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# Strong Metal-support Interactions in Photocatalysis: Fundamentals and Design Methods

Vien-Duong Quach,<sup>[a]</sup> Robert Wojcieszak,<sup>[b]</sup> and Mohamed Nawfal Ghazzal\*<sup>[a]</sup>



Engineering the composition and geometry of metallic sites has become a popular manner to boost reaction rate and control reaction selectivity in heterogeneous catalysis. Many studies have been devoted to enhancing the stability of metallic nanoparticles during catalytic reactions by dispersion on metal oxide supports such as TiO<sub>2</sub>, CeO<sub>2</sub> or Nb<sub>2</sub>O<sub>5</sub>. These supports not only modulate electronic properties and dispersion/stabilization of metallic nanoparticle but also influence catalytic selectivity, resulting in the so-called "strong metal-support interaction" (SMSI). In this minireview, we outlined the discovery and fundamentals of SMSI, as well as its extensive development over years. In addition, we summarized recent approaches

### 1. Introduction

Heterogeneous photocatalysts mediated by metallic nanoparticles, especially plasmonic metals, have shown superior properties and higher efficiency in the solar-to-energy conversion yield.<sup>[1-6]</sup> The metallic nanoparticles have been reported for boosting the photocatalytic efficiency of variable reactions, such as CO<sub>2</sub> reduction,<sup>[7-10]</sup> CO<sub>2</sub> methanation,<sup>[11-13]</sup> hydrogeneration,<sup>[14-16]</sup> and alcohol conversion.<sup>[17,18]</sup> However, further investigations are expected to understand the effect of photocatalyst design, the intrinsic activities of metallic nanoparticles, and the benefits of constructing strong interaction with reducible oxides as supports on the photocatalytic performance. Particularly, strong metal-support interaction (SMSI) between single metal atoms and semiconductors has become a central topic that much research involved to pinpoint SMSI effects on catalytic activities. This review aims to summarize several key factors of strong metal-support interaction, resulting in a brief and easily approachable view in this field.

The term "strong metal-support interaction" (SMSI) was first launched in 1978 by Tauster *et al.*<sup>[19]</sup> in order to explain the change in chemisorption properties of VIIIB group metals (Ru, Rh, Pd, Os, Ir, Pt) supported on TiO<sub>2</sub> when increasing reduction temperature. The research group used impregnation of metal salt solutions on commercial TiO<sub>2</sub> (P25 Degussa and Cab-O-Ti Cabot) to prepare catalysts. They recorded considerable H<sub>2</sub> and CO uptakes on catalysts at low reduction temperature (200 °C), demonstrating high dispersion of metal clusters. As the reduction temperature was increased to

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© 2023 The Authors. ChemNanoMat published by Wiley-VCH GmbH. This is an open access article under the terms of the Creative Commons Attribution Non-Commercial License, which permits use, distribution and reproduction in any medium, provided the original work is properly cited and is not used for commercial purposes. developed to induce the construction of SMSI between different metal nanoparticles and metal oxide supports. Associated characterization microscopic and spectroscopic techniques were emphasized. Despite being a prevalent concept in catalysis, the number of studies on SMSI in heterogeneous photocatalysis has been even in limitation. Herein, we highlighted the beneficial effects of SMSI on boosting photocatalytic activity for CO<sub>2</sub> reduction and H<sub>2</sub> evolution reactions. In general, despite some controversial aspects of the SMSI, this concept offers wide opportunities ahead and encourages researchers to rethink the local active site localization and photocatalyst design.

500 °C, the CO adsorption was found to decrease dramatically to near zero. The agglomeration of metal clusters due to the rise in temperature treatment and poisoning by impurities was found not inducing such a depletion in  $H_2$  and CO chemisorption. Thus, the encapsulation of metal particles occurring under  $H_2$  flow and at 500 °C was considered, despite the reversibility of the chemisorption. Indeed, metal particles should be accessible to gaseous molecules to experience reversible adsorption/desorption.<sup>[20]</sup>

The mechanism of SMSI encapsulation generally leans on the reduction of metal oxide support by H<sub>2</sub> at elevated temperature. It is well-accepted that the dissociation of H<sub>2</sub> molecules into H atoms on metal nanoparticle surface is the key driving force of reducing metal oxides.<sup>[21]</sup> The high temperature could promote redox reactions between metal nanoparticles and metal oxide supports, leading to changes in the oxidation state of metals.<sup>[22,23]</sup> Then, reduced metal cations are formed, which modulate Fermi levels of metal oxides, resulting in charge transfer as well as redistribution of electron density at the metal-support interface. The charge redistribution at the interface is the key of the construction of mixed oxides<sup>[24,25]</sup> and alloys,<sup>[25–27]</sup> which grow gradually and migrate onto metal nanoparticles, thus illustrating an overlayer of reduced metal oxide supports.<sup>[28]</sup>

In order to explain the unique chemisorption properties of the samples, Tauster et al.<sup>[19]</sup> suggested that the change in the chemical interactions between the noble metals and the support at surface, the so-called strong metal-support interaction (SMSI), is the main phenomenon modifying the chemisorption of CO. In particular, fully electron occupied "d" orbitals of noble metals are likely to overlap with vacant "d" orbitals of Ti<sup>[4+]</sup> cations, resulting in metal-metal bonding, under reducing conditions. The interaction between the "d" orbitals of donor-acceptor thermodynamically favourable to the formation of intermetallic bonds (Ti-M<sub>3</sub>), and presumably induces a change in the sorption properties at the surface of metal nanoparticles. The SMSI effect was expanded exclusively to reducible binary oxides including MnO<sub>2</sub>, V<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub> even when activation temperature was lower than 500°C.<sup>[29]</sup> The suppression of hydrogen and carbon monoxide were not recorded with SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, MgO, ZrO<sub>2</sub>, Sc<sub>2</sub>O<sub>3</sub>, HfO<sub>2</sub>, Y<sub>2</sub>O<sub>3</sub>. In other words, these non-reducible oxides typically exhibit no SMSI behaviour. The key solid-state transformations in reactive

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systems leading to the construction of strong metal-support interaction was the migration of reducible metal oxide support fully covering metal nanoparticles.<sup>[20,30]</sup> The metal oxide migration during the reduction process induces morphology change of metal particles. Large metal particles break up into smaller raftlike configurations that were fringed with the reduced surface of TiO2. In the meantime, the surface reduction of the support incentivized oxygen vacancies generation that would facilitate the proximity between metal atoms and reduced cations. The generation of oxygen vacancies at the metal/metal oxide overlayer was an indicative sign of the strong metal-support interaction in which titanium ions are the key. Several studies demonstrated chemical bonds between some VIIIB group metals, including Ni, Pt, Pd, Rh, and Ti<sup>[3+]</sup> cations. The TiO<sub>2</sub> overlayer was found to have variable effect regarding the main catalytic reaction. For instance, it was observed that the construction of SMSI using titanium oxide overlayer modify the chemisorption capacities towards H<sub>2</sub> and CO, as well as the catalytic activity in hydrogenolysis and dehydrogenation.<sup>[31]</sup> However, encapsulation of metal particles such as Pt in TiO2 or Nb2O5 was reported boosting the Fischer-Tropsch reaction rate of CO and H<sub>2</sub> to produce methane and other hydrocarbon.<sup>[32]</sup> While the construction of SMSI was restricted to group VIII metals, recent developments have been conducted and led to the extension of SMSI to other metallic nanoparticles. Over the past years, plenty of studies reported "SMSI" construction or effect to describe a change in the charge transfer, catalytic activity or molecules adsorption invoking the role of the metal/support interface, even when an encapsulation layer is not observed. In this review, we will mainly focus on variable strategies used for the



date in Material Chemistry at Institute of Physical Chemistry, CNRS UMR 8000, Université Paris-Saclay, France. He conducted an undergraduate study on Material Chemistry at Vietnam National University, Hanoi. He was then a scholar in the Erasmus Mundus Joint Master SERP + program (surface, electro, radiation, photochemistry) whereby he received double M.Sc. degrees in Physical Chemistry and Materials Science from Université Paris-Saclay (France) and Università di Genova (Italy). His work centers on strong metal support interaction-simulated core@shell nanostructures mediated by metallic nanoparticles and their applications on photocatalysis and photoelectrocatalysis.

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construction of SMSI using validated metal oxide overlayer covering metallic nanoparticles. We discuss herein the SMSI phenomenon on different metallic nanoparticles and the strategies to be applied to enhance the SMSI on various catalytic systems.

