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#### ARTICLE

# Carcinogenic dioxane detection using pristine and metal-doped 2D VSe<sub>2</sub>: Insights from density functional theory simulations **1**

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#### ABSTRACT

Dioxane (diethylene oxide) is a synthetic organic compound classified as heterocyclic ether, a potentially carcinogenic water toxicant. Prolonged exposure can cause eye irritation, carcinogenic liver reactions, and other severe issues. Therefore, efficient dioxane detectors are needed to be designed and developed. Inspired by the recent developments of 2D materials in biosensing, in this work, the dioxane detection potentiality of pristine and metal-doped (Al, Au, and Ag) 2D VSe<sub>2</sub> have been systematically examined using Density Functional Theory (DFT) based simulations. Among all the metallic dopants considered, Al binds energetically on the VSe<sub>2</sub> surface with an energy of -2.158 eV. The adsorption of dioxane was studied by considering the electronic properties, adsorption energy, orbital interactions, and charge transfer. Our DFT calculation suggests that dioxane adsorption in Al-doped VSe<sub>2</sub> is more promising than the pristine and other metal-doped VSe<sub>2</sub> systems, due to the reasonable adsorption energy of -0.80 eV, charge transfer of -0.567e, and strong orbital interaction between Al 3*p* and O 2*p* orbitals. Additionally, the room temperature structural solidity of the sensor has also been verified using the *ab initio* molecular dynamics simulations. The reported theoretical results inspire the fabrication and engineering of efficient dioxane sensors using Al-doped VSe<sub>2</sub>.

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#### I. INTRODUCTION

Rapid industrialization and population growth resulted in the discharge of large quantities of organic pollutants, undeniably contributing to water resource pollution.<sup>1–3</sup> Diethylene oxide (widely known as dioxane or 1,4-dioxane), classified as ether, with the chemical formula  $C_4H_8O_2$ , is a colorless liquid with a soft sweet smell like dimethyl ether. Dioxane is used on a large scale to produce paper, textiles, electronics, cosmetics, etc. They are often regarded as a powerful solvent and stabilizer for chlorinated solvents in laboratory applications. It is well known that dioxane is an environmental water toxicant with high water solubility, hydrophilicity, long-term

stability, and non-bio-degradability. Their excess presence can contaminate the ground, surface, and drinking water and are highly dangerous to living beings. Long term exposure can elicit severe eye irritation and carcinogenic reactions in the liver in various species. As per the studies, nearly one-fifth of the public drinking water sources in the United States (US) have measurable quantities of dioxane.<sup>4</sup> The International Agency for Research on Cancer (IARC) and the US Environmental Protection Agency (USEPA) have categorized dioxane as a potential human carcinogen.<sup>5</sup> Moreover, the Agency for Toxic Substances and Disease Registry (ATSDR) designated dioxane as a superfund precedence category due to the chemical harmful effects and proclivity to pollute drinking water.<sup>4</sup> Because



of these implications, it is urgent to develop sensors that selectively detect and remove the dioxane molecules from the environment.

In recent years, many sophisticated physicochemical and biological approaches have been used to successfully remove dioxane from affected environments. The unique chemical features of dioxane, such as its low Henry's law constant, high water solubility, and coexistence with the chlorinated solvents and other contaminants, make efficient removal difficult.<sup>6</sup> Complex water treatment procedures can be utilized to eradicate trace contaminants of dioxane.<sup>5</sup> Some of these include the photoinduced oxidation method (UV/H<sub>2</sub>O<sub>2</sub>) to substantially eliminate the structure of organic chemical; and, the Fenton's reagent (Fe and H<sub>2</sub>O<sub>2</sub>), advanced oxidation processes using systems, such as ozone  $(O_3)/H_2O_2$ , and H<sub>2</sub>O<sub>2</sub> - TiO<sub>2</sub> combined vacuum UV. The sonolytic decomposition chemistry of dioxane in an aqueous solution has also been investigated,<sup>7</sup> where dioxane is dissolved into smaller short-chain organic by-products using ultrasound, which can subsequently be destroyed more successfully by other methods like biological processes. Although several other advanced dioxane removal technologies, such as solid-phase microextraction (SPME), distillation, gas chromatography (GC), thermal methods, air stripping and membrane filtration techniques, etc., are incompetent due to their high rate of cost and energy consumption.<sup>8-11</sup> Recent studies including electrochemical dioxane-detection by using nickel oxide doped on neodymium oxide nanocomposites (NiO@Nd2O3 NCs) embedded on glassy carbon electrode<sup>12</sup> and reduced graphene oxide-curcumin nanocomposite synthesized by a simple solution approach<sup>13</sup> promotes the need of developing this water pollutant.

