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## Efficient continuous-wave Tm, Ho: CaF<sub>2</sub> laser at 2.1 μm

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We report on the first continuous-wave laser operation of a Tm³+,Ho³+-codoped calcium fluoride crystal at ~2.1 µm. Tm,Ho:CaF² crystals were grown by the Bridgman method and their spectroscopic properties were studied. The stimulated-emission cross-section for the  $^5 I_7 \rightarrow ^5 I_8$  Ho³+ transition is  $0.72\times 10^{-20}$  cm² at 2042 nm and the thermal equilibrium decay time is 11.0 ms. A 3 at.% Tm, 0.3 at.% Ho:CaF² laser generated 737 mW at 2062-2088 nm with a slope efficiency of 28.0% and a laser threshold of 133 mW. A continuous wavelength tuning between 1985 - 2114 nm (tuning range: 129 nm) was demonstrated. The Tm,Ho:CaF² crystals are promising for ultrashort pulse generation at ~2 µm. 2023 Optical Society of America

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Calcium fluoride (CaF<sub>2</sub>) is a cubic crystal (sp. gr. Fm3m). Ca<sup>2+</sup> cations are bonded in a body-centered cubic geometry to eight F<sup>-</sup> anions. Due to its broadband transparency (0.13 – 10  $\mu$ m), low refractive index (n=1.424 at ~2  $\mu$ m) and high thermal conductivity ( $\kappa=9.7$  Wm<sup>-1</sup>K<sup>-1</sup>), CaF<sub>2</sub> is widely used in optical components. It is also very attractive for doping with laser-active trivalent rare-earth ions (RE<sup>3+</sup>) replacing for the Ca<sup>2+</sup> cations [1-3]. Even at moderate doping levels (>0.1 at.% RE<sup>3+</sup>), calcium fluoride crystals exhibit RE<sup>3+</sup> ion clustering [4,5] resulting in a strong inhomogeneous broadening of absorption and emission bands (a "glassy-like" behavior) and enhancement of the energy-transfer processes [6]. The broadband emission and good thermal properties of RE<sup>3+</sup>-doped CaF<sub>2</sub> crystals determine their applications in high-power / high-energy ultrafast lasers and amplifiers at ~1  $\mu$ m (for Yb<sup>3+</sup> and Nd<sup>3+</sup> doping) [7,8], as well as broadly tunable lasers at ~2  $\mu$ m (Tm<sup>3+</sup> doping) [9,10].

Holmium ions (Ho³+) ions (electronic configuration: [Xe]4f¹0) are known for their emission in the eye-safe spectral range at  $\sim 2~\mu m$  originating from the  $^5l_7 \rightarrow ^5l_8$  transition [11]. Ho lasers emitting slightly above 2  $\mu m$  are of interest for LIght Detection And Ranging (LIDAR) systems for gas sensing and wind mapping, medical applications, material processing and pumping of mid-infrared laser sources, such as Cr²+ lasers and optical parametric oscillators based on non-oxide crystals.

There exist two approaches for excitation of Ho<sup>3+</sup> ions to the <sup>5</sup>I<sub>7</sub> metastable state: (i) Tm3+,Ho3+ codoping [11,12] and (ii) resonant (in-band) pumping [13,14]. In the first approach, the Tm<sup>3+</sup> ions (donors, D) efficiently absorb the pump radiation at  $\sim$ 0.8  $\mu m$  (the spectral range well addressed by commercial and high power AlGaAs laser diodes) and non-radiatively transfer their energy to  $\text{Ho}^{3+}$  ones (acceptors, A),  ${}^3F_4(\text{Tm}^{3+}) \rightarrow {}^5I_7(\text{Ho}^{3+})$ . The predominantly unidirectional energy transfer (ET)  $D \rightarrow A$  is maintained by an appropriate Ho<sup>3+</sup>/Tm<sup>3+</sup> codoping ratio, between 1:5 to 1:10, while the reduced energy-transfer upconversion (ETU) from the <sup>5</sup>I<sub>7</sub> Ho<sup>3+</sup> metastable state is ensured by choosing relatively low Ho3+ doping levels (<0.5 at.%). Efficient Tm, Ho lasers exist [12,15]. The codoping is also beneficial for combining the gain bandwidths of both ions, which is attractive for broadly tunable and mode-locked lasers at  $\sim$ 2 µm [16]. The thermal issues of Tm,Ho lasers can be mitigated by applying the fiber laser geometry [17].

