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Modelling of viscoelastic properties and crack growth in bituminous mixtures: Application to the simulation of crack growth in semi-circular samples subjected to oxidative ageing

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26 **1 Introduction**

27 Cracking represents one of the main risk of pavement damage over the time. It is often acceler-
28 ated by the ageing of bitumen in bituminous pavement. Understanding of the effect of ageing on
29 the mechanical and fracture properties of bituminous mixes is therefore a key step in predicting
30 the service life of pavement structures. Most of the studies in the literature deals with the effect
31 of ageing on rheological properties of binders and mixes [1–3]. These studies agree that the
32 stiffness modulus increases with ageing. However, the effect of ageing on the cracking resistance
33 of bituminous mixes is still under investigation. Only Brah’s et al. and Farhad et al. works [4–6]
34 combine these two aspects by limiting their investigations either to one loading rate or to a single
35 test temperature. At low temperatures, their studies show that the fracture parameters are even
36 more degraded as the ageing time increases. At these temperatures, bituminous mixes are often
37 considered as brittle or quasi-brittle elastic materials and under the assumption of a negligible
38 fracture process zone around the crack tip, several works have been conducted applying linear
39 elastic fracture mechanics (LEFM) to study the fracture behaviour of bituminous mixes [7–12].
40 Generally, the size of the fracture process zone is relatively large in these materials which limits
41 the accuracy of the predictive models based on the LEFM concepts.

42 The crack growth mechanism in bituminous mixes is relatively complex and involves the use
43 of fracture mechanics principles combined with time and temperature dependent constitutive
44 models to accurately capture crack initiation and growth. The high complexity inherent in
45 modelling bituminous mixes as a heterogeneous material leads to the assumption of an isotropic
46 viscoelastic homogeneous material in most modelling approaches. Relatively few studies deal
47 with the modelling of fracture in viscoelastic materials.

48 The first works on fracture mechanics based approaches in this domain include the important
49 contributions of Knauss [13, 14] and Schapery [15–17]. From the extended viscoelastic cor-
50 respondence principle [18], R. A. Schapery [17] proposed a viscoelastic integral $\mathcal{J}_{\dot{\gamma}}$ extending
51 Rice’s integral, used in linear elastic fracture mechanics, to viscoelastic materials. Based on this
52 theory, several time-dependent cracking models taking into account viscous deformations have
53 been proposed (e.g. [19, 20]). The application of this technique of calculation of the energy
54 release rate is very limited, as the conditions of application of the extended correspondence
55 principle do not lend itself to all real problems. Dubois et al. [21–25] developed a new method
56 of invariant surface integral $\mathcal{G}\theta_v$ to compute the rate of energy restitution. Based on Brincker’s
57 description [26], Dubois et al. introduced two intensity factors, namely the viscoelastic opening

58 intensity factor to determine the relative displacements in the vicinity of the crack bottom and
59 the stress intensity factor, already known in LEFM. They proposed an alternative formulation
60 of the $\mathcal{G}\theta_c$ integral calculation using the spectral decomposition of the reduced viscoelastic com-
61 pliance: starting from a generalized Kelvin-Voigt rheological model (KVG) of n Kelvin-Voigt
62 branches, these authors decoupled the general cracking problem into n elastic sub-problems by
63 assuming that the total displacement can be decomposed into a sum of elastic displacement of
64 each Kelvin-Voigt chain. This presents a concern due to the fact that the strain fields in the
65 Kelvin-Voigt chains are generally not geometrically compatible.

66 Cohesive zone models (CZMs) are the most widely used numerical local damage models for mod-
67 elling crack initiation and growth in bituminous mixes [27–35]. The different tensile-separation
68 laws of the CZM models are discussed in [36]. Li and Marasteanu [28], Kim and Buttlar [33]
69 investigated the fracture behaviour of asphalt mixtures under low temperature conditions using
70 the CZM bilinear law. Song et al.[30] and Arago et al.[35] modeled the crack growth in bitu-
71 minous mixes at room temperature by considering the exponential and bilinear approaches of
72 the time-independent cohesive model associated with the viscoelastic behaviour of the material,
73 respectively. While the CZM model is widely used in the bituminous mixes community, it is still
74 subject to debate. Indeed, although the method is academically interesting, one can naturally
75 question the fact that the viscoelastic behaviour of the material is locally disturbed by an elastic
76 law. To overcome this problem, C. Yoon et al. [37] recently proposed a non-linear viscoelastic
77 cohesive zone (NVCZ) model with a damage function making the model dependent on this.
78 Kim et al. [38] applied the NVCZ model to study a bituminous media fracture by considering
79 a Gaussian law as damage function. Their studies revealed that the proposed model takes into
80 account the rate-dependent effect and can successfully predict test results at different rates by
81 only taking the results at an arbitrary loading rate. One of the drawbacks of this model is that
82 it introduces additional parameters whose determination is not that easy. In addition, these
83 models may suffer a mesh dependence.

84 Non-local damage models are another technique for simulating material crack initiation and
85 growth. Several authors have recently extended some of these models to viscoelastic media: the
86 phase field approach [39–42], the Thick Level Set model [43]. The latest has been applied, in 1D
87 case, to model the uniaxial tensile stress tests of bituminous mixes at different temperatures.

88 In this paper, we focus on the modelling of crack propagation by using Griffith’s macroscopic
89 viewpoint [44].

90 Nguyen et al. [45] modeled crack propagation in a viscoelastic medium. In their work, the ther-

91 modynamic framework of viscoelastic crack propagation was revisited using the Burger model.
92 They showed that this approach allows to highlight the loading rate effect on the behaviour of
93 the crack growth driving force and the evolution of the crack length. They also investigated the
94 interaction between viscous dissipation and fracture growth in geological materials subjected to
95 long-term loading [46]. Relying on rigorous thermodynamic basis, this efficient approach could
96 be used to study crack growth in bituminous mixes. However, the used rheological model does
97 not allow to describe the thermomechanical behaviour of bituminous mixes over a wide range
98 of solicitation's frequencies.

99 In this study, this energetic approach is extended to the Generalized Maxwell model, in order
100 to predict the fracture behaviour of bituminous mixtures.

101 To this end, this paper is organized as follows.

102 Section 2 presents the materials and the ageing protocol of the loose mixtures. Section 3 de-
103 scribes test methods for the characterisation of viscoelastic (complex modulus test) and cracking
104 (semi-circular bending test) properties of bituminous mixtures. In section 4, in the perspective
105 of numerical developments of boundary value problems, an explicit incremental form of the vis-
106 coelastic behaviour law is presented. The thermodynamic framework of crack propagation in
107 viscoelastic media is then revisited. A crack growth criterion taking into account the viscous ef-
108 fects is derived. The energy release rate, representing the crack growth driving force, is identified
109 in a form that allows its numerical determination in the framework of the previous incremental
110 approach. The section 5 discusses the proposed energy release rate expressions for the case of
111 a semi-circular geometry subjected to 3-point bending. The numerical implementation of this
112 quantity is then illustrated by confronting the numerical results with the proposed expressions
113 and by examining the nature of the asymptotic responses of this driving force. In addition,
114 the stability of crack growth in dissipative media is discussed. It reveals the role played by the
115 resistance to fracture, i.e. the critical energy release rate. Finally, a validation of the proposed
116 crack growth criterion is presented in section 6. The selected form of the critical energy release
117 rate is discussed and different comparisons between simulations and experimental results are
118 carried out.

119 2 Materials and mix design

120 2.1 Materials properties

121 The mixes are produced with a paving grade bitumen 35/50, whose properties of needle penetra-
122 tion (EN 1426) [47] and softening temperature (EN 1427) [48] are respectively 42×0.1 mm and
123 53.2 °C, and porphyry aggregates which come from *Pont de Colonne* (in France) quarry whose
124 abrasion loss and fragmentation properties respectively known as Microdeval (EN 1097-1) [49]
125 value and Los Angeles (EN 1097-2) [50] value are 17% and 8%. It is used for surface and binder
126 courses and labelled AC10 mix. The mixture composition and the densities of the constituents
127 are given in Table 1.

Constituent	Volume fraction (in %)	Density (in $kg.m^{-3}$)
Limestone filler	2.8	2700
0/2	26.1	
2/6.3	23.7	2630
6.3/10	42.	
Bitumen	5.4	1040

Table 1: Composition of mixture AC10

128 2.2 Ageing of bituminous mixture

129 The ageing protocol adopted in the curent study is the RILEM ageing procedure described in
130 [51]. The mixes are manufactured at $165^{\circ}C$. After the mixing, the loose mixture are cooled
131 down to $135^{\circ}C$ and spread in thin layer (not more than 2.5 cm) into a tray and placed in a
132 ventilated oven. Part of the mixture is kept at this temperature for 8 hours to simulate a short
133 term ageing. Then, they are cooled down to $85^{\circ}C$ and kept respectively in the oven for 3 and
134 6 days to reproduce the long term ageing. 2 hours before the end of the ageing procedure, the
135 loose mixes are reheated at $165^{\circ}C$ and then compacted. The reference mix is compacted without
136 long-term ageing and designated V0 while those subject to 8 hours short-term ageing followed
137 by long-term ageing are labeled V3, V6 for respectively 3 and 6 days ageing.

138 3 Test methods

139 3.1 Complex modulus test: 2-point cantilever beam method

140 At the end of the ageing process, the loose mixes are compacted according to EN 12697-33 [52].
141 The manufactured bituminous slabs are then sawn into prismatic specimens of dimensions $4 \times$

142 4×12 cm and subjected to complex modulus measurements. Sinusoidal cyclic deflections are
 143 performed on a prismatic cantilever beam in accordance to EN 12697-26 (Annex A: 2PB-PR
 144 [53]). During the test, thermal conditioning of the specimens is ensured by a thermal chamber. A
 145 temperature probe located near the surface of the specimen is used to measure the temperature.
 Fig. 1 shows a general view of the experiment. Tests are performed at 4 frequencies (from 3 Hz

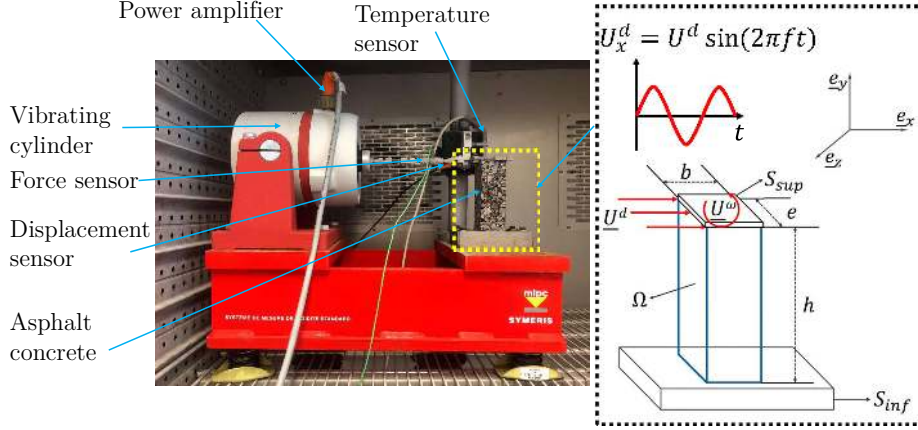


Fig. 1. Mixes complex modulus $E^*(\omega)$ test principle: $b= 40$ mm , $e= 40$ mm , $h= 120$ mm

146
 147 to 40 Hz) and at 8 temperatures (from -30 to 50°C). The measured force F and the imposed
 148 displacement U allow to calculate the complex modulus $|E^*|$ according to a calculation provided
 149 in [53]. The isotherms of $|E^*|$ are then used to build the master curves (see Fig. 2). Complex
 150 modulus test results are modeled using the Generalized Maxwell (GM) model [54]. The GM
 151 analytical expression of the complex Youngs modulus is given by the following equation (1):

$$E^*(\omega) = E_0 + \sum_{i=1}^{i=m} \left(\frac{E_i j\omega\tau_i}{1 + j\omega\tau_i} \right) \quad (1)$$

152 Where E_i and τ_i respectively represent the i^{th} spring modulus and characteristic time. E_i and τ_i
 153 are determined at each isotherm by minimizing the error between $E_{measured}^*$ and E_{GM}^* according
 154 to the least-squares method (see Fig. 2). The fitted parameters corresponding to each aged
 155 mixes are given in Table 2 at reference temperature $T_{ref}=20^\circ\text{C}$.

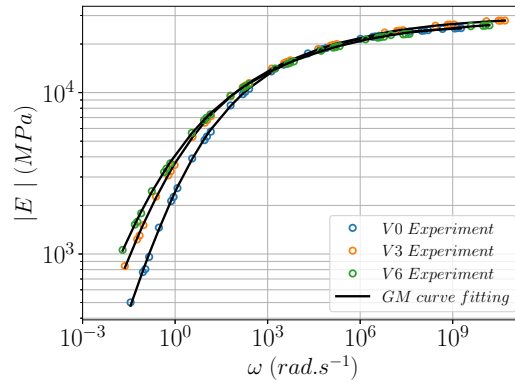


Fig. 2. Master curve of complex modulus of AC10 at $T_{ref}=20.^\circ\text{C}$

156 3.2 Crack growth in semi-circular bituminous mix specimen: experiment

157 3.2.1 Specimen preparation

158 The test specimens have been manufactured in laboratory and gyratory compacted. The ob-
 159 tained cylindrical samples have been sliced to obtain 50 mm thick cylinders with 150 mm in
 160 diameter then cut to obtain semi-circular specimens as schematized in Fig. 3. The specimens
 161 have been selected in such that their void contents are close to $(7\pm 0.6)\%$. At least four specimens
 162 have been used.

