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Marion Hingant, Stéphanie Mallarino, Egle Conforto, Emmanuel Dubillot, Pierrick Barbier, et al.. Artificial weathering of plastics used in oyster farming. *Science of the Total Environment*, 2023, 868, pp.161638. 10.1016/j.scitotenv.2023.161638 . hal-04371883

HAL Id: hal-04371883

<https://hal.science/hal-04371883>

Submitted on 22 Jan 2024

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1 **Artificial weathering of plastics used in oyster farming**

2

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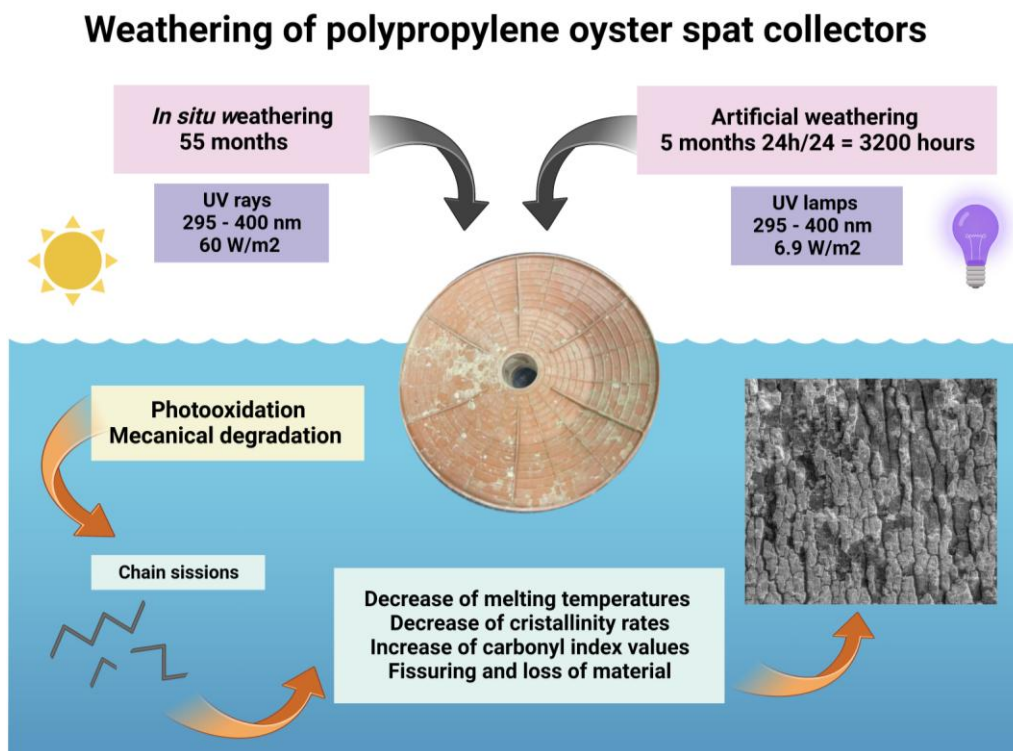
15 **ABSTRACT**

16 With the omnipresence of plastic litter from oyster farming in marine coastal areas, the objective of
17 this work was to better understand the weathering of plastics used in this field, focusing on oyster spat
18 collectors. During their use, around fifteen years, collectors made of polypropylene (PP) undergo
19 numerous degradations, alternatively submerged, emerged in seawater, and stored outdoor until the
20 next cycle. They weaken, crack, break, end up fragmenting and disseminated in the environment as
21 microplastics associated to persistent organic pollutants. In this work, a comparison of 55 months of *in*
22 *situ* weathering with five months of artificial weathering in air or in artificial seawater in a homemade
23 UV chamber was conducted to better understand the mechanisms involved. Chemical, thermal and
24 surface characterizations of virgin and weathered samples were conducted using Fourier Transform
25 Infrared Spectroscopy (FTIR), Differential Scanning Calorimetry (DSC) and Environmental Scanning
26 Electron Microscopy (ESEM). After 55 months of *in situ* weathering, collectors were notably
27 damaged with large fissures and loss of microplastics (MPs) associated with an increase of carbonyl

28 index values and a decrease of melting temperatures and crystallinity rates. Considering only UV
29 irradiation, five months of artificial weathering at 30°C under continuous irradiation of 6.9 W/m²
30 under UV lamps (295 - 400 nm) reproduced approximately 4.4 months of natural sunlight. Artificial
31 weathering confirmed that photooxidation by combined effects of UV rays and oxygen was the main
32 weathering mechanism and was reduced in seawater. These results help to understand the mechanisms
33 involved in the weathering of these collectors in the marine environment and provide valuable
34 informations for industrials and professionals. Our study suggests a better storage away from UV rays
35 and a reduction of the duration of use compared to current practices.