Resonant pumping of Ho<sup>3+</sup> ions directly to the upper laser level presents a costly alternative to the codoping scheme because of the need of expensive and complex pump sources, such as Tm fiber lasers [14] or GaSb-based laser diodes [15]. In-band pumping offers higher laser slope efficiencies [18] and weaker heat loading as compared to the codoping scheme.

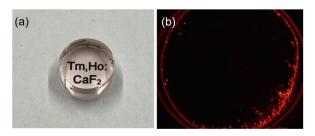
The multi-site behavior of low-doped Ho:CaF2 crystals was studied [19]. In 1965, the first lasing on the  ${}^5S_2 \rightarrow {}^5I_8$  transition in Ho:CaF<sub>2</sub> was reported at 77 K [20]. Recently, the interest shifted to in-band pumped Ho:CaF<sub>2</sub> lasers operating on the  ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$  transition [21-23]. Jelínek et al. reported on a cryogenic (83 K) continuouswave (CW) 0.5 at.% Ho:CaF<sub>2</sub> laser pumped by a Tm fiber laser at 1940 nm delivering an output power of 2.37 W at 2060 nm with a slope efficiency n of 23% [21]. Duan et al. reported on a continuouswave room-temperature 1 at.% Ho:CaF2 laser with a similar pump source generating 6.94 W at 2101 nm with a higher  $\eta$  of 57.9% [23]. Sulc et al. reported on a free-running 2 at.% Tm, 0.3 at.% Ho:CaF<sub>2</sub> ceramic laser diode-pumped at 785 nm delivering an output pulse energy of 4.1 mJ at 2099 nm with  $\eta = 10.8\%$  being tunable from 2016 to 2111 nm [24]. The same group employed a 1700 nm laser diode for Tm resonant pumping and a peak output power of 30 mW at 2074 nm was extracted with still low  $\eta$  of 5.3% [25].

In the present work, we report on the growth, spectroscopy and room-temperature continuous-wave laser operation of Tm<sup>3+</sup>, Ho<sup>3+</sup>-

codoped CaF2 crystals with the goal of developing gain media with broadband emission properties above 2  $\mu m$ .

Single-crystals of CaF<sub>2</sub> codoped with 3-5 at.% Tm<sup>3+</sup> and 0.3-0.5 at.% Ho<sup>3+</sup> (with respect to Ca<sup>2+</sup>) were grown by the Bridgman-Stockbarger technique using graphite crucibles (Φ8 mm, height: 40 mm). The starting reagents were CaF<sub>2</sub>, TmF<sub>3</sub> and HoF<sub>3</sub> powders (purity: 4N), the rare-earth fluorides were achieved by fluorination of the corresponding oxides. A special attention was paid to reduce the oxygen contamination: the growth chamber was sealed to vacuum (<10-5 mbar) and refilled with Ar + CF<sub>4</sub> gases. The starting reagents were mixed and placed into the crucible which was heated slightly above (~40 °C) the melting point (1418 °C) and the solution was homogenized for 3-4 h. The growth was ensured by translating the crucible in a vertical temperature gradient of 30-40 °C/cm. After the growth was completed, the crystals were slowly cooled down to room temperature within 48 h. The segregation coefficients for Tm<sup>3+</sup> and Ho<sup>3+</sup> ions in CaF<sub>2</sub> are 0.94 and 0.98, respectively. The actual ion densities are  $N_{\rm Tm} = 6.91$  and  $N_{\rm Ho} = 0.72$  [ $10^{20}$  at/cm<sup>3</sup>] for the 3 at.% Tm<sup>3+</sup>, 0.3 at.% Ho<sup>3+</sup>:CaF<sub>2</sub> crystal.

The as-grown crystals were transparent with a slight rose coloration due to Ho<sup>3+</sup> doping, Fig. 1(a). A Schlieren photography of a polished cylindrical barrel cut from the central part of the crystal confirmed its high optical quality, Fig. 1(b).