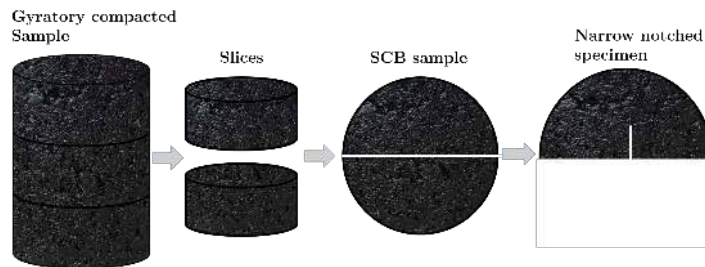


Fig. 3. SCB specimen's preparation

163 3.2.2 Experimental test procedure

164 The loading system consists of a three-point bending loading cell placed in UTICELL Type 650
 165 mechanical testing equipment. Before the test, the specimens are placed in a thermo-regulated
 166 chamber for 24 hours to reach temperature equilibrium before testing. They have been taken
 167 out of the chamber and placed in the loading cell rapidly to perform the test in less than a
 168 minute. The loading cell consists of two rigid roller supports and lateral thick metallic plates

169 which allows placing the specimen onto the bending fixture in order to avoid exentric loading.
170 Another roller support allows applying the load vertically. The test principle is represented in
171 Fig. 4. Lubrication is applied to the supports to mitigate friction during testing. The specimens



Fig. 4. SCB test device

171 diameters are $D=150$ mm, resulting in a span length of 120 mm between two cylindrical support
172 of 25 mm. The specimen thickness is $t=50$ mm. The notch length and width are respectively
173 $\ell = 10$ mm and $w_\ell = 0.35$ mm. Monotonic displacement rate is applied on the specimen until
174 failure. The force, the time and the vertical load-line displacement are recorded continuously
175 during the test by sensors integrated into the loading system. The tests are performed at 20°C
176 and a loading rate of $1\text{ mm}\cdot\text{min}^{-1}$ and $5\text{ mm}\cdot\text{min}^{-1}$ for the reference hot mix V0 and for the
177 aged mixes V3 and V6. Figure Fig. 5 shows typical force-displacement curves for an AC10 at 1
178 $\text{mm}\cdot\text{min}^{-1}$ and $5\text{ mm}\cdot\text{min}^{-1}$.

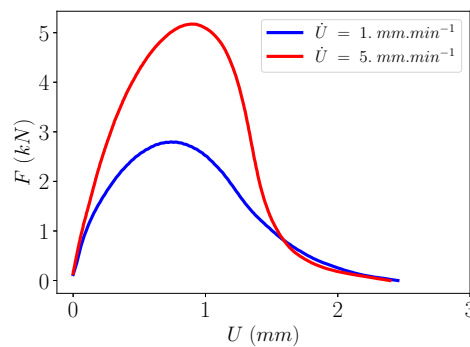


Fig. 5. Typical test results of AC10 at 20°C for two loading rate

179

180 4 Modelling of viscoelastic crack growth in SCB

181 4.1 Problem formulation

182 Let us consider the process of crack growth in opening mode in a notched two-dimensional
 183 viscoelastic body Ω subjected to prescribed displacement $\underline{U}^d(t)$ and force $\underline{T}^d(t)$ respectively on
 184 Γ_u and Γ_T (see Fig. 6). We assume that the crack does not appear on the edge part of $\Gamma_u \cup \Gamma_T$
 185 where either a displacement is prescribed or a force is non-zero. For the sake of simplicity, body
 186 forces and inertia effects are neglected and the small perturbation's hypotheses are considered.

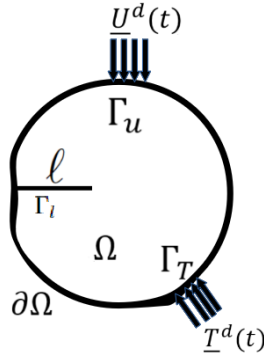


Fig. 6. Cracked viscoelastic domain

187 Based on these assumptions, the mechanical problem (\mathcal{P}) can be formulated as follows.

$$(\mathcal{P}) : \begin{cases} \operatorname{div} \underline{\sigma}(\underline{x}, t) = \underline{0} & \text{in } \Omega \\ \underline{u}(\underline{x}, t) = \underline{U}^d(t) & \text{on } \Gamma_u \\ \underline{\sigma}(\underline{x}, t) \cdot \underline{n} = \underline{T}^d(t) & \text{on } \Gamma_T \\ \underline{\sigma}(t) = \mathcal{R}_t \left[\begin{array}{c} \underline{\varepsilon}(\tau) \\ \tau = -\infty \end{array} \right] & \text{in } \Omega \\ \underline{\varepsilon}(\underline{u}) = \frac{1}{2} (\underline{\nabla} \underline{u} + {}^T \underline{\nabla} \underline{u}) & \text{in } \Omega \end{cases} \quad (2)$$

188 where $\mathcal{R}_t[\cdot]$ is termed *the modulus functional* [55], which maps the entire history strain to the
 189 stress value at the current time t . The Neumann-type boundary conditions in equation (2)
 190 indirectly assume that the stress vector is zero on the rest of the domain edge including the
 191 pre-crack.

192 The linear viscoelastic behaviour has historically been described by rheological models combining

193 springs and linear dampers, some of which provide an expression for the mathematical functional
 194 $\mathcal{R}_t[]$.

195 In most cases, instead of invoking boundary condition fields as $\underline{U}^d(t)$ and $\underline{T}^d(t)$, the loading
 196 mode may be described by a set of prescribed static parameters $(Q_1^d(t) \cdots Q_p^d(t))$ associated
 197 to kinematic variables $(q_1(t) \cdots q_p(t))$ and prescribed kinematic parameters $(q_{p+1}^d(t) \cdots q_n^d(t))$
 198 associated to static variables $(Q_{p+1}(t) \cdots Q_n(t))$ (see [56]). The whole set of loading parameters
 199 is thus given by a vector $\underline{\mathcal{C}}(t) = [Q_1^d(t), \cdots, Q_p^d(t), q_{p+1}^d(t), \cdots, q_n^d(t)]^T$ and the associated static
 200 and kinematic variables are defined by duality such that the power of external forces writes:

$$\begin{aligned} P_e &= \underline{Q} \cdot \dot{q} \\ &= \sum_{i=1}^p Q_i^d \dot{q}_i + \sum_{i=p+1}^n Q_i \dot{q}_i^d, \end{aligned} \quad (3)$$

201 where the whole static vector defining external forces contains both prescribed and resulting
 202 components $\underline{Q}(t) = [Q_1^d(t), \cdots, Q_p^d(t), Q_{p+1}(t), \cdots, Q_n(t)]^T$ and so is, by duality, the kinematic
 203 vector $q(t) = [q_1(t), \cdots, q_p(t), q_{p+1}^d(t), \cdots, q_n^d(t)]^T$.

204 4.2 Viscoelastic constitutive law for bituminous mixes

205 In this study, we chose to use the Generalized Maxwell (GM) model [54] to model the behaviour
 206 of bituminous mixtures in the time domain. It should be noted that the Huet [57] model and
 207 the 2S2P1D model [58] describe very well the bituminous mixtures behaviour. However, these
 208 models lead to store the loading history from the initial to the current time, which may be very
 209 costly in a finite element approach. As recalled in the sequel, the GM type of model is well
 210 adapted to an incremental formulation of the behaviour allowing to calculate a stress state at a
 211 given time step from the strain and stress states at the previous time. It is also interesting to
 212 note that the constitutive laws of bituminous mixtures under variable temperature have been
 213 proposed using Huet's [59] and generalized Maxwell's [41].

214 4.2.1 3D non-ageing isotropic linear viscoelastic formulation

215 The constitutive law of bituminous mixtures based on the GM model shown in Fig. 7 is given
 216 by the Stieltjes convolution integral below [55]:

$$\underline{\underline{\sigma}}(t) = \int_{t_0}^t \underline{\underline{R}}(t-t') : d\underline{\underline{\varepsilon}}(t'), \quad (4)$$

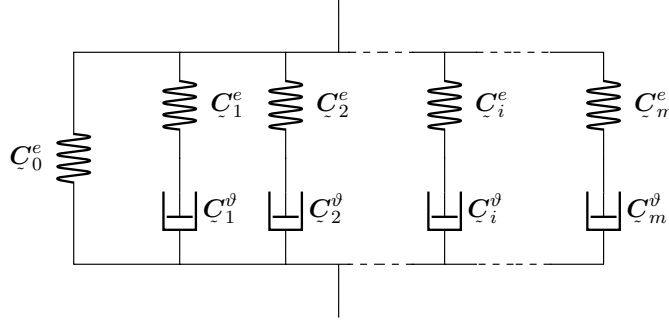


Fig. 7. Generalized Maxwell model (GM)

217 where

$$\underline{\mathbf{R}}(t) = \underline{\mathbf{C}}_0 + \sum_{i=1}^m \exp\left(-\frac{t}{\tau_i}\right) \underline{\mathbf{C}}_i. \quad (5)$$

218 In the case of a constant Poisson's ratio (which will be the case in this paper unless otherwise
219 stated), the relaxation tensor can be written in the following form:

$$\underline{\mathbf{R}}(t) = \bar{\mathbf{R}}(t) \underline{\mathbf{C}}^{el} \quad (6)$$

220 with :

$$\bar{\mathbf{R}}(t) = \frac{R(t)}{E^{el}}, \text{ where } R(t) = \left(E_0 + \sum_{i=1}^m E_i \exp\left(-\frac{t}{\tau_i}\right) \right) \text{ and } E^{el} = \sum_{i=0}^m E_i \quad (7)$$

$$\underline{\mathbf{C}}^{el} = \frac{E^{el}}{1-2\nu} \underline{\mathbf{J}} + \frac{E^{el}}{1+\nu} \underline{\mathbf{K}}$$

221 where $\underline{\mathbf{J}}_{ijkl} = \frac{1}{3} \delta_{ij} \delta_{kl}$ stands for the volumetric part components of the fourth-order unity
222 tensor $\underline{\mathbf{I}}$ ($\underline{\mathbf{I}}_{ijkl} = \frac{1}{2} (\delta_{ik} \delta_{jl} + \delta_{il} \delta_{jk})$), $\underline{\mathbf{K}} = \underline{\mathbf{I}} - \underline{\mathbf{J}}$ denotes the deviatoric part of $\underline{\mathbf{I}}$, $\bar{\mathbf{R}}$ and E^{el} are
223 respectively the normalized relaxation function and the instantaneous elastic modulus and E_i
224 the Young's modulus of the i th spring in the GM model.

225 4.2.2 Incremental formulation

226 Solving the viscoelastic problem formulated above (equation (4)) by the usual quadrature meth-
227 ods is less efficient and often leads to numerical instabilities. Indeed, these methods are stable
228 only if the computational time step is very small ($dt \ll \tau_i$), requiring several computational
229 steps in the case of very slow phenomena simulations. In order to overcome this problem, an

230 unconditionally stable algorithm, called exponential algorithm, developed by Z. P. Bazant [60],
 231 V. Šmilauer and Z. P. Bazant [61] is used. The physical time interval $T = [t_0, T_{max}]$ is dis-
 232 cretized into a sequence of intervals $\Delta_n t$ such that: $t_{n+1} = t_n + \Delta_n t$, T_{max} being the maximum
 233 simulation time. By rewriting equation (4) at time step t_{n+1} , exploiting the properties of the
 234 exponential function; splitting the time integral into two, one from t_0 to t_n where the solution
 235 is known and the other from t_n to t_{n+1} where the solution is unknown; and approximating the
 236 time derivative of the total strain tensor over the time interval $[t_n; t_{n+1}]$ by the following finite
 237 difference:

$$\frac{d\underline{\underline{\underline{\varepsilon}}}}{dt'}(t') \simeq \frac{\underline{\underline{\underline{\varepsilon}}}(t_{n+1}) - \underline{\underline{\underline{\varepsilon}}}(t_n)}{\Delta_n t} \quad \forall t' \in [t_n, t_{n+1}]; \quad (8)$$

238 it can be shown that equation (4) leads to an affine relationship between the total stress and
 239 total strain tensors:

$$\Delta_n \underline{\underline{\underline{\sigma}}} = \underline{\underline{\underline{C}}}(\Delta_n t) : \Delta_n \underline{\underline{\underline{\varepsilon}}} + \tilde{\underline{\underline{\underline{\sigma}}}}(t_n) \quad (9)$$

where

$$\underline{\underline{\underline{C}}}(\Delta_n t) = \underline{\underline{\underline{C}}}_0 + \sum_{i=1}^m \frac{\tau_i}{\Delta_n t} \left(1 - \exp\left(-\frac{\Delta_n t}{\tau_i}\right) \right) \underline{\underline{\underline{C}}}_i, \quad (10a)$$

$$\tilde{\underline{\underline{\underline{\sigma}}}}(t_n) = \sum_{i=1}^m \left(\exp\left(-\frac{\Delta_n t}{\tau_i}\right) - 1 \right) \underline{\underline{\underline{\sigma}}}_i(t_n), \quad (10b)$$

$$\underline{\underline{\underline{\sigma}}}_i(t_{n+1}) = \exp\left(-\frac{\Delta_n t}{\tau_i}\right) \underline{\underline{\underline{\sigma}}}_i(t_n) + \frac{\tau_i}{\Delta_n t} \left(1 - \exp\left(-\frac{\Delta_n t}{\tau_i}\right) \right) \underline{\underline{\underline{C}}}_i : \Delta_n \underline{\underline{\underline{\varepsilon}}} \quad (10c)$$

240 The formula (9) appears as a pseudo-elastic law with pre-stress where only the current and
 241 previously calculated fields are involved. The pre-stress $\tilde{\underline{\underline{\underline{\sigma}}}}(t_n)$ accounts for all past stress history
 242 updated and stored at the end of each time increment.

