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# 1:1 Ca<sup>2+</sup>:Cu<sup>2+</sup> A-site order in a ferrimagnetic double double perovskite

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**Abstract:** Cation ordering in ABX<sub>3</sub> perovskites is important to structural, physical and chemical properties. Here we report discovery of CaCuFeReO<sub>6</sub> with the tetragonal AA'BB'O<sub>6</sub> double double perovskite structure that was previously only reported for A' = Mn compositions. CaCuFeReO<sub>6</sub> occurs in the same phase field as CaCu<sub>3</sub>Fe<sub>2</sub>Re<sub>2</sub>O<sub>12</sub> demonstrating that different A-cation ordered perovskites may be obtained in the same chemical system. CaCuFeReO<sub>6</sub> has ferrimagnetic order of Fe, Re and Cu spins below T<sub>C</sub> = 567 K, in contrast to Mn analogues where the Mn spins order separately at much lower temperatures. The magnetoresistance of CaCuFeReO<sub>6</sub> displays low-field 'butterfly' hysteresis with an unusual change from negative to positive values as field increases. Many more AA'BB'O<sub>6</sub> double double perovskites may be accessible for A' = Cu and other divalent transition metals at high pressure, so the presently known phases likely represent only the 'tip of the iceberg' for this family.

ABO<sub>3</sub> perovskite oxides are important for a wide range of chemical and physical properties.<sup>[1,2]</sup> Cation ordering within the perovskite lattice can generate emergent physical properties,<sup>[3]</sup> notably in A<sub>2</sub>BB'O<sub>6</sub> double perovskites (DPV) where rock-salt ordering of B/B' transition metal cations<sup>[4,5,6]</sup> leads to ferrimagnetism and large magneto-resistance in Sr<sub>2</sub>FeMoO<sub>6</sub> and related materials.<sup>[7,8,9]</sup> More complex perovskites featuring order of alkali metal, alkaline earth or rare earth A cations with A' transition metal ions in addition to the latter B/B' order enable effects of the magnetic A' transition metal ions on the B/B' ferrimagnetism to be explored. Two such classes of compounds have been reported, both requiring high pressure synthesis to stabilise cation order based on size difference between the large A and small transition metal A' cations as substantial tilts of the B/B'O<sub>6</sub> octahedra reduce the A' coordination number from 12 in the ideal perovskite structure to 4 in the ordered derivatives. AA'<sub>3</sub>B<sub>2</sub>B'<sub>2</sub>O<sub>12</sub> (referred to as 1322-type) materials have a cubic (space group Pn-3) structure with 1:3 order of 12-coordinate A and 4-coordinate square-planar A' cations, and are formed predominantly

for A' = Cu. Ferrimagnets with Curie temperatures (T<sub>C</sub>) >300 K are found for CaCu<sub>3</sub>B<sub>2</sub>B'<sub>2</sub>O<sub>12</sub> (B/B' = Fe/Re, Fe/Os, Mn/Os)<sup>[9,10,11]</sup> and ACu<sub>3</sub>Fe<sub>2</sub>Os<sub>2</sub>O<sub>12</sub> (A = Na, La)<sup>[12,13]</sup> materials, with lower temperature ferri- or antiferro- magnetic orders in other analogues; CaCu<sub>3</sub>B<sub>2</sub>B'<sub>2</sub>O<sub>12</sub> (B/B' = Ga/Sb, Cr/Sb, Fe/Sb, Fe/Nb),<sup>[14,15,16,17]</sup> LaCu<sub>3</sub>Co<sub>2</sub>Re<sub>2</sub>O<sub>12</sub><sup>[18]</sup> and LaMn<sub>3</sub>Ni<sub>2</sub>Mn<sub>2</sub>O<sub>12</sub>.<sup>[19]</sup> The latter is the only A' = Mn (and non-Cu) 1322 phase reported to date.