However, properties, such as binding energy, adsorption geometry, adsorption configuration, charge transfer, etc., for detecting biomolecules cannot be directly determined by the timeconsuming real experiment methods. Computational techniques are the best way to perform quantum mechanical calculations to analyze materials' various properties and phenomena and model them.

In this study, we have adopted the widely applicable quantum mechanical theoretical simulation approach called Density Functional Theory (DFT)<sup>14</sup> to analyze the molecular interactions. In these first-principle based calculations, the energy is represented as functionals of density. Many theoretical investigations have recently been conducted to search for novel materials for sensing applications. Interestingly, some cutting-edge theoretical research on SiCNT was conducted utilizing DFT simulations to examine the adsorption and removal of dioxane.<sup>15</sup> Furthermore, the DFT-D3/revPBE calculation is verified using the MP2 theoretical method. The study showed that dioxane has significant adsorption toward this nanotube, but the higher recovery time limits the practical usage. However, most traditional sensing materials have several flaws due to their low sensitivity and selectivity. The interaction of biomolecules with 2D material is mainly via physisorption because of weak van der Waals forces. By modifying the functional qualities of 2D materials, such as defects engineering hybrid structure formation, the adsorption performances of these materials may be enhanced. Usually, metallic doping methods are adopted in the studies of 2D materials. Its unique and robust coupling of electromagnetic properties gives a charge transferring nature. They form covalent bonds with neighboring atoms via charge transfer, improving the conductivity of the pristine 2D material and hence the sensing properties.

Transition-metal dichalcogenides (TMDCs) are emerging 2D materials with a stoichiometric formula of MX2. M denotes a transition metal element, such as Sc, V, Mo, Ti, or W, while X stands for a chalcogen element, such as Te, Se, or S. Here, a single layer of M atoms is stacked between the bi-layer of X atoms in this tri-layered structure, creating the layered structure X-M-X. Long-range van der Waals (VdW) forces dominate between these layers, resulting in the weak interaction. It exhibits unique photoelectronic, thermal, chemical, and mechanical properties,<sup>16,17</sup> making it promising for gas or biosensing applications.<sup>18-20</sup> The uniqueness of TMDCs is due to the presence of the *d* orbitals of the heavy metals, which causes a robust spin-orbit interaction, leads to valley polarization. The TMDCs have out-of-plane and broken in-plane mirror symmetry. The polymorphic nature of TMDC structures is a fascinating feature. TMDCs can be mainly classified into two structural phases: 2H and 1T.<sup>21,22</sup> The 2H is semiconducting, and the coordination of metal atoms is trigonal prismatic, while the 1T structure is metallic with octahedral coordination. In view of the above, it is reasonable to conclude that TMDCs offer enormous untapped potential as a future choice for numerous biomolecule sensing applications with their wide range of characteristics.

2D vanadium dichalcogenides (VX2), such as VSe2, VTe2, and VS<sub>2</sub>, have recently attracted much interest<sup>23-26</sup> due to their exceptional optical, electrical, and chemical characteristics. Our research focused on the 1T VSe<sub>2</sub> monolayer, a paramagnetic material where the vanadium atoms are surrounded by a trigonally deformed cage of selenium atoms in an octahedral environment.<sup>27</sup> Zhang et al.<sup>28</sup> used a chemical vapor deposition (CVD) approach to establish a van der Waals epitaxial strategy for synthesizing stable 1T VSe<sub>2</sub> nanosheets on mica. Surprisingly, these 2D structures have exceptional metallic characteristics, including the electrical conductivity up to 10<sup>6</sup> S  $m^{-1}$ , higher than the other conductive 2D materials. Synthesizing VSe2 monolayer also have been reported by effective methods, such as liquid exfoliation,<sup>29</sup> from [(SnSe)<sub>1.15</sub>]<sub>m</sub>(VSe<sub>2</sub>)<sub>n</sub>] ferecrystals via physical vapor deposition,<sup>30</sup> low-pressure chemical vapor deposition (LPCVD),<sup>31</sup> etc. In very recent work, the potential of 1T VSe<sub>2</sub> for nitrobenzene detection was explored.<sup>26</sup> The DFT study reveals that the transition metal doped VSe2 is more sensitive to nitrobenzene molecules.

