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Kirill Eremeev, Pavel Loiko, Abdelmjid Benayad, Gurvan Brasse, Patrice Camy, et al.. Efficient continuous-wave Tm,Ho:CaF2 laser at 2.1  $\mu$ m. Optics Letters, 2023, 48 (7), pp.1730-1733. 10.1364/OL.486991 hal-04209401

## HAL Id: hal-04209401 https://hal.science/hal-04209401v1

Submitted on 2 Nov 2023

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# Efficient continuous-wave Tm,Ho:CaF<sub>2</sub> laser at 2.1 μm

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Received XX Month XXXX; revised XX Month, XXXX; accepted XX Month XXXX; posted XX Month XXXX (Doc. ID XXXXX); published XX Month XXXX

We report on the first continuous-wave laser operation of a Tm<sup>3+</sup>,Ho<sup>3+</sup>-codoped calcium fluoride crystal at ~2.1 µm. Tm,Ho:CaF<sub>2</sub> crystals were grown by the Bridgman method and their spectroscopic properties were studied. The stimulated-emission cross-section for the  ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$  Ho<sup>3+</sup> transition is 0.72×10<sup>-20</sup> cm<sup>2</sup> at 2042 nm and the thermal equilibrium decay time is 11.0 ms. A 3 at.% Tm, 0.3 at.% Ho:CaF<sub>2</sub> laser generated 737 mW at 2062-2088 nm with a slope efficiency of 28.0% and a laser threshold of 133 mW. A continuous wavelength tuning between 1985 - 2114 nm (tuning range: 129 nm) was demonstrated. The Tm,Ho:CaF<sub>2</sub> crystals are promising for ultrashort pulse generation at ~2 µm. 2023 Optical Society of America

http://dx.doi.org/10.1364/OL.99.099999

Calcium fluoride (CaF<sub>2</sub>) is a cubic crystal (sp. gr. *Fm*3*m*). Ca<sup>2+</sup> cations are bonded in a body-centered cubic geometry to eight F<sup>-</sup> anions. Due to its broadband transparency (0.13 – 10 µm), low refractive index (*n* = 1.424 at ~2 µm) and high thermal conductivity ( $\kappa$  = 9.7 Wm<sup>-1</sup>K<sup>-1</sup>), CaF<sub>2</sub> is widely used in optical components. It is also very attractive for doping with laser-active trivalent rare-earth ions (RE<sup>3+</sup>) replacing for the Ca<sup>2+</sup> cations [1-3]. Even at moderate doping levels (>0.1 at.% RE<sup>3+</sup>), calcium fluoride crystals exhibit RE<sup>3+</sup> ion clustering [4,5] resulting in a strong inhomogeneous broadening of absorption and emission bands (a "glassy-like" behavior) and enhancement of the energy-transfer processes [6]. The broadband emission and good thermal properties of RE<sup>3+</sup>-doped CaF<sub>2</sub> crystals determine their applications in high-power / high-energy ultrafast lasers and amplifiers at ~1 µm (for Yb<sup>3+</sup> and Nd<sup>3+</sup> doping) [7,8], as well as broadly tunable lasers at ~2 µm (Tm<sup>3+</sup> doping) [9,10].

Holmium ions (Ho<sup>3+</sup>) ions (electronic configuration: [Xe]4f<sup>10</sup>) are known for their emission in the eye-safe spectral range at ~2 µm originating from the <sup>5</sup>I<sub>7</sub>  $\rightarrow$  <sup>5</sup>I<sub>8</sub> transition [11]. Ho lasers emitting slightly above 2 µm are of interest for LIght Detection And Ranging (LIDAR) systems for gas sensing and wind mapping, medical applications, material processing and pumping of mid-infrared laser sources, such as Cr<sup>2+</sup> lasers and optical parametric oscillators based on non-oxide crystals.

