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Polysaccharide-based aerogels for thermal insulation and

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superinsulation: an overview

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11

12 **Abstract**

13 To reduce energy losses due to the insufficient thermal insulation is one of the current “hot” topics.
14 Various commercial porous materials are used with the best conductivity around 0.03 – 0.04 W/(m·K).
15 Aerogels are the only known materials with “intrinsic” thermal superinsulating properties, i.e. with
16 thermal conductivity below that of air in ambient conditions (0.025 W/(m·K)). The classical thermal
17 superinsulating aerogels are based on silica and some synthetic polymers, with conductivity 0.014 –
18 0.018 W/(m·K). Aerogels based on natural polymers are new materials created at the beginning of the
19 21st century. Can bio-aerogels possess thermal superinsulating properties? What are the bottlenecks
20 in the development of bio-aerogels as new high-performance thermal insulation materials? We try to
21 answer these questions by analyzing thermal conductivity of bio-aerogels reported in literature.

22

23 Key-words: cellulose; chitosan; pectin; alginate; starch; thermal conductivity; thermal insulation;
24 density; specific surface area

25

26

27

28 **1. Introduction**

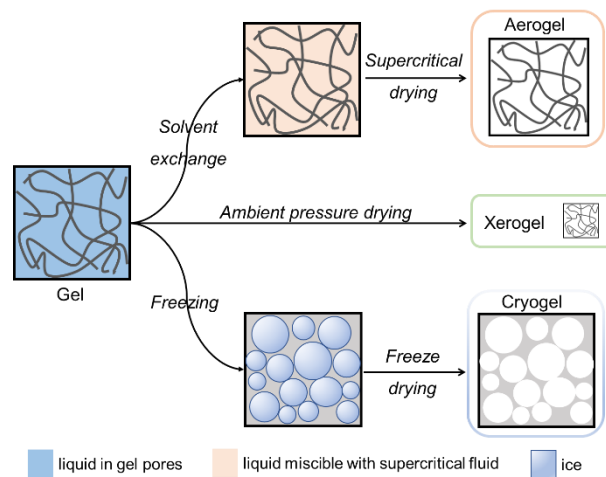
29 Insufficient thermal insulation leads to huge energy losses inducing massive energy consumption
30 and greenhouse gas emissions. To solve this problem, thermal insulation needs to be improved which
31 is usually done by using various porous materials. If willing to decrease the thickness of the insulating
32 layer, high performance materials with the lowest possible thermal conductivity are required. The best
33 commercial insulating materials, such as mineral wool and polymer foams, have thermal conductivity
34 around 0.030 – 0.040 W/(m·K) (Alam, Singh, & Limbachiya, 2011). The family of the most performing
35 materials and components with thermal conductivity below that of air in ambient conditions (0.025
36 W/(m·K)) are the *superinsulating* materials. Vacuum insulation panels possess the lowest conductivity,
37 around 0.007 – 0.008 W/(m·K), but they are still of rather high cost and with problems related to aging
38 and mechanical damage (Alam et al., 2011). Till now, only aerogels are known to be “intrinsically”
39 thermal superinsulating materials at the atmospheric pressure due to their mesoporosity (pore size from
40 2 to 50 nm) and low density (around 0.1 g/cm³). The lowest values of the thermal conductivity were
41 obtained for silica and resorcinol-formaldehyde aerogels, around 0.012 – 0.015 W/(m·K) (see, for
42 example, Koebel, Rigacci, & Achard (2012); Lu et al. (1992)), which decreases by more than half the
43 thickness of the insulation layer as compared to the commercial materials (Alam et al., 2011). These
44 extraordinary properties make aerogels very attractive materials for thermal insulation in buildings, pipe
45 lines, aerospace, appliances and transport, boosting current research and development towards the
46 improvement of aerogels’ properties and decrease of their price (Koebel, Huber, Zhao, & Malfait, 2016).
47 Some metal oxide aerogels possess enhanced heat resistance properties at high temperatures with
48 thermal conductivity down to 0.05 W/(m·K) (see, for example, Zu et al., 2014)

49 It is important to first define the term “aerogel”. IUPAC Gold Book describes an aerogel as “a gel

50 comprised of a microporous solid in which the dispersed phase is a gas” (IUPAC. Compendium of
51 Chemical Terminology 2014). This definition is rather restrictive as it includes the materials only with
52 pore sizes below 2 nm, thus excluding even classical silica aerogels which are mesoporous. It should
53 also be noted that in our days the term “aerogels” is somehow overused as various types of low-density
54 porous materials, even with low specific surface area and pores of diameter of several tens and
55 hundreds of microns, are called aerogels. Porous materials as expanded polystyrene or polyurethane
56 foam are not aerogels. As agreed between aerogel scientists, aerogels are dry open pores
57 nanostructured (mesoporous with small macropores) materials with high porosity (above 90%) and high
58 specific surface area (at least above 100 m²/g but no formal agreement exists (Pierre, 2011; Ganesan
59 et al., 2018; Budtova et al., 2020).

60 In general, aerogels are obtained from gels that are dried in supercritical conditions in order to
61 preserve gel morphology (Figure 1). It is also possible to obtain high specific surface area materials
62 using freeze-drying, but those are exceptional cases. In the following, gels that are freeze-dried will be
63 called “cryogels” (sometimes called “foams” in literature) (Figure 1). Finally, gels that are dried either
64 under ambient conditions or low vacuum will be called “xerogels” (Figure 1). In most cases, under these
65 drying conditions a gel loses its porosity because of structure collapse. Here again there are exceptional
66 cases when functionalization of the gel network helps the preservation of gel morphology after ambient
67 pressure drying; the well-known case is silylation of silica gels resulting in a material with aerogel-like
68 morphology and properties (Schwertfeger, Frank, & Schmidt, 1998).

69



70

71

Figure 1.

72

Schematic presentation of the preparation of aerogels, xerogels, and cryogels from a gel.

73

74

Various matters (silica and metal oxides, monomers-precursors of synthetic polymers) have been

75

used to synthesise aerogels; for the detailed information see Aerogels Handbook, 2011. In addition to

76

low thermal conductivity, aerogels are suggested to be used for oil/water separation (Wang X. et al.,

77

2020; Zou, Peng, Fu, Zhang, & Li, 2015), drug release (Follmann et al., 2020; Lim et al., 2020), energy

78

storage (Lin et al., 2018; Zhou et al., 2018), for adsorption (Maleki, Whitmore, & Hüsing, 2018) and as

79

catalyst supports (Wang M. et al., 2020; Zhang et al., 2017). At the beginning of the 21st century, a new

80

class of aerogels was made: they are based on natural polymers such as polysaccharides and proteins

81

(Zhao, Malfait, Guerrero-Alburquerque, Koebel, & Nyström, 2018; Budtova et al., 2020). Since the past

82

15 years the interest to bio-aerogels grew rapidly. One of the reasons is the use of renewable resources

83

in order to reduce the environmental impact of synthetic polymer industry and move towards more

84

sustainable society. Another reason is that bio-aerogels are biodegradable and often biocompatible

85

which makes them very promising for biomedical applications. Finally, bio-aerogels can potentially be

86

used in the same applications as classical aerogels: for separation, gas sorption, filtration, catalysts

87 and energy. The obvious “hot” question is “can a bio-aerogel be a thermal superinsulating material?”
88 Literature reports on the thermal conductivity of various bio-aerogels but a systematic analysis
89 summarizing the main trends and open questions is missing; an overview would help in moving forward
90 (or not) towards bio-aerogels’ application for thermal insulation.

91 In this review, we focus on the relationships between the thermal conductivity and properties of
92 aerogels based on polysaccharides. In some cases, the results on the thermal conductivity of bio-
93 cryogels and bio-xerogels are included for the comparison. We do not discuss the overall properties
94 and applications of each type of polysaccharide aerogel as each deserves a separate review article
95 (see, for example, reviews on nanocellulose aerogels and foams (De France, Hoare, & Cranston, 2017;
96 Lavoine & Bergström, 2017; Sun, Chu, Wu, & Xiao, 2021), on cellulose II aerogels (Budtova, 2019) and
97 on starch aerogels (Zhu, 2019)). We start with the background on the thermal conductivity of porous
98 materials and methods to obtain bio-aerogels and their characterization. Subsequently, we overview
99 the current status of the thermal conductivity of aerogels based on cellulose, chitin and chitosan, pectin,
100 starch and alginate. Finally, problems and prospects related to bio-aerogels’ applications for thermal
101 insulation are discussed.

102

103 **2. Background on the thermal conductivity of porous materials**

104 The thermal conductivity (λ) of an isotropic porous material is, in the first approximation, a sum of
105 gas (λ_g), solid (λ_s) and radiation (λ_r) components:

$$106 \quad \lambda = \lambda_s + \lambda_g + \lambda_r \quad (1)$$

107 In general, heat transfer by convection should have also been taken into account but this term is
108 negligible when the size of the pores is smaller than around 1 mm.

109 Thermal conductivity of a solid phase refers to the heat transfer along the network structure of a
110 porous material and it is mainly affected by density. For example, for monolithic resorcinol-formaldehyde
111 aerogels and their carbons the relationship between λ_s and aerogel bulk density ρ_{bulk} was shown to be
112 described by power law (Lu et al., 1992; Lu, Nilsson, Fricke, & Pekala, 1993):

$$113 \quad \lambda_s = C\rho_{bulk}^a \quad (2)$$

114 where C is the thermal conductivity of the backbone matter and a is a scaling exponent, usually
115 around 1.2 – 2 (see, for example, Ebert, 2011). Therefore, one way to decrease the thermal conductivity
116 of aerogel is to decrease its density.

117 Gas thermal conductivity refers to the heat transfer through the gaseous phase. At a given
118 temperature and pressure, the main factors affecting λ_g are porosity ϕ and pore size D ; convection
119 can be neglected (Reichenauer, Heinemann, & Ebert, 2007; Wiener, Reichenauer, Braxmeier,
120 Hemberger, Weis, Reichenauer, & Ebert, 2009) :

$$121 \quad \lambda_g = \frac{\phi\lambda_{g0}}{1+2\beta l/D} \quad (3)$$

122 where λ_{g0} is the thermal conductivity of the free gas, β is a constant value related to the interactions
123 between pore walls and gas molecules and is usually around 1.5 and l is the average free path of the
124 gas molecule (around 70 nm for the air in ambient conditions). For a material of a given porosity, gas
125 thermal conductivity can be decreased by decreasing pore diameter, and below l value λ_g may become
126 lower than λ_{g0} (Knudsen effect). For example, it was demonstrated that decreasing pore diameter of
127 a carbon aerogel from 10 μm to 200 nm, λ_g measured in argon at 0.1 MPa and 300 °C can be reduced
128 from around 0.05 to 0.015 W/(m·K) (Hemberger, Weis, Reichenauer, & Ebert, 2009).

129 Radiation thermal conductivity λ_r depends on the optical thickness of the porous material. For
130 optically thick aerogels, radiation thermal conductivity refers to the heat transfer due to the diffusion of

131 photons and can be described as follows:

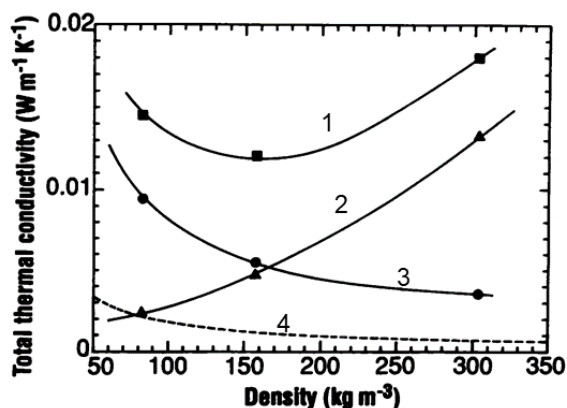
$$132 \quad \lambda_r = \frac{16\sigma n^2 T^3}{3\rho e} \quad (4)$$

133 where n is the effective refractive index of the porous material, σ is Stefan-Boltzmann constant
134 ($5.67 \times 10^{-8} \text{ W} \cdot \text{m}^{-2} \cdot \text{K}^{-4}$), T is the temperature, and e is specific extinction coefficient which depends on
135 the chemical composition and structure of the material and is inversely proportional to phonon mean
136 free path. Therefore, λ_r strongly increases with temperature and decreases with the density. To
137 decrease the radiation thermal conductivity, so-called infra-red opacifiers are sometimes added in a
138 small quantity (in order not to increase aerogel density): they are highly absorbing or scattering particles,
139 such as carbon black or titanium dioxide. In general, in optically thick porous materials the photon mean
140 free path is very small compared to the thickness of the sample, and at not elevated temperature the
141 radiation thermal conductivity is usually low.

142 Summarizing, in order to obtain a porous material with the lowest possible thermal conductivity (in
143 particular, a thermal superinsulating material), a compromise between the density (to get low λ_s) and
144 pore size (to get low λ_g) is needed. Each input is counterbalancing the other, as shown in Figure 2. The
145 minimum on this U-type curve is the ultimate value which is usually searched when making a thermal
146 insulating and superinsulating material. Low solid conductivity can be provided by low density; for a
147 porous material it is usually reached by decreasing the concentration of the matter which makes the
148 network. However, this usually increases the size of the pores. To have pores' dimensions below 70 nm
149 (mean free pass of air molecule) in a material filled with air in ambient conditions is the prerequisite for
150 having λ_g lower than that of free air. This is the condition to obtain a superinsulating material. To make
151 a low-density material with pores of such low dimensions is not straightforward, and thus most of the
152 best thermal insulating foams possess thermal conductivity above that of air, around 0.030 – 0.035

153 $W/(m \cdot K)$. The main reason is too large pores, much above 70 nm. Numerous process parameters, such
154 as matter type and concentration, mechanisms of network formation, way of solvent exchange (if
155 applicable) and gel drying influence porous materials' morphology in which porosity, pore size and pore
156 wall thickness are the key features influencing thermal conductivity. As explained in the next section, till
157 now there is no standard method for measuring pore size distribution in bio-aerogels; thus, other indirect
158 characteristics are needed to judge on bio-aerogels' pore dimensions. One of the criteria reflecting a
159 large number of mesopores is high specific surface area. In the following, thermal conductivity of bio-
160 aerogels will be analysed and correlated with material density and specific surface area.

161



162

163

Figure 2.

164 Example of the thermal conductivity of resorcinol-formaldehyde aerogels: (1) total conductivity, (2)

165 solid, (3) gaseous and (4) radiative (calculated). From Lu, X., Arduini-Schuster, M. C., Kuhn, J.,

166 Nilsson, O., Fricke, J., & Pekala, R. W. (1992). Thermal conductivity of monolithic organic aerogels.

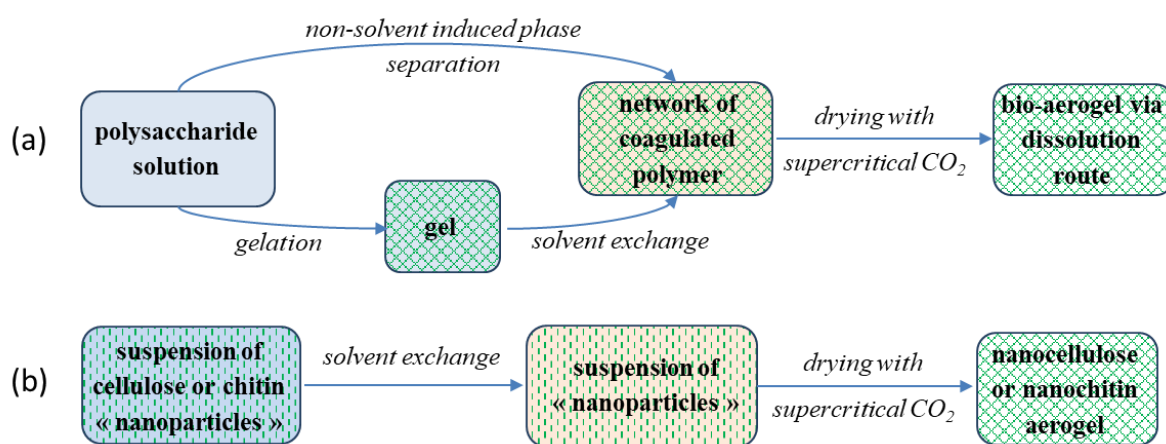
167 *Science*, 255, 971–972. Reprinted with permission from AAAS

168

169 **3. Preparation and characterization of aerogels, cryogels and xerogels based on**

170 **polysaccharides**

171 Bio-aerogels are prepared by polymer dissolution (except the case of nanocellulose and nanochitin),
 172 solution gelation (in some cases this step can be omitted), and drying which preserves network
 173 morphology, usually with supercritical CO₂ (Figure 3). Opposite to dissolved polysaccharides (Figure
 174 3a), nanocellulose and nanochitin make suspensions of nanofibrils or nanocrystals (Figure 3b), usually
 175 dispersed in water. In all the cases, the preparation of bio-aerogels is fundamentally different from the
 176 synthesis of silica- or synthetic polymer-based aerogels which is starting with the polymerization of
 177 monomers. As most of polysaccharide solvents are not miscible with CO₂, solvent exchange step is
 178 needed. Ethanol or acetone are used for this purpose, and in the majority of cases they are
 179 polysaccharides' anti-solvents. Two different mechanisms of structure formation in aerogel precursor
 180 before drying are possible: direct non-solvent induced phase separation if solution was not gelled, or
 181 gelation (Figure 3a). Opposite to many synthetic polymer-based gels that are collapsing in a non-solvent,
 182 polysaccharide solutions and gels are shrinking to a certain extent during solvent exchange, but they
 183 still keep a 3D shape and porosity.



184
 185 Figure 3.

186 Schematic presentation of the preparation of bio-aerogels: from polysaccharide solution (a) and
 187 from suspension (b).
 188

189 Drying is a challenging final step in the preparation of bio-aerogels as it should preserve network
190 morphology and avoid pores' collapse. During drying, a meniscus of the evaporating fluid is formed
191 within the pores of the network and capillary pressure P develops; it depends on the surface tension
192 γ of the evaporating liquid, angle θ between the liquid forming meniscus and the pore wall and pore
193 radius r :

$$194 \quad P = \frac{2\gamma\cos\theta}{r} \quad (5)$$

195 Drying in supercritical conditions is, till now, the best way to preserve gel morphology as no meniscus
196 is formed and the capillary pressure is theoretically zero. CO₂ in the supercritical state is the most often
197 used substance thanks to its mild critical temperature (31.3 °C) and pressure (72.9 atm).