Although some significant observational facts exist for dioxane sensing studies, evidence of investigating TMDCs as a sensing material, particularly monolayered VSe<sub>2</sub>, as a potential dioxane sensing material is unknown to the authors' available knowledge. In this current work, the adsorption and electronic properties of VSe<sub>2</sub> as a biosensor for dioxane have been examined using first principle-based DFT calculations. To overcome the weak interaction of VSe2 monolayer toward dioxane, VSe2 has been modified by doping with various metals (Al, Ag, Au, W, and Ti). Total Density of States (TDOS) and Partial Density of States (PDOS) analyses have been done to study the orbital interactions of states near the Fermi level and the charge transfer analysis via Lowdin Charge Analysis (LCA). Finally, because of its high adsorption energy toward dioxane, we suggest the Al-doped VSe<sub>2</sub> structure is among the most potential dioxane sensing materials. Furthermore, the structural durability of VSe<sub>2</sub> + Al was examined using ab initio MD simulations at 300 K, which revealed that the combination is extremely stable.

#### **II. COMPUTATIONAL DETAILS**

This study calculated all the structural, electronic, interfacial, and sensing properties of 2D VSe2 and metal doped structures using DFT with the Quantum Espresso package.<sup>32</sup> As an exchange-correlation energy functional approximation, Perdew-Burke-Ernzerhof (PBE) in the Generalized Gradient Approximation (GGA) function (PBE-GGA) were used.<sup>33</sup> As per the literature, GGA is appropriate for accurately predicting the electronic, magnetic, and optical properties of 2D TMDCs. To accommodate the long-range van der Waal's interactions, Grimme DFT-D2 were also introduced in the calculations. For the current work, we have modeled a  $2 \times 2$  supercell of 2D 1T VSe<sub>2</sub> monolayer containing 12 atoms. To eliminate the interactions between the periodic images, a 20 Å vacuum spacing is introduced in the vertical direction of the VSe2 monolayer, i.e., in the z-direction. The planewave-cutoff energy was set to 35 Ry. The geometrical structures were relaxed until each atom's total computed energy and force were less than  $10^{-6}$  and  $10^{-4}$  eV, respectively. The Brillion zone integration was achieved by  $4 \times 4 \times 1$  Monkhrost-Pack. For the most promising

system, molecular dynamic simulations at 300 K were also done to remark on the sensor material's integrity at ambient temperatures.

#### **III. RESULTS AND DISCUSSION**

## A. Electronic and structural properties of $\mathsf{VSe}_2$ monolayer

In this study, we have considered the 1T phase of pristine VSe<sub>2</sub>, in which a layer of metal Vanadium is placed between two selenium sheets. The optimized 2 × 2 supercell structure of VSe<sub>2</sub> is displayed in Fig. 1(a). The obtained lattice constants are a = b = 3.334 Å, correlating neatly with published primitive cell experimental value, a = b = 3.356 Å<sup>34</sup> and a = b = 3.36 Å,<sup>35</sup> and indeed corresponds to the previously determined DFT calculated value of a = b = 3.328Å.<sup>23</sup> The optimized V–Se bond length is 2.487 Å with the computed interlayer distance of 1.84 Å, matching the earlier available interlayer result of 1.56 Å.<sup>35</sup> The TDOS of VSe<sub>2</sub> have been calculated using the GGA + U approach (U value  $\approx 3.5$  eV for V<sup>36</sup>) and are displayed in Fig. 2. The continuous and asymmetric spin states at



24 May 2024 13:40:50

FIG. 1. The geometrically optimized structure of 1T VSe<sub>2</sub> monolayer in (a) armchair and (b) zigzag, (c) the relaxed structure of dioxane molecule, and (d) TDOS plot of 1T VSe<sub>2</sub>.