There exist two approaches for excitation of Ho3+ ions to the 5I7 metastable state: (i) Tm<sup>3+</sup>,Ho<sup>3+</sup> codoping [11,12] and (ii) resonant (in-band) pumping [13,14]. In the first approach, the Tm<sup>3+</sup> ions (donors, D) efficiently absorb the pump radiation at  $\sim 0.8 \ \mu m$  (the spectral range well addressed by commercial and high power AlGaAs laser diodes) and non-radiatively transfer their energy to Ho<sup>3+</sup> ones (acceptors, A),  ${}^{3}F_{4}(Tm^{3+}) \rightarrow {}^{5}I_{7}(Ho^{3+})$ . The predominantly unidirectional energy transfer (ET)  $D \rightarrow A$  is maintained by an appropriate Ho<sup>3+</sup>/Tm<sup>3+</sup> codoping ratio, between 1:5 to 1:10, while the reduced energy-transfer upconversion (ETU) from the 5I7 Ho3+ metastable state is ensured by choosing relatively low Ho3+ doping levels (<0.5 at.%). Efficient Tm,Ho lasers exist [12,15]. The codoping is also beneficial for combining the gain bandwidths of both ions, which is attractive for broadly tunable and mode-locked lasers at  $\sim 2 \mu m$  [16]. The thermal issues of Tm,Ho lasers can be mitigated by applying the fiber laser geometry [17].

Resonant pumping of Ho<sup>3+</sup> ions directly to the upper laser level presents a costly alternative to the codoping scheme because of the need of expensive and complex pump sources, such as Tm fiber lasers [14] or GaSb-based laser diodes [15]. In-band pumping offers higher laser slope efficiencies [18] and weaker heat loading as compared to the codoping scheme.

The multi-site behavior of low-doped Ho:CaF<sub>2</sub> crystals was studied [19]. In 1965, the first lasing on the  ${}^{5}S_{2} \rightarrow {}^{5}I_{8}$  transition in Ho:CaF<sub>2</sub> was reported at 77 K [20]. Recently, the interest shifted to in-band pumped Ho:CaF<sub>2</sub> lasers operating on the  ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$  transition [21-23]. Jelínek et al. reported on a cryogenic (83 K) continuouswave (CW) 0.5 at.% Ho:CaF<sub>2</sub> laser pumped by a Tm fiber laser at 1940 nm delivering an output power of 2.37 W at 2060 nm with a slope efficiency n of 23% [21]. Duan et al. reported on a continuouswave room-temperature 1 at.% Ho:CaF<sub>2</sub> laser with a similar pump source generating 6.94 W at 2101 nm with a higher  $\eta$  of 57.9% [23]. Sulc *et al.* reported on a free-running 2 at.% Tm, 0.3 at.% Ho:CaF<sub>2</sub> ceramic laser diode-pumped at 785 nm delivering an output pulse energy of 4.1 mJ at 2099 nm with  $\eta$  = 10.8% being tunable from 2016 to 2111 nm [24]. The same group employed a 1700 nm laser diode for Tm resonant pumping and a peak output power of 30 mW at 2074 nm was extracted with still low  $\eta$  of 5.3% [25].

In the present work, we report on the growth, spectroscopy and room-temperature continuous-wave laser operation of  $Tm^{3+}$ ,  $Ho^{3+}$ -

codoped CaF\_2 crystals with the goal of developing gain media with broadband emission properties above 2  $\mu m$ .

Single-crystals of CaF<sub>2</sub> codoped with 3-5 at.% Tm<sup>3+</sup> and 0.3-0.5 at.% Ho<sup>3+</sup> (with respect to Ca<sup>2+</sup>) were grown by the Bridgman-Stockbarger technique using graphite crucibles ( $\Phi 8$  mm, height: 40 mm). The starting reagents were CaF<sub>2</sub>, TmF<sub>3</sub> and HoF<sub>3</sub> powders (purity: 4N), the rare-earth fluorides were achieved by fluorination of the corresponding oxides. A special attention was paid to reduce the oxygen contamination: the growth chamber was sealed to vacuum ( $<10^{-5}$  mbar) and refilled with Ar + CF<sub>4</sub> gases. The starting reagents were mixed and placed into the crucible which was heated slightly above (~40 °C) the melting point (1418 °C) and the solution was homogenized for 3-4 h. The growth was ensured by translating the crucible in a vertical temperature gradient of 30-40 °C/cm. After the growth was completed, the crystals were slowly cooled down to room temperature within 48 h. The segregation coefficients for  $Tm^{3+}$  and  $Ho^{3+}$  ions in CaF<sub>2</sub> are 0.94 and 0.98, respectively. The actual ion densities are  $N_{\text{Tm}}$  = 6.91 and  $N_{\text{Ho}}$  = 0.72 [10<sup>20</sup> at/cm<sup>3</sup>] for the 3 at.% Tm<sup>3+</sup>, 0.3 at.% Ho<sup>3+</sup>:CaF<sub>2</sub> crystal.

