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Rapid identification of efficient photocatalysts by visualizing the spatial distribution of photoinduced charge carriers

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ABSTRACT. The design of high-performance photocatalysts is of great importance for solar energy conversion. However, the rapid identification of efficient photocatalysts is challenging because of the complex dynamics of photoinduced charge carriers. Herein, we develop an experimental descriptor, the average fluorescence intensity (AFI), to rapidly identify the high-performance photocatalysts based on the fluorescence microscopic visualization of charge dynamics. As a proof-of-concept, CuFeO₂/ZIF-67 heterojunction was used to visualize the spatial distribution of photoinduced charges with identified diffusion length. Furthermore, a linear relationship was obtained between photocatalytic activities and the AFI, indicating an improved charge separation efficiency in the heterojunction responsible for an enhanced photocatalytic activity. This work provides a fundamental understanding of the dynamics of photoinduced charge carriers and offers a promising descriptor for the future design of efficient photocatalysts.

INTRODUCTION

The conversion of solar energy into chemical energy or high value-added chemicals by photocatalysis is an effective approach to utilize a clean energy while avoiding energy waste and environmental issues.¹ Unfortunately, most of the photocatalysts suffer from critical charge recombination, which severely restricts their large-scale applications.^{2,3} Accordingly, developing an efficient strategy for catalyst design to overcome the charge separation bottleneck has a great value in both fundamental and applied research.⁴ In this context, the heterojunction engineering has emerged as a promising strategy owing to the optimal charge separation by the built-in electric field between two semiconductors and the enlarged specific surface area.^{5,6} Thus, the in depth understanding of the complex dynamic behavior of photogenerated charge could provide guidance for the identification of high-performance photocatalysts.^{7,8,9} Therefore, it is necessary to deeply explore the dynamics of charge carriers, and establish an experimental descriptor between activity and charge behaviors to quickly identify high-performance catalysts.

Fluorescence microscopy visualization is a fast imaging method, which facilitates the exploration of charge carriers by using suitable probe molecules. Compared with conventional spectroscopy and electrochemical methods,^{10,11} fluorescence microscopy visualization has advantages on studying spatial distribution and dynamics parameters of photoinduced charge carriers. Among the photocatalysts, the 2D heterojunctions have several fascinating merits including short carrier transfer distance from the bulk to the surface and reduced internal recombination of photoinduced charge carriers.^{12,13,14} For these reasons, the 2D heterojunction is an interesting model to study charge dynamics. Recent efforts towards the 2D heterojunction lead to the discovery of numerous photocatalysts.^{15,16} Among them, delafossite CuFeO₂ and Cobased ZIF-67 material show a great potential.^{17,18,19}.

In this article, we directly visualize the spatial distribution of photoinduced charge carriers in CuFeO₂/ZIF-67 heterojunction by fluorescence microscopy with two different fluorescent probe molecules. The designed 0D on 2D CuFeO₂/ZIF-67 heterojunction has distinct morphologies providing an easy to observe model for visualization. We found that the interfacial built-in electric field could operate the photoinduced holes and electrons distributed on different sites of CuFeO₂/ZIF-67 heterojunction, which improves the charge separation efficiency. Furthermore, we developed an experimental descriptor, the average fluorescence intensity (AFI), to rapidly identify high-performance photocatalysts based on charge dynamics. A linear relationship between the photocatalytic activities and the AFI demonstrates that the heterojunction improves charge separation efficiency, thus enhancing photocatalytic activity. As a proof of concept, the photocatalytic selective oxidation of benzyl alcohol (BA) to benzaldehyde (BAD) as well as photocatalytic hydrogen evolution reaction were selected to evaluate the activity of CuFeO₂/ZIF-67 heterojunction. Thus, we demonstrated that the photocatalyst with the optimal performance (quantum efficiency of 81.26% and 4.78% for BA conversion and H₂ evolution, respectively) was rapidly identified using the AFI descriptor.

RESULTS AND DISCUSSION

Our previous work showed that 2D CuFeO₂ nanosheets with a thickness of 40 nm could be prepared successfully, providing an appropriate template for the construction of heterojunctions.²⁰ Thus, in this work, 2D CuFeO₂/ZIF-67 heterojunctions with distinct morphological structure could be designed for visualizing the spatial distribution of photoinduced charge carriers. The detailed preparation of CuFeO₂ nanosheet are described in SI.