243 Considering a non-ageing isotropic linear viscoelastic behaviour, in which case the fourth-order
 244 tensor $\underline{\underline{\underline{C}}}$ can be completely defined by two scalar viscoelastic kernels $\lambda(t)$ and $\mu(t)$ [55], formula
 245 (9) can be expressed as follows:

$$\Delta_n \underline{\underline{\underline{\sigma}}} = \lambda(\Delta_n t) \text{tr}(\Delta_n \underline{\underline{\underline{\varepsilon}}}) \underline{\underline{\underline{I}}} + 2\mu(\Delta_n t) \Delta_n \underline{\underline{\underline{\varepsilon}}} + \tilde{\underline{\underline{\underline{\sigma}}}}(t_n), \quad (11)$$

where:

$$\lambda(\Delta nt) = \lambda_0 + \sum_{i=1}^m \frac{\tau_i}{\Delta nt} \left(1 - \exp\left(-\frac{\Delta nt}{\tau_i}\right)\right) \lambda_i, \quad (12a)$$

$$\mu(\Delta nt) = \mu_0 + \sum_{i=1}^m \frac{\tau_i}{\Delta nt} \left(1 - \exp\left(-\frac{\Delta nt}{\tau_i}\right)\right) \mu_i, \quad (12b)$$

In the case of a constant Poisson's ratio, the equations (12a) and (12b) reduce to:

$$\lambda(\Delta nt) = \frac{\nu E^{el}}{(1+\nu)(1-2\nu)} \bar{R}^*(\Delta nt) \quad (13a)$$

$$\mu(\Delta nt) = \frac{E^{el}}{2(1+\nu)} \bar{R}^*(\Delta nt) \quad (13b)$$

246 with:

$$\bar{R}^*(t) = \frac{R^*(t)}{E^{el}}, \text{ where } R^*(t) = \left(E_0 + \sum_{i=1}^m \frac{\tau_i}{t} \left(1 - \exp\left(-\frac{t}{\tau_i}\right)\right) E_i \right) \quad (14)$$

247 The incremental constitutive law given by equation (11) can be introduced in a finite ele-
 248 ment discretization to obtain solutions to complex viscoelastic problems, as will be discussed in
 249 sections 4.2.3 and 4.3.

250 4.2.3 3D modelling of the 2-point loads cantilever beam test

251 For FE modelling, the tested asphalt concrete AC10 is supposed to be homogeneous and
 252 isotropic, which is a common assumption. In addition, recent work by A. Gudmarsson et al. [62]
 253 and F. Allou et al. [63] on the measurement of the complex Poisson's ratio of bituminous mixes
 254 showed very low values of its imaginary part compared to its real part. The complexity of the
 255 measurement and the dispersion of the obtained results are additional factors to be considered.
 256 For these reasons, it is considered that the Poisson's ratio is constant at 0.35 (a value often used
 257 in structural calculations). For clarity, only the reference material V0 is chosen for the modelling
 258 and its properties are given in Table 2. The geometry and boundary conditions are presented
 259 in Fig. 1.

260 The mechanical problem, at each time increment, is then governed by the following system of

Table 2: Fitted GM model parameters at 20°C

V0		V3		V6	
E_i (MPa)	τ_i (s)	E_i (MPa)	τ_i (s)	E_i (MPa)	τ_i (s)
2.20×10^3	4.13×10^{-9}	2.41×10^3	4.12×10^{-10}	2.58×10^3	1.69×10^{-10}
2.49×10^3	5.16×10^{-7}	1.12×10^3	7.64×10^{-9}	1.57×10^3	9.27×10^{-9}
3.04×10^3	1.33×10^{-5}	2.57×10^3	1.34×10^{-7}	1.68×10^3	1.82×10^{-7}
2.99×10^3	1.65×10^{-4}	2.58×10^3	2.85×10^{-6}	9.39×10^2	1.38×10^{-6}
2.93×10^3	1.33×10^{-3}	7.22×10^2	8.01×10^{-5}	1.65×10^3	6.47×10^{-6}
1.88×10^3	6.35×10^{-3}	2.36×10^3	2.13×10^{-5}	2.02×10^3	5.35×10^{-5}
2.26×10^3	2.29×10^{-2}	2.98×10^3	2.82×10^{-4}	8.35×10^2	2.64×10^{-4}
3.74×10^2	2.34×10^{-2}	3.48×10^3	2.72×10^{-3}	1.42×10^3	5.32×10^{-4}
2.74×10^3	1.39×10^{-1}	1.17×10^3	1.38×10^{-2}	1.37×10^3	1.89×10^{-3}
1.57×10^2	5.77×10^{-1}	1.41×10^3	3.00×10^{-2}	1.56×10^3	5.34×10^{-3}
3.81×10^2	6.02×10^{-1}	1.29×10^3	4.73×10^{-2}	2.20×10^3	1.99×10^{-2}
1.19×10^3	9.35×10^{-1}	2.49×10^3	2.33×10^{-1}	2.57×10^3	1.09×10^{-1}
6.67×10^2	3.03×10^0	1.76×10^3	1.17×10^0	2.19×10^3	6.96×10^{-1}
5.54×10^2	1.19×10^1	1.40×10^3	8.03×10^0	1.44×10^3	4.48×10^0
2.48×10^2	8.21×10^1	1.67×10^2	7.67×10^1	8.17×10^2	2.86×10^1
1.07×10^2	–	3.37×10^2	7.76×10^1	2.38×10^2	1.48×10^2
		1.42×10^2	–	3.11×10^2	2.57×10^2
				2.09×10^2	–

261 equations:

$$(\mathcal{P}^{n+1})_s : \left\{ \begin{array}{l}
 \operatorname{div} \underline{\underline{\boldsymbol{\sigma}}}(\underline{\mathbf{x}}, t_{n+1}) = \underline{\mathbf{0}} \quad \text{in } \Omega \\
 \underline{\mathbf{u}}(\underline{\mathbf{x}}, t_{n+1}) = \begin{cases} \underline{U}^d(\underline{\mathbf{x}}, t_{n+1}) + \underline{U}^\omega(\underline{\mathbf{x}}, t_{n+1}) & \text{on } S_{sup} \\ \underline{\mathbf{0}} & \text{on } S_{inf} \end{cases} \\
 \underline{U}^d(\underline{\mathbf{x}}, t_{n+1}) = U^d \sin(2\pi f t_{n+1}) \underline{\mathbf{e}}_x \\
 \underline{U}^\omega(\underline{\mathbf{x}}, t_{n+1}) = \underline{\boldsymbol{\omega}} \wedge \underline{GM} \quad \text{where} \quad \begin{cases} \underline{\boldsymbol{\omega}}(t_{n+1}) = \omega(t_{n+1}) \underline{\mathbf{e}}_z \\ \underline{GM} = \left(x - \frac{b}{2}\right) \underline{\mathbf{e}}_x + \left(z - \frac{b}{2}\right) \underline{\mathbf{e}}_z \end{cases} \\
 \underline{\underline{\boldsymbol{\sigma}}}(\underline{\mathbf{x}}, t_{n+1}) \cdot \underline{\mathbf{n}} = \underline{\mathbf{0}} \quad \text{on } \partial\Omega \setminus \{S_{sup} \cup S_{inf}\} \\
 \underline{\underline{\boldsymbol{\sigma}}}(\underline{\mathbf{x}}, t_{n+1}) = \underline{\underline{\boldsymbol{\sigma}}}(\underline{\mathbf{x}}, t_n) + \lambda(\Delta_n t) \operatorname{tr}(\Delta_n \underline{\underline{\boldsymbol{\varepsilon}}}(\underline{\mathbf{x}})) \underline{\mathbf{I}} \\
 \quad + 2\mu(\Delta_n t) \Delta_n \underline{\underline{\boldsymbol{\varepsilon}}}(\underline{\mathbf{x}}) + \tilde{\underline{\underline{\boldsymbol{\sigma}}}}(\underline{\mathbf{x}}, t_n) \quad \text{in } \Omega \\
 \underline{\underline{\boldsymbol{\varepsilon}}}(\underline{\mathbf{u}}(\underline{\mathbf{x}}, t_{n+1})) = \frac{1}{2} (\underline{\nabla} \underline{\mathbf{u}}(\underline{\mathbf{x}}, t_{n+1}) + {}^T \underline{\nabla} \underline{\mathbf{u}}(\underline{\mathbf{x}}, t_{n+1})) \quad \text{in } \Omega
 \end{array} \right. \tag{15}$$

262 where the amplitude of the imposed displacement cycle U^d and the frequency f are input data

263 while the rotation ω is an additional unknown of the problem to solve. The unknown variable
 264 ω is a simple scalar that is not related to a finite element description insofar as it does not refer
 265 to a degree of freedom attached to any single node. Indeed, it accounts for the fact that the
 266 upper surface of the sample undergoes a rigid body motion and its value has to be adjusted so
 267 that the overall bending moment around \underline{e}_z acting over the upper surface cancels out.

268 Note that the kinematic boundary condition on S_{sup} in its general form is written: $\underline{u}(\underline{x}, t_{n+1}) =$
 269 $\underline{U}^d(\underline{x}, t_{n+1}) + \underline{\omega} \wedge \underline{GM}$ where the components of the displacement vector of the center of the
 270 plate \underline{U}^d and the rotation vector $\underline{\omega}$ define either prescribed parameters of the problem or La-
 271 grange multipliers associated to their corresponding dual static parameters (force or moment
 272 components). Here, for symmetry reasons, it is anticipated that both $\underline{U}^d \cdot \underline{e}_z = 0$ and $\underline{R} \cdot \underline{e}_z = 0$
 273 (\underline{R} is the resultant reaction on S_{sup}) as well as both $\underline{\omega} \cdot \underline{e}_x = 0$ and $\underline{M} \cdot \underline{e}_x = 0$ (\underline{M} is the resultant
 274 moment vector at the center G of S_{sup}) so that prescribed kinematic parameters ($\underline{U}^d \cdot \underline{e}_z = 0$
 275 and $\underline{\omega} \cdot \underline{e}_x = 0$) are preferably considered in the problem. In addition, since the loading system is
 276 screwed to the plate by applying a sinusoidal translation along x (thus $\underline{U}^d \cdot \underline{e}_x = U^d \sin(2\pi f t_{n+1})$
 277 and $\underline{R} \cdot \underline{e}_y = 0$), the plate prevents a torsional rotation, thus $\underline{\omega} \cdot \underline{e}_y = 0$. Finally, instead of impos-
 278 ing a null vertical force $\underline{R} \cdot \underline{e}_y = 0$, it is assumed by analogy with the study of a pure bending
 279 problem in classical structure mechanics that $\underline{U}^d \cdot \underline{e}_y = 0$ (null elongation of the neutral axis in
 280 absence of normal force) and we therefore obtain the kinematic boundary condition on S_{sup}
 281 given in equation (15). Finally, as mentioned above, the only (unknown) kinematic Lagrange
 282 multiplier is $\omega = \underline{\omega} \cdot \underline{e}_z$ associated with a prescribed null moment around \underline{e}_z .

