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1 **Crystal Growth Blocking Strategy Enabling**
2 **Efficient Solvent-Free Synthesis of Hierarchical**
3 **UiO-66 for Large-Molecule Catalysis**

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15 **KEY WORDS:** MOFs; UiO-66; hierarchical structure; DBT oxidation

16

17 **ABSTRACT**

18 Metal-organic frameworks (MOFs) with rich hierarchical structure have drawn great attention
19 in the process with large molecule, such as catalysis and adsorption. However, efficiently and
20 cost-effectively fabricating hierarchical MOFs with robust stability remains challenging. Herein,
21 we report a crystal growth blocking strategy to prepare hierarchical UiO-66 by a solvent-free
22 method. The addition of FeCl₃ in the synthetic system blocks the overgrowth of UiO-66 crystal
23 and limits the particle size to around just 12 nm. After the removal of Fe by ethanol treatment,
24 these small particles loosely aggregate and gain intercrystalline mesopores. The optimal sample
25 Mes-UiO-66 exhibits well-developed hierarchical structure, the BET surface area reaches 1161.9
26 m²g⁻¹, and the mesopore volume attains a state-of-art 1.21 cm³g⁻¹. The superior catalytic
27 performance and the robust stability of Mes-UiO-66 are demonstrated by the oxidation of DBT,
28 of which the DBT conversion surpasses 99% and shows slight decrease after five cycles. Our
29 finding may provide a novel platform for efficiently fabricating hierarchical MOFs for large
30 molecule catalysis.

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36 INTRODUCTION

37 Metal-organic frameworks (MOFs), as a well-developed class of porous materials, have drawn
38 much attention in gas adsorption, catalysis and pharmaceutical science, due to the characteristics
39 of large specific surface area and tunable framework structures¹⁻⁴. However, the pore size of
40 most reported MOFs can only be adjusted with micropore regime. The limited accessibility and
41 mass transformation greatly hinder their application when dealing with large guests, such as
42 macromolecule catalysis/ adsorption or large particle capturing⁵⁻⁸. Thus, the pivotal obstacle in
43 expanding the application of MOFs with large guests has been focused on enlarging the pore size
44 and the corresponding pore volume, but remains challenging.

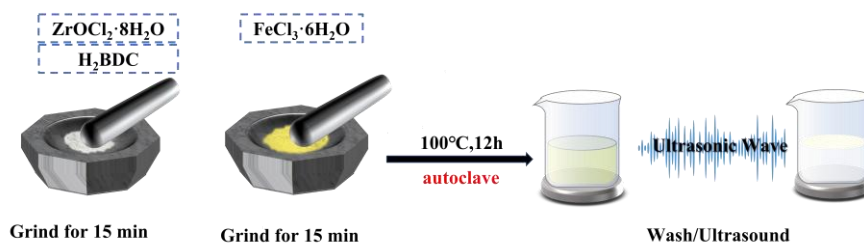
45 Pioneering efforts have been practiced to fabricate MOFs with relatively large pore sizes.
46 Considering the ligand of MOFs is more flexible in size compared with the metal center, the
47 splendid idea of exploiting large organic molecules as ligands has drawn a lots of attention and
48 has been demonstrated successfully⁹⁻¹¹. However, the complex molecular structures of these
49 ligands greatly increase the preparation cost and cause the problem of poor stability in thermal
50 and chemical solvent environments. In contrast, hierarchical MOFs gained by introducing
51 secondary pore structure in micropore MOFs are more cost-effective and robust. "Top-down"
52 and "bottom-up" are two common strategies to develop hierarchical MOFs from the micropore
53 ones. The "top-down" process employs post-treatment, such as solvent/ acid etching and linker
54 exchange, to excavate caves and holes on MOFs crystals. Despite its simplicity and efficiency,
55 the partial corrosion of the well-crystalized can still negatively affect the stability of MOFs. On
56 the other hand, the "bottom-up" strategy prefers to introduce abundant secondary pores into or
57 between the MOFs crystals during the synthetic process. The ideal hierarchical structure is

