



**HAL**  
open science

## Fluorescence decay enhancement and FRET inhibition in self-assembled hybrid gold CdSe/CdS/CdZnS colloidal nanocrystal supraparticles

V Blondot, C Arnold, Aymeric Delteil, D Gérard, A Bogicevic, T Pons, N  
Lequeux, J-P Hugonin, J-J Greffet, S Buil, et al.

### ► To cite this version:

V Blondot, C Arnold, Aymeric Delteil, D Gérard, A Bogicevic, et al.. Fluorescence decay enhancement and FRET inhibition in self-assembled hybrid gold CdSe/CdS/CdZnS colloidal nanocrystal supraparticles. *Optics Express*, 2023, 31, pp.4454. 10.1364/oe.476441 . hal-03955656

**HAL Id: hal-03955656**

**<https://hal.science/hal-03955656>**

Submitted on 25 Jan 2023

**HAL** is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Distributed under a Creative Commons Attribution 4.0 International License



# Fluorescence decay enhancement and FRET inhibition in self-assembled hybrid gold CdSe/CdS/CdZnS colloidal nanocrystal supraparticles

V. BLONDOT,<sup>1</sup> C. ARNOLD,<sup>1</sup> A. DELTEIL,<sup>1</sup>  D. GÉRARD,<sup>1</sup> A. BOGICEVIC,<sup>2</sup> T. PONS,<sup>2</sup> N. LEQUEUX,<sup>2</sup> J.-P. HUGONIN,<sup>3</sup> J.-J. GREFFET,<sup>3</sup>  S. BUIL,<sup>1</sup>  AND J.-P. HERMIER<sup>1,\*</sup>

<sup>1</sup>Université Paris-Saclay, UVSQ, CNRS, GEMaC, 78000, Versailles, France

<sup>2</sup>Laboratoire de Physique et d'Étude des Matériaux, ESPCI-Paris, PSL Research University, CNRS UMR 8213, Sorbonne Université, 10 rue Vauquelin, 75005 Paris, France

<sup>3</sup>Université Paris-Saclay, Institut d'Optique Graduate School, CNRS, Laboratoire Charles Fabry, F- 91127 Palaiseau, France

\*[jean-pierre.hermier@uvsq.fr](mailto:jean-pierre.hermier@uvsq.fr)

**Abstract:** We report on the synthesis of hybrid light emitting particles with a diameter ranging between 100 and 500 nm, consisting in a compact semiconductor CdSe/CdS/CdZnS nanocrystal aggregate encapsulated by a controlled nanometric size silica and gold layers. We first characterize the Purcell decay rate enhancement corresponding to the addition of the gold nanoshell as a function of the particle size and find a good agreement with the predictions of numerical simulations. Then, we show that the contribution corresponding to Förster resonance energy transfer is inhibited.

© 2023 Optica Publishing Group under the terms of the [Optica Open Access Publishing Agreement](#)

Plasmonic structures enable to modify the photoluminescence properties of individual fluorophores for a wide range of applications such as quantum optics [1,2], optoelectronics [3] or the detection of molecules of chemical or biological interest [4]. When colloidal semiconductor nanocrystals (NCs) with a CdSe core are considered, many configurations involving plasmonic surfaces or nanoparticles were reported [5–8]. They allow to enhance the emission intensity, to control the emission pattern or to suppress the NC residual blinking.

From a general point of view, as an alternative to photonic structures, confinement of multiple emitters in a small volume offers the possibility to benefit from their interaction in order to enhance the emission through coherent dipole-dipole interactions leading to superradiance or superfluorescence [9]. Plasmonic devices and cooperative effects can be combined and hybrid metal/emitters structures were designed to develop SPASER for example [10].

Dipole-dipole interactions can also generate a non-reversible and incoherent energy transfer which corresponds to FRET (Förster resonance energy transfer), especially when condensed matter nanoemitters showing a large inhomogeneity of their spectral properties such as NCs are considered [11,12]. Widely used for the investigation of biological or chemical systems [13,14], the modification and control of FRET through optical cavities - eventually leading to its suppression - is also the subject of numerous works in the field of optoelectronics for the development of devices such as LEDs or photovoltaic devices [15–17]. On a fundamental level, C. L. Cortes and Z. Jacob have shown that the fluorescence enhancement by the Purcell effect could be tuned independently of the increase or inhibition of FRET [18].

In a recent paper, we have demonstrated the synthesis of NC aggregates surrounded by a silica shell, called supraparticles (SPs) in the following, which exhibit a perfectly stable emission intensity at room temperature [20]. This study also highlighted that the decay rate corresponding

to FRET is of the order of the radiative decay rate. Here, we characterize the emission properties of similar structures when coupled to a nanoresonator consisting of a thin gold shell encapsulating the SPs. We first describe the synthesis of these nanoobjects and their elementary structural properties. Next, using several spectroscopic and/or complementary time resolved setups, we show the reduction of the FRET efficiency induced by the gold nanoshell.

## 1. Chemical synthesis and structural properties of GSPs

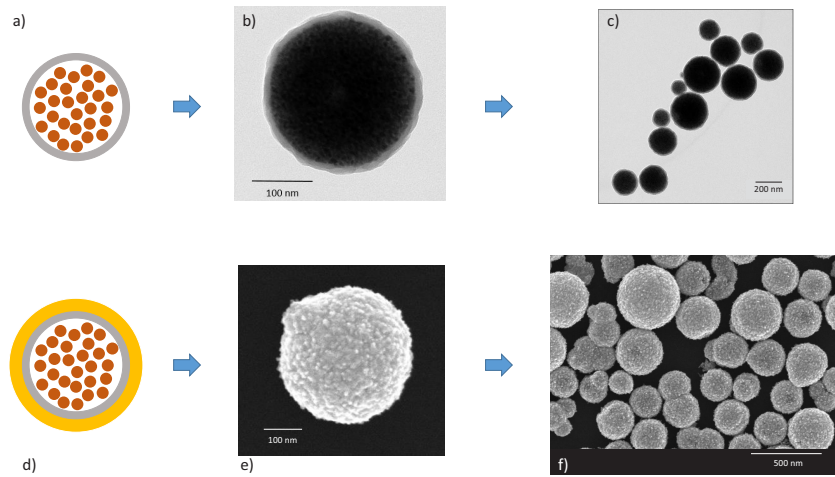
A multistep procedure was performed to prepare a hybrid structure containing few hundred NCs inside a gold nanoshell.