198 Another way to avoid capillary pressure is to freeze the liquid in the pores of the network and then
199 perform sublimation. The conventional procedure is freezing water in the pores of a gel; freezing usually
200 results in very large macropores that are replicas of ice crystals. Two main approaches are used,
201 random freezing resulting in no preferential pores orientation, and ice-templating resulting in the
202 formation of elongated pores and channels (Figure S1 in the Supporting Information). As it will be shown
203 in the following, freeze-drying is widely used to produce nanocellulose-based cryogels (or foams). If
204 willing to preserve gel morphology, the kinetics of ice crystals growth must be controlled which can be
205 done by either spray-freeze-drying which allows fast freezing resulting in sub-micron size pores, or by
206 using mixed solvents.

207 Ambient pressure or low vacuum evaporative drying usually does not preserve gel morphology due
208 to the high capillary pressure (eq. 5); xerogels with rather high density and low specific surface area
209 are formed. In rare cases, polysaccharides with certain meso- and macroporosity can be obtained if the
210 liquid in the network pores is with low surface tension; in this case solvent exchange step(s) are needed.

211 Bio-aerogels are characterized using the same methods as for inorganic and synthetic polymer
212 aerogels, but certain precautions must be taken in order not to obtain the artefacts. The first problem is
213 high hydrophilicity of native polysaccharides. Adsorption of water vapours may influence all
214 characteristics, from density to specific surface area and also mechanical and thermal properties.
215 Subsequent drying of a bio-aerogel which was exposed to humidity may not help as evaporative drying
216 usually leads to the irreversible pores' closure; for cellulose this phenomenon is called "hornification".
217 "Post-drying" may lead to aerogel shrinkage and change of density, morphology and, as a consequence,
218 of all other properties. The conditions of samples' storage are thus very important but, unfortunately,
219 are rarely reported. The aging of bio-aerogels is practically not studied.

220 Bulk (or apparent) density ρ_{bulk} is usually determined by measuring sample weight and volume, the
221 latter is often calculated from sample's dimensions. For samples with geometrically complex shapes a
222 powder densitometer, such as Geopyc from Micromeritics with DryFlo powder, is a useful option
223 (Demilecamps, Beauger, Hildenbrand, Rigacci, & Budtova, 2015; Rudaz et al., 2014). Skeletal density
224 ρ_{skeletal} of polysaccharides is known to be 1.5 – 1.7 g/cm³.

225 High resolution scanning electron microscopy (SEM) is a perfect tool to visualize aerogel
226 morphology, but cannot be used to quantitatively describe it. Specific surface area S_{BET} , obtained using
227 nitrogen adsorption technique and Brunauer–Emmett–Teller (BET) theory, is one of the main
228 parameters characterizing aerogel texture. Unfortunately, standard methods used to obtain pore
229 volume and size distribution (Barrett-Joyner-Halenda (BJH) approach via nitrogen adsorption or
230 mercury porosimetry) cannot be applied to bio-aerogels. Bio-aerogels often possess large macropores
231 (several hundreds of nanometers up to several microns). BJH method considers mainly mesopores and
232 small macropores (below 200 nm), which makes only 10–20% of the total pore volume in bio-aerogels

233 (Robitzer, Di Renzo, & Quignard, 2011; Rudaz et al., 2014; Jiménez-Saelices, Seantier, Cathala, &
234 Grohens, 2017; Groult & Budtova, 2018a). If mercury porosimetry is used, the change of bio-aerogel
235 volume is often due to a compression and not mercury penetration in the pores (Rudaz, 2013; Rudaz
236 et al. 2014). Imaging, such as SEM or 3D tomography, needs to “see” the mesopores and be able to
237 analyse pores’ complex shapes.

238 Till now, the mechanical properties of bio-aerogels have been tested under the uniaxial
239 compression. Bio-aerogels can usually be compressed without breakage till 70-80% strain, in the
240 majority of cases no shape recovery occurs. The linear window is very small, less than few per cent
241 strain, and the compressive modulus is usually approximated by power-law of density.

242 Finally, thermal conductivity is often measured using either hot-wire or hot-plate methods and
243 custom-made set-ups adapted for small samples. In the hot-wire method, a wire is placed between two
244 disks of the material under investigation. The temperature is raised and measured at a certain distance
245 from the wire as a function of time. The thermal conductivity is deduced from the temperature profile
246 which is fitted by a known analytical dependence. In the hot-plate method the sample is placed between
247 two plates; one of them is heated. The temperature of the other plate is recorded; temperature
248 difference depends on the thermal conductivity of the material. In both methods, especially for materials
249 with thermal conductivity comparable with that of air, the errors due to the “imperfections” in the set-up
250 may become critical and lead to wrong results. Calibration with the known thermal superinsulating
251 standard materials is a *sine qua non* condition for obtaining adequate results.

252

253 **4. Case studies**

254 **4.1 Cellulose aerogels**

255 Cellulose is the most abundant natural polymer that can be widely found in wood, plants, marine
256 animals and bacteria. Cellulose macromolecule is built of repeated β -1,4-linked anhydro-D-glucose
257 units, and the chains are organized in amorphous and crystalline sequences. The crystalline phase has
258 several polymorphs (cellulose I, II, III_i, III_{ii}, IV_i, and IV_{ii}) which reflect chains' packing; the most common
259 are cellulose I and II. Cellulose I corresponds to native cellulose with chains packed in the parallel
260 direction; this is determined by the biosynthesis. Cellulose II is produced by dissolving native cellulose
261 followed by coagulation (or regeneration) or by swelling native cellulose in a concentrated alkali and
262 then removing the swelling agent; the chains of cellulose II are packed in the anti-parallel direction
263 which is thermodynamically more stable. Till now only cellulose I and II have been used to make
264 aerogels.

265

266 **4.1.1 Cellulose I aerogels**

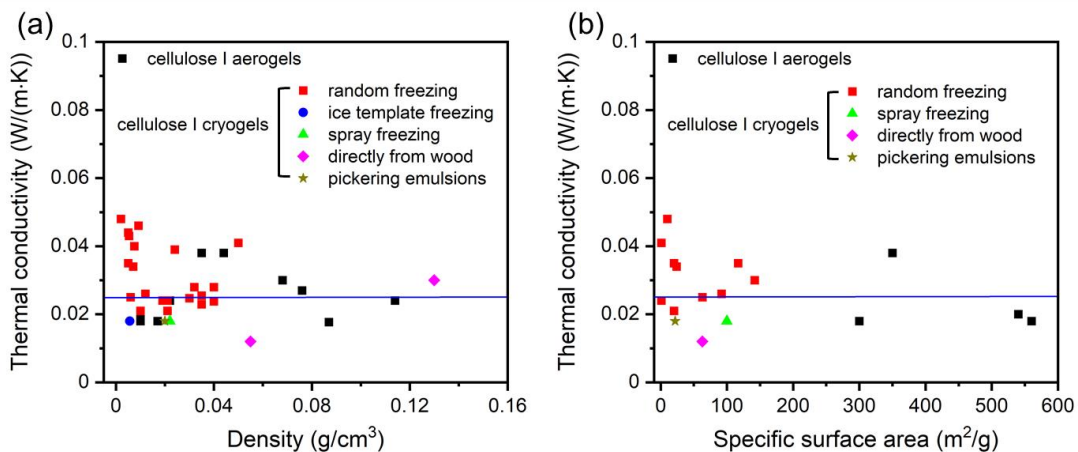
267 Nanocellulose (cellulose nanofibrils (CNF), bacterial cellulose (BC) and cellulose nanocrystals
268 (CNC)) is the most used cellulose type for making aerogels and foams. Excellent reviews on various
269 properties of porous nanocellulose materials have been published recently (Lavoine & Bergström 2017;
270 De France et al., 2017), and a recent review discusses nanocellulose materials for thermal applications
271 (Ahankari, Paliwal, Subhedar, & Kargarzadeh, 2021). Here we focus only on the works that report
272 thermal conductivity of nanocellulose-based aerogels. Nanocellulose was also mixed with other
273 substances; these materials will be called "composite aerogels". Porous materials with low thermal
274 conductivity, based on freeze-dried nanocellulose, neat or in composites, will also be considered.

275

276 **4.1.1.1 Neat cellulose I aerogels**

277

278 One of the first neat CNF aerogels with thermal conductivity below that of air were reported in 2014:
279 CNFs were from surface carboxylated softwood bleached kraft pulp (pretreated with 2,2,6,6-
280 tetramethylpiperidine-1-oxyl (TEMPO) as the catalyst) dispersed in dilute acid resulting in stiff free-
281 standing hydrogels (Kobayashi, Saito, & Isogai, 2014). The density of aerogels obtained under
282 supercritical drying was 0.01 – 0.02 g/m³ and specific surface area around 500 – 600 m²/g indicating a
283 highly mesoporous material. The lowest value of thermal conductivity was around 0.018 W/(m·K) at
284 aerogel density 0.017 g/cm³ (Figure 4). The homogeneous morphology with pore sizes below 100 nm
285 (as deduced from SEM, Figure 5a) shows that the obtained aerogel “fits” the characteristics of
286 superinsulating materials, recall Section 2. Slightly higher conductivity, 0.022 W/(m·K) (the lowest value),
287 was reported later by the same group, also for TEMPO-oxidized CNF aerogels (Sakai, Kobayashi, Saito,
288 & Isogai, 2016); in this work thermal conductivity was calculated as a product of density, thermal
289 diffusivity and specific heat capacity.



290

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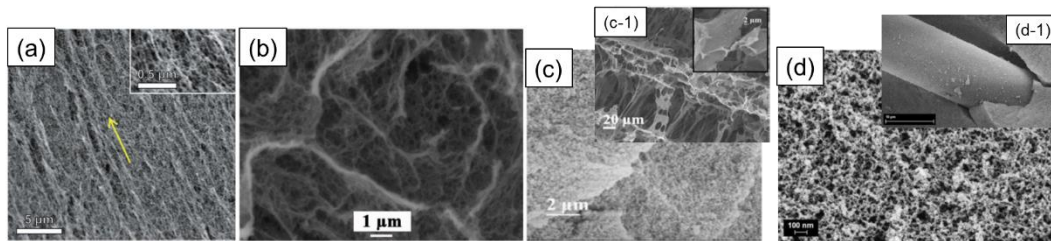
Figure 4.

292 Thermal conductivity of neat nanocellulose aero- and cryogels as a function of density (a) and specific

293 surface area (b). Blue solid line corresponds to the thermal conductivity of air, 0.025 W/(m·K). Data

294 are taken from the following references: Fan, Chen, Yao, Sun, & jin, 2017; Ge et al., 2018; Guo L.,
295 Chen, Lyu, Fu, & Wang, 2018; Guo W. et al., 2018; Gupta, Verma, & Maji, 2018; Jiménez-Saelices et
296 al., 2017, Jiménez-Saelices et al., 2018; Kobayashi et al., 2014; Lazzari, Perondi, Zampieri, Zattera,
297 & Santana, 2019; Li et al., 2018; Munier, Apostolopoulou-Kalkavoura, Persson, Bergström, 2020;
298 Plappert, Nedelec, Rennhofer, Lichtenegger, & Liebner, 2017; Qi et al., 2019; Sakai, Kobayashi,
299 Saito, & Isogai, 2016; Seantier, Bendahou, Bendahou, Grohens, & Kaddami, 2016; Song et al., 2018;
300 Zhao, et al., 2015.

301



302

303

Figure 5.

304

SEM images of

305

(a) TEMPO-oxidized CNF aerogel, from Kobayashi, Y., Saito, T., & Isogai, A. (2014). Aerogels with

306

3D ordered nanofiber skeletons of liquid-crystalline nanocellulose derivatives as tough and

307

transparent insulators. *Angewandte Chemie - International Edition*, 53, 10394–10397, reprinted with

308

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309

(b) spray freeze-dried TEMPO-oxidized CNF, reprinted from *Carbohydrate Polymers*, Vol 157,

310

Jiménez-Saelices, C., Seantier, B., Cathala, B., Grohens, Y., Spray freeze-dried nanofibrillated

311

cellulose aerogels with thermal superinsulating properties, Pages 105-113, Copyright 2017, with

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permission from Elsevier,

313 (c) composite freeze-dried methyltrimethoxysilane functionalised CNF-hydrophobised silica
314 aerogels (inset: (c-1) silylated CNF cryogels), from Zhao, S., Zhang, Z., Sèbe, G., Wu, R., Rivera
315 Virtudazo, R. V., Tingaut, P., & Koebel, M. M. (2015). Multiscale assembly of superinsulating silica
316 aerogels within silylated nanocellulosic scaffolds: Improved mechanical properties promoted by
317 nanoscale chemical compatibilization. *Advanced Functional Materials*, 25, 2326–2334, reprinted with
318 permission from John Wiley and Sons;

319 (d) hydrophobic silica aerogel-like composite reinforced with man-made fibers Tencel® (inset: (d-1)
320 Tencel® fiber within silica matrix), reprinted by permission from Springer, *J Mater Sci* 52:2210–2221,
321 Ambient-dried thermal superinsulating monolithic silica-based aerogels with short cellulosic fibers,
322 Markevicius, G., Ladj, R., Niemeyer, P., Budtova, T., & Rigacci, A, Copyright 2017

323
324 An original and intelligent approach for making superinsulating CNF aerogels was described in the
325 work of Plappert et al: in order to decrease pore dimensions and reach Knudsen regime, low-density
326 CNF aerogels were compressed (Plappert, Nedelec, Rennhofer, Lichtenegger, & Liebner, 2017). CNF
327 was obtained by mechanically fibrillating periodate/chlorite pre-treated cellulose resulting in 2,3-
328 dicarboxyl CNF. Aerogel compression to 20% of sample initial height allowed obtaining thermal
329 conductivity of around 0.017 W/(m·K); the initial aerogel density was around 0.02 – 0.025 g/cm³ and
330 after compression 0.087 g/cm³. This is a direct illustration of the importance of Knudsen effect and of
331 the compromise between not too low density and not too large pore dimensions.

332 Freeze-drying still remains the main drying method used to make porous nanocellulose materials,
333 most probably because lyophilization devices are much cheaper, easy to use and widely available
334 opposite to high-pressure supercritical drying reactors. If using classical freeze-drying from water, pore

335 dimensions are usually from several microns to tens or even hundreds of microns as pores are the
336 replicas of ice crystals. Knudsen effect is not working in this case (see Section 2), and despite ultra-low
337 density, it is difficult to expect thermal conductivity below that of air. Various attempts were made to
338 decrease pores' dimensions. Jiménez-Saelices et al. reported that cryogels prepared by spray freeze
339 drying method possess larger fraction of mesopores (Figure 4b) as compared to the conventional
340 freeze-drying, with specific surface area of 80-100 m²/g vs around 1 m²/g, respectively (Jiménez-
341 Saelices, Seantier, Cathala, & Grohens, 2017). As a result, thermal conductivity of spray freeze-dried
342 CNF was around 0.018 -0.021 W/(m·K) vs 0.024-0.028 W/(m·K) for conventionally freeze-dried CNF,
343 the density of both types of materials was within the same interval. Jiménez-Saelices reported another
344 method to obtain thermal superinsulating CNF cryogels (Jiménez-Saelices, Seantier, Grohens, &
345 Capron 2018). They were made from freeze-dried Pickering emulsions stabilized with TEMPO-oxidized
346 CNF; cryogels with ultra-low density (0.012 – 0.03 g/cm³), rather low specific surface area (15 – 65
347 m²/g) and thermal conductivity 0.019 – 0.023 W/(m·K) were made. In this case a U-shaped conductivity-
348 density dependence was obtained, recall Figure 2. The authors hypothesized that so low thermal
349 conductivity of cryogels with not high surface area (thus not highly mesoporous) is due to closed
350 porosity which decreases radiative heat transfer.

351 Another way of decreasing the size of the pores was suggested by Seantier et al (Seantier,
352 Bendahou, Bendahou, Grohens, & Kaddami, 2016): the dispersed in water bleached cellulose fibers
353 were mixed with CNF or CNC made from the same source and the whole lyophilized. The added
354 nanocellulose was “filling the space” between cellulose macro-fibers leading to the decrease in thermal
355 conductivity from 0.028 (no nanocellulose) to 0.023 W/(m·K) (best value, 10-20% of CNF). Specific
356 surface area was within 110 – 160 m²/g. A U-shape curves of conductivity vs density were obtained with

357 the lowest thermal conductivity at density around 0.03 - 0.04 g/cm³ (see example in Figure S2).

358 Surprisingly, several studies report thermal conductivity below that of air even of highly macroporous
359 freeze-dried CNF. For example, the conductivity of CNF cryogels with specific surface area around 20
360 m²/g was 0.019 – 0.021 W/(m·K) (the lowest values) which was even lower than those of aerogels
361 made from the same CNF with specific surface area around 300 – 350 m²/g and having similar solid
362 volume fraction (Sakai et al., 2016). Chen et al. reported CNF-based cryogels with extremely low density,
363 0.005 g/cm³, and thermal conductivity around 0.014 – 0.016 W/(m·K) (Chen et al., 2014). Specific
364 surface area was not reported, SEM images of some samples show pores of dimensions around 1 μm.

365 Varying the mode of freeze-drying, random or with the preferential direction of ice crystal growth
366 (see Figure S1), allows making nanocellulose cryogels with anisotropic thermal properties. For example,
367 CNF-based cryogel with anisotropic thermal conductivity was reported by Li et al. (2018). So-called
368 “nanowood” was made by the treatment of wood with NaOH and Na₂SO₃ heated to boiling temperatures,
369 followed by subsequent treatment with H₂O₂ to remove the lignin and most of the hemicellulose, and
370 freeze-dried. A material with density of 0.13 g/cm³ and tubular structure was obtained, the diameter of
371 tubes was several tens of microns. Thermal conductivity in the radial direction was 0.032 W/(m·K) and
372 in the axial direction 0.056 W/(m·K). The same approach was reported by Song et al. (2018): the
373 conductivity of “nanowood” in the radial direction was 0.028 W/(m·K) and 0.12 W/(m·K) in the axial
374 direction.

