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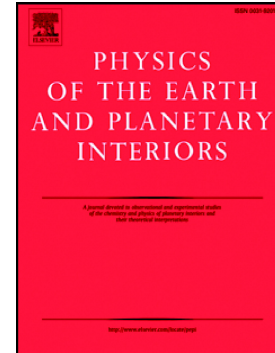
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## High pressure dissociation of $\text{CaTiO}_3$ perovskite into $\text{CaO}$ and $\text{CaTi}_2\text{O}_5$

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### Highlights:

- Phase diagram of the calcium titanate perovskite  $\text{CaTiO}_3$  up to 170 GPa and 4500 K.
- Thermal equation of state of the orthorhombic ( $Pbnm$ ) perovskite  $\text{CaTiO}_3$ .
- Dissociation of the  $\text{CaTiO}_3$  perovskite into  $\text{CaO}$  and  $\text{CaTi}_2\text{O}_5$  without going through a post-perovskite phase

### Key words:

Perovskite,  $\text{CaTiO}_3$ , Equation of State, Dissociation, High-pressure

### Abstract

In this study, we investigated phase transformations of  $\text{CaTiO}_3$  perovskite using x-ray diffraction at high pressure and high temperature up to 170 GPa and 4500 K in a laser-heated diamond-anvil cell. We report a high-pressure dissociation of  $\text{CaTiO}_3$  into  $\text{CaO}$ -B2 and  $\text{CaTi}_2\text{O}_5$  with a monoclinic  $P2/m$  structure, instead of the expected transformation of the orthorhombic distorted perovskite structure into a post-perovskite phase. We propose that this transition may be favored by the  $B1$  to  $B2$  phase change of  $\text{CaO}$  at around 60 GPa. In order to

provide additional information on the high pressure properties of  $\text{CaTiO}_3$  perovskite, we measured its melting temperature using  $\text{CO}_2$  laser heated diamond anvil cell up to 55 GPa yielding a fit of the melting curve to a Kraut-Kennedy empirical law according to :  $T_m(K) = 2188 * [1 + 4.23 * (\Delta V/V_0)]$ . To provide some further insight into the thermodynamic properties of  $\text{CaTiO}_3$ , we determined the P-V-T equation of state of the orthorhombic mineral perovskite, fitted by using a third order Birch-Murnaghan equation of state and a Berman thermal expansion model. The fit of the data yields to  $K_0 = 180.6(4)$  GPa,  $K'_0 = 4$  (fixed),  $\partial K/\partial T = -0.022(1)$  GPa  $\text{K}^{-1}$ ,  $\alpha_1 = 3.25(5) \times 10^{-5} \text{ K}^{-1}$ ,  $\alpha_2 = 1.3(1) \times 10^{-8} \text{ K}^{-2}$

### Plain language summary

Among 5000+ discovered exoplanets, some of them are classified as super-Earths. Based on their average densities derived from mass/radius observations, these planets are expected to be rocky but have a larger radius than that of the Earth and thus higher internal pressures. Bridgmanite, the orthorhombic perovskite structured compound with  $(\text{Mg,Fe})\text{SiO}_3$  composition, considered to be the most abundant mineral in the Earth's rocky mantle is likely abundant in super-Earths as well. Atomistic calculations have predicted that  $\text{MgSiO}_3$  dissociates between 1 to 3 terapascal ( $1 \text{ TPa} = 10^{12} \text{ Pa} = 10 \text{ Mbar}$ ) into oxide phases but such pressures remain outside of the realm of high pressure experiments. In this work, we focus on the archetypal perovskite structure  $\text{CaTiO}_3$ , and investigate its phase diagram using the laser-heated diamond anvil cell technique coupled with *in situ* x-ray diffraction. At pressures above 60 GPa and temperatures up to 4500 K, we show evidence of the dissociation of  $\text{CaTiO}_3$  into  $\text{CaO}$  and  $\text{CaTi}_2\text{O}_5$  instead of a transformation into a denser post-perovskite phase. Similar decomposition schemes have been proposed for  $\text{MgSiO}_3$  at ultra-high pressures in super-Earth's interiors; the decomposition reaction discovered here in  $\text{CaTiO}_3$  could allow to better understand their consequences by providing a model amenable to experiments under accessible pressures.

## 1 Introduction

Bridgmanite, the orthorhombic (*Pbnm* space group) perovskite-structured compound with  $(\text{Mg,Fe})\text{SiO}_3$  composition is considered to be the most abundant mineral in the Earth's rocky mantle. This mineral undergoes a well-documented transition to a *Cmcm*  $\text{CaIrO}_3$ -type post-perovskite phase at approximately 120 GPa (Murakami et al. 2004; Oganov & Ono, 2004). For super-Earth modelling, it is important to investigate  $\text{MgSiO}_3$  behaviour at much higher pressures and temperatures. Atomistic calculations have predicted that  $\text{MgSiO}_3$

dissociates between 1 to 3 TPa into simple oxides MgO and SiO<sub>2</sub> or more complex assembly of denser silicates and oxides (Mg<sub>2</sub>SiO<sub>4</sub>, MgSi<sub>2</sub>O<sub>5</sub>) (Umemoto et al., 2011, 2017, 2019 and references therein; Wu et al. 2014; Niu et al. 2015). However, such pressures remain outside of the realm of static high-pressure experiments and difficult to achieve together with relevant temperatures in dynamic compression experiments. Therefore, to experimentally acquire better knowledge about these important mechanisms and transitions, another approach is to study analogue compounds. An analogue material usually mimics the crystallographic structure of the material while having a different composition. Several compounds such as CaRhO<sub>3</sub>, MgGeO<sub>3</sub>, NaMgF<sub>3</sub>, GdFeO<sub>3</sub> and much more have been identified as model structures to explore high-density high-pressure evolution of ABX<sub>3</sub> compounds (Xu et al., 2015 and references herein). In this work, we focus on the mineral called perovskite (and archetypal perovskite structure) CaTiO<sub>3</sub>, which can be interesting to compare with the perovskite-structured materials of rocky planets such as MgSiO<sub>3</sub> and CaSiO<sub>3</sub> perovskite. Since CaTiO<sub>3</sub> has the *Pbnm* orthorhombic perovskite structure at ambient conditions, it is expected to undergo structural transitions at more moderate pressures than bridgmanite or CaSiO<sub>3</sub>. It has been also predicted that CaTiO<sub>3</sub> may undergo a series of transition perovskite (Pv) – post-perovskite (pPv) – post-post perovskite (ppPv) (Xu et al. 2015) that makes of this compound a very interesting case to study.

