

Correlation between deformation and total entropy change at the first-order magnetic transition of Fe2P-based magnetocaloric materials

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- Latent heat; First-order Magnetic Transition; Magnetism ; Magnetocaloric effect ;
- Thermomagnetic materials

 Magnetic transitions often involve simultaneous changes in the electronic (*S*el), lattice (*S*latt), and magnetic (*S*mag) properties. The expression *S* = *S*latt+*S*el+*S*mag is commonly employed to express how these degrees of freedoms and their entropy are expected to sum up. By extension, at first-28 order magnetic transition (FOMT) with transition temperature (T_{tr}), the transition entropy ΔS_{tr} = *L*/*T*tr related to the latent heat (*L*) is expected to originate from one or several of these contributions. Unfortunately, estimating individually each term is challenging. On top of that, the above summation relies on a strong and disputable assumption that the degrees of freedom are independent, *i.e.* there is no cooperativity between them.

 Understanding the driving mechanisms for the latent heat of giant magnetocaloric materials is of outmost importance for applications, as the optimization primarily consists in triggering a large latent heat for an external magnetic field change as small as possible. Following the discovery of a giant magnetocaloric effect with magneto-structural FOMT in Gd5Si2Ge2, a structural entropy change representing about half of the magnetocaloric effect was proposed as primary reason for overcoming the performances of elemental Gd [1,2]. This led to a general scaling that the larger the relative volume change Δ*V/V*, the stronger the lattice contribution and therefore the higher the magnetocaloric effect [3]. This approach may however be confusing as it considers the isothermal entropy change induced by a magnetic field change (referred to as ΔS), whose relationship with the transition entropy discontinuity in zero field (referred to as Δ*S*tr) is indirect. In addition, an accurate determination of Δ*S* can be challenging at FOMT due to the spike artefact. A Δ*V/V* scaling of the magnetocaloric effect appears also applicable to isotropic 45 magneto-elastic FOMT such as in La(Fe, Si)₁₃. For this system, after some early controversies on whether the lattice entropy change counteracts the magnetic one [4], direct insights using inelastic resonant x-ray scattering techniques have led to a consensus that the lattice entropy change adds up to the magnetic contribution and may be responsible for nearly half the isothermal entropy change [5-8]. However, Δ*V* or Δ*V/V* is not a convenient parameter for characterizing magneto-elastic anisotropic FOMT as in hexagonal Fe2P compounds where the contraction on the *a* axis is compensated by an expansion along the *c* axis leading to nearly negligible volume changes.

 In view of the strong interest in Fe2P compounds [9-28], insights into the mechanism of their FOMT are highly desirable. Fe2P-based compounds can exhibit ferromagnetic transitions with

 extreme behaviors. Depending on chemical compositions, it is possible to achieve: *i)* second- order transitions (typically Mn2-xFe*x*P0.5Si0.5 with *x* < 0.7) [9]; *ii)* FOMT with particularly small hysteresis and latent heat (case of the Fe2P binary parent) [10,11]; *iii)* FOMT with small hysteresis but significant latent heat (best compositions for magnetocaloric or thermomagnetic applications, *e.g.* MnFe0.95P0.585Si0.34B0.075) [12]; and *iv)* up to giant hysteresis and giant latent heat as in MnFe0.95P2/3Si1/3 [13]. In all these materials the volume discontinuity is either limited (Δ*V/V* < 0.2 %) or down to virtually zero. For the Fe2P-based family, no consensus has ever emerged about the dominant(s) contribution(s) to Δ*S*tr and even the most recent reports still led to puzzling results. First, there is a one order of magnitude difference on the integrated magnetic entropy between Fe2P and MnFe0.95P0.585Si0.34B0.075 [11,16]. This is hardly understandable in 65 view of their comparable saturation magnetizations (M_{sat} approximately 2.9 and 4.0 μ_B/f.u., respectively). Second, in contrast to early theoretical studies pointing out that the latent heat has mostly a magnetic origin [17], recent phonon calculations and inelastic scattering experiments 68 suggest that nearly all the transition entropy (ΔS_{tr}) has a structural origin [18,19]. Finally, former attempts at scaling the FOMT in Fe2P-based materials (either via *L*, hysteresis or even *ΔS*) failed to encompass the broad variety of properties observed in this system. Dung *et al.* pointed out that isothermal entropy changes (Δ*S*) are correlated with |∆*a/a*| and |∆*c/c*| in a couple of (Mn,Fe)2(P,Si) samples [25]. In MnFe(P,Si,B) compounds, Maschek *et al.* found that the thermal hysteresis varies like ∆(*c/a*), but no specific correlation could be established for Δ*S* [22]. In (Mn,Fe)2(P,Si) compounds, You *et al.* reported a linear relation between hysteresis and *c/a* (after a given threshold), while they pointed an absence of correlation between *L* and (*c/a*) [29]. Miao *et al.* claimed the existence of a striking relationship between thermal hysteresis and elastic 77 strain energy for a large panel of $Fe₂P$ compounds [20]. It must be emphasized, however, that all these previous observations carry some ambiguities. First, even though Δ*S* is correlated with Δ*S*tr, it also involves other parameters such as the sensitivity of the transition to the magnetic field *Ttr(B)* or even extrinsic ones like the transition width. Second, thermal hysteresis can also not be regarded as an intrinsic quantity for a FOMT. Indeed hysteresis is known to be impacted 82 by external features like the microstructure and is even not a necessity for FOMTs, as materials with nearly negligible hysteresis but sizable latent heat can be observed like in (Mn,V,Fe)2(P,Si,B) [23]. Actually, *ΔStr* (or *L=TtrΔStr*) are more genuine intrinsic quantities, directly connected to the Erhenfest definition of first-order transitions.

