

Debye's temperature and heat capacity for Sr0.15Ba0.85Bi 2 Nb 2 O 9 relaxor ferroelectric ceramic

A. Peláiz-Barranco, Y. González-Abreu, Pierre Saint-Grégoire, J.D.S. Guerra,

F. Calderón-Piñar

▶ To cite this version:

A. Peláiz-Barranco, Y. González-Abreu, Pierre Saint-Grégoire, J.D.S. Guerra, F. Calderón-Piñar. Debye's temperature and heat capacity for Sr0.15Ba0.85Bi 2 Nb 2 O 9 relaxor ferroelectric ceramic. Journal of Advanced Dielectrics, 2016, 06 (01), pp.1620001. 10.1142/S2010135X16200010. hal-04525529

HAL Id: hal-04525529 https://hal.science/hal-04525529

Submitted on 21 May 2024 $\,$

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers. L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Distributed under a Creative Commons Attribution 4.0 International License



Debye's temperature and heat capacity for Sr_{0.15}Ba_{0.85}Bi₂Nb₂O₉ relaxor ferroelectric ceramic

A. Peláiz-Barranco^{*,§}, Y. González-Abreu^{*}, P. Saint-Grégoire[†], J. D. S. Guerra[‡] and F. Calderón-Piñar^{*}

*Facultad de Física-IMRE, Universidad de La Habana

San Lázaro y L, Vedado, La Habana 10400, Cuba

[†]Department of Sciences and Arts, University of Nîmes 30021 Nîmes Cedex 01, France

[‡]Grupo de Ferroelétricos e Materiais Multifuncionais, Instituto de Física Universidade Federal de Uberlândia. 38400-902, Uberlândia - MG, Brazil [§]pelaiz@fisica.uh.cu

Received 2 September 2015; Revised 18 November 2015; Accepted 21 December 2015; Published 19 February 2016

A lead-free relaxor ferroelectric, $Sr_{0.15}Ba_{0.85}Bi_2Nb_2O_9$, was synthesized via solid-state reaction and the temperature-dependence of the heat capacity was measured in a wide temperature range. The dielectric permittivity was also measured between 500 Hz and 5 MHz in the same temperature range. No anomaly has been detected in the heat capacity curve for the whole temperature range covered in the present experiments, while broad peaks have been observed in the dielectric permittivity with high frequency dispersion. A typical relaxor behavior has been observed from the dielectric analysis. The Debye's temperature has showed a minimum value near the freezing temperature. The results are discussed considering the spin-glass model and the high frequency dispersion, which has been observed for the studied relaxor system.

Keywords: Relaxors; ferroelectrics; heat capacity; ceramics.

1. Introduction

It is known that normal ferroelectric materials go through a phase transition into the ferroelectric phase at the Curie temperature (T_C) , which is accompanied by a sharp dielectric permittivity peak without any frequency-dependence. On the other hand, relaxor ferroelectric systems show a broad maximum peak dielectric permittivity and a frequency-dependence on its corresponding temperature; the temperature of the corresponding maximum for the real (ε') and imaginary (ε'') components of the dielectric permittivity appears at different values.

An important number of researches have been developed on relaxor ferroelectrics.^{1–5} However, despite their very attractive physical properties, the identification of the nature of the dielectric response in relaxor systems still remains open and requires additional theoretical and experimental information, which can be very interesting to contribute to the explanation of the origin of the observed anomalies.

In order to understand the nature of the relaxor behavior, the thermodynamic information is very important and for that specific heat measurements over wide temperature ranges are indispensable. Heat-capacity experiments on relaxor ferro-electrics have showed different behaviors for typical relaxors such as $Pb(Mg_{1/3}Nb_{2/3})O_3$ and $Pb(Mg_{1/3}Ta_{2/3})O_3$ (see Ref. 6) than that of $BaTi_{0.65}Zr_{0.35}O_3$ relaxor ferroelectric.^{7,8} For the lead-based systems, the results have been associated to the formation of the ferroelectric polar nanoregions (PNRs)

showing an order–disorder type mechanism.⁶ A displacive type mechanism has been associated to the behavior which was observed for the lead-free ferroelectric relaxor.⁸

SrBi₂Nb₂O₉ system, which belongs to the Aurivillius family, shows a normal ferroelectric behavior in the ferroelectric–paraelectric phase transition.^{9,10} Dielectric studies in Sr_{1-x}A_xBi₂Nb₂O₉ system, with $A = Ca^{2+}$, Ba^{2+} and $0 \le x \le 30$ at.%, have also showed a normal ferroelectric behavior.¹¹ However, relaxor ferroelectric behavior has been reported for Sr_{1-x}Ba_xBi₂Nb₂O₉ when $x \ge 50$ at.%.^{5,12} The increase of the compositional disorder has been analyzed as the main cause for the relaxor behavior.⁵ It has been shown by using the refined atomic data that there is a mixing of different elements into *A* sites and bismuth sites in the layered structure. An increase of the frequency dispersion degree has also been observed with the increasing of the barium concentration.⁵

The present paper shows the heat capacity measurement results, which has been carried out on $Sr_{0.15}Ba_{0.85}Bi_2Nb_2O_9$ relaxor ferroelectric ceramic in order to study thermodynamic properties of this lead-free ferroelectric system.

