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# Imogolite Nanotubes and their Permanently Polarized Bifunctional Surfaces for Photocatalytic Hydrogen Production

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#### **Abstract**

To date, imogolite nanotubes (INTs) have been primarily used for environmental applications such as dye and pollutant degradation. However, imogolite's well- defined porous structure and distinctive electro- optical properties have prompted interest in the system's potential for energy- relevant chemical reactions. The imogolite structure leads to a permanent intrawall polarization arising from the presence of bifunctional surfaces at the inner and outer tube walls. Density functional theory simulations suggest such bifunctionality to encompass also spatially separated band edges. Altogether, these elements make INTs appealing candidates for facilitating chemical conversion reactions. Despite their potential, the exploitation of imogolite's features for photocatalysis is at its infancy, thence relatively unexplored. This perspective overviews the basic physical- chemical and optoelectronical properties of Imogolite nanotubes, emphasizing their role as wide bandgap insulator. Imogolite nanotubes have multifaceted properties that could lead to beneficial outcomes in energy-related applications. This work illustrates two case studies demonstrating a step- forward on photocatalytic hydrogen production achieved through atomic doping or metal co- catalyst. INTs exhibit potential in energy conversion and storage, due to their ability to accommodate functions such as enhancing charge separation and influencing the chemical potentials of interacting species. Yet, tapping into potential for energy- relevant application needs further experimental research, computational, and theoretical analysis.

**Keywords:** Imogolite, polarization, band-bending, photocatalysis, hydrogen production.

#### Introduction

The pressing need to attain carbon neutrality and zero emissions is driving the strong demand for sustainable chemical conversion and use of renewable green energy sources. [1,2] The goal is to reach 32% consumption by 2030 as established by the EU Energy Directive Council. [3] Solar energy, holding terawatt potential, stands as a candidate in facilitating the energy transition. [4] Notably, significant strides have been made in artificial photosynthesis, driven by photoelectrocatalytic contributions spanning the past two decades, especially in water splitting. [5–11] As a result, the proliferation of visible-light-responsive high-performance semiconductor materials—either bare or modified— has emerged as an appealing avenue. [12–14] Furthermore, the exploration of conventional metal oxides, chalcogenides, carbon-based materials, 1D nano-platforms are typical, functional and active catalysts.

Addressing the issue of photocatalysis involves processes taking place on short time scales but also on restricted distances. For the former, a variety of different time-resolved spectroscopies is now employed for tracking picosecond to microsecond events in photocatalytic materials such as surface photovoltage, transient absorption, or time-resolved spectroscopies (infrared, fluorescence, photoluminescence, microwave conductivity) to cite a few. Scientists have also worked to understand how fundamental chemical and/or physical principles differ when photocatalytic systems are restricted to nanoscale dimensions. The rise of nanoscience and nanotechnology has offered an ideal playground for exploring these concepts in a variety of different nanoreactors, from nanotubes

to more complex three-dimensional porous architectures. [25,26] The strong interest in nanotubes is due to their well-defined, oriented porous structure combined with unique electro-optical properties induced by the rolling up of the atomic lattices. [27] Early on, these iconic nanoscale objects were recognized as ultimate one-dimensional (1D) building blocks for next-generation devices that could outperform current technologies for nanofluidic, molecular sieving or energy storage. [28-31] In brief, the ion/energy storage capacity is primarily determined by the surface terminal groups, [32] likely hydroxyls in the case of INT. Furthermore, the hollow internal diameter enables ion hosting within the cavity, adding a secondary storage mechanism. As far as photocatalysis is concerned, literature is dominated by studies on TiO<sub>2</sub> nanotubes<sup>[33–35]</sup> although other photoactive metal-oxide semiconductors nanotubes have been proposed. [36] TiO<sub>2</sub> nanotubes stand out as a noteworthy example. [37,38] One prominent TiO<sub>2</sub> nanotubes platform exhibited competitive hydrogen production rates, ranging from 30 µmol h<sup>-1</sup> g<sup>-1</sup> to 80 mmol h<sup>-1</sup> g<sup>-1</sup>. [34] However, it's important to note that these rates were achieved in conjunction with the use of co-catalysts. As for bulk materials, coupling nanotubes with metallic nanoparticles is usually necessary to promote the generation and separation of photoexcited carriers in the structure. Although the loading of noble metals as co-catalysts is beyond the scope of this perspective, recent research findings highlight the potential of exploring this avenue as an alternative direction for improving hydrogen production rates. [39,40]

Another strategy is to exploit the polarization effect that originates from the non-centrosymmetric arrangement of atoms in the structure. [41,42] The permanent polarization induces positive and negative surfaces spatially separated into the catalyst, which may allow efficient electron-hole separation. Although several photo-active 2D and 3D porous nanostructures with permanent polarization have started to appear in the literature [43,44] the exploitation of 1D polarized nanotubes remains in its infancy. In the context, Imogolite nanotubes (INTs) emerge as a potentially interesting choice, given their prior use in environmental applications, *ca.* the degradation of dyes and pollutants. [45-47] Notably, INTs' cross-sectional separation of the band edges and permanent polarization may be exploited for more elaborate chemical conversion reactions, specifically in energy applications such as hydrogen evolution reaction (HER). [48]

In initial section, this perspective emphasizes the primary features and physical-chemical properties of INTs, defining their potential significance as a innovative wide bandgap insulator. In the following section delves two case studies that exemplify groundbreaking outcomes in  $H_2$  production achieved through cutting-edge material modifications: atomic doping and Schottky junction interfacing. Lastly, opportunities, directions, and modifications are proposed to continue unleashing INT's potential.