The as-grown crystals were transparent with a slight rose coloration due to Ho<sup>3+</sup> doping, Fig. 1(a). A Schlieren photography of a polished cylindrical barrel cut from the central part of the crystal confirmed its high optical quality, Fig. 1(b).



Fig. 1. (a) A photograph and (b) a Schlieren photography of a polished cylindrical barrel cut from the central part of an as-grown 3 at.%  $Tm^{3+}$ , 0.3 at.%  $Ho^{3+}$ :CaF<sub>2</sub> crystal.

The absorption spectra of Tm,Ho:CaF<sub>2</sub> crystals corresponding to the  ${}^{3}H_{6} \rightarrow {}^{3}H_{4}$  Tm<sup>3+</sup> pump transition are shown in Fig. 2(a). The peak absorption cross-section  $\sigma_{abs}$  is  $0.46 \times 10^{-20}$  cm<sup>2</sup> at 768.1 nm and the corresponding bandwidth (full width at half maximum, FWHM) is 12.6 nm, which is beneficial for diode-pumping. Figure 2(b) shows the luminescence spectrum of a codoped crystal around 2 µm. It is smooth and broad spanning from 1.55 to 2.25 µm due to two spectrally overlapping transitions,  ${}^{3}F_{4} \rightarrow {}^{3}H_{6}$  Tm<sup>3+</sup> and  ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$  Ho<sup>3+</sup>, with the Ho<sup>3+</sup> emission peaking at 2025 nm being more intense. As compared to singly Ho<sup>3+</sup>-doped crystals, the Tm<sup>3+</sup>,Ho<sup>3+</sup> codoping induces additional broadening of the Ho<sup>3+</sup> emission band due to more profound ion clustering at higher total RE<sup>3+</sup> doping levels.

The absorption,  $\sigma_{abs}$ , and stimulated-emission cross-section,  $\sigma_{SE}$ , spectra for the  ${}^{5}I_{B} \leftrightarrow {}^{5}I_{7}$  Ho<sup>3+</sup> transition in the 5 at.% Tm<sup>3+</sup>, 0.5 at.% Ho<sup>3+</sup> codoped CaF<sub>2</sub> crystal are shown in Fig. 2(c). The maximum  $\sigma_{SE}$  is 0.72×10<sup>-20</sup> cm<sup>2</sup> at 2042 nm, as calculated using the Füchtbauer–Ladenburg formula. The  ${}^{5}I_{7}$  radiative lifetime,  $\tau_{rad} = 12.6$  ms, was determined using a modified Judd-Ofelt theory [26]. According to the quasi-three-level nature of the Ho<sup>3+</sup> laser scheme, the gain cross-sections for Ho<sup>3+</sup> ions,  $\sigma_{gain} = \beta\sigma_{SE} - (1 - \beta)\sigma_{abs}$ , were calculated for different inversion ratios  $\beta = N_2/N_{Ho}$ , where  $N_2$  is the population of the upper laser level ( ${}^{5}I_7$ ), Fig. 2(d). The gain profiles are smooth and

broad and their maximum experiences a blue-shift with increasing the inversion ratio, from 2130 nm ( $\beta = 0.12$ ) to 2040 nm ( $\beta = 0.30$ ). For an intermediate  $\beta = 0.24$ , the gain bandwidth (FWHM) is 77 nm.