Perovskite and post-perovskite structures have been extensively studied during these past few years both by *ab initio* calculations and experiments (e.g. Murakami et al., 2004; Hirose et al., 2017 and references therein; Dutta et al., 2019; Umemoto et al., 2011, 2017, 2019; Xu et al., 2015; Tsuchiya et al., 2004). For CaTiO<sub>3</sub>, first-principles calculations by Xu et al. (2015) propose that the *Pbnm* perovskite structure transforms to a post-perovskite *Cmcm* CaIrO<sub>3</sub> phase at 22.7 GPa, followed by another transition to a post-post-perovskite structure at 71.7 GPa. Using diamond anvil cells at room-temperature, it has been demonstrated by Guennou et al. (2010) that CaTiO<sub>3</sub> remains under the *Pnma* (or equivalent *Pbnm*) orthorhombic perovskite structure up to 60 GPa. With CaTiO<sub>3</sub>, we focus on a class of materials in which the structural distortion is dominated by tilts of BO<sub>6</sub> octahedra (Zhao et al. 2011). It is interesting to identify the point where this structure destabilizes, for instance when the tilt angle ceases to evolve and bond compression becomes the main structural modification. The absence of heating in Guennou et al. (2010) may have, possibly, retained a metastable structure at the highest pressures. A dynamic compression study (Trunin et al.

1971) suggested a phase change in  $\text{CaTiO}_3$  at 60 GPa without giving any structural information for the new state.

In the present study, we investigate the phase diagram of calcium titanate  $\text{CaTiO}_3$  perovskite using the laser-heated diamond anvil cell technique coupled with *in situ* x-ray diffraction. We report the results of experiments conducted up to 170 GPa and temperatures up to 4500 K. We measure the thermal equation of state of the *Pbnm* orthorhombic perovskite phase. We also observe a gradual dissociation of  $\text{CaTiO}_3$  into oxides instead of transformations into post-perovskite or post-post-perovskite phases.

## 2 Methods

### 2.1 X-ray diffraction experiment

The *in-situ* experiments were carried out at the ID27 beamline at the European Synchrotron Radiation Facility (ESRF) using laser-heated diamond anvil cell (LH-DAC) methods. We characterised the samples using powder x-ray diffraction in angle-dispersive mode using an MCCD detector (Mezouar et al. 2000; Mezouar et al, 2017). In this setup, the laser heating system is composed of two high-power diode-pumped Nd-YAG infrared lasers entering the DAC from both sides with an angle of  $20^\circ$ . Temperatures were measured by multi-wavelength spectral radiometry methods (Schultz et al. 2005). The working wavelength was  $0.3738 \text{ \AA}$  (i.e., 33 keV) and remained the same during the whole experiment. We conducted eight runs in total (Table 1) with two types of starting materials. The first three runs were done using commercial synthetic  $\text{CaTiO}_3$  (purchased from Kurt J. Lesker Company with a purity of 99.9%), grinded and loaded without any pressure transmitting medium but with platinum black powder that served as IR laser heating absorber as well as pressure calibrant (series #1, #2 and #3). The following four runs were carried out with a natural  $\text{CaTiO}_3$  sample (Collection of Minerals, Sorbonne University, courtesy from J.C. Boulliard) compressed between KCl disks acting as thermal insulating pressure medium and pressure calibrant (series #4, #5, #6 and #7). X-ray diffraction patterns collected at ambient conditions confirmed the orthorhombic structure and the purity of the  $\text{CaTiO}_3$  sample, with an excellent match with reference diffraction patterns obtained on a single crystal by Guennou et al. (2010). In the natural  $\text{CaTiO}_3$  samples, trace elements in the crystal structure enabled significant YAG laser ( $1.064 \text{ \mu m}$ ) absorption and heating. The last experimental run was conducted with a natural  $\text{CaTiO}_3$  sample loaded with neon as a pressure medium and a small piece of gold was added for pressure measurements (series #8).

X-ray diffraction images were collected on a MAR165 2D-CCD detector with exposure time of the order of 30 seconds at ambient temperature conditions, and up to 90 seconds at the most extreme conditions. Each CCD image was then processed with the DIOPTAS software (Prescher & Prakapenka, 2015) to obtain intensity versus 2-theta angle integrated patterns. These patterns were subsequently treated using the FullProf Suite software (Rodriguez-Carvajal, 2001) with a Le Bail refinement. Pressures corresponding to each diffractogram were calculated using different equations of state of platinum (Matsui et al. 2009), gold (Takemura & Dewaele, 2008) and KCl (Dewaele et al. 2012), as described in supplementary materials. The EoSFit program (Angel et al., 2014; Gonzales-Plata et al., 2016) was used to fit the thermal equation of state parameters of  $\text{CaTiO}_3$  with the perovskite  $Pbnm$  structure.

## 2.2 Melting experiment

To provide additional information on the high-pressure behaviour of  $\text{CaTiO}_3$ , we performed *in situ* laser-heated diamond anvil cell experiments to determine the melting curve of  $\text{CaTiO}_3$  up to 60 GPa. These experiments were carried out at IMPMC laboratory, Sorbonne University. Single pieces of synthetic  $\text{CaTiO}_3$  (same sample as in the run #1-3 Table 1) were loaded in DACs equipped with anvils of 200 and 300  $\mu\text{m}$  culet sizes to reach pressures ranging between 16 and 55 GPa. The sample chamber drilled in the pre-indented Re-gasket of the DAC assembly was loaded with sample pieces of about 10-20  $\mu\text{m}$  thicknesses which were sandwiched between the pressure medium discs of KCl of 10-20  $\mu\text{m}$  thickness. KCl acted as pressure transmitting medium, but also as a thermal insulator and chemical barrier between the sample and the diamond anvils. A ruby sphere was placed at the edge of the sample chamber to measure pressure with ruby fluorescence. Pressures were measured before and after each heating run. Our experimental setup is equipped with one-sided  $\text{CO}_2$  laser heating system which is built with a Synrad Evolution 125  $\text{CO}_2$  laser delivering a gaussian beam (wavelength 10.6  $\mu\text{m}$ ,  $M^2 < 1.2$ ) focused by a ZnSe lens on the sample through the diamond in an axial geometry (focal length 25 mm). The use of a  $\text{CO}_2$  laser permits to heat non-metallic samples (*i.e.*, wide bandgap insulating materials) such as  $\text{CaTiO}_3$  without any additional laser absorber such as Pt black. Heating is also effective across the entire thickness of the sample, with lesser axial temperature gradient (Smith et al., 2018). Temperature measurements were performed by spectro-radiometry, *i.e.* the analysis of thermal emission of the sample in the visible region (from 500 to 800 nm). The imaging and pyrometry system was made of a two-mirrors Schwarzschild reflective objective lens (focal 850 mm) used to