FIG. 2. Optimized structures of (a) dioxane adsorbed pristine  $VSe_2$  and metal-doped  $VSe_2$  after geometrical relaxation, (b) AI, (c) Au, and (d) Ag.

the Fermi level indicate that 1T VSe<sub>2</sub> is magnetic and metallic.<sup>35</sup> The reasonable match of pristine VSe<sub>2</sub> bond length and other electronic parameters with previously reported results gives us confidence in the simulation procedures and methods' accuracy. 1T MX<sub>2</sub> structure shows two major directions, armchair and zigzag. For the monolayer 1T VSe<sub>2</sub>, the zigzag direction shows significantly stronger and more stretchy than armchair<sup>37</sup> [see Fig. 1(b)].

#### B. Adsorption studies of dioxane on pristine VSe<sub>2</sub>

Here, we have placed the dioxane molecule ~2 Å above the surface of the pristine VSe<sub>2</sub> at four different adsorption sites: hcp, fcc, top, and hollow sites.<sup>37</sup> Several combinations of the initial structures were explored; for example, the O atom of the dioxane molecule is close to the different sites of VSe<sub>2</sub> and is geometrically relaxed to minimum energy configuration. The most stable geometric

TABLE I. DFT calculated adsorption energy and bond length of metal-doped VSe<sub>2</sub> systems.

System	Adsorption energy (eV)	Bond length (Å)
$VSe_2 + Ti$	+3.623	2.340 (Se-Ti)
$VSe_2 + W$	+7.388	2.285 (Se-W)
$VSe_2 + Al$	-2.158	2.495 (Se-Al)
$VSe_2 + Au$	-1.412	2.437 (Se-Au)
$VSe_2 + Ag$	-1.124	2.534 (Se-Ag)

structure of VSe<sub>2</sub> + dioxane/dopant is attained by the adsorption energy ( $E_{ad}$ ) calculation using the following expression:

$$E_{ad}(D) = E(VSe_2 + D) - E(VSe_2) - E(D),$$
(1)

where  $E(VSe_2)$ , E(D), and  $E(VSe_2 + D)$  are, respectively, the energies of pristine VSe<sub>2</sub>, dioxane molecule or dopant, and VSe<sub>2</sub> + dioxane/ dopant system. Figure 2(a) depicts the DFT optimized structure of dioxane adsorbed VSe<sub>2</sub>. The top site of Se with adsorption energy of -0.15 eV at a bond distance of 3.71 Å is found to be the most favorable position for the dioxane adsorption. The adsorption energy is negative, indicating that the adsorption is spontaneous and energetically feasible. The Density of States (DOS) of the VSe<sub>2</sub> gets altered after the dioxane adsorption, and it is clearly visible in Fig. 3(b). The adsorption configuration of dioxane molecules on the VSe2 monolayer results in the transfer of charges between them. The LCA is performed to quantify the charge transfer between the dioxane and the pristine VSe<sub>2</sub> monolayer. As per the LCA, only a minute charge of 0.041e transfers from the VSe<sub>2</sub> to the dioxane molecule. The negligible charge transfer, weak adsorption energy, and large interaction distance suggest that the dioxane adsorption on pristine VSe2 monolayer is physisorption, which would be due to the long-range van der Waal's forces. The practical implementation of the sensor requires



FIG. 3. PDOS plots of (a) pristine and (b) dioxane adsorbed VSe<sub>2</sub>.



the dioxane molecules to be adsorbed strongly on the surface. In this respect, a metallic doping strategy was adopted.