36

37 GRAPHICAL ABSTRACT



38

39 KEYWORDS

40 Oyster farming, Polypropylene, Photooxidation, *In situ* weathering, Artificial weathering

41 ABBREVIATIONS

42 PP : Polypropylene, MPs : Microplastics, POPs : Persistent organic pollutants

43

44 1. INTRODUCTION

45 At now, a total of five ocean gyres with high concentrations of plastics debris have been identified
46 (Avio *et al.*, 2017; Law *et al.*, 2010). 80 % of these marine debris are land-based sources and 20 %
47 are ocean-based sources like fisheries and maritime activities (Avio *et al.*, 2017; Bringer *et al.*,
48 2021b). Since the arrival of plastic from petrochemicals in the 1950s, professionals in aquaculture
49 activities have progressively turned to this new material, which is more resistant, durable, and
50 economical. Due to their stability, plastics persist in the environment for hundreds of years (Cai *et*
51 *al.*, 2018; Gewert *et al.*, 2015a). Degradation of plastics can proceed by chemical, physical
52 degradations and biological degradations (Gewert *et al.*, 2015; Singh and Sharma, 2008; Song *et al.*,
53 2017). Degradation rates are dependent on the manufacturing process (Han *et al.*, 2018), the presence
54 of stabilizers, additives and the weathering conditions including sunlight, temperature, and mechanical
55 stress (Avio *et al.*, 2017a; Kalogerakis *et al.*, 2017a; Masry *et al.*, 2021a; Rajakumar, Sarasvathy,
56 Thamarai Chelvan, *et al.*, 2009). In seawater, degradation rates are lower and more uniform compared
57 to landfill due to lower and less variable conditions as UV irradiation, oxygen content and
58 temperatures (Cai *et al.*, 2018; Gewert *et al.*, 2015). But in any cases, degradations finally lead to the
59 release of microplastics (MPs) in the environment which can be accumulated by a wide range of
60 marine organisms from zooplankton (Cole *et al.*, 2013), to fishes through bivalves (Bringer *et al.*,
61 2021a) being transferred along food chains until humans (Avio *et al.*, 2017a). Some studies have
62 reported the presence of MPs in human organs (Hermabessiere *et al.*, 2017), blood (Leslie *et al.*, 2022)
63 and faeces (Zhang *et al.*, 2021). MPs exposure by ingestion, inhalation or dermal contact can cause
64 physical damages, with oxidative stress, or inflammatory lesions (Prata *et al.*, 2020).

65
66 Furthermore, plastics are associated with additives and stabilizers (Zhang *et al.*, 2021). Plasticizers as
67 di-n-butyl phthalate, antioxidant agents as bisphenol A, flame retardants as tris-2-chloro-éthyl
68 phosphate, are chemicals added during the manufacturing process to obtain certain desired properties
69 (Avio *et al.*, 2017a; Masry *et al.*, 2021a). Because they are usually not covalently bonded to the
70 polymers (Gewert *et al.*, 2015; Paluselli *et al.*, 2019; Wang *et al.*, 2020), these chemicals can migrate

71 from the polymeric matrix towards the surrounding environment. In seawater, leaching of these
72 chemicals by diffusion is dependent on their solubility, their polarity, and their affinity to the matrix
73 (Paluselli *et al.*, 2019). Some of these chemicals are known to be harmful to the marine ecosystems and
74 humans causing dysfunctions of the immune and reproductive systems, cancers and neurodegenerative
75 disorders (Avio *et al.*, 2017; Hermabessiere *et al.*, 2017).

76

77 Plastics are also able to absorb various pollutants called persistent organic pollutants (POPs) present in
78 the marine environment such as pesticides as dichlorodiphenyltrichloroethane, polycyclic
79 hydrocarbons as naphthalene, and heavy metals as cadmium and lead for examples (Boucher *et al.*,
80 2016; Gewert *et al.*, 2015; Hermabessiere *et al.*, 2017; Holmes *et al.*, 2012; León *et al.*, 2018; Paluselli
81 *et al.*, 2019). These pollutants can be ingested and bioaccumulated by marine organisms and humans,
82 often vectorized by MPs (Avio *et al.*, 2017; Gewert *et al.*, 2015; Guo and Wang, 2019; Prata *et al.*,
83 2020). The risks that MPs and these persistent organic pollutants (POPs) pose to marine life and
84 humans are widely recognized and have been included in national and international marine protection
85 strategies, policies, and legislation (EU Marine Strategy Framework Directive).