make an image of the hot spot on the entrance slit of the spectrometer (Acton 500i) equipped with an Andor-CDD camera. Optical pyrometry in the 500-800 nm range typically yields Planck's law fits within 10 K (uncertainty of the fit), which corresponds to an actual temperature uncertainty of about 130 K when emissivity changes and temperature variation are taken into account. Melting was detected by observation of a discontinuity in the curve of temperature vs laser power. This discontinuity in the absorption of the laser radiation, appears as a sudden increase of temperature at a given laser power, temperature which reaches then a so called 'plateau' attributed to melting. In few cases the laser power was slowly increased until the temperature reached a stable value ("plateau") due to melting at a given pressure (*e.g.* Boehler, 2000; Pradhan et al., 2015; Geballe & Jeanloz, 2012). We managed to fit the melting points using both the Kraut-Kennedy equation (Kraut & Kennedy, 1966) and the Simon equation (Simon, 1937). The formulation of the Kraut-Kennedy equation is:

(1)

$$T_m = T_m^0 \left( 1 + C \frac{\Delta V}{V_0} \right)$$

with  $T_m$  the melting temperature at a given pressure and  $T_m^0$  is the melting temperature at 1 atm.  $C$  is a constant depending on the material. The isothermal dilatation  $\Delta V/V_0$  is defined as:

(2)

$$\frac{\Delta V}{V_0} = \frac{V_0 - V}{V}$$

where  $V_0$  is the volume at ambient pressure and temperature.

The Simon equation is formulated as:

(3)

$$P_m = A \left[ \left( \frac{T_m}{T_m^0} \right)^B - 1 \right]$$

with  $T_m$  the melting temperature at a given pressure and  $T_m^0$  is the melting temperature at 1 atm.  $A$  and  $B$  are constants that depend on the material.

### 3 Results and discussion

#### 3.1 CaTiO<sub>3</sub> thermal equation of state



To establish the thermal equation of state of  $\text{CaTiO}_3$  perovskite, we used different data sets. Guennou et al. (2010) provide a reliable equation of state at ambient temperature up to 60 GPa obtained by single crystal x-ray diffraction using fluorescence of ruby as pressure gauge and helium as pressure-transmitting medium. However, above 40 GPa, they noticed a broadening of the peaks in the diffraction pattern attributed to a non-hydrostatic compression of the single crystal sample. Accordingly, we only included their data from 0 to 40 GPa in the refinement. In this work, we present a room-temperature dataset that can be compared to Guennou et al. (2010) in the same pressure range. The two datasets agree up to 47 GPa and we discuss below the evolution at higher pressures.

We coupled the Guennou et al. (2010) results with our data obtained up to temperatures in excess of 3000 K at 60 GPa. Most of the dataset comes from the series with platinum (see “Data EoS” in repository 10.17632/nmkzfcch752). In the refinement process, we avoided to include data that showed signs of structural transition to other phases (typically above 60 GPa). We also used the most recent data from Yashima et al. (2009) in the refinement. This work established the high-temperature thermal expansion of  $\text{CaTiO}_3$  perovskite and temperature-induced transition processes at ambient pressure up to 1720 K by neutron diffraction and Rietveld analysis refinement.

The refinement at  $T_0 = 300$  K was achieved with the software EoSFit (Angel et al., 2014) using a third order Birch-Murnaghan equation of state:

(4)

$$P(V, T_0) = \frac{3}{2} K_0 \left[ \left( \frac{V}{V_0} \right)^{-\frac{7}{3}} - \left( \frac{V}{V_0} \right)^{-\frac{5}{3}} \right] \left[ 1 + \frac{3}{4} (K'_0 - 4) \left[ \left( \frac{V}{V_0} \right)^{-\frac{2}{3}} - 1 \right] \right]$$

$V_0$  is the lattice volume at 300 K,  $V$  is the volume at a pressure point,  $K_0$  is the bulk modulus and  $K'_0$  is the first pressure derivative of the bulk modulus ( $\partial K / \partial P$ ).

In order to determine the pressure considering the temperature we introduced a thermal pressure term, according to:

(5)

$$P^{tot}(V, T) = P(V, T_0) + P^{th}(V, T)$$

To determine the thermal pressure, we used a Berman model:

(6)

$$\alpha \approx \alpha_0 + \alpha_1(T - T_0)$$

$\alpha$  is the thermal expansion. Once the thermal expansion is calculated, we have to consider a temperature dependent bulk modulus written as:

(7)

$$K_T = K_0 + \frac{\partial K}{\partial T}(T - T_0)$$

Then, the thermal pressure can be expressed as:

(8)

$$P^{th}(V, T) = \int_{T_0}^T \alpha K_T dT$$

The results are presented in Table 2 and Figure 1a. Our preferred model is  $V_0 = 33.62$  cm<sup>3</sup>/mol/f.u.,  $K_0 = 180.6$  GPa,  $K'_0 = 4$ ,  $\alpha_0 = 3.245 \times 10^{-5}$  K<sup>-1</sup>,  $\alpha_1 = 1.254 \times 10^{-8}$  K<sup>-2</sup> and  $\partial K / \partial T = -0.022$  GPa.K<sup>-1</sup>. All parameters were refined simultaneously except  $V_0$  which was fixed to the value reported by Guennou et al. (2010) which was obtained with high precision measurement on a single crystal. The refined  $K'_0$  was found very close to 4 (4.06). We decided to fix it to 4 in the final refinement process.

We also calculated the relative error between the volume measured during the experiments and obtained by refining the diffraction patterns and the volume calculated with the obtained equation of state. The minimum and maximum error we obtained were 0.7% and -1.2%, respectively, which corresponds to a volume difference of approximately 0.199 and -0.334 cm<sup>3</sup>/mol/f.u. (Figure 1b). The median error of our data is -0.07%. Thus, we can consider that the parameters of the thermal equation of state of CaTiO<sub>3</sub> are well defined with a  $\chi^2$  of 1.03.