### Main features and physical-chemical properties

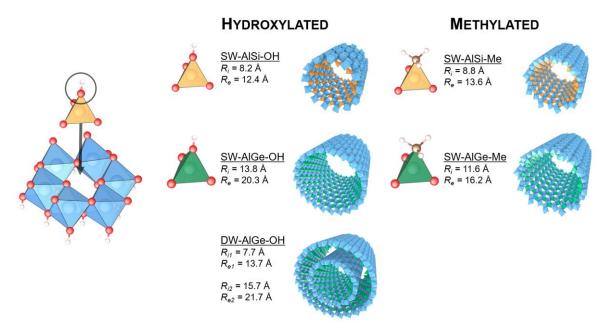
#### A) Bifunctional surface: inner and outer of the tube

Imogolite nanotubes are unique 1D nanostructures, first recognized in weathered volcanic soils as true hollow tubes,  $^{[49]}$  thirty years before the emergence of the "nanotube era".  $^{[50]}$  The chemical composition of INTs is defined as  $(OH)_3Al_2O_3Si(OH)$  from outside to inside the tube wall. It consists of an outer dioctahedral  $Al(OH)_3$  wall on which isolated tetrahedral of  $O_3Si(OH)$  are covalently bonded upright to the octahedral vacancy, forming the inner surface of the nanotube wall (Figure 1). The advantage of INTs in contrast to other nanotubes arises from a well-defined minimum of strain energy, which leads to a defined number of tetrahedra along the nanotube circumference.  $^{[52]}$ 

There are no imogolite deposits to speak of, and their extraction from soils requires several purification operations before a sufficient quantity can be obtained. On the other hand, synthetic INTs can be obtained by straightforward methods using low-temperature hydrothermal protocols. <sup>[53,54]</sup> The benefit is the ability to control or modify the nanotube crystallochemistry by changing the synthesis conditions and obtain new imogolite-like (or geo-inspired) nanostructures. <sup>[55,56]</sup> One of the most striking examples is the replacement of Si atoms by Ge ones, <sup>[57]</sup> which allows the direct synthesis of

monodisperse nanotubes in diameter with adjustable pore size (1-3 nm), shape (single-walled: SW; or double-walled: DW) and nanotube length through changes in the substitution rate ( $Si_xGe_{1-x}$ ) or the synthesis procedure (Figure 1).<sup>[58–63]</sup> These nanotubes all have the same chirality with a zig-zag configuration.<sup>[52,64,65]</sup>

The peculiar structure of INTs with inner and outer hydroxyl groups also makes them excellent candidates for surface modification. It has been widely documented that the outer wall can be functionalized with various coupling agents such as noble- and/or transition metals [66-68] and polymers. [55,69] More interestingly, the inner cavity of hydrophilic INTs can be rendered hydrophobic by a post-synthesis chemical modification. Post-functionalization can be obtained by different silane agents after a perfect dehydration of the hydrophilic INT samples, but the degree of inner surface substitution is lower than 35%. [70] A more convenient way is the direct, template-free, synthesis route by replacing, for instance, the initial alkoxide (prefiguring the tetrahedral layer) by a functionalized one. [71–73] Like for hydrophilic INTs, a dependence of the inner cavity diameter with the substitution rate is obtained from the silicon to the germanium analogue end-members with a progressive increase of the nanotube diameter. [74] Interestingly, methylated-INTs roll up into an armchair structure [75] unlike their zig-zag hydroxylated analogs, which may strongly impacted their optical and electronic properties. Overall, synthetic INTs offers a whole family of 1D nanoporous structures with functionalized interfaces, namely single -walled aluminosilicate (SW-AlSi-OH), aluminogermanate (SW-AlGe-OH), methylated aluminosilicate (SW-AlSi-Me) and methylated aluminogermanate (SW-AlGe-Me) nanotubes as well as double-walled aluminogermanate (DW-AlGe-OH) nanotubes (Figure 1).[76]

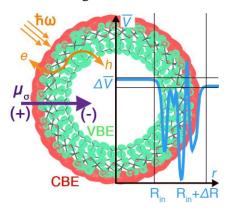


**Figure 1.** (*left*) Atomic structure of the dioctahedral aluminum layer (blue) and of the location of an  $SiO_4$  tetrahedron (orange). Oxygen atoms are in red and hydrogen ones are white. (*right*) Representation of the different members of the whole family of imogolite nanotubes with hydroxylated or methylated inner cavity. Orange tetrahedral correspond to aluminosilicate INTs while green tetrahedral are aluminogermanate INTs. Inner (outer) radius of the nanotubes is provided. Adapted from ref <sup>[76]</sup>.