### 1.1. SP synthesis

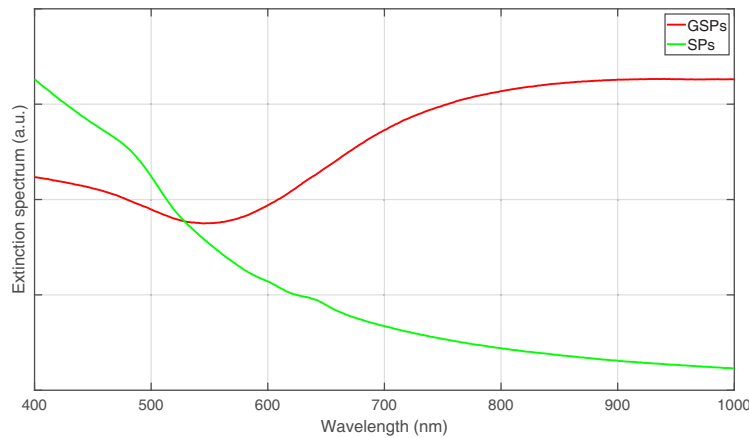
First, CdSe NCs were synthesized by a procedure slightly modified from Cao *et al.* [21] and shells of CdS/CdZnS were synthesized with continuous injection of shell precursors at high temperatures [22]. Typically, these reactions generate CdSe/CdS/CdZnS NCs of about  $7.7 \pm 1$  nm in size with an emission centered at 645 nm (FWHM = 30 nm). These NCs are then assembled into spherical aggregates using a micro-emulsion/evaporation method [23]. Typically, 6 nmol of NCs were re-dispersed in 1 mL of chloroform and added to 1 mL of dodecyltrimethylammonium bromide solution (20 mg/mL in water). The solution was vortexed for 30 seconds to obtain a stable oil in water emulsion. The chloroform was evaporated by heating the solution at 70°C in an oil bath for 10 minutes. The obtained NC assemblies stabilized by DTAB are hereafter referred to as NC SPs. They were purified by centrifugation at 6000 g for 5 minutes and re-dispersed in 4 mL of ethanol. NC SPs were coated with a silica shell by hydrolysis and condensation of tetraorthoethylsilicate (TEOS) precursors, see Fig. 1(a), (b), (c). Its thickness ( $15 \pm 2$  nm) prevents thereafter quenching of the emission of the NCs located close to the surface of the SP by the gold nanoshell [24,25]. For a typical reaction, 1 mL of NC SP solution was first sonicated in 9 mL of ethanol and 1.5 mL of deionized water for 10 minutes. Then 0.5 mL of ammonia solution (NH<sub>4</sub>OH, 28 wt% in water) and 10  $\mu$ L TEOS were added. The mixture was sonicated for 20 minutes to prevent aggregation [26]. The silica-coated NC SPs were purified by centrifugation and re-dispersed in 1 mL of ethanol.

### 1.2. Gold nanoshell deposition

Silica shells were functionalized by reaction with aminosilane precursors [27]. Typically, 500  $\mu$ L of NC SPs coated with silica were added to a mixture of 500  $\mu$ L of ethanol, 1 mL of water and 50  $\mu$ L of aminopropyltriethoxysilane and reacted at 60°C for 2 h. The solution was washed 3 times with ethanol and twice with water to remove excess silane and then re-dispersed in 1 mL of water. We follow the common synthesis route developed by Halas and coworkers [28] to obtain gold nanoshells with a thickness of  $19 \pm 5$  nm. Gold seeds were prepared by the Duff method [29]. 100  $\mu$ L from the previous 1 mL functionalized NC SPs were mixed with 1 mL of gold seeds solution for 30 minutes. The NC SPs were purified by centrifugation in water and re-dispersed in 300  $\mu$ L of water. 100  $\mu$ L from this solution were then dispersed in 5 mL of gold growth solution [28] followed by the addition of 25  $\mu$ L of formaldehyde and stirred for 2 h. The final gold-coated NC SPs (golden supraparticles, GSPs see Fig. 1(d), (e), (f)) were centrifuged in water and re-dispersed in 500  $\mu$ L of water. The full description and characterization of the NC GSP synthesis at the different steps of the preparation are described in a separate report [30]. The addition of gold on the silica shell results in plasmon resonances that induce a substantial modification of the extinction spectrum (see Fig. 2) [31,32,33,34]. Above 600 nm, we observe that the extinction progressively increases as the gold seed grows [30], which was used to monitor the gold shell synthesis.



**Fig. 1.** Schematic representations of single supraparticles of NCs after coating with silica (a) and gold (d). Transmission electron microscopy image of a single SP (b) and Scanning electron microscopy image of a GSP (e). Large field of view of images for SPs (c) and GSPs (f).



**Fig. 2.** Extinction spectra of SP and GSP solutions.

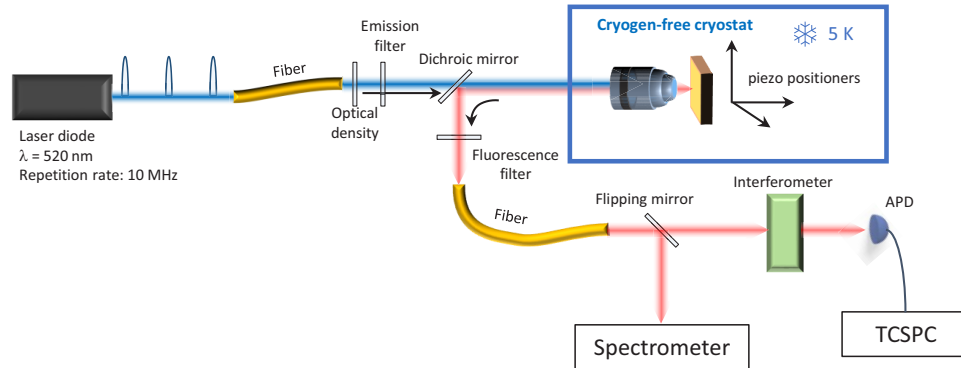
## 2. Emission properties

### 2.1. Experimental setup

To characterize the emission properties of SPs at the single object level, we spin coat the solution of the particles on a glass coverslip, with a  $\text{TiO}_2$  grid with numbered cells previously prepared by photolithography. In the case of the GSPs, the concentration is so low that we only deposit 40  $\mu\text{L}$  of the solution and let the solvent evaporate. Using an atomic force microscope (AFM), we locate the individual particles with respect to the grid and measure their size, so that the confocal experiments will be performed on supraparticles whose size are known.