By comparing our P-V-T data to other thermal equation of states, we extend the CaTiO<sub>3</sub> dataset to higher pressure and temperature and find a good agreement with other perovskite structures (Table 3). Our results on the thermal equation of state are comparable to thermoelastic parameters from other CaTiO<sub>3</sub> perovskite studies (Guennou et al. 2010, Webb et al. 1999, Kung & Rigden 1997, Fischer et al. 1993) done with different techniques (i.e.

ultrasonic interferometer, single crystal XRD in LH-DAC and powder XRD in LH-DAC) and extend the studied pressure range. The bulk modulus obtained in the present study is slightly higher but in good agreement with other studies (Webb et al. 1999). They found the same  $dK/dT$  as in the present study; however, they refined their thermal equation of state with constant thermal expansion ( $\alpha_0$ ) at ambient pressure whereas we include a temperature dependence term ( $\alpha_0$  and  $\alpha_1$ ) for a better fit. When refined with a constant thermal expansivity at 1 bar for comparison, we obtain a value of the same order ( $\alpha=3.80 \cdot 10^{-5} \text{ K}^{-1}$ ) as in Webb et al. (1999). Compared to other titanate perovskites (such as  $\text{SrTiO}_3$  or  $\text{BaTiO}_3$ ) (Guennou et al. 2010, Pruzan et al. 2002, Webb et al. 1999, Fischer et al. 1993),  $\text{CaTiO}_3$  is less compressible than  $\text{SrTiO}_3$  and  $\text{BaTiO}_3$ , which can be at first order a consequence of the cation size.

Importantly, when compared to other perovskite structures present in the Earth mantle (*i.e.*,  $\text{MgSiO}_3$ ,  $\text{CaSiO}_3$ ) (Tange et al. 2012, Ricolleau et al. 2009, Fiquet et al. 2000, Shim et al. 2000),  $\text{CaTiO}_3$  is more compressible but its compressibility diminishes less than  $\text{MgSiO}_3$  and  $\text{CaSiO}_3$  as temperature increases.

These data have implications for planetary science. They confirm that perovskite-structured titanates are less incompressible than perovskite-structured silicates (table 2) opening an interesting crystal-chemical perspective to better understand the effect of substitutions of silicon in the perovskite octahedral site. Moreover, such measurements open the way to a future thermodynamic modeling of  $\text{CaTiO}_3$  perovskite at high pressure and temperature which might have some relevance to better understand the behaviour of titanium at lower mantle conditions.

### 3.2 Orthorhombic and cubic perovskite structures of $\text{CaTiO}_3$

At ambient pressure and high temperature, the orthorhombic  $Pbnm$  phase of  $\text{CaTiO}_3$  is stable up to 1500K ( $\pm 30\text{K}$ ) (Naylor & Cook, 1946; Guyot et al., 1993; Redfern, 1996; Yashima & Ali, 2009) and further transforms into a cubic phase (space group  $Pm-3m$ ) at higher temperature. At ambient temperature and high-pressure, Guennou et al. (2010) observed that the  $Pbnm$  structure was stable at least up to 60 GPa, a pressure range that we have extended up to 110 GPa in the present study (Figure 2a). However, we observe a discontinuity of the  $c$  axis in the compression behaviour of the orthorhombic perovskite  $Pbnm$  structure at 60 GPa with a change in slope of the evolution of the unit cell parameter as a function of pressure (Figure 2b). Each cell parameter was obtained from a LeBail fit to our diffractograms using the FullProf software. In figure 2b, cell parameters are presented in a

pseudo-cubic lattice geometry for easier comparison. This discontinuity in compression cannot be ascribed to non-hydrostaticity in the Ne pressure medium which has no reported rheological nor phase change in this pressure range, but rather reveals a change in compression mechanism, possibly associated to a thermodynamic path that progressively drives the perovskite structure outside its stability field. At room-temperature though, it is likely that energetic barriers cannot be easily overcome to reach a new thermodynamic stable state. Up to 60 GPa in Figure 2b,  $\text{CaTiO}_3$  is still stable. Above 90 GPa, a axis becomes almost incompressible. Also, b axis remains constant on nearly the whole compression up to 100 GPa. However, it is clear that c axis experiences a change in compression above 60 GPa. It is striking to note that these changes in the compression slopes correspond to pressures at which a major transition is observed in  $\text{CaTiO}_3$  at high pressure and temperature (see below). Such behaviour has been described in other systems: a strongly distorted metastable phase is described for instance in cold compressed forsterite (Finkelstein et al., 2014).

The structure of the orthorhombic  $Pbnm$  perovskite is made of tilted  $\text{TiO}_6$  octahedra and of strongly distorted  $\text{CaO}_{12}$  dodecahedral sites. It is known from many studies (Glazer, 1972; Andrault & Poirier, 1991; Angel et al., 2005) that tilt and distortions change as a function of pressure and temperature and that the orthorhombic structure tends to transform to a cubic structure with increasing temperature. At pressure below 60 GPa, as temperature increases, we indeed observe a transformation of the  $Pbnm$  orthorhombic  $\text{CaTiO}_3$  perovskite into a more symmetric cubic structure (Figure 3). However, in most diffraction patterns, we still detect relics of the orthorhombic structure mixed with the cubic phase. These relics are seen in the form of a doubling of the peaks corresponding to the cubic phase with no appearance of the orthorhombic peaks between 2.5 Å ( $10^\circ 2\theta$ ) and 1.9 Å ( $11.5^\circ 2\theta$ ) and can be attributed to a gradient of temperature during the experiment. A tetragonal phase ( $I4/mcm$ ) has been reported in literature (e.g., Guyot et al., 1993; Redfern, 1996; Yashima & Ali, 2009) at intermediate temperatures between orthorhombic and cubic phases. Its restricted temperature stability range may however explain why it was not observed in the present study. According to Redfern (1996) or Yashima & Ali (2009), the existence of a tetragonal phase ( $I4/mcm$ ) between the perovskite and the cubic domain is shown in the diffractograms by the disappearance of the  $Pbnm$  reflections 120, 210, 121 and 103 in the d-spacing range between 2.37 and 2.10 Å, which only leaves the 211 reflection. It is clear from our diffractograms that the 211 reflection was never observed, so we conclude that either the tetragonal phase only exists in a narrow range of temperature that we missed here, or alternatively that the

tetragonal domain no longer exists at high pressure. A similar transition detailed by Komabayashi et al. (2007) is observed in  $\text{CaSiO}_3$  perovskite which transforms to a  $Pm\bar{3}m$  cubic phase, from a  $I4/mcm$  tetragonal phase, with a slightly positive  $dT/dP$  slope (1.1 K/GPa) and at relatively low temperature ( $\approx 500$  K) compared to  $\text{CaTiO}_3$  in the same range of pressure. The study also shows that the incorporation of aluminium stabilizes the tetragonal phase to higher temperature (Kurashina et al. 2004). Thomson et al. (2019) found the same stabilization process with the incorporation of titanium in  $\text{CaSiO}_3$  which brings an interesting analogy between  $\text{CaTiO}_3$  and  $\text{CaSiO}_3$ .