The second group of A/A' and B/B' ordered perovskites have AA'BB'O<sub>6</sub> stoichiometry and are labelled double double (or doubly ordered) perovskites (DDPv's). They have tetragonal P<sub>4</sub>/n symmetry with columnar ordering of 10-coordinate A and 4-coordinate A' cations, and all previously reported materials are based on A' = Mn. RMnMnB'O<sub>6</sub> (R = rare earth) series are reported for B' = Sb<sup>[20,21]</sup> and Ta,<sup>[22]</sup> and many CaMnBB'O<sub>6</sub> materials (B = Mn, Fe, Co, Ni for B' = Re,<sup>[23,24,25]</sup> and B/B' = Fe/Ta,<sup>[26]</sup> Cr/Sb and Fe/Sb<sup>[27]</sup>) were also synthesised at high temperatures and pressures. High temperature ferrimagnetism was reported in CaMnFeReO<sub>6</sub> (T<sub>C</sub> = 500 K) and also a Cu-doped derivative of nominal composition Ca(Mn<sub>0.5</sub>Cu<sub>0.5</sub>)FeReO<sub>6</sub> (T<sub>C</sub> = 560 K), with an unusual switch between negative and positive magnetoresistances in the latter at low temperatures.<sup>[23]</sup> The latter material suggests that DDPVs stabilised by A/Cu ordering, without any Mn content, may be accessible. This offers the exciting possibility of being able to stabilise AA'<sub>3</sub>B<sub>2</sub>B'<sub>2</sub>O<sub>12</sub> (1322) and AA'BB'O<sub>6</sub> (DDPv) cation ordered perovskites in the same A-A'(=Cu)-B-B' chemical system for the first time, and hence compare effects of the 1:3 and 1:1 A:Cu orders on the magnetic and electronic properties of the B/B' sublattice. Here we report synthesis of a Mn-free, high-T<sub>C</sub> DDPv material CaCuFeReO<sub>6</sub> and comparison of properties to the previously reported CaCu<sub>3</sub>Fe<sub>2</sub>Re<sub>2</sub>O<sub>12</sub>.<sup>[9]</sup>

Samples of nominal composition CaCuFeReO<sub>6</sub> were heated under pressures of 10 – 15.5 GPa using stoichiometric mixtures of Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub>, ReO<sub>2</sub> and CuO in a Walker-type module as described in Supplementary Information (SI). A component with the P<sub>4</sub>/n DDPv type structure was identified by powder X-ray diffraction within multiphase products. An optimum sample was synthesised at 15.5

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GPa and 1400 °C and this material was characterised by powder synchrotron X-ray diffraction at 300 K, neutron diffraction patterns from 5 to 550 K, and magnetic and electronic transport measurements (details in SI).

The crystal structure of  $\text{CaCuFeReO}_6$  was refined using synchrotron X-ray and neutron powder diffraction profiles. 6.8(2)% Fe/Re antisite disorder was found from the X-ray fit (full results are in ESI) and this value was fixed in all neutron refinements. The A' = Cu cation columns have two inequivalent sites with tetrahedral (Cu1) and square planar (Cu2) coordinations. Refinement against powder neutron data (Table 1 and Fig. 1) showed that 29(1) and 0(1)% of Fe was substituted at the Cu1 and Cu2 sites, respectively, so their occupancies were fixed as 70%Cu/30%Fe and 100%Cu in subsequent variable temperature neutron fits.

Magnetic diffraction peaks from  $\text{CaCuFeReO}_6$  (Fig. 1) were observed in all neutron patterns from 5 to 550 K. Peaks from ordered moments at both A-site Cu and B/B' Fe/Re are observed. The magnetic peaks are indexed by propagation vector [0 0 0] and good fits were obtained from a ferrimagnetic model with spins parallel to the *c*-axis (Fig. 2). The small Re spin was found to be unstable in initial magnetic refinements and was constrained to be antiparallel to the Fe moment as done in previous refinements of Mn-based  $\text{CaA}'\text{FeReO}_6$  materials.<sup>[23]</sup> The two Cu site spins were both found to be parallel to the Fe spin and were constrained to be equal in the final refinements. Results of the refinement of crystal and magnetic structures against powder neutron diffraction data are shown in Figs. 1 and 2 and Table 1. Further results are in SI.

Table 1. Atomic positions and main interatomic distances and angles of  $\text{CaCuFeReO}_6$  as refined against 320 K NPD data ( $P4_2/n$ ;  $a = 7.5381(2)$  Å and  $c = 7.6327(4)$  Å; residuals  $R_p = 2.87\%$ ,  $R_{wp} = 3.97\%$ ,  $R_B = 4.14\%$ ,  $R_I = 3.32\%$ ,  $R_{mag} = 5.86\%$ ,  $\chi^2 = 11.9$ ). Refined magnetic moments;  $\mu_{\text{Fe}} = 2.61(2)$   $\mu_B$ ,  $\mu_{\text{Re}} = 1.04(1)$   $\mu_B$ ,  $\mu_{\text{Cu}} = 1.18(1)$   $\mu_B$

