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# Full Communication

# Role of anodically electrogenerated hydroxyl radicals in minimizing mineral cathodic electroprecipitation in the presence of hard water

high current intensities.

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## **1. Introduction**

As a result of more stringent regulations on the release of wastewater into aquatic media which have been designed to preserve the natural environment and to promote water reuse, the conventional biological and physicochemical treatments used in wastewater treatment plants (WWTPs) may soon no longer be sufficient  $[1-3]$ . Thus, a variety of complementary treatments has been proposed in the literature, and some have been applied at an industrial scale [4–7]. The majority of these processes operate using chemical oxidation with strong oxidants to eliminate pollutants [8–11]. These processes include classical oxidation processes such as chlorination, ozonation and ultraviolet irradiation, and, more recently, advanced oxidation processes (AOPs) designed to degrade organic contaminants. However, this increased efficiency comes with a high price tag: the requirement to use chemical reagents (e. g.,  $H_2O_2$ ,  $Cl_2$ ,  $O_3$ , acid/base, Fe<sup>2+</sup>, etc.) [12,13], which goes against the environmentally friendly approach to wastewater treatment. One category of AOPs is electrochemical advanced oxidation processes (EAOPs) [3,5,12,14–18], in which potent, quasi non-selective oxidizing agents such as hydroxyl radicals (<sup>'</sup>OH) and other mediated oxidants (e.g., hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>), HClO/ClO<sup>-</sup>, Cl<sub>2</sub>, O<sub>3,</sub> etc.) are electrogenerated in situ [18–20].

In the last decade, the introduction of submillimetric reactors within EAOPs has broadened the perspective of water treatment using electrochemical techniques [21–28]. A micrometric interelectrode distance could lift the mass transport limitation often encountered in macrometric electrochemical reactors [21,29]. Redox activities on each electrode could also contribute and may result in changes to the overall electrochemical/chemical reactions in the system [30]. This approach also makes it possible to operate with low-conductivity solutions, while minimizing the energy requirement [31–33].

However, electrolytic systems remain prone to electrode passivation, even under microfluidic conditions [30,34]. Depending on the potential applied to the cathode, the reduction of dissolved oxygen  $(O_2)$  (Eq.  $(1)$ ) and/or water (Eq.  $(2)$ ) occurs  $[35,36]$ :

$$
O2 + 2H2O + 4e^- \rightarrow 4OH^-
$$
 (1)

$$
2H_2O + 2e^- \rightarrow H_2 + 2OH^-
$$
 (2)

These reactions lead to a gradient in the hydroxyl ion (OH<sup>-</sup>) concentration near the cathode compared to that in bulk solution. A high local pH [30,34,37] during electrolysis could easily convert  $CO<sub>3</sub><sup>2</sup>$  into its

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predominant form, according to the acid–base constants given in Eqs. (3)-(4) [38,39]:

$$
K_{\text{Al}} = \frac{[\text{H}^+] [\text{HCO}_3^-]}{[\text{H}_2\text{CO}_3]}, pK_{\text{Al}} = 6.3
$$
\n(3)

$$
K_{A2} = \frac{[\text{H}^+][\text{CO}_3^{-}]}{[\text{HCO}_3^-]}, pK_{A2} = 10.4
$$
 (4)

Thus, depending on the concentration of magnesium  $(Mg^{2+})$  and calcium  $(Ca^{2+})$  ions in the effluent to be treated, Mg- and Ca-based passivating layers could be formed on the cathode surface, as shown in Eqs. (5)-(7):

$$
Mg^{2+} + 2OH^- \rightarrow Mg(OH)_2
$$
 (5)