By comparing the measured unit cell parameters of the cubic perovskite phase of  $\text{CaTiO}_3$  and those calculated using the thermal equation of state of the orthorhombic perovskite, we found that the cubic phase is slightly smaller than the  $Pbnm$  phase extrapolated at these temperatures ( $\approx 3500$  K). Indeed, the difference between the calculated and measured volume of the cubic phase, using the perovskite thermal equation of state in the same condition, is around -8% in volume (Table S1).

### 3.3 Melting of $\text{CaTiO}_3$ perovskite

Several studies have attempted to determine the melting temperature of  $\text{CaTiO}_3$  at ambient pressure providing estimates ranging between 2188 K and 2262 K (Coughanour et al., 1954; DeVries et al., 1954; R. S. Roach, 1958, Jongejean & Wilkins, 1970; V. Daněk & I. Nerád, 2002). To anchor the melting curve at 1 bar, we have retained the value provided by Coughanour et al., i.e. 2188 K which was also selected by Robie et al. 1978. In Figure 4, we present the fitted solid-liquid line using two models for comparison, one from Kraut & Kennedy, 1956 and one from Simon, 1937.

(9)

$$T_m = 2188 * \left[ 1 + 4.23 * \left( \frac{\Delta V}{V_0} \right) \right] \quad (\text{Kraut} - \text{Kennedy})$$

(10)

$$Pm = 65.66 * \left[ \left( \frac{T_m}{T_0} \right)^{1.05} - 1 \right] \quad (\text{Simon})$$

Residuals of the two fits are given in Table S2.

Those melting results are compared with published experimental or calculated melting curves of  $\text{MgSiO}_3$  (Zerr & Boehler, 1993; DiPaola & Brodholt, 2016) and  $\text{CaSiO}_3$  (Zerr et al., 1997; Braithwaite & Stixrude, 2019) silicate perovskites. As shown in Figure 4, in the pressure range between 20 and 60 GPa, the melting curve of bridgmanite determined using a liquid-solid interface atomistic simulations (DiPaola & Brodholt, 2016) is indeed very close to the experimental melting curve of  $\text{CaTiO}_3$ . However, the experimental data from Zerr & Boehler (1993) suggest a different behaviour of  $\text{MgSiO}_3$  with a steeper slope and higher melting temperatures.  $\text{CaSiO}_3$  perovskite has a higher melting temperature than other perovskites and it has been determined both experimentally using  $\text{CO}_2$  LH-DAC (Zerr et al., 1997) and using DFT calculations (Braithwaite & Stixrude, 2019). Experimental slopes of melting curves of  $\text{MgSiO}_3$  and  $\text{CaSiO}_3$  appear steeper than in  $\text{CaTiO}_3$ . Unfortunately, to our best knowledge, no other melting studies on titanates (i.e.  $\text{BaTiO}_3$ ,  $\text{SrTiO}_3$ ,  $\text{MnTiO}_3$ ) at high pressure and temperature have been conducted. Therefore, the present work provides the first insights on the melting of titanate perovskites at high pressures.

It is interesting to note that our results on the melting of  $\text{CaTiO}_3$  perovskite under pressure suggest that titanate and silicate perovskites have melting temperatures of the same order of magnitude which could imply relatively weak dependencies of the melting temperature of mantle perovskites upon substitution of silicon in the octahedral sites. But these speculations remain to be confirmed experimentally and by ab initio simulations.

### 3.4 Dissociation of $\text{CaTiO}_3$ into $\text{CaO}$ and $\text{CaTi}_2\text{O}_5$ .

At pressures exceeding 50-60 GPa, x-ray diffraction patterns of  $\text{CaTiO}_3$  perovskite drastically change upon heating (Figure 5), which suggests a transition to another phase or an assemblage of new phase(s). We first analysed the high pressure diffractograms ranging from 90 to 170 GPa which showed no traces left of the former *Pbnm* orthorhombic initial perovskite structure. We used the tool DICVOL (Louër & Boulton, 2004) in the FullProf Suite to test several possible structures (space groups *Pnmm*, *Pnma*, *P2<sub>1</sub>/m*, *P2<sub>1</sub>/c*, *Pcca*, etc). We also tried to index those diffractograms with a post-perovskite  $\text{CaIrO}_3$ -type *Cmcm* phase, as observed for instance in  $\text{MgSiO}_3$ , but we could not find any satisfactory match with the data. We also extended our research by testing other structures found in high-pressure  $\text{ABX}_3$  systems before or after dissociation (cotunnite, brookite, baddeleyite, pseudobrookite, stibnite,  $\text{CaSi}_2\text{O}_5$ ,  $\text{Ca}_2\text{SiO}_4$ ,  $\text{SrSi}_2\text{O}_5$ ,  $\text{Fe}_2\text{P}$ , etc) but again, without any satisfying results. The best fit was obtained using a *P2<sub>1</sub>/m* space group for all diffractograms over the whole pressure range

including high temperature and quenched to ambient temperature data. This lattice volume is of around  $300 \text{ \AA}^3$ . This volume is much larger than any unit cell volumes of  $\text{CaTiO}_3$  extrapolated to those conditions ( $\approx 170 \text{ \AA}^3$ ), thus indicating that the stoichiometry is different from  $\text{CaTiO}_3$ . Along these lines, this volume of  $300 \text{ \AA}^3$  would perfectly match a stoichiometric volume of  $\text{CaO} + \text{CaTi}_2\text{O}_5$  in those conditions.

In all high-temperature diffractograms at pressures above 90 GPa, two reflections are also consistent with a  $\text{CaO}$ - $B2$  phase, according to the equation of state for  $B2$ -phase of  $\text{CaO}$  (Yamanaka et al., 2002) corrected for temperatures. One example of the refinement is given in Figure 6, where the two co-existing phases can be identified in that pressure domain.

Interestingly enough,  $\text{CaO}$  transforms from  $\text{NaCl}$ - $B1$  to  $\text{CsCl}$ - $B2$  structures at around 60 GPa at 300 K (Richet et al. 1988, Yamanaka et al. 2002), which corresponds to the pressure under which  $\text{CaTiO}_3$  starts to transform into the  $P2/m$  phase. Accordingly, we think the dissociation is likely driven by the change in structure of  $\text{CaO}$ .