### C. Modification of the VSe<sub>2</sub> monolayer by metallic doping

We have placed metals, such as Al, Ag, Au, W, and Ti ~2 Å, above the VSe<sub>2</sub> surface at different adsorption sites and allowed the systems to relax. The adsorption energies and bond lengths for the best configuration of VSe<sub>2</sub> + metal are calculated using Eq. (1) and are shown in Table I. From adsorption calculations at different sites, it was found that the top site above the Se atom was the best site for doped metals Al, Au, and Ag. The W and Ti showed positive adsorption energies, implying that they do not bind on the VSe<sub>2</sub> surface. The corresponding relaxed structures of the metaldoped VSe<sub>2</sub> monolayer are illustrated in Figs. 2(b)–2(d). As per the calculated adsorption energy values, Al binds strongly on the VSe<sub>2</sub> monolayer with an energy of -2.158 eV. At the same time, the Au and Ag have comparatively low adsorption energy of -1.412 and -1.124 eV, respectively. Hence, the adsorption energy analysis confirms Al as the best dopant. Figure 4(a) denotes the TDOS of pristine and metal-doped VSe<sub>2</sub> systems. The improvement of energy states close to the Fermi level indicates the increase in conductivity of metal-doped VSe<sub>2</sub>. The asymmetric spin states in the TDOS of VSe<sub>2</sub> + metal reminds the persistence of the magnetic nature. Additionally, Fig. 4(b) plots the TDOS plots of the pristine VSe<sub>2</sub> merged with the Al-doped VSe<sub>2</sub>. Unlike the original VSe<sub>2</sub> monolayer, the DOS of the Al-doped system differs in several energy levels, with some significant increase in the Fermi level, which may be due to the electron exchange between the VSe<sub>2</sub> monolayer to the Al atom.

The PDOS analysis provides a clear idea of the orbital interactions and the direction of transfer of charges between the dioxane molecule and the substrate VSe<sub>2</sub>. Figure 5 compares PDOS of the valance orbitals of VSe<sub>2</sub> (V 3*d* orbital and Se 4*p* orbital), 3*p* orbitals of isolated Al atom and Al atom in the VSe<sub>2</sub> + Al system. When the Al atom binds to the VSe<sub>2</sub> surface, a reduction in the energy states of the Al 3*p* orbital at the Fermi level occurs compared to that in the isolated Al atom. Simultaneously, the V 3*d* and Se 4*p* near the Fermi



24 May 2024 13:40:50

FIG. 5. The PDOS plots compare the valence orbitals of VSe<sub>2</sub> in the pristine and AI + VSe<sub>2</sub> system in the left panel, whereas in the right panel compare the 3*p* orbitals of isolated AI atom and the AI + VSe<sub>2</sub> system.

level get magnified, and the overall states are redistributed. This reduction in the Al and enhancement in VSe<sub>2</sub> energy states signifies a charge transfer from Al 3*p* orbital to 4*p* and 3*d* orbitals of Se and V in VSe<sub>2</sub> and is a result of the existence of an unpaired valence electron in the outermost *p* orbital. Hence, the Al atom tends to donate the electron and become an anion.

A similar redistribution of states can be observed from the PDOS plots of Au- and Ag-doped  $VSe_2$  systems (supplementary material, Fig. S1). Since Au and Ag have an unpaired electron in

their valence *s* orbitals, they can either act as an electron donator or an acceptor. On interacting with the VSe<sub>2</sub> system, the energy states of both the metals got depleted around the Fermi level (right panel of Fig. S1). This indicates the charge donating behavior of the Au and Ag metals. At the same time, the increased energy states near the Fermi level of the valence orbitals of the V and Se in VSe<sub>2</sub> denote the charge accumulation (left panel of Fig. S1), i.e., the VSe<sub>2</sub> gains charge from Au and Ag's 6*s* and 5*s* orbitals, respectively. According to the LCA, the Al atom loses a charge of 0.299*e* to the VSe<sub>2</sub> system.



In contrast, Au and Ag lose about 0.173e and 0.145e, respectively, to VSe<sub>2</sub> systems. The results obtained from LCA are pretty consistent with the PDOS.

Figures 6(a) and 6(b), respectively, shows the HOMO (highest occupied molecular orbital) and LUMO (lowest unoccupied molecular orbital) distributions of the Al-doped VSe<sub>2</sub> system. It is noted that a significant portion of HOMO and LUMO, basically the electron receiving/donating part, is distributed near the Se atom to which the Al is bonded. This signifies that the addition of Al atom enriched the active region on the VSe<sub>2</sub> surface and hence the sensitivity of the material. All the DFT calculations were performed at 0 K; therefore, it is highly essential to see the structural stability at ambient temperatures. Desorption of the molecules from the sensor materials is usually achieved by heating the material to higher temperatures. This also requires the consistency of the system at a higher temperature. To establish the same, we performed the ab initio Molecular Dynamics (AIMD) simulations at 300 K for the best system, i.e.,  $VSe_2 + Al$  system. Figure 6(c) depicts the MD snapshot of the VSe<sub>2</sub> + Al system for 3 ps. Although the structural plot of VSe<sub>2</sub> + Al shows minor bond length variations due to the TABLE II. The adsorption energy and bond length of dioxane adsorbed pristine and metal-doped  $\mathsf{VSe}_2$  systems.