 $OH^- + HCO_3^- \rightarrow H_2O + CO_3^2$  $\frac{2}{3}$  (6)

$$
Ca^{2+} + CO_3^{2-} \rightarrow CaCO_3 \tag{7}
$$

Such passivation layers can strongly impact the durability of EAOPs under continuous operation if regular and costly maintenance is not foreseen. Mechanical clean-in-place (CIP) techniques have demonstrated high descaling efficiency with smooth materials [40]. However, such methods are likely to be less useful for porous materials, bearing in mind that the electro-precipitates are located within the internal pores [41]. Most EAOPs require the use of porous material electrodes, particularly to implement flow-through design to increase mass transport [22,27,31,42]. Periodic polarity reversal is another successful descaling technique, which has been applied at industrial scale to prolong the electrodialysis process [43–46]. However, this technique may not be suitable for most EAOPs. Efficient descaling using this technique could only be obtained using a high applied current (*I*app) [43,44,47,48], which translates into an elevated overall operational cost. Also, only stable electrodes, such as several mixed metal oxides (MMO) (e.g. IrO<sub>2</sub>–RuO<sub>2</sub>/Ti, IrO<sub>2</sub>–RuO<sub>2</sub>–SnO<sub>2</sub>/Ti), Pt-coated Ti or boron-doped diamond (BDD), which are capable of resisting high current polarization in both cathode and anode configurations. Other common electrode materials (e.g., most steels and less noble metals) become corroded via anodic polarization at high *I*app, releasing ions into the solution. Carbonaceous materials can even be burnt when they are used as anodes under medium and high current densities [49]. Thus, reducing the frequency of polarity reversal or even proposing an alternative to avoid this technique by a self-cleaning mechanism would challenge the current practice and enlarge the application field of applied electrochemistry, including EAOPs.

Surprisingly, the role of • OH in mineral scale formation has never been studied and is thus the goal of this study. Moreover, the involvement of anodic activity is typically stronger under microfluidic conditions and may play a role in the mechanism of electroprecipitation on the cathode and on pollutant degradation via <sup>•</sup>OH generated at the anode or mediated oxidants in the bulk. HCO $_3^\circ$ /CO $_3^{2-}$  is a  $^\bullet$ OH scavenger via the formation of carbonate radicals [50,51], which could reduce the carbonate concentration available near the cathode, possibly resulting in less calcareous scaling. This hypothesis is studied for the first time in this paper using a systematic investigation using synthetic wastewater while progressively increasing the applied current density to assess the influence of anodic • OH on cathodic mineral scaling.

#### **2. Materials and methods**

## *2.1. Chemicals*

Information about the chemicals used in this study is given in the Supplementary Information (Text S1).

#### *2.2. Preparation of effluent*

Experiments were performed using a synthetic effluent containing 3.75 mmol L<sup>-1</sup> (150 mg of L<sup>-1</sup>) Ca<sup>2+</sup>, 0.21 mmol L<sup>-1</sup> (5 mg of L<sup>-1</sup>) Mg<sup>2+</sup> and 5 mmol  $L^{-1}$  (60 mg of C  $L^{-1}$ ) bicarbonates by dissolving 1.2887 g of CaSO<sub>4</sub>⋅2H<sub>2</sub>O, 0.1014 g of MgSO<sub>4</sub>⋅7H<sub>2</sub>O and 0.8400 g of NaHCO<sub>3</sub> in 2 L of ultrapure water (18.2 M $\Omega$  cm) provided by a PureLab ELGA water purification dispenser (Veolia Water, Antony, France). No additional supporting electrolyte was used. The initial pH was adjusted to 7.6 by adding 650 µL of 1 mol L<sup>-1</sup> sulfuric acid. The eventual electrical conductivity of the synthetic effluent was approximately 1000  $\mu$ S cm<sup>-1</sup>, with the ionic strength estimated to be 0.02 mol  $L^{-1}$ . These properties were selected to emulate the characteristics of municipal wastewater plant effluent [30,34].

#### *2.3. Electrochemical setup*

The electrochemical setup consisted of a vertical parallel-plate filter press reactor, composed of a stainless steel cathode and a BDD anode separated by a spacer of 500 µm thick. The setup is described elsewhere [29,30,34,41] and more details are given in the Supplementary Text S3.

#### *2.4. Analytical methods*

Details of the analytical methods used in this study are given in the Supplementary Text S2.