Following these observations reinforced by an analogy with Wu et al. (2009) who have suggested dissociation of a  $\text{FeTiO}_3$  perovskite into  $(\text{Fe}_{1-\delta}, \text{Ti}_\delta)\text{O}$  and  $\text{Fe}_{1+\delta}\text{Ti}_{2-\delta}\text{O}_5$ , we propose that above 60 GPa and at high temperature,  $\text{CaTiO}_3$  indeed breaks down into  $\text{CaO}$  ( $B2$ ) and  $\text{CaTi}_2\text{O}_5$  instead of going through a  $\text{pPv}$  phase, following the dissociation reaction:

(11)



To validate this hypothesis, we notice that in addition to those of the  $P2/m$  phase and of  $B2$  -  $\text{CaO}$ , we cannot observe any diffraction line of  $\text{TiO}_2$ . The summary of the  $\text{CaTiO}_3$  phases observed at different pressures and temperatures are all represented in a data-plot diagram in Figure 7a and a tentative phase diagram Figure 7b.

This dissociation of the perovskite  $\text{CaTiO}_3$  without going through a post-perovskite phase is an important experimental demonstration of the importance of simple oxides for the thermodynamic evolution of  $\text{ABX}_3$  systems. Tsuchiya et al. (2015) predicted a new  $\text{Fe}_2\text{P}$ -type phase for cotunnite- $\text{SiO}_2$  around 0.9 TPa which could impact  $\text{ABX}_3$  structures by favoring their dissociations in the silicate mantles ( $\text{MgSiO}_3$ ,  $\text{CaSiO}_3$ ) of super-Earths. Also, there is still no evidence of a  $\text{CaIrO}_3$ -type post-perovskite phase in  $\text{CaSiO}_3$  up to 149 GPa (Fujino et al. 2009). Though, the appearance of such a complex oxide as  $\text{CaTi}_2\text{O}_5$  is a transformation



that has similarities with the prediction made for  $\text{MgSiO}_3$  and the stabilization of  $\text{MgSi}_2\text{O}_5$  at extreme conditions, according to Umemoto et al. (2011, 2017).

In every high-pressure transition from a perovskite to a post-perovskite phase or to an assemblage containing oxides, a negative volume variation of around 1-2% has been observed (e.g., Hirose et al., 2005 and references therein). Considering the stoichiometry of equation (11), the volume of  $\text{CaTiO}_3$ , extrapolated to high-pressure and high-temperature ( $\approx 164\text{\AA}$ ), is slightly higher than that of  $\frac{1}{2} \text{CaO} + \frac{1}{2} \text{CaTi}_2\text{O}_5$  ( $\approx 156\text{\AA}$ ) in the same conditions. Using our high-pressure high-temperature equation of state of  $\text{CaTiO}_3$  and that of  $\text{CaO}$  as B2 (Yamanaka et al., 2002), we infer that the volume variation of reaction (11) is between -1% and -5% depending on the temperature. Although a high-pressure polymorph of  $\text{CaTiO}_3$  with  $\text{CaIrO}_3$  structure was predicted by first-principles calculations (Xu et al., 2015), the results of the present experimental study suggest that breakdown according to reaction (8) bypasses the expected post-perovskite phase and that dissociation to two other phases is favoured.

Between 60 and 90 GPa, the  $\text{CaTiO}_3$  perovskite structure can still be recognized in our patterns in coexistence with an assemblage of  $\text{CaO}$ -B2 and  $\text{CaTi}_2\text{O}_5$ - $P2/m$ . The  $\text{CaTiO}_3$  reflections weaken while the reflections attributed to  $\text{CaTi}_2\text{O}_5$  grow with increasing pressure. In this transition pressure range, however, few peaks remain unidentified. They could be due to yet unidentified defect structures derived from  $\text{CaTiO}_3$  perovskite, in a manner possibly similar to those reported in  $\text{MgSiO}_3$  by Tschäuner et al. (2008). Another important aspect of this transition mechanism is the effect of temperature on the dissociation process given in equation (11). At ambient temperature, the cold compression of  $\text{CaTiO}_3$  is unable to overcome energy barriers needed to drive the transition into  $\text{CaO}$  and  $\text{CaTi}_2\text{O}_5$ . We observe instead a change in the compression regime of the lattice parameters. As temperature is increased, the main peaks of the  $P2/m$  phase start to appear but this new phase still coexists with  $\text{CaTiO}_3$  perovskite for kinetic reasons over a 30 GPa pressure range. It is important to note that the complex mixed phase regime appearing in Figure 7 is prohibited according to thermodynamics and thus, draws its origin in kinetic effects.

Other authors have also addressed the complexity of different transition processes in  $\text{ABO}_3$  compounds. Wu et al. (2009) have shown that ilmenite  $\text{FeTiO}_3$  goes through a perovskite phase but then dissociates into a wüstite-type  $(\text{Fe}_{1-\delta}, \text{Ti}_\delta)\text{O}$  and a monoclinic  $C2/c$   $\text{Fe}_{1+\delta}\text{Ti}_{2-\delta}\text{O}_5$  structure, thus omitting the post-perovskite phase. More recently, Akaogi et al. (2016) proposed a dissociation of  $\text{FeTiO}_3$  into  $\text{Fe}_2\text{TiO}_4 + \text{TiO}_2$  or  $\text{Fe}_2\text{TiO}_4 + \text{FeTi}_2\text{O}_5$



depending on the temperature. They also highlighted the dissociation process of  $\text{ZnTiO}_3$  (Akaogi et al. 2015) where the ilmenite structure transforms into a perovskite and then dissociates directly into its simpler oxides with a rocksalt-type  $\text{ZnO}$  and a baddeleyite-type  $\text{TiO}_2$ . Kojitani et al. (2005) studied the dissociation of  $\text{SrSiO}_3$  into  $\text{Sr}_2\text{SiO}_4$  and  $\text{SrSi}_2\text{O}_5$  at 16 GPa and proceeded with a Rietveld analysis of the obtained species. In 2010, Okada et al. (2010) found in  $\text{MnTiO}_3$  the same process of decomposition as shown in the present study for  $\text{CaTiO}_3$  but with an orthorhombic  $\text{MnTi}_2\text{O}_5$  structure with lattice parameters indeed close to those of  $\text{FeTi}_2\text{O}_5$  from Akaogi et al. (2016). In addition, Kondo et al. (2000) described a high-pressure phase transition in  $\text{MnO}$  from  $B1$  to a distorted  $B1$  at 30 GPa, which echoes with the dissociation of  $\text{MnTiO}_3$  into  $\text{MnO}$  and  $\text{MnTi}_2\text{O}_5$ .