# 2. SMSIs regarding different metallic nanoparticles

### 2.1. SMSIs to platinum nanoparticles

Platinum nanoparticles might be one of the pioneering and widely-studied objects in the field of strong metal-support interaction catalysts.<sup>[19,33,34]</sup> Pt nanoparticles (larger size) are more likely to undergo SMSI encapsulation than Pt single atoms (smaller size) because of a difference in electronic structures. Belton and co-workers performed controlled coverage of Pt with thin titanium oxide and studied the electronic structures and charge transfers under ultrahigh-vacuum.<sup>[35]</sup> Even though the encapsulation of Pt nanoparticles has been investigated on different reducible oxides; such as Fe<sub>3</sub>O<sub>4</sub>.<sup>[36,37]</sup>  $TiO_2^{[38,39]}WO_3^{[39,40]}CeO_2^{[41-43]}$  etc., titanium dioxide remains by far the most studied support. However, the crystalline structure and receding of TiO<sub>x</sub> overlayers during catalytic reaction deteriorate the catalytic performance of support metal co-catalysts. The influence of annealing temperature and metal nanoparticle size on the encapsulation of Pt nanoparticles by support overlayers have been also studied. The migration of  $TiO_x$  (1 $\le$ x $\le$ 3) took place during encapsula-



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Mohamed Nawfal Ghazzal is an Associate Professor at the Université Paris-Saclay. He is now investigating strategies aiming at improving the light harvesting ability, the stability and designing photoactive material, such as core-shell, strong metal-support interactions, perovskites, graphdiyne for photocatalytic applications. tion process at a temperature above 150 °C even in the absence of H<sub>2</sub> as a reducing atmosphere,<sup>[44-47]</sup> while a sintering process would dominate the encapsulation process when annealing temperature is above 300 °C.<sup>[48,49]</sup> It was also reported that Pt nanoparticles and Pt single atoms could trigger SMSI with TiO<sub>2</sub> support but at different annealing temperatures.<sup>[50]</sup>

Wu et al.[51] studied the sensitivity of SMSI to the size of Pt particles supported on rutile TiO<sub>2</sub> by means of spectroscopic techniques. Pt nanoparticles are dominant at thicker coverage of 0.41 and 2.5 monolayer (ML), while Pt coverage of 0.06 ML generates both Pt clusters and Pt nanoparticles. Besides, 0.41 and 2.5 ML-Pt/TiO<sub>2</sub> showed the entire encapsulation of Pt nanoparticles into TiO<sub>2-x</sub> overlayers evidencing Pt-TiO<sub>2</sub> SMSI, and Pt nanoparticles on 0.06 ML-Pt/TiO<sub>2</sub> as well. However, Pt clusters do not exhibit the possible construction of SMSI. Moreover, no aggregation of Pt clusters into Pt nanoparticles was recorded on the sample after annealing at 727 °C. This indicated that high thermal stability of Pt clusters was due to the nucleation at surface oxygen vacancy sites of TiO<sub>2</sub> surface. The research group suggested that the influence of Pt particle size on SMSI might be related to their electronic structures. The charge transfer in Pt-TiO<sub>2</sub> SMSI typically operates from Pt to Ti<sup>[30]</sup> through Pt-Ti-O linkage. Indeed, Pt clusters with depleted charge in the "d" orbital and contracted lattice might be less capable of transferring charge to Ti, whereas larger Pt nanoparticles are more likely to form Pt-Ti-O interface and trigger the charge transfer through the construction of SMSI. The influences of particle size on SMSI, particularly Pt nanoparticles supported over Fe<sub>3</sub>O<sub>4.</sub><sup>[36]</sup> were also studied. The CO oxidation reaction, involving chemisorption of CO and O<sub>2</sub>, emphasized an important contribution of Langmuir-Hinshelwood mechanism between the two reactants on larger Pt nanoparticles (Figure 1a). Besides, the partial coverage of TiO<sub>x</sub> would stabilize Pt nanoparticles against sintering process at high annealing temperature.

Through a study on Pt-CeO<sub>2</sub> catalyst at microscopic level, Vayssilov et al.<sup>[26]</sup> identified two experimental hallmarks of SMSI: 1)- electron transfer from Pt NPs to CeO<sub>2</sub> support, and 2)- oxygen transfer from CeO<sub>2</sub> to Pt NPs. The electron transfer takes place on both surface and bulk of CeO<sub>2</sub>, while the oxygen transfer would be exclusive at the interface Pt-CeO<sub>2</sub>. In this study, the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO) in CeO<sub>2</sub> nanoparticles was regulated to CeO<sub>2</sub> (111) surface that HOMO and LUMO predominantly were assigned to O 2p orbitals and Ce 4f orbitals, respectively.<sup>[52]</sup> Introduction of Pt clusters fills the whole band gap HOMO-LUMO, as illustrated in Figure 1b. Because the highest occupied levels of Pt are in the vicinity of empty Ce 4f (LUMO), the author inferred electron transfer from Pt metal clusters to Ce to form Ce<sup>3+</sup> ions. The charge transfer would accompany with partial reduction of Pt clusters as well. Considering oxygen vacancy formation, theoretical calculation predicted Ce3+ ions deriving from oxygen detachment are detected exclusively at the Pt-CeO<sub>2</sub> interface. The changes in the oxidation state of CeO<sub>2</sub> due to molecular interactions, involving  $Ce^{3+}$  (4d<sup>10</sup>4f<sup>1</sup>) and  $Ce^{4+}$  (4d<sup>10</sup>4f<sup>0</sup>), were



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**Figure 1.** (a) Strong metal-support interactions between Pt nanoparticles and Fe<sub>3</sub>O<sub>4</sub> support. Reprinted with permission from.<sup>[36]</sup> Copyright American Chemical Society, 2020. (b) Density-of-states plot for the bare nanoparticle Ce<sub>40</sub>O<sub>80</sub> (dashed lines) and the nanoparticle system Pt<sub>8</sub>/Ce<sub>40</sub>O<sub>80</sub> (solid lines). Reprinted with permission from.<sup>[26]</sup> Copyright Springer Nature, 2011.

investigated by resonant photoelectron spectroscopy (RPES) at room temperature and then during annealing. The deposition of Pt on CeO<sub>2</sub> film axhibited the formation of Ce<sup>3+</sup> centres, which was not detectable before. The result evidences that Ce<sup>3+</sup> are generated on surface and in the region of Pt clusters. When elevating temperature over 300 °C, Ce<sup>3+</sup> concentration showed an increase. This is attributed to reverse spillover of oxygen from CeO<sub>2</sub> to Pt NPs, as expected through DFT calculation.

Baker et al.[21] demonstrated the role of Pt in dissociating  $H_2$  at 550 °C, which provides H atoms to reduce TiO<sub>2</sub> and form substoichiometric oxide (Ti<sub>4</sub>O<sub>7</sub>). In order to assess the necessary of H<sub>2</sub> dissociation in the generation of SMSI, the authors examined the behaviour of Ag nanoparticles on TiO<sub>2</sub> support with and without the presence of Pt nanoparticles. The specimen of Ag/TiO<sub>2</sub> outputted only a change in morphology of Ag NPs, including larger size and relatively dense and globular in outlines. Besides, TiO<sub>2</sub> support was in rutile form, no transformation was recorded. This confirmed SMSI interaction does not take place in this case. In contrary, (Aq + Pt)/TiO<sub>2</sub> exhibited remarkable change in the characteristics of these metal NPs. The metal crystallites of Ag became small, very thin, and especially angular in outline compared to the aforementioned globular outlines. With the presence of Pt, the rutile TiO<sub>2</sub> converted to Ti<sub>4</sub>O<sub>7</sub>, Ag NPs typically exhibited SMSI morphological features when supported on Ti<sub>4</sub>O<sub>7</sub>. The authors also inferred charge transfer between the reduced support and small Ag NPs through an EPR study that pointed out the formation of Ti<sup>3+</sup> and partial reduction of Ag<sup>+</sup> to Ag<sup>0</sup> during thermal treatment. In brief, SMSI interaction can be induced with metals which do not dissociate  $H_2$  if another metal was added to catch the mission, Pt in this specific study.