283 In the sequel, we adopt the following notation: $g(\underline{x}, t_n) = g^n$ for any volume or surface field g .
 284 For the approximation of (\mathcal{P}_s^{n+1}) by means of the finite element method (e.g. [64]), the weak
 285 formulation of this problem is needed. It is given by:

286 Find the quadruplet $(\underline{u}^{n+1}, \underline{p}^{n+1}, \underline{q}^{n+1}, \omega^{n+1})$ in $\mathcal{U} \times \mathcal{U}'(S_{inf}) \times \mathcal{U}'(S_{sup}) \times \mathbb{R}$:

$$\left\{ \begin{array}{l}
 \int_{\Omega} [\lambda(\Delta_n t) \text{tr}(\underline{\underline{\epsilon}}^{n+1}) \underline{\underline{I}} + 2\mu(\Delta_n t) \underline{\underline{\epsilon}}^{n+1}] : \underline{\underline{\epsilon}}(\delta \underline{u}) \, d\Omega - \int_{S_{inf}} \underline{p}^{n+1} \cdot \delta \underline{u} \, dS - \int_{S_{sup}} \underline{q}^{n+1} \cdot \delta \underline{u} \, dS = \\
 \int_{\Omega} [\lambda(\Delta_n t) \text{tr}(\underline{\underline{\epsilon}}^n) \underline{\underline{I}} + 2\mu(\Delta_n t) \underline{\underline{\epsilon}}^n - \underline{\underline{\sigma}}^n - \underline{\underline{\tilde{\sigma}}}^n] : \underline{\underline{\epsilon}}(\delta \underline{u}) \, d\Omega \quad \forall \delta \underline{u} \in \mathcal{U} \\
 \int_{S_{inf}} \underline{u}^{n+1} \cdot \delta \underline{p} \, dS = 0 \quad \forall \delta \underline{p} \in \mathcal{U}'(S_{inf}) \\
 \int_{S_{sup}} (\underline{u}^{n+1} - \underline{\omega}^{n+1} \wedge \underline{GM}) \cdot \delta \underline{q} \, dS = \int_{S_{sup}} \underline{U}^{d,n+1} \delta \underline{q} \, dS \quad \forall \delta \underline{q} \in \mathcal{U}'(S_{sup}) \\
 \int_{S_{sup}} (\underline{GM} \wedge \underline{q}^{n+1}) \cdot \underline{e}_z \delta \omega \, dS = 0 \quad \forall \delta \omega \in \mathbb{R}
 \end{array} \right. \quad (16)$$

287 where \mathcal{U} denotes the set of admissible displacement fields (set of smooth continuous displacement
 288 fields on Ω), $\mathcal{U}'(S_{inf})$ (respectively $\mathcal{U}'(S_{sup})$) is the set of admissible forces (Lagrange multipliers
 289 space) on S_{inf} (respectively S_{sup}) defined by duality with respect to \mathcal{U} , i.e. such that the integrals
 290 on S_{sup} (respectively S_{sup}) in (16) are defined for any kinematic field belonging to \mathcal{U} .

291 The Lagrangian corresponding to (16) is as follows:

$$\begin{aligned}
 \mathcal{L}(\underline{u}^{n+1}, \underline{p}^{n+1}, \underline{q}^{n+1}, \omega^{n+1}) &= \frac{1}{2} \int_{\Omega} \underline{\underline{\epsilon}}(\underline{u}^{n+1}) : \left((3\lambda(\Delta_n t) + 2\mu(\Delta_n t)) \underline{\underline{J}} + 2\mu(\Delta_n t) \underline{\underline{K}} \right) : \underline{\underline{\epsilon}}(\underline{u}^{n+1}) \, d\Omega + \\
 &\int_{\Omega} \left[\underline{\underline{\sigma}}^n + \underline{\underline{\tilde{\sigma}}}^n - \lambda(\Delta_n t) \text{tr}(\underline{\underline{\epsilon}}^n) \underline{\underline{I}} - 2\mu(\Delta_n t) \underline{\underline{\epsilon}}^n \right] : \underline{\underline{\epsilon}}(\underline{u}^{n+1}) \, d\Omega - \int_{S_{inf}} \underline{p}^{n+1} \cdot \underline{u}^{n+1} \, dS - \\
 &\int_{S_{sup}} \underline{q}^{n+1} \cdot \left(\underline{u}^{n+1} - \underline{U}^{d,n+1} - \underline{\omega}^{n+1} \wedge \underline{GM} \right) \, dS
 \end{aligned} \quad (17)$$

292 The incremental viscoelastic problem equation (16) is implemented in a generic finite element library
 293 GetFEM [65]. The resolution algorithm is described by the flowchart in Fig. 8. Fig. 9a to Fig. 9d compare
 294 the numerical results with the experimental results of force versus displacement values at a temperature
 295 of 20°C and for four loading frequencies (from 3Hz to 30Hz). A fairly good agreement between these
 296 results can be observed. This validates the implemented incremental method.

297 The developed program is used in the following section to model the crack propagation during the SCB
 298 test.

299 4.3 Crack growth modelling in a viscoelastic medium: Energetic approach

300 When the cracking and loading states are known, the problem (\mathcal{P}) can be solved using the approach
 301 presented above. All that remains is to define a crack growth criterion. For this purpose, the energetic
 302 approach to model mode I crack propagation in a **two-dimensional** viscoelastic medium adapted from the
 303 work of S. T. Nguyen et al. [68] is presented.

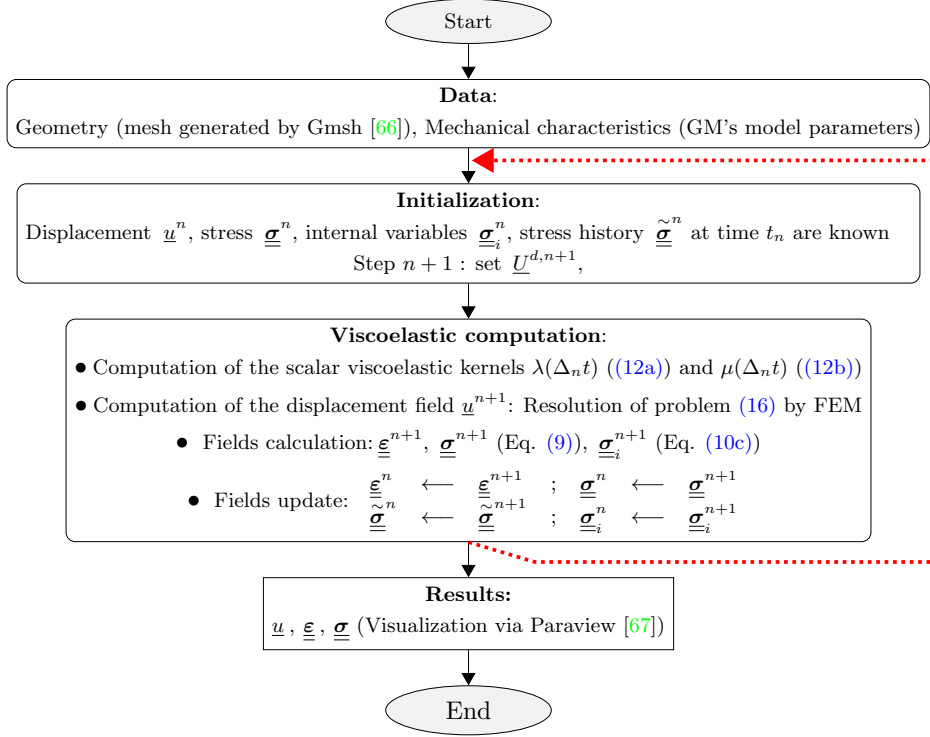


Fig. 8. Flowchart of the program implemented in GetFEM

304 4.3.1 Thermodynamic analysis

305 The developments of this section are much inspired by [68]. In the case of an isothermal and uniform
306 transformation, the first two principles of thermodynamics lead a dissipation given by:

$$\mathcal{D} = P_e - \dot{\mathcal{W}} \geq 0 \quad (18)$$

307 where \mathcal{W} denotes the elastic strain energy stored in the system and P_e the power of external forces.

308 For a finer analysis of the inequality (18), an alternative formulation of the state equation (4) is proposed
309 as follows:

$$\underline{\underline{\sigma}}(t) = \underline{\underline{C}}^{el} : (\underline{\underline{\epsilon}} - \underline{\underline{\epsilon}}^\vartheta) \quad (19)$$

310 where:

$$\underline{\underline{\epsilon}}^\vartheta(t) = \underline{\underline{C}}^{el^{-1}} : \sum_{i=1}^m \underline{\underline{C}}_i : \underline{\underline{\epsilon}}_i^\vartheta(t) \quad (20)$$

311 $\underline{\underline{\epsilon}}_i^\vartheta$ represents the viscous strain in the i^{th} Maxwell element and $\underline{\underline{\epsilon}}^\vartheta$ can be interpreted as an eigenstrain
312 tensor.

313 In order to analyze more in details the dissipation (18), it is worth observing first that the elastic
314 strain energy is a function of the loading parameters, which have been synthesized by the vector $\underline{\underline{\mathcal{L}}}(t) =$

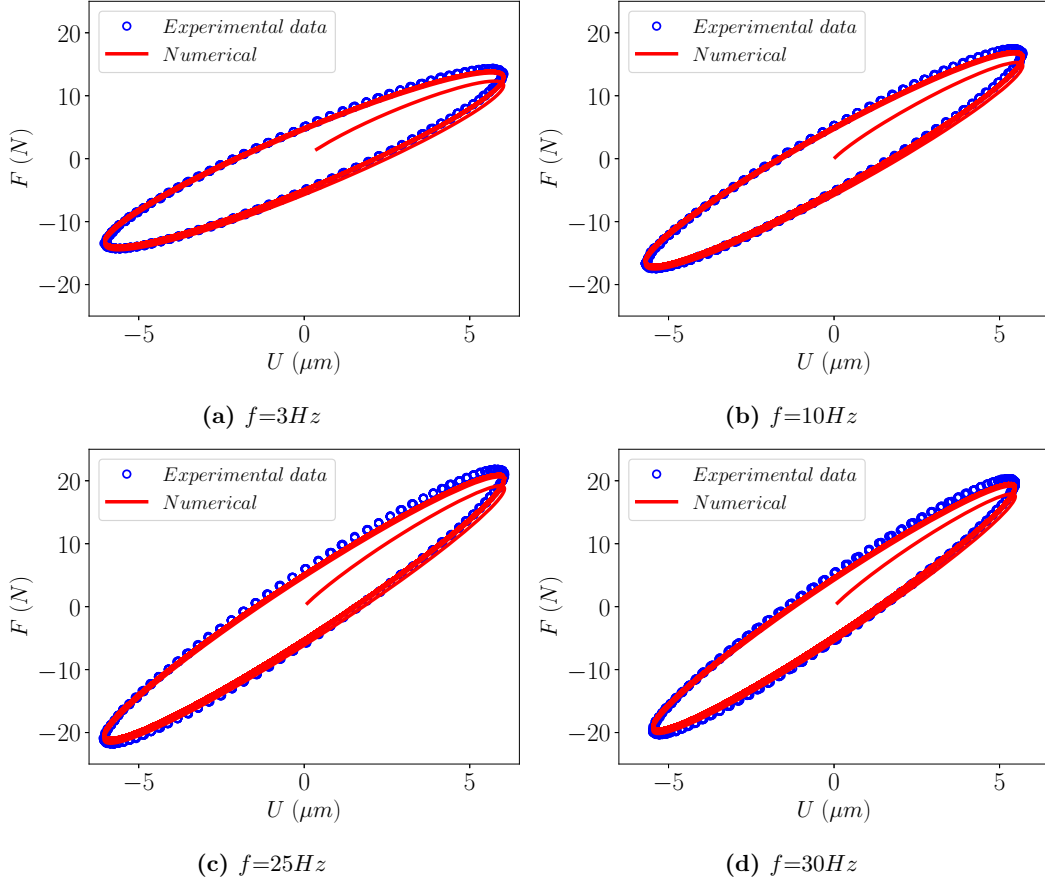


Fig. 9. Simulation of the 2-points cantilever beam test: Evolution of numerical and experimental forces versus the imposed displacement at 20°C

315 $[Q_1^d(t), \dots, Q_p^d(t), q_{p+1}^d(t), \dots, q_n^d(t)]^T$, of the crack length $\ell(t)$ as well as the internal variables fields
 316 $\{\underline{\underline{\epsilon}}_i^\vartheta(t), i = 1, m\}$ and writes:

$$\mathcal{W}(\underline{\underline{\mathcal{L}}}, \ell, \{\underline{\underline{\epsilon}}_i^\vartheta\}) = \frac{1}{2} \int_{\Omega} \left[\underline{\underline{\epsilon}} : \underline{\underline{C}}_0 : \underline{\underline{\epsilon}} + \sum_{i=1}^m (\underline{\underline{\epsilon}} - \underline{\underline{\epsilon}}_i^\vartheta) : \underline{\underline{C}}_i : (\underline{\underline{\epsilon}} - \underline{\underline{\epsilon}}_i^\vartheta) \right] d\Omega \quad (21)$$

317 Besides, the power of external forces has been expressed in (3) and it is clear that any evolution leaving
 318 ℓ and the internal fields $\underline{\underline{\epsilon}}_i^\vartheta$ unchanged while allowing changes of the loading parameters $\underline{\underline{\mathcal{L}}}$ corresponds
 319 to an elastic evolution which does not induce any dissipation by definition:

$$\left. \frac{\partial \mathcal{W}}{\partial \underline{\underline{\mathcal{L}}}_i} \right|_{(\ell, \{\underline{\underline{\epsilon}}_i^\vartheta\})} \dot{\underline{\underline{\mathcal{L}}}}_i = \underline{\underline{Q}} \cdot \left. \frac{\partial \underline{\underline{q}}}{\partial \underline{\underline{\mathcal{L}}}_i} \right|_{(\ell, \{\underline{\underline{\epsilon}}_i^\vartheta\})} \dot{\underline{\underline{\mathcal{L}}}}_i \quad (22)$$

320 It follows that the remaining terms of the dissipation stem only from the evolutions of ℓ and $\underline{\underline{\epsilon}}_i^\vartheta$:

$$\mathcal{D} = -\frac{\partial}{\partial \ell} (\mathcal{W} - \phi) \Big|_{(\underline{\mathcal{C}}, \{\underline{\underline{\epsilon}}_i^\vartheta\})} \dot{\ell} - \frac{\partial}{\partial \{\underline{\underline{\epsilon}}_i^\vartheta\}} (\mathcal{W} - \phi) \Big|_{(\underline{\mathcal{C}}, \ell)} \bullet \{\underline{\underline{\epsilon}}_i^\vartheta\} \quad (23)$$

321 where \bullet denotes a scalar product in the sense of functions (i.e. written under the form of integrals since
322 $\underline{\underline{\epsilon}}_i^\vartheta$ are actual fields) and ϕ is the work of given external forces written as:

$$\phi = \sum_{i=1}^p Q_i^d q_i, \quad (24)$$

323 Note that ϕ in (24) does not include terms corresponding to given kinematic parameters since the latter
324 disappear in the power of external forces for evolutions of \underline{q} at fixed $\underline{\mathcal{C}}$.

