



HAL
open science

Photo-engineered optoelectronic properties of indium tin oxide via reactive laser annealing

James Arthur Hillier, Panos Patsalas, Dimitrios Karfaridis, Sophie Camelio, Wayne Cranton, Alexei V Nabok, Christopher J Mellor, Demosthenes C Koutsogeorgis, Nikolaos Kalfagiannis

► **To cite this version:**

James Arthur Hillier, Panos Patsalas, Dimitrios Karfaridis, Sophie Camelio, Wayne Cranton, et al.. Photo-engineered optoelectronic properties of indium tin oxide via reactive laser annealing. *Scientific Reports*, 2022, 12 (1), pp.14986. <10.1038/s41598-022-18883-5>. <hal-04264243>

HAL Id: hal-04264243

<https://hal.science/hal-04264243v1>

Submitted on 30 Oct 2023

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



HAL Authorization



OPEN

Photo-engineered optoelectronic properties of indium tin oxide via reactive laser annealing

James Arthur Hillier¹, Panos Patsalas², Dimitrios Karfaridis², Sophie Camelio³, Wayne Cranton⁴, Alexei V. Nabok⁴, Christopher J. Mellor⁵, Demosthenes C. Koutsogeorgis¹ & Nikolaos Kalfagiannis¹✉

Transparent conductive oxides are appealing materials for optoelectronic and plasmonic applications as, amongst other advantages, their properties can be modulated by engineering their defects. Optimisation of this adjustment is, however, a complex design problem. This work examined the modification of the carrier transport properties of sputtered tin-doped indium oxide (ITO) via laser annealing in reactive environments. We relate the optical modifications to the structural, compositional, and electronic properties to elucidate the precise mechanisms behind the reactive laser annealing (ReLA) process. For sufficiently high laser fluence, we reveal an ambient-dependent and purely compositional modulation of the carrier concentration of ITO thin films. Hereby, we demonstrate that ReLA utilises the precise energy delivery of photonic processing to enhance the carrier mobility and finely tune the carrier concentration without significantly affecting the crystal structure. Exploitation of this phenomena may enable one to selectively engineer the optoelectronic properties of ITO, promising an alternative to the exploration of new materials for optoelectronic and photonic applications.

Transparent conductive oxides (TCOs) are used in optoelectronic devices such as flat panel displays due to their transparency, refractory character, and their capability for dynamic tuning of their optoelectronic behaviour through application of electric fields via the Pockels effect^{1–3}. An important additional asset of TCOs is the fact that they are CMOS compatible, unlike the noble metals, allowing for utilisation of well-matured manufacturing techniques and integration into current electronic devices⁴. Vitaly, the optoelectronic properties of TCOs are sensitive to the fabrication techniques and parameters⁵. For example, adjustment of the dopant level of the sputtering target, deposition and post-growth processing conditions enables one to tailor the optoelectronic properties of TCOs towards specific device requirements⁶. Such modifications, for tin-doped indium oxide (ITO) in particular, have been achieved through thermal annealing (TA) in controlled environmental conditions^{7–9}. This has been shown to improve the free carrier mobility and transparency by inducing crystallisation of the as grown amorphous film. TA has also been demonstrated to adjust the carrier concentration by probing the donor state variations, which result from Sn⁴⁺ ion substitutions and the oxygen vacancy concentration^{10–14}. TA within an ambient environment of 5% H₂ in N₂ increased the carrier concentration by creating oxygen vacancies while, conversely, an ambient environment of pure O₂ reduced the carrier concentration by filling oxygen vacancies⁷. However, TA of thin films suffers from long dwell times and high thermal budget, making the process, cumbersome and unable to be utilised for films where the characteristics of the substrate must not be compromised (i.e., flexible electronic devices or manufactured chips with heat-sensitive components). Recent results have demonstrated the ability of laser annealing (LA) to overcome these limitations and offer an ultra-fast, scalable, and low thermal budget post-growth processing technique to enhance the crystallinity of TCOs^{15–17}. LA operates through the application of a highly spatially and temporally localised energetic heating and offers an increased level of control over the manipulation of material properties¹⁸. This is achieved via access to an array of LA parameters such as pulse length, frequency, number of pulses, fluence and wavelength in addition to environmental parameters such as pressure and composition^{18,19}.

¹School of Science and Technology, Nottingham Trent University, Nottingham NG11 8NS, UK. ²Department of Physics, Aristotle University of Thessaloniki, 54124 Thessaloniki, Greece. ³UPR 3346, Institut Pprime, CNRS-Université de Poitiers, Poitiers, France. ⁴Materials and Engineering Research Institute, Sheffield Hallam University, Sheffield S1 1WB, UK. ⁵School of Physics and Astronomy, The University of Nottingham, Nottingham NG7 2RD, UK. ✉email: nikolaos.kalfagiannis@ntu.ac.uk

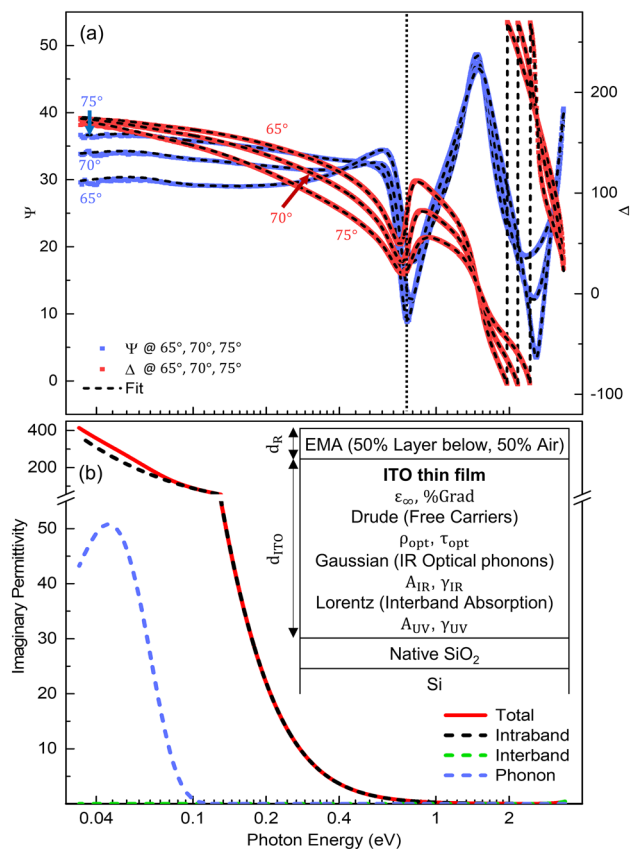


Figure 1. Ellipsometric measurement and as-fit permittivity of the seed ITO thin film. **(a)** Measured $\Psi(E)$ (blue squares) and $\Delta(E)$ (red squares) at 65°, 70°, and 75° for the higher resistivity seed ITO thin film alongside their corresponding fits (dashed black lines). **(b)** The fitted imaginary permittivity of the seed ITO thin film (solid red line), where the individual components of the imaginary permittivity for the free carriers, interband transitions and phonon mode are represented by the dashed black, blue and green dashed lines, respectively. A logarithmic scale in the x-axis is used in **(a,b)** to exaggerate the IR spectral region. The vertical dotted black line in **(a)** presents the cut-off between the spectral ranges of the two IR and NIR-VIS ellipsometers. The inset in **(b)** presents a schematic of the geometric and parametrised optoelectronic model.

In this work, we seek to synergise the advantages of LA (KrF, $\lambda = 248$ nm) with the ability to probe the defects by controlling the reactive ambient. Specifically, we investigate two “extremes” of a highly oxidising (100% O₂) and a highly reducing (5% H₂ in N₂) atmosphere during LA with a single pulse and a varied laser fluence (25–125 mJcm⁻²). We perform extensive characterisation utilising: four-point probe (4pp), Hall Effect, X-ray diffractometry (XRD), X-ray photon spectroscopy (XPS), transmission electron microscopy (TEM), energy-dispersive X-ray spectroscopy (EDX) and spectroscopic ellipsometry (SE) across a wide spectral range (0.034–3.34 eV). We relate the optical properties of the processed films to the electronic, structural, and compositional laser-induced modifications in order to gain knowledge of the physical mechanisms behind LA in reactive gas environments. We reveal a purely compositional, ambient-dependent modification of the carrier transport properties of ITO, demonstrating that “reactive LA” (ReLA) can enhance the carrier mobility and tune the carrier concentration without significantly affecting the crystal structure. Further exploitation opens a pathway to selectively engineer the defects of ITO.