System	Adsorption energy (eV)	Bond length (Å)
Pristine VSe <sub>2</sub> + dioxane	-0.15	3.71 (Se-O)
$VSe_2 + Al + dioxane$	-0.80	3.38 (Al-O)
$VSe_2 + Au + dioxane$	-0.30	2.26 (Au-O)
$VSe_2 + Ag + dioxane$	-0.20	2.62 (Ag-O)

thermal agitations, the system remains stable without any distortion at 300 K. Therefore, the Al-doped  $VSe_2$  system is quite feasible for practical sensor applications.

#### D. Adsorption of dioxane on metal-doped VSe<sub>2</sub>

To observe the dioxane adsorption on the metal-doped  $\rm VSe_2$  system, we have placed the dioxane molecule above the structurally



FIG. 8. TDOS plots of dioxane interaction on (a) pristine VSe<sub>2</sub>, (b) VSe<sub>2</sub> + AI, (c) VSe<sub>2</sub> + Au, and (d) VSe<sub>2</sub> + Ag systems.

relaxed metal-doped VSe<sub>2</sub> system such that the distance between the metal and the molecule is ~2 Å. The resulting structural combinations are allowed to optimize geometrically. The corresponding structures in their minimum energy configurations are shown in Fig. 7.

The adsorption energy and bond length of dioxane on different VSe<sub>2</sub> systems are tabulated (Table II). The dioxane has a greater binding on the Al-doped system with an adsorption energy of -0.8 eV. This high energy value indicates that dioxane molecules chemisorb on the Al-doped system. While dioxane adsorption on the Ag- and Au-doped VSe<sub>2</sub> systems are mostly physisorption, the adsorption energies are -0.30 and -0.20 eV, respectively. It is also noted that the dioxane adsorption strength increases by the strategy of metal doping, which in turn signifies the improvement in the sensing activity of the pristine VSe<sub>2</sub> system. The shortest distance between the O atom of the dioxane molecule and the Al atom in the VSe<sub>2</sub> + Al system is 3.38 Å. While the O–Au and O–Ag distances are 2.26 and 2.62 Å, respectively. The adsorption energy calculations conclude that the Al-doped VSe<sub>2</sub> system is well suited as the sensor material for detecting dioxane molecules from the atmosphere.

### E. Mechanism of dioxane adsorption in metal-doped $\mathsf{VSe}_2$ system

The DOS analysis explains how the electronic structure of the metal-doped VSe<sub>2</sub> system varies with the dioxane adsorption. Figure 8 compares the TDOS of the pristine and metal-doped VSe<sub>2</sub> system after the dioxane adsorption. The main reason for the origin of new energy states of the pristine and metal-doped VSe<sub>2</sub> system near the Fermi level is mainly because of the presence of dioxane molecules on its surface.

We have also analyzed the PDOS to clarify the orbital level interactions and charge transfer in the metal-doped VSe<sub>2</sub> systems. The PDOS of the Al 3p orbitals of VSe<sub>2</sub> + Al and VSe<sub>2</sub> + Al + dioxane systems are shown in the left top and bottom panel, respectively, in Fig. 9. At the same time, the right upper panel of the same figure represents the PDOS for the O 2p orbitals of the isolated dioxane, and the lower panel compares the PDOS for the O 2p orbitals of the VSe<sub>2</sub> + Al + dioxane system. It is worth noting that, after the interaction of the dioxane molecule with the VSe<sub>2</sub> + Al system, the energy states around the Fermi level of the Al 3p orbital diminished. At the same time, new states originated around the Fermi level of





TABLE III. LCA information on charge transfer of  $\mathsf{VSe}_2$  + metal systems to the dioxane molecule.