**Fig. 2.** Spectroscopy of Tm,Ho:CaF<sub>2</sub> crystals: (a) absorption spectra, the  ${}^{3}\text{H}_{6} \rightarrow {}^{3}\text{H}_{4}$  Tm<sup>3+</sup> transition; (b) luminescence spectrum around 2 µm, 5 at.% Tm<sup>3+</sup>, 0.5 at.% Ho<sup>3+</sup> codoped crystal, the spectra for singly-doped crystals are given for comparison; (c,d) 5 at.% Tm<sup>3+</sup>, 0.5 at.% Ho<sup>3+</sup>:CaF<sub>2</sub> crystal, the  ${}^{5}\text{I}_{8} \leftrightarrow {}^{5}\text{I}_{7}$  Ho<sup>3+</sup> transition: (c) absorption,  $\sigma_{abs}$ , and stimulated-emission,  $\sigma_{SE}$ , cross-sections, (d) gain cross-sections for Ho<sup>3+</sup> ions,  $\sigma_{gain}$ ,  $\beta = N_2({}^{5}\text{I}_7)/N_{Ho}$  – inversion ratio.

The luminescence decay curves from the  ${}^{3}F_{4}$  Tm<sup>3+</sup> and  ${}^{5}I_{7}$  Ho<sup>3+</sup> states were measured under resonant excitation for singly-doped 4 at% Tm:CaF<sub>2</sub> and 0.5 at% Ho:CaF<sub>2</sub> crystals, Fig. 3(a). The decay curves for both crystals are nearly single exponential yielding luminescence lifetimes  $\tau_{lum}$  of 13.75 ms and 14.07 ms, respectively. The luminescence dynamics were studied for a 4 at% Tm, 0.4 at% Ho:CaF<sub>2</sub> crystal under resonant Tm<sup>3+</sup> excitation, see Fig. 3(b). The decay curves are typical for the Tm<sup>3+</sup>,Ho<sup>3+</sup> coupled systems indicating an efficient and predominantly unidirectional  ${}^{3}F_{4}(Tm^{3+}) \rightarrow {}^{5}I_{7}(Ho^{3+})$  ET. The observation of two different time scales for the rise of Ho<sup>3+</sup> luminescence and the beginning of the Tm<sup>3+</sup> decay indicates the presence of two ion cluster compositions: (i) those containing both dopants and (ii) those being composed of only Tm<sup>3+</sup> ions. The thermal equilibrium decay time for both ions is 11.0 ms.



**Fig. 3.** Luminescence decay from the  ${}^{3}F_{4}$  Tm<sup>3+</sup> and  ${}^{5}I_{7}$  Ho<sup>3+</sup> states in CaF<sub>2</sub>: (a) singly-doped crystals (4 at% Tm<sup>3+</sup> and 0.5 at.% Ho<sup>3+</sup>), (b) a codoped crystal (4 at.% Tm<sup>3+</sup>, 0.4 at.% Ho<sup>3+</sup>).

A low-temperature (LT, 12 K) absorption spectrum of a 0.5 at.% Ho:CaF<sub>2</sub> crystal at ~2 µm is shown in Fig. 4(a). It shows sharp peaks assigned to isolated Ho<sup>3+</sup> ions in trigonal (C<sub>3v</sub>), tetragonal (C<sub>4v</sub>) and cubic (O<sub>h</sub>) sites, as well as intense and broader bands owing to Ho ion clusters. For a codoped crystal (5 at.% Tm<sup>3+</sup>, 0.5 at.% Ho<sup>3+</sup>), the LT absorption and emission spectra of Ho<sup>3+</sup> ions for the <sup>5</sup>I<sub>8</sub>  $\leftrightarrow$  <sup>5</sup>I<sub>7</sub> transition, Fig. 4(a,b), contain just broad bands which are different from those for single Ho<sup>3+</sup> doping indicating the entering of Ho<sup>3+</sup> ions into rare-earth (Tm-Ho) ion clusters.



**Fig. 4.** LT (12 K) (a) absorption and (b) luminescence spectra of Ho<sup>3+</sup> ions in CaF<sub>2</sub> ( ${}^{5}I_{8} \leftrightarrow {}^{5}I_{7}$  transition): a 5 at.% Tm<sup>3+</sup>, 0.5 at.% Ho<sup>3+</sup> codoped crystal and a 0.5 at.% Ho<sup>3+</sup> singly-doped one.



**Fig. 5.** (a) Set-up of the tunable Tm,Ho:CaF<sub>2</sub> laser: L- aspherical focusing lens, PM – pump mirror, OC – output coupler, F – long-pass filter; (b) a typical laser mode profile measured at 15 cm from the OC.