Site	x	y	z	Occ	$B_{\text{iso}}$ (Å <sup>2</sup> )
Ca (4e)	0.25	0.75	0.769(1)	1	0.2(2)
Cu1 (2a) /Fe	0.75	0.75	0.75	0.71/ 0.29(1)	1.1(1)
Cu2 (2b) /Fe	0.25	0.25	0.75	1.00/ 0.00(1)	1.1
Fe (4c) /Re	0.0	0.5	0.5	0.932/ 0.068	1.19(3)
Re (4d) /Fe	0.0	0.0	0.5	0.932/ 0.068	1.19
O1 (8g)	-0.044(1)	0.576(1)	0.247(2)	1	0.83(5)
O2 (8g)	-0.239(2)	-0.042(1)	0.580(1)	1	0.83
O3 (8g)	-0.265(1)	0.058(1)	-0.034(1)	1	0.83
Bond A-O	Length (Å)		Bond B-O	Length (Å)	
(Ca-O1)x2	2.914(7)		(Fe-O1)x2	2.04(1)	
(Ca-O1)x2	2.585(7)		(Fe-O2)x2	2.09(1)	
(Ca-O2)x2	2.465(8)		(Fe-O3)x2	2.06(1)	
(Ca-O3)x2	2.46(1)		<Fe-O>	2.06(1)	
(Ca-O3)x2	2.36(1)		(Re-O1)x2	2.00(1)	
<Ca-O>	2.56(1)		(Re-O2)x2	1.93(1)	
(Cu1-O2)x4	2.040(6)		(Re-O3)x2	1.84(1)	
(Cu2-O1)x4	2.029(7)		<Re-O>	1.92(1)	
Bond angles (°)	<Fe-O2-Re>		139.8(3)		
<Fe-O1-Re>	141.5(4)		<Fe-O3-Re>	150.5(4)	

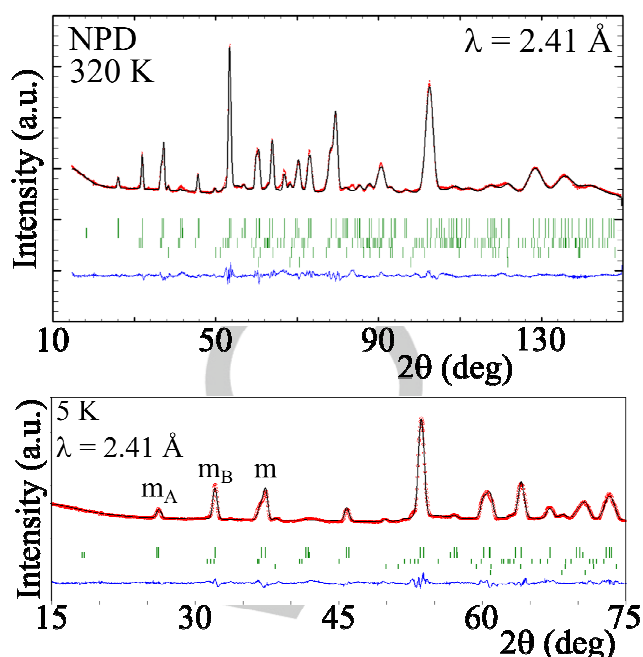


Figure 1. Rietveld fits of the DDPv structure of  $\text{CaCuFeReO}_6$  against NPD data at 320 K showing the full range of data (top) and at 5 K showing the low-angle magnetic peaks (bottom), labelled  $m_A/m_B/m$  to show peaks from spins at A/B/(A and B) type cation sites. Markers from top to bottom show (82(1) w.t.%),  $\text{CaCuFeReO}_6$  DPv (5.3(3) w.t.%),  $\text{ReO}_2$  (7.4(4) w.t.%) and Re (4.8(3)%).

The crystal structure refinements demonstrate that the present  $\text{CaCuFeReO}_6$  sample is slightly off-stoichiometric with refined composition  $\text{CaCu}_{0.85}\text{Fe}_{1.15}\text{ReO}_6$  and also with 7% Fe/Re antisite disorder. Fe shows a clear preference for substituting at the tetrahedral (Cu1) rather than the square planar (Cu2) site. Nevertheless the material shows a very high degree of order of four different metals over five cation sites, and demonstrates that the double double perovskite arrangement, previously reported only for Mn-based materials, can be stabilised at pressure by Ca/Cu ordering over A/A' columns.

Bond valence sum calculations using the 320 K bond distances in Table 1 give BVS values Ca = 2.3, Cu1 = 1.6, Cu 2 = 1.5, Fe = 2.6 and Re = 5.1. These are consistent with the ideal valence distribution  $\text{Ca}^{2+}\text{Cu}^{2+}\text{Fe}^{3+}\text{Re}^{5+}\text{O}_6$  which is analogous to those in related materials such as  $\text{Ca}^{2+}\text{Mn}^{2+}\text{Fe}^{3+}\text{Re}^{5+}\text{O}_6$ ,<sup>[23]</sup> allowing for disorder at some sites and structural strain induced by high pressure synthesis. Octahedral tilt angles calculated from Fe-O-Re angles in Table 1, along the *c* axis ( $\phi = 19.3(4)^\circ$ ) and within the *ab* plane ( $\theta = 17.4(2)^\circ$ ) are within the range of 15 to 19.5° for Mn-based analogues<sup>[20]</sup> and thus demonstrate that these substantial tilts are essential to stabilisation of this DDPv structure type.