86

87 With nearly 30,000 tons of Pacific oysters *Crassostrea gigas* produced each year, the department of
88 Charente-Maritime in France represents the first French production area with one third of the national
89 production (AGRESTE 2019). Different types of plastics such as polypropylene (PP), polyethylene
90 (PE) and polyvinyl chloride (PVC) are now used at each stage of production. Oyster spat collection is
91 the first phase of production in traditional oyster farming. Collectors have a specific design of a disc of
92 160 mm diameter, slightly concave, flexible and ribbed, with a thickness of 0.6 mm, designed in order
93 to collect a maximum of oyster spats in seawater during the reproductive period between July and
94 March. After collecting, collectors are taken out of seawater, oysters are detached and then transferred
95 to oyster bags. The environmental image of the profession is criticized today given the omnipresence
96 of plastic waste from oyster farming in coastal areas. Around 200 tons of plastic waste from oyster
97 farming, including 60 tons of oyster spat collectors, are found each year in the Atlantic south coast

98 (FEAMP 2020). During their use around fifteen years, collectors made of PP or PE undergo numerous
99 degradations. They weaken, crack, break, end up fragmenting, disseminating MPs and chemicals in
100 the marine environment. The toxicity of weathered MPs and associated chemicals from oyster farming
101 was recently evaluated on early stages of bivalve development (Bringer, Cachot, *et al.*, 2021).
102 Diméthylphtalate used as plasticizer and naphthalene as polycyclic hydrocarbons absorbed in the
103 marine environment have been identified on weathered oyster crops, collectors, and pipes (Bringer, Le
104 Floch, *et al.*, 2021) . Faced with the growing risk associated to plastics used in oyster farming for
105 marine life and humans, it is necessary understand and reduce their anthropogenic impacts.

106

107 *In situ* and artificial weathering of PP have been described in many studies (Badji *et al.*, 2018a; Lv *et*
108 *al.*, 2015; Rajakumar, Sarasvathy, Thamarai Chelvan, *et al.*, 2009; Tang *et al.*, 2019a) but were never
109 established for PP oyster spat collectors. These collectors are interesting because they have a chemical
110 composition and mechanical properties, with additives and stabilizers designed for aquaculture
111 activities specifications and because, *in fine*, oysters are destined for human consumption. In this
112 work, a comparison of 55 months *in situ* weathering with five months of artificial weathering in a
113 homemade UV chamber was conducted to better understand the mechanisms involved in PP collectors
114 degradations. With the aim of understanding the mechanisms of degradation of these collectors in the
115 marine environment, chemical, thermal, and structural characterizations of virgin and weathered
116 collectors were performed using Environmental Scanning Electron Microscopy (ESEM), Differential
117 Scanning Calorimetry (DSC), and Fourier Transform Infrared Spectroscopy (FTIR).

118

119 **2. MATERIALS AND METHODS**

120 ***2.1 Oyster collectors***

121 ***2.1.1 In situ weathering***

122 In 2017, a project conducted by the Center for Aquaculture, Fisheries, and the Environment of
123 Nouvelle-Aquitaine (CAPENA) had the objective to compare petrosourced and biosourced collectors
124 in terms of oyster spat collection efficiencies. Collectors have a specific design of a disc of 160 mm

125 diameter, slightly concave, flexible and ribbed, with a thickness of 0.6 mm. These collectors were
126 immersed in the area of the Pertuis Charentais, in the French department Charente-Maritime
127 (45°51'41.7"N 1°12'19.0"W) during three cycles of eight months collecting oyster spats in seawater
128 between 2017 and 2020. These collectors were alternatively submerged and emerged in seawater
129 during collection and were stored outdoors until the next collection cycle. Each cycle started in July
130 and stopped in March. In total, these collectors spent 24 months in seawater, and 31 months stored
131 outdoors until the start of this work in January 2022. A continuous environmental monitoring during
132 *in situ* weathering measured seawater temperature from 6 to 21 °C, salinity from 26 to 34 g/L,
133 outdoor temperature from 5 °C to 23 °C and a mean solar irradiance of 1200 kW/h/m²/year. Virgin
134 collectors, as references, were stored indoors and protected from humidity and UV rays. Virgin and *in*
135 *situ* weathered collectors were characterized by FTIR, DSC, and ESEM as described below.