The scheme of the laser is depicted in Fig. 5(a). Cylindric laser elements ( $\Phi$ 8 mm, thickness: 4.1 mm) were cut from the central parts of the as-grown Tm,Ho:CaF<sub>2</sub> crystal boules with three doping levels, 5/0.5, 4/0.4 and 3/0.3 at.% Tm<sup>3+</sup>/Ho<sup>3+</sup>. They were polished to laser-quality on both sides and left uncoated. The elements were mounted on a Cu-holder using a silver paste for better heat removal. The crystals were passively cooled to benefit from the good thermal properties of CaF<sub>2</sub>. The hemispherical laser cavity was formed by a flat pump mirror (PM) coated for high transmission (HT, T = 93%) at 0.77  $\mu$ m and high reflection (HR) at 1.9 – 2.3  $\mu$ m, and a set of concave output couplers (OCs) with a radius of curvature (RoC) of -100 mm and a transmission at the laser wavelength  $T_{\rm OC}$  in the range of 0.05% - 5%. The cavity length was ~99 mm. The laser crystal was placed near the PM with a separation <1 mm. The pump source was a CW Ti:Sapphire laser (3900S, Spectra Physics) emitting up to 4.0 W at 768.0 nm with a nearly diffraction limited beam ( $M^2 \approx 1$ ). The pump radiation was focused into the crystal through the PM using an antireflection (AR) coated aspherical lens (focal length, f = 100 mm). The pump spot size in the focus was  $2w_P = 70 \pm 10 \mu m$ . The pumping was in double pass owing to a high reflectivity of the OCs at the pump wavelength. E.g., for the 3 at.% Tm<sup>3+</sup>, 0.3 at.% Ho<sup>3+</sup> codoped crystal, the total pump absorption under lasing conditions was ~76% being weakly dependent on the output coupling.

Optionally, the pump beam was modulated using a mechanical chopper (duty cycle: 1:4). A long-pass filter (FEL1000, Thorlabs) was placed after the OC to filter out the residual pump. The laser spectra were recorded using an optical spectrum analyzer (AQ6376, Yokogawa) and a  $ZrF_4$  fiber. The laser mode profile was captured using a Pyrocam IIIHR camera (Ophir-Spiricon).

The best laser performance was achieved for the 3 at.% Tm<sup>3+</sup>, 0.3 at.% Ho<sup>3+</sup>:CaF<sub>2</sub> crystal. In the CW regime, the Tm,Ho:CaF<sub>2</sub> laser generated a maximum output power of 737 mW at 2062-2088 nm with a slope efficiency  $\eta$  of 28.0% (vs. the absorbed pump power) and a low laser threshold  $P_{\rm th}$  of 133 mW (for  $T_{\rm OC}$  = 1%), as shown in Fig. 6(a). For higher output coupling ( $T_{\rm OC}$  > 1%), the laser performance deteriorated which is attributed to the detrimental effect of ETU at increased populations of the upper laser level. For these OCs, a thermal roll-over was observed for  $P_{\rm abs}$  > 2.2 W. With reducing the output coupling, the laser threshold gradually decreased from  $P_{\rm th}$  = 312 mW (5% OC) to 65 mW (0.05% OC). Power scaling was limited by the available pump power.

The typical spectra of laser emission are shown in Fig. 3(b). The laser operated solely on the  ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$  Ho<sup>3+</sup> transition and no Tm<sup>3+</sup> colasing was observed. The laser wavelength experienced a blue-shift with increasing  $T_{OC}$ , from 2090-2116 nm (0.05% OC) to 2032-2039 nm (5% OC). This behavior is due to the quasi-three-level nature of the Ho<sup>3+</sup> laser scheme with reabsorption and it agrees well with the gain spectra of Ho<sup>3+</sup> ions in CaF<sub>2</sub>, see Fig. 2(d). The laser emission was unpolarized. The Tm,Ho:CaF<sub>2</sub> laser operated on the fundamental transverse mode, Fig. 5(b), with a measured beam quality factor M<sup>2</sup> < 1.1 at the highest output power.



**Fig. 6.** CW Tm,Ho:CaF<sub>2</sub> lasers: (a-b) a 3 at% Tm<sup>3+</sup>, 0.3 at% Ho<sup>3+</sup> codoped crystal: (a) input-output dependences,  $\eta$  – slope efficiency; (b) typical laser emission spectra; (c) tuning curve,  $T_{0C} = 1\%$ ; (d) effect of the Tm<sup>3+</sup>/Ho<sup>3+</sup> doping level on the laser performance,  $T_{0C} = 1\%$ .