The same ferrimagnetically ordered spin structure (Fig. 2a) was observed in  $\text{CaCuFeReO}_6$  from 5 to 550 K, and thermal variations of the refined magnetic moments are shown in Fig. 2b. The two independently refined magnetic moments (Fe/Re and Cu1/Cu2) both follow the same critical dependence  $\sim [1 - (T/T_C)]^\beta$  with fitted  $T_C = 566.70(3)$  K and  $\beta = 0.254(2)$ . The latter is close

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to the ideal mean field value of  $\beta = 0.25$  for a tricritical transition, on the cusp between first and second order behaviour. This may reflect the complex network of interactions between four distinct spin sublattices. Ordered moments at 5 K of 3.0 and 1.2  $\mu_B$  for Fe<sup>3+</sup> and Re<sup>5+</sup> (refined in a 5:2 ratio) are reduced from ideal values of 5 and 2  $\mu_B$  reflecting B/B' antisite disorder and spin-orbit coupling for Re<sup>5+</sup>. However the average moment at 5 K for the Cu sites of 1.4  $\mu_B$  exceeds the ideal 1  $\mu_B$  value for Cu<sup>2+</sup> showing that the Fe spins substituted at the Cu sites are also aligned.

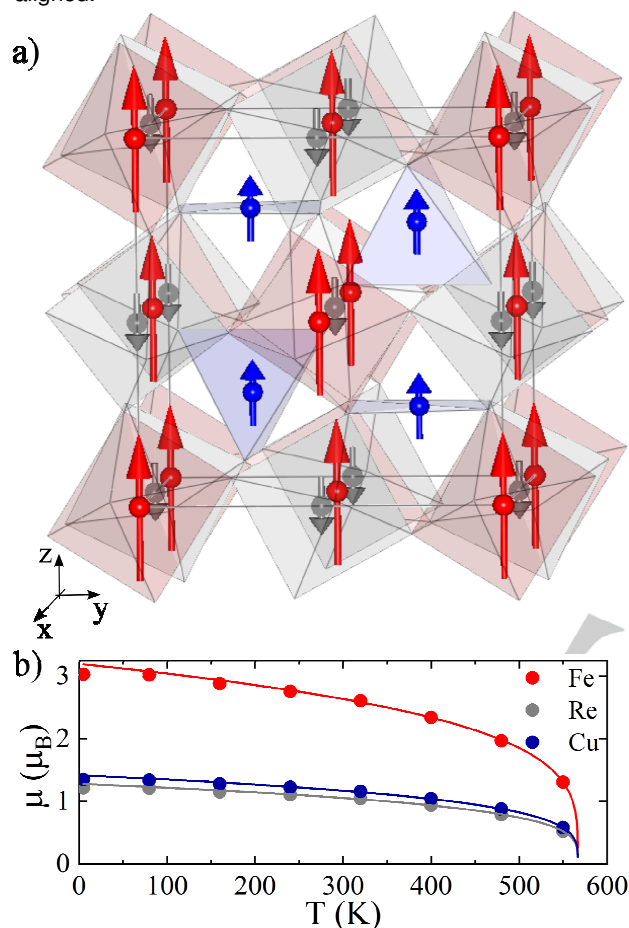


Fig. 2 a) Refined crystal and magnetic structures for CaCuFeReO<sub>6</sub> with Fe, Re and Cu magnetic moments as red, grey and blue arrows, respectively. b) Refined magnetic moments vs. temperature with critical functions fitted in the  $T_C/2 < T < T_C$  temperature range.

Physical properties of CaCuFeReO<sub>6</sub> were explored through magnetisation and resistivity measurements in zero and applied fields. Magnetic susceptibility, shown in Figure 3a, reveals a single Curie transition. The estimated  $T_C = 551$  K, from extrapolation of the steepest negative  $\chi$ -T slope to  $\chi = 0$ , is in agreement with the value of  $T_C = 567$  K from fitting the zero field neutron moments above. A Curie-Weiss fit to data at 600 – 625 K gives realistic results despite the narrow fitting range; an effective paramagnetic moment of  $\mu_{\text{eff}} = 6.52 \mu_B/\text{f.u.}$ , close to the spin-only estimate of 6.78  $\mu_B/\text{f.u.}$ , and a Weiss temperature of  $\theta = 522$  K, comparable to  $T_C$  and showing that 3D ferrimagnetic order occurs with little frustration. Magnetisation-field hysteresis loops confirm the ferrimagnetic order, with a saturated magnetic moment of 4  $\mu_B/\text{f.u.}$  (Fig. 3b) at 2 and 100

K identical to the value expected for order of  $S = 1/2 \uparrow \text{Cu}^{2+}$ ,  $S = 5/2 \uparrow \text{Fe}^{3+}$ , and  $S = 1 \downarrow \text{Re}^{5+}$  spins, confirming the arrangement derived from neutron diffraction. The saturated moment decreases to 2.7  $\mu_B$  at 400 K (see SI). Small coercive fields of 120 and 97 mT are observed at 2 and 100 K respectively, reflecting the weak anisotropy of the tetragonal DDPv structure.