2. Experimental Procedure

 $Sr_{0.15}Ba_{0.85}Bi_2Nb_2O_9$ ferroelectric ceramic was prepared by the solid-state reaction method. The powders of the starting materials were mixed with a desired weight ratio. The oxide

This is an Open Access article published by World Scientific Publishing Company. It is distributed under the terms of the Creative Commons Attribution 4.0 (CC-BY) License. Further distribution of this work is permitted, provided the original work is properly cited.

A. Peláiz-Barranco et al.

mixtures were calcined, in air atmosphere, at 950°C for 2 h in a closed alumina crucible and ball-milled again. The powders were pressed uniaxially by using 200 MPa and sintered in a closed alumina crucible at 1100°C for 1 h. A density value higher than 97% of the theoretical density values was obtained. The structural analysis by using X-ray diffraction⁵ has showed an orthorhombic structure with the space group A2₁am. The refined atomic data, which was obtained from using the X-ray diffraction pattern, has showed the mixing of different elements into A sites of the perovskite blocks and bismuth sites in the layered structure.⁵ The studied system has also been evaluated concerning dielectric, piezoelectric, pyroelectric and ferroelectric properties.^{5,13,14}

The differential scanning calorimetry (DSC) measurements were performed between 40° C and 500° C in argon atmosphere (Netzsch 404C with TASC 414/4 controller). Silver painted electrodes were applied to the opposite faces of disk-like samples by a heat treatment at 590°C. The dielectric measurements were carried out by using a computer controlled HIOKI 3532-50 LCR HiTESTER over a wide frequency range (50 Hz–1 MHz) from room temperature up to 500°C, approximately, by applying 200 mV AC to the samples.

3. Results and Discussion

Figure 1 shows the temperature-dependence of the specific heat (C_P) and also the dielectric permittivity (ε') . It is known that relaxor ferroelectrics do not exhibit thermodynamic anomalies in C_P versus *T*. However, the corresponding Debye's temperature (T_D) curves, which can be obtained by

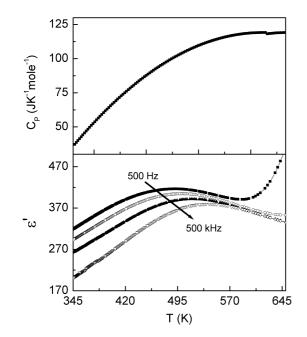


Fig. 1. Temperature-dependence of the heat capacity (C_P) and the dielectric permittivity (ε') for $Sr_{0.15}Ba_{0.85}Bi_2Nb_2O_9$ ferroelectric ceramic system.

using differential scanning calorimetric measurements, exhibit other kinds of anomalies for some relaxors.^{6,8} $Pb(Mg_{1/3}Nb_{2/3})O_3$ and $Pb(Mg_{1/3}Ta_{2/3})O_3$ have showed a minimum value for T_D at high temperatures, which has been associated to the formation of the ferroelectric PNRs showing an order-disorder type mechanism.⁶ On the other hand, no anomalous behavior for T_D has been reported for $BaTi_{0.65}Zr_{0.35}O_3$, where a displacive type mechanism has been considered to explain the results.8 Even though DSC measurement is a poor technique to analyze the relaxor ferroelectric materials, T_D versus T curves can provide elements to analyze the mechanisms, which are associated to the relaxor behavior and the formation of PNRs. All previous reports were obtained under adiabatic conditions, which are not our case, and C_P versus T curves did not show any anomaly. Under our conditions, no anomaly has been found in C_P versus T for the studied relaxor system (Fig. 1).