In order to carry out the optical measurements (see Fig. 3), a confocal microscope (Attocube Attodry 1100, numerical aperture of the objective = 0.82) operating at 4 K (Cryomech cryocooler) focuses a pulsed laser diode beam (Picoquant, LDH-D-C-520, FWHM  $\sim 160$  ps) of low fixed power (the probability to excite one NC of the aggregate is of the order of 10 %) at the diffraction

limit on particles that were previously identified with the AFM. Photons are detected with single photon avalanche photodiodes (SPAD) (MPD, time resolution 50 ps) connected to a Time Correlated Single Photon Counting (TCSPC) device (Picoquant, PicoHarp 300) that enables to measure the photoluminescence decay. An interferometer (GEMINI, Nireos model) can be placed before the SPAD, which provides after a Fourier transform the fluorescence as a function of detection wavelength and the photon detection time (maximal spectral resolution  $\sim 0.7$  nm). Finally, a spectrometer and a cooled CCD camera allow to reduce the spectral accuracy down to 0.05 nm, at the cost of a time resolution of several seconds.



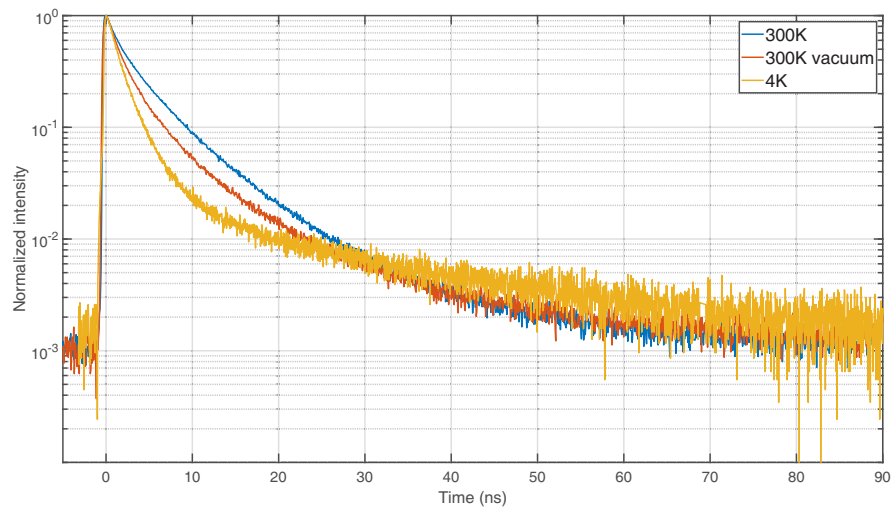
**Fig. 3.** Setup for time and spectrally resolved photoluminescence experiments.

## 2.2. Emission of supraparticules and Purcell effect

Figure 4 shows the PL decay of a typical GSP under low excitation power and under various conditions: at 300 K in air or vacuum (low helium pressure  $\sim 10^{-3}$  mbar), or 4 K (same low helium pressure). In vacuum, thick shell CdSe/CdS NCs are well known to become ionized since no water or oxygen molecules can neutralize the emitter once it has been photoionized [35]. The photoluminescence (PL) lifetime then decreases since the trion recombines twice as fast as the exciton [36]. At 4 K, the trion lifetime is even shorter due to the modification of the conduction band offset at cryogenic temperature [35]. For the single CdSe/CdS/CdZnS NCs used to synthesize the SPs and GSPs, we measured a PL lifetime about 5.6 ns in vacuum. As a result, the increase of the PL decay rate of GSPs at 300 K when placed in vacuum instead of air (Fig. 4) shows that NCs inside a GSP become ionized. Indeed, the porosity of the chemically synthesized silica and gold shells does not prevent the escape of water or oxygen molecules from the SPs or GSPs. In the following, the experiments are performed at 4 K and we will consider that the NCs forming the aggregates are always ionized.

If we consider a single NC inside an aggregate, the close proximity of the surrounding NCs composed of materials with refractive indices around 2.5 is likely to modify its electromagnetic environment compared to single NCs in vacuum. Moreover, the sub-wavelength dimension of the SPs results in the confinement of the electromagnetic field, leading to a Purcell-enhanced emission. In order to investigate the influence of the surrounding environment for NCs inside the SPs, we use a multilayer spherical model based on Mie theory (see section 1 of the [Supplement 1](#)). We model the aggregate part of our structures as a homogeneous material with a permittivity being the average of the permittivities of the different materials composing the NCs weighted by their relative fraction in the total volume. We assume a packing density of NCs of 66 % in the aggregates [38].

To model gold-coated aggregates, the roughness of the metallic layer is described by a homogeneous layer following the method of Ref. [39]. The gold refractive index is defined