# B) Permanent polarization

Owing to their tubular geometry and compositional radial symmetry, the INTs invariably present a permanent dipole density across the single or double wall interface. This feature, first inferred from electrophoresis measurements by Gustaffson in 2001, has been directly or indirectly or indirectly confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) or indirectly last the confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) last the confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) last the confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) last the confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) last the confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) last the confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) last the confirmed by Density Functional Theory (DFT) as well as Tight-Binding DFT (TB-DFT) last the confirmed by Density Functional Theory (DFT) last the confirmed by Density Funct

simulations of several members of the imogolite family. Regardless of the zigzag or armchair rolling, the INTs present, at DFT level, accumulation of negative and positive charge on the inner and outer surface, respectively. One notable exception is provided by the results for the SW-AlGe-Me nanotubes, which presents an inverted dipole density with the negative and positive ends pointing towards the outer and inner surface, respectively.<sup>[75]</sup> As shown in Figure 2, the dipole density across the nanotube walls is accompanied by a marked separation of the Valence Band edge (VBE) and Conduction Band edge (CBE) on opposite sides of the nanotube cavity, which has been suggested to be conducive to the presence of long-range charge-transfer optical excitations. [79,80,84] General, composition agnostic electrostatic models describing the interplay between the permanent dipole density and associated electrostatic potential step, thence polarizing electrostatic field (vide infra) across the nanotube wall(s) have been provided in a prior study. [80] Preliminary simulations indicate such a permanent polarization to be partially resilient to the presence of point-defects, which introduce very local perturbations in the nanotube electrostatics. [78,84] Notably, recent simulations of highly idealized, thence inevitably approximated, structures for the tube ends<sup>[81]</sup> suggest the INTs can develop, in addition to a radial dipole density, also a longitudinal polarization and ensuing band bending due to physical truncation and ensuing structural relaxation. [85]



**Figure 2.** Schematic representation of the permanent dipole density ( $\mu\sigma$ ) at the wall of imogolite nanotubes, and associated interface electrostatic potential step ( $\Delta V$ ), that can be used to polarize differently reactants inside and outside the NT cavity, offering potential control of their electron-transfer kinetics. The panel shows also the  $\mu\sigma$ -induced real-space separation between the VBE (green) and CBE (red), and intra-wall charge-transfer excitations (wiggly arrow) separating electrons (e<sup>-</sup>) and holes (h<sup>+</sup>) on different sides of the NT-cavity. Picture from Ref. [G4]. Reproduced with permission. [80] Copyright 2023, Wiley.

# C) Band-bending of adsorption and desorption species

DFT simulation of different hydroxylated and methylated INTs indicates that the dipole density across the wall, and associated electrostatic potential steps (Figure 2), is effective in tuning not only the adsorption geometry and energies of interacting molecules such as  $H_2O$  but also the energy alignment of the molecules' electron acceptor and donor levels. Combined with the demonstrated effects of the nanotubes structure and electrostatic environment in altering the molecular motions at the nanotube- $H_2O$  interfaces, these findings prompt for additional research in the potential of INTs and associated radial polarization for the design and control of electron transfer kinetics at  $H_2O$  interfaces as necessary for improved, up-scalable photocatalytic strategies to hydrogen generation.

The possibility of introducing optical absorption in the visible range of the spectrum by heteroatom doping leading to potentially beneficial photocatalytic pathways not present in the pristine nanotubes has motivated growing computational and experimental research in the subject. [46,89–93]

The nanotube polarization will self-evidently polarize differently different molecular species (with different polarizability). Suggestions have accordingly been put forward also on the potentially beneficial use of INTs as co-photocatalyst *i.e.* as support for grafted molecular or nanoparticle-based

photocatalysts.<sup>[79]</sup> As the polarizability *i.e.* response to the electric field from the nanotube dipole density will be invariably different for  $H_2O$  molecules and the grafted photocatalysts, novel and favorable band alignments may/might be engineered and realized, in principle even using systems of known unsuitability for water photo-reduction (*e.g.* Fe<sub>2</sub>O<sub>3</sub> and WO<sub>3</sub>)<sup>[94]</sup> due to their insufficiently high CBE with respect to the  $H_2/H_2O$  redox potential.

## Benefit summary of imogolite nanotubes as photoelectrocatalysts

The distinctive combination of bifunctional surface, permanent polarization, and band-bending modulation brings specific properties in alumino(germano)silicate nanotubes, enabling new chemical potentials for driving the typical energetic reactions, namely  $H_2O$  splitting,  $CO_2$  reduction, and  $N_2$  fixation. The adaptability of nanostructuring single or double walls becomes particularly interesting for improved photon management. Unlike conventional semiconductors, INTs demonstrate the ability to transport photogenerated carriers across their walls. As a result, these multifaceted nanotubes hold the potential to manipulate the chemical potentials of model photocatalytic reactions with minimal material modifications.