375 The majority of nanocellulose-based cryogels, with pore size from few microns to several hundreds
376 of microns, have thermal conductivity higher than that of air, as expected (Figure 4). The conductivity
377 mainly depends on cryogel density. For example, Qi et al. (2019) made cryogels based on CNF from
378 cotton, poplar wood, bamboo and rice straw. At the same density, the samples had similar conductivity,

379 0.040 – 0.046 W/(m·K). For cotton-based CNF cryogels the conductivity decreased from around 0.049
380 to 0.028 W/(m·K) with increase the porosity from around 97.7% to 99.4%, respectively. The values of
381 the thermal conductivity as a function of density and specific surface area of different nanocellulose
382 cryogels are shown in Figure 4a,b.

383

384 **4.1.1.2 Composite cellulose I aerogels**

385 One of the ways to decrease the thermal conductivity of a porous material is to fill the pores with a
386 material of conductivity below that of air but without a significant increase in the whole bulk density.
387 Silica aerogels with the conductivity around 0.013 – 0.016 W/(m·K) are the best candidates for this
388 purpose. It should also be taken into account that cellulose is hydrophilic and adsorbs water vapours.
389 This increases thermal conductivity, as expected, and as demonstrated by Apostolopoulou-Kalkavoura
390 et al : the conductivity of oven-dried foamed CNF increased from ~ 0.048 W/(m·K) to 0.085 W/(m·K)
391 with relative humidity (RH) increase from 20 to 80 %, respectively (Apostolopoulou-Kalkavoura,
392 Gordeyeva, Lavoine, Bergström, 2018). Surprisingly, the opposite trend was reported by the same
393 group: thermal conductivity in the radial direction of unidirectionally freeze-dried CNF decreased with
394 the increase of the relative humidity from 0.03 to 0.019 W/(m·K) for RH from 1 to 65 %, respectively,
395 despite the increase in moisture uptake (Munier, Apostolopoulou-Kalkavoura, Persson, Bergström,
396 2020). The authors do not provide an explanation of this anomalous behavior. To avoid water vapours'
397 adsorption, cellulose hydrophilization (often silylation) is performed. Below various composite aerogels
398 based on nanocellulose mixed with other components and/or functionalized, are discussed.

399 A somewhat complex but efficient strategy to obtain thermal superinsulating composites was
400 suggested by Zhao et al., 2015. CNF was hydrophobized with methyltrimethoxysilane (MTMS), freeze-

401 dried, filled with tetraethoxysilane (TEOS)-based sol, the latter also hydrophobized and the whole dried
402 with supercritical CO₂. While both non-hydrophobised and silylated CNF cryogels had very large
403 macropores (inset of Figure 5c), negligibly low specific surface area and thus thermal conductivity
404 above that of air (0.031 – 0.039 W/(m·K)), hydrophobic composites with silica aerogels in the pores of
405 CNF possessed conductivity 0.014 – 0.018 W/(m·K) (Zhao et al., 2015). Thanks to mesoporous silica
406 aerogel (Figure 5c), specific surface area of composite aerogels was high, 450 – 630 m²/g. Higher was
407 the fraction of CNF, lower was the surface area and higher thermal conductivity. The mechanical
408 properties of composite aerogels under compression were improved as compared to the neat silica
409 aerogels. It should be noted that CNF and two dryings, lyophilization and supercritical, are not needed
410 to obtain strong hydrophobic silica aerogel-like materials with thermal superinsulation properties:
411 natural or man-made short cellulose fibers were dispersed in the prepolymerised TEOS, the latter
412 hydrophobized with hexamethyldisilazane and the whole dried at ambient pressure and 140 °C (Figure
413 5d) (Jaxel, Markevicius, Rigacci, & Budtova, 2017; Markevicius, Ladj, Niemeyer, Budtova, & Rigacci,
414 2017). The thermal conductivity of these composites was 0.016 – 0.018 W/(m·K), specific surface area
415 from 570 to 730 m²/g, and mechanical properties under 3-point bending were significantly improved as
416 compared to silica aerogels.

417 Some composite cryogels based on CNF with thermal superinsulating properties were reported
418 despite very large macropores of the materials. For example, CNF/graphene oxide/zirconium
419 phosphate cryogels with low specific surface area (around 20 m²/g), density 0.009 g/cm³ and pores
420 with the size of several tens of microns were reported to possess thermal conductivity of around 0.018
421 W/(m·K) (Wang D. et al., 2020). Similar values were reported for CNF/graphene oxide/sepiolite
422 nanorods and CNF/graphene oxide/sepiolite nanorods/boric acid cryogels made via unidirectional ice-

423 templating (Figure S1) (Wicklein et al., 2015). The composites had tubular morphology with tubes'
424 diameter of few tens of microns, and thermal conductivity in the radial direction was surprisingly low,
425 0.016 – 0.018 W/(m·K). The density was around 0.007 g/cm³ and specific surface area was not reported.
426 Lazzari et al. produced nanocellulose-biochar cryogels with density around 0.01 – 0.03 g/cm³, high
427 macroporosity as deduced from SEM (no specific surface area reported) and conductivity 0.021 – 0.026
428 W/(m·K) (Lazzari et al., 2019). Unfortunately, all these authors were not explaining how it was possible
429 to overcome Knudsen effect.

430 Zhang X. et al. (2020) compared the influence of different freeze-drying methods (random freezing,
431 unidirectional freezing, and bidirectional freezing) on the structure and thermal conductivity of
432 polyimide/bacterial cellulose cryogels. Bidirectional freezing corresponds to a setup inducing two
433 temperature gradients during freezing resulting in aligned lamellar structures. Cryogels prepared by
434 random freezing had isotropic thermal conductivity 0.042 W/(m·K). Cryogels obtained with
435 unidirectional freezing (tubular structure) possess thermal conductivity 0.037 W/(m·K) in the radial and
436 0.066 W/(m·K) in the axial directions. The thermal conductivity of bidirectional cryogels with lamellar
437 morphology was 0.023 and 0.044 W/(m·K) in the radial and axial direction, respectively. The density of
438 bidirectional cryogels was 0.05 – 0.1 g/cm³, specific surface area was not reported; SEM images
439 showed the distance between lamellas around 10 μm and pore walls of 1 to 2 μm. The authors interpret
440 the lowest thermal conductivity of bidirectional cryogels in the radial direction as follows: “the parallel
441 lamellar structure of b-PI/BC (here, bidirectional) aerogel would greatly reduce the heat transfer in the
442 radial direction since the absence of interlamellar heat conduction” (Zhang X. et al., 2020).

443 The majority of nanocellulose-based composite cryogels reported in literature have low thermal
444 conductivity but higher than that of air, as expected from material morphology with very large

445 macropores. A non-cellulose component is often used to increase hydrophobicity and/or improve fire
446 retardant properties by adding various substances, but these modifications usually increase the thermal
447 conductivity. For example, hydrophobised CNC-polyurethane cryogels possessed long-lasting
448 superhydrophobicity, anti-icing properties and oil absorbency efficiency, but conductivity was high,
449 0.075 W/(m·K) (Cai et al., 2020). Flame-retarding modification of CNF-based cryogels increased
450 thermal conductivity from 0.026-0.027 for neat CNF cryogel to 0.039 W/(m·K) for CNF-sepiolite (Gupta,
451 Verma, & Majie, 2019), to 0.032 W/(m·K) for CNF-N-methylol dimethylphosphonopropionamide/1,2,3,4-
452 butanetetracarboxylic acid (Guo L., Chen, Lyu, Fu, & Wang, 2018) and to 0.028 W/(m·K) for CNF-
453 sodium bicarbonate cryogels (Farooq, Sipponen, Sepplälä, & Österberg, 2018). Conductivity around
454 0.040 W/(m·K) were reported for CNF-hydroxyapatite (Guo W. et al. 2018) and CNF-aluminium based
455 metal organic frameworks cryogels (Zhou S. et al., 2020). The details on the conductivity and materials'
456 properties of various nanocellulose-based composite aerogels and cryogels are presented in Table 1.
457 As in the case of the reinforcement of silica aerogels with cellulose fibers and keeping thermal
458 superinsulating properties (Jaxel et al., 2017; Markevicius et al., 2017), nanocellulose is not a pre-
459 requisite for obtaining lightweight fire retardant and thermal insulating cellulose-based materials: for
460 example, wood fibers/palygorskite cryogels possess conductivity of 0.033 – 0.047 W/(m·K) at density
461 0.033 – 0.072 g/cm³, respectively (Zhou X. et al., 2020)..

Table 1. Composite aerogels and cryogels involving cellulose I.

Material	Additional component	Density (g/cm ³)	Thermal conductivity (W/(m·K))	Specific surface area (m ² /g)	Young's modulus (MPa)	Water contact angle (°)	Drying method	Reference
CNF	nanozeolites		0.018	100	0.11	-	freeze-drying	Bendahou, Bendahou, Seantier, Grohens, & Kaddami, 2015
CNF	graphene oxide and sepiolite nanorods	0.0075	0.015	-	0.54	-	freeze-drying	Wicklein et al., 2015
	-	0.0056	0.018	-	0.08	-		
CNF	hydrophobized with MTMS	0.01-0.03	0.031-0.036	13-2	-	-	freeze-drying	Zhao et al., 2015
	without hydrophobized	0.007-0.018	0.034-0.039	24-9	-	-		
	neat CNF cryogels mixed with prepolymerised TEOS (polyethoxydisiloxane)	0.130-0.146	0.014-0.018	631-454	2.3-2.8	-	supercritical CO ₂ drying	
	cryogels made from MTMS hydrophobized CNF mixed with PEDS-P75E20 (prepolymerized form of TEOS	0.128-0.133	0.018-0.020	756-675	1.8-1.3	-		

	containing a water-to-TEOS molar ratio of 1.5 and a SiO ₂ content of 20% w/w in ethanol)							
CNF	-	0.055	-	11	5.93	-	ambient	Fu et al., 2016
	tetraethylorthosilicate	0.059-0.295	0.023	14-700	5.42	-	pressure drying	
bleached cellulose fibers (palm tree)	-	0.04	0.028	-	0.013	-	freeze-drying	Seantier et al., 2016
	cellulose nanofibrils	0.04	0.023-0.028	161	0.062	-		
	cellulose nanocrystals	0.04	0.025-0.028	142	0.062	-		
CNF	-	0.002	0.048	10	-	-	freeze-drying	Fan et al., 2017
	AIOOH	0.0025	0.039	67	-	-		
Tencel fibers	Prepolymerised TEOS (polyethoxydisiloxane) hydrophobized with methyltrimethoxysilane (MTMS)	0.1-0.13	0.017-0.018	730-570	-	133-139	ambient pressure drying	Markevicius, Ladj, Niemeyer, Budtova, & Rigacci, 2017
		0.1-0.12	0.016-0.017	730-570	-	136-142	supercritical CO ₂ drying	

CNF	triblock poly(ethylene glycol)-poly(propylene glycol)-poly (ethylene glycol) polyoxamer copolymer /Gluconic acid D-lactone/CaCO ₃	0.012	0.043	-	-	-	freeze-drying	Apostolopoulou-Kalkavoura et al.,2018
CNF	sodium bicarbonate	-	0.028	-	-	-	freeze-drying	Farooq et al., 2018
Carboxymethyl cellulose	-	0.024	0.039	-	0.23	-	freeze-drying	Ge et al., 2018
	2D graphene oxide/boric acid	0.026	0.042	-	1.03	-		
CNF	-	0.006	0.025	63	0.002	-	freeze-drying	Guo L. et al., 2018
	1,2,3,4-butanetetracarboxylic acid/N-methylol di-methylphosphonopropionamide	-	0.033	43	0.005	-		
CNF	-	0.050	0.041	1	0.17	-	freeze-drying	Guo W. et al., 2018
	hydroxyapatite	0.088	0.039	39	0.46	-		
CNF	-	0.012	0.026	180	-	-	freeze-drying	Gupta et al. , 2019
	sepiolite	0.024	0.035	341	-	-		
unbleached long fiber	-	0.01-0.019	0.021-0.024	-	-	-	freeze-drying	Lazzari, Perondi, Zampieri, Zattera, & Santana, 2019
	biochar	0.021-0.027	0.023-0.026	-	-	-		
CNF	-	0.006	0.025	12	0.04	-	unidirectional	Munier et al., 2020

	isotropic silica	0.019	0.021	94	0.15	-	freeze-drying	
	anisotropic silica	0.021	0.020	153	0.3	-		
CNF	-	0.004	0.049	10	0.08	-	freeze-drying	Wang D. et al., 2020
	hierarchical ZrP/RGO nanosheets	0.009	0.018	23	0.994	-		
bacterial cellulose	polyimide	0.046	0.042	-	-	-	freeze-drying	Zhang X. et al., 2020
			0.037 radial direction	-	-	-	unidirectional	
			0.06 axial direction	-	-	-	freezing (ice templating)	
			0.023 radial direction 0.044 axial direction	-	-	-	bidirectional freeze-drying	
CNF	-	-	0.043	103	-	-	freeze-drying	Zhou S. et al., 2020
	metal-organic frameworks	-	0.041	227	-	-		

'-' without additional component or the data are not reported.

464 **4.1.2 Cellulose II aerogels**

465 Cellulose II aerogels, as well as other man-made cellulose materials, can be obtained via two main
466 routes: either derivatization followed by cellulose regeneration or by dissolution in a direct solvent
467 followed by coagulation (or precipitation). The former method has been used very rarely: cellulose
468 aerogels were made from viscose (Ookuna et al., 1993) and from cellulose carbamate (Pinnow, Fink,
469 Fanter, & Kunze, 2008). Thermal conductivity was not measured in these two cases. Various direct
470 solvents have been used to make cellulose II aerogels (aqueous alkali, *N*-methylmorpholine-*N*-oxide
471 monohydrate (NMMO), ionic liquids, zinc chloride ($\text{ZnCl}_2 \cdot 6\text{H}_2\text{O}$), calcium thiocyanate ($\text{Ca}(\text{SCN})_2 \cdot 6\text{H}_2\text{O}$)
472 and LiCl/*N,N*-dimethylacetamide (LiCl/DMAc)) with only one, 8% NaOH-water, used for gelation of
473 cellulose solutions. The details on cellulose II aerogels preparation, properties and applications can be
474 found in a recent review (Budtova, 2019).

475

476 **4.1.2.1 Neat cellulose II aerogels**

477 Thermal conductivity of neat cellulose II aerogels as a function of density and specific surface area
478 is shown in Figure 6. The reader will notice much less experimental data on cellulose cryogels, as
479 compared to nanocellulose-based cryogels (Figure 4). What is even more important is that opposite to
480 nanocellulose aero- and cryogels, no value of thermal conductivity of neat cellulose II aero- and
481 cryogels falls in the thermal superinsulation region.

482

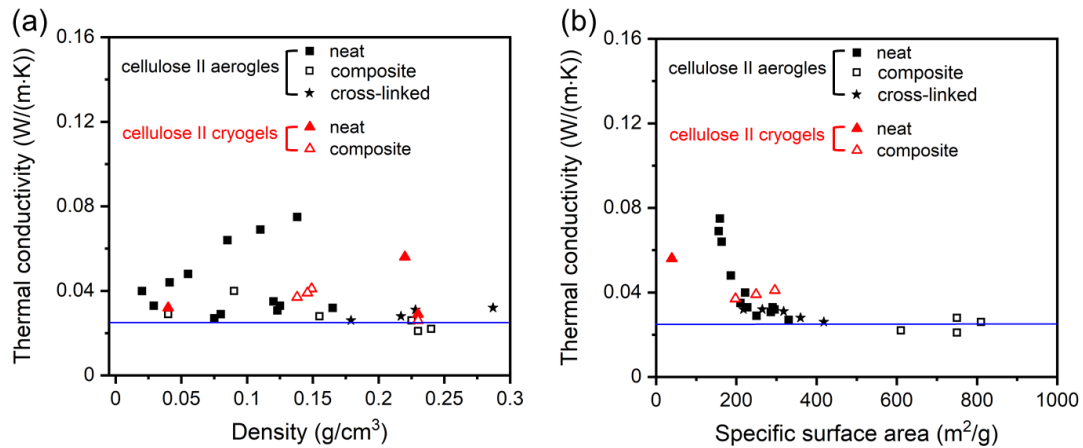


Figure 6

483

484

485 Thermal conductivity of neat (filled points) and composite (open points) cellulose II aerogels and
 486 cryogels as a function of density (a) and specific surface area (b). Blue solid line corresponds to the

487 thermal conductivity of air. Data are taken from references: Demilecamps et al., 2015; Demilecamps,

488 Alves, Rigacci, Reichenauer, & Budtova., 2016; Feng et al., 2016; Han, Zhang, Wu, & Lu, 2015;

489 Laskowski, Milow, & Ratke, 2015; Mi, Ma, Yu, He, & Zhang, 2016; Rudaz, 2013; Karadagli et al.,

490 2015; Shi, Lu, Guo, Sun, & Cao, 2013; Shi, Lu, Guo, Zhang, & Cao, 2013

491

492 Why cellulose II cryogels cannot be thermal superinsulating materials is clear from their morphology

493 (Figure 7a): the density is low (0.01 – 0.2 g/cm³) but the pores are large and the pore walls are thick

494 and not porous. Specific surface area is rather low, from maximum 50 m²/g for low molecular weight

495 cellulose to maximum 130 m²/g for high molecular weight cellulose (Buchtová & Budtova, 2016;

496 Buchtová, Pradille, Bouvard, & Budtova, 2019).

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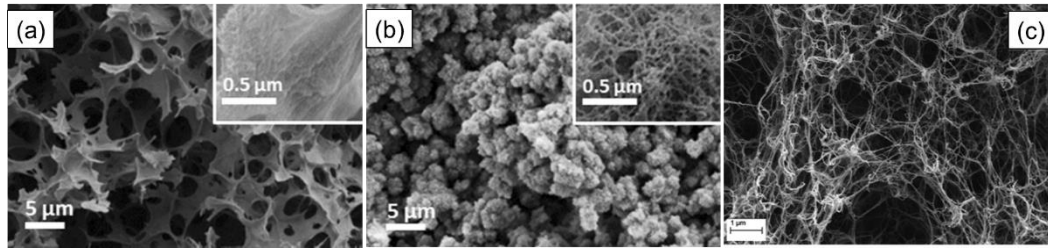


Figure 7

498

499

500

SEM images of cellulose II cryogel (a) and aerogel (b) both from microcrystalline cellulose

501

dissolved in 1-ethyl-3-methylimidazolium acetate/DMSO and coagulated in ethanol, reprinted by

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permission from Springer, Cellulose, Cellulose aero-, cryo- and xerogels: towards understanding of

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morphology control, Buchtova N., Budtova T., Copyright 2016, and

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(c) aerogel from cellulose dissolved in $\text{Ca}(\text{SCN})_2 \cdot 4\text{H}_2\text{O}$ and coagulated in ethanol, reprinted from

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J. of Supercritical Fluids, 106, Karadagli I., Schulz B., Schestakow M., Milow B., Gries T., Ratke L.,

506

Production of porous cellulose aerogel fibers by an extrusion process, 105-114., Copyright 2015, with

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509

Cellulose II aerogels possess a finer fibrous morphology as compared to cryogels but seem to have

510

too many large macropores (Figure 7b,c) and/or probably too thick pore walls, both being not favorable

511

for thermal superinsulating properties (eq. 1). This is different from nanocellulose-based aerogels,

512

compare aerogels' morphology shown in Figure 5a,b for nanocellulose-based aerogels and Figure 7

513

for cellulose II. In the view of decreasing pores' size and thus decreasing gaseous phase conductivity

514

(eq. 3), cellulose was cross-linked with epichlorohydrin (ECH) in 8% NaOH-water (Rudaz, 2014). This

515

allowed decreasing the conductivity from around 0.033 – 0.035 to 0.026 W/(m·K) (Figure 6). As

516

mentioned in Section 3, unfortunately there is no suitable method to determine pore size distribution in

517

bio-aerogels which is slowing down the understanding of their thermal conductivity properties.