Figure 1. Synthesis route for CuFeO₂/ZIF-67 heterojunction (a); SEM of CuFeO₂ and CuFeO₂/ZIF-67 heterojunction (b, c); EDS mapping of CuFeO₂/ZIF-67 (8h) heterojunction (d); TEM (e), HRTEM (f), and FFT (g) images of CuFeO₂/ZIF-67 heterojunction; XRD patterns of CuFeO₂, ZIF-67, and CuFeO₂/ZIF-67 (8h) heterojunction (h); Cu 2p XPS spectra in pure CuFeO₂ and CuFeO₂/ZIF-67 (8h) heterojunction (i).

The synthesis route is schemed in Figure 1a. SEM images (Figure 1b, c, Figure S1) show that ZIF-67 nanoparticles are grown on the CuFeO₂ surface, and with the increase of reaction time from 4 to 8 hours, ZIF-67 nanoparticles with a mean particle size of ca. 50 nm are gradually covered on the surface of CuFeO₂ nanosheets. The as prepared CuFeO₂/ZIF-67 heterojunction remain of hexagon nanosheet morphology with a diameter of 400 nm and a thickness of 40 nm. Scanning TEM coupled with EDS elemental mapping (Figure 1d) show elemental Cu and Fe are uniformly distributed on the surface of the hexagonal substrate, while the distribution of Co element is more dispersed in small islands structure. Such observationproves the successful synthesis of CuFeO₂/ZIF-67 heterojunction. The specific surface area of CuFeO₂, ZIF-67, and $CuFeO_2/ZIF-67$ heterojunction are shown in Table S1. The TEM image reveals the hexagon nanosheet decorated with 30 nm size particles =on the surface (Figure 1e). The FFT patterns from the selected area in HRTEM image (Figure 1f) are indexed to the CuFeO₂ [001] zone axis (Figure 1g). The XRD diffraction peaks match perfectly with CuFeO₂ (JCPDS No. 75-2146, 79-1546) and ZIF-67 (CCDC No. GITTOT) without any impurities (Figure 1h). The XPS spectra (Figure S2) demonstrate the existence of Cu, Fe, Co, C, N, and O in CuFeO₂/ZIF-67 and Cu, Fe, C, and O in CuFeO₂. From the Cu 2p XPS spectra (Figure 1i), two contributions at 934.1 and 932.2 eV were attributed to Cu^{II} and Cu^I, respectively. After construction of the heterojunction, the peak area ratio of $Cu^{I}/(Cu^{I} + Cu^{II})$ increases from 50% to 60%, which indicates the electron transfer from ZIF-67 to CuFeO₂ resulting in the increase of Cu¹.



Figure 2. The UV-vis-NIR absorption spectra of CuFeO₂, ZIF-67, and CuFeO₂/ZIF-67 heterojunction (8 h) (a); Mott-Schottky plots of CuFeO₂ and ZIF-67 in 0.5 M Na₂SO₄ (b); Chopped I-t curves (c) and electrochemical impedance spectra (d) of ZIF-67, CuFeO₂ and CuFeO₂/ZIF-67 heterojunction at -0.5 V vs. Ag/AgCl performed in 0.5 M Na₂SO₄ electrolyte; the schematic representation of energy band diagram with the calculated band gap and charge transfer process (e).

The optical UV-vis-NIR absorption spectra from 200 to 2000 nm are shown in Figure 2a. The CuFeO₂ sample presents two obvious absorption bands, including one in the UV-vis region (<1000 nm) and one in near IR region (1050-1500 nm), which are assigned to intrinsic band to band transition from Cu 3d and O 2p to Cu/Fe 3d orbitals, and d-d transitions at Fe^{III} site.^{18,21} The Fe-O interatomic distances are comparable in CuFeO₂ (2.0559 Å, COD ID: 1533117) and in Fe₂O₃ (1.9443/2.1136 Å, COD ID: 2101167). The ZIF-67 sample exhibits three absorption regions: UV (<380 nm), visible (420-720 nm), and near IR (720-1400 nm), which are attributed to the LMCT transition, the higher-lying $[{}^{4}A_{2}(F) \rightarrow {}^{4}T_{1}(P)]$ and lower-lying $[{}^{4}A_{2}(F) \rightarrow {}^{4}T_{1}(F)]$ d-d ligand field transitions at Co^{II} site, respectively.²² The absorption in visible region of green and yellow light makes the material of bluish violet color. Previous work reported that long-lived charge separation state occurs in ZIF-67 material after light irradiation, which is featured as semiconductor behavior strongly implying the potential application in photocatalysis.^{22,23,24} The CuFeO₂/ZIF-67 heterojunction and CuFeO₂ have similar absorption features below 1000 nm, while the absorption intensity above 1000 nm is higher for CuFeO₂/ZIF-67 heterojunction due to the absorption of ZIF-67. The transformed Tauc plots give the energy level difference between the ground state and the excited state of CuFeO₂ and ZIF-67 as shown in Figure S3.