#### *2.5. Development of the electroprecipitation model*

Our model of electroprecipitation kinetics is based on recent studies in which the electroprecipitations occurring within submillimetric to millimetric interelectrode distances were modeled [30,34]. Continuous reduction of dissolved  $O_2$  (Eq. (1)) and H<sub>2</sub>O (Eq. (2)) to OH<sup>-</sup> could be modeled via a modified Faraday's law as shown in Eqs. (8)-(9):

$$
r_{\text{OH}^{-}} = + \frac{\text{d}[\text{OH}^{-}]}{\text{d}t} = \frac{\nu_{\text{O}_2} j_{\text{app}}}{n_{\text{O}_2} F} C E_{\text{O}_2} + \alpha \frac{j_{\text{app}}}{F}
$$
(8)

$$
\alpha = \nu_{\text{H}_2\text{O}} \frac{CE_{\text{H}_2\text{O}}}{n_{\text{H}_2\text{O}}} - \nu_{\text{H}_2} \frac{CE_{\text{H}_2}}{n_{\text{H}_2}}
$$
(9)

where  $r_{OH^-}$  is the heterogeneous production rate of OH<sup>–</sup> (mol m<sup>-2</sup>  $min^{-1}$ ) (the concentration is expressed in mol m<sup>-2</sup> due to negligible thickness of the reaction zone with respect to the electrode surface [30]);  $\alpha$  is the gas hindrance factor related to  $H_2$  bubble production on the cathode originating from the H<sub>2</sub>O reduction reaction shown in Eq.  $(2)$ [30]; *j*<sub>app</sub> is the applied current density (A m<sup>-2</sup>); *F* is the Faraday constant (96,485 C mol<sup>-1</sup>);  $n_{O_2}$  (=4) and  $n_{H_2O}$  (=2) are the numbers of electrons participating in Eq. (1) and Eq. (2), respectively  $[30,34]$ ;  $n_{\text{H}_2}$  (=2) reacted to produce  $H_2$  with respect to the stoichiometry of OH<sup> $-$ </sup> production from Eq. (2);  $\nu_{O_2}$  (=4) and  $\nu_{H_2O}$  (=2) are the stoichiometric coefficients of OH<sup>-</sup> produced by Eq. (1) and Eq. (2), respectively;  $\nu_{\text{H}_2}$  (=1) is the stoichiometry of  $H_2$  production with respect to that of OH<sup>-</sup> in Eq. (2);  $CE_{O_2}$  and  $CE_{H_2O}$  are the current efficiencies attributed to the Faradaic reactions in Eq.  $(1)$  and Eq.  $(2)$ ; and  $CE_{H_2}$  is the proportion of Faradaic current spent due to the concomitant  $H_2$  evolution in Eq. (2).  $CE_{O_2}$  and  $CE_{\text{H}_2\text{O}}$  were determined experimentally at different applied currents using the Ti(IV)- $H_2O_2$  complexation method described elsewhere [30,52]. Because  $CE_{H_2O}$  and  $CE_{H_2}$  appear in the same half equation of Eq. (2) whereby the reduction of  $H_2O$  and the evolution of  $H_2$  take place on the same cathodic surface, it has been hypothesized that the current efficiency  $CE_{H_2}$  equals  $CE_{H_2O}$  [30]. Thus,  $\alpha$  (Eq. (9)) can also be deduced. The current efficiencies and the value of  $\alpha$  as a function of the applied current are summarized in Table S1. Detailed explanations and calculations are given in the Supplementary Text S4.