58 commonly obtained by rationally using templates, exquisitely choosing modulators or precisely
59 tuned synthetic conditions ¹²⁻¹⁴. For instance, exploiting surfactants (CTAB) as sacrificial
60 templates could create considerable mesoporous in various MOFs ^{15,16}, introducing fatty acids
61 with different chain lengths as regulator provides a splendid platform to regulate the mesoporous
62 ^{17,18}, moreover employing unique synthetic condition such as nano-fusion or supercritical fluid
63 can even fabricate hierarchical MOFs without any additive ¹³. Because of the slight impact on the
64 metal-organic frameworks, the original structure is largely preserved under the "bottom-up"
65 strategy, but the insufficient synthesis efficiency and high synthesis costs are still challenging.
66 Together, although remarkable development has been gained, the search for a novel strategy to
67 efficiently fabricate robust hierarchical MOFs with easy-operated process and low-cost is still
68 desired.

69 UiO-66, made of zirconium base metal nodes and 1,4-dicarboxybenzene (H₂BDC), is one of
70 the most used MOFs. Its excellent stability has been confirmed in water, acidic/ alkaline
71 solutions, most organic solvents and thermal condition up to 500 °C ¹⁹. Moreover, the original
72 micropore structure of UiO-66 can be developed into hierarchical structure by rationally
73 designed "bottom-up" strategy, although with relatively poor crystalline and low efficiency ^{20,21}.
74 Recently, well-crystalized MOFs including UiO-66, can be synthesized efficiently by solvent-
75 free method, which inspired our design of fast fabricating hierarchical UiO-66 with robust
76 stability by modified solvent-free approach ²²⁻²⁵. Herein, we report a crystal growth blocking
77 strategy for efficiently solvent-free synthesis of hierarchical UiO-66. The addition of Fe salt in
78 solvent-free system successfully limits the crystal size of UiO-66 to between 12 ±2 nm, but
79 leaves slight Fe residue. Moreover, the aggregation of these nano-crystals introduces abundant
80 secondary mesopores to gain well-crystalized hierarchical UiO-66. The optimal sample Mes-

81 UiO-66 exhibits a remarkable mesopore volume of $1.21\text{cm}^3\text{g}^{-1}$, which is one of the largest in
82 UiO-66. The oxidative desulfurization evaluation demonstrates the superior catalytic ability of
83 Mes-UiO-66 when dealing with large molecules, such as DBT. This work may open up a novel
84 strategy for preparing robust MOFs with rich hierarchical structures.

85 RESULTS AND DISCUSSION



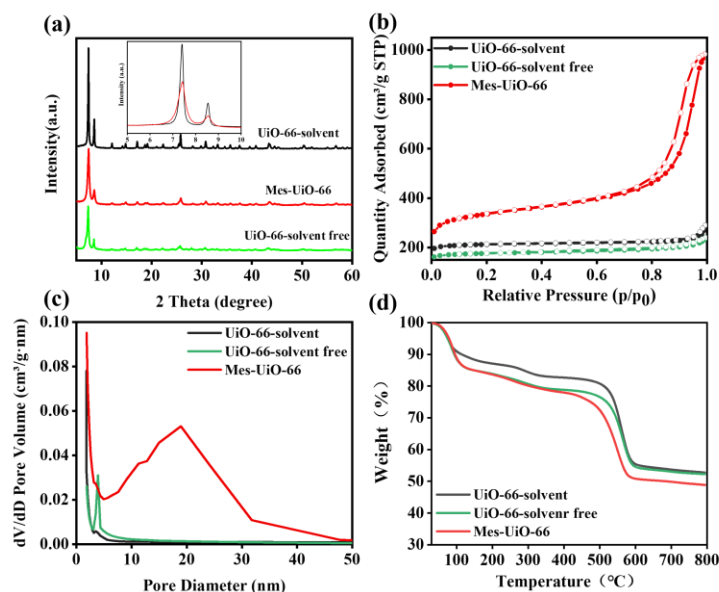
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Figure 1. Schematic diagram for the preparation of Mes-UiO-66.