136

137 **2.1.2 Artificial weathering**

138 Inspired by ISO 4892:2016 part 3 and ATSM G154:06, a homemade UV chamber was equipped with
139 five UV lamps with wavelength between 295 and 400 nm (KFMS reference PH710932 15W). These
140 UV lamps were selected because they can give a good reproduction of natural sunlight in the UV
141 range (Andrade *et al.*, 2019; Cai *et al.*, 2018) and because this medium energy is particularly efficient
142 in facilitating photodegradation of PP (Andrady, 2015). Virgin PP collectors were weathered under
143 UV rays in two different conditions: in air (UV+) or immersed in artificial seawater at 30 g/L
144 (UV+/SW). Negative controls, not exposed to UV rays, were installed in a box in the chamber (UV-
145 and UV-/SW). Artificial seawater was made by mixing milliQ water and commercial salt. Virgin PP
146 collectors were installed directly under UV lamps at 15 cm (UV+) or in borosilicate crystallizers filled
147 with 500 mL of 30 g/L of artificial seawater (SW) and closed with a borosilicate lid to limit
148 evaporation but letting pass UV rays through. This experiment was conducted for five months, at
149 30°C, under 24h/24 continuous UV irradiation. Temperature was monitored weekly with a mercury
150 thermometer. Intensity was monitored weekly with a radiometer (RS Pro reference IM-213) placed at
151 15 cm from the lamps and was approximately equal to 6.9 W/m² in a empty crystallizer and 3 W/m² in

152 artificial seawater. Salinity was monitored weekly with a multiparameter (WMW reference Multi3430
153 with Tetracon 925 probe) and was adjusted to 30 g/L by adding milliQ water if necessary. Each
154 month, one triplicate for each condition was removed and stored in glass petri dishes at 4 °C, in the
155 dark, to stop weathering.

156

157 ***2.2 Characterization***

158 ***2.2.1 Environmental Scanning Electron Microscopy (ESEM)***

159 The scanning electron microscope used in this work is an Environmental FEI Quanta 200 –
160 ESEM/FEG (LaSIE, La Rochelle University) working at 9 KV of electron beam accelerating voltage
161 and 0.1 nA of beam current under 200 Pa of water vapor pressure in the specimen chamber (Conforto
162 *et al.*, 2015). Secondary electron-type images, providing topographical information from the surfaces,
163 were obtained in environmental mode using a “Gaseous Large-Field” detector. The low-energy beam
164 and environmental conditions used are necessary to observe plastic materials, which are thermally
165 sensitive and poor electrical and thermal conductors, without any preparation. In fact, if a surface
166 metallization (which is part of the sample preparation procedure) was performed, it could partially or
167 totally mask cracks. The latter are an important parameter used to evaluate the weathering of plastic.
168 Surface topography of different parts of virgin and weathered collectors was observed. We made sure
169 that the results were coherent, homogenous, reproducible. Ten fissures lengths were measured on
170 random fissures and averaged with FIJI software.

171

172 ***2.2.2 Fourier Transform Infrared Spectroscopy (FTIR)***

173 In order to identify the different types of chemical bonds, FTIR was used. Infrared frequency identical
174 to the vibrational frequency of a bond results in absorption, creating a spectrum acting as a molecular
175 “fingerprint” of the sample. The position, shape, and intensity of peaks in the spectrum reveal details
176 about the molecular structure of the sample. Different parts of virgin and weathered collectors were
177 analyzed using a thermoscientific Nicolet iS50 FTIR spectrophotometer equipped with an attenuated
178 total reflectance mode (ATR). We made sure that the results were coherent, homogenous,

179 reproducible. Each spectrum was recorded as the average of 32 scans in the spectral range of 500 to
180 4000 cm^{-1} , with a spectral resolution at 4.0 cm^{-1} . Ten spectras were done randomly at the surface of
181 each sample averaged with a corrected baseline, and analyzed with Spectragryph 1.2 software.

182

183 **2.2.3 Differential Scanning Calorimetry (DSC)**

184 Specific temperatures of the samples such as melting temperature and crystallization, as well as the
185 enthalpy values associated with these phenomena were examined on a differential scanning
186 calorimeter TA Instrument Q100. Samples were cut, weighed, and heated from -60 °C to 210 °C in an
187 aluminum pan, then cooled to 40 °C. All experiments were performed at a rate of 20 °C/min under
188 nitrogen flow (50 ml/min) to avoid thermal degradation. For the sake of repeatability, at least two
189 samples for each type of collector were used for the DSC measurements. The analysis of DSC
190 thermograms was carried out with TA Universal Analysis software.