The rate of CaCO<sub>3</sub> electroprecipitation  $(r_{CaCO<sub>3</sub>})$  was modeled using

# the kinetic laws given in Eqs.  $(10)-(11)$ :

$$
r_{Ca^{2+}} = -r_{CaCO_3} = -(k_{CaCO_3} - k_{det, CaCO_3}) \left[ Ca^{2+} \right] \left[ CO_3^{2-} \right]_{int}
$$
 (10)

$$
\left[\text{CO}_3^{2-}\right]_{\text{int}} = \frac{K_{\text{A2}} \times \left[\text{TIC}\right]}{\frac{K_{\text{W}}}{\left[\text{OH}^-\right]} + K_{\text{A2}}} - k_{\bullet \text{OH}} \cdot \left[\text{TIC}\right].t
$$
\n(11)

where  $r_{\text{Ca}^{2+}}$  is the rate of heterogeneous Ca<sup>2+</sup> precipitation (mol m<sup>-2</sup>  $\text{min}^{-1}$ );  $k_{\text{CaCO}_3}$  is the second-order rate constant of heterogeneous electroprecipitation of CaCO<sub>3</sub> (3.6 × 10<sup>-3</sup> m<sup>4</sup> mol<sup>-1</sup> min<sup>-1</sup> [30,34];  $k_{\text{det,CaCO_3}}$  $(m^4 \text{ mol}^{-1} \text{ min}^{-1})$  is the CaCO<sub>3</sub> detachment rate constant due to concomitant evolving gas, which is dependent on the applied current;  $[Ca<sup>2+</sup>]$  is the molar concentration of  $Ca<sup>2+</sup>$  (mol m<sup>-3</sup>); *K*<sub>A2</sub> is the acidity constant of the HCO<sub>3</sub>/CO<sub>3</sub><sup>2</sup> acid–base couple (Eq. (4)) ( $pK_{A2} = 10.4$  $\frac{[38]}{[38]}$ ; and  $\left[\text{CO}_3^2\right]$  $\int_{\text{int}}$  is the theoretical interfacial concentration of carbonates on the cathode (mol  $m^{-3}$ ), which is modeled using Eq. (11). Eq.  $(11)$  is a function of **(1)** the concentration (mol m<sup>-3</sup>) of total inorganic carbon ([TIC], defined as  $[HCO_3^-] + [CO_3^{2-}]$ ), ignoring carbonic acid given the range of investigated pH values in the bulk and **(2)** the concentration of OH<sup>-</sup> produced via Eqs.  $(8)$ - $(9)$ .  $k$ <sub>OH</sub> is newly introduced and represents the pseudo first-order rate constant for conversion of carbonates into carbonate radicals  $(CO_3^{\bullet-})$  by the intermediary  $\bullet$ OH  $(\text{min}^{-1})$  under conditions where the latter is present.

The modeling of  $Ca^{2+}$  concentration evolution was performed using Aquasim® software [53]. The root mean square error (RMSE) (Eq.  $(12)$ ) [54] was used to determine the fit between experimental and modeled points:

RMSE = 
$$
\sqrt{\frac{\sum_{i=1}^{K} (y_i - y_i')^2}{K}}
$$
 (12)

where  $y'_i$  is the experimental value,  $y_i$  is the model value and *K* is the number of iteration.

# **3. Results and discussion**

#### 3.1. *Kinetics of electroprecipitation of CaCO<sub>3</sub> at different applied currents*

Fig. 1 shows the evolution of the experimental and modeled  $Ca^{2+}$  and total inorganic carbon (TIC) concentrations during electrolysis at 40, 200 and 800 mA.  $Ca^{2+}$  precipitated at 54.4  $\pm$  1.9%, 49.8  $\pm$  5.6% and  $46.9 \pm 1.2$ % at 40, 200 and 800 mA, respectively, while TIC precipitated at 92.4  $\pm$  3.9%, 87.6  $\pm$  1.7% and 84.4  $\pm$  4.8% with increasing  $I_{\rm app}$ . A rise in CaCO3 electroprecipitation was noted as *I*app was increased. The associated changes in bulk pH and ionic conductivity are displayed in Fig. S1 and discussed in the Supplementary Text S5.