325 Then, by designating by $P = \mathcal{W} - \phi$ the potential energy and by denoting:

$$\mathcal{D}_\vartheta = -\frac{\partial P}{\partial \{\underline{\underline{\epsilon}}_i^\vartheta\}} \Big|_{(\underline{\mathcal{C}}, \ell)} \bullet \{\underline{\underline{\epsilon}}_i^\vartheta\}, \quad (25)$$

326 the energy dissipated by the crack growth appears to be the complementary of the viscous term in the
327 total dissipation:

$$\mathcal{D} - \mathcal{D}_\vartheta = -\frac{\partial P}{\partial \ell} \Big|_{(\underline{\mathcal{C}}, \{\underline{\underline{\epsilon}}_i^\vartheta\})} \dot{\ell}, \quad (26)$$

328 where $\mathcal{G}_\vartheta = -\frac{\partial P}{\partial \ell} \Big|_{(\underline{\mathcal{C}}, \{\underline{\underline{\epsilon}}_i^\vartheta\})}$ represents the thermodynamic force associated with the crack growth.

329 The specificity of this energy release rate relies on the fact that its value depends on the viscous strain
330 field which, in turn, depends on the loading history $\underline{\mathcal{C}}(t)$ and the evolution of the crack length $\ell(t)$ over
331 the considered time interval.

332 By following the assumption formulated in the work of S. T. Nguyen et al. [68] that the dissipation
333 related to crack growth is proportional to the propagation rate $\dot{\ell}$, we have:

$$\mathcal{D} - \mathcal{D}_\vartheta = \mathcal{R} \dot{\ell} \quad (27)$$

334 where \mathcal{R} can be interpreted as a resistance strength that must be overcome for crack growth to occur.

335 By merging equations (26) and (27), the crack growth criterion can be formulated as follows:

$$\begin{cases} \mathcal{G}_\vartheta < \mathcal{R} & \implies \dot{\ell} = 0 \quad (\text{no crack growth}) \\ \mathcal{G}_\vartheta = \mathcal{R} & \implies \dot{\ell} \geq 0 \quad (\text{possible crack growth}) \end{cases} \quad (28)$$

336 The main difficulty in the study of cracking in dissipative media lies in the determination of these two
337 terms and will be discussed in the next section.

338 4.3.2 Energy released rate computation

339 The energy release rate expression \mathcal{G}_ϑ highlights a partial derivative with respect to the crack length ℓ
 340 of the potential energy. This partial derivative is approximated, from two crack states ℓ and $\ell + \delta\ell$, by a
 341 difference between the potential energies calculated for these two states. This is a small elastic variation
 342 related to an increase $\delta\ell$ of the crack length since the derivation is performed by blocking the viscous
 343 strains and the imposed loading. To demonstrate this, let us consider the alternative formulation of the
 344 behaviour law equation (19) for the real state of cracking corresponding to length ℓ :

$$\underline{\underline{\sigma}}(\ell) = \underline{\underline{C}}^{el} : \underline{\underline{\varepsilon}}(\ell) + \underline{\underline{\sigma}}^r(\ell) \quad (29)$$

where:

$$\underline{\underline{\sigma}}^r(\ell) = -\underline{\underline{C}}^{el} : \underline{\underline{\varepsilon}}^\vartheta(\ell)$$

345 The auxiliary elementary calculation for the fictitious cracking state $\ell + \delta\ell$ is instead governed by the
 346 following state equation:

$$\underline{\underline{\sigma}}(\ell + \delta\ell) = \underline{\underline{C}}^{el} : \underline{\underline{\varepsilon}}(\ell + \delta\ell) + \underline{\underline{\sigma}}^r(\ell) \quad (30)$$

347 By combining equations (29) and (30), the variation in stress between these two cracking states is thus
 348 given by equation (31):

$$\Delta_\ell \underline{\underline{\sigma}} = \underline{\underline{C}}^{el} : \Delta_\ell \underline{\underline{\varepsilon}}, \quad (31)$$

349 which represents the instantaneous response of viscoelastic behaviour. The approximated expression for
 350 the energy release rate is given by:

$$\begin{aligned} \mathcal{G}_\vartheta &= -\frac{\delta P}{\delta\ell} \\ &= -\frac{(\mathcal{W} - \phi)(\ell + \delta\ell) - (\mathcal{W} - \phi)(\ell)}{\delta\ell} \end{aligned} \quad (32)$$

351 In summary, the energy release rate at time t characterised by the cracking state ℓ is calculated in two
 352 steps:

353 Step (1): Solving the problem (\mathcal{P}) with the viscoelastic behaviour law and deriving $\mathcal{W}(\ell)$ by the equation
 354 (33) and the work of given external forces by (34).

$$\mathcal{W}(\ell) = \frac{1}{2} \int_{\Omega} \left[\underline{\underline{\varepsilon}}(\ell) : \underline{\underline{C}}_0 : \underline{\underline{\varepsilon}}(\ell) + \sum_{i=1}^m \left(\underline{\underline{\varepsilon}}(\ell) - \underline{\underline{\varepsilon}}_i^\vartheta(\ell) \right) : \underline{\underline{C}}_i : \left(\underline{\underline{\varepsilon}}(\ell) - \underline{\underline{\varepsilon}}_i^\vartheta(\ell) \right) \right] d\Omega \quad (33)$$

355

$$\phi(\ell) = \sum_{i=1}^p Q_i^d q_i(\ell) \quad (34)$$

356 where $q_i(\ell)$ is the kinematic parameter resulting from solving the problem (\mathcal{P}).

357 Step (2): Solving an auxiliary elastic problem characterised by the cracking state $\ell + \delta\ell$ subjected to the

358 same loading with the behaviour law (31) (where the viscous strain fields are those calculated in step (1))
 359 and deriving $\mathcal{W}(\ell + \delta\ell)$ by the equation (35) and the work of given external forces by (36).

$$\mathcal{W}(\ell + \delta\ell) = \frac{1}{2} \int_{\Omega} \left[\underline{\underline{\epsilon}}(\ell + \delta\ell) : \underline{\underline{C}}_0 : \underline{\underline{\epsilon}}(\ell + \delta\ell) + \sum_{i=1}^m \left(\underline{\underline{\epsilon}}(\ell + \delta\ell) - \underline{\underline{\epsilon}}_i^{\vartheta}(\ell) \right) : \underline{\underline{C}}_i : \left(\underline{\underline{\epsilon}}(\ell + \delta\ell) - \underline{\underline{\epsilon}}_i^{\vartheta}(\ell) \right) \right] d\Omega \quad (35)$$

360

$$\phi(\ell + \delta\ell) = \sum_{i=1}^p Q_i^d q_i(\ell + \delta\ell) \quad (36)$$

361 where $q(\ell + \delta\ell)$ is the kinematic parameter obtained by solving the auxiliary problem.

362 Note that $[Q_1^d, \dots, Q_p^d]$ is the set of prescribed static parameters and that when the boundary conditions
 363 are only defined by prescribed kinematic parameters q^d then $\phi = 0$, this will be the case in the sequel.

364 In the following section, the validity of this approach will be examined.

365 5 Numerical applications

366 The energy approach, developed in this paper, is implemented in the generic finite element library Get-
 367 FEM [65]. In order to validate this approach, we consider a **two-dimensional** viscoelastic notched semi-
 368 circular geometry of 75 mm in radius with a central notch of depth ℓ under three-point bending: a
 369 vertical downward displacement $q^d(t)$ is imposed over a segment on top of the sample (see Fig. 10) For
 370 the numerical simulations, the loading and support surfaces are set to 10^{-3} mm and the notch width is
 371 assumed to be zero. **We assume a plane state of strain and we reason per unit thickness unless explicitly**
 372 **stated otherwise.** In this configuration, the crack propagates in an opening mode. The focus of this
 373 section is to numerically implement the energy approach introduced in section 4.3 for dealing with crack
 374 growth in asphalt mixtures.

375 5.1 Notched semi-circular beam without crack growth

376 5.1.1 Numerical computation energy release rate validation

377 This section is devoted to the comparison between numerical and semi-analytical energy restitution rates
 378 under two different loading conditions defined by a prescribed displacement, i.e. a unique kinematic
 379 loading parameter $\mathcal{C}(t) = q^d(t)$ so that the static variable Q as well as the residual kinematic variable
 380 q^{ϑ} and the overall compliance S^e are here scalar. To simplify the analytical approach, the homogeneous
 381 isotropic material is assumed to behave as a non-ageing linear viscoelastic material with a constant
 382 Poisson's ratio ($\nu = 0.35$) and we consider the plane strain framework.

383 **Following the same reasoning as the energetic approach in microporoelasticity, it can be shown that the**
 384 **elastic strain energy can be decomposed into a recoverable energy \mathcal{W}^e and a frozen energy \mathcal{W}^{res} (see A**

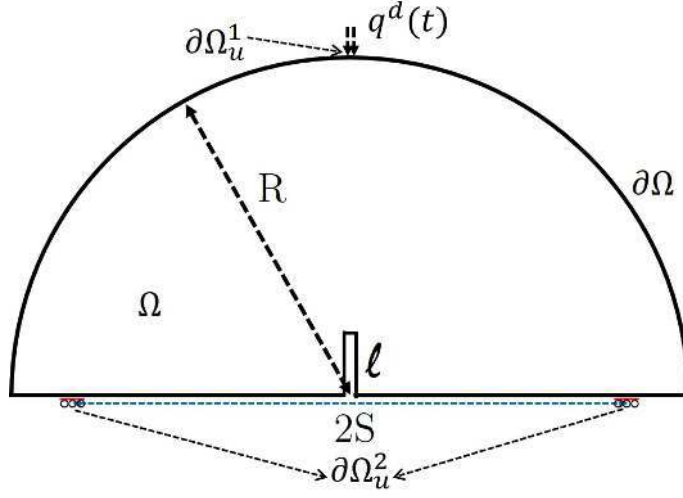


Fig. 10. Semi-circular geometry $R= 75$ mm, $S= 60$ mm

385 for more details):

$$\mathcal{W} = \mathcal{W}^e + \mathcal{W}^{res} \quad (37)$$

386 where:

$$\mathcal{W}^e = \frac{1}{2} (\underline{q} - \underline{q}^\vartheta) \cdot \underline{\underline{S}}^e \cdot (\underline{q} - \underline{q}^\vartheta), \quad (38)$$

387 $\underline{\underline{S}}^e$ represents the instantaneous global compliance of the system Ω , \underline{q} denotes the global displacement
388 vector and \underline{q}^ϑ the residual displacement parameter. We have:

$$\underline{q} = \underline{\underline{S}}^e \cdot \underline{Q} + \underline{q}^\vartheta \quad (39)$$

389 If we consider the alternative form of the elastic strain energy equation (37), the energy release rate
390 equation (32) takes the form:

$$\mathcal{G}_\vartheta(t) = -\frac{1}{2} \frac{d}{d\ell} \left(\frac{1}{S^e} \right) (q - q^\vartheta)^2 + \frac{1}{S^e} \frac{\partial q^\vartheta}{\partial \ell} \Big|_{(q^d, \{\underline{\underline{\epsilon}}_i^\vartheta\})} (q - q^\vartheta) - \frac{\partial \mathcal{W}^{res}}{\partial \ell} \Big|_{(q^d, \{\underline{\underline{\epsilon}}_i^\vartheta\})} \quad (40)$$

391 Assuming that the contributions of the derivatives of the residual energy and kinematic parameter are
392 negligible (S. T. Nguyen et al.[68] have rigorously proved this assumption in the case of 1D heterogeneous
393 structures and we propose to verify it in the 3D case), one gets the following reduced form of \mathcal{G}_ϑ :

$$\mathcal{G}_\vartheta(t) \simeq -\frac{1}{2} \frac{d}{d\ell} \left(\frac{1}{S^e} \right) (q - q^\vartheta)^2 \quad (41)$$

394 Substituting equation (39) into equation (41), this one becomes:

$$\mathcal{G}_\vartheta(t) \simeq \frac{1}{2} \frac{dS^e}{d\ell} Q^2 \quad (42)$$