Results and discussion

Optoelectronic properties of the ReLA-ITO films. In order to reveal the properties of the ITO films subjected to ReLA we firstly examine the optoelectronic properties (i.e., the complex permittivity), which define the key characteristics of TCOs that are associated with most applications^{20,21}. Figure 1a shows the measured ellipsometric angles $\Psi(E)$ (blue squares) and $\Delta(E)$ (red squares), measured at 65°, 70°, and 75°, of an indicative seed ITO thin film. $\Psi(E)$, $\Delta(E)$, $T_{IR}(E)$, and $R_{VIS}(E)$ for all samples can be found in the Supplementary, section B. For all samples, there is a perfect overlap between the IR and near-IR-visible (NIR-VIS) measurements. Therefore, we fit the total combined spectra, with $T_{IR}(E)$ and $R_{VIS}(E)$ appended to the data. By fitting all measurements simultaneously, we reduce the correlation between the fitting parameters and thus we improve the confidence in the uniqueness and physical reality of the extracted values. The best fit across the entire spectral range is presented with dashed black lines in Fig. 1a. To fit the measured $\Psi(E)$ and $\Delta(E)$ to the seed and laser processed ITO, we apply a geometric model that was derived from the architecture of the sample (ITO/2.63 nm-SiO₂/Si;

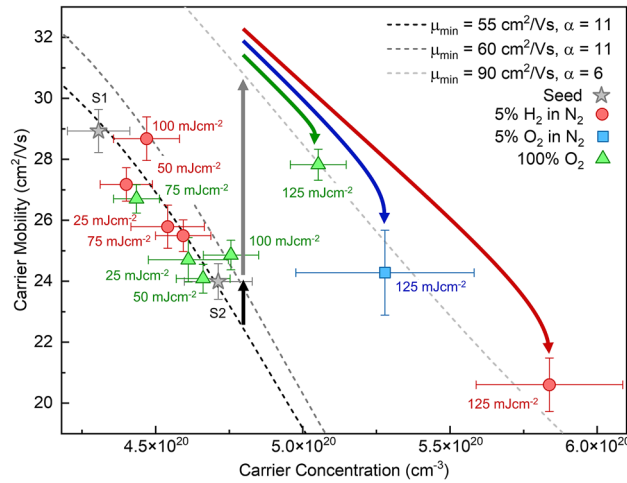


Figure 2. Optoelectronic properties of all seed and laser processed ITO thin films. “Optical” carrier concentration, N_{opt} , and mobility, μ_{opt} , of the seed ITO thin films (grey stars) and those subject to single-pulse ReLA at 25–125 $mJcm^{-2}$ in 5% H_2 in N_2 (red circles) and 100% O_2 (green triangles). We also present the sample processed with 125 $mJcm^{-2}$ in 5% O_2 in N_2 with the blue square. The dashed lines indicate the theoretical trend of $\mu(N)$ considering various scattering mechanisms. The parameters of $\mu(N)$ that are related to ionised cluster scattering (μ_{min} and α) were fitted to the seed materials and those subject to ReLA at below 100 $mJcm^{-2}$ (black dashed line), at 100 $mJcm^{-2}$ (dark grey dashed line) and at 125 $mJcm^{-2}$ (light grey dashed line). The black and grey arrows indicate the mobility-only transition to the next cluster and the red, blue, and green arrows indicate the ambient dependent transition to higher N_{opt} after ReLA at 125 $mJcm^{-2}$ in 5% H_2 in N_2 , 5% O_2 in N_2 and 100% O_2 , respectively.

presented schematically in Fig. 1b). To improve the fit, we follow the suggestions of previous reports to include a layer of surface roughness, described by an effective medium approximation (EMA) of 50% ITO and 50% void, and incorporate a simple gradient of the permittivity across the depth of the ITO film^{22,23}. The optical model describes $\tilde{\epsilon}(E)$ for each layer. The imaginary permittivity of the ITO layer, $\epsilon_2(E)$, is described by a summation of multiple oscillators (Eq. 1), which are described further in the Supplementary, section C.

$$\tilde{\epsilon}(E) = \epsilon_\infty + \frac{-\hbar^2}{\epsilon_0 \rho_{opt} (\tau_{opt} E^2 + i\hbar E)} + \sum_{n=1}^N \tilde{\epsilon}_n(E) \tag{1}$$

where \hbar is Planck’s constant, ϵ_0 is the vacuum permittivity, ρ_{opt} is the optical resistivity, and τ_{opt} is the scattering time. The third term summates multiple oscillators that describe phonon, defect and/or interband absorption. The real permittivity, $\epsilon_1(E)$, is calculated from a Kramers–Kronig transformation of $\epsilon_2(E)$ with a background contribution to the permittivity, ϵ_∞ ²⁴. By fitting the permittivity and geometry (thickness and roughness) of the ITO layer to the measurements, we extract the unknown variables in our model (e.g., thickness, d_{ITO} , surface roughness, d_R , “optical” resistivity, ρ_{opt} , and “optical” scattering time, τ_{opt}). The designation “optical” refers to the fact that the parameter is defined in relation to an alternating driving electromagnetic field instead of by a static field as is the case of 4 pp and Hall effect measurements. The “optical” free carrier concentration, N_{opt} , and mobility, μ_{opt} , are calculated from ρ_{opt} and τ_{opt} via Eqs. (2 and 3).

$$\mu_{opt} = \frac{\tau_{opt} e}{m_e^*} \tag{2}$$

$$N = \frac{m_e^*}{\rho_{opt} e^2 \tau_{opt}} \tag{3}$$

where e is the electron charge and $m_e^* = m^* m_e$ (m^* is the effective mass ratio and m_e is the electron mass). The set of fitting parameters and independently measured properties for all ITO thin films are summarised in Supplementary Tables S1 and S2. The effective electron mass, m_e^* , is determined for each sample by considering an increased non-parabolicity in the conduction band due to the free carrier population²⁵. This process required independent measurement of the “Hall” carrier concentration, N_{Hall} , via Hall Effect and is outlined further in a previous publication²⁶. N and μ are linked through the influence of scattering mechanisms. The individual mechanisms that comprise the mobility dependence to the carrier concentration, $\mu(N)$, have been described for the case of ITO by Ellmer et al.^{27,28}. Therefore, to reveal the full picture of the transport properties of each seed and laser processed film, we present μ_{opt} against N_{opt} and for the seed ITO thin films (grey stars) and those subject to single-pulse ReLA at 25–125 $mJcm^{-2}$ in 5% H_2 in N_2 (red circles), 5% O_2 in N_2 (blue square), and 100% O_2 (green triangles) in Fig. 2. The labels indicate the fluence used during laser processing. Three distinct trends of $\mu(N)$ are fit to individual “clusters” of the films: 25–75 $mJcm^{-2}$, 100 $mJcm^{-2}$, and 125 $mJcm^{-2}$, and are

indicated in Fig. 2 with the black, dark grey, and light grey dashed lines, respectively. The “scattering equation” that defines $\mu(N)$ is expressed fully elsewhere^{27,28}.

Consequently, we observe that the clusters relate to groups of samples where the scattering mechanisms that influence μ_{opt} in relation to N are identical. When $N > 10^{20} \text{ cm}^{-3}$, the dominant scattering mechanism of ITO is ionised cluster scattering (ICS)²⁸. Within the small range of N studied, only the parameters for the ionised impurity scattering (IIS) diminished mobility, μ_{min} , and the coefficient of ICS, α_{ICS} , matter (due to Matthiessen’s rule). α_{ICS} depicts the scattering efficiency of the ionised clusters and, within the small range of N studied here, can be perceived as the negative gradient of $\mu(N)$ (where μ_{min} represents the intercept). Thus, these are set as free parameters when fitting $\mu(N)$ to each cluster. All other parameters are fixed to those found by examining a larger range of N ^{26,27}. The first cluster comprises the two seed films and those annealed in either 5% H_2 in N_2 (red circles) and 100% O_2 (green triangles) at lower fluences ($25\text{--}75 \text{ mJcm}^{-2}$). The two seed films cover the range of ρ_{opt} demonstrated by the cluster of samples taken from the seed ITO film (see the Supplementary, section A). The films subject to ReLA at $25\text{--}75 \text{ mJcm}^{-2}$ lie along the theoretical trend of $\mu(N)$, where $\mu_{\text{min}} = 55 \text{ cm}^2/\text{Vs}$ and $\alpha = 11$ (black dashed line), within the range of N covered by the seed films. Therefore, we observe in Fig. 2 that single-pulse ReLA at $\leq 75 \text{ mJcm}^{-2}$ provides almost no alteration of the carrier transport properties from the seed ITO film. ReLA at 100 mJcm^{-2} results in a minor improvement to the film quality (μ_{min} is increased to $60 \text{ cm}^2/\text{Vs}$; black arrow in Fig. 2). ReLA at 125 mJcm^{-2} appears to cross a critical fluence threshold which leads to an even greater improvement of the film quality (μ_{min} is increased to $90 \text{ cm}^2/\text{Vs}$) alongside an apparent reduction in the “strength” of ICS (α is reduced from 11 to 6). These enhancements of the film quality likely arise from either structural and/or compositional modifications^{16,17}, to be discussed later. Furthermore, for the films annealed at 125 mJcm^{-2} , a change in the mobility alone (grey arrow) is not able to explain the post-ReLA transport properties. To do so, an additional change in N_{opt} is required. Vitaly, the increase in N_{opt} in an oxidising atmosphere (100% O_2 ; green arrow) is less than in a reducing atmosphere (5% H_2 in N_2 ; red arrow), indicating that this translation is ambient-dependent.