 Seeking to bring back together the diversity in FOMTs observed in Fe2P materials, **Figure 1** presents the transition entropy Δ*S*tr as a function of the elastic deformation quantified as the discontinuity in the cell parameters ratio Δ(*c/a*). The MnFeP0.66-xSi0.34B*^x* series is particularly important as it covers a wide Δ*S*tr range while keeping chemical substitutions limited [13]. Results for Fe2P, MnFe(P,Si) and MnFe(P,Ge) were included depending upon availability of Δ*S*tr and temperature dependent structural data. The criteria used to determine Δ(*c/a*) and Δ*S*tr as well as the list of the eighteen considered samples are presented in Supplementary Materials.

 A striking correlation is observed between the transition entropy Δ*Str* and the deformation Δ(*c/a*). We emphasize, however, that this correlation does not imply a predominant lattice component in the entropy change; it neither addresses the respective weights of the magnetic, lattice or electronic contributions within the net Δ*S*tr value. First of all, one must keep in mind that the experimental determinations of Δ*S*tr can only yield a global quantity incorporating all the degrees of freedom. Second, we stress that even when using additional experimental data, the aim at separating the various contributions in Δ*S*tr is basically impossible owing to the intrication between degrees of freedom that characterizes FOMTs. Recently, it has been proposed that the 101 elastic deformation energy could be expressed as $\Delta U_{\text{elastic}}/V_{\text{mol}} \approx (c_{11}+c_{12})\epsilon_1^2+2c_{13}\epsilon_1\epsilon_3+(c_{33}\epsilon_3^2)/2$ where *ε*ij denotes the strains and *c*ij the elastic constants [20]. Considering the theoretical elastic constants for MnFeSi0.33P0.58B0.08 [21] and the experimental strains for MnFe0.95P0.595Si0.33B0.075 104 [13], the above equation leads to a $\Delta S_{elastic} = \Delta U_{elastic}/T_{tr}$ estimated in the range 8 – 18 J kg⁻¹ K⁻¹, 105 which would be compatible with the experimental $\Delta S_{tr} = 16$ J kg⁻¹ K⁻¹. Nevertheless, the above equation involves squared strains, whereas Δ*S*tr appears to linearly scale with Δ(*c/a*). Even though large uncertainties make it difficult to distinguish the two possibilities, a Δ*S*tr-Δ(*c/a*) scaling was found to be the most reliable.

 The main result of Figure 1 is that the deformation appears as the source of the *total* transition entropy change Δ*S*tr. But no complementary information about the various contributions to this overall result is accessible. Let us also specify that the correlation appears so far limited to Fe2P materials. Considering iso-structural FOMTs in other hexagonal materials, (Sc,Ti)Fe² (Δ(*c/a*) ≈ 0.004 and Δ*S*tr ≈ 7 J kg-1 K-1 [31]) would be close to fit in, but (Hf,Ta)Fe² (Δ(*c/a*) ≈ 0.014 and Δ*S*tr $114 \approx 4$ J kg⁻¹ K⁻¹ [32]) strongly departs from the trend. This might reflect that each materials family has its own underlying FOMT mechanism, leading to distinct Δ*S*tr-deformation relations.

 To conclude, the scaling shown in Fig. 1 is a first step towards a systematization of the FOMT properties in the Fe2P materials family. From an applicative viewpoint, this correlation between Δ*S*tr and Δ(*c/a*) also illustrates the challenges met when optimizing Fe2P-related magnetocaloric materials. In order to reach large isothermal entropy changes, one needs sizable Δ*S*tr, but the latter necessarily goes along with sizable deformations having negative consequences such as hysteresis or mechanical embrittlement. Besides tailoring the microstructure, fine tuning of Δ*S*tr is therefore a crucial issue to target magnetocaloric materials compatible with applications.

123

124 **Figure 1:** Transition entropy of the FOMT as a function of deformation Δ(*c/a*) of the lattice 125 parameters for Fe₂P-based materials. The dashed line marks out the linear regression with 126 \blacksquare intercept fixed at 0 $(R^2 = 98.7\%)$.

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