Together the polarization, surface, and band-bending of INTs may enable unusual but distinctive effects with promising performance benefits:

- Increase the charge carrier's concentration and continuous flux after photon activation,
- Enhancement of carriers' migration to the exposed outer surface since the distance length between the single and double walls are relatively short,
- Improvement on time scale of adsorption and desorption species by polarization and surface charge affinity with reactant molecules,
- Downward and upward of chemical potentials of model photocatalytic reactions when modified.

# Hydrogen production upon photocatalytic and radiolytic activation

Two case studies were selected from two French teams with extensive experience in INT modifications over a decade. These teams have started to explore their catalytic performance using different activation sources. The discussion of this section centers on appreciating the unique aspects of each study, given the substantial differences in experimental conditions, rather than directly comparing the material's modifications nor performance. Particularly, regarding the activation sources (nature: photon or radiation, lamps: monochromatic, full spectrum, and power intensities). Nevertheless, each catalytic case study plays a role in positioning imogolite's in the energy applications (hydrogen evolution reaction) and both are carefully analyzed to reveal its potential as nano-reactors.

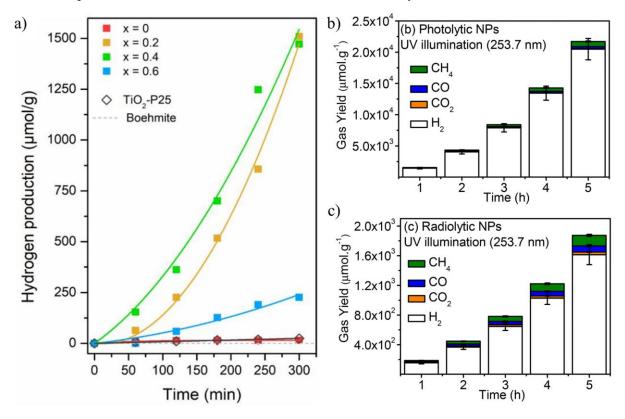
Firstly, Jiménez *et al.* developed Ti-modified aluminogermanate double-wall imogolite nanotubes. <sup>[95]</sup> The authors reported 1500 µmol<sup>-1</sup> g<sup>-1</sup> HER (Figure 3a) for the optimal 0.4 Ti/Ge-INT composite with typical sacrificial agent concentration (methanol 33% volume) and full Xe lamp (300 W) irradiation. <sup>[95]</sup> Interestingly, the present result was obtained without any co-catalyst thus positioning in the metal-free category. The high photoactivity was attributed to low recombination rates and accumulated electrons at the available Ti-sites, which presumably present longer lifetimes.

Rather than relying on metal nanoparticles as co-catalysts and electron collectors, the hybrid active composite's inherent polarization facilitates enhanced photogenerated charge separation. Methanol decomposition functions to block the oxygen evolution reaction (OER) while generating protons simultaneously. These processes occur within the bifunctional surface nanotube, leading to effective

spatial separation of useful charge carriers, with proton reduction taking place in the outer wall of the NT.

Secondly, Patra *et al.* reported a value of  $1443~\mu mol^{-1}~g^{-1}$  HER (Figure 3b) in the case of aluminosilicate INT-CH<sub>3</sub> containing 20% Au NPs as co-catalyst with propan-2-ol 20% as sacrificial agent and monochromatic light (254 nm, 15 W) irradiation. Notably, such INT-CH<sub>3</sub>/Au photoactive composite exhibits an efficient metal-semiconductor interface, leading to a substantial gold catalytic surface contribution, while the possibility of polarization and confinement effects from the INT-CH<sub>3</sub> cannot be disregarded.

INT-CH<sub>3</sub>/Au composite has proven another interesting trend in function of size-dependent. Patra *et al* reported the comparison of 5 and 10 nm average size of Au NPs given a pronounced difference on hydrogen production of 10 and 90 times higher than bare INT-CH<sub>3</sub>/Au. This outcome indicates the presence of metallic Au and highlights the active site on the outer surface of INT-CH<sub>3</sub>. This site effectively converts nearby protons into H<sub>2</sub>. The second source used was gamma radiation from <sup>137</sup>Cs in argon atmosphere (Figure 3c). However, the H<sub>2</sub> production was one order of magnitude lower than when photo-activated, suggesting that the reduction mechanism via radiation does not bring the same benefits as photon to create a feasible metal-semiconductor Schottky barrier.



**Figure 3. a)** Photocatalytic hydrogen production of Ti-modified Imogolite NTs under Xenon lamp illumination, [95] **b)** Photolytic and **c)** radiolytic hydrogen production of INTs functionalized with -CH<sub>3</sub> groups and containing Au NPs. [96]

# Conclusions and outlook

The chosen case studies on  $H_2$  generation via photocatalytic, photolytic, and radiolytic highlights the experimental proof-of-concept, evidence the clear potential of INTs as a strong candidate to be added to the list of topical semiconductor photoelectrocatalysts. Extensive progress has been achieved in characterizing diverse aspects like physico-chemical, optoelectronic, magnetic, bulk, and surface properties of this nanotube. However, it is crucial to include an examination of charge carrier dynamics in order to complete the comprehensive understanding of its behavior.