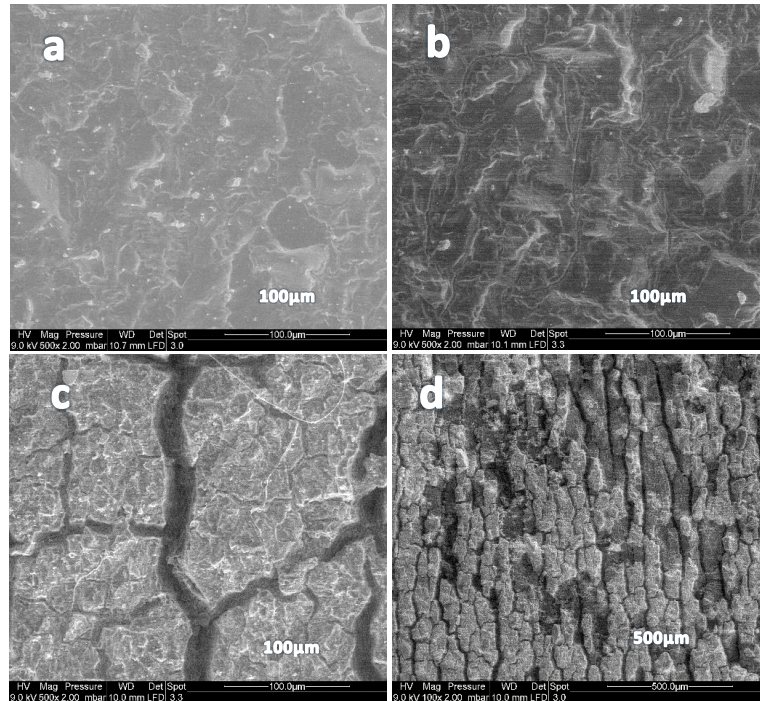
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192 **3. RESULTS**

193 **3.1 Surface analysis**

194 Surface images were obtained by using environmental scanning electron microscopy (ESEM). Virgin
195 PP collectors exhibited relatively homogeneous, smooth and compact textures without any fissures
196 (Fig 1a). *In situ* weathered PP collectors (Fig 1c and 1d) exhibited heterogeneous surfaces, with large
197 and deep averaged fissures up to 24 μm . Loss of material was observed at lower magnification (Fig
198 1d) caused by mechanical stress as tides, currents during submersion, wind and rain during emersion
199 but also handling. This loss of material suggests a release of MPs in the environment. Crystals of
200 sodium chloride salts and algae and diatom residues were also observed and confirmed that the
201 collectors stayed in seawater. Artificially weathered collectors under UV rays (UV+) start fissuring
202 after five months with the observation of 2 μm length fissures (Fig 1b). No fissures were observed for
203 the other conditions as negative control (UV-) and immersion in artificial seawater for five months
204 (UV-/SW and UV+/SW) (Results not presented). Artificial weathering confirmed that fissuring started
205 after five months exposed to UV rays.

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217 Figure 1. ESEM pictures of (a) virgin PP collector x 500; (b) five months artificially weathered PP
218 collector under UV rays x 500; (c) and (d) 55 months *in situ* weathered PP collector (x 500 and x 100
219 respectively). Arrow point to a small fissure of 2 μm .

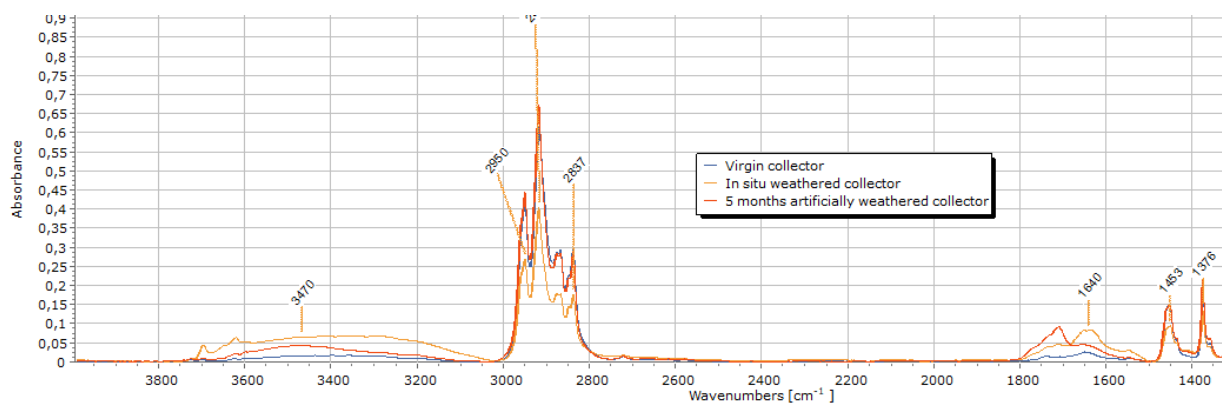
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3.2 Chemical analysis