88 Figure 1 illustrates the synthetic process of the hierarchical UiO-66. Firstly, $\text{ZrOCl}_2 \cdot 8\text{H}_2\text{O}$
89 and H_2BDC are mixed and ground for 15 min. Then $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ is added into the mixture and
90 ground for another 15 min. After crystallizing in an autoclave, the obtained powder is washed by
91 ethanol to remove Fe salt. After drying in vacuum, powder-like hierarchical UiO-66 is gained,
92 which is labeled as Mes-UiO-66. For comparison, samples synthesized without the addition of
93 Fe (UiO-66-solvent free) and synthesized by traditional hydrothermal method (UiO-66-solvent)
94 are also obtained²². The effect of the synthetic condition on Mes-UiO-66 is also investigated,
95 including the molar ratio of Fe and Zr ($R_{\text{Fe}/\text{Zr}}$), crystallization time and crystallization
96 temperature. Based on the evaluation of the crystallinity, BET surface area and mesopore
97 volume, the optimal synthetic condition is indicated as crystallizing at 100 °C for 12 h with a
98 $R_{\text{Fe}/\text{Zr}}$ of 3. (Figure S1-3, Table S1-3)

99 Figure 2a shows the XRD patterns of various UiO-66. In agreement with UiO-66-solvent and
 100 UiO-66-solvent free, Mes-UiO-66 displays well-resolved diffraction peaks of UiO-66 structure
 101 at 7.4 and 8.5 degree, and no peaks of any crystal impurity are observed. The broader peak width
 102 of Mes-UiO-66 compared with that of UiO-66-solvent imply its relatively small particle size.
 103 The results of XRD suggest that the addition of Fe salt is incapable of impacting the
 104 crystallization process or inducing the nucleation of any undesired crystal, agreed with the well
 105 thermal stability of Mes-UiO-66 before and after activation (Figure 2b and Figure S4).
 106 Moreover, the ICP and XPS results confirmed the negligible residue of Fe in Mes-UiO-66,
 107 (Table S4 and Figure S5), demonstrating the Fe salt is more favorable to act as mesoporous-
 108 inducing agent than to combine into the framework of Mes-UiO-66.



109
 110 **Figure 2.** (a) XRD patterns, (b) TG curves, (c) N₂ sorption isotherms and (d) pore size
 111 distribution of UiO-66-solvent, UiO-66-solvent free and Mes-UiO-66.

112 The hierarchical structure of Mes-UiO-66 is firstly investigated by N₂ adsorption-desorption
 113 test, and the N₂ sorption isotherms are shown in Figure 2c. UiO-66-solvent and UiO-66-solvent

114 free exhibit the typical type I isotherms, indicating the predominant micropore structure in the
 115 two samples. In comparison, Mes-UiO-66 displays type I and IV isotherm with a hysteresis loop,
 116 demonstrating the existence of both micropore and mesopore structure, which solid defined the
 117 hierarchical structure of Mes-UiO-66. The detailed N₂ sorption results are listed in Table 1. Mes-
 118 UiO-66 displays a BET surface area of 1161.9 m²g⁻¹, slightly higher than that of UiO-66-solvent
 119 and UiO-66-solvent free. Interestingly, the mesopore volume of Mes-UiO-66 reaches 1.21 cm³g⁻¹
 120 ¹, over five times as that of UiO-66-solvent (0.21 cm³g⁻¹) and UiO-66-solvent free (0.24 cm³g⁻¹).
 121 Moreover, the affluent mesopore structure of Mes-UiO-66 surpasses most of the reported
 122 hierarchical UiO-66 (Table S5). The pore distributions are calculated based on the BJH
 123 desorption branch. As shown in Figure 2d, in agreement with the isotherms, UiO-66-solvent and
 124 UiO-66-solvent free exhibit almost no pores between 2-50 nm (the peaks around 4 nm are caused
 125 by the instrumental error), confirming the absence of mesopore. But for Mes-UiO-66, a peak
 126 centers around 18 nm is observed between the range of 5 to 32 nm. Considering the good
 127 crystallinity of Mes-UiO-66, the relatively broad mesopore distribution may be the result of the
 128 accumulation of small crystal.