Fig. 2 shows the molar ratio of  $Ca^{2+}$  concentration with respect to TIC concentration over the course of a 5 h electrolysis as well as the molar ratio of precipitated  $Ca^{2+}$  compared to precipitated carbonates. In Fig. 2a, the ratio  $\left[\text{Ca}^{2+}\right] / \left[\text{TIC}\right]$  was less than one (0.64 to 0.77) because there were more moles of carbonate than of  $Ca^{2+}$ , meaning that the concentration of  $Ca^{2+}$  was the limiting factor. As the electrolysis time increased, the ratio  $\left[\text{Ca}^{2+}\right] / \left[\text{TIC}\right]$  increased due to a decrease in both  $Ca<sup>2+</sup>$  and carbonate concentrations, which suggests that the consumption of the latter was quicker than that of the former. This result is supported by the TIC plot in Fig. 1b. The quicker decrease in carbonate compared to  $Ca^{2+}$ , whatever the applied current, could be due to their protonation into carbonic acid and then into  $CO<sub>2</sub>$ , caused by the local acidification at the anode surface through  $H_2O$  oxidation. The deduction made from Fig. 1 was replicated in this study (see Fig. 2b). The molar ratio of precipitated  $\Delta$ [Ca<sup>2+</sup>] over  $\Delta$ [TIC] at the end of electrolysis was determined to be  $0.39 \pm 0.02$ ,  $0.40 \pm 0.04$  and  $0.35 \pm 0.02$  for 40, 200 and 800 mA current, respectively. The ratio  $\Delta [Ca^{2+}]/\Delta [TIC]$  was the lowest at the highest *I*app (800 mA). This lower ratio was the outcome of lower precipitation of  $Ca^{2+}$  (i.e., higher  $Ca^{2+}$  content in the bulk) and could also be due to the reaction of carbonates with other compounds, which is discussed in Sections 3.3 and 3.4.

# 3.2. Role of measured electrode potentials on CaCO<sub>3</sub> electroprecipitation *at different applied currents*

When an electrochemical cell is polarized at a given *I*<sub>app</sub> using a current generator, a specific potential will eventually be applied to both the anode and the cathode. The applied potentials depend on the applied current and on the configuration of the electrochemical reactor [29,30]. Fig. 3 shows the cathode potential  $(E_C)$  as a function of the percentage of CaCO<sub>3</sub> deposited, together with the anode potential  $(E_A)$ , cell voltage (Δ*U*) and ohmic drop ( <sup>∑</sup>*RI*) at all investigated *I*app values. In agreement with Fig. 3a, *E*<sub>C</sub> was found to be -1.40, -1.54 and -2.30 V/Ag-AgCl when the current was 40, 200 and 800 mA, respectively. A previous study showed that as  $E_C$  falls deeper into the water reduction region  $[30]$ ,  $E<sub>C</sub>$  moves further away from the standard potential of O<sub>2</sub> reduction into OH<sup> $-$ </sup> via four electrons (Eq.  $(1)$ ), which occurs with minimum parasitic hydrogen gas  $(H<sub>2</sub>)$  formation [55,56]. The importance of gas evolution on cathode scaling was emphasized in a recent article dealing with the effect of  $d_{\text{elec}}$  [30]. The increase in  $d_{\text{elec}}$  decreased  $E_C$  and thus increased scale detachment. Similarly, as a higher *I*<sub>app</sub> is applied in this study, a lower percentage of cathodic  $CaCO<sub>3</sub>$  deposition was observed, partly due to the detachment associated with intense  $H_2$  evolution.

Also, as  $I_{\text{app}}$  increased, the measured  $E_A$  and  $\Delta U$  also increased.  $E_A$ was experimentally determined to be 2.19, 2.50 and 2.78 V/Ag-AgCl,



**Fig. 1.** Experimental and modeled kinetics of the concentration of (a) Ca<sup>2+</sup> and (b) TIC at 40, 200 and 800 mA.



**Fig. 2.** Molar ratio of (a) Ca<sup>2+</sup> concentration with respect to that of TIC and (b) precipitated Ca<sup>2+</sup> with respect to precipitated carbonates.