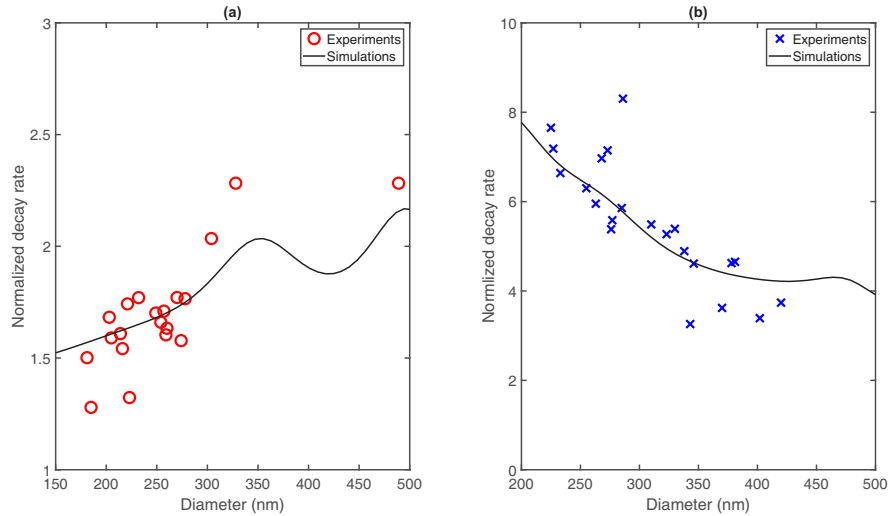
**Fig. 4.** Photoluminescence of a single GSP in air and at 300 K (blue), in vacuum and at 300 K (red) and at 4 K (yellow).

thanks to a Drude model adjusted from Ref. [40]. To account for the specific optical properties of gold obtained by chemical process, we followed Ref. [39] where a fivefold increase of the optical losses and a slight adjustment of the plasma frequency is done on the Drude model (see also Section 2 of Supplement 1 for more details). These high Joule losses result in plasmonic modes which show no resonances.

Using Mie theory, we expand the field emitted by a single dipole on a basis of spherical harmonics. We can then compute the power emitted by the dipole inside the supraparticle [41]. Transition rates of atoms near spherical surfaces are averaged over all wavelengths and over the different positions of the dipole inside the aggregate. From the emitted power, we can derive the decay rate (additional details concerning the different steps of the calculation are provided in the section 3 of the Supplement 1). In principle, we should also take into account the dielectric interface corresponding to the glass coverslip that modifies a dipole emission located at a distance lower than typically  $0.1 \lambda = 60 \text{ nm}$  [42–44]. However, we checked that this interface only slightly modifies the dissipated power of supraparticles (see the gap between the curves for which the interface is either taken into account or not, section 4 Supplement 1), so we neglected it for all the calculations presented in this article. Since the modification of the dipole emission by a dielectric interface is essentially a near field effect [43], the negligible role of the glass substrate is consistent with the large size of the SPs (greater than 200 nm) and the shell thickness (15 nm for the SPs and 34 nm for the GSPs). Finally, to calculate the Purcell factor, we use a single dipole above a glass substrate as reference (the latter is obtained following [42]).

Experimentally, we first fitted the PL decay rates of single NCs at 4 K under vacuum by a bi-exponential function, the first component corresponding to the decay of the trion [35]. The luminescence decay of individual aggregates are fitted with a log-normal distribution with a central decay rate  $\Gamma_{LN}$  [20] for the NC emission, plus a mono-exponential function to take into account a slow emission from traps [45]. Figure 5(a) shows the decay rates of non-gold coated SPs as a function of their diameters, normalized by the mean decay rate of single NCs deposited on a glass substrate. A good agreement is obtained between the measured decay rates  $\Gamma_{LN}$  and the decay rates predicted by the model. Emission enhancement is measured between 1.3 and 2.3 in comparison with single NCs. These accelerations come from the high refractive index of

the aggregate as well as Mie resonances. A small discrepancy is observed since non radiative recombination channels may appear during the synthesis.



**Fig. 5.** Decay rates of SPs (a) and GSPs (b) normalized by the decay rate of individual emitters. Black curves correspond to the decay rates obtained by the numerical simulations. Crosses and circles are the decay rates of the log-normal distribution used to fit measurements.

We now turn to decay rates of gold-coated GSPs (Fig. 5(b)). As expected, the emission enhancement is even more pronounced than for non-metallized aggregates, particularly for the smaller GSPs where accelerations of 8 compared to individual NCs have been obtained. Calculations show that the fraction of radiative recombination channels is about 30 %, meaning that the enhancement of the decay rate (as well as the resulting decrease of the FRET efficiency, see section 2.3) mainly comes from the generation of non-radiative channels (see section 5 of the Supplement 1). It should be noted that the good agreement between model predictions and measurements for GSPs and SPs does not necessitate accounting for FRET. Indeed, we now show with a basic model that FRET does not contribute to the overall PL decay time if the quantum efficiency and the radiative decay rate of the "blue" NCs (corresponding to the NCs with the shortest wavelength emission) are the same as the "red" ones. Let us consider the basic situation of excited donor (D) and acceptor (A) emitters characterized by single fluorescence ( $k$ ) and FRET ( $k_{FRET}$ ) decay rates. This simplified system can be depicted by the following rate equations:

$$\begin{aligned}\frac{dN_D}{dt} &= -k_{FRET}N_D - kN_D \\ \frac{dN_A}{dt} &= +k_{FRET}N_D - kN_A\end{aligned}$$

By summing the two equations, one finds that the total number of excited emitters  $N = N_D + N_A$  follows:

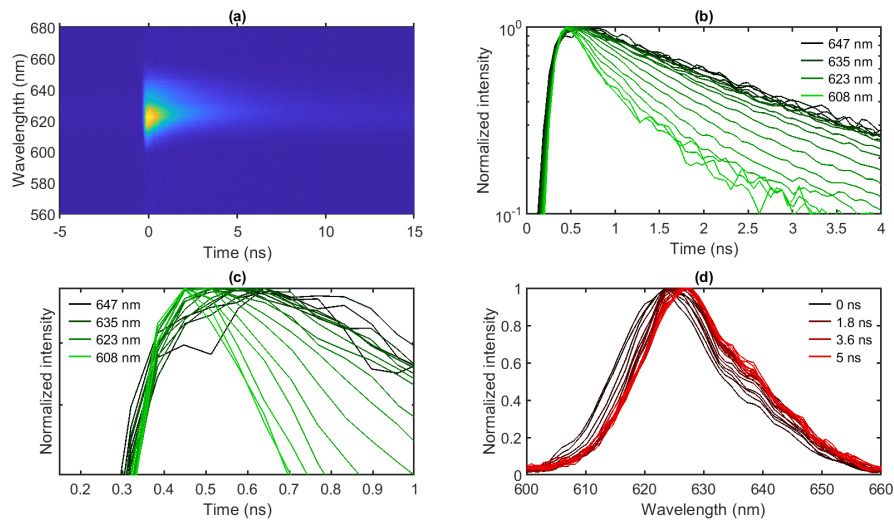
$$\frac{dN}{dt} = -kN$$

which means that the overall decay rate is left unchanged. The generalization to multiple population classes is straightforward and leads to the same conclusion. The FRET only modifies the emission spectrum of the NCs ensemble. The emission spectrum of the SP is redshifted with respect to the NC solution used for the SP synthesis. More precisely, the emission spectrum as a whole is translated by about 5 nm towards low energies [20]. This first analysis focused on

Purcell effect, to compare FRET between NCs in the SPs and GSPs. In the next section, we turn into the measurements allowing spectrally-resolved measurements of the decay rates.