129 **Table 1.** Detailed N₂ sorption results of various samples.

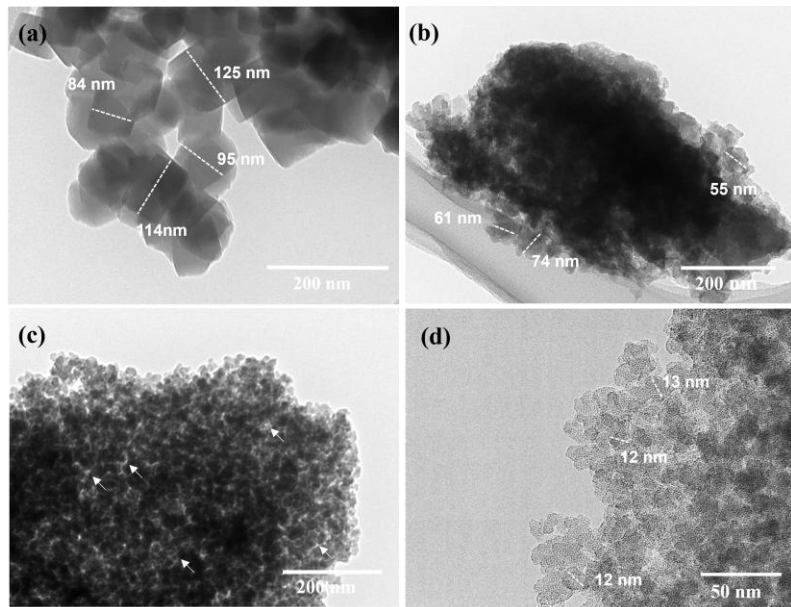
Sample	S _{BET} (m ² g ⁻¹) ^a	V _{total} (cm ³ g ⁻¹) ^b	V _{micro} (cm ³ g ⁻¹) ^a	V _{meso} (cm ³ g ⁻¹) ^c	S _{external} (m ² g ⁻¹) ^a
Mes-UiO-66	1161.9	1.51	0.30	1.21	412.4
UiO-66-solvent free	1078.0	0.65	0.41	0.24	114.9
UiO-66-solvent	798.0	0.50	0.31	0.21	87.6