To obtain the valence band (VB) edges of CuFeO₂ and ZIF-67, Mott-Schottky plots were measured. Both plots show negative slops indicating p-type conductivity (Figure 2b). From the intercept of plots with X axis, the flatband potentials (regarded as the VB) can be measured at 1.46 and 1.09 V vs. Ag/AgCl for CuFeO₂ and ZIF-67, respectively. The photoelectrochemical characteristics of the CuFeO₂ and CuFeO₂/ZIF-67 heterojunction electrodes were analyzed as shown in Figure 2c. The CuFeO₂ and CuFeO₂/ZIF-67 heterojunction display an instant photocathodic response when the sample is illuminated. The CuFeO₂/ZIF-67 heterojunction shows higher photocurrent than CuFeO₂ electrode. In addition, the CuFeO₂/ZIF-67 (8h) heterojunction presents the highest photocurrent of -7.37 μ A/cm², which demonstrates the presence of an enhanced charge separation in the heterojunction. The ZIF-67 exhibits no response, showing that a serious charge recombination exists in ZIF-67. The charge transfer behavior at the interface between photoelectrodes and electrolyte was investigated by electrochemical impedance spectroscopy (EIS) as shown in Figure 2d. For the photoelectrode prepared from the CuFeO₂/ZIF-67 heterojunction material, the CuFeO₂ particles are separated by ZIF-67, so the interfacial charge transfer within a single particle is included in the whole charge transfer channel of the photoelectrode. The interfacial charge transfer resistance within a single particle will affect the whole charge transfer resistance. So the EIS method can be used to analyze the interfacial charge transfer process in heterostructures. The CuFeO₂/ZIF-67 (8h) heterojunction shows a smaller semicircle diameter than other samples indicating a much lower charge-transfer resistance after heterojunction construction. Under the action of the interface electric field, the photogenerated electrons and holes are separated in different directions. This phenomenon inhibits recombination, improves the carrier life and enhances the photocatalytic efficiency. Additionally, the fluorescence spectra of CuFeO₂, ZIF-67, and CuFeO₂/ZIF-67

heterojunction were measured to illustrate the charge recombination process as shown in Figure S4. The CuFeO₂/ZIF-67 (8h) sample shows the weakest intensity among the different heterostructures indicating the enhanced charge separation efficiency benefited from the construction of heterojunction. Therefore, combined with the band gap from the Tauc plots (Figure S3), the conduction band (CB) is determined at 0.21 and -2.26 V vs. Ag/AgCl for CuFeO₂ and ZIF-67, respectively. CuFeO₂ has a lower CB and VB than ZIF-67. From these band gaps, a precise energy band diagram with the calculated band gap and charge transfer process can be proposed (Figure 2e). Under light ($\lambda = 400$ nm) irradiation, photoinduced holes transfer from the VB of CuFeO₂ to the VB of ZIF-67, which suppress the recombination of photoinduced charge.

To visualize the photoinduced charge transfer behavior, fluorescence microscopy was employed with amplex red and resazurin as probes of photoinduced holes and electrons, respectively.^{25,26} The absorption and photoluminescence spectra of the probes and their fluorescence state are shown in Figure S5. The fluorescence microscopy images of different photocatalysts were measured via dispersing in amplex red or resazurin aqueous solution. Under the excitation wavelength of 400 nm, the nonfluorescent amplex red and resazurin molecules may capture, respectively, photoinduced holes and photoinduced electrons from pure CuFeO₂ nanosheet and convert to fluorescence images is from 559-689 nm. Although the fluorescence bands of resorufin and resazurin are overlapped partially (Figure S5), the fluorescence of resazurin is treated as the background and has been subtracted in the fluorescence images. The fluorescence intensity is proportional to the concentration of photoinduced charge carriers. Therefore, the probe molecule can detect the behaviour of photoinduced carriers at any site, and

allows to evaluate the photocatalytic activity of this site. With the help of probe molecules, we successfully visualize the distribution of photoinduced charge carriers in $CuFeO_2$ sample (fluorescence microscopy images with electrons and holes probe molecules shown in Figure 3). The transition processes of nonfluorescent and fluorescent states of the probe molecules are illustrated in Figure 3a.