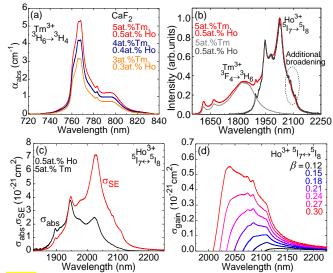


**Fig. 1.** (a) A photograph and (b) a Schlieren photography of a polished cylindrical barrel cut from the central part of an as-grown 3 at.%  $Tm^{3+}$ , 0.3 at.%  $Ho^{3+}$ :CaF<sub>2</sub> crystal.

The absorption spectra of Tm,Ho:CaF<sub>2</sub> crystals corresponding to the  ${}^3H_6 \rightarrow {}^3H_4$  Tm ${}^{3+}$  pump transition are shown in Fig. 2(a). The peak absorption cross-section  $\sigma_{abs}$  is  $0.46 \times 10^{-20}$  cm ${}^2$  at 768.1 nm and the corresponding bandwidth (full width at half maximum, FWHM) is 12.6 nm, which is beneficial for diode-pumping. Figure 2(b) shows the luminescence spectrum of a codoped crystal around 2  $\mu$ m. It is smooth and broad spanning from 1.55 to 2.25  $\mu$ m due to two spectrally overlapping transitions,  ${}^3F_4 \rightarrow {}^3H_6$  Tm ${}^3+$  and  ${}^5I_7 \rightarrow {}^5I_8$  Ho ${}^3+$ , with the Ho ${}^3+$  emission peaking at 2025 nm being more intense. As compared to singly Ho ${}^3+$ -doped crystals, the Tm ${}^3+$ ,Ho ${}^3+$  codoping induces additional broadening of the Ho ${}^3+$  emission band due to more profound ion clustering at higher total RE ${}^3+$  doping levels.

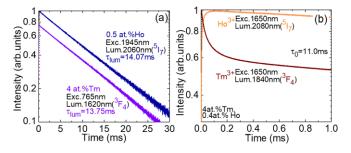
The absorption,  $\sigma_{\text{abs}}$ , and stimulated-emission cross-section,  $\sigma_{\text{SE}}$ , spectra for the  ${}^5\text{I}_8 \leftrightarrow {}^5\text{I}_7$  Ho ${}^{3+}$  transition in the  ${}^6\text{S}$  at  ${}^9\text{M}$  Tm ${}^{3+}$ , 0.5 at. ${}^9\text{M}$  Ho ${}^{3+}$  codoped CaF $_2$  crystal are shown in Fig. 2(c). The maximum  $\sigma_{\text{SE}}$  is  $0.72 \times 10^{-20}$  cm $^2$  at 2042 nm, as calculated using the Füchtbauer–Ladenburg formula. The  ${}^5\text{I}_7$  radiative lifetime,  $\tau_{\text{rad}} = 12.6$  ms, was determined using a modified Judd-Ofelt theory [26]. According to the quasi-three-level nature of the Ho ${}^{3+}$  laser scheme, the gain cross-sections for Ho ${}^{3+}$  ions,  $\sigma_{\text{gain}} = \beta \sigma_{\text{SE}} - (1 - \beta) \sigma_{\text{abs}}$ , were calculated for different inversion ratios  $\beta = N_2/N_{\text{Ho}}$ , where  $N_2$  is the population of the upper laser level ( ${}^5\text{I}_7$ ), Fig. 2(d). The gain profiles are smooth and

broad and their maximum experiences a blue-shift with increasing the inversion ratio, from  $21\frac{30}{10}$  nm ( $\beta = 0.1\frac{2}{10}$ ) to  $20\frac{40}{10}$  nm ( $\beta = 0.30$ ). For an intermediate  $\beta = 0.24$ , the gain bandwidth (FWHM) is 77 nm.