130 ^a t-plot method

131 ^b desorption total pore volume

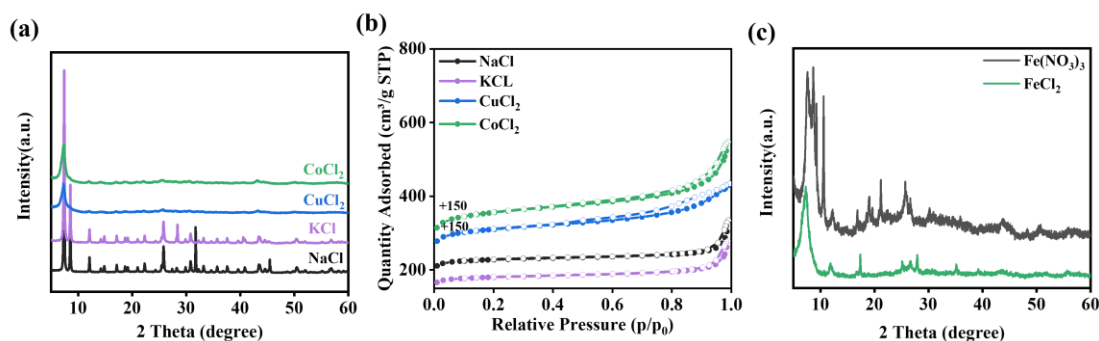
132 ^c BJH desorption branch

133 To further reveal the origin of the hierarchical structure, the morphology of UiO-66
134 synthesized by different methods is investigated. The SEM images of Mes-UiO-66, UiO-66-
135 solvent and UiO-66-solvent-free all disclose the unordered aggregation of particles, but the size
136 and the packing degree of different MOFs are discrepant (Figure S6-8). As shown in Figure 3,
137 the TEM images confirmed that the average diameter of Mes-UiO-66 is around 12 ± 2 nm, much
138 smaller than that of UiO-66-solvent (105 ± 20 nm) and UiO-66-solvent-free (65 ± 10 nm), in
139 good agreement with the broader diffraction peaks of Mes-UiO-66. Moreover, Mes-UiO-66
140 displays a relatively loose aggregation of the small particles, in which abundant interstices and
141 spaces can be obviously observed (Figure 3c-d and Figure S9). In contrary, the particles of UiO-
142 66-solvent and UiO-66-solvent-free clung tightly, leaving just a little space between each
143 particle. Thus, it can be demonstrated that the abundantly hierarchical structure in Mes-UiO-66
144 stem from the nano-scale crystal and their unconsolidated accumulation, which should be
145 attributed to the addition of Fe salt during the synthetic process.



146
147 **Figure 3.** TEM images of (a) UiO-66-solvent, (b) UiO-66-solvent free, (c, d) Mes-UiO-66.

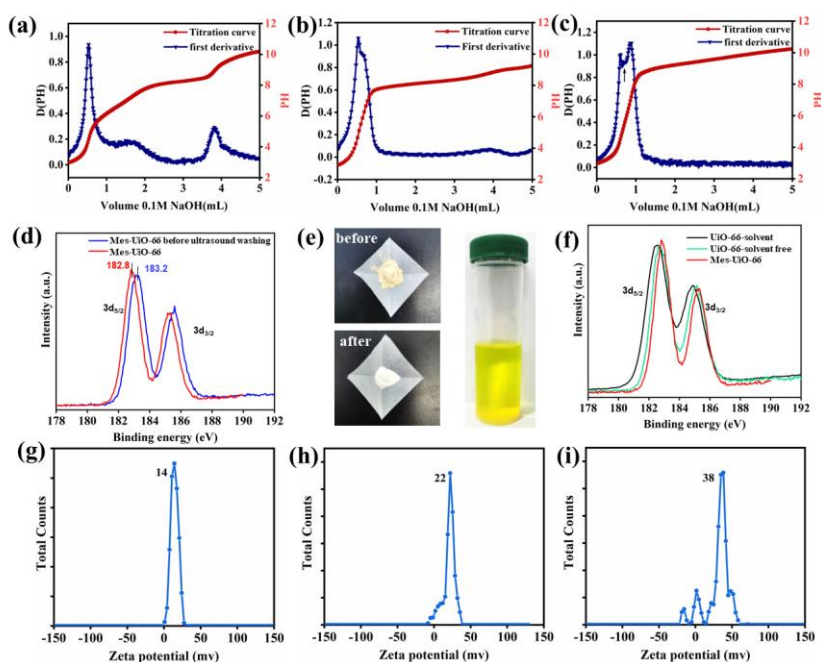
148 In order to clarify the role of FeCl_3 in the formation of hierarchical structure in Mes-UiO-
 149 66, other chloride salts, such as NaCl , KCl , CuCl_2 and CoCl_2 , are employed as substitute of
 150 FeCl_3 to prepare UiO-66. As shown in Figure 4a, the addition of CuCl_2 and CoCl_2 results in quite
 151 weak diffraction peaks of UiO-66. Although the samples synthesized with the addition of NaCl
 152 and KCl show well-resolved diffraction peak of UiO-66, but the low BET surface area and
 153 mesopore volume reveal the insufficient secondary pore structure (Figure 4b and Table S6).
 154 Moreover, FeCl_2 and $\text{Fe}(\text{NO}_3)_3$ are also used as additive to synthesize hierarchical UiO-66, but
 155 fail to gain crystalline UiO-66. Moreover, it can be indicated that the FeCl_3 does not just act as a
 156 salt template for mesopores, despite it is can be totally removed by ethanol washing after the
 157 synthetic process. The FeCl_3 should participate the crystal growth of UiO-66 and block it from
 158 growing big. The above results reveal the uniqueness of FeCl_3 in inducing hierarchical structure
 159 in the solvent-free synthesis of UiO-66, which may benefit from the easy interaction and
 160 dissociation of FeCl_3 with UiO-66 framework.