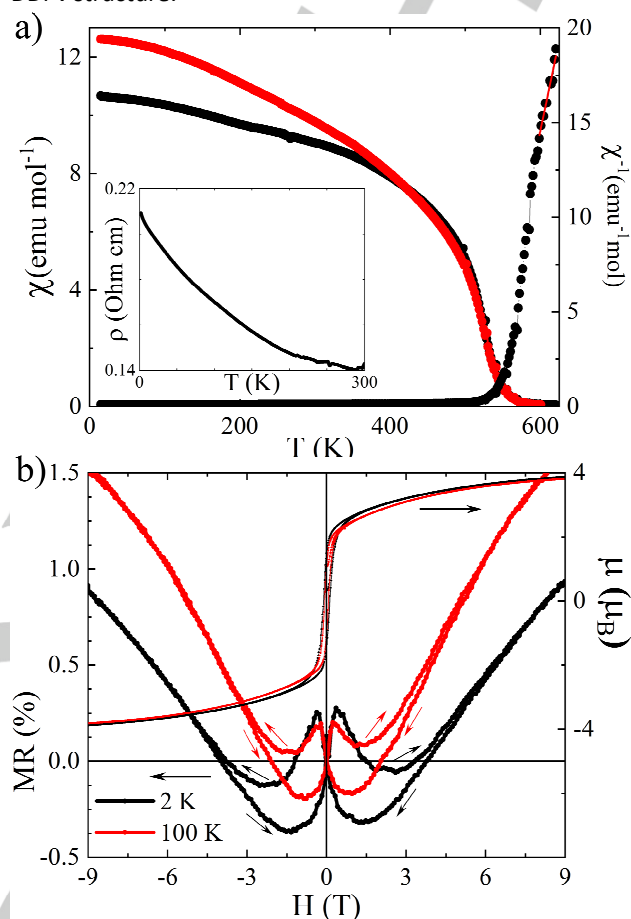


Figure 3. a) FC-ZFC (red-black) magnetic susceptibility (left) and inverse (right) of CaCuFeReO<sub>6</sub>. Inset shows T-dependence of resistivity. b) Hysteresis loops and butterfly-shaped magnetoresistance at 2 and 100 K.

The resistivity of a ceramic pellet of CaCuFeReO<sub>6</sub> is 140 m $\Omega$ ·cm at room temperature increasing to 210 m $\Omega$ ·cm to 4 K (Figure 2a). This change corresponds to an energy gap of  $\sim 1$  meV which is unrealistically small for intrinsic semiconducting behaviour suggesting that grain boundary resistances mask an underlying metallic conductivity. Magnetoresistance (MR) values in Fig. 3b are small, most likely due to the Fe/Re antisite disorder, but show interesting variations with field. The negative low field effect is a typical spin-valve-type magnetoresistance due to the intergrain tunneling of spin-polarized conduction carriers, as observed in Sr<sub>2</sub>FeMoO<sub>6</sub> and other ferromagnetic double perovskites. However a positive MR effect dominates at high fields. A similar switch from negative to positive MR was reported in the related Ca(Mn<sub>0.5</sub>Cu<sub>0.5</sub>)FeReO<sub>6</sub> DDPv<sup>[23]</sup> and in the DPv Mn<sub>2</sub>FeReO<sub>6</sub><sup>[28,29]</sup> due to canting of Fe and Re spins arising from frustration of their order with respect to the Mn spins as field increases, and a similar mechanism may operate here due to their interactions with the Cu site spins. Low field 'butterfly' MR hysteresis curves are



observed but the hysteresis in MR differs slightly from that in the magnetization as the peak-to-peak MR separations do not coincide with the coercive fields. The behavior is also observed in  $\text{Sr}_2\text{FeMoO}_6$  and the 1322-type analogue to  $\text{CaCuFeReO}_6$ ,  $\text{CaCu}_3\text{Fe}_2\text{Re}_2\text{O}_{12}$ ,<sup>[9]</sup> and for epitaxial  $\text{Fe}_3\text{O}_4$  films and  $\text{NiCo}_2\text{O}_4$ .<sup>[30,31]</sup>