#### 2.2. Extension of SMSIs to IB group metals

Transition metals having work functions and surface energies exceeding a certain value would be likely to undergo the above-mentioned encapsulation and thus practice SMSIs.<sup>[53]</sup> Meanwhile, metals outside group VIIIB with lower work functions and surface energies was reported not to be sensitive to SMSI encapsulation.<sup>[19]</sup> However, over the past decade, research groups pioneered classical SMSI between reducible oxides (TiO<sub>2</sub>, ZnO) and IB group metals (such as Au and Ag),<sup>[54–58]</sup> leading to an extension of studies in SMSI afterward.

Mei et al.<sup>[54]</sup> reported the agglomeration of  $TiO_x$  in the vicinity of Au nanoparticles and the generation of Au/TiO<sub>2</sub> core-shell structures on an account of SMSI. The overlayer of  $TiO_x$  over Au probably depends on its loading. The authors inferred the relation between Au/TiO<sub>2</sub> composites with variable Ti loading and photocatalytic H<sub>2</sub> evolution (Figure 2). Particularly, less extent of the Au/TiO<sub>2</sub> core-shell structure would be accompanied by lower number of active sites and hence lower amount of H<sub>2</sub>. Indeed, a sample in which the agglomeration of TiO<sub>x</sub> was deliberately blocked produced negligible amount of H<sub>2</sub>. The results pointed out diffusion-controlled photo-deposition lead to smaller size of Au NPs due to strong metal-support interactions.

Tang *et al.*<sup>[56]</sup> reported the construction of classical strong metal-support interaction between Au NPs and TiO<sub>2</sub>, evidenced by suppression of CO adsorption, electron transfer from oxide support to metal nanoparticles, and gold encapsulation by reduced TiO<sub>2</sub> overlayer at high-temperature reducing environment. In this study, Au NPs were deposited on different polymorphs of titania, including P25 (Degussa), anatase, and rutile, these samples underwent high temperature reduction (HTR) treatment to 500°C in  $H_2/He$  (10% v/v) before being reoxidized at 400°C in  $O_2/He$  (10% v/v). *In situ* diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS,



**Figure 2.** Hydrogen evolution measured during the photodeposition process of Au (0.25 wt.%) on Ti(x)/SBA materials with increasing Ti loading. Reprinted with permission from.<sup>[54]</sup> Copyright American Chemical Society, 2013.

Figure 3a) indicated a band of absorbed CO on metallic Au at 2103 cm<sup>-1</sup>, the band intensity depleted with elevating reduction temperature and nearly faded after 500 °C. This evidenced the loss of CO adsorption sites due to SMSI encapsulation. Through X-ray photoelectron spectroscopy (XPS), the authors recorded the existence of negative Au surface species during reduction at over 400 °C and their reverse to the metallic state after reoxidation step, observed by a small shift of binding energy from 83.5 eV to 83.1 eV then back to 83.5 eV (Figure 3b). It was attributed to reversible electron transfer between TiO<sub>2</sub> and Au, which reveals the occurrence of SMSI, likewise, TiO<sub>2</sub>-supported platinum group metals previously reported. Besides, they highlighted an increase in Ti<sup>3+</sup> defects from electron paramagnetic resonance (EPR, Figure 3c), which evidenced the reduction of TiO<sub>2</sub> after high-temperature reduction treatments. It emphasized that titania reduction is a key factor in inducing SMSI encapsulation.

Among polymorphs of titania oxides, anatase was found to be the most susceptible to encapsulation, while rutile, the most stable phase, is less favourable to SMSI encapsulation. Surprisingly, the SMSI encapsulation of Au occurs not only for TiO<sub>2</sub> but also for Fe<sub>3</sub>O<sub>4</sub> and CeO<sub>2</sub> after a reduction treatment at 500 °C, which is much lower than 700 °C for  $Pt/CeO_2$  and Rh/CeO<sub>2</sub><sup>[59,60]</sup> Tang et al.<sup>[56]</sup> investigated the SMSI of other IB group metals (such as Cu and Ag) supported over TiO<sub>2</sub>. Experimental outputs showed reversible CO adsorption and encapsulation of Ag NPs at 500 °C, indicating classical SMSI between Ag NPs and TiO<sub>2</sub>. At the same time, the migration of the metal oxide support over Cu NPs was found difficult, due to a lack of TiO<sub>2</sub> reduction, evidenced by a decrease of Ti<sup>3+</sup>. The SMSI construction on Cu/TiO<sub>2</sub> required reduction treatment at higher temperatures (above 700  $^\circ\text{C}\textsc{)}.$  In the contrary to previous studies, which reported the limited application of Au and Ag in catalysis due to their poor thermal stability at high temperatures, classical SMSI in Au/TiO<sub>2</sub> demonstrated its excellent stability toward catalytic CO oxidation and even slow deactivation. The stability was ascribed to metal-promoted oxygen vacancy generated in Au/metal oxide interface.

Zhang et al.[61] provided solid experimental evidence that divulged two key factors of inducing strong metal-support interactions on Au/TiO<sub>2</sub> catalyst: 1)- particle size of supported Au NPs and 2)- facets of TiO2 supports. This study focused on Au NPs of 2 nm and 5 nm in size which were separately supported on different TiO<sub>2</sub> supports with different facets, including {100}, {001}, and {101}. The Au/TiO<sub>2</sub> SMSI was induced by high-temperature annealing treatment. The authors adjusted the size of Au NPs by changing the annealing environment.<sup>[46]</sup> They obtained Au NPs of 5 nm after doing treatment in air (oxidative O<sub>2</sub>) while the size of Au NPs was smaller, around 2 nm, when replacing air by  ${\rm H}_2$  reductive atmosphere. In fact, O2 in air would strengthen Ostwald ripening of metal NPs during annealing. The suppression of CO adsorption, investigated by in situ DRIFT spectroscopy, signifies the modification of surface structure after the annealing treatment (Figure 3d). 5 nm Au/TiO<sub>2</sub> samples recorded no CO adsorption, while much weaker peaks of CO adsorption still appeared in DRIFT spectra of 2 nm Au/TiO<sub>2</sub>

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**Figure 3.** (a) In situ DRIFT spectra of CO adsorption over Au/TiO<sub>2</sub> fresh and those reduced at elevated temperature from 200 °C to 500 °C (H200 to H500) before being reoxidized at 400 °C (H500 + O400). (b) Negatively shifted Au 4f XPS peak of Au/TiO<sub>2</sub> high-temperature reduction at 500 °C (H500). (c) EPR spectrum of Au/TiO<sub>2</sub> treated at 500 °C reveals a new signal at g = 1.930, assigned for Ti<sup>3+</sup> defects. Reprinted with permission from.<sup>[50]</sup> Copyright American Association for the Advancement of Science, 2017. (d) In situ DRIFT spectra of CO adsorption deduced on Au/TiO<sub>2</sub> catalysts with different size of Au NPs and various facets of TiO<sub>2</sub> support. Reprinted with permission from.<sup>[61]</sup> Copyright Wiley-VCH GmbH, 2021.

samples. Among facets of TiO<sub>2</sub>, samples with {101} facet showed no change of CO adsorption peak before and after the SMSI treatment, but samples with {100} and {001} did. TEM and HRTEM visualized SMSI encapsulation on the series of Au/ TiO<sub>2</sub> samples. Full encapsulation took place with 5 nm Au NPs whereas for 2 nm Au NPs the partial encapsulation occurred instead. In brief, Au NPs with the size of 5 nm are likely to undergo SMSI than those of 2 nm and the SMSI are more favourable on {100} and {001} facets of TiO<sub>2</sub>. DFT calculation explained that Au-TiO<sub>2</sub> SMSI promotes the reduction of TiO<sub>2</sub>, especially at a greater extent on {100} and {001}. In addition, Zhang *et al.*<sup>[61]</sup> detected downward surface band bending of TiO<sub>2</sub> after SMSI treatment due to XPS binding energy shifts of Ti 2p and O 1s. The shift of Au 4f, Ti 2p, and O 1s supposed that charge transfer derives from surface oxygen vacancies of reduced TiO<sub>x</sub> to Au NPs, not stoichiometric TiO<sub>2</sub> surface. Additionally, charge transfer at the Au-TiO<sub>x</sub> interface is less extensive in the case of 2 nm Au NPs because of contracting lattice and depleted "d" orbitals, this impeding Au-TiO<sub>2</sub> SMSI.<sup>[61]</sup>

# 3. Approaches to generate strong metal-support interactions

Various techniques supply high energy to induce defects and valence variation of metallic species that are key factors for the migration of transition metal oxide support, leading to the so-called strong metal-support interaction.