The results for ε' show typical behavior of relaxor ferroelectrics; (i) wide peaks in ε' vs *T* curves, (ii) the temperature for its corresponding maximum increases with frequency. The dielectric losses factor (ε'') also shows relaxor behavior characteristics.⁵

It is known that the relaxor behavior can be described by using two characteristic temperatures, the Burn's temperature (T_B) (see Ref. 15) and the freezing temperature (T_F) .⁵ PNRs are formed below T_B , whose volume, and also their interactions, increase when the temperature decreases. Then, the dielectric permittivity also increases on cooling from T_B . However, its increasing takes care until the relaxation frequency of the dipole fluctuation becomes lower than the measurement frequency. After that the dielectric permittivity decreases on cooling. The size and the interaction between PNRs become negligible below the freezing temperature (T_F) . For $T < T_F$, the applied electric field is not strong enough to break such interactions and only the smallest regions can switch with the electric field, that is to say, there is a frustration of the cooperative effect.

In order to carry out a better analysis of the heat capacity behavior, the Debye's temperature (T_D) was calculated⁸ by using Eq. (1), where n = 8 (number of atoms in the unit cell for the Aurivillius family with m = 2) and R is the gas constant. Figure 2 shows the results for the studied temperature range. The minimum values can be observed at 388 K.

$$C_P = 3nR\left(\frac{T}{T_D}\right)^3 \int_0^{T_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx,$$
 (1)

The study of the dielectric behavior for the present system has been previously made in a wide temperature and frequency ranges.⁵ The dynamics of the PNRs and the spinglass model have been considered to analyze the relaxor behavior. By using the Vogel–Fulcher equation, the freezing temperature has been obtained ($T_F = 378$ K).⁵ Thus, the minimum value for T_D occurs near T_F .

Previous study¹⁶ on BaTi_{0.65}Zr_{0.35}O₃ relaxor system has showed two anomalies in $C_p(T)$, one around 250–350 K and

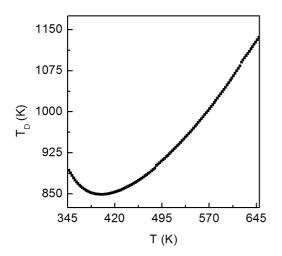


Fig. 2. Debye's temperature (T_D) for Sr_{0.15}Ba_{0.85}Bi₂Nb₂O₉ ferroelectric ceramic system.

the other one around 150–200 K. The former occurs in the same region where the dielectric permittivity starts to deviate from the Curie–Weiss law (T_B). The second one appears in the T_m region (where the dielectric permittivity shows a maximum value, which depends on frequency).

On the other hand, no obvious anomalies have been reported in $Pb(Mg_{1/3}Nb_{2/3})O_3$ relaxor system for the temperature-dependence of the heat capacity.⁶ However, the temperature-dependence for T_D has showed a minimum around 320 K, a region where the nucleation and growth of PNRs occur.

For the present studied system, no anomalies have been found in $C_p(T)$ like in Pb(Mg_{1/3}Nb_{2/3})O₃ relaxor system but T_D shows a minimum value near T_F , which is quite different from previous reports.^{6,16}

The freezing temperature is determined by the randomfield effects and the random intercluster interactions, which influence the PNRs' size distribution. A high frequency dispersion degree has been obtained for $Sr_{0.15}Ba_{0.85}Bi_2Nb_2O_9$ $(\Delta T = 54^{\circ}C)$, which is higher than that for Pb $(Mg_{1/3}Nb_{2/3})O_3$.⁵ A higher frequency dispersion degree suggests a higher size distribution of PNRs. Then, a quite different distribution of T_m values could be expected concerning Pb $(Mg_{1/3}Nb_{2/3})O_3$ system.

Considering the spin-glass model, which has been previously applied to the present system,⁵ if there are strong dipolar interactions among PNRs, an anomalous relaxation and/or hysteresis must be observed in C_P versus *T* around the freezing temperature. It has not been observed, suggesting that the dipolar interactions among PNRs are not strong enough to be thermally detected relative to the inside of PNRs. It could be associated to the high frequency dispersion around T_F for the studied system in any frequency range. However, T_D has exhibited a minimum near T_F , which has not been previously reported for other relaxor systems with lower frequency dispersion.

4. Summary

As summary, the heat capacity and the dielectric behavior were analyzed for $Sr_{0.15}Ba_{0.85}Bi_2Nb_2O_9$ lead-free relaxor ferroelectric in a wide temperature range. No anomaly was detected in the heat capacity while broad peaks were observed in the dielectric permittivity showing high frequency dispersion. The Debye's temperature showed a minimum value near the freezing temperature. The results were discussed considering the spin-glass model and the high frequency dispersion, which was observed for the studied relaxor system.

Acknowledgments

The authors wish to express their thanks to the Third World Academy of Sciences (RG/PHYS/LA Nos. 99-050, 02-225 and 05-043) and to ICTP, Trieste, Italy, for the financial support of Latin-American Network of Ferroelectric Materials (NET-43). Thanks to CNPq and FAPEMIG agencies from Brazil. APB acknowledges to the Conseil Régional Languedoc-Roussillon for her invitation to the University of Nîmes, France. Thanks to the Embassy of France in Havana, Cuba, for financial support for the scientific cooperation between the University of Nîmes and the Havana University.