These data show that titanates tend to dissociate more easily without going through a post-perovskite phase whereas germanates adopt more easily post perovskite phases. (Ito & Matsui, 1979). Tateno et al. (2006) describe a transition from perovskite to a  $\text{CaIrO}_3$ -type post-perovskite phase in  $\text{MnGeO}_3$  at 60 GPa and 1800 K, exactly the same process as in  $\text{MgGeO}_3$  at 63 GPa and 1800 K (Hirose et al., 2005). More recently,  $\text{ZnGeO}_3$  also joined the Pv-pPv transition group with a similar process as  $\text{MgSiO}_3$  as it transforms around 114 GPa (Yusa et al., 2014). In 2017, However, Dutta et al. triggered the dissociation of  $\text{FeGeO}_3$  into  $\text{FeO}$  and  $\text{GeO}_2$  at high temperature without adopting a pPv structure, following the same transition path as in  $\text{FeSiO}_3$  (Dutta et al., 2017).

Based on those literature observations and on the experimental data of the present study, we thus confirm that when chemical bonding characteristics are met as it is the case in titanates,  $\text{ABO}_3$  systems often break down into different oxides compounds. As the dissociation of  $\text{CaTiO}_3$  (Pv) without transition to a post-perovskite phase is a phenomenon that has already been seen in other  $\text{ABO}_3$  systems, it further brings us to the conclusion that this type of transition is common, especially in  $\text{ATiO}_3$  systems (table 4). Moreover, according to Akaogi et al. (2016) (and references therein), it seems that the titanate perovskites do not transform into post-perovskite but instead dissociate into oxides, a behaviour seemingly related to a small degree of covalency of Ti-O bonds.

#### 4 Implications for planetary interiors

It is interesting to compare our observation of the dissociation of  $\text{CaTiO}_3$  with the prediction by Umemoto et al. (2011) using atomistic calculations of a dissociation of  $\text{MgSiO}_3$  into  $\text{MgO}$  and  $\text{MgSi}_2\text{O}_5$  phase with a  $P2_1/c$ -type structure at pressures of the order of 900

GPa. Moreover, Tsuchiya & Tsuchiya (2015) predicted the dissociation of  $\text{CaSiO}_3$  and  $\text{MgSiO}_3$  into their corresponding oxides, with the change in structure of  $\text{SiO}_2$  as a possible cause, which echoes with the dissociation of  $\text{CaTiO}_3$  likely favored by the  $\text{CaO}$  *B1-B2* transformation.

To summarize, the decomposition of several perovskites (titanates, germanates and silicates) under pressure is often observed into a combination of their simplest constitutive oxides ( $\text{AO}$  and  $\text{BO}_2$ ). However, a thorough examination of the literature reveals more complex stoichiometries of these breakdowns. With  $\text{CaTiO}_3$ , these transformations and dissociations can be observed in a pressure range accessible to accurate experimental studies which could usefully shed light on certain aspects of the evolution of  $\text{MgSiO}_3$  or  $\text{CaSiO}_3$  compounds at ultra-high pressures in exoplanets. Moreover, further detailed study of the  $\text{CaTi}_2\text{O}_5$  phase could usefully shed light on some properties of  $\text{ASi}_2\text{O}_5$  compounds in super-Earth mantles

## 5 Conclusion

In conclusion, we explored the high-pressure high temperature transformations of  $\text{CaTiO}_3$  perovskite up to 170 GPa and 4500 K. We measured the thermal equation of state in the stability domain of  $\text{CaTiO}_3$  perovskite. We also measured the melting curve of this perovskite up to 55 GPa. At higher pressures and temperatures, at odds with what had been predicted by theoretical calculations, we show that  $\text{CaTiO}_3$  does not transform into a post-perovskite  $\text{CaIrO}_3$  *Cmcm* structure but instead dissociates into *B2*- $\text{CaO}$  and monoclinic  $\text{CaTi}_2\text{O}_5$ . The transition process starts at 60 GPa and is spread, for kinetic reasons, over a large range of pressure. The transition toward *B2*- $\text{CaO}$  and  $\text{CaTi}_2\text{O}_5$  is complete at pressures around 100 GPa.

Strengthening knowledge of the evolution of  $\text{ABX}_3$  structures at high pressures is of first importance to infer the mineralogical state of exoplanetary mantles. We propose that future studies of the  $\text{CaTi}_2\text{O}_5$  phase discovered in this work in a pressure range amenable to precise experimental studies will provide insights into exoplanetary ultra-high pressure phenomena that are still beyond reach of experimental mineral physics studies.