Considering the two presented case studies, it is evident that INT holds great potential to be further developed into a widely in-demand metal oxide with remarkable properties, not just for energy conversion but also for energy storage. Although the latter aspect was not covered in this article, interested readers can find comprehensive details elsewhere. On the other hand, the introduction of imogolite double-wall nanotubes provides a novel 1D nanostructure with controlled photon management for activation, opening up potentially exciting avenues for new research and advances in hydrogen/oxygen generation, carbon dioxide reduction, and nitrogen fixation, all of which guarantee further investigations.

On the horizon, there are additional opportunities for incorporating metal dopants, coupling with medium band gap semiconductors, and/or depositing bimetallic catalysts to continue harnessing and exceeding the current efficiencies of solar fuels technologies.

To further and better support future research efforts in the potential of INT for photo-catalytic applications, two lines of simulation development would be very beneficial. First, definition and validation of scalable yet sufficiently accurate routes to modelling realistic electronic ground state interfaces between the nanotube's surfaces, interacting molecules and surrounding media has remained challenging over the last ten to fifteen years. Current advances in DFT-derived machine learning interatomic potentials hold great promise to this end. Second, viable simulation methods for optical electronic excitations in the nanotubes beyond the single-particle picture used so far as well as first principles approaches to excited state (excitonic) dynamics in model Imogolite-water (solvent) nanotube interfaces are currently missing or yet to be applied. While the continuous increase in academically accessible high performance computing resources authorizes outlooks towards the use of many-body approaches to valence-electron optical spectroscopy, progress on the latter front will require intense method development efforts. At least to best of our knowledge, the needed computational machinery is yet to be developed or made available.

As for electrocatalytic applications, it is recommended to measure/report the INTs conductivity and/or resistivity. One can think of implementing significant vacancies or atoms in their lattice that can enhance their conductivity, and therefore electron collection for pure electrocatalysis.

#### **Conflict of interest**

The authors declare no conflict of interest.

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#### References

- [1] I. F. Teixeira, P. Jiménez-Calvo, in *Renew. Energy Prod. Distrib. Vol.* 2, Academic Press, **2023**, pp. 145–180.
- [2] E. C. Okonkwo, M. Al-Breiki, Y. Bicer, T. Al-Ansari, *Int. J. Hydrogen Energy* **2021**, *46*, 35525.
- (3) "Promotion of the use of energy from renewable sources," can be found under https://eurlex.europa.eu/legal-content/EN/TXT/?uri=CELEX%3A32018L2001, **2018**.
- [4] P. Jiménez-Calvo, V. Caps, V. Keller, Renew. Sustain. Energy Rev. 2021, 149, 111095.