**Fig. 3.** Relationship between (a) measured  $E<sub>C</sub>$  with the percentage of electroprecipitation and (b) measured *E*A, Δ*U* and the corresponding ∑*RI* as a function of applied current.

while Δ*U* was measured to be 3.07, 4.03 and 4.90 V at currents of 40, 200 and 800 mA, respectively. This increase in cell voltage was expected because a higher *I*app was applied. As *I*app increased, overpotentials on both the anode and cathode also increased with respect to the equilibrium potentials  $[57]$ . H<sub>2</sub> and O<sub>2</sub> gas bubbles produced on the cathode and anode, respectively, were insulating in nature [58]. Thus, the

internal resistance of the cell [30] also increased from 2.28 to 4.42  $Ω$  as  $I_{\text{ann}}$  increased from 40 to 800 mA (Fig. 3b). Elsewhere, the measured  $E_A$ has a fundamental impact on the presence of radical species, as presented in the next section.

#### *3.3. Validity of carbonate reactivity under advanced electrooxidation*

The possible parallel reactions of sulfates and carbonates have also been investigated. A synthetic solution containing only  $HCO_3^-/CO_3^{2-}$  at 5 mmol L<sup>-1</sup> (60 mg-C L<sup>-1</sup>) and SO<sup>2</sup><sup>-</sup> at 5 mmol L<sup>-1</sup> was studied at 40, 200 and 800 mA (Fig. 4). The evolution of sulfate concentration remained constant regardless of the applied current (Fig. 4a). This demonstrates the lack of reactivity of sulfates during electrolysis over this range of current densities with an interelectrode gap of 500 µm, while stronger operating conditions are known to form the sulfate radical at BDD anode surfaces [59]. In contrast, the evolution of TIC (i.e., carbonates) differed according to the applied current, see Fig. 4b. The carbonate concentration remained constant throughout electrolysis at 40 and 200 mA. However, at a higher current (800 mA), their concentration decreased exponentially with time  $(R^2 = 0.91)$  (Fig. 4c). At this intensity,  $E_A$  is around 3.0 V/Normal hydrogen electrode (NHE) (Fig. 3b). This is sufficient to generate physically sorbed "OH at the BDD anode surface from H<sub>2</sub>O oxidation (Eq.  $(13)$ ), given its high O<sub>2</sub> evolution overvoltage (2.2–2.8 V/NHE) as well as the standard potential of the • OH/H2O redox couple, which is equal to 2.8 V/NHE [4–7]:

$$
BDD + H_2O \rightarrow BDD(\bullet OH) + e^- + H^+ \tag{13}
$$

<sup>•</sup>OH radicals are known to react with HCO<sub>3</sub> and  $CO<sub>3</sub><sup>2</sup>$  to form carbonate radicals ( $CO<sub>3</sub><sup>•</sup>$ ), via a pseudo first-order reaction assuming a quasi steady-state approximation toward <sup>\*</sup>OH [50,51,60]. Thus, none of the carbonates was available to react with  $Ca^{2+}$  when the local pH was sufficiently high. This result could explain the lower  $\Delta$ [Ca<sup>2+</sup>]/ $\Delta$ [TIC] molar ratio at 800 mA, as shown in Fig. 2b. Additionally, the microdistance (500  $\mu$ m) used in the cell could favor the transport of carbonates towards the • OH present in the vicinity of the anode surface, because such a small distance enhances mass transfer, as reported previously with a cell of the same design [21,29].

This result also highlighted the unprecedented anodic contribution to the limitation of cathodic CaCO<sub>3</sub> electroprecipitation at the current density leading to the formation of • OH. The contribution of radicals has been considered in a new mathematical model described in the following section and in the Supplementary Information.



**Fig. 4.** Variation of (a)  $SO_4^{2-}$  and ((b), (c)) TIC in blank solution containing only HCO<sub>3</sub>/CO<sup>2−</sup> (5 mmol L<sup>−1</sup>; 60 mg-C L<sup>−1</sup>) and SO<sup>2−</sup> (5 mmol L<sup>−1</sup>) during electrooxidation at 40, 200 and 800 mA.