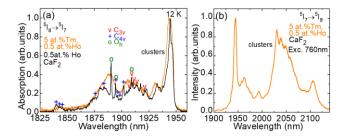
**Fig. 2.** Spectroscopy of Tm,Ho:CaF<sub>2</sub> crystals: (a) absorption spectra, the  ${}^{3}\text{H}_{6} \rightarrow {}^{3}\text{H}_{4}$  Tm ${}^{3+}$  transition; (b) luminescence spectrum around 2 μm, 5 at.% Tm ${}^{3+}$ , 0.5 at.% Ho ${}^{3+}$  codoped crystal, the spectra for singly-doped crystals are given for comparison; (c,d) 5 at.% Tm ${}^{3+}$ , 0.5 at.% Ho ${}^{3+}$ :CaF<sub>2</sub> crystal, the  ${}^{5}\text{I}_{8} \leftrightarrow {}^{5}\text{I}_{7}$  Ho ${}^{3+}$  transition: (c) absorption,  $\sigma_{abs}$ , and stimulatedemission,  $\sigma_{SE}$ , cross-sections, (d) gain cross-sections for Ho ${}^{3+}$  ions,  $\sigma_{gain}$ ,  $\beta = N_{2}({}^{5}\text{I}_{7})/N_{Ho}$  – inversion ratio.

The luminescence decay curves from the  ${}^3F_4$  Tm ${}^{3+}$  and  ${}^5I_7$  Ho ${}^{3+}$  states were measured under resonant excitation for singly-doped 4 at% Tm:CaF2 and 0.5 at% Ho:CaF2 crystals, Fig. 3(a). The decay curves for both crystals are nearly single exponential yielding luminescence lifetimes  $\tau_{lum}$  of 13.75 ms and 14.07 ms, respectively. The luminescence dynamics were studied for a 4 at.% Tm, 0.4 at.% Ho:CaF2 crystal under resonant Tm $^{3+}$  excitation, see Fig. 3(b). The decay curves are typical for the Tm $^{3+}$ ,Ho $^{3+}$  coupled systems indicating an efficient and predominantly unidirectional  ${}^3F_4$ (Tm $^{3+}$ )  $\rightarrow$   ${}^5I_7$ (Ho $^{3+}$ ) ET. The observation of two different time scales for the rise of Ho $^{3+}$  luminescence and the beginning of the Tm $^{3+}$  decay indicates the presence of two ion cluster compositions: (i) those containing both dopants and (ii) those being composed of only Tm $^{3+}$  ions. The thermal equilibrium decay time for both ions is 11.0 ms.

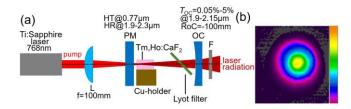


**Fig. 3.** Luminescence decay from the  ${}^3F_4$  Tm $^{3+}$  and  ${}^5I_7$  Ho $^{3+}$  states in CaF<sub>2</sub>: (a) singly-doped crystals (4 at% Tm $^{3+}$  and 0.5 at% Ho $^{3+}$ ), (b) a codoped crystal (4 at% Tm $^{3+}$ , 0.4 at% Ho $^{3+}$ ).

A low-temperature (LT, 12 K) absorption spectrum of a 0.5 at.% Ho:CaF<sub>2</sub> crystal at ~2  $\mu m$  is shown in Fig. 4(a). It shows sharp peaks assigned to isolated Ho³+ ions in trigonal (C₃v), tetragonal (C₄v) and cubic (Oh) sites, as well as intense and broader bands owing to Ho ion clusters. For a codoped crystal (5 at.% Tm³+, 0.5 at.% Ho³+), the LT absorption and emission spectra of Ho³+ ions for the  $^5I_8 \leftrightarrow ^5I_7$  transition, Fig. 4(a,b), contain just broad bands which are different from those for single Ho³+ doping indicating the entering of Ho³+ ions into rare-earth (Tm-Ho) ion clusters.



**Fig. 4.** LT (12 K) (a) absorption and (b) luminescence spectra of  ${\rm Ho^{3+}}$  ions in  ${\rm CaF_2}$  ( ${}^5{\rm I}_8 \leftrightarrow {}^5{\rm I}_7$  transition): a 5 at.%  ${\rm Tm^{3+}}$ , 0.5 at.%  ${\rm Ho^{3+}}$  codoped crystal and a 0.5 at.%  ${\rm Ho^{3+}}$  singly-doped one.



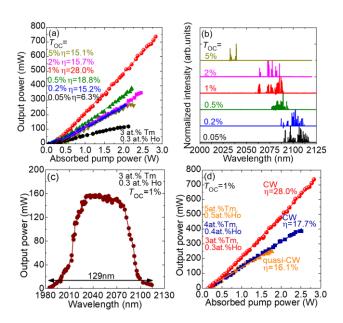
**Fig. 5.** (a) Set-up of the tunable  $Tm,Ho:CaF_2$  laser: L- aspherical focusing lens, PM – pump mirror, OC – output coupler, F – long-pass filter; (b) a typical laser mode profile measured at 15 cm from the OC.