To assess the ambient-dependence further, we performed an additional single-pulse ReLA at 125 mJcm^{-2} in an intermediately oxidising environment (5% O_2 in N_2). We present the results of this investigation with the blue square in Fig. 2. The post-ReLA transport characteristics for this sample are found to lie on the same cluster with N between that of the samples annealed at 125 mJcm^{-2} in oxidising and reducing environments (blue arrow), confirming that intermediate mixtures of reducing and oxidising gasses results in intermediate carrier concentrations. Further investigations into the precise relation between the oxygen partial pressure and additional ambient atmospheres (e.g., N_2 , Ar, Vacuum) are highly warranted, but are beyond the scope of this work. In summary, ReLA induces modifications of the carrier transport properties of the ITO film. This comprises an enhancement of μ_{opt} at $\geq 100 \text{ mJcm}^{-2}$ coupled with an ambient-dependent modulation of N at 125 mJcm^{-2} . The increase in N was greater for the environments with less oxygen. The modifications of the carrier transport properties translate into changes in the complex permittivity of ITO (see Supplementary Fig. S15), which has implications for the potential of ITO as a plasmonic material component^{29,30}.

Structural ReLA-induced modifications. To examine whether these modifications arise from changes in the crystal structure, we investigated the crystallinity of the seed and processed films with XRD. Figure 3a presents the X-ray diffractograms for the seed film (grey squares) and the corresponding films processed with a single laser pulse at 125 mJcm^{-2} in 100% O_2 (green squares) and in 5% H_2 in N_2 (red squares). The X-ray diffractograms for all samples can be found in the Supplementary, section D.

The observed broad, asymmetric peak at $\sim 32.5^\circ$ may arise through various ways: (i) a mixing of (222) cubic In_2O_3 ($a = 1.0118 \text{ nm}$ ³¹, $2\theta = 30.61^\circ$) and (321) cubic In_2O_3 ($a = 1.0118 \text{ nm}$ ³¹, $2\theta = 33.127^\circ$; pink dashed line Fig. 3a) preferred orientations (ii) a mixing of (222) In_2O_3 and $\text{In}_4\text{Sn}_3\text{O}_{12}$ ($a = 0.9467 \text{ nm}$ ³², $2\theta = 32.77^\circ$; blue dashed line) phases, present in amorphous ITO films^{33–36}, and/or (iii) a (211) preferred orientation for orthorhombic In_2O_3 ($a = 0.7912 \text{ nm}$, $b = 0.5477 \text{ nm}$, $c = 0.5592 \text{ nm}$ ³⁷, $2\theta = 32.07^\circ$; dark blue dashed line). To account for the asymmetry and potential mixing, the diffractograms are fit with two *pseudo*-Voigt functions (grey shaded areas in Fig. 3a) with shared Lorentzian and Gaussian broadening factors (β_L and β_G , respectively)³⁸. Employing Scherrer’s formula³⁹, we extract a grain size, L , from the Lorentzian integral breadth, $\tau_L = (\pi/2)\beta_L$, that has been reported to give more accurate determination of L ³⁸. For the seed ITO film, we find that $L = 3.2 \pm 0.3 \text{ nm}$, confirming that the film is nanocrystalline. The offset of the (222) peak from the “bulk” value (dark pink dashed line in Fig. 3a) indicates that the film is under tensile stress ($\delta_S = 7.0 \pm 0.3 \text{ GPa}$) due to the interfacial mismatch between the atomic spacings for the Si and In_2O_3 lattices⁴⁰. From this, we calculate the “bulk” peak position for the 2nd, larger, peak to be $33.17^\circ \pm 0.04^\circ$, indicating that the 2nd peak likely arises from a (321) preferred orientation for cubic In_2O_3 . However, we cannot exclude the possibility of a co-existence of all three potential sources mentioned above. To summarise, the observed large peak broadening and asymmetry for the seed film arises from the nanocrystalline nature of the room temperature deposited ITO films film. In this case, we note from Fig. 3a that the nanocrystalline nature is unaffected during ReLA at $\leq 125 \text{ mJcm}^{-2}$ in either atmosphere. Indeed, there is no resolvable change in the internal stress, grain size, or the ratio of the two component peak areas for any of the laser processed ITO films (Supplementary, section D).

However, XRD averages over the entire sample and, due to the inherent depth-dependence of the LA process (see Supplementary, section F), the modifications may be highly localised. To elucidate any microscopic modifications to the film structure, we examined high-resolution TEM images for the seed ITO thin film (Fig. 3b) and the ITO films subjected to single-pulse processing at 125 mJcm^{-2} in 100% O_2 (Fig. 3c) and 5% H_2 in N_2 (Fig. 3d) environments. All the samples are confirmed to be nanocrystalline and appear uniform across the entire sample depth, both before and after ReLA at 125 mJcm^{-2} . It should be noted that in previous reports on LA of ITO, structural changes have been observed at lower fluences^{16,17,41}. The deviation between these reports and this work