### 2.3. FRET

Aggregates of NCs are known to exhibit FRET [12–14,37]. In a SP, the energy transferred from a NC is all the greater as its wavelength emission is shifted towards the blue, as a result of an increased overlap between its emission spectrum and the average absorption spectrum of its neighbors [20]. We now show that the gold nanocavity modifies the impact of FRET on the PL emission properties of the GSPs with respect to the SP ones. As already mentioned, the Fourier transform of the signal recorded by the SPAD at the output of the interferometer enables to plot a fluorescence map as a function of the detection wavelength and the time of photon detection  $\tau$  (Fig. 6(a)). From this data, we can plot the PL decay for a given wavelength (Fig. 6(b)) or, alternatively, the evolution of the emission spectrum at a given time after the pulsed excitation (Fig. 6(d)). Both data sets demonstrate the strong effect of FRET on non-metallized SPs. First, PL decays for shorter wavelengths are faster than for longer ones. FRET is also evidenced by the wavelength dependence of the delay between the pulse excitation and the PL signal peak. Due to energy transfer from the "blue" NCs to the "red" ones, the maximum emission of the "red" is delayed with respect to the "blue" one. Lastly, the normalized spectrum is redshifted over time (Fig. 6(d)).



**Fig. 6.** (a): example of a fluorescence map of a single SP. (b): PL decay for a SP as a function of the wavelength from 647 nm (black curve) to 608 nm (lightest green). (c): zoom of the same PL decay. (d): Evolution of the emission spectrum of the same SP during a time interval of 5 ns.

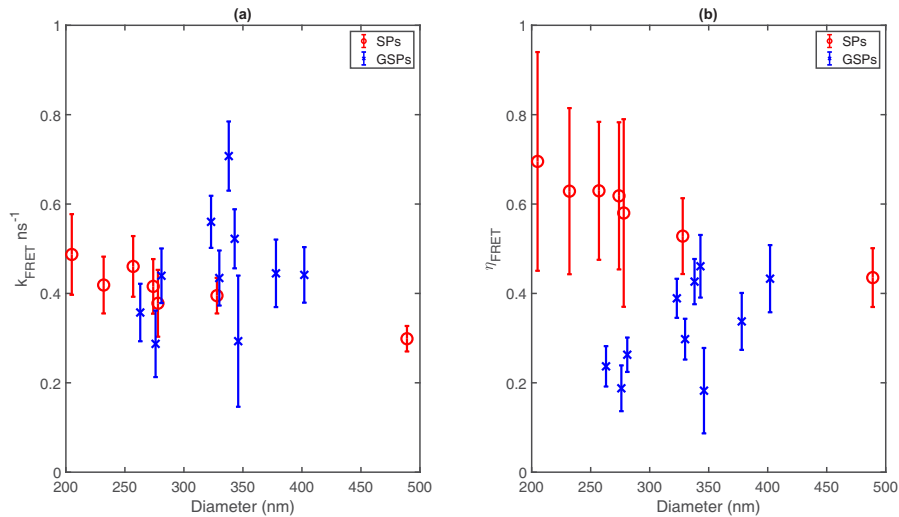
Simulations done on SPs show negligible variation of the Purcell-enhanced decay rates when the wavelength varies inside the NC emission spectrum (section 6 of the Supplement 1). Since the decay rate of the trion  $k$  does not depend on the NC emission wavelength [20], the variations of the decay rate with the wavelength can be entirely attributed to FRET. As a result, the decay rate  $k(\lambda)$  for NCs inside a GSP emitting at a wavelength  $\lambda$  writes:

$$k(\lambda) = k + k_{FRET}(\lambda)$$

where  $k_{FRET}(\lambda)$  is the decay rate due to FRET. The variations that we measure between the fast decay rates of the "blue" NCs and the slower "red" NCs come from the FRET decay rate



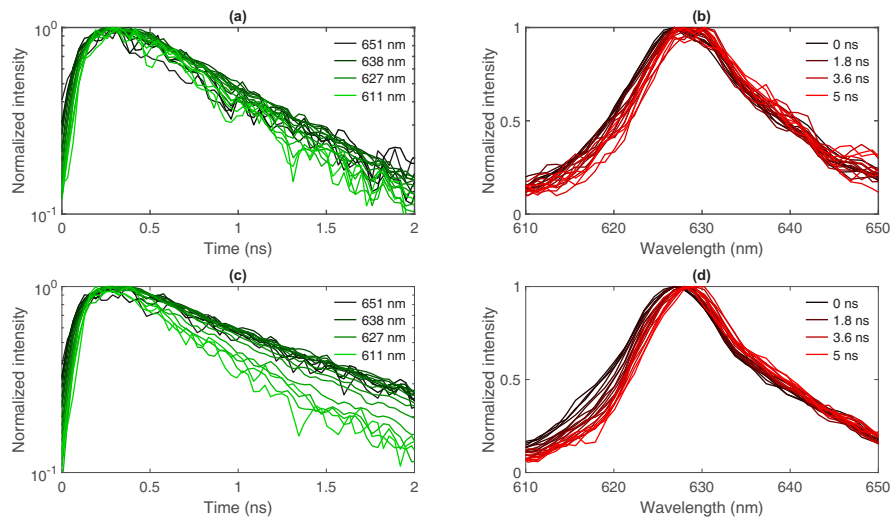
$k_{FRET}$ . In [20], we have also shown that the energy transferred by FRET from a given NC decreases with its wavelength emission and vanishes for the highest wavelength due to a vanishing number of acceptors, that is  $\lambda = 647$  nm. Figure 7(a) shows the value of  $k_{FRET}$  deduced by the difference between the decay rates measured at 640 nm and at 615 nm. An average value of  $k_{FRET}(\lambda = 615 \text{ nm}) = 0.45 \text{ ns}^{-1}$  is obtained. The variations between SPs can be due to the difficulty to fit the PL decay, the spectral filtering reducing the number of counts in particular at the edge of the emission spectrum.