The scheme of the laser is depicted in Fig. 5(a). Cylindric laser elements (Φ8 mm, thickness: 4.1 mm) were cut from the central parts of the as-grown Tm, Ho: CaF2 crystal boules with three doping levels, 5/0.5, 4/0.4 and 3/0.3 at.% Tm<sup>3+</sup>/Ho<sup>3+</sup>. They were polished to laser-quality on both sides and left uncoated. The elements were mounted on a Cu-holder using a silver paste for better heat removal. The crystals were passively cooled to benefit from the good thermal properties of CaF<sub>2</sub>. The hemispherical laser cavity was formed by a flat pump mirror (PM) coated for high transmission (HT. T = 93%) at 0.77  $\mu m$  and high reflection (HR) at 1.9 - 2.3  $\mu m$ , and a set of concave output couplers (OCs) with a radius of curvature (RoC) of-100 mm and a transmission at the laser wavelength  $T_{\rm OC}$  in the range of 0.05% - 5%. The cavity length was  $\sim$  99 mm. The laser crystal was placed near the PM with a separation < 1 mm. The pump source was a CW Ti:Sapphire laser (3900S, Spectra Physics) emitting up to 4.0 W at 768.0 nm with a nearly diffraction limited beam ( $M^2 \approx 1$ ). The pump radiation was focused into the crystal through the PM using an antireflection (AR) coated aspherical lens (focal length, f = 100 mm). The pump spot size in the focus was  $2w_P = 70\pm10 \,\mu\text{m}$ . The pumping was in double pass owing to a high reflectivity of the OCs at the pump wavelength. E.g., for the 3 at.% Tm3+, 0.3 at.% Ho3+ codoped crystal, the total pump absorption under lasing conditions was ~76% being weakly dependent on the output coupling.

Optionally, the pump beam was modulated using a mechanical chopper (duty cycle: 1:4). A long-pass filter (FEL1000, Thorlabs) was placed after the OC to filter out the residual pump. The laser spectra were recorded using an optical spectrum analyzer (AQ6376, Yokogawa) and a  $ZrF_4$  fiber. The laser mode profile was captured using a Pyrocam IIIHR camera (Ophir-Spiricon).

The best laser performance was achieved for the 3 at.%  $\rm Tm^{3+}$ , 0.3 at.%  $\rm Ho^{3+}$ : $\rm CaF_2$  crystal. In the CW regime, the Tm,Ho: $\rm CaF_2$  laser generated a maximum output power of 737 mW at 2062-2088 nm with a slope efficiency  $\eta$  of 28.0% (vs. the absorbed pump power) and a low laser threshold  $P_{\rm th}$  of 133 mW (for  $T_{\rm OC}$  = 1%), as shown in Fig. 6(a). For higher output coupling ( $T_{\rm OC}$  > 1%), the laser performance deteriorated which is attributed to the detrimental effect of ETU at increased populations of the upper laser level. For these OCs, a thermal roll-over was observed for  $P_{\rm abs}$  > 2.2 W. With reducing the output coupling, the laser threshold gradually decreased from  $P_{\rm th}$  = 312 mW (5% OC) to 65 mW (0.05% OC). Power scaling was limited by the available pump power.

The typical spectra of laser emission are shown in Fig. 3(b). The laser operated solely on the  $^5I_7 \rightarrow ^5I_8$  Ho³+ transition and no Tm³+ colasing was observed. The laser wavelength experienced a blue-shift with increasing  $T_{\rm OC}$ , from 2090-2116 nm (0.05% OC) to 2032-2039 nm (5% OC). This behavior is due to the quasi-three-level nature of the Ho³+ laser scheme with reabsorption and it agrees well with the gain spectra of Ho³+ ions in CaF₂, see Fig. 2(d). The laser emission was unpolarized. The Tm,Ho:CaF₂ laser operated on the fundamental transverse mode, Fig. 5(b), with a measured beam quality factor  $M^2$  < 1.1 at the highest output power.