$\text{CaCuFeReO}_6$  is an important discovery in the field of cation-ordered perovskites as it demonstrates that the  $P4_2/n$   $\text{AA}'\text{BB}'\text{O}_6$  double double perovskite (DDPv) structure already reported for many  $\text{RMnBB}'\text{O}_6$  and  $\text{CaMnBB}'\text{O}_6$  materials can also be stabilised for  $\text{A}' = \text{Cu}$ .  $\text{CaCuFeReO}_6$  is the first reported Mn-free DDPv. Synthesis of  $\text{CaCuFeReO}_6$  has proved more challenging than for the  $\text{CaMnFeReO}_6$  analogue.<sup>[23]</sup> The present sample of  $\text{CaCuFeReO}_6$  was recovered from 15.5 GPa and 1400 °C, has 82 % phase purity, with 7% Fe/Re antisite disorder and 30% Fe substituted at one of the Cu sites, whereas  $\text{CaMnFeReO}_6$ , prepared at 10 GPa and 1400 °C was >99% phase pure with 3% Fe/Re antisite disorder and 8 and 16% Fe at the two Cu sites. Nevertheless this study indicates that increasing pressure significantly above 10 GPa may enable many more  $\text{AA}'\text{BB}'\text{O}_6$  DDPvs to be synthesised for  $\text{A}' = \text{Cu}$  and perhaps other divalent  $\text{A}' = \text{Fe, Co, Ni...}$  cations, so the presently known phases likely represent only the 'tip of the iceberg' of DDPv materials.

Interesting differences between the physical properties of  $\text{CaCuFeReO}_6$  and  $\text{CaMnFeReO}_6$  are apparent. Both have high  $T_C$ 's for ferrimagnetic Fe/Re spin order, of 567 and 500 K, typical for  $\text{A}_2\text{FeReO}_6$  double perovskites, but the Cu spins also order at  $T_C$  in  $\text{CaCuFeReO}_6$  whereas the A-site Mn spins in  $\text{CaMnFeReO}_6$  order at a separate transition at  $T_A = 70$  K.<sup>[9]</sup> Order of  $\text{Mn}^{2+}$  spins at  $T_A = 75$  K, well below  $T_C = 520$  K, is also seen in the double perovskite  $\text{Mn}_2\text{FeReO}_6$ .<sup>[28,29]</sup> These differences evidence stronger Fe/Re-O-A' superexchange for  $\text{A}' = \text{Cu}$  compared to  $\text{A}' = \text{Mn}$ , and perhaps also a shift in relative values of the various exchange interactions so that A' spins in  $\text{CaCuFeReO}_6$  are less magnetically frustrated.  $\text{CaMnFeReO}_6$  has negative MR diverging to large values at low temperature and high field (-32% at 20 K and 7 T), typical for ferrimagnetic double perovskites like  $\text{Sr}_2\text{FeMoO}_6$ ,<sup>[7]</sup> whereas  $\text{CaCuFeReO}_6$  shows a switch from negative to positive MR with increasing field. A similar switch in  $\text{Mn}_2\text{FeReO}_6$  results from increased canting of Fe and Re spins due to frustration of their order with respect to the Mn spins as field increases. High field neutron diffraction experiments would be needed to verify the same effect in  $\text{CaCuFeReO}_6$ . It is also interesting to compare the mixed DDPv  $\text{Ca}(\text{Mn}_{0.5}\text{Cu}_{0.5})\text{FeReO}_6$ <sup>[23]</sup> to  $\text{CaA}'\text{FeReO}_6$  ( $\text{A}' = \text{Cu}$  and Mn).  $\text{Ca}(\text{Mn}_{0.5}\text{Cu}_{0.5})\text{FeReO}_6$  has  $T_C = 560$  K and a separate ordering transition at  $T_A = 150$  K unlike in  $\text{CaCuFeReO}_6$ , but the increase relative to  $T_A = 70$  K in  $\text{CaMnFeReO}_6$  demonstrates the effect of Cu in strengthening Fe/Re-O-A' superexchange. However  $\text{Ca}(\text{Mn}_{0.5}\text{Cu}_{0.5})\text{FeReO}_6$  displays a switch from negative to positive MR on warming, as seen in  $\text{CaCuFeReO}_6$  (e.g. by comparing 2 and 100 K MR values in a 3 T field in Fig. 3b) but not in  $\text{CaMnFeReO}_6$ . Such complex MR behaviour is therefore not directly linked to the presence (or absence) of a distinct ordering transition of A-site spins, but rather

to field effects on competing A'-O-B, A'-O-B' and B-O-B' interactions within  $\text{AA}'\text{BB}'\text{O}_6$  DDPvs with ordered A', B and B' spin sublattices.