**Fig. 7.** FRET analysis on SPs (red circles) and GSPs (blue crosses). (a): FRET decay rates obtained by doing the differences of the decay rates at 615 and 640 nm as a function of the diameters. (b): FRET efficiencies, defined as the ratio between  $k_{FRET}$  and the decay rates measured at short wavelengths. The error bars correspond to the uncertainties provided by the fits.

In contrast with SPs, the effect of FRET is far less visible for the GSPs. The differences between the decay rates of "blue" NCs and "red" NCs are less marked (Fig. 8(c)) than on SPs and even hardly visible on some GSPs (Fig. 8(a)). This is explained by the emission enhancement by Purcell effect, which dominates over the FRET rates since  $k(\lambda) = Fpk + k_{FRET}(\lambda)$ . We also point out that the shift of the PL decay maximum (Fig. 6(c)) with the wavelength is no more visible. Simultaneously, the modification of the normalized spectrum over time is small (Fig. 8(b) and Fig. 8(d), to be compared with Fig. 6(b) and Fig. 6(d)), and all the more so as the contribution of FRET is low. Figure 7(a) shows that we obtain comparable  $k_{FRET}$  on GSPs and SPs. As shown by C. L. Cortes and Z. Jacob in their detailed modeling [18], the effect of the gold plasmonic resonator on the FRET rate is not straightforward: it can either have no effect, increase or decrease the ratio between the total decay rate and the FRET rate in a way that strongly depends on the position of the donor-acceptor pair with respect to the photonic mode, independently of the evolution of the radiative rate.

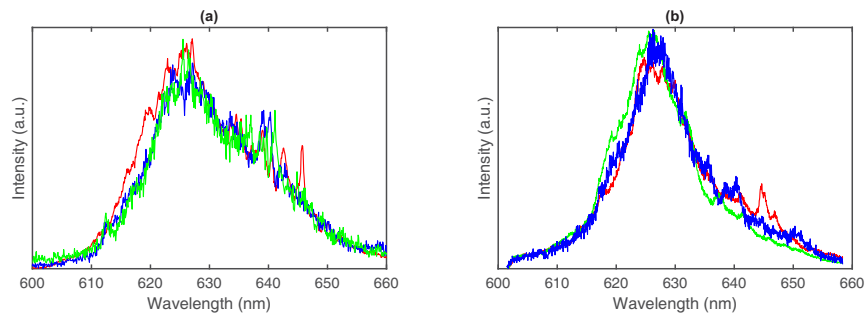
In order to determine the probability that a NC emitting around 615 nm decays by FRET rather than other recombination channels, the FRET efficiency (which is defined as the ratio between  $k_{FRET}$  and the total decay rate [19]) is deduced from measurements (Fig. 7(b)). For the smallest structures (below 300 nm), the FRET efficiency of GSPs is reduced by a factor of 3 compared to non-metallized SPs, showing the strong impact of the metallic layer to reduce FRET of NCs inside SPs. The fluctuations are larger for GSPs than for SPs due to the GSP-dependent creation of non-radiative channels during the gold shell synthesis [46]. We can also notice that FRET



**Fig. 8.** (a): PL decay for GSP<sub>1</sub> as a function of the wavelength from 651 nm (black curve) to 611 nm (lightest green). (b): Emission spectrum of GSP<sub>1</sub> as a function of time from 0 ps (black curve) to 5 ns (lightest red). (c): PL decay for GSP<sub>2</sub> as a function of the wavelength from 651 nm (black curve) to 611 nm (lightest green). (d): Evolution of the emission spectrum of the same GSP during a time interval of 5 ns.

efficiencies become comparable for larger diameters, which was expected since the Purcell effect becomes similar for SPs and GSPs when the size increases.

The modification of the FRET efficiency is also confirmed by the shape of the emission spectrum integrated over a few tens of seconds. FRET broadens the overall spectrum for the SP with respect to the GSP one (Fig. 9). In the case of SPs, FRET favors the emission of the "red" NCs through their excitation by the "blue" ones. The result is a relative increase of the "red" part of the emission spectrum and a relative decrease of the central peak height (the systematic asymmetry of the emission spectrum towards the red indicates that a small fraction of the NCs are slightly larger). The envelop of the SP and GSP emission spectra also exhibits fluctuations that have been analyzed in detail in [46].



**Fig. 9.** Emission spectrum of three single SPs (a) and three single GSP (b).

### 3. Conclusion

In conclusion, we first have presented the synthesis of hybrid structures consisting in NCs aggregates surrounded by a gold nanoresonator. As expected, the PL decay rate is increased by the metallic structure. We have shown that the mean Purcell factor is well predicted by a model and depends on the particle size. In addition, through the study of the SPs with or without a gold shell, we have also analyzed in detail the influence of the metallic resonator on FRET. Especially, interferometric measurements providing time resolved fluorescence spectra show that the relative contribution of FRET decreases by a factor of 3 for the smallest GSPs. Beyond the fine structure of the PL temporal dynamics, we have also found that the overall spectrum of the GSPs is narrowed with respect to the SPs ones. In terms of applications, the GSPs are submicronic bright and photostable sources that could be used for example for lighting. More fundamentally, the inhibition of FRET, which is an incoherent energy transfer process, opens the possibility to achieve coherent interactions between NCs that could enable to achieve quantum collective emission regime such as superradiance.

**Funding.** Agence Nationale de la Recherche (GYN project ANR-17-CE24-0046).

**Disclosures.** The authors declare no conflicts of interest.

**Data Availability.** Data underlying the results presented in this paper are not publicly available at this time but may be obtained from the authors upon reasonable request.