# 3.1. High temperature annealing in a reductive environment to induce permeable/porous SMSI overlayers

One of the most common methods to generate SMSI encapsulation is high temperature annealing in a reductive atmosphere. Beck et al.<sup>[62]</sup> induced SMSI encapsulation of Pt on  $TiO_2$  by annealing Pt/TiO\_2 catalyst at 600 °C in H<sub>2</sub> or O<sub>2</sub>. The encapsulation can be constructed through the migration of metal oxide support onto metal nanoparticles, and bimetallic alloy formation. This study also revealed the role of O<sub>2</sub> in the formation process and stabilization of the metal oxide overlayer. In situ transmission electron microscopy (TEM) was used to study the formation of SMSI over the time. After the samples were heated at 600°C under 1 bar of H<sub>2</sub>, platinum surface was gradually decorated and then fully covered within minutes. Once the samples were submitted to oxidative environment (under O<sub>2</sub> flow), an increase of the volume and crystallinity of metal oxide overlayer was observed. Electron energy loss spectra (EELS) of Ti under H<sub>2</sub> atmosphere showed a downward shift energy of 1 eV, reflecting partial titania reduction. TiO<sub>2</sub> overlayer covering Pt NPs was found to shrunk in size and Pt lattice at TiO2-Pt interface showed a deformation. When  $O_2$  oxidation atmosphere was replaced by  $H_{2r}$ structural defects appeared and the reconstruction of Pt NPs took place followed by a crystalline layer covering up of metal NPs. This layer involved multilayer of titania, which suggested a multistep support migration. In situ XPS provided insight on the surface-sensitivity of Pt/TiO<sub>2</sub> catalyst. Under H<sub>2</sub> atmosphere, Pt was found in fully reduced state, assigned with a peak of Pt 4f<sub>7/2</sub> at 71.3 eV. The binding energy peak shifted of 0.8 eV to 72.1 eV when inlet gas was changed to O<sub>2</sub>. The authors inferred the formation of Pt-O at the interface between Pt and TiO<sub>2</sub> overlayer, whereby electron transfer could take a place among the metal species and the metal oxide support. Near edge X-ray absorption fine structure (NEXAFS) - an extremely surface sensitive technique confirmed the amorphous phase of TiO<sub>2</sub> overlayer under H<sub>2</sub> treatment while a typical spectrum of rutile was observed under O<sub>2</sub>. The annealing method did not generate TiO<sub>2</sub> anatase, which is the most active phase and common-used in photocatalysis. Pt-Ti alloy formation, which is the second mechanism for SMSI encapsulation, was further clarified by in situ X-ray diffraction. A shift of Bragg peak of Pt into higher angles derived from lattice contraction which is attributed to the incorporation of Ti into Pt crystal structure. In other words, high temperature annealing under H<sub>2</sub> generated Pt-Ti bimetallic bonds as expectation.

Zhao et al.<sup>[63]</sup> used H<sub>2</sub> reduction treatment at high temperature to induce SMSI on TiO<sub>2</sub>-supported Ir NPs. Strong metalsupport interaction among Ir NPs and TiO<sub>2</sub> support was found effective on boosting the pairwise selectivity of propene hydrogeneration due to electronic and geometric blocking effects. Scanning transmission electron microscopy - high angle annular dark field (STEM-HAADF) images illustrated the discrepancy between two fresh catalyst specimens, which underwent calcination at 200 °C and 500 °C. Ir NPs observed larger size and flatter shape after reduction at 500 °C that could be a result of SMSI. In addition, any traces of TiO<sub>2</sub> amorphous layer were detected on the top of Ir NPs by STEM - annular bright field (STEM-ABF) imaging. <sup>1</sup>H NMR spectroscopy recorded reversible changes in the pairwise selectivity on a recycled catalyst specimen, which was re-oxidized in air at 400 °C and then reduced at 200 °C in H<sub>2</sub>. Pairwise selectivity and overall catalytic activity remained unchanged after the recycling treatment. These findings further confirmed the presence of SMSI by means XPS analysis of the changes in oxidation states of metal species related to SMSI. Indeed, it would be challenging to determine the oxidation state of Ir because of the overlap between Ir 4f and Ti 3s peaks. Surprisingly, XPS spectra registered a broaden of Ti  $2p_{3/2}$  peak (at 458.3 eV) in the case of Ir/TiO<sub>2</sub> annealed at 500 °C in H<sub>2</sub>, ascribed to the reduction of Ti<sup>4+</sup> to Ti<sup>3+</sup>, which is a typical hallmark of SMSI formation.

Whilst previous studies identified high temperature treatment in a reductive environment as a key dimension of generating SMSI encapsulation, Liu et al.<sup>[64]</sup> has taken a reversed approach which constructed a permeable TiO<sub>2</sub> overlayer onto Au NPs under oxidative atmosphere, illustrated in Figure 4 (pathway a-d-e). Compared to the conventional pathway a-b-c, the reversed approach offered permeable SMSI overlayer without receding and re-oxidizing after being retreated in air over 400 °C. The Au/TiO<sub>2</sub> catalyst was synthesized using deposition-precipitation method and then subjected to a modification step with melamine at 600 °C in N<sub>2</sub> followed by calcination at 800 °C in air. High-resolution transmission electron microscopy (HRTEM) depicted that the migration of TiO<sub>2</sub> encapsulated Au NPs after the calcination and melamineinduced encapsulation seems to withstand the sintering of Au NPs. In fact, the size of Au NPs observed an insignificant extension from 3.6 nm at the beginning to 7.5 nm after the final step. The metal nanoparticles without passing the melamine pre-treatment enlarged up to 32.6 nm in size because of an intensive sintering. However, the morphology and encapsulation of Au NPs with the size of less than 2 nm was not presented in this article.

This research assumed that high-temperature annealing with melamine in oxidative environment would encourage electron transfers from Au NPs to  $TiO_2$  support, resulting in the reduction of  $Ti^{4+}$  to  $Ti^{3+}$  and positive charge on Au metal species, confirmed also by EELS and DRIFTS.<sup>[64]</sup> In addition, X-ray absorption fine structure techniques, including EXAFS and XANES, revealed almost fully occupied d orbital of Au and Au-Ti bond with a similar bond length to AuTi alloy. This finding could explain the migration of  $TiO_2$  onto Au NPs. In other

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**Figure 4.** Two pathways to reach out  $TiO_x$  overlayer under high-temperature treatment: 1, Conventional SMSI encapsulation pathway starts from bare Au NPs deposited on  $TiO_2$  support (a), followed by annealing at 500 °C in 10 vol % H<sub>2</sub>/He to form an impermeable  $TiO_x$  overlayer (b). The SMSI-induced overlayer recedes as retreating at 400 °C under oxidative condition (c). 2, Melamine-induced SMSI encapsulation pathway involves annealing bare  $TiO_2$ -supported Au NPs (a) with melamine at 600 °C in N<sub>2</sub> to achieve a permeable  $TiO_x$  overlayer (d). This melamine-modified overlayer remains stable as retreating in air up to 600 °C (e). Reprinted with permission from.<sup>[64]</sup> Copyright Springer Nature, 2019.

words, SMSI encapsulation could be induced by a crystalline reconstructing of  $TiO_2$  in the vicinity of Au NPs under an appropriate condition.  $TiO_2$  overlayer was in anatase phase, which turns to rutile phase after calcination at 800 °C. While no trace of anatase phase was detected in the sample without undergoing melamine pre-treatment, the melamine-induced  $TiO_2$  overlayer preserved 12% of anatase. Although the author reported SMSI between Au and  $TiO_2$  is one of the key contributors to resist  $TiO_2$  phase conversion, this approach suffers from the limited portion of the most active  $TiO_2$  phase that hinders its availability in photocatalysis.

A great deal of previous research into SMSI has gone through high-temperature treatment. However, encapsulation processes under these conditions are generally uncontrollable and prompt, leading to limited exposure of active sites (Figure 5a). In fact, SMSI encapsulation layer are necessarily permeable to allow reactants to reach active sites, while the overlayer should fully encapsulate metal nanoparticles to preclude them from sintering and agglomeration, to cite a few, during catalytic reactions. To overcome these obstacles of the conventional SMSI, Wu et al.[65] proposed a porous yolkshell structure of Pd-Fe<sub>3</sub>O<sub>4</sub> which Pd atoms were expected to migrate into Fe<sub>3</sub>O<sub>4</sub> crystalline structure (Figure 5b). During the procedure, Pd-FeO<sub>x</sub> NPs were annealed at 300°C under a mixture of  ${\rm H_2}$  and Ar atmospheres to trigger SMSI, in the meantime, another sample was treated at the same temperature but in air for comparison.