395 Under the stated assumptions especially constant Poisson's ratio and by application of the principle of
 396 superposition, the evolutionary solution defined by the kinematic loading parameter $q^d(t)$ is given by
 397 equation (43) [55]:

$$Q(t) = K^e \int_{t_0}^t \bar{R}(t-t') dq^d(t') \quad (43)$$

398 where $K^e = \frac{1}{S^e}$ and \bar{R} is given by equation (7) Substituting equations (43) into the equation (42), the
 399 energy release rate thus takes the form:

$$\mathcal{G}_\vartheta(t) \simeq -\frac{1}{2} \frac{dK^e}{d\ell} \left(\int_{t_0}^t \bar{R}(t-t') dq^d(t') \right)^2 \quad (44)$$

400 Let us discuss in more detail the two types of loading envisaged:

401 • A constant displacement is prescribed $\underline{u}(t) = -U_0 \mathcal{H}(t) \underline{e}_y$ ($q^d(t) = U_0 \mathcal{H}(t)$ and $\mathcal{H}(t)$ is the Heaviside
 402 function). In this case, \mathcal{G}_ϑ is given by:

$$\mathcal{G}_\vartheta(t) \simeq -\frac{1}{2} \frac{dK^e}{d\ell} (\bar{R}(t) U_0)^2 \quad (45)$$

403 The asymptotic energy release rate ($t \rightarrow \infty$) yields:

$$\mathcal{G}_\vartheta(t) \xrightarrow{t \rightarrow \infty} \mathcal{G}_\vartheta^\infty = -\frac{1}{2} \frac{dK^e}{d\ell} \left(\frac{E_0 U_0}{E^{el}} \right)^2 \quad (46)$$

404 • A displacement is prescribed in the form $\underline{u}(t) = -\dot{U} t \mathcal{H}(t) \underline{e}_y$ ($q^d(t) = \dot{U} t \mathcal{H}(t)$ and \dot{U} stands for the
 405 prescribed constant displacement rate). The energy release rate is expressed as follows:

$$\mathcal{G}_\vartheta(t) \simeq -\frac{1}{2} \frac{dK^e}{d\ell} (\bar{R}^*(t) \dot{U} t)^2 \quad (47)$$

406 where \bar{R}^* is given by equation (14) The asymptotic energy release rate ($t \rightarrow \infty$) yields:

$$\mathcal{G}_\vartheta(t) \xrightarrow{t \rightarrow \infty} \mathcal{G}_\vartheta^\infty = \begin{cases} -\frac{1}{2} \frac{dK^e}{d\ell} \left(\frac{E_0 \dot{U} t}{E^{el}} \right)^2 \rightarrow \infty & \text{if } E_0 \neq 0 \\ -\frac{1}{2} \frac{dK^e}{d\ell} \left(\frac{\dot{U} \sum_{i=1}^m E_i \tau_i}{E^{el}} \right)^2 & \text{if } E_0 = 0 \end{cases} \quad (48)$$

407 It can be shown that $K^e(\ell) = E^{el} K_1^e(\ell)$ where $K_1^e(\ell)$ is the effective stiffness per unit modulus and imposed
 408 loading. A numerical determination of $K_1^e(\ell)$ may be easily achieved by standard finite elements. So, the
 409 expressions giving \mathcal{G}_ϑ are semi-analytical in the sense that they are time-based analytical whereas K^e is
 410 obtained by solving an elasticity problem using the finite element approach.

411 The energy release rate can also be defined:

$$\mathcal{G}_1^e(\ell) = -\frac{1}{2} \frac{dK^e}{d\ell} \quad (49)$$

412 Fig. 11 show the effective stiffness and energy release rate calculated for different crack lengths in the
 413 semi-circular bending test subjected to a unit displacement and for a unit modulus. The fitted curve

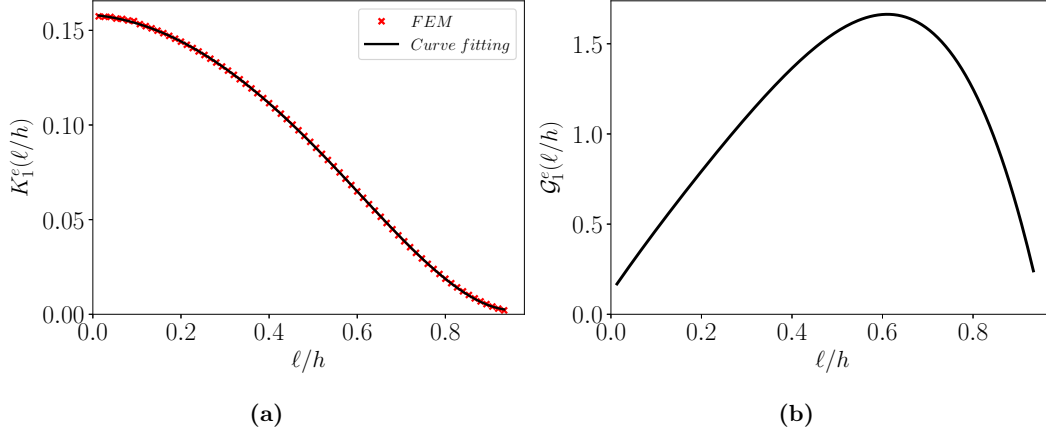


Fig. 11. Effective stiffness and energy release rate calculated for different crack lengths in the SCB test subjected to a unit displacement

413
 414 equation is given by:

$$K_1^e(x) = 0.209x^5 - 0.18x^4 + 0.11x^3 - 0.275x^2 - 0.018x + 0.158 \quad \text{where } x = \frac{\ell}{R} \quad (50)$$

415 In order to verify the accuracy of the numerical technique for determining \mathcal{G}_ϑ , numerical resolutions are
 416 performed by simulating a semi-circular bending test without crack growth. The viscoelastic properties of
 417 the material is given in the Table 2 (material named V0 in the table). The numerical results are compared
 418 with our semi-analytical developments (equation (45) and equation (47)). Fig. 12a shows the evolution
 419 of the viscoelastic energy release rate over time for a constant imposed displacement and in Fig. 12b, the
 420 variation of \mathcal{G}_ϑ is plotted for a constant imposed displacement rate. A very good agreement between the
 421 numerical and semi-analytical solutions can be observed. Under prescribed constant displacement, the
 422 decrease of \mathcal{G}_ϑ as a function of time can also be remarked. This fact reflects the relaxation phenomenon
 423 due to the viscoelastic effect. Figures Fig. 13a and Fig. 13b show the asymptotic responses of the energy
 424 release rate for the two considered loading cases taking $E_0 = 0$.

425 5.1.2 Progressive or brutal propagation

426 The notion of stability in the cracking process is very crucial in Griffith's theory [44]. It is proposed here
 427 to deal with this question in the viscoelastic framework.

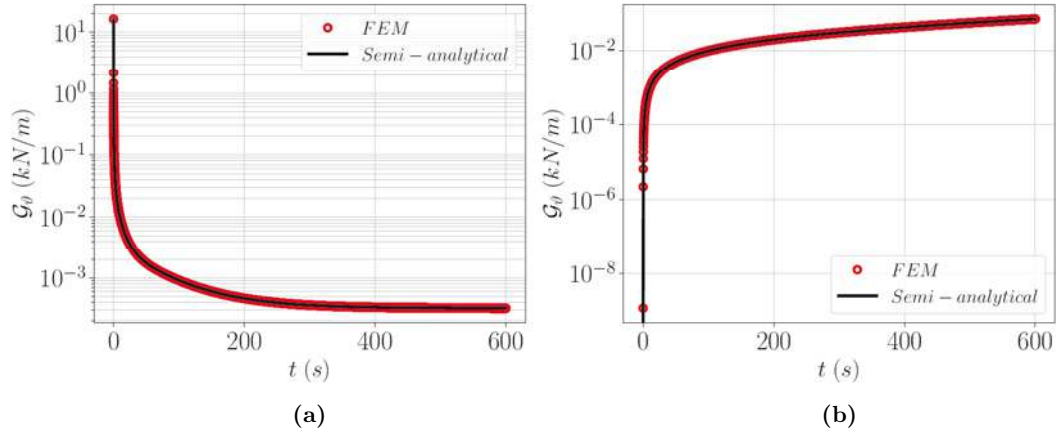


Fig. 12. Evolution of numerical and approximate energy release rates at 20°C without crack growth: Fig. 12a $U_0 = 1$ mm, Fig. 12b $\dot{U} = 1$ $\text{mm}\cdot\text{min}^{-1}$ ($E_0 = 0$)

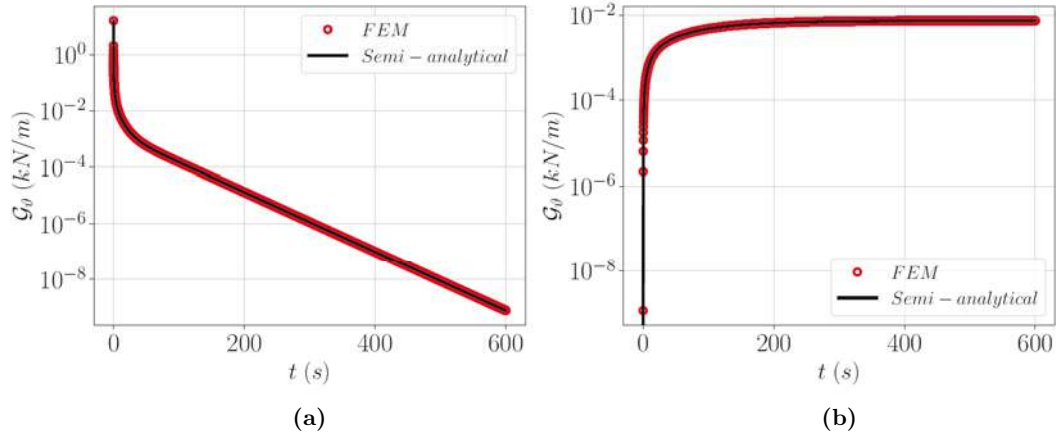


Fig. 13. Evolution of numerical and approximate asymptotic energy release rates at 20°C without crack growth for $E_0 = 0$: Fig. 13a $U_0 = 1$ mm, Fig. 13b $\dot{U} = 1$ $\text{mm}\cdot\text{min}^{-1}$

428 According to Griffith, the stability criterion can be summarised as:

$$\mathcal{G}_\theta \leq \mathcal{R} \quad (51)$$

429 This implies that, at fixed loading and viscous strains, the variation of the ratio between the energy
 430 release rate and the crack growth resistance allows to identify the two evolutions of the crack. If this
 431 ratio is decreasing with respect to the crack length, then the propagation is stable. To increase cracking,
 432 the loading must be increased. However, if this ratio is increasing, then the crack evolution is no longer
 433 controlled. If the loading is maintained, the crack will abruptly propagate. In a new state of cracking,
 434 under this same loading, the crack may or may not stabilise.

435 Let's illustrate this by reconsidering the three-point bending test on the semi-circular configuration (see

436 Fig. 10) subjected to a constant loading rate \dot{U} . By assuming that the crack growth resistance is constant
 437 ($\mathcal{R} = G_c$) and using relation (47), the ratio $\frac{\mathcal{G}_\vartheta}{\mathcal{R}}$ gives:

$$\frac{\mathcal{G}_\vartheta}{\mathcal{R}} = \mathcal{G}_1^e(\ell) \left(\sqrt{\frac{E^{el}}{G_c} R^*(t) \dot{U} t} \right)^2 \quad (52)$$

438 At a fixed loading, the term $\left(\sqrt{\frac{E^{el}}{G_c} R^*(t) \dot{U} t} \right)^2$ being constant, the variation of the ratio $\frac{\mathcal{G}_\vartheta}{\mathcal{R}}$ as a function
 439 of ℓ is therefore imposed by $\mathcal{G}_1^e(\ell)$. Under the indicated assumptions, and referring to Fig. 11b, the two
 440 situations are distinguished: a phase where the crack propagates abruptly (increasing ratio) followed by
 441 a phase of stable propagation (decreasing ratio).

442 For bituminous mixes, it is well known that the non-linear zone ahead of the crack tip is relatively large
 443 compared to the dimensions of the structure and, therefore, its effect cannot be neglected as in linear
 444 fracture mechanics. X. Li et al. [69] studied the effect of process fracture zone in bituminous mixtures.
 445 They used an eight-channel acoustic emissions system to monitor the fracture process zone during the
 446 crack propagation test using an SCB geometry in crack opening mode. They showed that the size of
 447 the process zone varies with temperature (the process zone at low temperatures has a greater length
 448 than that at high temperatures, while there is no significant difference in width). The authors explain
 449 this phenomenon by the viscous behaviour of bituminous mixtures at high temperatures. To take this
 450 non-linearity into account, \mathcal{R} -curve models have been developed and successfully used to treat cracking
 451 problems in wood [70, 71] and concrete [72]. The shape of the \mathcal{R} -curve can play a very crucial role in the
 452 stability of the crack growth as will be seen in the next section.