**Supplemental document.** See [Supplement 1](#) for supporting content.

### References

1. M. Tame, M. K. McEney, S. Özdemir, J. Lee, S. A. Maier, and M. S. Kim, "Quantum plasmonics," *Nat. Phys.* **9**(6), 329–340 (2013).
2. Z. K. Zhou, J. Liu, Y. Bao, L. Wu, C. E. Png, X. H. Wang, and C. W. Qiu, "Quantum plasmonics get applied," *Progress Quantum Electron.* **65**, 1–20 (2019).
3. Z. Liang, J. Sun, Y. Jiang, L. Jiang, and X. Chen, "Plasmonic Enhanced Optoelectronic Devices," *Plasmonics* **9**(4), 859–866 (2014).
4. A. Brolo, "Plasmonics for future biosensors," *Nat. Photonics* **6**(11), 709–713 (2012).
5. D. E. Gómez, K. C. Vernon, P. Mulvaney, and T. J. Davis, "Surface Plasmon Mediated Strong Exciton-Photon Coupling in Semiconductor Nanocrystals," *Nano Lett.* **10**(1), 274–278 (2010).
6. T. B. Hoang, G. M. Akselrod, C. Argyropoulos, J. Huang, D. R. Smith, and M. H. Mikkelsen, "Ultrafast spontaneous emission source using plasmonic nanoantennas," *Nat. Commun.* **6**(1), 7788 (2015).
7. Y.-C. Yao, Z.-P. Yang, J.-M. Hwang, H.-C. Su, J.-Y. Haung, T.-N. Lin, J.-L. Shen, M.-H. Lee, M.-T. Tsai, and Y.-J. Lee, "Coherent and Polarized Random Laser Emissions from Colloidal CdSe/ZnS Quantum Dots Plasmonically Coupled to Ellipsoidal Ag Nanoparticles," *Adv. Opt. Mater.* **5**(3), 1600746 (2017).
8. A. Coste, L. Moreaud, G. Colas des Francs, S. Buil, X. Quélin, E. Dujardin, and J.-P. Hermier, "Dramatic decrease of the Joule losses in the coupling between a crystalline gold and single colloidal CdSe/CdS nanocrystals at 4 K," *Phys. Rev. B* **101**(7), 075406 (2020).
9. R. H. Dicke, "Coherence in Spontaneous Radiation Processes," *Phys. Rev.* **93**(1), 99–110 (1954).
10. S. I. Azzam, A. V. Kildishev, R.-M. Ma, C.-Z. Ning, R. Oulton, V. M. Shalaev, M. I. Stockman, J.-L. Xu, and X. Zhang, "Ten years of spasers and plasmonic nanolasers," *Light: Sci. Appl.* **9**(1), 90 (2020).
11. M. Anni, L. Manna, R. Cingolani, D. Valerini, A. Creti, and M. Lomascolo, "Förster energy transfer from blue-emitting polymers to colloidal CdSe/ZnS core shell quantum dots," *Appl. Phys. Lett.* **85**(18), 4169–4171 (2004).
12. J. A. Jolene Mork, M. C. Weidman, F. Prins, and W. A. Tisdale, "Magnitude of the Förster Radius in Colloidal Quantum Dot Solids," *J. Phys. Chem. C* **118**(25), 13920–13928 (2014).
13. I. L. Medintz, A. R. Clapp, H. Mattoussi, E. R. Goldman, B. Fisher, and J. M. Mauro, "Self-assembled nanoscale biosensors based on quantum dot FRET donors," *Nat. Mater.* **2**(9), 630–638 (2003).
14. T. Pons, I. L. Medintz, M. Sykora, and H. Mattoussi, "Spectrally resolved energy transfer using quantum dot donors: Ensemble and single-molecule photoluminescence studies," *Phys. Rev. B* **73**(24), 245302 (2006).
15. H. Zhang, Q. Su, and S. Chen, "Suppressing Förster Resonance Energy Transfer in Close-Packed Quantum-Dot Thin Film: Toward Efficient Quantum-Dot Light-Emitting Diodes with External Quantum Efficiency over 21.6%," *Adv. Opt. Mater.* **8**(10), 1902092 (2020).
16. J. Li, S. K. Cushing, F. Meng, T. R. Senty, A. D. Bristow, and N. Wu, "Plasmon-induced resonance energy transfer for solar energy conversion," *Nat. Photonics* **9**(9), 601–607 (2015).
17. L.-Y. Hsu, W. Ding, and G. C. Schatz, "Plasmon-Coupled Resonance Energy Transfer," *J. Phys. Chem. Lett.* **8**(10), 2357–2367 (2017).

