121, p. 598, 2013. http://dx.doi.org/10.2109/jcersj2.121.598.
- [24] H. Fukushima, H. Uchida, H. Funakubo, T. Katoda, and K. Nishida Journal of the Ceramic Society of Japan, vol. 125, p. 445, 2017. https://doi.org/10.2109/jcersj2.16262.
- [25] M. Guo, J. Lu, Y. Wu, Y. Wang, and M. Luo Langmuir, vol. 27, p. 3872, 2011. https://doi.org/10.1021/la200292f.