518 As for the case of nanocellulose aerogels, efforts were made to decrease cellulose hydrophilicity
519 expecting this would decrease thermal conductivity. This indeed helped but still the “barrier” of air
520 conductivity was not overcome: thermal conductivity of tritylcellulose aerogels was 0.027 – 0.029
521 W/(m·K) (Demilecamps, Alves, Rigacci, Reichenauer, & Budtova, 2016), methyltrimethoxysilane
522 (MTMS) treatment of cellulose cryogel decreased the conductivity from 0.032 to 0.029 W/(m·K)
523 (Nguyen et al., 2014) and the same lowest value of conductivity, 0.029 W/(m·K), was obtained for
524 cellulose cryogels treated with cold plasma (Shi, Lu, Guo, Sun, & Cao, 2013).

525

526 **4.1.2.2 Composite cellulose II aerogels**

527 In order to decrease gas phase conductivity in cellulose II aerogels, silica aerogels were used to
528 replace air in the pores, as in the case of nanocellulose-based aerogels. To make an interpenetrated
529 cellulose II-silica network turned out to be less evident than dispersing nanocellulose in silica sol. An
530 attempt of using a common solvent of cellulose and sodium silicate, 8%NaOH-water, and one-pot
531 approach, was tested; however, cellulose coagulated in the presence of silica and silica formed particles
532 and not aerogels (Demilecamps, Reichenauer, Rigacci, Budtova, 2014). Mixing silica aerogel particles
533 with cellulose dissolved in calcium thiocyanate slightly reduced the thermal conductivity as compared
534 to neat cellulose aerogel but still remained high, 0.04 – 0.05 W/(m·K) (Laskowski, Milow, & Ratke, 2015).

535 Interpenetrated cellulose-silica networks with silica making aerogel phase in the pores of cellulose
536 skeleton can be formed when impregnating coagulated cellulose (before drying) with silica sol followed
537 by silica gelation and supercritical or freeze-drying. However, silica is very sensitive to pH and cellulose
538 may be ionized after being dissolved in aqueous alkali; surface charges on the walls of the pores of
539 cellulose network may perturb the formation of silica aerogels. This is what probably happened when

540 coagulated cellulose was impregnated with tetraethyl orthosilicate (TEOS): the conductivity surprisingly
541 increased from 0.025 W/(m·K) for the neat cellulose aerogel to 0.04–0.05 W/(m·K) for composites (Cai
542 et al., 2012). A possible reason could be high density of composite aerogel (0.3 – 0.6 g/cm³) as
543 compared to neat components (0.14 g/cm³ and 0.19 g/cm³ for cellulose and silica aerogels,
544 respectively). A decrease of the thermal conductivity was reported for composite cellulose-silica
545 aero/cryogels made via impregnation technique: from around 0.033 – 0.030 W/(m·K) for the neat
546 cellulose aerogel or cryogel to 0.026 W/(m·K) for the composites (Shi, Lu, Guo, Zhang, & Cao., 2013).

547 It seems that the only one case known in literature when impregnation approach resulted in thermal
548 conductivity of cellulose II-silica composite aerogel below that of air was when coagulated tritylcellulose
549 network was impregnated with polyethoxydisiloxane, the latter gelled, silica hydrophobised and the
550 whole dried with supercritical CO₂ (Demilecamps et al., 2016). Thermal conductivity was 0.021 – 0.022
551 W/(m·K) and SEM images confirm that silica phase completely fills the pores of cellulose matrix (Figure
552 S3) (Demilecamps et al., 2016).

553

554 **4.3 Chitin and chitosan aerogels**

555 Chitin is the second most abundant natural polysaccharide. It is a linear polymer composed of (1→
556 4)-β linked D-(N-acetyl)glucosamine. Chitin is similar to cellulose but with one hydroxyl group on each
557 AGU replaced by an acetyl amine group. Chitin is organized in microfibrils embedded in proteins and
558 associated with calcium salts. As for cellulose, aero- and cryogels based on chitin can be prepared
559 either from various nanochitin forms, nanocrystals or nanofibers, or via polymer dissolution (Figure 3).

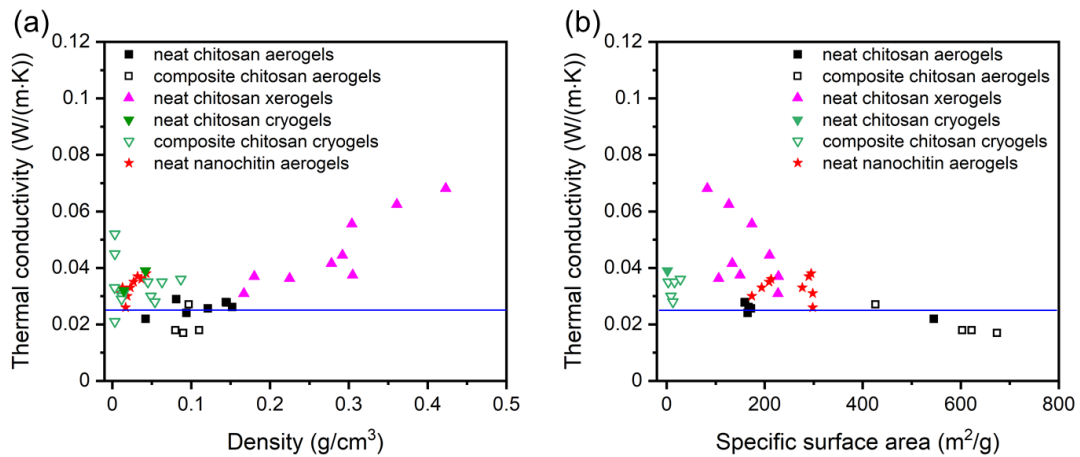
560 Very few publications report on the thermal conductivity of nanochitin aerogels: the lowest values
561 were obtained for the neat nanochitin aerogel (0.027 W/(m·K)), silylation of this aerogel using chemical

562 vapor deposition increased the contact angle with water but also increased the conductivity to 0.030 –
563 0.036 W/(m·K) (Yan et al., 2020). Because of some difficulties in chitin dissolution, only few articles
564 describe the production of chitin aerogels and, to the best of our knowledge, none report their thermal
565 conductivity. As cellulose, chitin was shown to be soluble in imidazolium-based ionic liquids (Silva, Mano,
566 & Reis, 2017), which has not been explored yet for testing chitin aerogels' thermal conductivity.

567 Chitosan is a cationic polysaccharide obtained from chitin by decreasing the degree of acetylation
568 below 50 wt%. Replacing acetyl with amino group allows chitosan easy dissolution in acidic media such
569 as hydrochloric, dilute nitric, acetic, formic and lactic acids; the most commonly used solvent is 1%
570 acetic acid at pH around 4. Aerogels and cryogels can then be produced in the ways similar to cellulose
571 II.

572 The majority of chitosan aero- and cryogels that were tested for thermal insulation were made from
573 gels based on crosslinked chitosan, either with formaldehyde (Takeshita & Yoda, 2015), or with
574 glutaraldehyde (Zhu et al., 2020), or with ECH (Tang et al., 2019), or gelled in the presence of urea
575 (Guerrero-Alburquerque et al., 2020) (Figure 8). The lowest values of thermal conductivity of neat
576 chitosan aerogels were reported by Takeshita et al.: their aerogels were not only with thermal
577 superinsulating properties (0.022 W/(m·K)), but also with high specific surface area (545 m²/g),
578 translucent and bendable (see aerogel morphology in Figure 9a) (Takeshita et al., 2015). Low values
579 of thermal conductivity of chitosan aerogels, around 0.024 – 0.025 W/(m·K), were obtained when
580 chitosan solutions were gelled in the presence of urea, but specific surface area was not high, 160 –
581 170 m²/g (Guerrero-Alburquerque et al., 2020). The authors also explored drying the same gels at
582 ambient pressure, at ambient temperature and at 65 °C (Figure 9b, c). In both cases thermal
583 conductivity was significantly increased, from 0.036 to 0.068 W/(m·K) for the gels dried in ambient

584 conditions and from 0.031 to 0.056 W/(m·K) for those dried at higher temperature; interestingly,
 585 xerogels made at elevated temperature had higher specific surface area, 170-230 m²/g, as compared
 586 to their aerogel counterparts (Guerrero-Alburquerque et al., 2020).
 587



588

589

Figure 8.

590 Thermal conductivity of nanochitin and chitosan aero-, cryo- and xerogels as a function of density (a)

591 and specific surface area (b). Blue solid line corresponds to the thermal conductivity of air. Data are

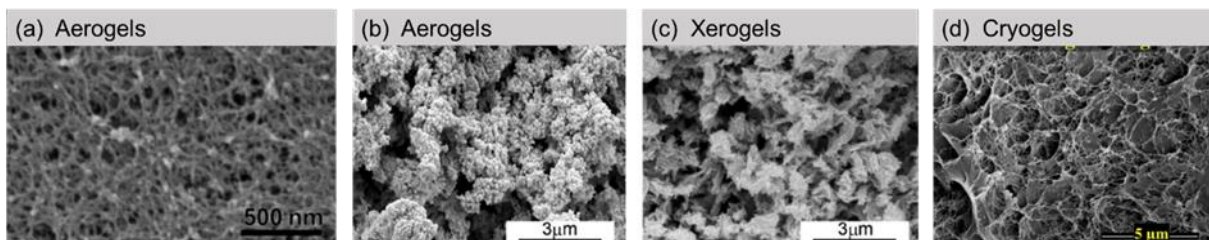
592 taken from references: (Guerrero-Alburquerque et al., 2020; Takeshita et al., 2015; Wang et al., 2017;

593 Zhang, Feng, Feng, Jiang, & Li, 2018; Zhao et al., 2016; Zhu et al., 2020; Tang et al., 2019; Yan et al.,

594

2020)

595



596

597

Figure 9

598

SEM images of chitosan aero-, cryo- and xerogels:

599 (a) formaldehyde-crosslinked chitosan aerogel, reprinted with permission from Takeshita, S., &
600 Yoda, S. Chitosan Aerogels: Transparent, Flexible Thermal Insulators. *Chemistry of Materials*, 27,
601 7569–7572. Copyright 2015 American Chemical Society;

602 (b) and (c) urea-crosslinked chitosan aerogel (b) and xerogel (c), reprinted with permission from
603 Guerrero-Alburquerque, N., Zhao, S., Adilien, N., Koebel, M. M., Lattuada, M., & Malfait, W. J. Strong,
604 Machinable, and Insulating Chitosan-Urea Aerogels: Toward Ambient Pressure Drying of Biopolymer
605 Aerogel Monoliths. *ACS Applied Materials and Interfaces*, 12, 22037–22049. Copyright 2020
606 American Chemical Society;

607 (d) composite cryogels based on esterified cellulose nanocrystals/agar/chitosan, reprinted by
608 permission from Springer Nature, Cellulose, Synthesis of light weight, high strength biomass-derived
609 composite aerogels with low thermal conductivities, Tang, L., Zhuang, S., Hong, B., Cai, Z., Chen, Y.,
610 & Huang, B. 26, 8699–8712, Copyright 2019.

611

612 Composite chitosan aerogels were prepared either with organic or inorganic co-network. Chitosan-
613 silica aerogels based on interpenetrated networks hydrophobized with hexamethyldisilazane were
614 made by co-gelation of chitosan and sodium silicate: higher chitosan concentration increased aerogel
615 density, improved compressive stress but slightly increased thermal conductivity from 0.0167 for the
616 reference silica aerogel to 0.017 - 0.018 W/(m·K) (Zhao et al., 2016). Mixing chitosan with polyvinyl
617 alcohol improved aerogel shape stability during processing and increased specific surface area from
618 308 to 426 m²/g for the neat chitosan to composite chitosan-PVA aerogel, respectively; thermal
619 conductivity of composite aerogel was 0.027 W/(m·K) (Zhang, Feng, Feng, Jiang, & Li, 2018).

620 Few publications report on the thermal conductivity of chitosan cryogels and its composites. Despite
621 rather large pores (Figure 9d), the lowest thermal conductivity, 0.021 W/(m·K), was obtained for ECH-
622 crosslinked chitosan/esterified cellulose nanocrystals/agar aerogels (Tang et al., 2019). Neat agar and
623 chitosan/agar aerogels had much higher conductivity, 0.057 and 0.033 W/(m·K), respectively (Tang et
624 al., 2019). Rather low conductivity was obtained for chitosan/NFC cryogels; it decreased from 0.032 for
625 the neat chitosan aerogel to 0.029 W/(m·K) in the presence of NFC (Wang et al., 2017). Ice templating
626 (recall Figure S1) was used to make anisotropic composites based on glutaraldehyde cross-linked
627 chitosan/amino modified hydroxyapatite nanocrystals (HAP) (Zhu et al., 2020). Channels with rather
628 flat pore walls were formed and specific surface area was low (from few m²/g to few tens of m²/g),
629 similar to cellulose cryogels. Thermal conductivity in the radial direction first decreased from 0.039 to
630 0.028 W/(m·K) for the neat chitosan to composite chitosan cryogel, respectively, and increased to
631 around 0.037 W/(m·K) with the increase of HAP concentration. On the opposite, in the axial direction
632 the conductivity of composite cryogel was higher than that of the neat one (0.043 vs 0.039 W/(m·K),
633 respectively (Zhu et al., 2020).

634

635 **4.4 Pectin aerogels**

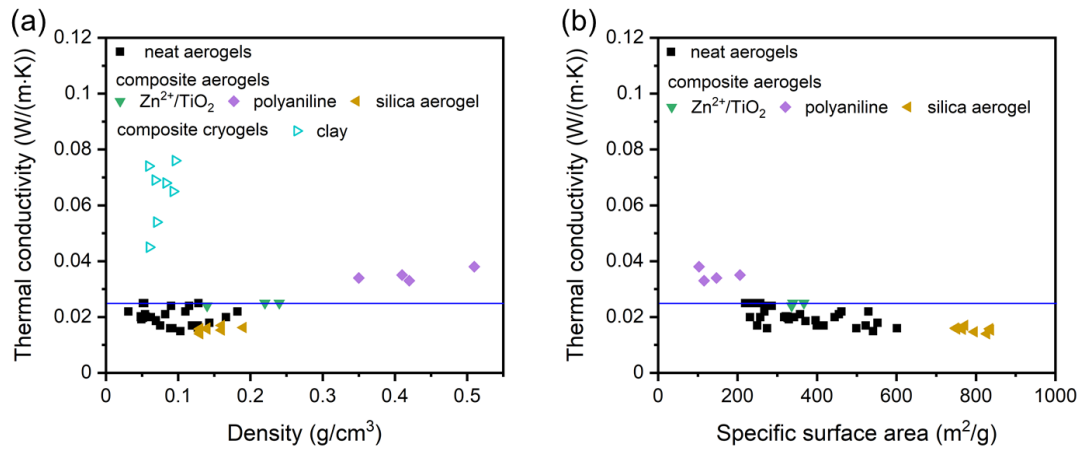
636 Pectin is a ramified heteropolysaccharide usually extracted from citrus peel or apple pomace; it is
637 widely used in food industry as a gelling and thickening agent. Pectin is a branched polymer with the
638 main chain consisting of α -1,4-linked D-galacturonic acids which can be partly methylated or amidated
639 at C6 position, or acetylated at O2 or O4 positions. Pectin degree of esterification (DE) defines the
640 amount of methyl-esterified galacturonic acid resulting in high-methylated pectin (HM) with DE > 50%
641 and low-methylated pectin (LM) with DE < 50%.

642 Opposite to cellulose and chitosan solutions, pectin solutions are “easy gelling”: esterified groups of
643 galacturonic acid are prone to hydrophobic interactions while non-esterified groups may interact via
644 hydrogen bonding if being in the protonated state or crosslink with polyvalent metal ions if deprotonated.
645 Depending on the intrinsic properties of pectin (chain length, branching, DE) and external conditions
646 (pH, temperature, ionic strength, presence of metal ions, sugars, etc.) pectin chains may associate in
647 different ways leading to the formation of various types of gels.