- [5] W.-J. Ong, L.-L. Tan, Y. H. Ng, S.-T. Yong, S.-P. Chai, Chem. Rev. 2016, 116, 7159.
- [6] K. Takanabe, ACS Catal. 2017, 7, 8006.
- [7] M. Melchionna, P. Fornasiero, *ACS Catal.* **2020**, *10*, 5493.
- [8] M. Jerigova, Y. Markushyna, I. F. Teixeira, B. Badamdorj, M. Isaacs, D. Cruz, I. Lauermann, M. Á. Muñoz-Márquez, N. V. Tarakina, N. López-Salas, O. Savateev, P. Jiménez-Calvo, ChemRxiv 2022, DOI 10.26434/CHEMRXIV-2022-C21JC.
- [9] I. F. Teixeira, N. V. Tarakina, I. F. Silva, N. López-Salas, A. Savateev, M. Antonietti, *Adv. Sustain. Syst.* **2022**, 2100429.
- [10] L. Schumacher, R. Marschall, *Top. Curr. Chem.* 2022 3806 **2022**, 380, 1.
- [11] J. Yang, D. Wang, H. Han, C. Li, Acc. Chem. Res. 2013, 46, 1900.
- [12] M. Isaacs, J. Garcia-Navarro, W.-J. Ong, P. Jiménez-Calvo, Glob. Challenges 2022, 2200165.
- [13] P. Jiménez-Calvo, L. Michel, V. Keller, V. Caps, ACS Appl. Mater. Interfaces 2021, 13, 61015.
- [14] U. Caudillo-Flores, M. J. Muñoz-Batista, R. Luque, M. Fernández-García, A. Kubacka, *Chem. Eng. J.* **2019**, *378*, 122228.
- [15] J. Ma, T. J. Miao, J. Tang, Chem. Soc. Rev. 2022, 51, 5777.
- [16] C. Colbeau-Justin, M. Kunst, D. Huguenin, J. Mater. Sci. 2003, 38, 2429.
- [17] C. Colbeau-Justin, M. A. Valenzuela, *Time-Resolved Microwave Conductivity (TRMC) a Useful Characterization Tool for Charge Carrier Transfer in Photocatalysis, a Short Review*, **2013**.
- [18] R. Godin, Y. Wang, M. A. Zwijnenburg, J. Tang, J. R. Durrant, *J. Am. Chem. Soc.* **2017**, *139*, 5216.
- [19] W. Wang, Y. Tao, L. Du, Z. Wei, Z. Yan, W. K. Chan, Z. Lian, R. Zhu, D. L. Phillips, G. Li, *Appl. Catal. B Environ.* **2021**, 282, 119568.
- [20] T. Dittrich, S. Fiechter, A. Thomas, *Appl. Phys. Lett.* **2011**, *99*, 84105.
- [21] P. Jiménez-Calvo, C. Marchal, T. Cottineau, V. Caps, V. Keller, *J. Mater. Chem. A* **2019**, 7, DOI 10.1039/c9ta01734h.
- [22] J. Liu, H. Wang, M. Antonietti, Chem. Soc. Rev. 2016, 45, 2308.
- [23] X. Wang, K. Maeda, A. Thomas, K. Takanabe, G. Xin, J. M. Carlsson, K. Domen, M. Antonietti, *Nat. Mater.* **2009**, *8*, 76.
- [24] P. Jiménez-Calvo, V. Caps, M. N. Ghazzal, C. Colbeau-Justin, V. Keller, *Nano Energy* **2020**, 75, 104888.
- [25] B. Višić, L. S. Panchakarla, R. Tenne, J. Am. Chem. Soc. 2017, 139, 12865.
- [26] P. Jimenéz-Calvo, M. J. Muñoz-Batista, M. Isaacs, V. Ramnarain, D. Ihiawakrim, X. Li, M. Ángel Muñoz-Márquez, G. Teobaldi, M. Kociak, E. Paineau, *Chem. Eng. J.* **2023**, *459*, 141514.
- [27] M. Serra, R. Arenal, R. Tenne, *Nanoscale* **2019**, *11*, 8073.
- [28] S. Guo, E. R. Meshot, T. Kuykendall, S. Cabrini, F. Fornasiero, Adv. Mater. 2015, 27, 5726.
- [29] H. G. Park, Y. Jung, Chem. Soc. Rev. 2014, 43, 565.
- [30] S. Kumar, G. Saeed, L. Zhu, K. N. Hui, N. H. Kim, J. H. Lee, *Chem. Eng. J.* **2021**, *403*,

126352.

- [31] L. Bocquet, Nat. Mater. 2020 193 2020, 19, 254.
- [32] O. Brummel, D. Besold, T. Döpper, Y. Wu, S. Bochmann, F. Lazzari, F. Waidhas, U. Bauer, P. Bachmann, C. Papp, H. P. Steinrück, A. Görling, J. Libuda, J. Bachmann, *ChemSusChem* **2016**, *9*, 1424.
- [33] I. Paramasivam, H. Jha, N. Liu, P. Schmuki, *Small* **2012**, *8*, 3073.
- [34] M. Ge, Q. Li, C. Cao, J. Huang, S. Li, S. Zhang, Z. Chen, K. Zhang, S. S. Al-Deyab, Y. Lai, M. Z. Ge, Q. S. Li, C. Y. Cao, J. Y. Huang, S. H. Li, S. N. Zhang, K. Q. Zhang, Y. K. Lai, Z. Chen, S. S. Al-Deyab, *Adv. Sci.* 2017, 4, 1600152.
- [35] M. Abdullah, S. K. Kamarudin, Renew. Sustain. Energy Rev. 2017, 76, 212.
- [36] R. S. Devan, R. A. Patil, J. H. Lin, Y. R. Ma, Adv. Funct. Mater. 2012, 22, 3326.
- [37] Y. Tao, Z. Ma, W. Wang, C. Zhang, L. Fu, Q. Zhu, Y. Li, G. Li, D. Zhang, *Adv. Funct. Mater.* **2023**, *33*, DOI 10.1002/adfm.202211169.
- [38] F. Li, H. Huang, G. Li, D. Y. C. Leung, J. Nanoparticle Res. 2019, 21, 29.
- [39] S. Xiao, P. Liu, W. Zhu, G. Li, D. Zhang, H. Li, Nano Lett. 2015, 15, 4853.
- [40] W. Wang, X. Zhao, Y. Cao, Z. Yan, R. Zhu, Y. Tao, X. Chen, D. Zhang, G. Li, D. L. Phillips, *ACS Appl. Mater. Interfaces* **2019**, *11*, 16527.
- [41] L. Li, P. A. Salvador, G. S. Rohrer, *Nanoscale* **2013**, *6*, 24.
- [42] F. Chen, H. Huang, L. Guo, Y. Zhang, T. Ma, *Angew. Chemie Int. Ed.* **2019**, DOI 10.1002/anie.201901361.
- [43] F. Sastre, V. Fornés, A. Corma, H. García, J. Am. Chem. Soc. 2011, 133, 17257.
- [44] R. Su, Y. Shen, L. Li, D. Zhang, G. Yang, C. Gao, Y. Yang, Small 2015, 11, 202.
- [45] K. ichi Katsumata, X. Hou, M. Sakai, A. Nakajima, A. Fujishima, N. Matsushita, K. J. D. MacKenzie, K. Okada, *Appl. Catal. B Environ.* **2013**, *138–139*, 243.
- [46] E. Shafia, S. Esposito, M. Armandi, E. Bahadori, E. Garrone, B. Bonelli, *Catal. Today* **2016**, 277, 89.
- [47] E. Bahadori, V. Vaiano, S. Esposito, M. Armandi, D. Sannino, B. Bonelli, *Catal. Today* **2018**, *304*, 199.
- [48] J. Garcia-Navarro, M. A. Isaacs, M. Favaro, D. Ren, W.-J. Ong, M. Grätzel, P. Jiménez-Calvo, *Glob. Challenges* **2023**, 2300073.
- [49] N. Yoshinaga, S. Aomine, *Soil Sci. Plant Nutr.* **1962**, 8, 22.
- [50] S. Iijima, *Nature* **1991**, *354*, 56.
- [51] L. Guimarães, A. N. Enyashin, J. Frenzel, T. Heine, H. A. Duarte, G. Seifert, *ACS Nano* **2007**, *1*, 362.
- [52] A. D'Angelo, E. Paineau, S. Rouzière, É. Elkaim, C. Goldmann, D. Toquer, S. Rols, P. Launois, *Appl. Clay Sci.* **2023**, *242*, 107043.
- [53] V. C. Farmer, A. R. Fraser, in *Dev.* [[Sedimentology]], Elsevier, **1979**, pp. 547–553.
- [54] V. C. Farmer, M. J. Adams, A. R. Fraser, F. Palmieri, *Clay Miner.* **1983**, *18*, 459.
- [55] E. Paineau, Appl. Sci. **2018**, 8, 1921.