# *3.4. Kinetics modeling and mechanistic scheme of CaCO3 electroprecipitation*

Considering the results presented in the previous sections, a novel kinetic model was developed for CaCO<sub>3</sub> electroprecipitation in a microfluidic thin film reactor under different current intensity conditions. This model is described in detail in the Methods section, the associated curves are plotted in Fig. 2 against experimental data, and Table S2 shows the primary constants involved in the model. Table S3 summarizes the data (initial conditions, variables and processes) computed using the Aquasim® software assuming a mixed reactor compartment type as well as a constant volume ( $5 \times 10^{-4}$  m<sup>3</sup>) with zero input (batch system). The difference in the kinetics of electroprecipitation depending on the applied current was modeled by considering **(i)** the gas hindrance factor (*α*) [30]; **(ii)** scale detachment  $(k<sub>det</sub>)$  at the cathode due to high H<sub>2</sub> gas evolution at sufficiently low  $E_C$ values (200 and 800 mA conditions) [30]; and **(iii)** carbonate reactions with <sup>•</sup>OH ( $k_{\bullet$ OH) at higher currents (800 mA) when  $E_A$  was sufficiently high to produce <sup>•</sup>OH (newly introduced in this study).  $k_{\bullet\rm OH}$  was determined experimentally by considering the slope in Fig. 4c (i.e., 3.83  $\times$  $10^{-4}$  min<sup>-1</sup>).

RMSE values indicated good fits, particularly with the kinetics of Ca2<sup>+</sup> concentration (RMSE *<* 0.19), while RMSE values were below 0.42 for TIC evolution. These results confirmed the critical role of current density in scale detachment and carbonate reactivity. A mechanistic scheme illustrating the different factors that affect  $CaCO<sub>3</sub>$  electroprecipitation is shown in Fig. S2.

## **4. Conclusions**

The present work was stimulated by an interest in developing microreactors that not only allow electrical energy consumption to be minimized, but also operate with low-salinity effluents while decreasing electroprecipitation phenomena at a sufficiently high current density. The results demonstrate the possibility of limiting the frequency of polarity reversal at an industrial scale, though the anodic role (accumulation of  $^{\bullet}$ OH and H<sup>+</sup>) could be minimized depending on the operating conditions and the characteristics of the solution being treated. Moreover, this study highlights an additional feature to consider: the kinetics of • OH reactivity in the degradation of organic pollutants during advanced electro-oxidation treatment in the presence of hard water.

There is still a need to carry out studies in presence of more complex matrices, whilst the generation of unwanted byproducts under various applied currents should be monitored and controlled. Other radical scavengers (nitrate, organic matter, etc.) could be in competition with carbonates and therefore limit the anti-scaling role of • OH. Moreover, the lower treatment capacity and higher clogging issues of microreactors encourages researchers to consider innovative designs [25,27,61]. A reactive electro-mixing reactor that combines both micro- and macroreactors could be one possibility [25,27,61]. Finally, further studies are required to compare the cost efficiency of the anti-scaling effect demonstrated in this paper with other conventional descaling technologies (e.g. mechanical techniques).

# **CRediT authorship contribution statement**

**Faidzul Hakim Adnan:** Methodology, Investigation, Software, Formal analysis, Writing – original draft. **Steve Pontvianne:** Investigation. Marie-Noëlle Pons: Supervision, Writing – review & editing, Funding acquisition. **Emmanuel Mousset:** Conceptualization, Methodology, Software, Validation, Supervision, Writing – original draft, Writing – review & editing, Project administration, Resources, Funding acquisition.

# **Declaration of Competing Interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### **Data availability**

Data will be made available on request.

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# **Appendix A. Supplementary data**

Supplementary data to this article can be found online at [https://doi.](https://doi.org/10.1016/j.elecom.2023.107493)  [org/10.1016/j.elecom.2023.107493](https://doi.org/10.1016/j.elecom.2023.107493).

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