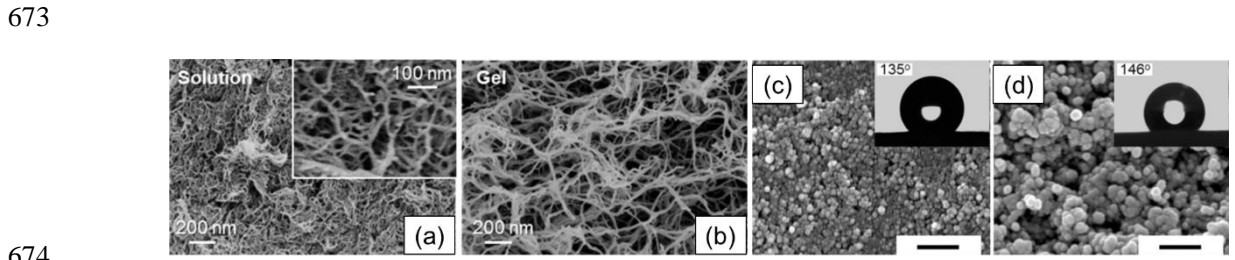
648 Pectin aerogels can be prepared either from gelled pectin solutions or by direct non-solvent induced
649 phase separation (Figure S4). The pathway of structure stabilization (or the state of the matter before
650 solvent exchange, gel or solution) is one of the key parameters influencing aerogel morphology and
651 density. When strong gels are formed via calcium-induced gelation at pH close pK_a , the aerogels are
652 with large pores, very low density ($0.04\text{--}0.05\text{ g/cm}^3$) and not very high specific surface area (around
653 $250\text{--}350\text{ m}^2/\text{g}$). When pectin network is formed via non-solvent induced phase separation from non-
654 gelled pectin solutions during solvent exchange step, the aerogels are mesoporous, with higher density
655 ($0.12\text{--}0.15\text{ g/cm}^3$) and high specific surface area (up to $600\text{ m}^2/\text{g}$) (Groult & Budtova, 2018b).

656 Pectin aerogels turned out to be as performing thermal superinsulating materials as nanocellulose
657 aerogels, compare Figure 10 for pectin and Figure 4 for nanocellulose. The thermal conductivity of
658 pectin cryogels (Lu, Wu, & Chen, 2015) is much higher than that of aerogels, as expected (Figure 10).
659 A complete U-shape curve of thermal conductivity vs density was obtained for pectin aerogels (Figure
660 S5) (Groult et al., 2018a). Their density and morphology was varied by polymer concentration, gelation
661 conditions (hydrogel bonding or calcium crosslinking) and using non-solvent induced phase separation.
662 The lowest value of conductivity, around 0.015 W/m K , was obtained for aerogels made from non-gelled
663 pectin solutions, at pH 2 and pectin concentration of 2 wt%. Calcium-induced gelation results in LM

664 pectin aerogels of very low density (around 0.05 g/cm³) but with too many macropores, thus reducing
 665 the contribution of Knudsen effect (Figure 11a, b).
 666



667
 668 Figure 10.
 669 Thermal conductivity of neat and composite pectin aerogels (filled points) and cryogels (open points)
 670 as a function of density (a) and specific surface area (b). Blue solid line corresponds to the thermal
 671 conductivity of air. Data taken from references: Groult et al., 2018a; Lu, Wu, & Chen, 2015; Nešić et
 672 al., 2018; Rudaz et al., 2014; Zhao, Chen, & Chen, 2017; Zhao et al., 2015.



674
 675 Figure 11
 676 SEM images of
 677 (a) neat pectin aerogels made from dissolved 3 wt% not cross-linked LM pectin at pH 2,

678 (b) the same but cross-linked with calcium, reprinted from Carbohydrate Polymers, Vol 196, Groult, S.
679 & Budtova, T. Thermal conductivity/structure correlations in thermal superinsulating pectin aerogels,
680 Pages 73-81, Copyright 2018, with permission from Elsevier;

681 (c) hydrophobized HM pectin-silica composite aerogels made from the mixtures at pH 1.5 and
682 (d) the same at pH 3, the insets in (c) and (d) show the water contact angle. Reprinted with
683 permission from John Wiley and Sons, Zhao, S., Malfait, W. J., Demilecamps, A., Zhang, Y., Brunner,
684 S., Huber, L., Tingaut, P., Rigacci, A., Budtova, T., & Koebel, M. M. (2015). Strong, Thermally
685 Superinsulating Biopolymer-Silica Aerogel Hybrids by Cogelation of Silicic Acid with Pectin.
686 Angewandte Chemie - International Edition, 54, 14282–14286.

687
688 Very low thermal conductivity, 0.014 – 0.017 W/(m·K) (Figure 10), was also obtained for HM pectin-
689 silica composite aerogels hydrophobized with hexamethyldisilazane (Zhao et al., 2015). By varying
690 pectin:sodium silicate ratio in the mixture and tuning gelation conditions by pH, homogenous
691 mesoporous and “neck-free” composite aerogels were prepared (Figure 11c, d). Specific surface area
692 of these composite aerogels was very high due to silica aerogel phase. Other pectin-based composite
693 aerogels with low thermal conductivity (0.022-0.025 W/(m·K)) were made by mixing LM amidated pectin
694 with TiO₂ and Zn²⁺; aerogels possessed antibacterial properties but the conductivity increased in the
695 presence of TiO₂ (Nešić et al., 2018). When LM pectin was mixed with polyaniline, the composite
696 aerogels were of thermal conductivity 0.033-0.038 W/(m·K) and showed flame retardant property (Zhao,
697 Chen, & Chen, 2017). These higher values can be explained by high density of aerogels, 0.41 – 0.51
698 g/cm³ and rather low specific surface area, 100 – 205 m²/g, indicating low mesoporosity (Zhao, Chen,
699 & Chen, 2017).

700 **4.5 Starch aerogels**

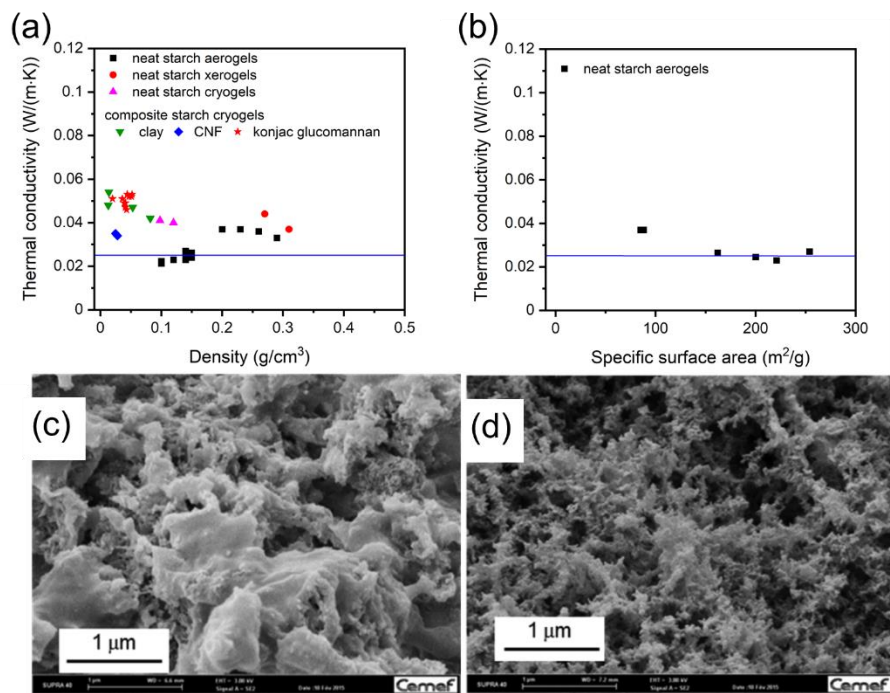
701 Starch is composed of amylose, a linear (1→4)- α linked glucan, and amylopectin, a (1→4)- α -linked
702 D-glucose with (1 → 6)- α branches. There are two main ways of starch processing: via
703 thermomechanical treatment and via dissolution. The first one leads to starch melting; it can then be
704 processed as a thermoplastic polymer. This processing route can be used to obtain starch foams but
705 not aerogels. The second one is gelatinization: heating starch granule suspension in water resulting in
706 granule swelling, loss of crystallinity, amylose leaching out and granule burst. Gelatinization is a
707 complex process not always leading to full starch dissolution: after granules' burst, a colloidal
708 suspension consisting of granule remnants and polysaccharides in solution is formed. The amount,
709 integrity and size of granules' remnants depend on several factors such as pasting temperature; for
710 example, melting temperature of amylose-based double helices, which stabilize granules' ghosts, is
711 above 130 °C. Another factor influencing dissolution is the presence of proteins/lipids which form a skin
712 and prevent granule swelling. If willing to obtain starch aerogels with specific surface area as high as
713 possible, starch needs to be well dissolved as granules and their remnants are non-porous. The
714 influence of non-dissolved entities has been reported for cellulose aerogels: the former decrease
715 cellulose aerogel specific surface area and increase the density (Korhonen & Budtova, 2020), both
716 trends being not favourable for reaching low thermal conductivity.

717 Most of starch solutions are gelling upon temperature decrease which is known as retrogradation
718 process. Numerous parameters affect starch aerogel properties and morphology: starch type and
719 concentration, retrogradation conditions and type of non-solvent (Druel, Bardl, Vorweg, & Budtova,
720 2017; García-González, Uy, Alnaief, & Smirnova 2012; Glenn & Irving, 1995; Ubeyitogullari & Ciftci,
721 2016; Zou & Budtova, 2020). Till now, very few works report on the thermal conductivity of starch

722 aerogels and even less provide both the density and specific surface area (Figure 12a,b). The first
 723 authors who reported the low conductivity of starch aerogels (they were called “microcellular foams”),
 724 0.024-0.036 W/(m·K), were Glenn & Irving (1995). Next, Druel et al. reported the influence of starch
 725 type and retrogradation time on the thermal conductivity of starch aerogels (Druel et al., 2017). Pea
 726 starch aerogels possess the lowest thermal conductivity (0.021-0.023 W/(m·K)) compared to waxy
 727 potato starch, regular potato starch and high amylose corn starch due to the lowest density and finest
 728 morphology (Figure 12c,d).

729

730



731

732

Figure12.

733 (a) and (b): thermal conductivity of starch porous materials as a function of density and specific
 734 surface area, respectively. Blue solid line corresponds to the thermal conductivity of air. Data are

735 taken from Druel et al., 2017; Glenn et al., 1995; Wang et al., 2018; Ye et al., 2019; Yildirim, Shaler,

736

Gardner, Rice, & Bousfield, 2014.

737 (c) and (d): SEM images of regular potato (c) and pea (d) starch aerogels with thermal conductivity
738 0.037 and 0.023 W/(m·K), respectively. Reprinted with permission from Druel, L., Bardl, R., Vorwerg,
739 W., & Budtova, T. Starch Aerogels: A Member of the Family of Thermal Superinsulating Materials.
740 *Biomacromolecules*, 18, 4232–4239. Copyright 2017 American Chemical Society.

741
742 The thermal conductivity of neat and composite starch cryogels is much higher (0.040-0.060
743 W/(m·K)) than that of aerogels due to very large macropores formed during freezing process (Wang et
744 al., 2018; Ye et al., 2019; Yildirim, Shaler, Gardner, Rice, & Bousfield, 2014).

745

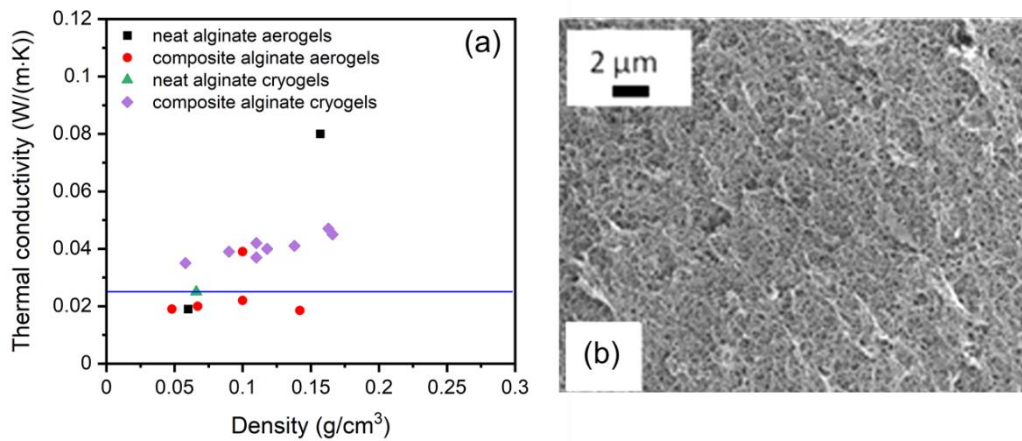
746 **4.6 Alginate aerogels**

747 Alginate is a linear polyuronic acid composed of β -D-mannuronic acid and α -L-guluronic acid in
748 various proportions. Their ratio practically does not vary within algae specie but may vary between 0.45
749 and 1.85 among different algae sources. As pectin, it is an “easy gelling” polysaccharide, and aerogels
750 can be prepared either from alginate solutions gelled with polyvalent metal ions (Gurikov, Raman,
751 Weinrich, Fricke, & Smirnova, 2015; Raman, Gurikov, & Smirnova, 2015) or via non-solvent induced
752 phase separation (Tkalec, Knez, & Novak, 2015). Alginate aerogels have been extensively studied as
753 drug delivery systems, but only few publications report their thermal conductivity (Figure 13). This is
754 surprising as alginate aerogels are known to have low density and high specific surface area, both
755 favourable for low thermal conductivity.

756 Some alginate aerogels and their composites have been reported as thermal superinsulating
757 materials with conductivity around 0.018-0.022 W/(m·K) (Gurikov et al., 2015; Raman et al. 2015), their
758 specific surface area was 370 – 670 m²/g and 210 – 810 m²/g, respectively. Tkalec et al. reported high

759 conductivity (for aerogels), 0.08 W/(m·K), but specific surface area was only 150 m²/g (Tkalec et al.,
 760 2015). When alginate was gelled within PET nonwoven, the thermal conductivity of the composite
 761 aerogel was 0.039 W/(m·K) and specific surface area was surprisingly very low, only 18 m²/g (Ahmad
 762 Ulker, & Erkey, 2018)). Alginate cryogels have higher thermal conductivity, 0.035-0.047 W/(m·K), as
 763 compared to alginate-based aerogels, and as expected from cryogel morphology and low specific
 764 surface area.

765



766

767

Figure13.

768 (a) Thermal conductivity of alginate aerogels and cryogels as a function of density. Blue solid line

769 corresponds to the thermal conductivity of air. Data are taken from the references Gurikov et al.,

770 2015; Li, He, Qin, Chen, & Chen, 2019; Raman et al., 2015; Shang et al., 2016; Tkalec et al., 2015;

771 Ahmad et al., 2018;

772 (b) SEM image of composite alginate-lignin aerogels, reprinted from J. of Supercritical Fluids, 106,

773 S.P. Raman, P. Gurikov, I. Smirnova, Hybrid alginate based aerogels by carbon dioxide induced

774 gelation:Novel technique for multiple applications, Pages 23-33, Copyright 2015, with permission from

775 Elsevier.

776

777 **4.7 Non-polysaccharide bio-aerogels**

778 Bio-aerogels can be made from natural polymers such as proteins. For example, silk fibroin aerogels
779 with density 0.02 g/cm^3 , specific surface area $412 \text{ m}^2/\text{g}$ and thermal conductivity $0.026 \text{ W}/(\text{m}\cdot\text{K})$ were
780 reported (Maleki, Montes, Hayati-Roodbari, Putz, & Huesing, 2018; Maleki, Whitmore, & Hüsing, 2018).
781 In the view of improving aerogel mechanical properties, silk fibroin was mixed with organosilanes
782 (tetramethylorthosilicate and 5-(trimethoxysilyl)pentanoic acid or polymethylsilsesquioxane and 5-
783 (trimethoxysilyl)pentanoic acid) in the presence of surfactant, byproducts extracted by solvent
784 exchange with methanol and aerogels were made from such interpenetrated networks. Density and
785 specific surface area increased as compared to the neat silk fibroin aerogels, but, unfortunately, thermal
786 conductivity also increased to $0.033 - 0.043 \text{ W}/(\text{m}\cdot\text{K})$ (Maleki et al., 2018; Maleki et al., 2018). Silk
787 fibroin is a very interesting but expensive matter with rather complex extraction and purification
788 procedure; its aerogels can probably have potential in biomedical applications. Some other proteins
789 have been used to make aerogels but their thermal conductivity was not reported.

790 Finally, amino acid-based (phenylalanine and leucine, dissolved in toluene or tetralin) hydrophobic
791 supramolecular aerogels were reported (Jamart-Grégoire et al., 2016). The density was very low,
792 $0.004-0.03 \text{ g/cm}^3$ and the specific surface area around $90 \text{ m}^2/\text{g}$. These aerogels possessed thermal
793 conductivity of $0.026-0.027 \text{ W}/(\text{m}\cdot\text{K})$.

794

795 **5. Discussion: overview, problems and prospects**

796 The results presented above show that some bio-aerogels are thermal superinsulating materials,
797 with conductivity comparable to that of superinsulating silica aerogels. This is a very promising result.

798 Except for nanocellulose based aerogels and cryogels, for which data are very scattered, the general
799 trend for bio-aerogels shows that in order to have the lowest possible thermal conductivity, the density
800 should be around 0.1 g/cm^3 and specific surface area above several hundreds of m^2/g . High specific
801 surface area can be seen as a prerequisite to have Knudsen effect, as demonstrated for aerogels based
802 on cellulose II composites, chitosan, pectin, starch and alginate. In other words, in addition to high
803 porosity, pore sizes should be below the mean free path of air molecule, recall Section 2 and Figure 2.
804 This is in the agreement with previously reported trends for inorganic and synthetic polymer aerogels.

805 One of the most intriguing questions is why it is not possible (at least never reported) to make thermal
806 superinsulating cellulose II neat aerogels while this is feasible for cellulose I counterparts and other bio-
807 aerogels? During coagulation cellulose chains are packing in a thermodynamically stable antiparallel
808 direction stabilized by intra- and intermolecular hydrogen bonds. Is this the reason of the “unfavourable”
809 morphology with rather large pores and/or thick pore walls? Molecular modelling of structure formation
810 during coagulation could help a better understanding of thermal conductivity properties of cellulose-
811 based aerogels. Here supercritical drying could be a way to visualize the morphology as a function of
812 processing conditions, e.g. polymer concentration and type of solvent and non-solvent. It should be
813 noted that such important parameter as cellulose crystallinity was never considered when analyzing
814 thermal conductivity. Modelling of thermal conductivity of bio-aerogels is totally absent, which makes
815 this area developing according to trials and errors approach.

816 If willing to use a material for thermal insulation, several important characteristics need to be
817 considered. One is the price of the starting matter. We will not make here a technico-economical
818 evaluation of the production of bio-aerogels; it is a topic of a separate study which should be
819 accompanied by life cycle analysis. However, it should be noted that polysaccharides described above

820 used to make bio-aerogels are natural, renewable and not expensive.