Figure 3. Transition processes of nonfluorescent and fluorescent states of probe molecules (a); fluorescence microscopy images of CuFeO₂ nanosheet in 50 nM resazurin (b-d); fluorescence intensity from line scanning of fluorescence microscopy images vs. the distance by using

resazurin (e); fluorescence microscopy images of CuFeO₂ nanosheet in 50 nM amplex red (f-h); fluorescence intensity from line scanning of fluorescence microscopy images vs. the distance by using amplex red (i); TEM (j) and element mapping images (k) of CuFeO₂ after photo-deposition experiment by using Mn^{2+} ; TEM (l) and element mapping images (m) of CuFeO₂ after photodeposition experiment by using Ag^+ ; calculated DOS of CuFeO₂ (001) and (-120) facets (n); schematic diagram of energy level (o) and photoinduced charge transfer process (p).

By using the electron probe molecule of resazurin, a strong fluorescence signal is observed for the CuFeO₂ nanosheet as shown in Figure 3b-d. The fluorescence intensity from line scanning of fluorescence microscopy images vs. the distance by using resazurin is shown in Figure 3e. When using the amplex red (probe of holes), a weak fluorescence signal is observed (Figure 3f-h) as few photoinduced holes are located at the surface. The fluorescence intensity from line scanning of fluorescence microscopy images vs. the distance by using amplex red is shown in Figure 3i. CuFeO₂ is a p-type semiconductor with a downward surface band bending. Accordingly, under the effect of the built-in electric field in the surface space charge region, the photoinduced electrons shift to the surface and the photoinduced holes move to the bulk. Furthermore, the fluorescence intensity at the center of CuFeO₂ particles is higher than that at the edge, and the uneven distribution of fluorescence intensity originates from the uneven charge concentration distribution at CuFeO₂ (001) facets. According to the theory of semiconductor physics, the behavior of excess carriers with time and in space in the presence of electric fields and density gradients are described by the Haynes-Shockley model²⁷. In such conditions, the continuity equation is expressed as:

$$\frac{\partial \Delta n}{\partial t} = D \frac{\partial^2 \Delta n}{\partial x^2} - \mu \varepsilon \frac{\partial \Delta n}{\partial t} - \frac{\Delta n}{\tau}$$
(1)

Where Δn , D, μ , τ and ε corresponds to the carrier concentration, diffusion constant, mobility, lifetime, and the strength of internal electric field, respectively. In our model, a horizontal polarized electric field (HPEF), which originates from the difference of energy levels between CuFeO₂ (001) and (-120) facets, exists and drives the photoinduced charge carriers movements in x-y plane (Figure 3p). The HPEF is also confirmed by the heterogeneous distribution of the fluorescence intensity. In other words, if the HPEF is absent, the fluorescence intensity should be homogeneous as the particles are excited at any sites. The solution of equation (1) is:

$$\Delta n = \frac{N_n}{\sqrt{4\pi Dt}} exp\left[-\frac{(x-\mu\varepsilon t)^2}{4Dt}\right] exp\left[-\frac{t}{\tau}\right]$$
(2)

Equation (2) could be simplified as:

$$\Delta n \propto e^{-\frac{(x-\mu\varepsilon\tau)^2}{4D\tau}} \ (3)$$

It can be seen that the carrier concentration distribution satisfies the Gaussian function. In the line scanning of fluorescence microscopy images, the fluorescence intensity is proportional to carrier concentration, which also agrees with the Gaussian distribution:

$$\Delta n \propto I \propto \frac{A}{\sigma\sqrt{2\pi}} e^{\left(-\frac{(x-x_c)^2}{2\sigma^2}\right)}$$
(4)

Where A is a constant, x is the variable, x_c is the mean, and σ is the standard deviation. By comparing the equation (3) and (4), the following equation is obtained:

$$D\tau = \frac{\sigma^2}{2} \ (5)$$

So the carrier diffusion length (L) is extracted from the standard deviation:

$$\mathcal{L} = \sqrt{D\tau} = \frac{\sigma}{\sqrt{2}} \ (6)$$

The L value of pure CuFeO₂ obtained from the Gaussian fitting shown in Figure 3e and 3i are 195.17 nm and 87.77 nm for photoinduced electrons and holes, respectively. Photo-deposition experiments were carried out to further confirm the dynamic behavior of photoinduced charge carriers in this model by using Mn^{2+} and Ag^+ as hole and electron probe, respectively. The element mapping images show that MnO_x and Ag nanoparticles deposited preferentially to deposit on side facets and top facets, respectively (Figure 3j-m). This distribution indicates that (-120) and (001) facets are enriched in holes and electrons, respectively. Additional density functional theory (DFT) calculations were carried out to explain the charge dynamics. The calculated DOS confirms that the conduction band and valence band of CuFeO₂ (001) face is lower than the ones of the (-120) face (Figure 3n). Such phenomenon generates a facet junction between the two facets. The difference of energy levels between the two facets provides the driving force, which lead to the photoinduced electrons transfer from (-120) to (001) face, and the photoinduced holes transfer from (001) to (-120) face (Figure 3o, p).