X-ray diffraction patterns indicated the amorphous phase of iron oxide shell as treating in air, while characteristic patterns of  $\gamma$ -Fe<sub>3</sub>O<sub>4</sub> lattice appeared in the case of sample treated under H<sub>2</sub> reductive condition.<sup>[65,66]</sup> The g-Fe<sub>3</sub>O<sub>4</sub> shell showed a microporous structure in HR-STEM images. The





**Figure 5.** A reversed route to induce SMSI on Fe<sub>3</sub>O<sub>4</sub>@Pd core-shell nanostructure under H<sub>2</sub> reductive environment. The approach aims to compensate shortcomings of conventional SMSI. Reprinted with permission from.<sup>[65]</sup> Copyright Springer Nature, 2020.

research inferred that  $H_2$  might reduce metal oxides at high temperature to generate oxygen vacancies in the oxide matrix. Meanwhile, the crystallization accompanies with redistribution of atoms in oxide grains that might accelerate the formation of oxygen vacancies, and thus a crystalline structure showing voids and cavities. Reactants are more likely to diffuse into pores and expose to metallic active sites, while the core-shell structure still guarantees SMSI such an electron transfer from Pd to Fe<sub>3</sub>O<sub>4</sub>.

A great number of groups have paid effort on SMSI of platinum-group metals, mainly Pt and Pd, under H<sub>2</sub> reductive condition. To escape from these boundaries, Li *et al.*<sup>[67]</sup> conducted research on Ni and anatase TiO<sub>2</sub> and pinpointed the role of NH<sub>3</sub> treatment together with H<sub>2</sub> treatment in SMSI formation. Ni species were deposited on two different

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supports: 1)- anatase TiO<sub>2</sub> treated at 600 °C in NH<sub>3</sub> and 2)anatase TiO<sub>2</sub> without experiencing the NH<sub>3</sub> treatment, both samples went through annealing in  $H_2$  at 500 °C (Figure 6a). Surprisingly, the sample which Ni NPs were deposited on NH<sub>3</sub>treated anatase TiO<sub>2</sub> showed a superior catalytic activity and selectivity for CO<sub>2</sub> methanation, raising a question about the influence of NH<sub>3</sub> on the surface catalyst modification. XRD patterns showed no change in the anatase TiO<sub>2</sub> phase before and after NH<sub>3</sub> treatment. X-ray photoelectron spectroscopy showed the presence of nitrogen component as N 1s signal was recorded for anatase TiO<sub>2</sub>. The N1s signal disappeared upon exposition to H<sub>2</sub> reduction, demonstrating that N-doping is unrelated to the enhancement of catalytic reaction yield. EPR studies showed an obvious signal at g-value of 1.987, which is typical of Ti<sup>3+</sup> in bulk,<sup>[68,69]</sup> for NH<sub>3</sub>-treated anatase TiO<sub>2</sub>, whilst none was recorded for anatase TiO<sub>2</sub> (Figure 6b). Further investigation reported a proportional relation between the intensity of the Ti<sup>3+</sup> EPR signal and CO<sub>2</sub> conversion yield (Figure 6c). HAADF-STEM, quasi - in situ XPS, and XRD provided some evidence of SMSI encapsulation. Ni NPs were efficiently dispersed over anatase TiO<sub>2</sub> without through NH<sub>3</sub> treatment, resulting in smaller crystal size of 9.5 nm (14.2 nm observed for Ni NPs on NH<sub>3</sub>-treated anatase TiO<sub>2</sub>). A larger size of Ni NPs and reduced Ti species (Ti<sup>3+</sup>) in bulk might suppress the migration of TiO<sub>2</sub>, leading partial encapsulation of Ni active sites. When skipping NH<sub>3</sub> treatment, Ti species remain in Ti<sup>4+</sup> oxidation state, which were reduced by  $H_2$  in the annealing step that would generate full encapsulation of Ni. In this study, the authors considered that TiO<sub>2</sub> overlayer could inhibit the exposure of reactants to Ni active sites, and thus deteriorate catalytic activities.

#### 3.2. The SMSI construction mediated by adsorbates

Matsubu et al.<sup>[70]</sup> reported the customization of SMSI on TiO<sub>2</sub>supported Rh catalyst to tune selectivity of CO<sub>2</sub> reduction reaction through use of treatment in a CO<sub>2</sub>-H<sub>2</sub> mixed environment at 150–300 °C. The catalytic CO<sub>2</sub> reduction on Rh/TiO<sub>2</sub> catalyst (Figure 7a), which experienced a treatment in  $CO_2$ :  $H_2$ : He (20:2:78 v/v ratio) at 250 °C, registered a decrease in CH<sub>4</sub> production rate but an increase in CO production rate. At the beginning, the disintegration of Rh NPs into isolated Rh atoms was supposed. However, ex situ scanning electron microscopy does not show any change in Rh structure. In situ diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS) revealed a high coverage of HCO<sub>x</sub> adsorbates, including formate and bicarbonate-like species, on TiO<sub>2</sub> surface during the 20CO<sub>2</sub>:2H<sub>2</sub> treatment. At temperature higher than 100°C, the adsorbate coverage could generate oxygen vacancies on TiO<sub>2</sub> surface, resulting in a migration of TiO<sub>2</sub> onto Rh active sites. In situ STEM and in situ EELS showed that the TiO<sub>2</sub> overlayer exists in amorphous phase, involving Ti<sup>3+</sup> (30%) and Ti<sup>4+</sup> (70%). In contrary to a conventional SMSI overlayer exclusively induced in H<sub>2</sub> at 500 °C, HCO<sub>x</sub>-induced SMSI overlayer was permeable, porous, and partially reduced. In addition, HCO<sub>x</sub> adsorbates coordinated with Ti species and stabilized SMSI encapsulation during catalytic reactions.

The reduction of metal oxide support has been identified as a prerequisite for SMSI encapsulation. However, critical conditions, involving high temperature at 100 °C in either generally inert or reductive atmosphere, became main obstacles for experimental approaches of inducing SMSI. Some research reported that alcohol could foster the reduction of metal oxides,<sup>[72,73]</sup> especially below 100 °C in the case of TiO<sub>2</sub>.<sup>[74]</sup> In order to address the aforementioned issues, Polo-Garzon *et al.*<sup>[71]</sup> studied alcohol-induced SMSI at low temperature. TiO<sub>2</sub>-supported Au catalyst was treated in *iso*-propanol or methanol at 30 °C prior to another treatment in argon at



**Figure 6.** (a) Schematic illustration of inducting SMSI in high-temperature  $H_2$  atmosphere on anatase TiO<sub>2</sub> support with and without undergoing ammonia treatment. (b) EPR spectra recorded at 110 K for anatase TiO<sub>2</sub> and anatase TiO<sub>2</sub> after being annealing in NH<sub>3</sub> atmosphere within different durations. (c) The correlation between EPR signal intensity and CO<sub>2</sub> conversion yield. Reprinted with permission from.<sup>[67]</sup> Copyright American Chemical Society, 2019.

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**Figure 7.** (a) Schematic illustration of adsorbate-induced SMSI formation: 1, bare  $TiO_2$ -supported Rh nanoparticles favours to reduce  $CO_2$  to  $CH_4$ . 2, Treatment in  $CO_2$  and  $H_2$  produces a permeable overlayer consisting of  $TiO_x$  species and  $HCO_x$  adsorbates. 3, The SMSI overlayer remains intact during and after catalytic  $CO_2$  reduction reactions. Reprinted with permission from.<sup>[70]</sup> Copyright Springer Nature, 2016. (b) 95% of Au active site encapsulated by  $TiO_2$  support after a treatment in iso-propanol at room temperature followed by desorption at 150 °C. (c) The time-on-stream (TOS) rate of CO conversion after treating in  $O_2$  at different temperature to retract  $TiO_2$  overlayer, divulging the so-called "self-activating" (right). Reprinted with permission from.<sup>[71]</sup> Copyright American Chemical Society, 2020.