821 The mechanical properties are the key parameter to consider if willing to use a material for thermal
822 insulation. The majority of literature on polysaccharide-based aerogels study their mechanical
823 properties under the uniaxial unconfined compression, most probably because of the easiness in the
824 preparation and handling of cylindrical samples. The results from bio-aerogels' 3-point bending test,
825 which would be very useful for practical applications, are strongly missing. All literature on bio-aerogels
826 agrees that under the uniaxial compression, a typical stress-strain curve is similar to that of foams
827 (Gibson & Ashbi, 1999) with a small linear region (few % strain and less) followed by a stress plateau
828 corresponding to progressive cell walls buckling and collapse, and densification region (see, for
829 example, Rudaz et al., 2014; Kobayashi et al., 2014; Druel, Bardl, Vorweg, & Budtova, 2017; Plappert
830 et al., 2017; Plappert et al., 2018; Rennhofer et al., 2019; Buchtová, Pradille, Bouvard, & Budtova 2019).
831 Bio-aerogels, including those that are thermal superinsulating, are not breaking under compression,
832 with strain at densification reaching 70-80% (Kobayashi, Saito, & Isogai, 2014; Rudaz et al., 2014;
833 Plappert, Nedelec, Rennhofer, Lichtenegger, & Liebner, 2017). This is a very important factor in favour
834 of using bio-aerogels for thermal insulation applications. Silica aerogels are known to be brittle, and
835 thus various reinforcement strategies are used. They involve either short or long fibers (inorganic,
836 synthetic polymer or cellulose-based) (Bardy, Mollendorf, & Pendergast, 2007; Jaxel et al. 2017) with
837 the classical example being so-called aerogel "blankets". Another option to reinforce silica aerogels is
838 to use modified precursors and/or cross-linking (G. Zhang et al., 2004; Hayase et al., 2014; Hayase,
839 Kanamori, Maeno, Kaji, & Nakanishi, 2016). Such modifications are not required for bio-aerogels.

840 Depending on bio-aerogel density, the compressive modulus varies from hundreds KPa to tens MPa.

841 As for organic and inorganic aerogels, compressive modulus is power law dependent on bio-aerogel

842 bulk density with exponent varying from 3 to 4. The majority of literature reports zero Poisson ratio for
843 bio-aerogels, however, recently it was demonstrated that Poisson ratio of low-density cellulose II
844 aerogels is around 0.15 – 0.20, as for silica aerogels (Gross, Reichenauer, & Fricke, 1988), and it
845 decreases with bio-aerogel density increase (Buchtová et al., 2019).

846 There are two strong disadvantageous properties of bio-aerogels that may explain why they are still
847 not considered for a large-scale thermal insulation application. One is drying in supercritical conditions.
848 Despite that supercritical CO₂ is used in various industrial sectors (for separation and extraction, in
849 polymer processing for plasticization and foaming, in chemical and biochemical reactions, for drying
850 samples for scanning electron microscopy), supercritical drying involves high-pressure technology and
851 it is a batch process. To obtain bio-based porous materials with the thermal conductivity and mechanical
852 properties equivalent to those of bio-aerogels but avoiding drying in supercritical conditions is a
853 challenging task which till now is not solved.

854 If drying polysaccharide gels at ambient pressure or in vacuum, pores are closing due to capillary
855 pressure (eq. 5) and remain closed as chains collapse and also due to strong hydrogen bonds that may
856 be formed between numerous hydroxyl groups. To avoid pore closing and network strong densification,
857 various procedures were suggested: for example, a multi-step liquid exchange procedure replacing
858 water with less polar liquids (Svensson, Larsson, Salazar-Alvarez, & Wågberg, 2013), surface
859 esterification which was applied on nanofibrillated cellulose to obtain hydrophobic porous nanopaper
860 (Sehaqui, Zimmermann, & Tingaut, 2014), foaming (Cervin et al., 2013) and etherification of cellulose
861 with hydrophobic moieties in homogeneous conditions (Pour, Beauger, Rigacci, & Budtova, 2015). In
862 all cases mentioned above thermal conductivity was not reported but specific surface area was not high
863 suggesting low mesoporosity and thus limited input of Knudsen effect.

864 One option to solve the problem of irreversible mesopores' closure might be performing surface
865 functionalization of a polysaccharide gel pore walls before drying in the same way as it is done for silica
866 gels. To make aerogel-like material from ambient pressure dried alkoxide- or sodium silicate gels, their
867 pore walls are grafted with $\text{Si}(\text{CH}_3)_3$ groups using hexamethyldisilazane (HMDZ), trimethylchlorosilane
868 (TMCS) or hexamethyldisiloxane (HMDS) (see, for example, Schwertfeger et al., (1998)). This leads to
869 the so-called "spring-back effect," i.e. re-opening of the pores at the final stage of drying due to the
870 repulsion of the grafted groups (see, for example, Hayase et al. (2016)). However, due to the inherent
871 brittleness of the silica network which is built of nanometer-size particles organized in a "pearl necklace",
872 the initially monolithic gel breaks into sub-centimeter size pieces. These pieces possess aerogel
873 properties, i.e. are of low density, mesoporous and their packed bed can be of very low thermal
874 conductivity, around $0.017 \text{ W}/(\text{m}\cdot\text{K})$ (Koebel, Huber, Zhao, & Malfait, 2016). The advantage of such
875 silylation is that it also makes the functionalised silica hydrophobic resisting aging.

876 The second disadvantage is bio-aerogel aging which occurs due to the adsorption of water vapours
877 by naturally hydrophilic polysaccharides. This is especially pronounced for bio-aerogels based on
878 water-soluble polysaccharides. As mentioned above for the case on CNF foams, a two-fold increase of
879 the thermal conductivity was recorded for the relative humidity increasing from 20 to 80 RH%
880 (Apostolopoulou-Kalkavoura et al., 2018). A similar result was obtained for cellulose II cryogels (Shi, Lu,
881 Guo, Sun, & Cao, 2013). When the latter were cold-plasma hydrophobized, the thermal conductivity
882 remained around $0.030 - 0.035 \text{ W}/(\text{m}\cdot\text{K})$ but did not vary as a function of humidity. To the best of our
883 knowledge, the aging of bio-aerogels, i.e. the evolution of density, specific surface area, thermal
884 conductivity and mechanical properties, is practically not studied. Till now, no strategy is developed to
885 "save" thermal superinsulation properties of bio-aerogels. As for silica aerogels, pore surface

886 functionalization of polysaccharide gels could, potentially, help the preservation of network structure
887 against collapse during drying at ambient pressure and decrease water vapours adsorption. However,
888 harsh chemistry should be avoided.

889

890 **6. Conclusions**

891 This review presents the current status on thermal conductivity of bio-aerogels and some other
892 porous polysaccharides, in the view of their potential use for thermal insulation. A special attention is
893 paid on the possibility of reaching the conductivity below that of air, i.e. thermal superinsulation region.
894 The influence of polymer source and processing conditions on the thermal conductivity is analysed via
895 material density and specific surface area. The latter is used to judge on the mesoporosity of the
896 material and thus on the applicability of Knudsen effect.

897 Within the well selected preparation conditions, all bio-aerogels made up to now – based on
898 nanocellulose, pectin, alginate, starch and chitosan – can possess conductivity in thermal
899 superinsulating region, except cellulose II aerogels. Why it is not possible to obtain thermal
900 superinsulating cellulose II aerogels remains an open question. Modelling of structure formation upon
901 coagulation of polysaccharides and of bio-aerogels' thermal conductivity could help better tuning their
902 properties.

903 The opportunities and challenges of bio-aerogels in the view of their use for thermal insulation are
904 discussed. Bio-aerogels have attractive mechanical properties, but they are aging and supercritical
905 drying should be replaced by ambient pressure drying provided the properties of aerogels are preserved.
906 Solving these problems involves interesting and in-depth fundamental research. A technico-economical

907 evaluation of bio-aerogels' production is also needed. There is thus still a lot of exciting research to do
908 for making high performance thermal insulating bio-based materials.

909 **References**

- 910 Aerogels Handbook (2011), in *Advances in Sol-Gel Derived Materials and Technologies*, Eds. M. A.
911 Aegerter, M. Prassas, Springer Science and Business Media LLC: Berlin/Heidelberg, Germany.
- 912 Ahankari, S., Paliwal, P., Subhedar, A., & Kargarzadeh, H. (2021). Recent Developments in
913 Nanocellulose-Based Aerogels in Thermal Applications: A Review. *ACS Nano*, 15, 3849–3874.
- 914 Ahmad, F., Ulker, Z., & Erkey, C. (2018). A novel composite of alginate aerogel with PET nonwoven with
915 enhanced thermal resistance. *Journal of Non-Crystalline Solids*, 491, 7–13.
- 916 Alam M. , Singh H., & Limbachiya. M. C. (2011). Vacuum Insulation Panels (VIPs) for building
917 construction industry - A review of the contemporary developments and future directions. *Applied*
918 *Energy*, 11, 3592–3602.
- 919 Apostolopoulou-Kalkavoura, V., Gordeyeva, K., Lavoine, N., & Bergström, L. (2018). Thermal
920 conductivity of hygroscopic foams based on cellulose nanofibrils and a nonionic polyoxamer.
921 *Cellulose*, 25, 1117–1126.
- 922 Bardy, E. R., Mollendorf, J. C., & Pendergast, D. R. (2007). Thermal conductivity and compressive
923 strain of aerogel insulation blankets under applied hydrostatic pressure. *Journal of Heat Transfer*,
924 129, 232–235.
- 925 Bendahou, D., Bendahou, A., Seantier, B., Grohens, Y., & Kaddami, H. (2015). Nano-fibrillated
926 cellulose-zeolites based new hybrid composites aerogels with super thermal insulating properties.
927 *Industrial Crops and Products*, 65, 374–382.
- 928 Buchtová, N., & Budtova, T. (2016). Cellulose aero-, cryo- and xerogels: towards understanding of
929 morphology control. *Cellulose*, 23, 2585–2595.
- 930 Buchtová, N., Pradille, C., Bouvard, J. L., & Budtova, T. (2019). Mechanical properties of cellulose

931 aerogels and cryogels. *Soft Matter*, 15, 7901–7908.

932 Budtova, T. (2019). Cellulose II aerogels: a review. *Cellulose*, 26, 81-121.

933 Budtova, T., Aguilera, D. A., Beluns, S., Berglund, L., Chartier, C., Espinosa, E., Gaidukovs, S., Klimek-
934 kopyra, A., Kmita, A., Lachowicz, D., Liebner, F., Platnieks, O., Rodríguez, A., Navarro, L. K. T.,
935 Zou, F., & Buwalda, S. J. (2020). Biorefinery approach for aerogels. *Polymers*, 12, 1–63.

936 Cai, C., Wei, Z., Huang, Y., Ding, C., Wang, P., Song, J., Deng, L., Fu, Y., & Zhong, W. H. (2020).
937 Ultralight Programmable Bioinspired Aerogels with an Integrated Multifunctional Surface for Self-
938 Cleaning, Oil Absorption, and Thermal Insulation via Coassembly. *ACS Applied Materials and*
939 *Interfaces*, 12, 11273-11286.

940 Cai, J., Liu, S., Feng, J., Kimura, S., Wada, M., Kuga, S., & Zhang, L. (2012). Cellulose-silica
941 nanocomposite aerogels by in-situ formation of silica in cellulose gel. *Angewandte Chemie -*
942 *International Edition*, 51, 2076–2079.

943 Cervin, N. T., Andersson, L., Ng, J. B. S., Olin, P., Bergström, L., & Waišgberg, L. (2013). Lightweight
944 and strong cellulose materials made from aqueous foams stabilized by nanofibrillated cellulose.
945 *Biomacromolecules*, 14(2), 503–511.

946 Chen, W., Li, Q., Wang, Y., Yi, X., Zeng, J., Yu, H., Liu, Y., & Li, J. (2014). Comparative study of aerogels
947 obtained from differently prepared nanocellulose fibers. *ChemSusChem*, 7, 154–161.

948 De France, K. J., Hoare, T., & Cranston, E. D. (2017). Review of Hydrogels and Aerogels Containing
949 Nanocellulose. *Chemistry of Materials*, 29, 4609–4631.

950 Demilecamps, A., Alves, M., Rigacci, A., Reichenauer, G., & Budtova, T. (2016). Nanostructured
951 interpenetrated organic-inorganic aerogels with thermal superinsulating properties. *Journal of*
952 *Non-Crystalline Solids*, 452, 259–265.

- 953 Demilecamps, A., Beauger, C., Hildenbrand, C., Rigacci, A., & Budtova, T. (2015). Cellulose-silica
954 aerogels. *Carbohydrate Polymers*, *122*, 293–300.
- 955 Demilecamps, A., Reichenauer, G., Rigacci, A., & Budtova, T. (2014). Cellulose-silica composite
956 aerogels from “one-pot” synthesis. *Cellulose*, *21*, 2625–2636.
- 957 Druel, L., Bardl, R., Vorweg, W., & Budtova, T. (2017). Starch Aerogels: A Member of the Family of
958 Thermal Superinsulating Materials. *Biomacromolecules*, *18*, 4232–4239.
- 959 Ebert, H.P. (2011). Thermal Properties of Aerogels, in *Aerogels Handbook (Advances in Sol-Gel Derived*
960 *Materials and Technologies)*. M. A. Aegerter, N. Leventis, & M. M. Koebel (Eds.). New York
961 Dordrecht Heidelberg London: Springer Science and Business Media, p. 537
- 962 Fan, B., Chen, S., Yao, Q., Sun, Q., & Jin, C. (2017). Fabrication of cellulose nanofiber/AlOOH aerogel
963 for flame retardant and thermal insulation. *Materials*, *10*, 1–10.
- 964 Farooq, M., Sipponen, M. H., Seppälä, A., & Österberg, M. (2018). Eco-friendly Flame-Retardant
965 Cellulose Nanofibril Aerogels by Incorporating Sodium Bicarbonate. *ACS Applied Materials and*
966 *Interfaces*, *10*, 27407–27415.
- 967 Feng, J., Le, D., Nguyen, S. T., Tan Chin Nien, V., Jewell, D., & Duong, H. M. (2016). Silica–cellulose
968 hybrid aerogels for thermal and acoustic insulation applications. *Colloids and Surfaces A:*
969 *Physicochemical and Engineering Aspects*, *506*, 298–305.
- 970 Follmann, H. D. M., Oliveira, O. N., Martins, A. C., Lazarin-Bidóia, D., Nakamura, C. V., Rubira, A. F.,
971 Silva, R., & Asefa, T. (2020). Nanofibrous silica microparticles/polymer hybrid aerogels for
972 sustained delivery of poorly water-soluble camptothecin. *Journal of Colloid and Interface Science*,
973 *567*, 92–102.
- 974 Fu, J., Wang, S., He, C., Lu, Z., Huang, J., & Chen, Z. (2016). Facilitated fabrication of high strength

975 silica aerogels using cellulose nanofibrils as scaffold. *Carbohydrate Polymers*, 147, 89–96.

976 Ganesan, K., Budtova, T., Ratke, L., Gurikov, P., Baudron, V., Preibisch, I., Niemeyer, P., Smirnova, I.,
977 & Milow, B. (2018). Review on the production of polysaccharide aerogel particles. *Materials*, 11,
978 1–37.

979 García-González, C. A., Uy, J. J., Alnaief, M., & Smirnova, I. (2012). Preparation of tailor-made starch-
980 based aerogel microspheres by the emulsion-gelation method. *Carbohydrate Polymers*, 88,
981 1378–1386.

982 Ge, X., Shan, Y., Wu, L., Mu, X., Peng, H., & Jiang, Y. (2018). High-strength and morphology-controlled
983 aerogel based on carboxymethyl cellulose and graphene oxide. *Carbohydrate Polymers*, 197,
984 277–283.

985 Glenn, G. M., & Irving, D. W. (1995). Starch-Based Microcellular Foams. *American Association of*
986 *Cereal Chemists*, 72, 155–161.

987 Gibson, L. J., & Ashby, M. F. (1999). Cellular Solids. Structure and Properties (2nd ed.). New York:
988 Cambridge University Press, pp: 184–192.

989 Gross, J., Reichenauer, G., & Fricke, j. (1988). Mechanical properties of SiO₂ - aerogels. *J. Phys. D:*
990 *Appl. Phys.*, 21, 1447–1451.

991 Groult, S., & Budtova, T. (2018a). Thermal conductivity/structure correlations in thermal super-insulating
992 pectin aerogels. *Carbohydrate Polymers*, 196, 73–81.

993 Groult, S., & Budtova, T. (2018b). Tuning structure and properties of pectin aerogels. *European Polymer*
994 *Journal*, 108, 250–261.

995 Guerrero-Alburquerque, N., Zhao, S., Adilien, N., Koebel, M. M., Lattuada, M., & Malfait, W. J. (2020).
996 Strong, Machinable, and Insulating Chitosan-Urea Aerogels: Toward Ambient Pressure Drying of

- 997 Biopolymer Aerogel Monoliths. *ACS Applied Materials and Interfaces*, 12, 22037–22049.
- 998 Guo, L., Chen, Z., Lyu, S., Fu, F., & Wang, S. (2018). Highly flexible cross-linked cellulose nanofibril
999 sponge-like aerogels with improved mechanical property and enhanced flame retardancy.
1000 *Carbohydrate Polymers*, 179, 333–340.
- 1001 Guo, W., Wang, X., Zhang, P., Liu, J., Song, L., & Hu, Y. (2018). Nano-fibrillated cellulose-hydroxyapatite
1002 based composite foams with excellent fire resistance. *Carbohydrate Polymers*, 195, 71–78.
- 1003 Gupta, P., Verma, C., & Maji, P. K. (2019). Flame retardant and thermally insulating clay based aerogel
1004 facilitated by cellulose nanofibers. *Journal of Supercritical Fluids*, 152, 104537.
- 1005 Gurikov, P., Raman, S. P., Weinrich, D., Fricke, M., & Smirnova, I. (2015). A novel approach to alginate
1006 aerogels: Carbon dioxide induced gelation. *RSC Advances*, 5, 7812–7818.
- 1007 Han, Y., Zhang, X., Wu, X., & Lu, C. (2015). Flame Retardant, Heat Insulating Cellulose Aerogels from
1008 Waste Cotton Fabrics by in Situ Formation of Magnesium Hydroxide Nanoparticles in Cellulose
1009 Gel Nanostructures. *ACS Sustainable Chemistry and Engineering*, 3, 1853–1859.
- 1010 Hayase, G., Kanamori, K., Abe, K., Yano, H., Maeno, A., Kaji, H., & Nakanishi, K. (2014).
1011 Polymethylsilsesquioxane-cellulose nanofiber biocomposite aerogels with high thermal insulation,
1012 bendability, and superhydrophobicity. *ACS Applied Materials and Interfaces*, 6, 9466–9471.
- 1013 Hayase, G., Kanamori, K., Maeno, A., Kaji, H., & Nakanishi, K. (2016). Dynamic spring-back behavior
1014 in evaporative drying of polymethylsilsesquioxane monolithic gels for low-density transparent
1015 thermal superinsulators. *Journal of Non-Crystalline Solids*, 434, 115–119.
- 1016 Hemberger, F., Weis, S., Reichenauer, G., & Ebert, H. P. (2009). Thermal transport properties of
1017 functionally graded carbon aerogels. *International Journal of Thermophysics*, 30, 1357–1371.
- 1018 Jamart-Grégoire, B., Son, S., Allix, F., Felix, V., Barth, D., Jannot, Y., Pickaert, G., & Degiovanni, A.