161
 162 **Figure 4.** (a) XRD patterns and (b) N_2 sorption isotherms of UiO-66 prepared with the addition
 163 of CoCl_2 , CuCl_2 , KCl and NaCl , (c) XRD patterns of samples prepared with the addition of
 164 Fe_2Cl and $\text{Fe}(\text{NO}_3)_3$.

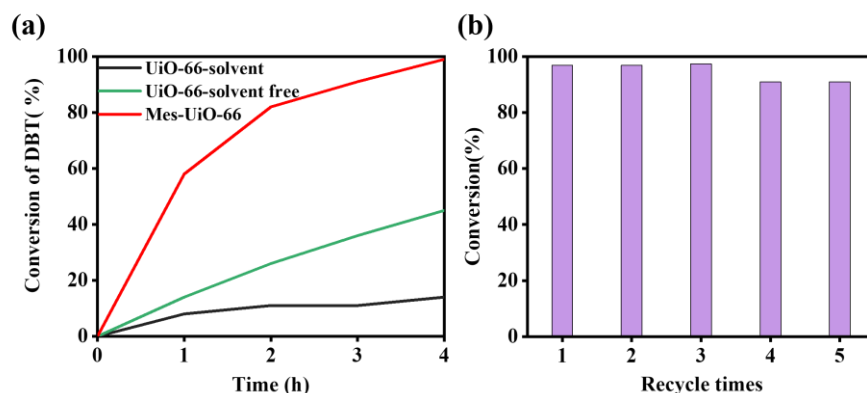
165 The impaction of the interaction of Zr center and the H₂BDC (linker) caused by the addition of
166 hetero metal salts should contribute to the small particle size of Mes-UiO-66. Thus, we first
167 practice the acid-base titration to investigate the effect of the FeCl₃ on the missing linker. Figure
168 5a-c show the acid-base titration curves and the corresponding first derivative curves. The acid-
169 base titration curve of UiO-66-solvent illustrates three inflection points (Figure 5a), which
170 belong to the μ_3 -OH proton stem from the framework of UiO-66, Zr-OH₂ proton and Zr-OH
171 proton caused by the missing linker, respectively. The UiO-66-solvent free shows similar
172 inflection points (Figure 5b). Interestingly, the Mes-UiO-66 only displays the inflection of μ_3 -
173 OH proton, the inflections of Zr-OH₂ proton and Zr-OH proton are hard to be detected (Figure
174 5c). Based on the first derivative curve, the number of missing linkers in Mes-UiO-66 is
175 calculated to 0.125, much smaller than in UiO-66-solvent (1.625) and in UiO-66-solvent-free
176 (2.155, Table S7-9). The results of the acid-base titration demonstrate that the FeCl₃ has slight
177 impaction on the initial combination of H₂BDC into Mes-UiO-66 to form small crystal. To
178 reveal the impaction of FeCl₃ on the Zr center, Zr 3d XPS is characterized over Mes-UiO-66
179 before and after ethanol washing. As shown in Figure 5d, the Zr 3d_{5/2} binding energy of the
180 sample without ethanol washing (just synthesized Mes-UiO-66) is confirmed as 183.2 eV, in
181 good agreement with the reported Zr(IV)-O-Fe(III) coordination in UiO-66, indicating the
182 temporary connection of Fe on the surface of Mes-UiO-66 particles by O bridge²⁶. However,
183 after the ultrasonic treatment in ethanol, the Zr 3d_{5/2} binding energy of Mes-UiO-66 negatively
184 shifting to 182.8 eV, suggesting the removal of the Fe. The stripping of the connecting Fe from
185 Mes-UiO-66 can also be confirmed by the color change of Mes-UiO-66 from yellow to white,
186 the bright yellow appearance of the used ethanol and the absence of the Fe-O vibration around
187 520 cm⁻¹ in FT-IR spectra (Figure 5e and Figure S10)²⁷⁻²⁹. Moreover, the Mes-UiO-66 still