300 °C to flush adsorbed alcohol molecules. A decrease of infrared spectroscopic (IR) peaks of CO adsorption evidenced the inaccessibility of Au active sites. The authors further studied to elucidate at which process, either exposure to 2propanol at 30°C or desorption of 2-propanol by Ar flushing at elevated temperature. They found metal sites were only partially covered during the first step. A full coverage was really reached upon 150°C when the desorption/decomposition of alcohols might trigger chemical transformation of catalyst surface. The alcohol-modified catalyst then experienced the final treatment in  $O_2$  at around 250 °C (Figure 7b). The authors recorded partial receding of the TiO<sub>x</sub> overlayer on Au species, unfolding a larger number of catalytic sites and thus curtailing catalytic deactivation, which was characterized by the increasing red curve in time-on-stream rate (TOS, Figure 7c).

#### 3.3. Laser treatment and photoinduced SMSI encapsulation

So far, universal methods of constructing SMSI encapsulation in various environment under ambient conditions are still in high demand. Laser ablation has known as a universal, green, and practical method in synthesizing metal-mediated nanomaterials. However, nanosecond and longer pulse irradiation generally suffer from high temperature and high pressure that might initiate unfavourable phase transformation and/or sintering of metal active sites. Thus, ultrafast pulse laser, involving femtosecond and picosecond, could address the drawbacks based on its ultrahigh intensity but ultrashort irradiation time. Taking these advantages of femtosecond laser excitation (35 fs), Zhang et al. succeeded in inducing SMSI on CeO<sub>2</sub>-supported Pt NPs, evidenced by porous/permeable overlayer of CeO<sub>2</sub> onto Pt NPs.<sup>[75]</sup> When triggering the pulse laser at the interface, local plasmon resonance of Pt contributes to electric field enhancement, resulting in ionization of CeO<sub>2</sub>. In particular, the flux of photons is likely to excite electrons to conduction band and leave holes at valence band. Surface oxygen atoms would donate electrons to compensate the electronic imbalance, leading to a high probability to be peeled off. The mechanism was shown in Figure 8a. As a result, surface defects, including  $Ce^{3+}$  and oxygen vacancies were detected by EPR and XPS. Such CeO<sub>x</sub> species tend to reorganize its structure and then migrate to encapsulate partially Pt NPs. This study showed that the thickness of the metal oxide overlayer was proportional to laser exposition time.<sup>[44]</sup> The author recorded none of these phenomena on laser-treated pure CeO2. This emphasized the role of the enhanced electric field at metal - metal oxide interface in the formation of surface defects. The study reported SMSI encapsulation of metal NPs not only on reducible metal oxide such as Pt/TiO<sub>2</sub>, Pd/TiO<sub>2</sub>, and Au/TiO<sub>2</sub>, but also non-reducible metal oxides, including Pt/Al<sub>2</sub>O<sub>3</sub>, Pt/SiO<sub>2</sub>, and Au/MgO.

Another photochemistry-driven methodology was reported for the construction of SMSI on TiO<sub>2</sub>-supported Pd catalyst under ambient conditions.<sup>[14]</sup> The SMSI generation





**Figure 8.** (a) Ultrafast laser induces oxygen vacancies and  $Ce^{3+}$ , leading to SMSI encapsulation of Pt NPs by  $CeO_x$  overlayer. Reprinted with permission from.<sup>[75]</sup> Copyright Springer Nature, 2021. (b) 1. Two-step pathway to achieve SMSI encapsulation of Pd NPs supported over TiO<sub>2</sub> in isopropanol under UV irradiation 2. STEM-ADF images and EELS spectrum at Pd site show no coverage of TiO<sub>2</sub> 3. Suppressed CO adsorption on Pd sites after 30 minutes of UV illumination 4. STEM-ADF images and EELS spectrum at Pd site after UV treatment demonstrates TiO<sub>2</sub> overlayer onto these metal sites. Reprinted with permission from.<sup>[14]</sup> Copyright American Chemical Society, 2021.

leans on the formation of Ti<sup>3+</sup> and oxygen vacancies which were triggered by photo-excited charge carriers (reductive electrons and oxidative holes). Instead of using ultrafast laser, a dispersed solution of Pd/TiO<sub>2</sub> in isopropanol was irradiated by a UV lamp at room temperature (Figure 8b.1). Photogenerated electrons in TiO<sub>2</sub> bulk could diffuse to surface and reduce  $Ti^{4+}$  to  $Ti^{3+}$  species, which are then dissolved in the solution. The Ti<sup>3+</sup> species accumulate on Pd NPs due to Ti<sup>3+</sup> -metal interaction, resulting in Pd-O<sub>vacant</sub>-Ti<sup>3+</sup> sites at the interface. These interfacial sites seem to be the prerequisite for SMSI encapsulation. The STEM-ADF images and EELS mapping evidenced the formation of TiO<sub>2</sub> overlayer onto Pd species (Figure 8b.2 and Figure 8b.4). Through XRD patterns, the author recorded the anatase phase of the overlayer and metallic phase of Pd (111), while the characteristic peak of Pd (111) showed a downward trend over UV irradiation time. Surface-induced amorphization, decreased crystalline size, and lattice expansion of Pd NPs under a long UV-irradiation time. The suppression of CO adsorption, investigated by CO-DRIFTS, once again confirmed the SMSI encapsulation of Pd NPs. Particularly characteristic peaks of linear CO adsorption and bridged CO adsorption on edge and terrace of  $Pd^0$  (green, orange, and greenish beige) were observed in CO-DRIFTS spectrum of Pd/TiO<sub>2</sub> after 30 minutes of irradiation (Figure 8b.3), while the peak of threefold-hollow-bonded CO remained insignificant (pink).

#### 3.4. Wet chemistry to construct SMSI encapsulation

In general, conventional method to create SMSI overlayer involves the migration of reducible metal oxide support onto metal nanoparticle active sites through redox treatments at high temperature. However, those metal NPs are likely to undergo agglomeration and sinter into bigger clusters. In addition, a receding of the metal oxide overlayer could operates during the catalytic reaction. Zhang *et al.* proposed a "wet-chemistry" SMSI (wcSMSI) to produce TiO<sub>x</sub> overlayer onto Au NPs but at room temperature.<sup>[55]</sup> TiO<sub>2</sub>-supported Au catalyst was firstly synthesized by deposition-precipitation method

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(Figure 9a). Then, a TiO<sub>x</sub> colloid solution, prepared from the gradual hydrolysis of TiCl<sub>3</sub> in a basic media NaHCO<sub>3</sub>, was dropped into the Au/TiO<sub>2</sub> in liquid phase. In this case, Au NPs typically play a role of nucleation seeds for TiO<sub>x</sub> encapsulation. This study reported the mean diameter of Au NPs at 2 nm, which is smaller than that reported in previous articles.<sup>[56,76]</sup> Operando FTIR unveiled partial coverage of Au NPs by TiO<sub>x</sub> overlayer and electronic interaction between Au NPs and TiO<sub>x</sub>, evidenced by weaker band of CO adsorption and their shift under different treatments. In fact, the authors identified Au-Ti<sup>3+</sup> interaction as the key for the wcSMSI generation, which was highlighted by XPS. Looking at Au 4f region in XPS spectra (Figure 9b) pre-treated Au NPs in O<sub>2</sub> showed two peaks at 83.8 eV and 84.7 eV assigned to metallic Au<sup>o</sup> and oxidized  $Au^{\delta+}$  species. These peaks became negligible when adding Ti<sup>3+</sup> into Au colloid, while characteristic peaks of reduced  $Au^{\delta-}$  species became predominant.

Meantime, Ti 2p region in XPS spectra (Figure 9c) exhibited the disappearance of  $Ti^{3+}$  peaks at 457.1 eV. This could be explained by the reduction of Au species by  $Ti^{3+}$ .