222 The averaged infrared spectra of a virgin PP collector, used as reference, is presented in the Figure 2
223 (blue). The spectra showed absorption peaks located at wavenumbers of 971, 995, 1167, 1375, 1454,
224 2838, 2865, 2918, and 2953 cm^{-1} corresponding to specific vibrations of rocking, bending and
225 stretching and identified as the native hydrocarbon bonds of virgin PP (C_3H_6)_n (Badji *et al.*, 2018b;
226 Tang *et al.*, 2019b). Photooxidation, by combined effects of UV rays and oxygen, is known to be the
227 main weathering process for thermoplastics such as PP (Andrady, 2015; Gewert, 2015; Kalogerakis *et*
228 *al.*, 2017b; Masry *et al.*, 2021b). UV radiation has sufficient energy to break the C—C and C—H bonds
229 (Singh & Sharma, 2008). Diminution of C—C and C—H bonds on the *in situ* weathered collector (Fig
230 2 – orange) were observed at wavenumbers between 2800 - 3000 cm^{-1} confirming their break. The free
231 radicals produced during photooxidation reacted freely with atmospheric oxygen to form hydroxyl
232 groups (O—H) peaking at wavenumbers between 3000 - 3700 cm^{-1} and carbonyl groups (C=O)

233 including carboxylic acids and ketones peaking at wavenumber 1711 cm^{-1} , esters and aldehydes
234 peaking at wavenumber 1740 cm^{-1} and lactones peaking at wavenumber 1780 cm^{-1} (Badji *et al.*,
235 2018a; Lv *et al.*, 2015; Masry *et al.*, 2021b). These new significant peaks at around wavenumbers
236 $3000 - 3700\text{ cm}^{-1}$ and $1500 - 1800\text{ cm}^{-1}$ were identifiable on *in situ* and artificially weathered collector
237 infrared spectra (Fig 2) and their areas were a function of weathering conditions (Cai *et al.*, 2018;
238 Rajakumar, Sarasvathy, Thamarai Chelvan, *et al.*, 2009; Tang *et al.*, 2019b).

239



240 Figure 2. FTIR spectra of a virgin PP collector (blue), *in situ* weathered PP collector (orange) and a 5
241 months artificially weathered collector (UV+) (yellow)

242

243 The carbonyl index (CI) has been used to quantify the degree of weathering of various samples
244 (Almond *et al.*, 2020; Badji *et al.*, 2018a; Julienne *et al.*, 2019; Rajakumar, Sarasvathy, Chelvan, *et*
245 *al.*, 2009; Song *et al.*, 2017b; Tang *et al.*, 2019a). This index is defined as the ratio of the integrated
246 area of the carbonyl group peak at around wavenumbers 1700 cm^{-1} to an internal constant peak. In this
247 work, boundaries between wavenumbers 1390 and 1340 cm^{-1} were chosen for the constant peak
248 because it remained unchanged during weathering. Consequently, boundaries between wavenumbers
249 1820 and 1570 cm^{-1} were chosen for the carbonyl peak. Carbonyl index of *in situ* and artificially
250 weathered samples calculated on the average of ten spectra for each sample and after baseline
251 correction are resumed in Table 1. Carbonyl index values for the virgin collector and negative control
252 (UV-), as references, were similar and equal to 0.7 and 0.5 respectively. These values were not equal
253 to zero probably due to thermal degradation of the material during injection. Negative control (UV-)

254 also confirmed that no additional degradation occurred during the five months of artificial weathering.
255 Consequently, only UV ray exposure and immersion in seawater can be considered in this experiment.
256 After five months of artificial weathering in seawater (UV-/SW), the carbonyl index value reached
257 1.9. Additionally immersed and exposed to UV rays (UV+/SW), this value increased to 2.1. Finally,
258 upon exposure to UV rays only during five months (UV+), the carbonyl index reached the value of 3.
259 In this experiment, the highest value for carbonyl index was therefore found after five months
260 artificially exposed to UV rays only (UV+). After 55 months of *in situ* weathering, the carbonyl index
261 value reached 4.2.

262

263 **3.3 Thermal analysis**

264 Semi-crystalline PP has a heterogeneous structure, consisting of crystalline and amorphous phases
265 (Masry *et al.*, 2021b). The crystallinity rate and the melting temperature are impacted by weathering
266 (Badji *et al.*, 2018b). Monitoring the evolution of these parameters using Differential Scanning
267 Calorimetry (DSC) gave a global indication of weathering degree of the samples. On the DSC
268 thermograms (Fig 3), the heat flow exhibited endothermic peaks related to the melting of PP allowing
269 the measure of melting temperatures and melting enthalpies (ΔH_m) corresponding to the energy needed
270 to melt the sample. Crystallinity rates can be calculated as this relation: Crystallinity rate (%) = $\Delta H_m /$
271 $(\Delta H_{inf}) \times 100$ (equation 1) with ΔH_{inf} equal to 209 J/g for PP corresponding to the theoretical melting
272 enthalpy of a fully crystalline PP (Badji *et al.*, 2018; Han *et al.*, 2018) ΔH_m can be calculated with TA
273 analysis software. Melting temperatures and crystallinity rates and of virgin, *in situ* and artificially
274 weathered PP collectors are summarized in Table 1. For the virgin collector, as reference, melting
275 temperature was equal to 169 °C, melting enthalpy was equal to 59 J/g and crystallinity rate was equal
276 to 29 % (equation 1). These values were equivalent for the negative control after five months of
277 artificial weathering without UV rays (UV-). After *in situ* weathering, significant decreases of the
278 melting temperature from 169 to 164 °C, and crystallinity rates from 29 to 25 % were observed.
279 Changes in melting temperature and crystallinity rate were observed on samples exposed to UV rays
280 (UV+). No differences were observed for the other conditions in seawater (UV+/SW, UV-/SW).