1019 (2016). Monolithic organic aerogels derived from single amino-acid based supramolecular gels:
1020 Physical and thermal properties. *RSC Advances*, 6, 102198–102205.

1021 Jaxel, J., Markevicius, G., Rigacci, A., & Budtova, T. (2017). Thermal superinsulating silica aerogels
1022 reinforced with short man-made cellulose fibers. *Composites Part A: Applied Science and*
1023 *Manufacturing*, 103, 113–121.

1024 Jiménez-Saelices, C., Seantier, B., Cathala, B., & Grohens, Y. (2017). Spray freeze-dried nanofibrillated
1025 cellulose aerogels with thermal superinsulating properties. *Carbohydrate Polymers*, 157, 105–113.

1026 Jiménez-Saelices, C., Seantier, B., Grohens, Y., & Capron, I. (2018). Thermal Superinsulating Materials
1027 Made from Nanofibrillated Cellulose-Stabilized Pickering Emulsions. *ACS Applied Materials and*
1028 *Interfaces*, 10, 16193–16202.

1029 Karadagli, I., Schulz, B., Schestakow, M., Milow, B., Gries, T., & Ratke, L. (2015). Production of porous
1030 cellulose aerogel fibers by an extrusion process. *Journal of Supercritical Fluids*, 106, 105–114.

1031 Kobayashi, Y., Saito, T., & Isogai, A. (2014). Aerogels with 3D ordered nanofiber skeletons of liquid-
1032 crystalline nanocellulose derivatives as tough and transparent insulators. *Angewandte Chemie -*
1033 *International Edition*, 53, 10394–10397.

1034 Koebel, M. M., Huber, L., Zhao, S., & Malfait, W. J. (2016). Breakthroughs in cost-effective, scalable
1035 production of superinsulating, ambient-dried silica aerogel and silica-biopolymer hybrid aerogels:
1036 from laboratory to pilot scale. *Journal of Sol-Gel Science and Technology*, 79, 308–318.

1037 Koebel, M., Rigacci, A., & Achard, P. (2012). Aerogel-based thermal superinsulation: An overview.
1038 *Journal of Sol-Gel Science and Technology*, 63, 315–339.

1039 Korhonen, O., & Budtova, T. (2020). All-cellulose composite aerogels and cryogels. *Composites Part A:*
1040 *Applied Science and Manufacturing*, 137, 106027.

- 1041 Laskowski, J., Milow, B., & Ratke, L. (2015). The effect of embedding highly insulating granular aerogel
1042 in cellulosic aerogel. *Journal of Supercritical Fluids*, *106*, 93–99.
- 1043 Lavoine, N., & Bergström, L. (2017). Nanocellulose-based foams and aerogels: Processing, properties,
1044 and applications. *Journal of Materials Chemistry A*, *5*, 16105–16117.
- 1045 Lazzari, L. K., Perondi, D., Zampieri, V. B., Zattera, A. J., & Santana, R. M. C. (2019). Cellulose/biochar
1046 aerogels with excellent mechanical and thermal insulation properties. *Cellulose*, *26*, 9071–9083.
- 1047 Li, T., Song, J., Zhao, X., Yang, Z., Pastel, G., Xu, S., Jia, C., Dai, J., Dai, C., Gong, A., Jiang, F., Yao,
1048 Y., Fan, T., Yang, B., Wågberg, L., Yang, R., & Hu, L. (2018). Anisotropic, lightweight, strong, and
1049 super thermally insulating nanowood with naturally aligned nanocellulose. *Science Advances*, *4*,
1050 1–10.
- 1051 Lim, M. B., Ganas, A. S., Hanson, J. L., Zhou, X., Hellner, B., Manandhar, S., Garipey, R. E., Baneyx,
1052 F., Wilbur, D. S., & Pauzauskie, P. J. (2020). Crystalline loading of lipophilic Coenzyme Q10
1053 pharmaceuticals within conjugated carbon aerogel derivatives. *Carbon*, *164*, 451–458.
- 1054 Lin, D., Yuen, P. Y., Liu, Y., Liu, W., Liu, N., Dauskardt, R. H., & Cui, Y. (2018). A Silica-Aerogel-
1055 Reinforced Composite Polymer Electrolyte with High Ionic Conductivity and High Modulus.
1056 *Advanced Materials*, *30*. 1802661.
- 1057 Lu, X., Arduini-Schuster, M. C., Kuhn, J., Nilsson, O., Fricke, J., & Pekala, R. W. (1992). Thermal
1058 conductivity of monolithic organic aerogels. *Science*, *255*, 971–972.
- 1059 Lu, Xianping, Nilsson, O., Fricke, J., & Pekala, R. W. (1993). Thermal and electrical conductivity of
1060 monolithic carbon aerogels. *Journal of Applied Physics*, *73*, 581–584.
- 1061 Lu, Y., Wu, H., & Chen, Q. (2015). Facile preparation of fracture-free pectin/clay aerogel monoliths.
1062 *Materials Research Innovations*, *19*, S246–S251.

- 1063 Maleki, H., Montes, S., Hayati-Roodbari, N., Putz, F., & Huesing, N. (2018). Compressible, Thermally
1064 Insulating, and Fire Retardant Aerogels through Self-Assembling Silk Fibroin Biopolymers Inside
1065 a Silica Structure - An Approach towards 3D Printing of Aerogels. *ACS Applied Materials and*
1066 *Interfaces*, 10, 22718–22730.
- 1067 Maleki, H., Whitmore, L., & Hüsing, N. (2018). Novel multifunctional polymethylsilsesquioxane-silk
1068 fibroin aerogel hybrids for environmental and thermal insulation applications. *Journal of Materials*
1069 *Chemistry A*, 6, 12598–12612.
- 1070 Markevicius, G., Ladj, R., Niemeyer, P., Budtova, T., & Rigacci, A. (2017). Ambient-dried thermal
1071 superinsulating monolithic silica-based aerogels with short cellulosic fibers. *Journal of Materials*
1072 *Science*, 52, 2210–2221.
- 1073 IUPAC (2014) Compendium of Chemical Terminology, 2nd ed. (the "Gold Book"). Compiled by
1074 McNaught AD, Wilkinson A. Blackwell Scientific Publications, Oxford (1997). XML on - line
1075 corrected version: <http://goldbook.iupac.org> (2006) created by M. Nic, J. Jirat, B. Kosata; updates
1076 compiled by A. Jenkins. ISBN 0 - 9678550 - 9 - 8. <https://doi.org/10.1351/goldbook>. Last update
1077 2014 - 02 - 24; version: 2.3.3.
- 1078 Mi, Q. Y., Ma, S. R., Yu, J., He, J. S., & Zhang, J. (2016). Flexible and Transparent Cellulose Aerogels
1079 with Uniform Nanoporous Structure by a Controlled Regeneration Process. *ACS Sustainable*
1080 *Chemistry and Engineering*, 4, 656–660.
- 1081 Munier, P., Apostolopoulou-Kalkavoura, V., Persson, M., & Bergström, L. (2020). Strong silica-
1082 nanocellulose anisotropic composite foams combine low thermal conductivity and low moisture
1083 uptake. *Cellulose*, 27, 10825–10836.
- 1084 Nešić, A., Gordić, M., Davidović, S., Radovanović, Ž., Nedeljković, J., Smirnova, I., & Gurikov, P. (2018).

1085 Pectin-based nanocomposite aerogels for potential insulated food packaging application.
1086 *Carbohydrate Polymers*, 195, 128–135.

1087 Nguyen, S. T., Feng, J., Ng, S. K., Wong, J. P. W., Tan, V. B. C., & Duong, H. M. (2014). Advanced
1088 thermal insulation and absorption properties of recycled cellulose aerogels. *Colloids and Surfaces*
1089 *A: Physicochemical and Engineering Aspects*, 445, 128–134.

1090 Ookuna, S., Igarashi, K., Hara, M., Aso, K., Yoshidone, H., Nakayama, H., Suzuki, K., Nakajima, K.
1091 (1993). Porous ion-exchanged fine cellulose particles, method for production thereof, and affinity
1092 carrier. USOO5196527A

1093 Pierre, A.C. (2011). Aerogels Handbook (Advances in Sol-Gel Derived Materials and Technologies). In
1094 M. A. Aegerter, N. Leventis, & M. M. Koebel (Eds.), History of Aerogels (pp. 3-18). New York
1095 Dordrecht Heidelberg London: Springer Science and Business Media.

1096 Pinnow, M., Fink, H. P., Fanter, C., & Kunze, J. (2008). Characterization of highly porous materials from
1097 cellulose carbamate. *Macromolecular Symposia*, 262, 129–139.

1098 Plappert, S. F., Nedelec, J. M., Rennhofer, H., Lichtenegger, H. C., Bernstorff, S., & Liebner, F. W.
1099 (2018). Self-Assembly of Cellulose in Super-Cooled Ionic Liquid under the Impact of Decelerated
1100 Antisolvent Infusion: An Approach toward Anisotropic Gels and Aerogels. *Biomacromolecules*, 19,
1101 4411–4422.

1102 Plappert, S. F., Nedelec, J. M., Rennhofer, H., Lichtenegger, H. C., & Liebner, F. W. (2017). Strain
1103 Hardening and Pore Size Harmonization by Uniaxial Densification: A Facile Approach toward
1104 Superinsulating Aerogels from Nematic Nanofibrillated 2,3-Dicarboxyl Cellulose. *Chemistry of*
1105 *Materials*, 29, 6630–6641.

1106 Pour, G., Beauger, C., Rigacci, A., & Budtova, T. (2015). Xerocellulose: lightweight, porous and

1107 hydrophobic cellulose prepared via ambient drying. *Journal of Materials Science*, 50(13), 4526–
1108 4535

1109 Qi, J., Xie, Y., Liang, H., Wang, Y., Ge, T., Song, Y., Wang, M., Li, Q., Yu, H., Fan, Z., Liu, S., Wang, Q.,
1110 Liu, Y., Li, J., Lu, P., & Chen, W. (2019). Lightweight, Flexible, Thermally-Stable, and Thermally-
1111 Insulating Aerogels Derived from Cotton Nanofibrillated Cellulose. *ACS Sustainable Chemistry
1112 and Engineering*, 7, 9202–9210.

1113 Raman, S. P., Gurikov, P., & Smirnova, I. (2015). Hybrid alginate based aerogels by carbon dioxide
1114 induced gelation: Novel technique for multiple applications. *Journal of Supercritical Fluids*, 106,
1115 23–33.

1116 Reichenauer, G., Heinemann, U., & Ebert, H. P. (2007). Relationship between pore size and the gas
1117 pressure dependence of the gaseous thermal conductivity. *Colloids and Surfaces A:
1118 Physicochemical and Engineering Aspects*, 300, 204–210.

1119 Rennhofer, H., Plappert, S. F., Lichtenegger, H. C., Bernstorff, S., Fitzka, M., Nedelec, J. M., & Liebner,
1120 F. W. (2019). Insight into the nanostructure of anisotropic cellulose aerogels upon compression.
1121 *Soft Matter*, 15, 8372–8380.

1122 Robitzer, M., Renzo, F. Di, & Quignard, F. (2011). Natural materials with high surface area.
1123 Physisorption methods for the characterization of the texture and surface of polysaccharide
1124 aerogels. *Microporous and Mesoporous Materials*, 140, 9–16.

1125 Rudaz, C. (2013). *Cellulose and Pectin Aerogels: Towards their nano-structuration*. PhD Thesis.
1126 CEMEF - Centre de Mise en Forme des Matériaux, Mines ParisTech.

1127 Rudaz, C., Courson, R., Bonnet, L., Calas-Etienne, S., Sallée, H., & Budtova, T. (2014). Aeropectin:
1128 Fully biomass-based mechanically strong and thermal superinsulating aerogel.

- 1129 *Biomacromolecules*, 15, 2188–2195.
- 1130 Sakai, K., Kobayashi, Y., Saito, T., & Isogai, A. (2016). Partitioned aers at microscale and nanoscale:
1131 Thermal diffusivity in ultrahigh porosity solids of nanocellulose. *Scientific Reports*, 6, 1–7.
- 1132 Schwertfeger, F., Frank, D., & Schmidt, M. (1998). Hydrophobic waterglass based aerogels without
1133 solvent exchange or supercritical drying. *Journal of Non-Crystalline Solids*, 225, 24–29.
- 1134 Seantier, B., Bendahou, D., Bendahou, A., Grohens, Y., & Kaddami, H. (2016). Multi-scale cellulose
1135 based new bio-aerogel composites with thermal super-insulating and tunable mechanical
1136 properties. *Carbohydrate Polymers*, 138, 335–348.
- 1137 Sehaqui, H., Zimmermann, T., & Tingaut, P. (2014). Hydrophobic cellulose nanopaper through a mild
1138 esterification procedure. *Cellulose*, 21(1), 367–382.
- 1139 Shang, K., Liao, W., Wang, J., Wang, Y. T., Wang, Y. Z., & Schiraldi, D. A. (2016). Nonflammable Alginate
1140 Nanocomposite Aerogels Prepared by a Simple Freeze-Drying and Post-Cross-Linking Method.
1141 *ACS Applied Materials and Interfaces*, 8, 643–650.
- 1142 Shi, J., Lu, L., Guo, W., Sun, Y., & Cao, Y. (2013). An environment-friendly thermal insulation material
1143 from cellulose and plasma modification. *Journal of Applied Polymer Science*, 130, 3652–3658.
- 1144 Shi, J., Lu, L., Guo, W., Zhang, J., & Cao, Y. (2013). Heat insulation performance, mechanics and
1145 hydrophobic modification of cellulose-SiO₂ composite aerogels. *Carbohydrate Polymers*, 98, 282–
1146 289.
- 1147 Silva, S. S., Mano, J. F., & Reis, R. L. (2017). Ionic liquids in the processing and chemical modification
1148 of chitin and chitosan for biomedical applications. *Green Chemistry*, 19, 1208–1220.
- 1149 Song, J., Chen, C., Yang, Z., Kuang, Y., Li, T., Li, Y., Huang, H., Kierzewski, I., Liu, B., He, S., Gao, T.,
1150 Yurker, S. U., Gong, A., Yang, B., & Hu, L. (2018). Highly Compressible, Anisotropic Aerogel with

- 1151 Aligned Cellulose Nanofibers. *ACS Nano*, 12, 140–147.
- 1152 Sun, Y., Chu, Y., Wu, W., & Xiao, H. (2021). Nanocellulose-based lightweight porous materials: A review.
1153 *Carbohydrate Polymers*, 255, 117489.
- 1154 Svensson, A., Larsson, P. T., Salazar-Alvarez, G., & Wågberg, L. (2013). Preparation of dry ultra-porous
1155 cellulosic fibres: Characterization and possible initial uses. *Carbohydrate Polymers*, 92(1), 775–
1156 783.
- 1157 Takeshita, S., & Yoda, S. (2015). Chitosan Aerogels: Transparent, Flexible Thermal Insulators.
1158 *Chemistry of Materials*, 27, 7569–7572.
- 1159 Tang, L., Zhuang, S., Hong, B., Cai, Z., Chen, Y., & Huang, B. (2019). Synthesis of light weight, high
1160 strength biomass-derived composite aerogels with low thermal conductivities. *Cellulose*, 26,
1161 8699–8712.
- 1162 Tkalec, G., Knez, Ž., & Novak, Z. (2015). Formation of polysaccharide aerogels in ethanol. *RSC*
1163 *Advances*, 5, 77362–77371.
- 1164 Ubeyitogullari, A., & Ciftci, O. N. (2016). Formation of nanoporous aerogels from wheat starch.
1165 *Carbohydrate Polymers*, 147, 125–132.
- 1166 Wang, D., Peng, H., Yu, B., Zhou, K., Pan, H., Zhang, L., Li, M., Liu, M., Tian, A., & Fu, S. (2020).
1167 Biomimetic structural cellulose nanofiber aerogels with exceptional mechanical, flame-retardant
1168 and thermal-insulating properties. *Chemical Engineering Journal*, 389, 124449.
- 1169 Wang, M., Zhang, B., Ding, J., Xu, N., Bernards, M. T., He, Y., & Shi, Y. (2020). Three-Dimensional
1170 Nitrogen-Doped Graphene Aerogel-Supported MnO Nanoparticles as Efficient Electrocatalysts for
1171 CO₂ Reduction to CO. *ACS Sustainable Chemistry & Engineering*, 8, 4983–4994.
- 1172 Wang, X., Liu, F., Li, Y., Zhang, W., Bai, S., Zheng, X., Huan, J., Cao, G., Yang, T., Wang, M., Jiang, Z.,

1173 Wang, C., & Ho, S. H. (2020). Development of a facile and bi-functional superhydrophobic
1174 suspension and its applications in superhydrophobic coatings and aerogels in high-efficiency oil-
1175 water separation. *Green Chemistry*, 22, 7424–7434.

1176 Wang, Y., Wu, K., Xiao, M., Riffat, S. B., Su, Y., & Jiang, F. (2018). Thermal conductivity, structure and
1177 mechanical properties of konjac glucomannan/starch based aerogel strengthened by wheat straw.
1178 *Carbohydrate Polymers*, 197, 284–291.