Another research from our group introduced a facile procedure to obtain SMSI via soft chemistry.<sup>[77,78]</sup> The key idea lies in depositing substoichiometric  $TiO_x$  onto Au NPs which are both supported on  $SiO_2$  microspheres, constructing core@-shell nanostructure (Figure 9d). Compared to amorphous and non-stoichiometric  $TiO_x$  overlayer induced by classical method under reductive and high temperature conditions, the soft-chemistry approach offered SMSI-like  $TiO_x$  layer in photoactive anatase phase with adjustable thickness (Figure 9e). Combining with local surface plasmonic resonance of Au NPs, the authors firstly reported the use of soft-chemistry SMSI concept

in photocatalysis. Under UV-visible illumination, photogenerated electrons were excited to jump to conduction band of TiO<sub>2</sub>. These electrons transferred from TiO<sub>2</sub> conduction band to Au NPs, which act as electron reservoirs due to metal semiconductor Schottky junction. Surprisingly, the electrons were injected back from Au NPs to TiO<sub>2</sub> even accompanied with the injection of gold LSPR "hot" electrons, evidenced by time-resolved microwave conductivity analysis. In addition, the catalysts prepared by soft-chemistry method were also active in thermal catalysis. The SiO2@Au@TiO2 was studied in base-free furfural oxidation at relatively low temperature (110 °C). The enhanced catalytic behaviour of these materials compared to classical supported Au catalysts suggests an essential role of the TiO<sub>2</sub> overlayer on Au nanoparticles. Encapsulation prevents Au not only from leaching but also from Au deactivation by irreversible furfural adsorption.<sup>[79]</sup> This soft-chemistry left a perspective on developing SMSI encapsulation on other metals such as Pt, Cu, and Ni.

#### 3.5. Mechanochemistry-driven SMSI encapsulation

Solid-state approaches to chemical synthesis and materials science have gained more and more attention in last years. Mechanochemistry became potential to carry out solid-state reactions because it is clean, safe, and facile.<sup>[80]</sup> In fact, simple methods, which are able to induce controllable SMSI, specifically under ambient conditions, and tailor encapsulation extent as well as SMSI structures, are in high demand. As a result, several research groups developed mechanochemistry-driven approaches to construct SMSI. Zhang *et al.*.<sup>[81]</sup> reported



**Figure 9.** (a) Wet-chemistry SMSI encapsulation of Au NPs through Ti<sup>3+</sup>-Au interaction in liquid phase. (b) Au 4f XPS spectra of Au NPs when pre-treated in O<sub>2</sub> (above) and adding Ti<sup>3+</sup> (below). (c) Ti 2p XPS spectra of TiO<sub>x</sub> colloid before (above) and after (below) interacting with Au NPs. Reprinted with permission from.<sup>[55]</sup> Copyright American Chemical Society, 2019. (d) Schematic illustration of constructing SMSI-like TiO<sub>x</sub> overlayer on Au NPs via a soft-chemistry pathway. (e) HRTEM image of an encapsulated Au NPs. Reprinted with permission from.<sup>[77]</sup> Copyright Royal Society of Chemistry, 2020.

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SMSI encapsulation on  $TiO_2$ -supported Pd and TiC-supported Pd electrocatalysts through the use of mechanical milling under ambient atmosphere (Figure 10a). HRTEM illustrated  $TiO_x$  overlayers on Pd/TiO\_2 in which their thickness could be adjusted by milling time, while XRD determined amorphous phase of these overlayer. In addition, SMSI imposed some modification on electronic structure of Pd NPs. Particularly, the decreasing intensity of Pd<sup>2+</sup> peaks and the increasing of Pd<sup>8-</sup> peaks in XPS spectra over milling time was observed (Figure 10b). A similar pattern was seen on Ti 2*p* with a dominant intensity of low valence state and a fading intensity of high valence state (Figure 10c). These experimental outputs demonstrate a reduction of both Ti and Pd. In other words, SMSI might accelerate a reduction of TiO<sub>2</sub> and hence a reduction of Pd due to charge transfer.

Recently, Ti<sup>3+</sup> and oxygen vacancies, which are the key of SMSI encapsulation, were generated on Pd/anatase TiO<sub>2</sub> catalyst using a NaBH<sub>4</sub>-assissted mechanochemical method under ambient conditions (Figure 10d).<sup>[82]</sup> Strong reducing agent such as NaBH<sub>4</sub> can reduce Pd<sup>2+</sup> and Ti<sup>4+</sup> to Pd<sup>0</sup> and Ti<sup>3+</sup>, respectively, as evidenced by XPS. SMSI encapsulation of Pd NPs by TiO<sub>x</sub> overlayer leaded to lower chemisorbed CO amount. The CO adsorption showed a downward trend upon increasing ball milling time, reflecting its capability of regulating encapsulation extent. High-speed vibrant ball milling encourages energetic solid-state interactions, resulting in rapid transfer of these reactive intermediates. Indeed, Ti<sup>3+</sup>

could adsorbed on Pd NPs during high-interactive conditions. Combining with oxygen vacancies,  $Pd-O_v-Ti^{3+}$  sites could be generated and thus SMSI encapsulation (fig. 10e). This study demonstrated the mechanochemistry-driven pathway could initiate SMSI on different metal NPs (Pd, Ru, and Pt) supported on a variety of reducible metal oxides, including TiO<sub>2</sub>, ZnO, Fe<sub>3</sub>O<sub>4</sub>, and Nb<sub>2</sub>O<sub>5</sub>.

# 4. Strong metal-support interaction in photocatalysis

So far, strong metal-support interaction has been largely neglected in photocatalytic reaction, and only few studies have been reported. Lin *et al.*<sup>[33]</sup> constructed SMSI between Pt nanoparticles homogenously dispersed on oxygen-vacancy defective TiO<sub>2</sub> crystalline for gas-phase photocatalytic CO<sub>2</sub> reduction. Pt NPs were partially covered by disordered TiO<sub>2</sub> layers through a high-temperature flame aerosol synthesis. While CO<sub>2</sub> photoreduction on bare TiO<sub>2</sub> produced CO only, CH<sub>4</sub> was detected additionally from the photoreduction reactions on Pt-modified TiO<sub>2</sub>. Indeed, SMSI between Pt and TiO<sub>2</sub> obviously tuned the selectivity of CO<sub>2</sub> photoreduction thanks to electron transfer from TiO<sub>2</sub> to Pt NPs. The authors reported that Pt-supported oxygen-vacancy induced TiO<sub>2</sub> photocatalyst resisted catalytic deactivation after eight-hour



**Figure 10.** (a) Encapsulation of  $TiO_x$  overlayer onto Pd NPs supported on  $TiO_2$  P25 or TiC through mechanochemical ball milling. (b) Pd 3d XPS spectra and (c) Ti 2p XPS spectra of Pd/TiO<sub>2</sub> over prolonging milling time. Reprinted with permission from.<sup>[81]</sup> Copyright American Chemical Society, 2022. (d) Mechanochemical NaBH<sub>4</sub>-driven SMSI encapsulation of Pd NPs on anatase  $TiO_2$ . (e) HRTEM images of the SMSI TiO<sub>x</sub> overlayer. Reprinted with permission from.<sup>[82]</sup> Copyright American Chemical Society, 2022.



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photoreaction because of SMSI on ruling out Pt and oxygen vacancies. In this case, oxygen vacancies tend to strengthen  $CO_2$  adsorption on photocatalyst surface, thus boosting photocatalytic efficiency. The selectivity toward methane was explained by studying different pathways of  $CO_2$  adsorption and activation during photocatalytic reduction in the presence of H<sub>2</sub>O (Figure 11a). In details, adsorbed  $CO_2$  would react with OH groups on the surface of bare  $O_v$ -defected TiO<sub>2</sub> to form

 $CO_3^{2-}$  intermediates, which can be further converted to CO and CH<sub>4</sub>. In the case of Pt-mediated photocatalysts, Pt NPs become active sites for heterolytic dissociation of H<sub>2</sub>O to H<sup>+</sup> protons. Besides, electron transfers from TiO<sub>2</sub> to Pt resulted from SMSI made these metal NPs act as electron reservoirs. CO<sub>2</sub> molecules in the vicinity of SMSI regions would capture those electrons and together with H<sup>+</sup> protons to produce



**Figure 11.** (a) Different pathways of CO<sub>2</sub> photoreduction with  $H_2O$  on bare TiO<sub>2</sub> and defective TiO<sub>2</sub>-supported Pt photocatalysts. Reprinted with permission from.<sup>[33]</sup> Copyright Elsevier, 2020. (b) Hydrocarbons, CH<sub>4</sub> and C<sub>2</sub>H<sub>6</sub>, production on single-atom binary metal components supported on TiO<sub>2</sub> nanotubes from CO<sub>2</sub> photoreduction. Reprinted with permission from.<sup>[83]</sup> Copyright Elsevier, 2021. (c) A proposed mechanism for CO<sub>2</sub> reduction to CO on TiO<sub>2</sub>-coated dendritic fibrous nano-silica supported Cu catalysts. Reprinted with permission from.<sup>[84]</sup> Copyright American Chemical Society, 2023.