- [56] J. Govan, N. Arancibia-Miranda, M. Escudey, B. Bonelli, F. Tasca, *Mater. Chem. Front.* **2021**, *5*, 6779.
- [57] S. I. Wada, K. Wada, Clays Clay Miner. 1982, 30, 123.
- [58] G. I. Yucelen, R. P. Choudhury, A. Vyalikh, U. Scheler, H. W. Beckham, S. Nair, *J. Am. Chem. Soc.* **2011**, *133*, 5397.
- [59] A. Thill, B. Guiose, M. Bacia-Verloop, V. Geertsen, L. Belloni, *J. Phys. Chem. C* **2012**, *116*, 26841.
- [60] A. Thill, P. Maillet, B. Guiose, O. Spalla, L. Belloni, P. Chaurand, M. Auffan, L. Olivi, J. Rose, *J. Am. Chem. Soc.* **2012**, *134*, 3780.
- [61] M. S. Amara, E. Paineau, M. Bacia-Verloop, M. E. M. Krapf, P. Davidson, L. Belloni, C. Levard, J. Rose, P. Launois, A. Thill, *Chem. Commun.* **2013**, *49*, 11284.
- [62] J. F. Moore, E. Paineau, P. Launois, M. S. P. Shaffer, ACS Appl. Mater. Interfaces 2021, 13, 17940.
- [63] C. H. Lam, A. C. Yang, H. Y. Chi, K. Y. Chan, C. C. Hsieh, D. Y. Kang, *ChemistrySelect* **2016**, *1*, 6212.
- [64] L. Scalfi, G. Fraux, A. Boutin, F.-X. Coudert, *Langmuir* **2018**, *34*, 6748.
- [65] P. D. G. Cradwick, V. C. Farmer, J. D. Russell, C. R. Masson, K. Wada, N. Yoshinaga, *Nat. Phys. Sci.* 1972, 240, 187.
- [66] E. Cruces, N. Arancibia-Miranda, K. Manquián-Cerda, F. Perreault, N. Bolan, M. I. Azócar, V. Cubillos, J. Montory, M. A. Rubio, B. Sarkar, *ACS Appl. Nano Mater.* **2022**, *5*, 1472.
- [67] C. Hamon, E. Beaudoin, P. Launois, E. Paineau, J. Phys. Chem. Lett. 2021, 12, 5052.
- [68] L. M. Liz-Marzan, A. P. Philipse, J. Phys. Chem. 1995, 99, 15120.
- [69] W. Ma, W. O. Yah, H. Otsuka, A. Takahara, J. Mater. Chem. 2012, 22, 11887.
- [70] D.-Y. Kang, J. Zang, C. W. Jones, S. Nair, J. Phys. Chem. C 2011, 115, 7676.
- [71] C. Zanzottera, A. Vicente, E. Celasco, C. Fernandez, E. Garrone, B. Bonelli, *J. Phys. Chem. C* **2012**, *116*, 7499.
- [72] M. Boyer, E. Paineau, M. Bacia-Verloop, A. Thill, *Appl. Clay Sci.* **2014**, *96*, 45.
- [73] P. Picot, F. Gobeaux, T. Coradin, A. Thill, Appl. Clay Sci. 2019, 178, 105133.
- [74] M. S. Amara, E. Paineau, S. Rouzière, B. Guiose, M.-E. M. Krapf, O. Taché, P. Launois, A. Thill, *Chem. Mater.* **2015**, *27*, 1488.
- [75] G. Monet, M. S. Amara, S. Rouzière, E. Paineau, Z. Chai, J. D. Elliott, E. Poli, L.-M. Liu, G. Teobaldi, P. Launois, *Nat. Commun.* **2018**, *9*, 2033.
- [76] G. Monet, Nanotubes Géo-Inspirés : Structure Atomique, Transformation En Température et Dynamiques Corrélées Nanotube-Eau Moléculaire, University of Paris Saclay, **2019**.
- [77] J. P. Gustafsson, Clays Clay Miner. 2001, 49, 73.
- [78] G. Teobaldi, N. S. Beglitis, A. J. Fisher, F. Zerbetto, W. A. Hofer, *J. Phys. Condens. Matter* **2009**, *21*, 195301.
- [79] E. Poli, J. D. Elliott, L. E. Ratcliff, L. Andrinopoulos, J. Dziedzic, N. D. M. Hine, A. A. Mostofi, C.-K. Skylaris, P. D. Haynes, G. Teobaldi, *J. Phys. Condens. Matter* **2016**, 28, 74003.
- [80] J. D. Elliott, E. Poli, I. Scivetti, L. E. Ratcliff, L. Andrinopoulos, J. Dziedzic, N. D. M. Hine, A.