Figure 4. Fluorescence microscopy images of CuFeO₂/ZIF-67 (4, 6, 8, 10 h) heterojunction samples in 50 nM resazurin (a-d) and 50 nM amplex red (e-h) aqueous solution; the line scanning of fluorescence microscopy images of the particle 1 derived from (a)-(d) by using resazurin (i)-(1); the line scanning of fluorescence microscopy images of the particle 1 derived from (e)-(h) by using amplex red (m)-(p).

We further visualize the distribution of photoinduced charge carriers in different CuFeO₂/ZIF-67 heterojunctions, and carried out a statistical analysis from the fluorescence

microscopy images of at least three particles for each sample (Figure 4). In comparison to CuFeO₂, the fluorescence intensity distribution for CuFeO₂/ZIF-67 heterojunctions using electron probe molecule remains uneven with the central region exhibiting a higher intensity of the fluorescence (Figure 4a-d). Interestingly, when using holes probe molecules, an obvious fluorescence signal region located on the center of the heterojunction can be observed for all CuFeO₂/ZIF-67 samples (Figure 4e-h). This observation indicates that photoinduced holes populate the center of the catalyst under the action of the interface built-in electric field and the facet junction of CuFeO₂ simultaneously. ZIF-67 particles are located on the surface of CuFeO₂ nanosheet, which proves that the photoinduced holes are localized on the outer ZIF-67 particles. For CuFeO₂/ZIF-67 (10 h), the excess quantity of ZIF-67 hinders the absorption of light in the heterojunction region decreasing the charge concentration. Such phenomenon results in a weaker fluorescence signal in comparison to CuFeO₂/ZIF-67 (8 h). The line scanning of fluorescence microscopy images are shown in Figure 4i-p. The carrier diffusion length was obtained from the line scanning of fluorescence microscopy images as shown in Table S2, and the L value for CuFeO₂/ZIF-67 heterojunctions are longer than the L value of pure CuFeO₂. It should be noted that, the regions concentrated in electrons prefer to undergo catalysis reduction reaction, while the regions concentrated in holes are more inclined to catalysis oxidation reaction.^{28,29} All these observations promise that such CuFeO₂/ZIF-67 heterojunctions are favorable catalysts toward photocatalytic reactions.



Figure 5. The quantum efficiencies (QE) of oxidation of BA and H_2 evolution of CuFeO₂/ZIF-67 heterojunctions (2, 4, 6, 8, 10 h) (a); the QE of BA conversion versus descriptor (AFI) (b); the QE of H_2 evolution versus descriptor (AFI) (c).

As a proof of concept, the performance of photocatalytic selective oxidation of BA and H_2 evolution performance of the different as prepared heterojunctions were analyzed. At the beginning of the reaction, reactant molecules are adsorbed on different catalyst surface sites. The DFT calculations indicate that the adsorption energies of BA/H₂O are comparable for CuFeO₂ and ZIF-67 surfaces (Figure S6). Therefore, one can conclude that the photocatalytic activity mainly depends on the efficiency of photoinduced charge separation. The photocatalytic quantum efficiencies of the selective oxidation of BA and H₂ evolution are gradually increased from CuFeO₂/ZIF-67 (4 h) heterojunction to CuFeO₂/ZIF-67 (8 h) heterojunction (Figure 5a). The CuFeO₂/ZIF-67 (8 h) heterojunction exhibits the highest quantum efficiency (QE) of 81.26% and 4.78% for the photocatalytic selective oxidation of BA and H₂ evolution, respectively. The quantum efficiency for CuFeO₂/ZIF-67 (10 h) decreases, which resulted from the excess of ZIF-67 particles accumulated on the CuFeO₂ surface as this excess hinders the light absorption of CuFeO₂, weakens the effect of the heterojunction region and thus reduces the charge separation efficiency. The photocatalytic selective oxidation of BA is the thermodynamically downhill

reaction, and the photocatalytic H₂ evolution is thermodynamically uphill reaction (regarded as photosynthesis). Although the two types of reactions are obviously different in thermodynamics, from the material point of view, the considerations of photocatalyst properties, such as light absorption, charge separation and charge transfer, are analogous.³⁰ The recombination process of photoinduced charge carriers is important for photocatalysis, which affects the lifetime and the average distance of excess minority carriers from the location where the charge carriers are generated to the location where they are utilized.³¹