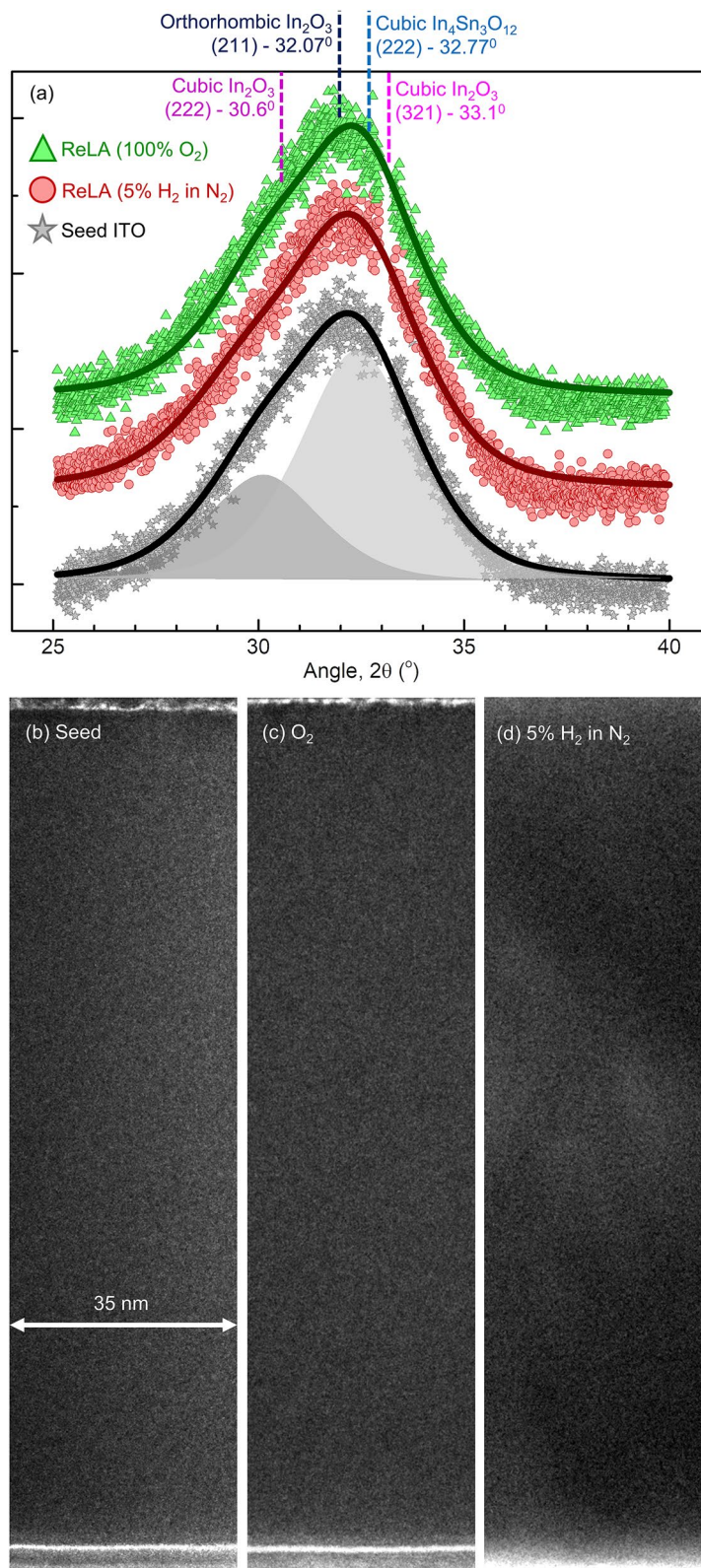


Figure 3. Structural properties of seed ITO films and laser processed at 125 mJcm^{-2} . X-ray diffractograms for seed ITO film (grey stars) and those subject to ReLA in 5% H_2 in N_2 (red circles) and 100% O_2 (green triangles) at 125 mJcm^{-2} . The solid black, red, and green lines represent the corresponding fit of two pseudo-Voigt functions to the data. For the seed film, we present the component peaks, representing cubic In_2O_3 (222) and rhombohedral $\text{In}_4\text{Sn}_3\text{O}_{12}$, with the grey shaded areas. The sharp (200) Si peak at $32.9^\circ - 33.1^\circ$ has been manually removed from all diffractograms. Also shown are the TEM cross-sectional images of the (b) seed ITO film and those subject to ReLA at 125 mJcm^{-2} in (c) 100% O_2 and (d) 5% H_2 in N_2 .

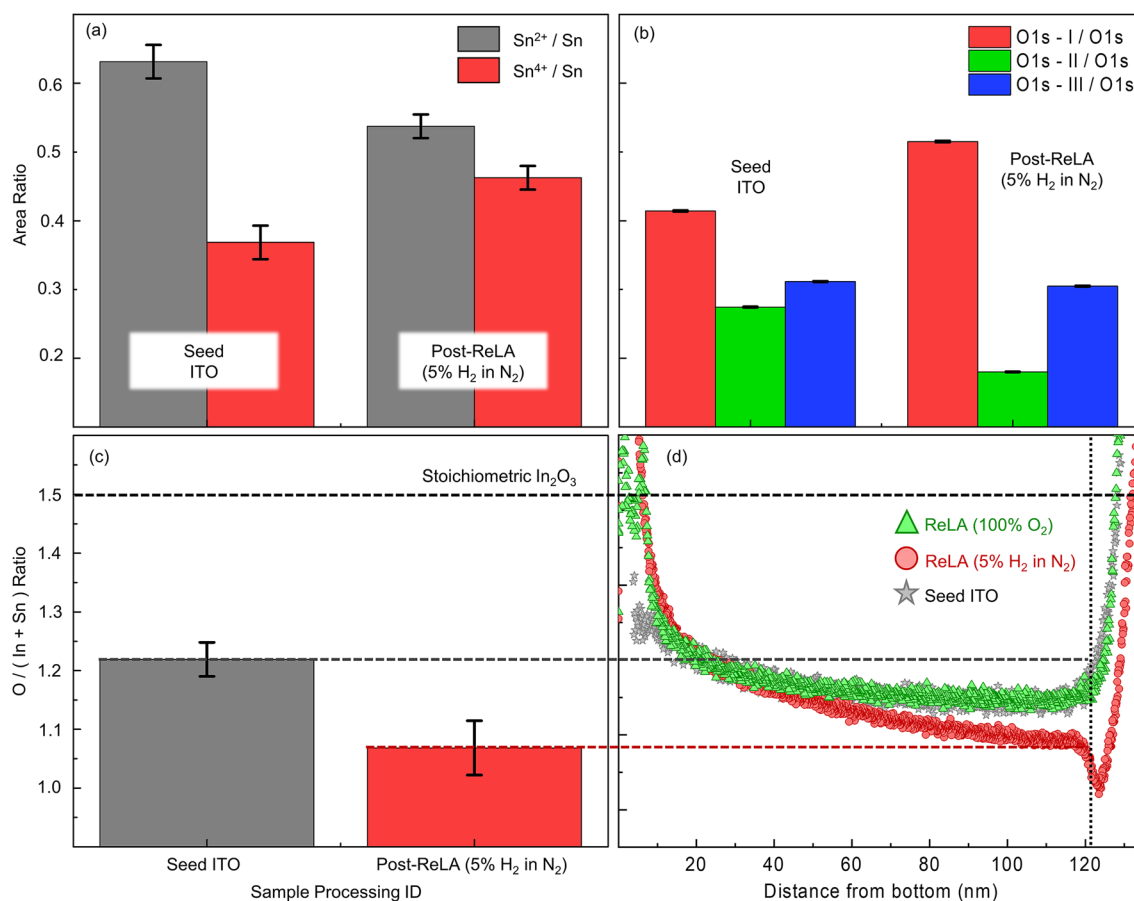


Figure 4. Compositional properties of seed ITO films and laser processed at 125 mJcm^{-2} . **(a,b)** Relative areal intensities of the deconvoluted component peaks of the core level X-ray photoelectron spectra for $\text{Sn}:\text{Sn}^{2+}$ (grey bar) and Sn^{4+} (red bar) and O1s peaks: O_{I} (red bar), O_{II} (green bar) and O_{III} (blue bar), respectively, for the seed ITO film and that subject to a single ReLA pulse at 125 mJcm^{-2} in $5\% \text{ H}_2$ in N_2 . **(c)** The $\text{O}/(\text{In} + \text{Sn})$ ratio for the seed ITO film (grey bar) and that subject to a single ReLA pulse at 125 mJcm^{-2} in $5\% \text{ H}_2$ in N_2 (red bar), calculated from surface-XPS results. **(d)** Depth profile of $\text{O}/(\text{In} + \text{Sn})$ for the seed ITO film (grey stars) and those subject to single ReLA pulse at 125 mJcm^{-2} in $5\% \text{ H}_2$ in N_2 (red circles) and $100\% \text{ O}_2$ (green triangles), as calculated from EDX imaging. $\text{O}/(\text{In} + \text{Sn})$ for a stoichiometric In_2O_3 film is indicated by the black dashed line. We map the XPS results onto the “surface” (black vertical dotted line) of the EDX depth profile with the horizontal grey and red dashed lines.

could may be due to the different thickness of the ITO film and the intermediate SiO_2 layer (acting as a thermal barrier and an optical spacer), and/or the growth process (solution versus sputtering)^{16,17,41}. However, we also note that an increased ambient pressure is typically employed to suppress and/or eliminate sample ablation during LA^{42,43}. We conclude that ReLA, at high pressure and up to and including 125 mJcm^{-2} in either $5\% \text{ H}_2$ in N_2 or $100\% \text{ O}_2$, is a “low-stress” process and the modifications to the optoelectronic properties induced during ReLA cannot be explained through structural changes.

Compositional ReLA-induced modifications. As the structure of the ITO films does not significantly change during ReLA, we focus our attention to the compositional alterations. To elucidate the compositional changes that govern the tailoring of the optical properties of ITO, we examined the surface oxidation states of the individual elements, via XPS, for the seed ITO film and one that was subjected to single-pulse ReLA at 125 mJcm^{-2} in the reducing environment ($5\% \text{ H}_2$ in N_2). A key dominant donor mechanism in ITO is the introduction of the Sn dopant to In_2O_3 ²⁰. Therefore, we initially examine the peak in the core-level X-ray photoelectron spectra (see Supplementary, section E) that is related to Sn atoms. We deconvolute the $\text{Sn}3d_{5/2}$ peak and extract the areal intensities of the Sn^{4+} and Sn^{2+} components, which are related to the “activated” SnO_2 and the “un-activated” Sn–O within the film, respectively⁷. As the donor state of ITO originates from Sn^{4+} ions substituting into In^{3+} and oxygen vacancy positions (becoming “activated”), the areal intensity of the Sn^{4+} component can be related to adjustments of the carrier concentration^{44,45}.