453 5.2 Numerical crack growth algorithm

454 In order to implement crack growth modelling in finite element software, we present an algorithm which
 455 allows finite element coupling between the viscoelastic behaviour and the crack propagation criterion.
 456 This paper is limited to the case of a straight crack growth. Thus, the crack path is a priori known. Then
 457 we apply the technique known as "unbuttoning nodes" to treat numerically the crack growth of a length
 458 $\delta\ell$ when the criterion equation (28) is verified.

459 In practice, for a virtual crack advance $\delta\ell$, we look for the loading q^d (prescribed displacement for clarity)
 460 for which the criterion is fulfilled, i.e. $\mathcal{G}_\vartheta(\ell, q^d) - \mathcal{R} = 0$. The first step is to determine the loading q_c^d
 461 that would propagate the crack by a length $\delta\ell$. For a constant loading rate (\dot{q}^d), the loading parameter
 462 q^d can be written in the incremental form equation (53).

$$q_{n+1}^d = q_n^d + \dot{q}^d \Delta_n t \quad ; \quad \Delta_n t = t_{n+1} - t_n \quad (53)$$

463 where q_{n+1}^d is determined by looking for the time increment $\Delta_n t$ that ensures a propagation of $\ell + \delta\ell$.
 464 Thus for each q_i^d , the energy release rate \mathcal{G}_ϑ^i is computed. Let us denote by q_k^d , the loading at which

465 $\mathcal{G}_\vartheta^k > \mathcal{R}$ for the first time. The criterion is then unfulfilled and it becomes necessary to determine the
 466 displacement $q_k^{d,j}$, close to q_k^d such that :

$$f(q_k^{d,j}) = \mathcal{G}_\vartheta(q_k^{d,j}) - \mathcal{R} = 0 \quad (54)$$

467 The equation (54) is resolved using the secant method which provides the following recurrence relation:

$$q_k^{d,j+1} = q_k^{d,j} - \frac{q_k^{d,j} - q_k^{d,j-1}}{f(q_k^{d,j}) - f(q_k^{d,j-1})} f(q_k^{d,j}) \quad (55)$$

468 with $q_k^{d,0} = q_{k-1}$ and $q_k^{d,1} = q_k$.

469 Or:

$$\Delta_k^{j+1}t = \Delta_k^j t - \frac{\Delta_k^j t - \Delta_k^{j-1} t}{f(q_k^{d,j}) - f(q_k^{d,j-1})} f(q_k^{d,j}) \quad (56)$$

470 with $\Delta_k^0 t = \Delta_{k-1} t$ and $\Delta_k^1 t = \Delta_k t$.

471 The calculation is interrupted when:

$$|f(q_k^{d,j})| \leq \varepsilon_\ell \quad (57)$$

472 where ε_ℓ represents the approximation error committed on the accuracy of the numerical solution. Once
 473 the displacement q_c^d is determined (with a tolerance ε_ℓ), the crack is allowed to propagate by a length
 474 $\delta\ell$. If the sequence $(q_k^{d,j})$ diverges, which would mean that the reference solution q_k^d is very far from the
 475 zero of the function f , then it is necessary to decrease the loading step (and thus the time increment)
 476 and restart the calculation. The algorithm for solving the crack growth problem as described above is
 477 summarised by the scheme Fig. 14.

478 6 Comparison between numerical model and experiments

479 In the framework of numerical simulation, the three-point bending problem is implemented using the
 480 finite element library Getfem++ [65] and compared to the experimental results.

481 6.1 Numerical simulation of the semi-circular bending test

482 The objective of this section is to evaluate the relevance of the model by comparing the response of the
 483 numerical simulations with the experimental observations. The standard crack propagation test on a
 484 semi-circular asphalt mixtures specimen is again considered (see Fig. 10). The dimensions used in the
 485 simulations are given in section 3.2.2. In order to conduct the 2D numerical simulations of Mode I crack
 486 propagation, the following assumptions are adopted:

- 487 • Isotropic linear viscoelastic material, homogeneous, whose viscoelastic parameters of the MG model
 488 are given in Table 2 for the reference material and for the materials subjected to the 3 and 6 days
 489 ageing times.

490 • Constant Poisson's ratio $\nu = 0.35$

491 In this study, the formulation of the \mathcal{R} curve developed by Bathias [73, 74] is used:

$$\mathcal{R} = [\alpha + \beta (\ell - \ell_0)] \frac{1}{\lambda} \quad (58)$$

492 where α , β and λ are constants. In the crack growth process, it is shown that the total internal energy,
 493 which is equal to the external work done, is composed of the recoverable strain energy, the energy
 494 dissipated by the fracture process and the energy dissipated due to the viscoelasticity of the material.
 495 The latter term is difficult to identify in the energy analysis balance. The force-displacement method
 496 used by Morel et al. [70] cannot be applied to determine the \mathcal{R} -curve in viscoelastic media in which the
 497 viscous dissipation may not be neglected. For this reason, the \mathcal{R} -curve parameters are determined by
 498 an optimisation approach. These parameters can be determined by calibrating the peak load and the
 499 area under the force-displacement curve between the numerical simulations and the experimental results.
 The \mathcal{R} -values and summarized in Table 3. Figures 15a and 15b illustrate the SCB test simulation results

Table 3: Optimised parameters of the \mathcal{R} -curve expression

	\dot{U}	α	β	λ
V0	1	6.41×10^{-2}	1.43×10^3	1.31
	5	8.81×10^{-2}	4.04×10^3	1.20
V3	1	6.4×10^{-2}	9.49×10^3	1.77
	5	6.53×10^{-2}	4.23×10^3	1.10
V6	1	8.40×10^{-2}	1.44×10^3	1.06
	5	6.37×10^{-2}	5.02×10^3	1.05

500

501 obtained by the finite element method.

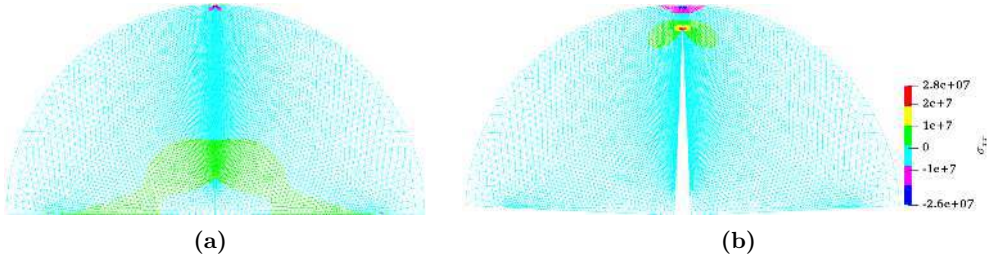


Fig. 15. Crack paths in the numerical simulation for the reference material V0: 15a start of crack growth, 15b end of crack growth

502 Figures Fig. 16a and Fig. 16b show the numerical variations in crack growth during loading for the two
 503 loading rates ($1 \text{ mm}\cdot\text{min}^{-1}$ and $5 \text{ mm}\cdot\text{min}^{-1}$) and for the reference material (V0) and the materials
 504 conditioned to long-term ageing for 3 and 6 days (V3 and V6).

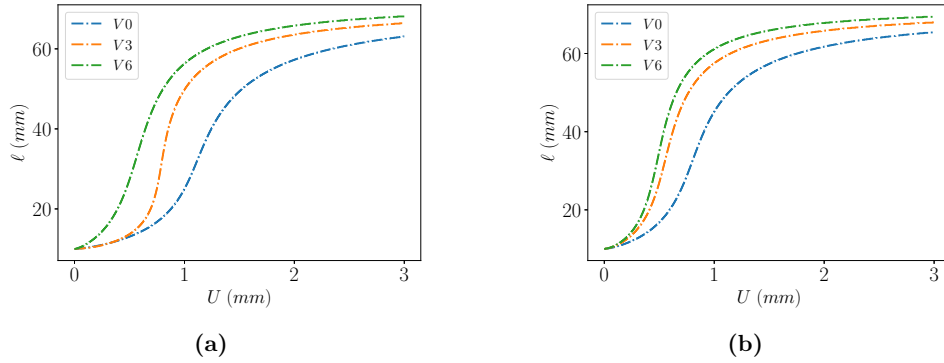


Fig. 16. Evolution of crack length as a function of applied displacement at 0, 3 and 6 day ageing times: **16a** for $1 \text{ mm}\cdot\text{min}^{-1}$, **16b** for $5 \text{ mm}\cdot\text{min}^{-1}$

505 It is noted that as the loading rate increases, the crack propagates more quickly. This result is in
 506 agreement with Nguyen et al. conclusions [68]. These figures also show that oxidative ageing leads to
 507 an increase in crack growth rate which becomes more pronounced at $5 \text{ mm}\cdot\text{min}^{-1}$ (see Fig. 16). Figures
 508 Fig. 17a and Fig. 17b show the evolution of the \mathcal{R} -curve versus the crack length $(\ell - \ell_0)$. An increase in
 509 \mathcal{R} is observed as a function of the loading rate and the ageing time. The reason for this may be that the
 510 bituminous mix exhibits less viscous and less ductile behaviour and the development of defects becomes
 511 significant with increasing ageing time.

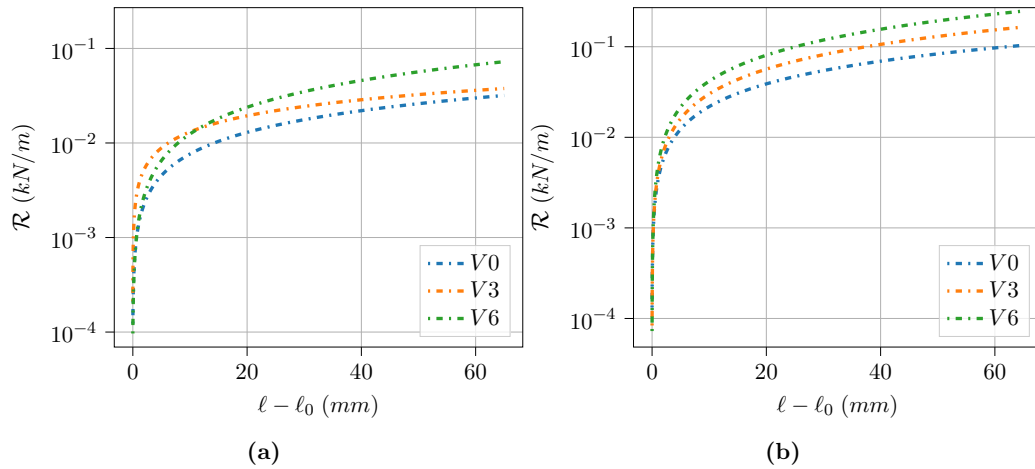


Fig. 17. \mathcal{R} -curve evolution as a function of crack length at 0, 3 and 6 day ageing times: **17a** for $1 \text{ mm}\cdot\text{min}^{-1}$, **17b** for $5 \text{ mm}\cdot\text{min}^{-1}$

512 The numerical simulation results, in terms of load-displacement, are validated by comparing them with
 513 the experimental results, as shown in Fig. 18a-18f. In a general way, it can be seen that, for each ageing
 514 time and loading rate, the numerical and experimental load-displacement curves are consistent for the
 515 selected specimens. Nevertheless, we note that the accuracy of the model decreases when the material

516 is more exposed to ageing and under a higher loading rate. We think that these discrepancies between
517 the numerical simulations and the experimental results may be related to the increase in the size of the
518 process zone (with the ageing duration) limiting its consideration by the \mathcal{R} -curve model. An attempt can
519 be made to overcome this problem by using another form of \mathcal{R} -curve expression like the one proposed
520 by J. Lemaitre and J-L Chaboche [75] which make the \mathcal{R} -expression depend on the crack velocity. Thus,
521 the proposed crack growth criterion, which assumes that the part of the dissipation due to the creation
522 of a crack surface is proportional to the crack velocity, needs to be reconsidered. However, when several
523 specimens are tested, account should be taken of the heterogeneity between specimens which may lead
524 to a dispersion of the test results.