To identify a descriptor for the CuFeO₂/ZIF-67 heterojunction photocatalysts, we examined the relationship between the catalytic activities and the photoinduced charge carrier behavior. As an important elementary step in photocatalytic reaction, charge separation is crucial for increasing the performance of photocatalyst. Thus, a descriptor is desirable to optimize the charge separation efficiency and photocatalytic performance. The carrier diffusion length determines whether electron/hole created deeper in the bulk material can reach the surface, and the carrier concentration dominates the quantity of the electron/hole participating in the photocatalytic reaction.^{32,33} Therefore, a new experimental descriptor, the average fluorescence intensity (AFI), which quantifies the diffusion length and concentration of photoinduced charge carriers simultaneously, is proposed to rationalize the catalytic activity trends. The AFI is defined as the integral of line scanning curve divided by distance, which meets the abovementioned requirements. The physical meaning of AFI is the number of photoinduced charge carriers per unit length distance on the surface (Number/nm). The statistical AFI values of different CuFeO₂/ZIF-67 heterojunction samples are shown in Table S3. An excellent linear relationship between BA conversion and the AFI, as well as between H₂ evolution and the AFI, can be observed (Figure 5b, 5c), suggesting that the AFI can serve as a descriptor for predicting

the photocatalytic performance of heterojunction photocatalysts. The CuFeO₂/ZIF-67 (8 h) exhibits the highest AFI and shows the highest catalytic activity. Other samples with the shorter or longer growth time of ZIF-67 display weaker AFI, because of a lack of ZIF-67 leading to an insufficient photoinduced charge carriers generation or an excess of ZIF-67 leading to an insufficient light transmission reducing the action range of the junction region. All these results successfully demonstrated that the AFI is a practical descriptor for rapidly identifying efficient photocatalysts in both thermodynamically uphill and downhill reactions.

CONCLUSIONS

In summary, we focused on visualizing the distribution of photoinduced charge carriers in $CuFeO_2/ZIF-67$ heterojunction by fluorescence microscope with probe molecules. The heterojunction enhanced the separation efficiency of photoinduced charge carriers under the effect of interfacial built-in electric field, which improved the carrier diffusion length and concentration. Furthermore, we reveal that AFI can be used as an experimental descriptor to rapidly identify the high-performance photocatalyst. Thus, heterojunction samples with different AFI are demonstrated to display photocatalytic performance following the trends of $CuFeO_2/ZIF-67$ (2 h) < $CuFeO_2/ZIF-67$ (4 h) < $CuFeO_2/ZIF-67$ (6 h) < $CuFeO_2/ZIF-67$ (10 h) < $CuFeO_2/ZIF-67$ (8 h). Thus, we believe that the AFI could be a promising descriptor for the future screening of potential photocatalysts.

EXPERIMENTAL

Preparation of CuFeO₂ nanosheet.

The CuFeO₂ nanosheet was prepared by hydrothermal method based on our previous work.²⁰ More details are shown in Supporting Information (SI).

Preparation of CuFeO₂/ZIF-67 heterojunction.

First, 0.65 mmol CuFeO₂, 0.1 mmol Co(NO₃)₂·6H₂O and 2 mmol 2-methylimidazole were dispersed in 10 mL of DI water under stirring. Next, the mixture was transferred into a Teflonlined stainless autoclave heating at 120 °C for 4, 6, 8 h to synthetize the heterojunction samples, which were named as CuFeO₂/ZIF-67 (4/6/8 h). Finally, the products were washed three times with water and ethanol, respectively and dried at 80 °C for 12 h. ZIF-67 was also prepared in the absence of CuFeO₂ under the same condition.

Characterization.

Scanning electron microscopy (SEM) images were obtained using a Zeiss Supra 55 electron microscope. High resolution transmission electron microscopy (HRTEM) characterizations were performed using a Tecnai G2 F30 S-TWIN transmission electron microscope operated at 200 kV. X-ray diffraction (XRD) patterns of the products were recorded using a Bruker D8 Focus diffractometer by using Cu K α radiation ($\lambda \Box = 1.54178$ Å). X-ray photoelectron spectroscopy (XPS) spectra were obtained using an ESCALAB 250Xi Thermo Scientific X-ray photoelectron spectrometer. The CuFeO₂/ZIF-67 based work electrodes were fabricated using a dip-coating method. Typically, 0.008 g of the CuFeO₂ material was added in 5 mL of ethylene glycol monomethylether solution under ultrasound. Then, the sample suspension was dropped onto the FTO substrate and dried at 90 °C following a heating at 200 °C in a tube furnace for 30 min under N₂ atmosphere. Photoelectrochemical measurements were performed in a three-electrode system using an electrochemical workstation (Autolab).