Formation of  $\text{CaCuFeReO}_6$  is also important because a 1322 type phase  $\text{CaCu}_3\text{Fe}_2\text{Re}_2\text{O}_{12}$  (written here as  $\text{Ca}_{1/2}\text{Cu}_{3/2}\text{FeReO}_6$  for comparison) is also known and so this is the first demonstration that two different A-cation ordered perovskites can be prepared within the same chemical system. Comparison of their properties is again instructive.  $\text{CaCuFeReO}_6$  and  $\text{Ca}_{1/2}\text{Cu}_{3/2}\text{FeReO}_6$  have similar  $T_C$ 's of 567 and 560 K with Cu spins ordering parallel to Fe spins at  $T_C$  leading to large saturated magnetisations of 4.0 and 4.35  $\mu_B$  respectively, the latter value reflecting the extra 0.5  $\text{Cu}^{2+}$  spin per double perovskite unit. Both materials have apparent metallic conductivity but with resistive grain boundaries in ceramic samples. A high spin-polarization of conduction electrons was reported for  $\text{Ca}_{1/2}\text{Cu}_{3/2}\text{FeReO}_6$  and both materials display low-field 'butterfly' spin-valve MR effects although the magnitude of MR is small. However,  $\text{Ca}_{1/2}\text{Cu}_{3/2}\text{FeReO}_6$  has negative MR while  $\text{CaCuFeReO}_6$  shows an unusual switch to positive values at high field. Exploration and comparison of DDPv, 1322 type, and perhaps other types of A and B-site ordered perovskites in mixed transition metal oxide systems is likely to lead to discovery of spintronic and other types of interesting property.

In conclusion, discovery of  $\text{CaCuFeReO}_6$  marks an important milestone in development of A- and B- cation ordered perovskites, as it demonstrates that the tetragonal  $\text{AA}'\text{BB}'\text{O}_6$  double double perovskite structure, previously only reported for  $\text{A}' = \text{Mn}$ , can be stabilised by Cu and perhaps other metals at pressures above 10 GPa.  $\text{CaCuFeReO}_6$  occurs in the same phase field as 1322-type  $\text{CaCu}_3\text{Fe}_2\text{Re}_2\text{O}_{12}$  demonstrating that different A-cation ordering types may be obtained in the same chemical system. Ferrimagnetic Fe (up), Re (down) and Cu (up) spin order occurs in  $\text{CaCuFeReO}_6$  below  $T_C = 567$  K and magnetoresistance shows low-field spin-valve-type butterfly hysteresis with an unusual change from negative to positive values as field increases.

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**Keywords:** double perovskites • high-pressure chemistry • magnetic properties • solid-state structures

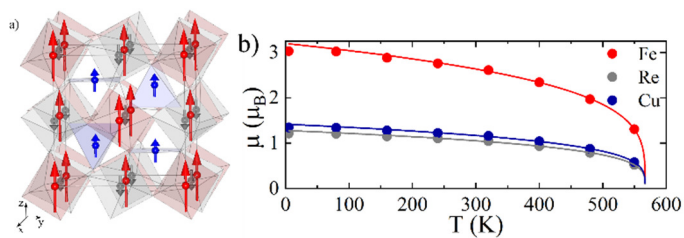
- 1 R. J. Tilley, *Perovskites: structure-property relationships*, John Wiley & Sons, **2016**.
- 2 J. P. Attfield, P. Lightfoot, R. E. Morris, *Dalton Trans.*, **2015**, 44, 10541-10542.
- 3 G. King, P. M. Woodward, *J. Mater. Chem.* **2010**, 20, 5785-5796.
- 4 M. T. Anderson, K. B. Greenwood, G. A. Taylor, K. R. Poeppelmeier, *Prog. Solid State Chem.* **1993**, 22, 197-233.

- 5 C. J. Howard, B. J. Kennedy, P. M. Woodward, *Acta Crystallogr. Sect. B: Struct. Sci.* **2003**, 59, 463-471.
- 6 S. Vasala, M. Karppinen, *Prog. Solid State Chem.* **2015**, 43, 1-36.
- 7 K.-I. Kobayashi, T. Kimura, H. Sawada, K. Terakura, Y. Tokura, *Nature* **1998**, 395, 677-680.