References

- ¹J. H. Ko, D. H. Kim and S. Kojima, Correlation between the dynamics of polar nanoregions and temperature evolution of central peaks in Pb[(Zn_{1/3}Nb_{2/3})_{0.91}Ti_{0.09}]O₃ ferroelectric relaxors, *Appl. Phys. Lett.* **90**, 112904 (2007).
- ²V. V. Shvartsman and D. C. Lupascu, Lead-free relaxor ferroelectrics, J. Am. Ceram. Soc. 95, 1 (2012).
- ³H. Zhang, H. Yan and M. J. Reece, High temperature lead-free relaxor ferroelectric: Intergrowth Aurivillius phase BaBi₂Nb₂O₉–Bi₄Ti₃O₁₂ ceramics, *J. Appl. Phys.* **107**, 104111 (2010).
- ⁴L. E. Cross, Relaxor ferroelectrics, *Ferroelectrics* **76**, 241 (1987).
- ⁵Y. González-Abreu, A. Peláiz-Barranco, J. D. S. Guerra, Y. Gagou and P. Saint-Grégoire, From normal ferroelectric transition to relaxor behavior in Aurivillius ferroelectric ceramics, *J. Mater. Sci.* **49**, 7437 (2014).
- ⁶Y. Moriya, H. Kawaji, T. Tojo and T. Atake, Specific-heat anomaly caused by ferroelectric nanoregions in Pb(Mg_{1/3}Nb_{2/3})O₃ and Pb (Mg_{1/3}Ta_{2/3})O₃ relaxors, *Phys. Rev. Lett.* **90**, 205901 (2003).
- ⁷M. V. Gorev, V. S. Bondarev, I. N. Flerov and Ph. Sciau, Heat capacity study of double perovskite-like compounds $BaTi_{1-x}Zr_xO_3$, *Phys. Solid State* **47**, 2304 (2005).
- ⁸M. Nagasawa, H. Kawaji, T. Tojo and T. Atake, Absence of the heat capacity anomaly in the Pb-free relaxor BaTi_{0.65}Zr_{0.35}O₃, *Phys. Rev. B* **74**, 132101 (2006).
- ⁹S. Huang, Ch. Feng, L. Chen and Q. Wang, Relaxor behavior of Sr_{1-x}Ba_xBi₂Nb₂O₉ ceramics, *J. Am. Ceram. Soc.* **89**, 328 (2006).
- ¹⁰C. Miranda, M. E. V. Costa, M. Avdeev, A. L. Kholkin and J. L. Baptista, Relaxor properties of Ba-based layered perovskites, *J. Eur. Ceram. Soc.* **21**, 1303 (2001).

- ¹¹Y. Wu, M. J. Forbess, S. Seraji, S. J. Limmer, T. P. Chou, C. Nguyen and G. Z. Cao, Doping effect in layer structured SrBi₂Nb₂O₉ ferroelectrics, *J. Appl. Phys.* **90**, 5296 (2001).
- ¹²A. Peláiz-Barranco and Y. González-Abreu, Dielectric relaxation mechanisms in relaxor Bi-layered perovskites, *Ferroelectrics* 426, 122 (2012).
- ¹³Y. González-Abreu, A. Peláiz-Barranco, J. D. S. Guerra and P. Saint-Grégoire, Piezoelectric behavior in $Sr_{1-x}Ba_xBi_2Nb_2O_9$ Aurivillius-type structure ferroelectric ceramics, *Phys. Status Solidi B* **250**, 1551 (2013).
- ¹⁴A. Peláiz-Barranco, Y. González-Abreu, J. de los Santos Guerra, J. Wang, T. Yang and P. Saint-Grégoire, *Ferroelectric Materials: Synthesis and Characterization*, ed. A. Peláiz-Barranco, Chapter 4 (InTech, 2015), pp. 85–103.
- ¹⁵G. Burns and F. H. Dacol, Crystalline ferroelectrics with glassy polarization behavior, *Phys. Rev. B* 28, 2527 (1983).
 ¹⁶M. Gorev, V. Bondarev, P. Sciau and J.-M. Savariault, Heat ca-
- ¹⁶M. Gorev, V. Bondarev, P. Sciau and J.-M. Savariault, Heat capacity study of relaxors BaTi_{0.65}Zr_{0.35}O₃ and BaTi_{0.60}Zr_{0.40}O₃, *J. Phys., Condens. Matter.* **18**, 4407 (2006).