Fluorescence microscopy imaging.

The catalysts powder (10 mg) was dispersed in 1 mL of ethanol and was sonicated for 1 minute. A portion (10 μ L) of the above colloidal suspension was then dipped onto a cleaned glass slide heat at 90 °C to remove the ethanol getting a well dispersed catalysts particles layer. A portion (10 µL) of 50 nM resazurin/amplex red aqueous solution was then dipped onto the above prepared catalysts layer, and the liquid drop was carefully covered with a glass coverslip avoiding any gas bubbles. Then the catalysts on glass slide were used to perform the fluorescence microscopy imaging. A Zeiss LSM 880 NLO two-photon laser confocal microscopy system was used consisting of a Zeiss Axio Imager 2 optical microscope with oilimmersion objective lens (×100). The resolution limit of the fluorescence microscopy (ZEISS LSM 880 NLO, with the commercial ZEISS Airyscan detrctor) is of 130 nm in x, y and 350 nm in z. (https://www.zeiss.com) The operating software has the function of image magnification by adjusting the zoom settings, which could increase the magnification to see more details without objective change. To conclude, based on the above information, it is possible to identify the charge separation in a micro-sized particle. The well-isolated, non-aggregated catalyst particles visualized by bright field microscopy imaging were selected as regions of interest. Two photon laser line was typically used while recording the fluorescence images. Fluorescence images were analysed by ZEN software, and the line scanning of fluorescence microscopy images were obtained.

Photocatalytic reaction.

The photocatalytic reaction was carried out in a 30 mL quartz test bottle with a diameter of 26 mm and the light source is a 450 nm LED with light intensity of 17.93 W/m^2 . The photocatalytic

selective oxidation of BA was carried out under air atmosphere by dispersing 30 mg catalyst in 5 mL of DMF with 1 mmol BA and 1.5 mg KOH, and the reaction time is 4 h. The BAD yields were determined by gas chromatography-mass spectrometry (Agilent 7890B). The photocatalytic hydrogen evolution reaction was carried out by dispersing 30 mg catalyst in 60 mL of 10% triethanolamine aqueous solution and the reaction time is 1.5 h. The H₂ evolution rate was determined by gas chromatography (GC-2014C). The quantium efficiency is calculated by the following equation^{34,35}:

$$QE = 2 \times \frac{N(Product)}{N(Photons)}$$

where N(Product) and N(Photons) denote the number of product molecules (BAD or H₂) and the number of photons reaching the surface of the reaction solution, respectively.

DFT calculation.

The electronic structure of the system was investigated based on the DFT approach with the CASTEP program. Both structures were optimized at the generalized gradient approximation (GGA) level with Perdew–Burke–Ernzerhof (PBE) exchange–correlation functional and the ultra-soft pseudopotentials. To obtain the density of states (DOS) of CuFeO₂, we performed the PBE + U calculations with a value of U = 4.1 eV for the Fe 3d states, which is in agreement with previous reports for Fe atoms.³⁶ For geometry optimization, ultra-soft pseudopotentials with a kinetic energy cut-off of 351 eV and a G-centered $2\times 2\times 1$ k-point mesh for sampling the Brillouin zone. During the optimization, the lattice parameters were fixed and only the atoms were allowed to relax. The vacuum layer of the models is set to be 15 Å to minimize the interaction between the periodic images.

ASSOCIATED CONTENT

Supporting Information. This material is available free of charge via the Internet at http://pubs.acs.org. Experimental details, SEM, XPS spectra, Tauc plots, fluorescence spectra, UV-vis-NIR absorption spectra, fluorescence microscopy images, DFT results and gas chromatograms data.

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Author Contributions

Tengfei Jiang conceived and designed the experiments, analysed the data, and wrote the paper. Jingying Wei performed the experiments and revised the manuscript. Jianyang Li performed the experiments and collected the data. Huaiguo Xue designed the project and revised the manuscript. Jingqi Tian conceived and designed the experiments, and revised the manuscript. Romain Gautier conceived and designed the experiments, and revised the manuscript.

Notes

The authors declare no competing financial interest.

SUPPORTING INFORMATION

Experimental

Preparation of CuFeO₂ nanosheet.

A mixture of 1 mmol CuSO₄·5H₂O, 1 mmol FeSO₄·7H₂O, and 10 mL sodium alginate of 5 g/L were dissolved into 10 mL of deionized water by ultrasonic stirring for 10 min. The mixture was added in a Teflon-lined autoclave (50 mL), and then adding 10 mL 10 M NaOH into the autoclave and ultrasonic stirring for 5 min and putting the autoclave into drying oven at 160 °C for 6 h. Finally, the synthesized nanoflakes were washed three times with water and ethanol and dried at 80 °C for 12 h.²⁰

RESULTS AND DISCUSSION



Fig. S1. SEM images of CuFeO₂/ZIF-67 (4h) (a), CuFeO₂/ZIF-67 (6h) (b), CuFeO₂/ZIF-67 (8h)

(c), and $CuFeO_2/ZIF-67$ (10h) (d).