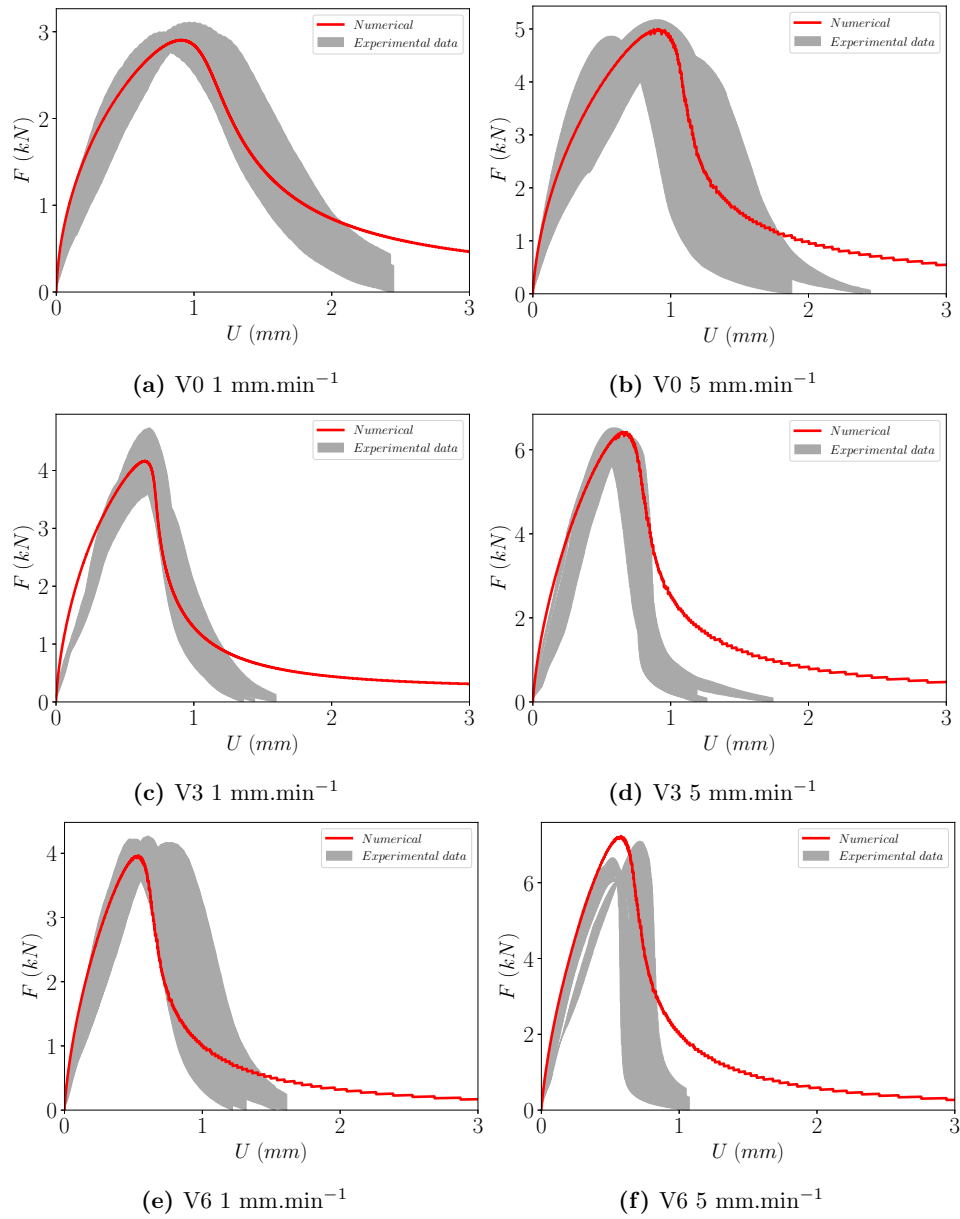


Fig. 18. Comparison of force-displacement curves between numerical and experimental results at 20°C

525 7 Conclusion

526 The development of an energetic crack growth model for viscoelastic media is presented in this paper.
 527 This model is intended for the study of the fracture of bituminous mixes. For this purpose, the analogue
 528 Generalized Maxwell model is used to represent the 3D viscoelastic discretized behaviour of the latter.

529 This model is implemented in a finite element software GetFem. At the same time, The discretized law
530 is applied to the analysis of a part of the complex modulus tests (2PB-PR) carried out in the laboratory
531 on bituminous mixes at room temperature (20°C) and under 4 loading frequencies ranging from 3 to
532 40 Hz. A good agreement between the experimental and numerical results has been obtained, which
533 led to the validation of the implemented incremental law. Next, a mechanical model of crack growth
534 in dissipative media has been described. For the first time, analytical expressions of the crack growth
535 driving force as a function of the type of loading is formulated for a semi-circular geometry subjected
536 to three-point bending. This analytical strategy provided a basis for the validation of the numerical
537 determination of the driving force implemented in a Getfem FE software. Finally, the developed model
538 is applied for the very first time to simulate the SCB test performed in the laboratory on bituminous
539 mixtures at 20°C and under two imposed displacement rates (1 mm.min⁻¹ and 5 mm.min⁻¹) for the
540 reference and aged mixtures. By dependently setting the \mathcal{R} -curve parameters on the loading rate and
541 ageing, a good match between the experimental and numerical results in terms of the force-displacement
542 curve is obtained for all investigated test conditions, thus proving the ability of our model to simulate the
543 ductile crack growth in bituminous mixes under monotonic loading. **Taking into account the dependence
544 of the cracking resistance \mathcal{R} on a parameter that is a function of the loading rate (a hardening parameter
545 defined by the residual displacement rates \underline{q}^ϑ by analogy with the hardening laws in plasticity) can thus
546 improve the predictive potential of the model and will be investigated in future.**

547 Aiming at refining the model by representing the heterogeneous composition of the herein considered
548 materials, while remaining within reasonable numerical simulation times and further approximating the
549 real material composition (heterogeneity), evolutionary multi-model formulations, articulated within the
550 multi-scale and multi-model Arlequin framework, initiated in [76, 77] and where the overall behaviour
551 can be modelled by homogenisation schemes extended to the viscoelastic medium (e.g. [78, 79]) will be
552 conducted and implemented in the future.

553 Acknowledgements

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555 out the semi-circular bending crack propagation tests.

556 A Decomposition of elastic strain energy

557 Following the same reasoning as the energetic approach in microporoelasticity (see [68, 80]), the local
558 problem of (\mathcal{P}) at fixed crack length $\ell(t)$ with the alternative constitutive law equation (19) can be
559 separated in two loading cases:

560 1. The first load case we consider is the generalized forces $\underline{\mathcal{C}}$, while pre-strain field $\underline{\underline{\epsilon}}^\vartheta$ is zero:

$$\left(\mathcal{P}'\right) : \begin{cases} \operatorname{div}\underline{\underline{\sigma}}' = \underline{0} & \text{in } \Omega \\ \underline{\underline{\sigma}}' = \underline{\underline{C}}^{el} : \underline{\underline{\epsilon}}' & \text{in } \Omega \\ \underline{\underline{\sigma}}' & \text{is statically admissible with } \underline{Q} \end{cases} \quad (59)$$

561 The solution of the evolution problem $\left(\mathcal{P}'\right)$ is given by:

$$\begin{aligned} \underline{\underline{\epsilon}}' &= \underline{\underline{A}} \cdot \underline{Q} \\ \underline{\underline{\sigma}}' &= \underline{\underline{B}} \cdot \underline{Q} \\ \underline{q}' &= \underline{\underline{S}}^e(\ell) : \underline{Q} \end{aligned} \quad (60)$$

562 where $\underline{\underline{A}}$ and $\underline{\underline{B}}$ are respectively third order strain and stress concentration tensors fields. $\underline{\underline{S}}^e$
563 represents the instantaneous global compliance of the system Ω which is symmetric by virtue of
564 the Maxwell-Betti reciprocal theorem.

565 2. The second load case corresponds to a loading defined by the eigenstrain $\underline{\underline{\epsilon}}^\vartheta$ field (actually defined
566 by all the fields $\underline{\underline{\epsilon}}_i^\vartheta$) only:

$$\left(\mathcal{P}''\right) : \begin{cases} \operatorname{div}\underline{\underline{\sigma}}'' = \underline{0} & \text{in } \Omega \\ \underline{\underline{\sigma}}'' = \underline{\underline{C}}^{el} : (\underline{\underline{\epsilon}}'' - \underline{\underline{\epsilon}}^\vartheta) & \text{in } \Omega \\ \underline{\underline{\sigma}}'' & \text{is statically admissible with } \underline{Q} = \underline{0} \end{cases} \quad (61)$$

567 Let us denote by \underline{q}^ϑ the residual displacement parameter, solution of the problem $\left(\mathcal{P}''\right)$. According
568 to the virtual work theorem one has:

$$\begin{aligned} \underline{q}^\vartheta \cdot \hat{Q} &= \int_{\Omega} (\underline{\underline{\epsilon}}^\vartheta + \underline{\underline{C}}^{el^{-1}} : \underline{\underline{\sigma}}'') : \hat{\underline{\underline{\sigma}}} \, d\Omega \\ &= \left(\int_{\Omega} \underline{\underline{\epsilon}}^\vartheta : \underline{\underline{B}} \, d\Omega \right) \cdot \hat{Q} + \int_{\Omega} \underline{\underline{\sigma}}'' : \hat{\underline{\underline{\epsilon}}} \, d\Omega \quad \forall \hat{Q} \quad (62) \\ \underline{q}^\vartheta \cdot \hat{Q} &= \left(\int_{\Omega} \underline{\underline{\epsilon}}^\vartheta : \underline{\underline{B}} \, d\Omega \right) \cdot \hat{Q}, \text{ because } \underline{\underline{\sigma}}'' \text{ is self-equilibrating stress} \end{aligned}$$

569 Therefore we have:

$$\underline{q}^\vartheta = \int_{\Omega} \underline{\underline{\epsilon}}^\vartheta : \underline{\underline{B}} \, d\Omega \quad (63)$$

570 The global vector \underline{q} can then be written:

$$\underline{q} = \underline{\underline{S}}^e \cdot \underline{Q} + \underline{q}^\vartheta \quad (64)$$

571 On the other hand, the elastic strain energy can be written as follows:

$$\begin{aligned}
\mathcal{W} &= \sum_{i=0}^m \int_{\Omega} \frac{1}{2} (\underline{\boldsymbol{\sigma}}'_i + \underline{\boldsymbol{\sigma}}''_i) : \mathbf{C}_i^{-1} : (\underline{\boldsymbol{\sigma}}'_i + \underline{\boldsymbol{\sigma}}''_i) \, d\Omega \\
&= \sum_{i=0}^m \int_{\Omega} \frac{1}{2} \underline{\boldsymbol{\sigma}}'_i : \underline{\boldsymbol{\varepsilon}}'_i \, d\Omega + \sum_{i=0}^m \int_{\Omega} \underline{\boldsymbol{\sigma}}''_i : \underline{\boldsymbol{\varepsilon}}'_i \, d\Omega + \sum_{i=0}^m \int_{\Omega} \frac{1}{2} \underline{\boldsymbol{\sigma}}''_i : \mathbf{C}_i^{-1} : \underline{\boldsymbol{\sigma}}''_i \, d\Omega \\
&= \sum_{i=0}^m \int_{\Omega} \frac{1}{2} \underline{\boldsymbol{\sigma}}'_i : \underline{\boldsymbol{\varepsilon}}'_i \, d\Omega + \sum_{i=0}^m \int_{\Omega} \frac{1}{2} \underline{\boldsymbol{\sigma}}''_i : \mathbf{C}_i^{-1} : \underline{\boldsymbol{\sigma}}''_i \, d\Omega, \text{ because } \underline{\boldsymbol{\sigma}}'' \text{ is statically admissible} \\
&\quad \text{with } \underline{\mathbf{Q}}=0 \text{ and } \underline{\boldsymbol{\varepsilon}}' \text{ is geometrically compatible}
\end{aligned} \tag{65}$$

$$\mathcal{W} = \mathcal{W}^e + \mathcal{W}^{res}$$

572 where \mathcal{W}^e denotes the recoverable energy and \mathcal{W}^{res} the frozen energy that remains after unloading:

$$\begin{aligned}
\mathcal{W}^e &= \sum_{i=0}^m \int_{\Omega} \frac{1}{2} \underline{\boldsymbol{\sigma}}'_i : \underline{\boldsymbol{\varepsilon}}'_i \, d\Omega \\
&= \frac{1}{2} \underline{\mathbf{q}}' : \underline{\mathbf{Q}} \\
\overline{\mathcal{W}}^e &= \frac{1}{2} (\underline{\mathbf{q}} - \underline{\mathbf{q}}^\vartheta) : \underline{\mathbf{S}}^{e-1} : (\underline{\mathbf{q}} - \underline{\mathbf{q}}^\vartheta)
\end{aligned} \tag{66}$$

573 and

$$\begin{aligned}
\mathcal{W}^{res} &= \sum_{i=0}^m \int_{\Omega} \frac{1}{2} \underline{\boldsymbol{\sigma}}''_i : \mathbf{C}_i^{-1} : \underline{\boldsymbol{\sigma}}''_i \, d\Omega \\
&= \sum_{i=0}^m \int_{\Omega} \frac{1}{2} (\underline{\boldsymbol{\varepsilon}}''_i - \underline{\boldsymbol{\varepsilon}}^\vartheta_i) : \underline{\boldsymbol{\sigma}}''_i \, d\Omega \\
\mathcal{W}^{res} &= - \sum_{i=0}^m \int_{\Omega} \frac{1}{2} \underline{\boldsymbol{\varepsilon}}^\vartheta_i : \underline{\boldsymbol{\sigma}}''_i \, d\Omega
\end{aligned} \tag{67}$$

574 It is well demonstrated by equations (63) and (67) that the residual displacement field $\underline{\mathbf{q}}^\vartheta$ and the frozen
575 energy \mathcal{W}^{res} both depend on the viscous strains field.

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Highlights

- The viscoelastic crack growth criterion is reviewed and extended to the behaviour of bituminous mixtures.
- A new analytical expression of the energy release rate is presented and applied in detail to the semi-circular bending configuration.
- Numerical simulations of the semi-circular bending test are performed.
- Comparison between numerical and experimental results is analyzed using a new approach combining the proposed model and the R-curve model.

This study is not subject to any conflict of interest.