formate species (HCOO $^{-}),$  which is further reduced to  $\mathsf{CH}_4$  (Figure 11a)

Another research on enhancing gas-phase photocatalytic CO<sub>2</sub> reduction in water vapour phase was reported on strong metal-support interaction between single-atom binary metal components Pt/Au and  $\text{TiO}_{2\text{-}x}$  nanotube.  $^{\scriptscriptstyle [83]}$  These metal components were supposed to trap electrons and create bonds with uncoordinated Ti<sup>3+</sup> sites, inducing electron transfers from Ti<sup>3+</sup> to Pt and Au. This study assigned the conversion of CO<sub>2</sub> into CO<sub>2</sub><sup>•-</sup> as a rate-limiting step of the photocatalytic reduction. Electron-rich bimetal components might boost multi-electron reactions, evidenced by the modification of one-electron water oxidation to four-electrons, leading to O<sub>2</sub> evolution instead of 'OH generation. Catalysts mediated with atomic scale Pt and Au clearly promoted the polarization of  $CO_2$  and protonation of  $CO_2^{\bullet-}$  to  ${}^{\bullet}CH_3$  radicals, resulting in  $CH_4$ or C–C coupling to C<sub>2</sub>H<sub>6</sub>. The catalytic activity remained stable after five testing cycles (Figure 11b).A recent study has reported strong metal-supported interaction on TiO<sub>2</sub>-supported Cu catalyst and its use in CO<sub>2</sub> reduction reaction.<sup>[84]</sup> TiO<sub>2</sub> support partially encapsulated and induced SMSI overlayer around Cu active sites where Ti<sup>3+</sup> species found them strongly bond with copper atoms, accompanying with defective oxygen vacancies on TiO<sub>2</sub> surface. This study proposed a mechanism of CO<sub>2</sub> reduction, illustrated in Figure 11c. CO<sub>2</sub> molecules adsorbed on TiO<sub>2</sub> surface through filling oxygen vacancy defects before reacting with hydroxyl group to produce bicarbonate species and then formate intermediates by dissociated H from Cu NPs. The formate intermediates would break down into CO and leave OH on TiO<sub>2</sub> surface which desorbed as water molecules after reacting with another dissociated H species. Afterwards, the catalyst regenerated oxygen vacancies and Ti<sup>3+</sup> species as a fresh one.<sup>[84]</sup>

Strong metal-support interaction between Au NPs and TiO<sub>x</sub> was firstly reported in the context of photocatalytic hydrogen evolution in a research by Gesesse et al.[77] In this study, the authors deliberately covered stoichiometric TiO<sub>2</sub> anatase overlayer onto plasmonic Au NPs dispersed on SiO<sub>2</sub> microspheres. The SMSI-like core@shell nanostructure recorded 10 times higher photocatalytic H<sub>2</sub> production yield than a conventional photocatalyst in which Au NPs were deposited on TiO<sub>2</sub> surface (Figure 12a). This could be attributed to hot electron injection from Au NPs to  $TiO_2$  conduction band and superior charge carrier dynamics within the nanostructure. Time-resolved microwave conductivity (TRMC, Figure 12b) evidenced that Au NPs enable to prolong photogenerated charge carrier lifetime in which electrons could be not only transferred from TiO<sub>2</sub> to Au NPs, but also injected back from Au NPs to TiO<sub>2</sub> through SMSI region, leading to superior charge carrier separation and boost photocatalytic efficiency. Density functional theory (DFT) calculation was performed to understand adsorption/ desorption energy in the photocatalyst designs (Figure 12c). The stronger adsorption/desorption energy of methanol on  $TiO_2$  surface (-1.0788 eV) in the case of conventional design explained its lower photocatalytic activity because of restraining desorption of methanol from the surface which is a crucial step in methanol aqueous solution of H<sub>2</sub> production. In order to achieve an optimal reaction yield, the adsorption sites of hydrogen ions should most likely occur at the  $Au/TiO_2$  interface. In fact, the active sites cover a larger surface in the encapsulation design compared to the conventional design. In the case of conventional design, Au NPs are the active site. The distance between adsorption site and Au NPs is longer, leading to the limited probability of photocatalytic reactions.

### 5. Conclusions and Perspectives

In this manuscript, we reviewed the classical SMSI between platinum-group metals and reducible metal oxide such as TiO<sub>2</sub>. It was believed that the SMSI encapsulation is exclusively able to occur onto the platinum-group metals which have high work function (>5.3 eV) and great surface energy (>2 J.m<sup>-[2]</sup>) .<sup>[85]</sup> However, recent studies have already demonstrated that SMSI occurred with Cu (4.46 eV)<sup>[86]</sup> and Au (5.3 eV)<sup>[87]</sup> even on different metal oxide supports (TiO<sub>2</sub>, ZnO, Nb<sub>2</sub>O<sub>5</sub>, CeO<sub>2</sub> etc.), leading to potential extension of SMSI concept in solid-state chemistry and material science. This review summarized a variety of synthetic approaches to induce SMSI, including high-temperature annealing, alcohol adsorbate-induced encapsulation, laser/photon irradiation treatment, wet chemistry, and the most recent mechanochemical technique. High temperature annealing in reductive environment is the most commonly used method. Because of impermeable TiO<sub>2</sub> overlayer after the treatment which could insulate metallic active sites and deactivate catalytic activity, some efforts have been devoted to generate a permeable overlayer through modifying synthesis procedure with melamine or architecting a porous structure of metal oxide support. The high temperature synthetic condition would be an obstacle that incentivized later attempt on alternative approaches at milder or ambient conditions such as isopropanol-adsorbate treatment and ultrafast laser inducing. Even though of successfully generating TiO<sub>x</sub> overlayer onto metal NPs, these techniques generally failed to obtain high crystalline TiO<sub>2</sub> in anatase phase, but amorphous TiO<sub>2</sub>. This typically limited the application of SMSI in the context of heterogeneous photocatalysis. In addition, it is necessary to call for methods that SMSI overlayer thickness is adjustable in purpose. Wet chemistry and mechanochemistry methods nominated themselves as potential choices to overcome these challenges. Strong metal-support interaction typically accompanies with fully or partially encapsulation of metal nanoparticles by metal oxide support. Electronic interactions through bonding between Ti element of TiO<sub>2</sub> and supported metal species induce electron transfer from TiO<sub>2</sub> to metal NPs, leading to  $Ti^{3+}$  species and oxygen vacancies  $O_v$  in SMSI region. These surface defects importantly play as active sites during (photo)catalytic reactions. In addition to electronic interaction, SMSI encapsulation exerts some effects on morphology of supported metal nanoparticles. The overlayer would exclude sintering, agglomeration and leaching of metal NPs or even metal nanoparticle lost during synthesis and catalytic reactions.

Review



**Figure 12.** (a) Hydrogen production yield on SMSI-like core@shell SiO<sub>2</sub>@Au@TiO<sub>2</sub> and conventional core@shell SiO<sub>2</sub>@TiO<sub>2</sub>@Au photocatalysts in methanol/ water solution. (b) high intensity and long decay of time-resolved microwave conductivity signal reveal superior charge carrier dynamics within the SMSI-like nanostructure. (c) A proposed mechanism by DFT of methanol dissociation, reduction of H<sup>+</sup> species on the two different core@shell nanostructures. Reprinted with permission from.<sup>[77]</sup> Copyright Royal Society of Chemistry, 2020.

In conclusion, strong metal-support interaction has become promising to tune electronic properties and surface chemistry of metal nanoparticles highly dispersed on metal oxide support. There are still high demands of universal approaches to trigger SMSI at ambient conditions. Besides, a limited number of studies mentioning SMSI in photocatalysis leave a huge space for perspectives of architecting different SMSI-liked nanostructures, replacing noble metals (Pt, Pd, Au, etc.) by earth-abundant metals (Cu, Co, Ni, Fe, etc.) which could demonstrate improved photocatalytic activity, elevated stability, and reproducibility.

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### **Conflict of Interests**

The authors declare that they have no conflict of interest.

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Review

## REVIEW

This review focuses on recent efforts to enhance photocatalytic activities of metal nanoparticle-mediated photocatalysts through "strong metalsupport interaction" (SMSI). Herein, we discuss the fundamentals of "strong metal-support interaction" and the methodology to practice the concept, involving synthesis and characterization techniques. The strengths and limitations of SMSI are also discussed, thus outlining future perspectives.



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Strong Metal-support Interactions in Photocatalysis: Fundamentals and Design Methods