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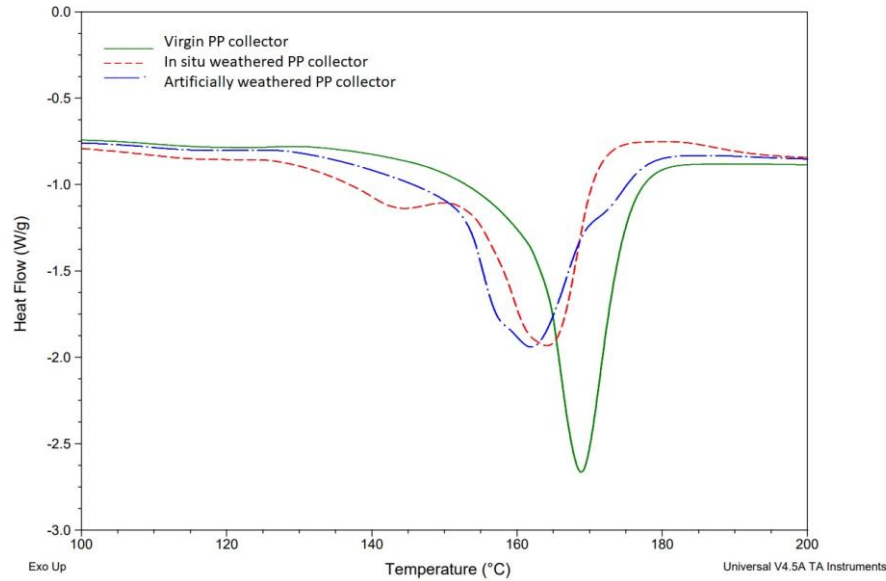


Figure 3. DSC thermograms of a virgin PP collector (green line), *in situ* weathered PP collector (red line) and artificially weathered PP collector (blue line) with endothermic peak indicating the melting of the material.

Virgin PP collector	55 months <i>in situ</i> weathered PP collector	5 month artificially weathered PP collectors			
		UV-	UV+	UV-/SW	UV+/SW
0.7	4.2	0.5	3.0	1.9	2.1
169	164	169	164	169	169
29	25	31	26	30	30
0	24	0	2	0	0

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Table 1: Carbonyl index values, melting temperatures, crystallinity rates, and fissure lengths for virgin PP collectors, 55 months *in situ* weathered PP collectors, and five months artificially weathered PP collectors.

4. DISCUSSION

During their use, PP oyster spat collectors undergo various degradations as the result of photooxidation, thermal degradation, mechanical degradation, and biodegradation, all based on free-

302 radicals mechanisms. (Avio *et al.*, 2017a; Singh & Sharma, 2008). After *in situ* weathering, increase
303 of carbonyl index values, and decrease of crystallinity rates, melting temperatures and apparition of
304 fissures were observed (Table 1). Artificial weathering was conducted to better understand the
305 mechanisms involved. Because the temperature in the UV chamber was fixed, homogenous and
306 relatively low, and because no additional mechanical stress was applied, only UV irradiation was
307 considered in this experiment. The highest degradation rate obtained after UV irradiation in air
308 (UV+) and negative control confirmed that no additional degradation than photooxidation occurred.
309 Abiotic degradations precede biotic degradations. Photooxydation by combined effect of oxygen and
310 UV rays is known to be the main weathering mechanisms for PP. Initiated by UV rays absorbed by
311 structural abnormalities or impurities, photooxidation cause chain scissions and produce shorter and
312 more mobile chains with free radicals reacting freely with atmospheric oxygen (Gewert *et al.*, 2015).
313 This mechanism usually starts in the amorphous phase (Andrady, 2015; Julienne *et al.*, 2019a; Masry
314 *et al.*, 2021, Han *et al.*, 2018) but results of this work obtained after artificial weathering showed a
315 decrease of crystallinity rate indicating that photooxidation already caused damage in the amorphous
316 phase and then occurred in the crystalline phase. This mechanism, already described by (Badji *et al.*,
317 2018; Lv *et al.*, 2015; Tang *et al.*, 2019), finally lead to a decrease of the hydrophobicity and the
318 molecular weigh of the polymer. However, the molecular weight remaining high, biodegradation of
319 the long chains stays limited (Gewert *et al.*, 2015; Singh and Sharma, 2008). In fact, PP is often used
320 as a negative control in biodegradation experiments (Kalogerakis *et al.*, 2017; Lott *et al.*, 2020).