18. C. L. Cortes and Z. Jacob, "Fundamental figures of merit for engineering Förster resonance energy transfer," *Opt. Express* **26**(15), 19371 (2018).
19. D. Nettel, D. Haenni, S. Maillot, M. Gueye, A. Barth, V. Hirschfeld, C. G. Hübner, J. Léonard, and B. Schuler, "Excited-state annihilation reduces power dependence of single-molecule FRET experiments," *Phys. Chem. Chem. Phys.* **17**(48), 32304–32315 (2015).
20. V. Blondot, A. Bogicevic, A. Coste, C. Arnold, S. Buil, X. Quélin, T. Pons, N. Lequeux, and J.-P. Hermier, "Fluorescence properties of self assembled colloidal supraparticles from CdSe/CdS/ZnS nanocrystals," *New J. Phys.* **22**(11), 113026 (2020).
21. Y. A. Yang, H. Wu, K. R. Williams, and Y. C. Cao, "Synthesis of CdSe and CdTe nanocrystals without precursor injection," *Angew. Chem. Int. Ed.* **44**(41), 6712–6715 (2005).
22. K. Boldt, N. Kirkwood, G. A. Beane, and P. Mulvaney, "Synthesis of Highly Luminescent and Photo-Stable, Graded Shell CdSe/CdxZn1-xS Nanoparticles by In Situ Alloying," *Chem. Mater.* **25**(23), 4731–4738 (2013).
23. F. Bai, D. Wang, Z. Huo, W. Chen, L. Liu, X. Liang, C. Chen, X. Wang, Q. Peng, and Y. Li, "A Versatile Bottom-up Assembly Approach to Colloidal Spheres from Nanocrystals," *Angew. Chem. Int. Ed.* **46**(35), 6650–6653 (2007).
24. E. P. A. M. Bakkers, A. W. Marsman, L. W. Jenneskens, and D. Vanmaekelbergh, "Distance-Dependent Electron Transfer in Au/Spacer/Q-CdSe Assemblies," *Angew. Chem. Int.* **39**, 2297–2299 (2000).
25. J. R. Lakowicz, "Radiative decay engineering 5: metal-enhanced fluorescence and plasmon emission," *Anal. Biochem.* **337**(2), 171–194 (2005).
26. M. Ammar, F. Mazaleyrat, J.-P. Bonnet, P. Audebert, A. Brosseau, G. Wang, and Y. Champion, "Synthesis and characterization of core-shell structure silica-coated Fe<sub>29.5</sub>Ni<sub>70.5</sub> nanoparticles," *Nanotechnology* **18**(28), 285606 (2007).
27. S. Villa, P. Riani, F. Locardi, and F. Canepa, "Functionalization of Fe<sub>3</sub>O<sub>4</sub> NPs by Silanization: Use of Amine (APTES) and Thiol (MPTMS) Silanes and Their Physical Characterization," *Materials* **9**(10), 826 (2016).
28. S. J. Oldenburg, R. D. Averitt, S. L. Westcott, and N. J. Halas, "Nanoengineering of optical resonances," *Chem. Phys. Lett.* **288**(2-4), 243–247 (1998).
29. D. G. Duff, A. Baiker, and P. P. Edwards, "A New Hydrosol of Gold Clusters. 1. Formation and Particle Size Variation," *Langmuir* **9**(9), 2301–2309 (1993).
30. A. Bogicevic, X. Xu, S. Buil, C. Arnold, J.-P. Hermier, T. Pons, and N. Lequeux, "Synthesis and characterization of colloidal quantum dot superparticles - plasmonic gold nanoshell hybrid nanostructures," *ChemRxiv*, chemrxiv-2022-1m9q5 (2022).
31. B. E. Brinson, J. B. Lassiter, C. S. Levin, R. Bardhan, N. Mirin, and N. J. Halas, "Nanoshells made easy: improving Au layer growth on nanoparticles surfaces," *Langmuir* **24**(24), 14166–14171 (2008).
32. H. Wang, D. W. Brandl, F. Le, P. Nordlander, and N. J. Halas, "Nanorice: A hybrid plasmonic nanostructure," *Nano Lett.* **6**(4), 827–832 (2006).
33. H. S. Zhou, I. Honma, H. Komiyama, and J. W. Haus, "Controlled synthesis and quantum-size effect in gold-coated nanoparticles," *Phys. Rev. B* **50**(16), 12052–12056 (1994).
34. C. Loo, A. Lin, L. Hirsch, M.-H. Lee, J. Barton, N. Halas, J. West, and R. Drezek, "Nanoshell-Enabled Photonics-Based Imaging and Therapy of Cancer," *Technol. Cancer Res. Treat.* **3**(1), 33–40 (2004).
35. C. Javaux, B. Mahler, B. Dubertret, A. Shabaev, A. V. Rodina, A. L. Efros, D. R. Yakovlev, F. Liu, M. Bayer, G. Camps, L. Biadala, S. Buil, X. Quélin, and J.-P. Hermier, "Thermal activation of non-radiative Auger recombination in charged colloidal nanocrystals," *Nat. Nanotechnol.* **8**(3), 206–212 (2013).
36. C. Galland, Y. Ghosh, A. Steinbrück, J. A. Hollingsworth, H. Htoon, and V. I. Klimov, "Lifetime blinking in nonblinking nanocrystal quantum dots," *Nat. Commun.* **3**(1), 908 (2012).
37. M. Rafipoor, R. Koll, J.-P. Merkl, L. S. Fruhner, H. Weller, and H. Lange, "Resonant Energy Transfer can Trigger Multiexciton Recombination in Dense Quantum Dot Ensembles," *Small* **15**(5), 1803798 (2019).
38. A. Zaccone, "Explicit Analytical Solution for Random Close Packing in d=2 and d=3," *Phys. Rev. Lett.* **128**(2), 028002 (2022).
39. J. Botao, G. Emerson, H. Benjamin, S. Piernicola, M. Nasiowski, X. Xu, N. Lequeux, J.-P. Hugonin, F. Marquier, J.-J. Greffet, and B. Dubertret, "Non-blinking quantum dot with a plasmonic nanoshell resonator," *Nat. Nanotechnol.* **10**(2), 170–175 (2015).
40. P. B. Johnson and R. W. Christy, "Optical Constants of the Noble Metals," *Phys. Rev. B* **6**(12), 4370–4379 (1972).
41. H. Chew, "Transition rates of atoms near spherical surfaces," *J. Chem. Phys.* **87**(2), 1355–1360 (1987).
42. R. R. Chance, "Lifetime of an emitting molecule near a partially reflecting surface," *J. Chem. Phys.* **60**(7), 2744–2748 (1974).
43. W. Lukosz and R. E. Kunz, "Light emission by magnetic and electric dipoles close to a plane interface. i Total radiated power," *J. Opt. Soc. Am.* **67**(12), 1607 (1977).
44. A. Egel, S. W. Kettlitz, and U. Lemmer, "Efficient evaluation of Sommerfeld integrals for the optical simulation of many scattering particles in planarly layered media," *J. Opt. Soc. Am. A* **33**(4), 698 (2016).
45. P. H. Sher, J. M. Smith, P. A. Dalgarno, R. J. Warburton, X. Chen, P. J. Dobson, S. M. Daniels, N. L. Pickett, and P. O'Brien, "Power law carrier dynamics in semiconductor nanocrystals at nanosecond timescales," *Appl. Phys. Lett.* **92**(10), 101111 (2008).
46. A. Delteil, V. Blondot, S. Buil, and J.-P. Hermier, "First-order coherence of light emission from inhomogeneously broadened mesoscopic ensembles," *Phys. Rev. B* **106**(11), 115302 (2022).