188 exhibits higher oxidative state even after ethanol washing (Figure 5f), which results in the higher
 189 zeta potential of Mes-UiO-66 in ethanol (38 mV) compared with that of UiO-66-solvent (14 mV)
 190 and UiO-66-solvent-free (22 mV, Figure 5g-i), demonstrating the increasing positive charge on
 191 the surface of Mes-UiO-66. Therefore, the forming pathway of the hierarchical Mes-UiO-66 can
 192 be disclosed. Firstly, Zr species react with H₂BDC to form small UiO-66 particles. Then, the Fe
 193 species connect with the Zr at the surface of particles, which blocks the further growth of these
 194 small particles with H₂BDC. Finally, after the removal of Fe by ethanol washing, the relatively
 195 high surface positive charge enables loose aggregation of these particles, which lead to the
 196 abundant hierarchical structure.



197
 198 **Figure 5.** Acid-base titration curve and first derivative curve of (a) UiO-66-solvent, (b) UiO-66-
 199 solvent free and (c) Mes-UiO-66. (d) Zr 3d XPS of the just synthesized and ethanol washed Mes-
 200 UiO-66. (e) Digital images of the as synthesized and ethanol washed Mes-UiO-66 and the used
 201 ethanol. (f) Zr 3d XPS spectra of various samples. Zeta potential of (g) UiO-66-solvent, (h) UiO-
 202 66-solvent free and (i) Mes-UiO-66 in ethanol.

203 The oxidizing Zr center, abundant hierarchical structure and rich surface positive charge of
204 Mes-UiO-66 provide an ideal platform for the catalytic oxidation of aromatic large molecules.
205 Thus, the oxidation of dibenzothiophene (DBT) with tert-butyl hydroperoxide (TBHP) at 60 °C
206 is firstly employed to evaluate the catalytic performance of Mes-UiO-66. As shown in Figure 6a,
207 only 14% of DBT can be oxidized when UiO-66-solvent is used as catalyst. The UiO-66-solvent
208 free displays a DBT conversion of 38% after 4 h. The Mes-UiO-66 shows the best ability to
209 trigger the oxidation of DBT. After 4 hours almost all the DBT is removed by TBHP (DBT
210 conversion surpasses 99%). Moreover, the DBT conversion over Mes-UiO-66 just shows slight
211 decrease after five cycles (Figure 6b), indicating its robust stability. The superior catalytic
212 performance of Mes-UiO-66 can be attributed to its hierarchical structure that reduces the mass
213 transfer limitation and the rich surface positive that enhances the interaction with aromatic
214 molecule. Furthermore, the superior catalytic performance of Mes-UiO-66 is also confirmed in
215 the oxidation of cyclohexene (Figure S11).



216
217 **Figure 6.** (a) DBT conversion over Mes-UiO-66, UiO-66-solvent and UiO-66-solvent-free. (b)
218 stability tests of Mes-UiO-66.