System	Charge transfer	
$VSe_2 + Al + dioxane$	–0.567e (Al 3p orbital – O 2p orbital)	
$VSe_2 + Au + dioxane$	–0.101e (Au 6s orbital – O 2p orbital)	
$VSe_2 + Ag + dioxane$	–0.132e (Ag 5s orbital – O 2p orbital)	

the O 2*p* orbital. These states were absent in the same energy range of the PDOS of isolated dioxane. This formation of new states in the O 2*p* orbital and decrement in the intensity of available energy states in the Al 3*p* orbital implies that the electronic charges are transferred from the O 2*p* to Al 3*p* orbital. In addition, the strong dioxane adsorption is because of the p-p orbital overlap. The quantitative charge transfer study by LCA also justifies this charge transfer, i.e., ~0.567*e* charges are lost by the 3*p* orbitals of Al in the VSe<sub>2</sub> + Al system to the O 2*p* orbital of the dioxane molecule.

A comparable reallocation of the energy states is observed in the PDOS of VSe<sub>2</sub> + Au and VSe<sub>2</sub> + Ag systems after the dioxane adsorption (Fig. S2). Similar to the Al 3*p* orbitals, the energy states of Ag 5*s* and Au 6*s* orbitals near the Fermi level decrease. Meanwhile, there is an enhancement in the energy states in the vicinity of the Fermi level of O 2*p* orbital of dioxane in the adsorbed system. The charge transfer from Ag and Au's *s* orbitals to the O 2*p* orbitals causes this reduction and enhancement in states. According to LCA, Au 6*s* lose 0.101*e*, and Ag 5*s* lose 0.132*e* charges to the O 2*p* orbital of dioxane, shown in Table III.

A large amount of charge transfer between the Al-doped VSe<sub>2</sub> system and dioxane molecule is due to the strong chemical adsorption. The minor charge transfer from the Ag and Au to the dioxane molecule is due to the physisorption. Since adsorption energy and charge transfer are the important parameters in sensing, the Al-doped VSe<sub>2</sub> system will be beneficial for the fabrication of electrochemical dioxane sensors.

#### **IV. CONCLUSION**

In conclusion, we used the first principle density functional theory technique to explore the dioxane sensing abilities of pristine and metal-doped (Al, Au, and Ag) 2D VSe<sub>2</sub> systems. We have investigated various adsorption sites of the dioxane molecules on the pristine VSe<sub>2</sub> surface. Our adsorption energy calculation suggests that the dioxane interaction is more with the metal-doped VSe<sub>2</sub> system, particularly the Al-doped VSe<sub>2</sub>, than its pristine counterpart. The Al atom binds strongly on the VSe2 surface than the other metals with higher energy and charge transfer. The metal doping improved the conductivity and hence the adsorption ability of the VSe2 system. The strong adsorption of the dioxane molecules with the VSe<sub>2</sub> + Al system is mainly due to the extensive charge transport and orbital interaction between the O 2p of dioxane and the 3p orbitals of Al. The MD simulation at 300 K verified the room temperature stability of the VSe<sub>2</sub> + Al system; therefore, the Al-doped VSe<sub>2</sub> system could be a more promising dioxane sensor. This work aims to provide a theoretical foundation and an interpretation of the adsorption behaviors of VSe2-based biosensors to facilitate future empirical creations.

#### SUPPLEMENTARY MATERIALS

See the supplementary material for the PDOS analysis of Au and Ag doping in pristine VSe<sub>2</sub> and the dioxane interaction on Au and Ag doped VSe<sub>2</sub>.

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#### AUTHOR DECLARATIONS

#### **Conflict of Interest**

The authors have no conflicts to disclose.

#### **Author Contributions**

Sreejith Pallikkara Chandrasekharan: Conceptualization (equal); Data curation (equal); Formal analysis (equal); Funding acquisition (equal); Investigation (equal); Methodology (equal); Software (equal); Visualization (equal); Writing - original draft (equal); Writing - review & editing (equal). See tha Lakshmy: Conceptualization (equal); Data curation (equal); Formal analysis (equal); Investigation (equal); Methodology (equal); Software (equal); Writing original draft (equal); Writing - review & editing (equal). Saju Joseph: Conceptualization (equal); Data curation (equal); Formal analysis (equal); Investigation (equal); Methodology (equal); Project administration (equal); Software (equal); Supervision (equal); Writing - original draft (equal); Writing - review & editing (equal). Nandakumar Kalarikkal: Conceptualization (equal); Formal analysis (equal); Funding acquisition (equal); Project administration (equal); Resources (equal); Supervision (equal); Validation (equal); Writing - original draft (equal).

#### DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding authors upon reasonable request.

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