- A. Mostofi, C.-K. Skylaris, P. D. Haynes, G. Teobaldi, Adv. Sci. 2017, 4, 1600153.
- [81] E. Poli, J. D. Elliott, N. D. M. Hine, A. A. Mostofi, G. Teobaldi, https://doi.org/10.1179/1432891715Z.0000000001560 2015, 19, S272.
- [82] F. Alvarez-Ramírez, *Phys. Rev. B* **2007**, *76*, 125421.
- [83] M. P. Lourenço, L. Guimarães, M. C. da Silva, C. de Oliveira, T. Heine, H. A. Duarte, J. Phys. Chem. C 2014, 118, 5945.
- [84] E. Poli, J. D. Elliott, S. K. Chulkov, M. B. Watkins, G. Teobaldi, *Front. Chem.* **2019**, *7*, DOI 10.3389/fchem.2019.00210.
- [85] E. Poli, J. D. Elliott, Z. Chai, G. Teobaldi, Cryst. 2020, Vol. 10, Page 1051 2020, 10, 1051.
- [86] E. Poli, J. D. Elliott, L. E. Ratcliff, L. Andrinopoulos, J. Dziedzic, N. D. M. Hine, A. A. Mostofi, C.-K. Skylaris, P. D. Haynes, G. Teobaldi, J. Phys. Condens. Matter 2016, 28, 074003.
- [87] G. Monet, E. Paineau, Z. Chai, M. S. Amara, A. Orecchini, M. Jimenéz-Ruiz, A. Ruiz-Caridad, L. Fine, S. Rouzière, L. M. Liu, G. Teobaldi, S. Rols, P. Launois, *Nanoscale Adv.* 2020, 2, 1869.
- [88] S. Le Caër, M. C. Pignié, Q. Berrod, V. Grzimek, M. Russina, C. Carteret, A. Thill, J. M. Zanotti, J. Teixeira, *Nanoscale Adv.* **2021**, *3*, 789.
- [89] F. Alvarez-Ramírez, J. Chem. Theory Comput. 2009, 5, 3224.
- [90] A. Avellan, C. Levard, N. Kumar, J. Rose, L. Olivi, A. Thill, P. Chaurand, D. Borschneck, A. Masion, *RSC Adv.* **2014**, *4*, 49827.
- [91] E. Shafia, S. Esposito, M. Manzoli, M. Chiesa, P. Tiberto, G. Barrera, G. Menard, P. Allia, F. S. Freyria, E. Garrone, B. Bonelli, *J. Nanoparticle Res.* **2015**, *17*, 1.
- [92] E. Shafia, S. Esposito, M. Armandi, M. Manzoli, E. Garrone, B. Bonelli, *Microporous Mesoporous Mater.* **2016**, 224, 229.
- [93] I. S. Popov, A. N. Enyashin, *Phys. status solidi* **2021**, 258, 2100188.
- [94] M. Grätzel, Nature 2001, 414, 338.
- [95] P. Jimenéz-Calvo, A. Sobolewska, M. Isaacs, Y. Zhang, A. Leforestier, J. Degrouard, S. Rouzière, C. Goldmann, D. Vantelon, P. Launois, M. N. Ghazzal, E. Paineau, *ChemRxiv* **2023**, DOI 10.26434/CHEMRXIV-2023-QCXSH.
- [96] S. Patra, F. Testard, F. Gobeaux, L. Sicard, D. Shaming, S. Le Caër, A. Thill, *Nanoscale* **2023**, *15*, DOI 10.1039/d2nr05796d.