For the wavelength tunable operation, a Lyot filter was inserted into the laser cavity at the Brewster angle close to the laser crystal. It was made of a 2 mm-thick quartz plate with an optical axis lying in the surface plane. When using  $T_{OC} = 1\%$ , the laser wavelength was continuously tunable from 1985 to 2114 nm (tuning range: 129 nm at the zero-power level).

The laser performance of Tm,Ho:CaF<sub>2</sub> crystals with different doping levels is compared in Fig. 6(d) (for the same  $T_{OC} = 1\%$ ). With increasing the doping level, despite the increased pump absorption, the maximum output power and the slope efficiency decreased to 386 mW at 2070-2086 nm with  $\eta = 17.7\%$  (4 at.% Tm<sup>3+</sup>, 0.4 at.% Ho<sup>3+</sup>:CaF<sub>2</sub>) and 249 mW at 2083-2087 nm with  $\eta = 16.1\%$  (5 at.% Tm<sup>3+</sup>, 0.5 at.% Ho<sup>3+</sup>:CaF<sub>2</sub>). For these crystals, a thermal roll-over in the output dependences was observed for absorbed pump powers above 2.5 W and 1.75 W, respectively (for the latter crystal, the laser was operated in quasi-CW regime). The observed deterioration of the laser performance is ascribed to a drop in the thermal conductivity of CaF<sub>2</sub> crystals upon rare-earth doping increase.

To conclude, we report on the first continuous-wave Tm,Ho:CaF<sub>2</sub> laser delivering an output power up to 737 mW at 2062-2088 nm with a maximum slope efficiency of 28.0% and a laser threshold down to 65 mW, operating on the fundamental transverse mode. A continuous wavelength tuning over a broad range of 129 nm (1985 - 2114 nm) was demonstrated. These results are superior to the previous reports on Tm,Ho:CaF2 ceramic lasers (operating in freerunning / quasi-CW regimes) [24,25], mainly due to the better optical and thermal properties of Tm,Ho:CaF<sub>2</sub> single-crystals. In the present work, the crystal quality was improved by carefully controlling the growth atmosphere. The spectroscopic properties of Ho<sup>3+</sup> ions in Tm,Ho:CaF<sub>2</sub> crystals were also studied. They exhibit inhomogeneously broadened emission spectra extending above  $2 \,\mu m$  (up to at least 2.25  $\mu m$ ) owing to the profound rare-earth ion clustering, an efficient  $Tm^{3+} \rightarrow Ho^{3+}$  energy-transfer and a relatively long thermal equilibrium decay time. Evidence of existence of two cluster compositions (Tm and Tm/Ho clusters) is presented.

Thanks to its low melting point and good thermal properties, Tm,Ho:CaF<sub>2</sub> appears as an attractive competitor to other recently studied Tm<sup>3+</sup>,Ho<sup>3+</sup>-codoped laser crystals with broadband emission properties, Tm,Ho:CaGaAlO<sub>4</sub> [15] and Tm,Ho:CNGG [28]. Further improvement of the laser efficiency for Tm,Ho:CaF<sub>2</sub> is possible by: optimizing the Ho<sup>3+</sup>/Tm<sup>3+</sup> codoping ratio to suppress the Ho-free Tm-clusters which do not contribute significantly to laser emission >2 µm, and implementing a Tm resonant pumping, e.g., by Ramanshifted Erbium fiber lasers [27]. Further power scaling is expected under pumping at ~0.8 µm using AlGaAs laser diodes and applying active cooling. In this case, lower total Tm+Ho doping levels are preferable for maintaining better thermal properties of Tm,Ho:CaF<sub>2</sub> crystals, as evidenced by the present work.

**Funding.** Agence Nationale de la Recherche (ANR-19-CE08-0028); Région Normandie (Chaire d'excellence "RELANCE").

Disclosures. The authors declare no conflicts of interest.

**Data availability.** Data underlying the results presented in this paper are not publicly available at this time but may be obtained from the authors upon reasonable request.

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