Fig. S2. XPS survey spectrum of CuFeO₂/ZIF-67 and pure CuFeO₂ (a); Co 2p spectrum in CuFeO₂/ZIF-67 (b); Fe 2p spectra in pure CuFeO₂ and CuFeO₂/ZIF-67 (c); O 1s spectra in pure CuFeO₂ and CuFeO₂/ZIF-67 (d).

The XPS survey spectra (Fig. S2a) demonstrate the existence of Cu, Fe, Co, C, N, and O in CuFeO₂/ZIF-67 and Cu, Fe, C, and O in CuFeO₂. From the Co 2p XPS (Fig. S2b), the contribution at 781.1 eV was attributed to fully coordinated cobalt within the Co-N₄ environment.^{37,38}. The observable satellite features at 786 eV suggest the valence state of cobalt is Co^{II}. The Fe 2p XPS peak at 710.8 eV (Fig. S2c) are contributed to Fe^{III}. From the O 1s XPS (Fig. S2d), two contributions at 531.8 and 529.5 eV were attributed to the surface hydroxyl and lattice oxygen, respectively. Thus, the increased surface hydroxyl in CuFeO₂/ZIF-67 is mainly attributed to the adsorbed O-species.



Fig. S3. Tauc plots of CuFeO₂ and ZIF-67 from UV-vis-NIR spectra and the intercept between absorption band edge and baseline.



Fig. S4. Fluorescence spectra of ZIF-67, $CuFeO_2$ and $CuFeO_2/ZIF$ -67 heterostructure dispersed

in DMF at the excitation wavelength of 330 nm.



Fig. S5. The UV-vis-NIR spectra (a-c) and PL spectra with λ_{EX} =400 nm (d-f) of 25 nM

Resorufin, Resazurin and Amplex Red ethanol solution.



Fig. S6. DFT calculated adsorption energy of BA and H₂O on catalyst surfaces (a); the calculated adsorption configurations of BA and H₂O on catalyst surfaces (b-e).

Samples	Specific surface area (m ² /g)	Pore volume (cm ³ /g)	Pore size (nm)
CuFeO ₂	18.411	0.081	3.934
ZIF-67	1122.223	0.603	2.486
CuFeO ₂ /ZIF-67 (4h)	25.730	0.111	3.133
CuFeO ₂ /ZIF-67 (6h)	49.040	0.165	3.317
CuFeO ₂ /ZIF-67 (8h)	64.382	0.285	3.499
CuFeO ₂ /ZIF-67 (10h)	73.249	0.265	3.133

Table S1. Specific surface area, pore volume, and average pore size of CuFeO₂, ZIF-67, and CuFeO₂/ZIF-67 heterojunction samples.

	Carrier diffusion length (L, nm)		
Sample			
	in Resazurin	in Amplex red	
	105 17	07 77	
CuFeO ₂	195.17	8/.//	
$CuFeO_2/ZIF-67$ (4 h)	659.31	192.33	
CuFeO ₂ /ZIF-67 (6 h)	299.46	100.08	
CuFeO ₂ /ZIF-67 (8 h)	464.68	160.99	
$C_{\rm uEeO_{2}}/ZIE_{-67}$ (10 h)	344 21	378 71	
$Cur CO_2/2m^{-}Or (10m)$	JTT.21	570.71	
CuFeO ₂ /ZIF-67 (10 h)	344.21	378.71	

Table S2. The carrier diffusion length of different CuFeO₂/ZIF-67 heterojunction samples.

Comple	Statistical AFI (Number/nm)		
Sample	in Resazurin	in Amplex red	
CuFeO ₂ /ZIF-67 (4 h)	3321.904	3078.398	
CuFeO ₂ /ZIF-67 (6 h)	8609.697	6271.438	
CuFeO ₂ /ZIF-67 (8 h)	22073.600	9351.519	
CuFeO ₂ /ZIF-67 (10 h)	5699.618	6110.161	

Table S3. The statistical AFI values of different CuFeO₂/ZIF-67 heterojunction samples.

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SYNOPSIS.



An experimental descriptor, the average fluorescence intensity (AFI), was developed to rapidly identify the high-performance photocatalyst using fluorescence microscopic visualization.