**Fig. 6.** CW Tm,Ho:CaF<sub>2</sub> lasers: (a-b) a 3 at% Tm<sup>3+</sup>, 0.3 at% Ho<sup>3+</sup> codoped crystal: (a) input-output dependences,  $\eta$  – slope efficiency; (b) typical laser emission spectra; (c) tuning curve,  $T_{\text{OC}} = 1\%$ ; (d) effect of the Tm<sup>3+</sup>/Ho<sup>3+</sup> doping level on the laser performance,  $T_{\text{OC}} = 1\%$ .

For the wavelength tunable operation, a Lyot filter was inserted into the laser cavity at the Brewster angle close to the laser crystal. It was made of a 2 mm-thick quartz plate with an optical axis lying in the surface plane. When using  $T_{\rm OC}$  = 1%, the laser wavelength was

continuously tunable from 1985 to 2114 nm (tuning range: 129 nm at the zero-power level).

The laser performance of Tm,Ho:CaF<sub>2</sub> crystals with different doping levels is compared in Fig. 6(d) (for the same  $T_{\rm OC}$  = 1%). With increasing the doping level, despite the increased pump absorption, the maximum output power and the slope efficiency decreased to 386 mW at 2070-2086 nm with  $\eta$  = 17.7% (4 at.% Tm³+, 0.4 at.% Ho³+:CaF<sub>2</sub>) and 249 mW at 2083-2087 nm with  $\eta$  = 16.1% (5 at.% Tm³+, 0.5 at.% Ho³+:CaF<sub>2</sub>). For these crystals, a thermal roll-over in the output dependences was observed for absorbed pump powers above 2.5 W and 1.75 W, respectively (for the latter crystal, the laser was operated in quasi-CW regime). The observed deterioration of the laser performance is ascribed to a drop in the thermal conductivity of CaF<sub>2</sub> crystals upon rare-earth doping increase.

To conclude, we report on the first continuous-wave Tm, Ho:CaF<sub>2</sub> laser delivering an output power up to 737 mW at 2062-2088 nm with a maximum slope efficiency of 28.0% and a laser threshold down to 65 mW, operating on the fundamental transverse mode. A continuous wavelength tuning over a broad range of 129 nm (1985 - 2114 nm) was demonstrated. These results are superior to the previous reports on Tm,Ho:CaF2 ceramic lasers (operating in freerunning / quasi-CW regimes) [24,25], mainly due to the better optical and thermal properties of Tm, Ho: CaF2 single-crystals. In the present work, the crystal quality was improved by carefully controlling the growth atmosphere. The spectroscopic properties of Ho<sup>3+</sup> ions in Tm,Ho:CaF<sub>2</sub> crystals were also studied. They exhibit inhomogeneously broadened emission spectra extending above  $2 \mu m$  (up to at least 2.25  $\mu m$ ) owing to the profound rare-earth ion clustering, an efficient Tm<sup>3+</sup> → Ho<sup>3+</sup> energy-transfer and a relatively long thermal equilibrium decay time. Evidence of existence of two cluster compositions (Tm and Tm/Ho clusters) is presented.

Thanks to its low melting point and good thermal properties,  $Tm,Ho:CaF_2$  appears as an attractive competitor to other recently studied  $Tm^{3+},Ho^{3+}$ -codoped laser crystals with broadband emission properties,  $Tm,Ho:CaGaAlO_4$  [15] and Tm,Ho:CnGG [28]. Further improvement of the laser efficiency for  $Tm,Ho:CaF_2$  is possible by: optimizing the  $Ho^{3+}/Tm^{3+}$  codoping ratio to suppress the Ho-free Tm-clusters which do not contribute significantly to laser emission  $>2~\mu m$ , and implementing a Tm resonant pumping, e.g., by Ramanshifted Erbium fiber lasers [27]. Further power scaling is expected under pumping at  $\sim$ 0.8  $\mu m$  using AlGaAs laser diodes and applying active cooling. In this case, lower total Tm+Ho doping levels are preferable for maintaining better thermal properties of  $Tm,Ho:CaF_2$  crystals, as evidenced by the present work.

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**Disclosures.** The authors declare no conflicts of interest.

**Data availability.** Data underlying the results presented in this paper are not publicly available at this time but may be obtained from the authors upon reasonable request.

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