219 CONCLUSION

220 In conclusion, we report a simple and efficient method for synthetic hierarchical UiO-66, by
221 grinding $\text{ZrOCl}_2 \cdot 8\text{H}_2\text{O}$, H_2BDC with $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ and crystalizing at $100\text{ }^\circ\text{C}$ without solvent.
222 The addition of FeCl_3 successfully limits the crystal size of UiO-66 to around $12 \pm 2\text{nm}$. After
223 the totally removing of Fe by ethanol washing, the loose aggregation of these small particles
224 leads to rich intercrystalline mesopores. The resultant hierarchical sample Mes-UiO-66 exhibits a
225 BET surface area of $1161.9\text{ m}^2\text{g}^{-1}$ and a state-of-art mesopore volume of $1.21\text{ cm}^3\text{g}^{-1}$. The
226 abundant hierarchical structure of Mes-UiO-66 enables superior catalytic performance in the
227 oxidation of DBT. The DBT conversion surpasses 99% after 4 h. Our finding may provide a
228 novel platform to efficiently fabricating hierarchical MOFs for large molecule catalysis.

229 **EXPERIMENTAL SECTION**

230 **Synthetic method**

231 Mes-UiO-66: $\text{ZrOCl}_2 \cdot 8\text{H}_2\text{O}$ (1.8mmol) and H_2BDC (1.8mmol) were mixed at room
232 temperature, manually ground in mortar for about 15 min, and then $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ was added in a
233 certain proportion for further grinding for 15 min. The obtained material was transferred to an
234 autoclave and crystallized at a certain temperature and time. After cooling to room temperature,
235 the resulting solid was washed with ethanol at $60\text{ }^\circ\text{C}$ and sonicated for 90 min, followed by
236 vacuum drying at $80\text{ }^\circ\text{C}$ for 12 h.

237 UiO-66-solvent: UiO-66-solvent was prepared by the method reported ¹⁹. The detailed steps
238 are shown in the Supporting Information.

239 UiO-66-solvent free: UiO-66-solvent mainly refer to the reported method ²². The detailed steps
240 are shown in the Supporting Information.

241 **ASSOCIATED CONTENT**

242 **Supporting Information**

243 The online version of this article (http://dx.doi.org/10.1007/s12274-***-****-) contains
244 supplementary material, which is available to authorized users.

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252 **Author Contributions**

253 The manuscript was written through contributions of all authors. All authors have approved the
254 final version of the manuscript.

255 **Notes**

256 The authors declare no conflict of interest.

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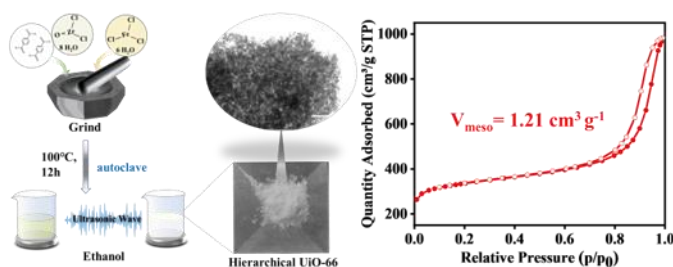
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356 **For Table of Contents Use Only**

357 **Crystal Growth Blocking Strategy Enabling Efficient Solvent-Free Synthesis of**
358 **Hierarchical UiO-66 for Large-Molecule Catalysis**

359 *Jianting Zhang¹, Jingbo Shi¹, Hang Zhao¹, Jinbo Bai², Xiaolin Li³, * and Kunyue Leng¹, **



360

361 we report a crystal growth blocking strategy for efficiently solvent-free synthesis of
362 hierarchical UiO-66. The addition of Fe salt in solvent-free system successfully limits the crystal
363 size of UiO-66 to between 12 ± 2 nm, the loose aggregation of these nanocrystals introduced
364 abundant intercrystalline mesopores to obtain well-crystallized stratified UiO-66. The resultant
365 Mes-UiO-66 exhibits well-developed hierarchical structure, the BET surface area reaches 1161.9
366 $\text{m}^2 \text{g}^{-1}$, and the mesopore volume attains a state-of-art $1.21 \text{ cm}^3 \text{g}^{-1}$.

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