1179 Wang, Y., Uetani, K., Liu, S., Zhang, X., Wang, Y., Lu, P., Wei, T., Fan, Z., Shen, J., Yu, H., Li, S., Zhang,
1180 Q., Li, Q., Fan, J., Yang, N., Wang, Q., Liu, Y., Cao, J., Li, J., & Chen, W. (2017). Multifunctional
1181 Bionanocomposite Foams with a Chitosan Matrix Reinforced by Nanofibrillated Cellulose.
1182 *ChemNanoMat*, 3, 98–108.

1183 Wicklein, B., Kocjan, A., Salazar-Alvarez, G., Carosio, F., Camino, G., Antonietti, M., & Bergström, L.
1184 (2015). Thermally insulating and fire-retardant lightweight anisotropic foams based on
1185 nanocellulose and graphene oxide. *Nature Nanotechnology*, 10, 277–283.

1186 Wiener, M., Reichenauer, G., Braxmeier, S., Hemberger, F., & Ebert, H. P. (2009). Carbon aerogel-
1187 based high-temperature thermal insulation. *International Journal of Thermophysics*, 30, 1372–
1188 1385.

1189 Yan, Y., Ge, F., Qin, Y., Ruan, M., Guo, Z., He, C., & Wang, Z. (2020). Ultralight and robust aerogels
1190 based on nanochitin towards water-resistant thermal insulators. *Carbohydrate Polymers*, 248,
1191 116755.

1192 Ye, D. D., Wang, T., Liao, W., Wang, H., Zhao, H. B., Wang, Y. T., Xu, S., & Wang, Y. Z. (2019). Ultrahigh-
1193 Temperature Insulating and Fire-Resistant Aerogels from Cationic Amylopectin and Clay via a
1194 Facile Route. *ACS Sustainable Chemistry and Engineering*, 7, 11582–11592.

- 1195 Yildirim, N., Shaler, S. M., Gardner, D. J., Rice, R., & Bousfield, D. W. (2014). Cellulose nanofibril (CNF)
1196 reinforced starch insulating foams. *Cellulose*, *21*, 4337–4347.
- 1197 Zhang, G., Dass, A., Rawashdeh, A. M. M., Thomas, J., Counsil, J. A., Sotiriou-Leventis, C., Fabrizio,
1198 E. F., Ilhan, F., Vassilaras, P., Scheiman, D. A., McCorkle, L., Palczer, A., Johnston, J. C., Meador,
1199 M. A., & Leventis, N. (2004). Isocyanate-crosslinked silica aerogel monoliths: Preparation and
1200 characterization. *Journal of Non-Crystalline Solids*, *350*, 152–164.
- 1201 Zhang, L. M., Sui, X. L., Zhao, L., Huang, G. S., Gu, D. M., & Wang, Z. B. (2017). Three-dimensional
1202 hybrid aerogels built from graphene and polypyrrole-derived nitrogen-doped carbon nanotubes as
1203 a high-efficiency Pt-based catalyst support. *Carbon*, *121*, 518–526.
- 1204 Zhang, S., Feng, J., Feng, J., Jiang, Y., & Li, L. (2018). Ultra-low shrinkage chitosan aerogels trussed
1205 with polyvinyl alcohol. *Materials and Design*, *156*, 398–406.
- 1206 Zhang, X., Zhao, X., Xue, T., Yang, F., Fan, W., & Liu, T. (2020). Bidirectional anisotropic
1207 polyimide/bacterial cellulose aerogels by freeze-drying for super-thermal insulation. *Chemical*
1208 *Engineering Journal*, *385*, 123963.
- 1209 Zhao, H. B., Chen, M., & Chen, H. B. (2017). Thermally Insulating and Flame-Retardant
1210 Polyaniline/Pectin Aerogels. *ACS Sustainable Chemistry and Engineering*, *5*, 7012–7019.
- 1211 Zhao, S., Malfait, W. J., Demilecamps, A., Zhang, Y., Brunner, S., Huber, L., Tingaut, P., Rigacci, A.,
1212 Budtova, T., & Koebel, M. M. (2015). Strong, Thermally Superinsulating Biopolymer-Silica Aerogel
1213 Hybrids by Cogelation of Silicic Acid with Pectin. *Angewandte Chemie - International Edition*, *54*,
1214 14282–14286.
- 1215 Zhao, S., Malfait, W. J., Guerrero-Alburquerque, N., Koebel, M. M., & Nyström, G. (2018). Biopolymer
1216 Aerogels and Foams: Chemistry, Properties, and Applications. *Angewandte Chemie -*

1217 *International Edition*, 57, 7580–7608.

1218 Zhao, S., Malfait, W. J., Jeong, E., Fischer, B., Zhang, Y., Xu, H., Angelica, E., Risen, W. M., Suggs, J.
1219 W., & Koebel, M. M. (2016). Facile One-Pot Synthesis of Mechanically Robust Biopolymer-Silica
1220 Nanocomposite Aerogel by Cogelation of Silicic Acid with Chitosan in Aqueous Media. *ACS*
1221 *Sustainable Chemistry and Engineering*, 4, 5674–5683.

1222 Zhao, S., Zhang, Z., Sèbe, G., Wu, R., Rivera Virtudazo, R. V., Tingaut, P., & Koebel, M. M. (2015).
1223 Multiscale assembly of superinsulating silica aerogels within silylated nanocellulosic scaffolds:
1224 Improved mechanical properties promoted by nanoscale chemical compatibilization. *Advanced*
1225 *Functional Materials*, 25, 2326–2334.

1226 Zhou, G., Kim, N. R., Chun, S. E., Lee, W., Um, M. K., Chou, T. W., Islam, M. F., Byun, J. H., & Oh, Y.
1227 (2018). Highly porous and easy shapeable poly-dopamine derived graphene-coated single walled
1228 carbon nanotube aerogels for stretchable wire-type supercapacitors. *Carbon*, 130, 137–144.

1229 Zhou, S., Apostolopoulou-Kalkavoura, V., Tavares da Costa, M. V., Bergström, L., Strømme, M., & Xu,
1230 C. (2020). Elastic Aerogels of Cellulose Nanofibers@Metal–Organic Frameworks for Thermal
1231 Insulation and Fire Retardancy. *Nano-Micro Letters*, 12, 1–13.

1232 Zhou, X., Jin, H., Xu, T., Wang, J., Zhu, Y., Ding, S., Hu, T., Yun, S., & Chen, J. (2020). Excellent flame
1233 retardant and thermal insulated palygorskite/wood fiber composite aerogels with improved
1234 mechanical properties. *Applied Clay Science*, 184, 105402.

1235 Zhu, F. (2019). Starch based aerogels: Production, properties and applications. *Trends in Food Science*
1236 *and Technology*, 89, 1–10.

1237 Zhu, J., Xiong, R., Zhao, F., Peng, T., Hu, J., Xie, L., Xie, H., Wang, K., & Jiang, C. (2020). Lightweight,
1238 High-Strength, and Anisotropic Structure Composite Aerogel Based on Hydroxyapatite

1239 Nanocrystal and Chitosan with Thermal Insulation and Flame Retardant Properties. *ACS*
1240 *Sustainable Chemistry and Engineering*, 8, 71–83.

1241 Zou, F., & Budtova, T. (2020). Tailoring the morphology and properties of starch aerogels and cryogels
1242 via starch source and process parameter. *Carbohydrate Polymers*, 255, 117344.

1243 Zou, F., Peng, L., Fu, W., Zhang, J., & Li, Z. (2015). Flexible superhydrophobic polysiloxane aerogels
1244 for oil-water separation via one-pot synthesis in supercritical CO₂. *RSC Advances*, 5, 76346–
1245 76351.

1246 Zu, G., Shen, J., Wang, W., Zou, L., Lian, Y., Zhang, Z., Liu, B., & Zhang, F. (2014). Robust, highly
1247 thermally stable, core-shell nanostructured metal oxide aerogels as high-temperature thermal
1248 superinsulators, adsorbents, and catalysts. *Chemistry of Materials*, 26(19), 5761–5772.

1249

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Supporting Information

Polysaccharide-based aerogels for thermal insulation and superinsulation: an overview

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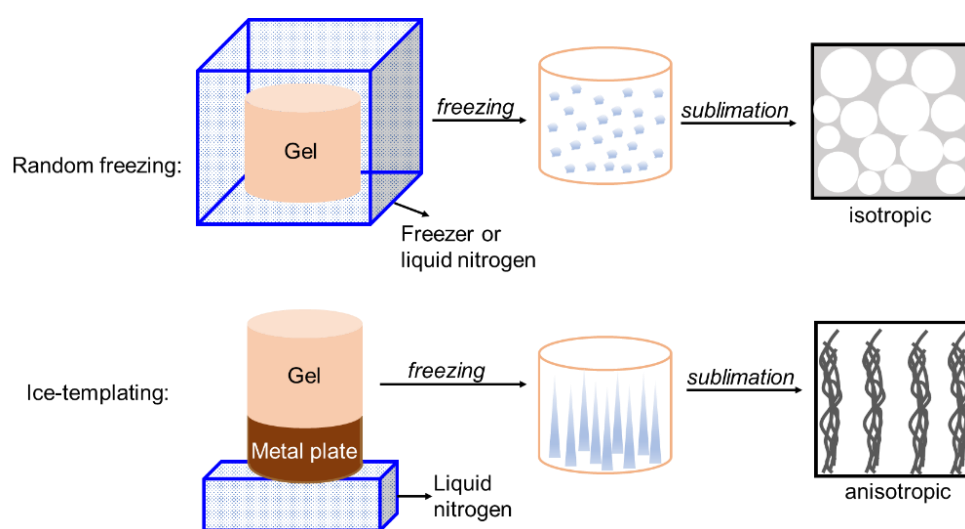
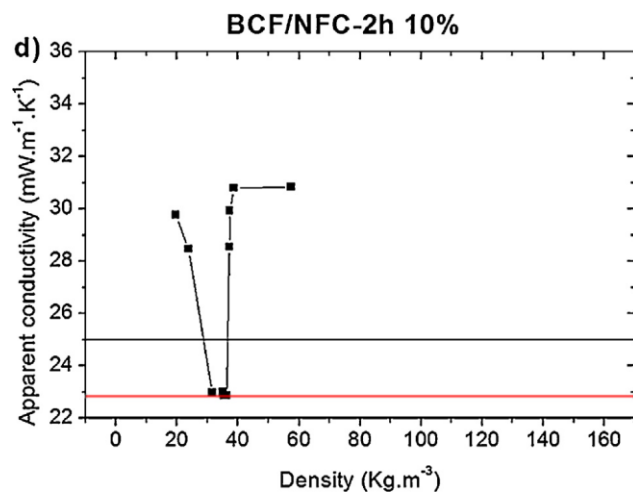


Figure S1.

Schematic presentation of random freezing and ice-templating methods.

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Figure S2.

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Example of U-shape conductivity-density dependence for freeze-dried bleached cellulose fibers

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mixed with CNF. Reprinted from Carbohydrate Polymers, Vol 138, Seantier, B., Bendahou, D.,

1277

Bendahou, A., Grohens, Y., & Kaddami, H., Multi-scale cellulose based new bio-aerogel composites

1278

with thermal super-insulating and tunable mechanical properties, Pages 335–348, Copyright 2016,

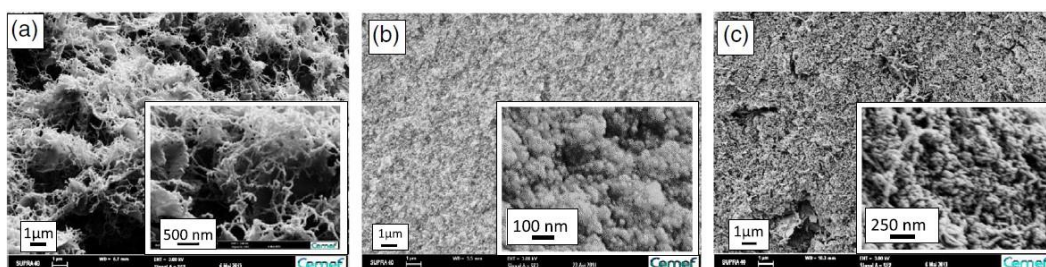
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Figure S3.

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Aerogels based on tritylcellulose (a), polyethoxydisiloxane (b) and

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tritylcellulose/polyethoxydisiloxane (c). Reprinted from Journal of Non-Crystalline Solids, Vol 452,

1287

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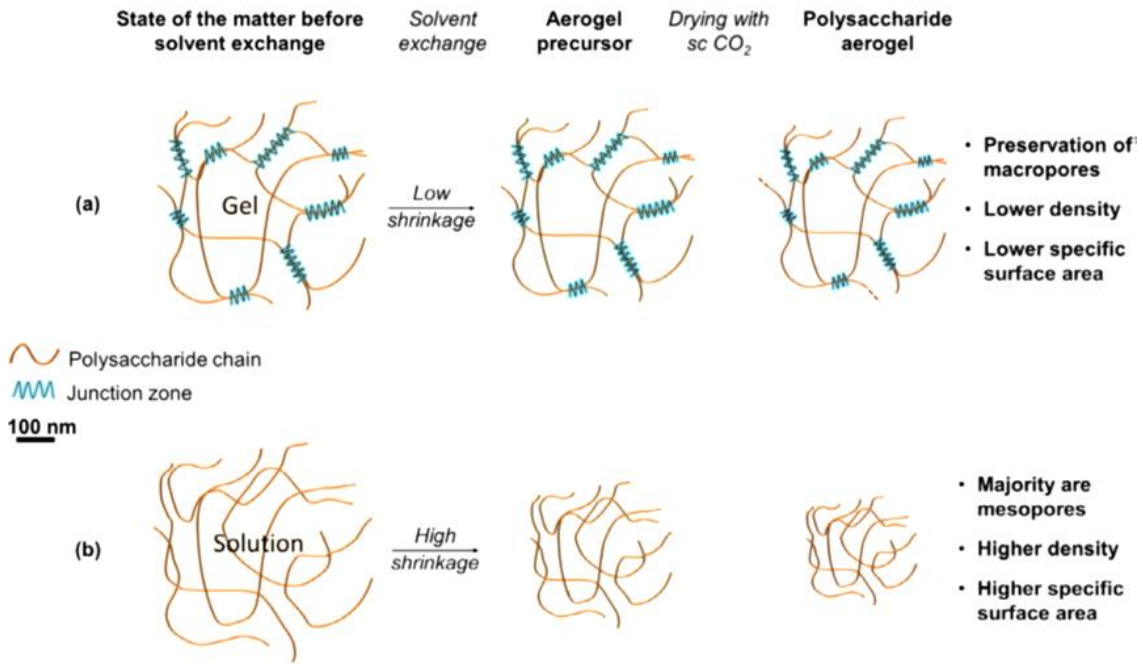
Demilecamps, A., Alves, M., Rigacci, A., Reichenauer, G., & Budtova, T., Nanostructured

1289

interpenetrated organic-inorganic aerogels with thermal superinsulating properties, Pages 259–265,

1290

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Figure S4.

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A schematic presentation of the two main ways of structure formation in pectin aerogels: starting

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from a gel (a) and from solution via non-solvent induced phase separation (b). Reprinted from

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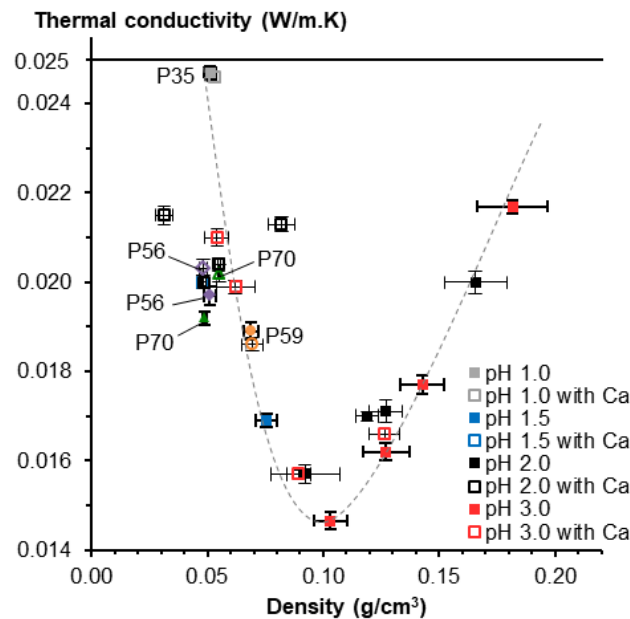
European Polymer Journal, Vol 108, Groult, S., & Budtova, T., Tuning structure and properties of

1297

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1301 Figure S5.

1302 Thermal conductivity as a function of density for LM pectin aerogel with DE 35 (most of the
1303 experimental point) and pectins with various DE: 56% (“P56”, diamonds), 59% (“P59”, circles) and
1304 70% (“P70”, triangles). pH in the legend corresponds to the pH of pectin solution. Solid line is
1305 conductivity of air in ambient conditions; dashed line is given to guide the eye. Reprinted from
1306 *Carbohydrate Polymers*, Vol 196, Groult, S. & Budtova, T. Thermal conductivity/structure correlations
1307 in thermal super-insulating pectin aerogels, Pages 73-81, Copyright 2018, with permission from
1308 Elsevier.

1309
1310
1311 **References:**

- 1312
- 1313 Demilecamps, A., Alves, M., Rigacci, A., Reichenauer, G., & Budtova, T. (2016). Nanostructured
1314 interpenetrated organic-inorganic aerogels with thermal superinsulating properties. *Journal of*
1315 *Non-Crystalline Solids*, 452, 259–265.
- 1316 Groult, S., & Budtova, T. (2018a). Tuning structure and properties of pectin aerogels. *European Polymer*
1317 *Journal*, 108, 250–261.
- 1318 Groult, S., & Budtova, T. (2018b). Thermal conductivity/structure correlations in thermal super-insulating
1319 pectin aerogels. *Carbohydrate Polymers*, 196, 73–81.
- 1320 Seantier, B., Bendahou, D., Bendahou, A., Grohens, Y., & Kaddami, H. (2016). Multi-scale cellulose
1321 based new bio-aerogel composites with thermal super-insulating and tunable mechanical
1322 properties. *Carbohydrate Polymers*, 138, 335–348.