We present the areal intensities of the Sn^{4+} and Sn^{2+} components in relation to the $\text{Sn}3d_{5/2}$ peak in Fig. 4a. We note that in the seed film the Sn^{2+} component is dominant and that there is an increased Sn^{4+} content after ReLA in $5\% \text{ H}_2$ in N_2 . Evidently, Sn activation at 125 mJcm^{-2} accounts for the enhancement of N_{opt} observed in Fig. 2. We infer that a similar process of Sn activation occurs for the film processed in $100\% \text{ O}_2$, resulting in

the increase in N_{opt} (Fig. 2). This is despite the highly oxidising environment, which can be reasonably expected to reduce N_{opt} by filling oxygen vacancies^{10–14}. To investigate the role of the oxygen vacancies as donor states, we examine the O1s core-level X-ray photoelectron peak (see Supplementary, section E, for the measurements and peak fits). The O1s peak is de-convoluted into O_{I} , O_{II} and O_{III} components⁷. The O_{I} and O_{III} components are related to the metal-oxide binding (In-O)⁴⁵ and surface contaminants⁴⁶, respectively⁴¹. The origin of the O_{II} peak is commonly associated with oxygen atoms existing near to a neighbouring oxygen vacancy. However, this is still under debate, and it is also possible that the O_{II} peak is assigned to the amorphous phase of ITO and/or oxygen atoms bound to Sn^{10,47,48}. The relative areal intensities of the O1s components are shown in Fig. 4b. The film subjected to ReLA in 5% H_2 in N_2 exhibits an enhancement of the metal oxide peak (O_{I} ; red bar in Fig. 4b) in relation to the seed film. This corresponds to a removal of oxygen interstitials that activates the Sn^{4+} donors and corroborates the improvement in μ_{opt} observed in Fig. 2 (particularly the increased μ_{min})⁴⁹. There is also a marked decrease in the relative areal intensity for the peak commonly associated with oxygen vacancies (O_{II} , green bar in Fig. 4b). As the decrease in the O_{II} peak corresponds to a reduction in the amount of oxygen (relative to In and Sn) alongside an increase in the carrier concentration, it seems unlikely that the O_{II} peak is a result of oxygen vacancies. In addition, the lack of a significant change in the structure (Fig. 3) eliminates the amorphous phase as the source of the O_{II} peak. The correlation between the reductions in the relative areal intensity of the O_{II} peak and the increase in the $\text{Sn}^{4+}/\text{Sn}^{2+}$ ratio suggests, instead, that the oxygen atoms bound to Sn are the primary source. It is, however, possible that the peak arises from a summation of the above-mentioned sources, which cannot be individually resolved during the peak fitting. Finally, there exists a reduction in the surface contaminants after annealing in 5% H_2 in N_2 (O_{III} ; blue bar in Fig. 4b). Due to the surface nature of XPS, it is somewhat unclear how these modifications correlate to the bulk of the ITO film.

To reveal the compositional modifications induced across the entire depth of the sample we turn to the EDX measurements, presented in Fig. 3. Across the image, the average emitted X-ray intensity at the characteristic energy of each element within the sample was measured in order to build a depth profile of the relative elemental concentration across the film. The intensities, plotted relative to their maxima, are presented in the Supplementary, section E. To calibrate the EDX intensities to the absolute surface region abundance determined via XPS, we first calculate the surface region O/(In + Sn) ratio from the total areal intensities of the O1s, In3d_{3/2}, In3d_{5/2}, Sn3d_{3/2}, and Sn3d_{5/2} XPS peaks (considering the relative sensitivity factors)^{50,51}. We present the surface region O/(In + Sn) ratio for the seed ITO film (grey bar) and that subject to single-pulse ReLA at 125 mJcm⁻² in 5% H_2 in N_2 (red bar) in Fig. 4c. The likelihood that annealing in 5% H_2 in N_2 has created new oxygen vacancies is supported by the drop in the relative O content, from 1.21 to 1.06. Both values are a departure from the stoichiometric case of O/(In + Sn) = 1.5 (dashed black line in Fig. 4c). Significantly, the observed modulation of the oxygen vacancy contribution may give rise to the ambient dependence of the change in the carrier concentration during ReLA. However, this is only relevant to the surface of the film and the ReLA process is depth-dependent (see Supplementary, section F). Figure 4c is used to calibrate the EDX-calculated depth profile of the normalised O/(In + Sn) ratio across the sample depth for the seed ITO film (grey stars) and those subject to single-pulse ReLA at 125 mJcm⁻² in 5% H_2 in N_2 (red circles) and in 100% O_2 (green triangles).

We present this as a function of the distance from the bottom of the film in Fig. 4d. We offset the EDX results so that the values of O/(In + Sn) for the seed and 5% H_2 in N_2 annealed film, at roughly 10 nm below the imaged surface (to account for the sampling depth)⁵², match the values determined from XPS. Further details on the analysis of the EDX images is presented in the Supplementary, section E. These steps allow us to examine how the oxygen content varies across the film. In Fig. 4d, a smooth gradient is observed in the relative oxygen concentration across the seed film. The gradient in O/(In + Sn) is more substantial for the film processed in 5% H_2 in N_2 , and that near the surface there is a sharp dip in O/(In + Sn). Such a gradient has been reported previously using secondary ion mass spectroscopy⁵³, where a dip of O/(In + Sn) at the surface of the film was also found for reducing annealing environments. The gradient of the oxygen throughout the depth of the film is reflected in the presence of the simply graded inhomogeneity utilised during SE analysis.

Reflecting the oxygen gradient in the ellipsometric model. To reflect the observed gradient in the oxygen content across the film (Fig. 4d), an exponential gradient in ρ_{opt} and τ_{opt} across the depth of the ITO films was included in the ellipsometric analysis using Eq. (4) and (5).

$$\rho_{\text{opt}}(z) = \rho_{\text{opt},0} \left(1 + \frac{\rho_{\% \text{Gr}}}{100} \left(\frac{z}{d_{\text{B}}} \right)^{\chi} \right) \quad (4)$$

$$\tau_{\text{opt}}(z) = \tau_{\text{opt},0} \left(1 + \frac{\tau_{\% \text{Gr}}}{100} \left(\frac{z}{d_{\text{B}}} \right)^{\chi} \right) \quad (5)$$

where $\rho_{\text{opt},0}$ and $\tau_{\text{opt},0}$ are the values of the optical resistivity and mean free time, respectively, at the bottom of the film (the ITO/Si interface). $\rho_{\% \text{Gr}}$ and $\tau_{\% \text{Gr}}$ are the percent gradient of the optical resistivity and mean free time, respectively. z is the distance from the bottom of the layer, d_{B} is the layer thickness, and χ is the exponent of the gradient. The gradient is described by a series of 100 discrete slices such that $z = (c + 0.5)/n$ where c is the current slice, n is the total number of slices and z falls at the middle of each slice. The resulting MSE for fitting this film using a gradient in ρ_{opt} and τ_{opt} was reduced (from 14.7 to 10.3). The values of $N_{\text{opt}}(z)$ were calculated from $\rho_{\text{opt}}(z)$ and $\tau_{\text{opt}}(z)$ and are presented in Fig. 5 for the two of the seed optimised ITO films (black and grey line) and those subject to ReLA in 5% H_2 in N_2 (red line) and 100% O_2 (green line) at 100 mJcm⁻² (lighter shade) and 125 mJcm⁻² (darker shade). The films annealed at < 100 mJcm⁻² showed a negligible change in the gradient of $N_{\text{opt}}(z)$ as compared to the seed films, and so are excluded from Fig. 5 for the sake of clarity. For the seed

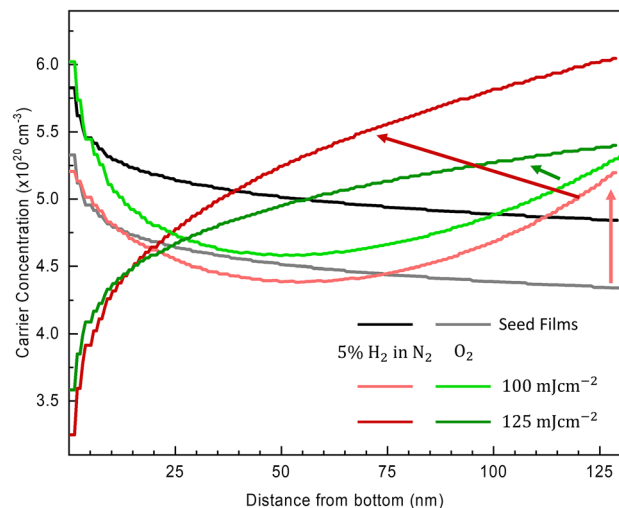


Figure 5. The optical carrier concentration as a function of the distance from the ITO/Si interface, $N_{\text{opt}}(z)$, for two representative samples at different positions on the seed ITO film (grey and black line) and those subject to single-pulse ReLA at 100 mJcm^{-2} (lighter shade) and 125 mJcm^{-2} (darker shade) in both $5\% \text{ H}_2$ in N_2 (red line) and $100\% \text{ O}_2$ (green line).