- 8 D. Serrate, J. De Teresa, M. Ibarra, *J. Phys.: Condens. Matter* **2006**, *19*, 023201.
- 9 W.-T. Chen, M. Mizumaki, H. Seki, M. S. Senn, T. Saito, D. Kan, J. P. Attfield, Y. Shimakawa, *Nat. Commun.* **2014**, *5*, 3909.
- 10 S. S. Ahmed, G. D. Ngantso, L. Bahmad, A. Benyoussef, A. E. Kenza, *Physics Letters A* **2018**, **382**, 186–192.
- 11 L. Gao, X. Wang, X. Ye, W. Wang, Z. Liu, S. Qin, Z. Hu, H.-J. Lin, S.-C. Weng, C.-T. Chen, P. Ohresser, F. Baudelet, R. Yu, C. Jin, Y. Long, *Inorg. Chem.* **2019**, *58*, 15529–15535.
- 12 X. Wang, M. Liu, X. Shen, Z. Liu, Z. Hu, K. Chen, P. Ohresser, L. Nataf, F. Baudelet, H.-J. Lin, C.-T. Chen, Y.-L. Soo, Y.-f. Yang, C. Jin, Y. Long, *Inorg. Chem.* **2019**, *58*, 320–326.
- 13 X. Ye, X. Wang, Z. Liu, B. Zhou, L. Zhou, H. Deng, Y. Long, *Dalton Trans.*, **2022**, *51*, 1745–1753.
- 14 S.-H. Byeon, M. W. Lufaso, J. B. Parise, P. M. Woodward, T. Hansen, *Chem. Mater.* **2003**, *15*, 3798–3804.
- 15 S.-H. Byeon, S.-S. Lee, J. B. Parise, P. M. Woodward, N. H. Hur, *Chem. Mater.* **2005**, *17*, 3552–3557.
- 16 H. Xiang, J. Wang, J. Meng, Z. Wu, *Computational Materials Science* **2009**, *46*, 307–309.
- 17 M. Senn, W.-t. Chen, T. Saito, S. García-Martín, J. P. Attfield, Y. Shimakawa, *Chem. Mater.* **2014**, *26*, 4832–4837.
- 18 Z. Liu, X. Wang, X. Ye, X. Shen, Y. Bian, W. Ding, S. Agrestini, S.-C. Liao, H.-J. Lin, C.-T. Chen, S.-C. Weng, K. Chen, P. Ohresser, L. Nataf, F. Baudelet, Z. Sheng, S. Francoual, J. R. L. Mardegan, O. Leupold, Z. Li, X. Xi, W. Wang, L. H. Tjeng, Z. Hu, Y. Long, *Phys. Rev. B*, **2021**, *103*, 014414.
- 19 Y.-Y. Yin, M. Liu, J.-H. Dai, X. Wang, L. Zhou, H. Cao, C. de la Cruz, C.-T. Chen, Y. Xu, X. Shen, R. Yu, J. A. Alonso, A. Muñoz, Y.-F. Yang, C. Jin, Z. Hu, Y. Long, *Chem. Mater.* **2016**, *28*, 8988–8996.
- 20 E. Solana-Madruga, Á. M. Arévalo-López, A. J. Dos Santos-García, E. Urones-Garrote, D. Ávila-Brandé, R. Sáez-Puche, J. P. Attfield, *Angew. Chem. Int. Ed.* **2016**, *55*, 9340–9344.
- 21 E. Solana-Madruga, Á. M. Arévalo-López, A. J. Dos Santos-García, C. Ritter, C. Cascales, R. Sáez-Puche, J. P. Attfield, *Phys. Rev. B* **2018**, *97*, 134408.
- 22 K. Ji, Y. Yuan, G. T. Moy, C. Ritter, J. P. Attfield, *J. Solid State Chem.* **2022**, *313*, 123329.
- 23 G. M. McNally, Á. M. Arévalo-López, P. Kearins, F. Orlandi, P. Manuel, J. P. Attfield, *Chem. Mater.* **2017**, *29*, 8870–8874.
- 24 G. M. McNally, Á. M. Arévalo-López, F. Guillou, P. Manuel, J. P. Attfield, *Phys. Rev. Mater.* **2020**, *4*, 064408.
- 25 E. Solana-Madruga, Y. Sun, Á. M. Arévalo-López, J. P. Attfield, *Chem. Commun.* **2019**, *55*, 2605–2608.
- 26 P. Kearins, E. Solana-Madruga, K. Ji, C. Ritter, J. P. Attfield, *J. Phys. Chem. C* **2021**, *125*, 9550–9555.
- 27 E. Solana-Madruga, P. S. Kearins, K. N. Alharbi, C. T. Lennon, C. Ritter, J. P. Attfield, *Phys. Rev. Mater.* **2021**, *5*, 054412.
- 28 M. Rong Li, M. Retuerto, Z. Deng, P. W. Stephens, M. Croft, Q. Huang, H. Wu, X. Deng, G. Kotliar, J. Sánchez-Benítez, J. Hadermann, D. Walker, M. Greenblatt, *Angew. Chem. Int. Ed.* **2015**, *54*, 12069–12073.
- 29 A. M. Arévalo-López, G. M. McNally, J. P. Attfield, *Angew. Chem. Int. Ed.* **2015**, *54*, 12074–12077.
- 30 P. Li, L. T. Zhang, W. B. Mi, E. Y. Jiang, H. L. Bai, *J. App. Phys.* **2009**, *106*, 033908.
- 31 P. Li, C. Xia, D. Zheng, P. Wang, C. Jin, H. Bai, *Phys. Status Solidi RRL* **2016**, *10*, 2, 190–196.

## COMMUNICATION

## Entry for the Table of Contents



CaCuFeReO<sub>6</sub> adopts the tetragonal AA'BB'O<sub>6</sub> double double perovskite structure that was previously only reported for A' = Mn compositions. Ferrimagnetic order of Fe, Re and Cu spins is observed at  $T_C = 567$  K and the magnetoresistance displays low-field 'butterfly' hysteresis with an unusual change from negative to positive values as field increases.