321
322 At the location of the *in situ* weathering, a mean solar irradiance of 1200 kWh/m²/year was monitored
323 (https://re.jrc.ec.europa.eu/pvg_tools/fr/#DR) which corresponds to an irradiance of 60 kWh/m²/year
324 in the total UV range (295 and 400 nm). During artificial weathering, the intensity of the UV lamps in
325 this same range was averaged at 6.9 W/m² and was maintained for 3200 hours which correspond to a
326 total cumulated irradiance of 22 kWh/m² (= 6.9 x 3200). Consequently, artificial weathering
327 reproduced approximately 134 days (22 / 60 x 365) corresponding to 4.4 months (Gewert *et al.*, 2018).

328 This calculation of solar simulation equivalent is theoretical because it is dependant of numerous
329 environmental factors, which varies throughout the seasons.

330

331 After five months of artificial weathering in seawater, an increase of carbonyl index values was
332 observed whereas the crystallinity rates and melting temperatures were little impacted (Table 1).
333 Absorption of seawater in PP can allow the entrance of dissolved oxygen into the matrix favoring
334 chain oxidation and hydrolysis (Kalogerakis *et al.*, 2017a) but the kinetics of these degradation
335 mechanisms are slower than those for photooxidation. Moreover, intensity of UV irradiance in
336 artificial seawater (3 W/m^2) was lower than for direct exposure to a UV lamp (6.9 W/m^2), reducing the
337 photooxidation mechanism in seawater, wich it is expected to occur in the marine environment. No
338 fissures were observed for these conditions (UV+/SW and UV-/SW).

339

340 Collectors artificially weathered during five months gave approximately the same results than after 55
341 months weathered *in situ* except for the fissure lengths due to additional degradations involved *in situ*.
342 It is important to consider that *in situ*, degradation of PP collectors was influenced by alternations of
343 tides, day and night cycles, weather and seasonal conditions, presence of spats, biofooling and crystal
344 salts. In conclusion, *in situ* weathering of these PP collectors was mainly caused by photooxidation
345 during emersion and outdoor storage and was enhanced with additional effects of mechanical stresses
346 such as tides, currents, wind and rain but also professional handling. Comparing *in situ* and artificial
347 weathering must be done with caution since there is still not an adequate simulation of the combined
348 effects and synergies of various environmental factors (Andrade *et al.*, 2019; Rajakumar,
349 Sarasvathy, Thamarai Chelvan, *et al.*, 2009; Yang & Ding, 2006).

350

351 **5. CONCLUSIONS**

352 This is the first study comparing *in situ* and artificial weathering of PP oyster spat collectors. This
353 work shows that after only 55 months *in situ*, alternatively submerged and emerged, the collectors
354 were notably damaged, with large fissures and loss of material such as MPs. These damages were

355 mainly due to photooxidation during exposure to UV rays, enhanced by mechanical stress as tides,
356 current, wind, rain and handling all along their use and storage. Considering only irradiation in the UV
357 range, five months of artificial weathering reproduced approximately 4.4 months of natural sunlight *in*
358 *situ* and confirmed that photooxidation by combined effect of UV rays and oxygen was the main
359 weathering mechanism and was higher in air than in seawater. Degradations as the apparition of the
360 first fissures were observed relatively soon under UV irradiation. This first comparative study gave
361 preliminary results on this topic and help to better understand the mechanisms involved in the
362 degradation of these collectors. This work allows to make recommendations to industrials and
363 professionals including a better storage protected from UV rays and a reduction of the duration of use.
364 It is important to note that there is still a lack of information on the toxicity and the potential
365 environmental hazard of these altered plastics in the marine environment, as vectors of MPs, POPs and
366 microorganisms (Avio *et al.*, 2017b; Gewert *et al.*, 2015b). Consequently, a future work will be
367 focused on the relationship between weathering, fragmentation and toxicity including promising
368 alternatives such as biosourced materials.

369

370 **Funding and acknowledgements:**

371 The PLASTIC LAB project (Dégradabilité, effets et comportement des PLASTIques Conchylicoles en
372 LABoratoire), was funded by the FEAMP (Fonds Européen des Affaires Maritimes et de la Pêche)
373 and the SCIC TEO (Taho'E Eco Organisation). The authors thank the CAPENA, and Pierrick Barbier,
374 for providing the oyster collectors used in this work, Latta Socciligame and Fanon Julienne for their
375 help. The authors are gratefull to all contributors including Linda Sperling for their proofreading
376 services.

377

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