ITO films (black and grey solid lines in Fig. 5), $N_{\text{opt}}(z)$ is larger towards the bottom of the film where it rapidly falls by $\sim 1 \times 10^{20} \text{ cm}^{-3}$ within the first $\sim 10 \text{ nm}$ before coming to a low plateau. A sharper gradient in $N_{\text{opt}}(z)$ near the bottom of the film is observed for the film annealed at 100 mJcm^{-2} in both the reducing and oxidising environments (light shaded red and green, respectively, lines in Fig. 5). At the surface of the film, however, N_{opt} is increased from the case of the seed films, and this results in an almost parabolic shape of $N_{\text{opt}}(z)$ across the film, where there is a lower N_{opt} near the middle of the film.

The local minimum in $N_{\text{opt}}(z)$ arises from the interaction between a decreasing $\rho_{\text{opt}}(z)$ and $\tau_{\text{opt}}(z)$. As was seen in Fig. 2, the “average” N_{opt} of the film annealed at 100 mJcm^{-2} is larger for the film annealed in $100\% \text{ O}_2$ rather than $5\% \text{ H}_2$ in N_2 . However, a reasonable assumption can be that the films used for laser processing for each ambient were slightly different due to the slight disuniformity of the deposition. This can be seen to occur by noting that N_{opt} at the bottom of the film is larger for the film annealed in $100\% \text{ O}_2$, and how it is very similar to the seed film with the largest N_{opt} (black line). For ReLA in $5\% \text{ H}_2$ in N_2 , $N_{\text{opt}}(z = 0)$ conversely is very similar to the seed film with the smallest N_{opt} (black line). By accounting for this, it is seen that the increase of N_{opt} at the surface (relative to the seed film) is larger for the film annealed in $5\% \text{ H}_2$ in N_2 , than for the film annealed in $100\% \text{ O}_2$. This is indicated by the light shaded red and green arrows in Fig. 5. For the films annealed at 125 mJcm^{-2} in $5\% \text{ H}_2$ in N_2 and in $100\% \text{ O}_2$ (dark red and green lines in Fig. 5, respectively), N_{opt} at the surface is increased even further and the increase is, again, dependent on the ambient. The steepness of the $N_{\text{opt}}(z)$ curve for the seed and laser processed films agree well with the trend in the $\text{O}/(\text{In} + \text{Sn})$ ratio observed in Fig. 4d. This is expected from the increased likelihood of oxygen vacancy donors in the oxygen deficient lattice. Closer to the substrate, $N_{\text{opt}}(z)$ is reduced in comparison to the seed film for both the films annealed in $5\% \text{ H}_2$ in N_2 and $100\% \text{ O}_2$. This is reflected by how the $\text{O}/(\text{In} + \text{Sn})$ ratio is increased at the bottom of the film (Fig. 4d). The overall trend can be described by a shift of the gradient to lower in the film, so that the “dip” in N_{opt} lies near the bottom of the film.

These results are used to build a physical interpretation of the ReLA process. In response to the photo-induced localised heating, the temperature at the surface rises sufficiently (see Supplementary, section F) to allow the oxygen atoms to become mobile within the lattice. In the reducing environment, the oxygen overflows from the thin films into the environment. This causes the reduced oxygen content in the surface region, resulting in the formation of oxygen vacancies⁵⁴. While the temperature within the film is high enough, oxygen atoms remain mobile and they continue to be drawn towards this region, resulting in the final increased oxygen gradient within the film. In the oxidising environment, however, the oxygen atoms that escape at the surface are quickly replaced with oxygen atoms from within the pressurised chamber. Therefore, the system (regarding the distribution of oxygen atoms) is at equilibrium and the oxygen content across the film remains highly similar to that of the seed film. Depth-profile EDX does not elucidate whether the oxygen atoms within the film are more likely to end up in metal oxide or interstitial positions after ReLA. However, from the increased mobility observed in the optical results (Fig. 2) we can infer a reduction in oxygen interstitial concentration as no structural modifications were observed (Fig. 3). Finally, it is important to note that annealing in hydrogen has a potential to promote hydrogen doping within the lattice^{54,55}. This has been proposed to be a shallow donor that is more energetically favourable than oxygen vacancies⁵⁶. The effect of H-doping during ReLA in $5\% \text{ H}_2$ in N_2 may further enhance the carrier concentration modulation, but analysis of this mechanism is beyond the scope of this work. From this physical picture, we conclude that the ambient dependence of the free carrier modulation during ReLA arises primarily from the probing of the oxygen vacancies. In reducing environments, the oxygen that becomes mobile in the film escapes, leaving behind increased oxygen vacancies, increasing N_{opt} . However, due to the activation of Sn^{4+} within the film, we note an increase in the carrier concentration even within oxidising environments. As single-pulse

ReLA in 100% O₂ does not change the oxygen content within the film from that of the seed film, we are unable here to reduce the carrier concentration. For applications where a lower carrier concentration is desired in order to shift the plasma energy to lower photon energy, such as for low-loss NIR and IR plasmonics^{29,30,57}, a different strategy is required.

Conclusions

It was shown that ReLA offers a low stress method to engineer the defects of the ITO films by enhancing μ and selectively increasing N. The modifications arise from a combination of Sn⁴⁺ activation and manipulation of the oxygen migration during the annealing process. The low-stress nature of the process means that ReLA can promote defect mobility within the lattice without causing structural changes. This potentially makes the technique applicable to more complex prefabricated nano/microstructures, where changes to the structure may degrade device performance. Furthermore, the low-thermal budget of LA allows for the application of ReLA to annealing of materials on heat sensitive substrates, such as those used for flexible electronics, provided that the bulk of the thermal dose is contained within the ITO film (Supplementary, section F)^{58,59}. It should be noted that this is just for a single pulse and as such, multi-pulse ReLA is likely to be able to enhance the capability to engineer the defects of TCOs and thus tailor their carrier transport properties. Further investigations into the precise role of an intermediate oxygen partial pressure during ReLA and/or other ambient environments are also warranted.

Methods

Thin film deposition. ITO films (thickness of ~ 130 nm) were deposited onto double side polished n-type (1–10 Ωcm), 4", (100) oriented Si wafers with a 2 nm native oxide using radio frequency magnetron sputtering. The base pressure was $\sim 10^{-5}$ Pa ($\sim 10^{-7}$ mbar). Intentional substrate heating or bias was not applied to the substrate. We utilised a 3" target of 10 wt.% Sn:In₂O₃ (99.99% purity). Ar and O₂ were introduced to the chamber with an O₂ concentration of $0.25 \pm 0.02\%$ and the deposition was performed with a sputtering pressure of 2 mTorr and an RF power of 40 W. The substrate to target distance was 10.8 ± 0.2 cm. Further details on the deposition process can be found elsewhere^{60,61}. The seed film exhibited non-uniformity in the thickness and resistivity across the Si wafer (see Supplementary, section A). To account for this, the wafer was diced into $\sim 8 \times 8$ mm² die and a cluster of the seed films with the most similar resistivity were selected for further processing. Two samples with resistivities at the extreme ends of this cluster ($4.89 - 5.4 \times 10^{-4}$ Ωcm) remained un-processed to represent the seed film. These samples are denoted as "S1" and "S2" for the higher and lower resistivity seed films, respectively, within the manuscript and the supplementary.

Reactive laser annealing. LA was performed at room temperature with a KrF (248 nm) excimer laser (LAMBDA PHYSIK LPX 305i), which can deliver unpolarised light, with an average energy per pulse of up to 1.5 J, at a 1–50 Hz pulse repetition rate and 25 ns pulse length (i.e., average power per pulse of 60 MW) to the sample surface. Further details of the LA system are published elsewhere^{60,61}. Here, we also utilise a pressure cell with a UV-transparent window to enable LA within pressurised (6.89×10^5 Pa or 100 psig) environments of 5% H₂ in N₂ and 100% O₂. We call the process of LA within a reactive gas environment "reactive LA" (ReLA). For each ambient composition, a single laser pulse was delivered to the sample with a fluence, J_L , of 25–125 mJcm⁻² (power density of 1–5 MWcm⁻²) in steps of 25 mJcm⁻². We also performed single-pulse ReLA at 125 mJcm⁻² in 5% O₂ in N₂. We note that 125 mJcm⁻² was the maximum fluence allowed by our experimental setup that could maintain a readily measurable spot size of 1.20×1.20 cm².

Thin film characterisation. To reveal a complete picture of the laser-matter interactions that govern the conversion of the optoelectronic properties of the seed samples, an extensive set of characterisation techniques were utilised to relate the optical properties to the electronic, structural, compositional, and morphological modifications of ITO during ReLA.

Optoelectronic characterisation. Optical characterisation was performed using a J. A. Woollam Mark II infrared (IR)-SE in the spectral range of 0.034–0.8 eV (1.4–40 μm) and a J. A. Woollam M2000 SE in the spectral range of 0.74–3.34 eV (0.37–1.7 μm) at incident angles of 65°, 70°, and 75°. An optical model, reflecting the sample geometry, comprised the complex permittivity, $\tilde{\epsilon}(E)$, of each material and was fitted to $\Psi(E)$ and $\Delta(E)$. From the fitting process, we extract the geometric features (film thickness, surface roughness and uniformity) and the parameterised oscillators that describe $\tilde{\epsilon}(E)$ (interband transitions, phonon modes and/or defect states absorption, free carriers etc.) for the ITO thin films⁶². The J. A. Woollam Mark II IR-SE and a normal-incidence optical reflectance probe were also utilised to measure the IR transmittance, $T_{\text{IR}}(E)$, and visible reflectance, $R_{\text{VIS}}(E)$, respectively. Electrical characterisation of the resistivity was performed via 4pp in both collinear and Van-der-Pauw configurations ($\rho_{4\text{pp}}$ and ρ_{Hall} , respectively). Characterisation of the "Hall" carrier concentration, N_{Hall} , was performed with an Ecopia HMS-3000 Hall Measurement System at room temperature ($B = 0.553$ T). Further details of these techniques were provided in a recent publication²⁶. It should be noted that the typical back-surface finish of single-side polished Si wafers is insufficiently rough to ensure that there are no substrate back-reflections during IR-SE measurements. To account for this, the backsides of the Si wafers were roughened using a Dremel tool with a diamond head (following measurements of $T_{\text{IR}}(E)$).

Structural characterisation. Structural characterisation was performed by XRD employing a PanAnalytical Pro Diffractometer. The spot size was set to 5×5 mm² via a 10 mm height-limiting slit, 2^{rad} Soller slits and a programmable divergence slit. The scan step size was 0.0083556° with a time per step of 1998.345 s in a range of

25°–40°. The very long step time was required due to the small sample size and amorphous nature of the room-temperature sputtered ITO films^{9,53}. The range was chosen to cover the most clearly identifiable (222) and (400) Bragg peaks for bixbyite In₂O₃ ($a = 1.0118 \text{ nm}$)³¹, at 30.6° and 35.5°, respectively. Cross-sectional images of an indicative set of ITO thin films were obtained via high resolution TEM (FEI™ Talos sTEM) using a Schottky-type field emission gun operated with an electrostatic potential of 200 kV. Prior to TEM, the films were subjected to focused ion beam milling process using a JEOL 4500 FIB/SEM.

Compositional characterisation. Surface elemental characterisation was performed with a Kratos Analytical Ltd. AXIS Ultra XPS system using a monochromated Al-K_{α1} X-ray beam excitation source (1486.6 eV) in an ultra-high vacuum chamber (base pressure $\sim 10^{-9}$ bar). The relative elemental abundance across the entire depth of the ITO thin films was determined from EDX measurements with the FEI™ Talos sTEM.

Data availability

All data generated or analysed during this study are included in this published article and its supplementary information files.

Received: 13 June 2022; Accepted: 22 August 2022

Published online: 02 September 2022

References

- Haffner, C. *et al.* Low-loss plasmon-assisted electro-optic modulator. *Nature* **556**, 483–486 (2018).
- Ma, Z., Li, Z., Liu, K., Ye, C. & Sorger, V. J. Indium-tin-oxide for high-performance electro-optic modulation. *Nanophotonics* **4**, 198–213 (2015).
- Babicheva, V. E., Boltasseva, A. & Lavrinenko, A. V. Transparent conducting oxides for electro-optical plasmonic modulators. *Nanophotonics* **4**, 165–185 (2015).
- Babicheva, V. E. *et al.* Towards CMOS-compatible nanophotonics: Ultra-compact modulators using alternative plasmonic materials. *Opt. Exp.* **21**, 27326 (2013).
- Ginley, D. S. *Handbook of Transparent Conductors* (Springer, 2011).
- Mei, F., Yuan, T., Li, R., Qin, K. & Huang, J. Microstructure evolution and grain orientation in ITO targets and their effects on the film characteristics. *J. Mater. Sci.: Mater. Electron.* **29**, 14620–14634 (2018).
- Kim, J.-W. *et al.* Atomistic aspects of carrier concentration variation in post-annealed indium tin oxide films. *J. Phys. D Appl. Phys.* **48**, 395307 (2015).
- Wang, R. X. *et al.* Influence of gaseous annealing environment on the properties of indium-tin-oxide thin films. *J. Appl. Phys.* **97**, 033504 (2005).
- Hu, Y., Diao, X., Wang, C., Hao, W. & Wang, T. Effects of heat treatment on properties of ITO films prepared by rf magnetron sputtering. *Vacuum* **75**, 183–188 (2004).
- Mori, N. *et al.* Effects of postannealing in ozone environment on opto-electrical properties of Sn-doped In₂O₃ thin films. *Thin Solid Films* **411**, 6–11 (2002).
- Kanehara, M., Koike, H., Yoshinaga, T. & Teranishi, T. Indium tin oxide nanoparticles with compositionally tunable surface plasmon resonance frequencies in the near-IR region. *J. Am. Chem. Soc.* **131**, 17736–17737 (2009).
- Li, S. Q. *et al.* Infrared plasmonics with indium-tin-oxide nanorod arrays. *ACS Nano* **5**, 9161–9170 (2011).
- Wang, Y. *et al.* Tunability of indium tin oxide materials for mid-infrared plasmonics applications. *Opt. Mater. Exp.* **7**, 2727 (2017).
- Warzecha, M. *et al.* High mobility annealing of transparent conductive oxides. *IOP Conf. Ser. Mater. Sci. Eng.* **34**, 012004 (2012).
- Palneedi, H. *et al.* Laser processing of metal oxides: Laser irradiation of metal oxide films and nanostructures: Applications and advances. *Adv. Mater.* **30**, 1870094 (2018).
- Noh, M. *et al.* Spectroscopic ellipsometry investigation on the excimer laser annealed indium thin oxide sol-gel films. *Curr. Appl. Phys.* **16**, 145–149 (2016).
- Noh, M. *et al.* Post-annealing effect on the optical property of indium tin oxide sputtered films. *Curr. Appl. Phys.* **16**, 1576–1580 (2016).
- Yarali, E. *et al.* Recent progress in photonic processing of metal-oxide transistors. *Adv. Func. Mater.* **30**, 1906022 (2020).
- Siozios, A. *et al.* Laser-matter interactions, phase changes and diffusion phenomena during laser annealing of plasmonic AlN: Ag templates and their applications in optical encoding. *J. Phys. D Appl. Phys.* **48**, 285306 (2015).
- Gao, J. *et al.* Physics of transparent conductors. *Adv. Phys.* **65**, 553–617 (2016).
- Fox, M. *Optical Properties of Solids*. (Oxford University Press, 2010).
- Tamanai, A., Dao, T. D., Sendner, M., Nagao, T. & Pucci, A. Mid-infrared optical and electrical properties of indium tin oxide films. *Phys. Status Solidi (A)* **214**, 1600467 (2017).
- Gonçalves, G. *et al.* Influence of post-annealing temperature on the properties exhibited by ITO, IZO and GZO thin films. *Thin Solid Films* **515**, 8562–8566 (2007).
- Jellison, G. E. & Modine, F. A. Parameterization of the optical functions of amorphous materials in the interband region. *Appl. Phys. Lett.* **69**, 371–373 (1996).
- Fujiwara, H. & Kondo, M. Effects of carrier concentration on the dielectric function of ZnO: Ga and In₂O₃: Sn studied by spectroscopic ellipsometry: Analysis of free-carrier and band-edge absorption. *Phys. Rev. B* **71**, 075109 (2005).
- Hillier, J. A. *et al.* When ellipsometry works best: A case study with transparent conductive oxides. *ACS Photon.* **7**, 2692–2702 (2020).
- Ellmer, K. & Mientus, R. Carrier transport in polycrystalline ITO and ZnO: Al II: The influence of grain barriers and boundaries. *Thin Solid Films* **516**, 5829–5835 (2008).
- Ellmer, K. & Mientus, R. Carrier transport in polycrystalline transparent conductive oxides: A comparative study of zinc oxide and indium oxide. *Thin Solid Films* **516**, 4620–4627 (2008).
- Wang, Z., Chen, C., Wu, K., Chong, H. & Ye, H. Transparent conductive oxides and their applications in near infrared plasmonics. *Phys. Status Solidi (A)* **216**, 1700794 (2019).
- Naik, G. V., Shalae, V. M. & Boltasseva, A. Alternative plasmonic materials: Beyond gold and silver. *Adv. Mater.* **25**, 3264–3294 (2013).
- Seiler, W., Nistor, M., Hebert, C. & Perrière, J. Epitaxial undoped indium oxide thin films: Structural and physical properties. *Sol. Energy Mater. Sol. Cells* **116**, 34–42 (2013).
- Jiamprasertboon, A. *et al.* Understanding structure, optical, and electrical properties of In₄Sn₃O₁₂ and In_{4.5}Sn₂M_{0.5}O₁₂ (M = Nb and Ta). *J. Alloys Compds.* **783**, 28–36 (2019).

33. Teixeira, V., Cui, H. N., Meng, L. J., Fortunato, E. & Martins, R. Amorphous ITO thin films prepared by DC sputtering for electrochromic applications. *Thin Solid Films* **420–421**, 70–75 (2002).
34. Thilakan, P. & Kumar, J. Studies on the preferred orientation changes and its influenced properties on ITO thin films. *Vacuum* **48**, 463–466 (1997).
35. Mei, F., Yuan, T. & Li, R. Effects of second-phase particles and elemental distributions of ITO targets on the properties of deposited ITO films. *Ceram. Int.* **43**, 8866–8872 (2017).
36. Warschkow, O., Ellis, D. E., González, G. B. & Mason, T. O. Defect structures of tin-doped indium oxide. *J. Am. Ceram. Soc.* **86**, 1700–1706 (2003).
37. Walsh, A. & Scanlon, D. O. Polymorphism of indium oxide: Materials physics of orthorhombic In₂O₃. *Phys. Rev. B* **88**, 161201 (2013).
38. de Keijsers, Th. H., Langford, J. I., Mittemeijer, E. J. & Vogels, A. B. P. Use of the Voigt function in a single-line method for the analysis of X-ray diffraction line broadening. *J. Appl. Crystallogr.* **15**, 308–314 (1982).
39. Holzwarth, U. & Gibson, N. The Scherrer equation versus the “Debye-Scherrer equation”. *Nat. Nanotechnol.* **6**, 534–534 (2011).
40. Bouhdjer, A. *et al.* Structural, morphological, optical, and electrical properties of In₂O₃ nanostructured thin films. *Optik (Stuttg)* **127**, 7319–7325 (2016).
41. Kim, H. J. *et al.* Chemical and structural analysis of low-temperature excimer-laser annealing in indium-tin oxide sol-gel films. *Curr. Appl. Phys.* **19**, 168–173 (2019).
42. Siozios, A. *et al.* Sub-surface laser nanostructuring in stratified metal/dielectric media: A versatile platform towards flexible, durable and large-scale plasmonic writing. *Nanotechnology* **26**, 155301 (2015).
43. Delli, E. *et al.* Laser-driven structural modifications and diffusion phenomena of plasmonic AlN/Ag stratified films. *Surf. Coat. Technol.* **295**, 46–53 (2016).
44. González, G. B. *et al.* Defect structure studies of bulk and nano-indium-tin oxide. *J. Appl. Phys.* **96**, 3912–3920 (2004).
45. Mergel, D., Schenkel, M., Ghebre, M. & Sulkowski, M. Structural and electrical properties of In₂O₃: Sn films prepared by radio-frequency sputtering. *Thin Solid Films* **392**, 91–97 (2001).
46. Kim, J. S. *et al.* X-ray photoelectron spectroscopy of surface-treated indium-tin oxide thin films. *Chem. Phys. Lett.* **315**, 307–312 (1999).
47. Fan, J. C. C. & Goodenough, J. B. X-ray photoemission spectroscopy studies of Sn-doped indium-oxide films. *J. Appl. Phys.* **48**, 3524–3531 (1977).
48. Park, J. H. *et al.* The effect of post-annealing on Indium Tin Oxide thin films by magnetron sputtering method. *Appl. Surf. Sci.* **307**, 388–392 (2014).
49. de Wit, J. H. W. Electrical properties of In₂O₃. *J. Solid State Chem.* **8**, 142–149 (1973).
50. Wagner, C. D. Sensitivity factors for XPS analysis of surface atoms. *J. Electron Spectrosc. Relat. Phenom.* **32**, 99–102 (1983).
51. Powell, C. J. Recommended Auger parameters for 42 elemental solids. *J. Electron Spectrosc. Relat. Phenom.* **185**, 1–3 (2012).
52. Powell, C. J. & Jablonski, A. Progress in quantitative surface analysis by X-ray photoelectron spectroscopy: Current status and perspectives. *J. Electron Spectrosc. Relat. Phenom.* **178–179**, 331–346 (2010).
53. Zhang, K., Zhu, F., Huan, C. H. A. & Wee, A. T. S. Indium tin oxide films prepared by radio frequency magnetron sputtering method at a low processing temperature. *Thin Solid Films* **376**, 255–263 (2000).
54. Husein, S. *et al.* Carrier scattering mechanisms limiting mobility in hydrogen-doped indium oxide. *J. Appl. Phys.* **123**, 245102 (2018).
55. Macco, B., Knoops, H. C. M. & Kessels, W. M. M. Electron scattering and doping mechanisms in solid-phase-crystallized In₂O₃: H prepared by atomic layer deposition. *ACS Appl. Mater. Interfaces.* **7**, 16723–16729 (2015).
56. Limpijumnong, S., Reunchan, P., Janotti, A. & van de Walle, C. G. Hydrogen doping in indium oxide: An ab initio study. *Phys. Rev. B* **80**, 193202 (2009).
57. Guler, U., Kildishev, A. V., Boltasseva, A. & Shalaev, V. M. Plasmonics on the slope of enlightenment: the role of transition metal nitrides. *Faraday Discuss.* **178**, 71–86 (2015).
58. Kim, S. & Choi, Y.-K. Resistive switching of aluminum oxide for flexible memory. *Appl. Phys. Lett.* **92**, 223508 (2008).
59. Gergel-Hackett, N. *et al.* A flexible solution-processed memristor. *IEEE Electron Dev. Lett.* **30**, 706–708 (2009).
60. Hillier, J. A. *Photo-Engineered Optoelectronic Properties of Transparent Conductive Oxides via Reactive Laser Annealing (ReLA): The Consequence of Defects.* (2021).
61. El Hamali, S. O. *et al.* Enhanced electrical and optical properties of room temperature deposited aluminium doped zinc oxide (AZO) thin films by excimer laser annealing. *Opt. Lasers Eng.* **80**, 45–51 (2016).
62. Hilfiker, J. N. & Tompkins, H. G. *Spectroscopic Ellipsometry: Practical Application to Thin Film Characterization.* (Momentum Press, 2015).

Acknowledgements

All authors acknowledge Prof. E. Lidorikis (University of Ioannina) for providing access to the opto-thermal code used to perform the calculations presented in the Supplementary, section F. J.A.H. acknowledges funding from the Nottingham Trent University Vice Chancellor’s Ph.D. Bursary Scheme. N.K. acknowledges support from Nottingham Trent University, School of Science and Technology capital equipment funding scheme for acquiring the Infrared Spectroscopic Ellipsometry system. S.C. acknowledges the “Région Nouvelle-Aquitaine” and the European Regional Development Fund (ERDF 773 2014–2020) for funding within the IMATOP Project No. P-2016-BAFE-209.

Author contributions

J.A.H. conceived and conducted the experiments and formal analysis and wrote the main manuscript text. P.P. and D.K. curated and analysed data. S.C., W.C., A.V.N., and C.J.M. curated data. D.C.K. and N.K. supervised and administrated the project. N.K. reviewed & edited the manuscript and performed the opto-thermal calculations. All authors reviewed the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary Information The online version contains supplementary material available at <https://doi.org/10.1038/s41598-022-18883-5>.

Correspondence and requests for materials should be addressed to N.K.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher's note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons licence, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons licence, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons licence and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this licence, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2022