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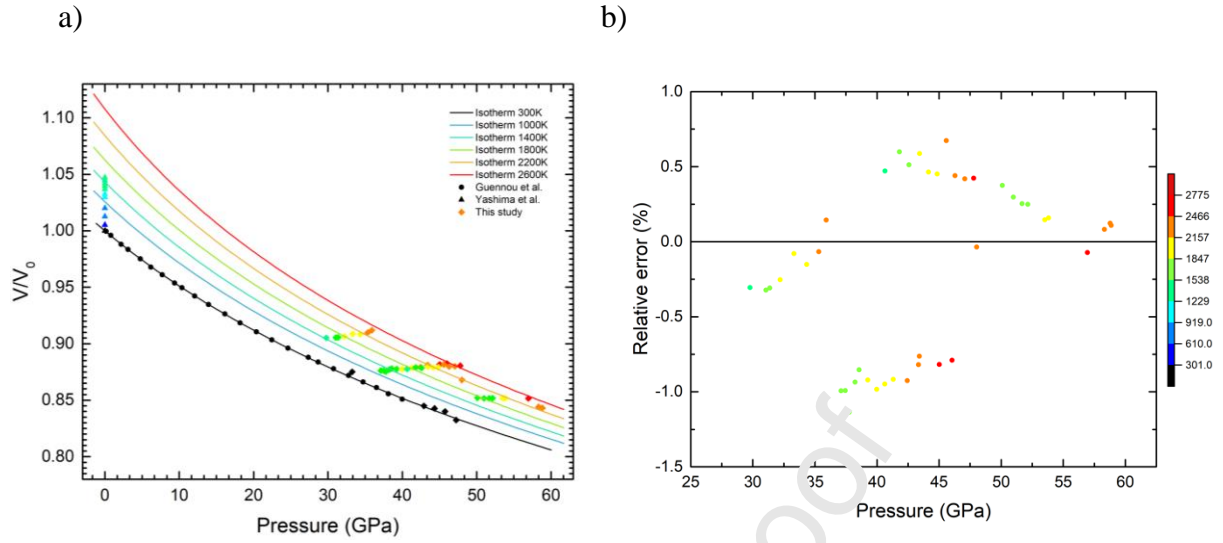
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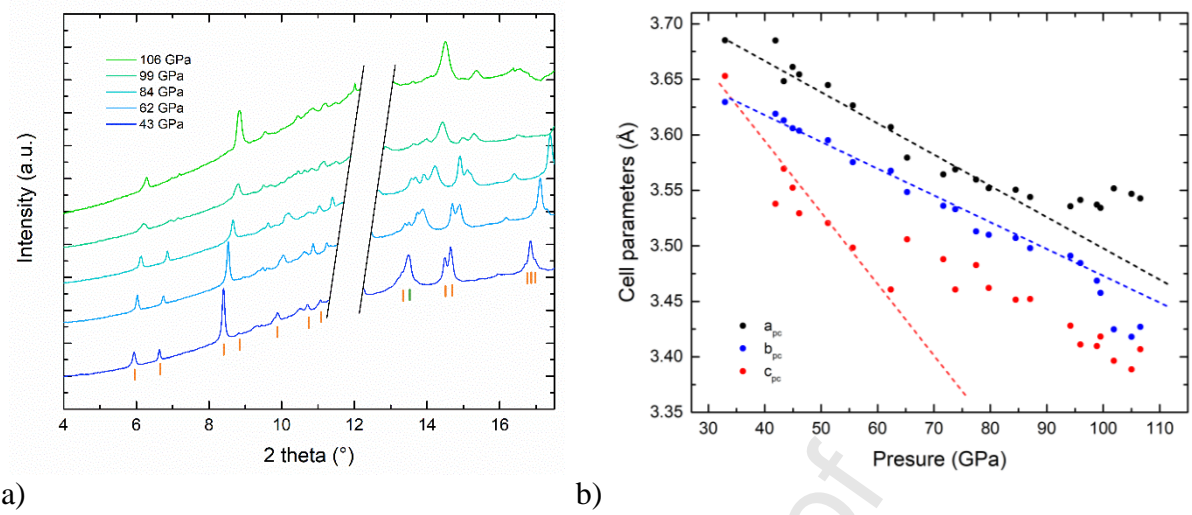
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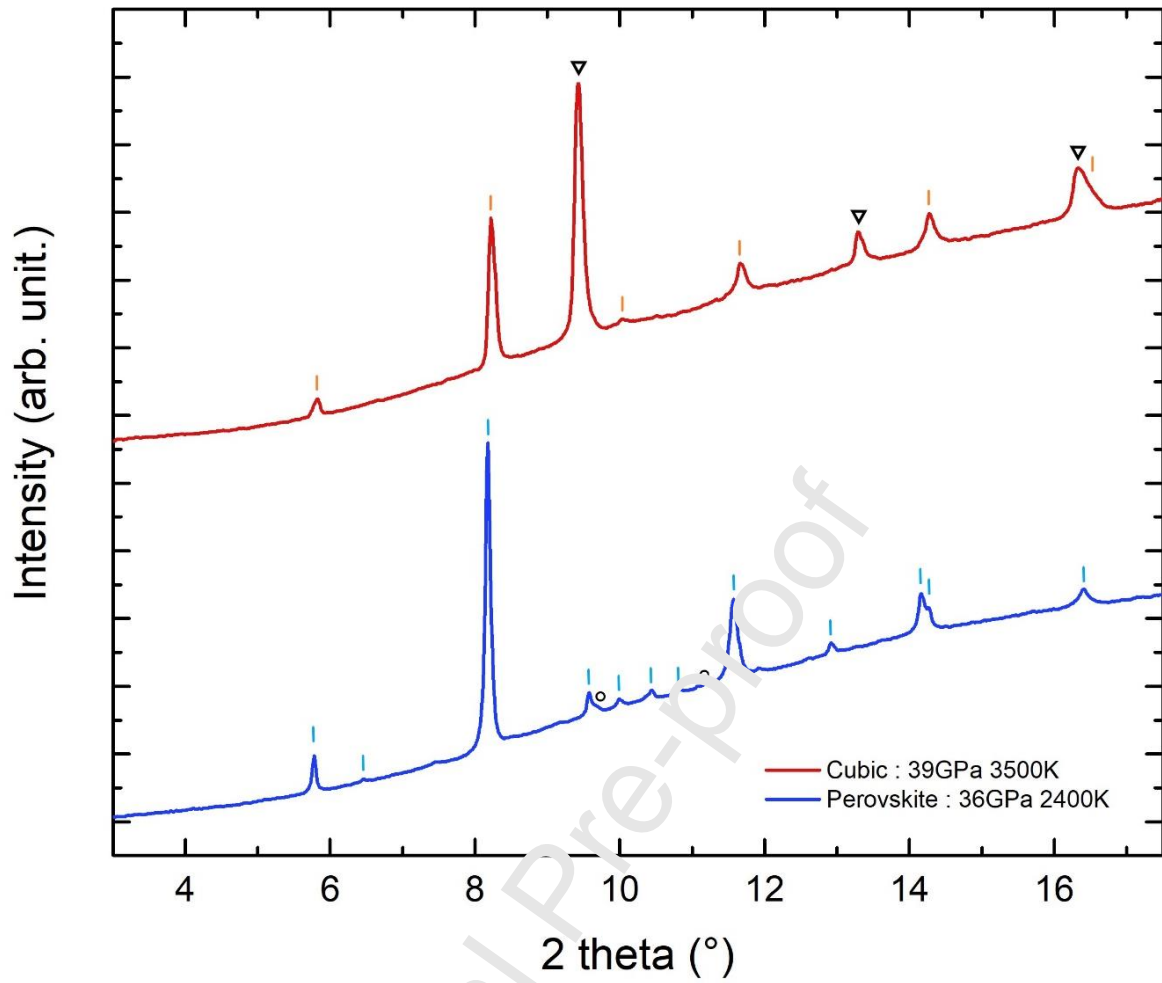
## Figures



**Figure 1:** a) Thermal equation of state of  $\text{CaTiO}_3$  perovskite with isotherms from 300 K to 2600 K, along with experimental data used in the refinement. This dataset includes: Guennou et al. (2010) data from 0 to 40 GPa (black dots), data from Yashima et al. (2008) at ambient pressure (represented by the coloured triangles), our own data collected at room-temperature from 40 to 47 GPa (black diamonds) and data collected at high-temperature from 30 to 60 GPa (coloured diamonds) b) Relative error between the volume calculated with the thermal EoS and the volume measured on each diffractograms. The colour scale corresponds to the measured temperatures in K.

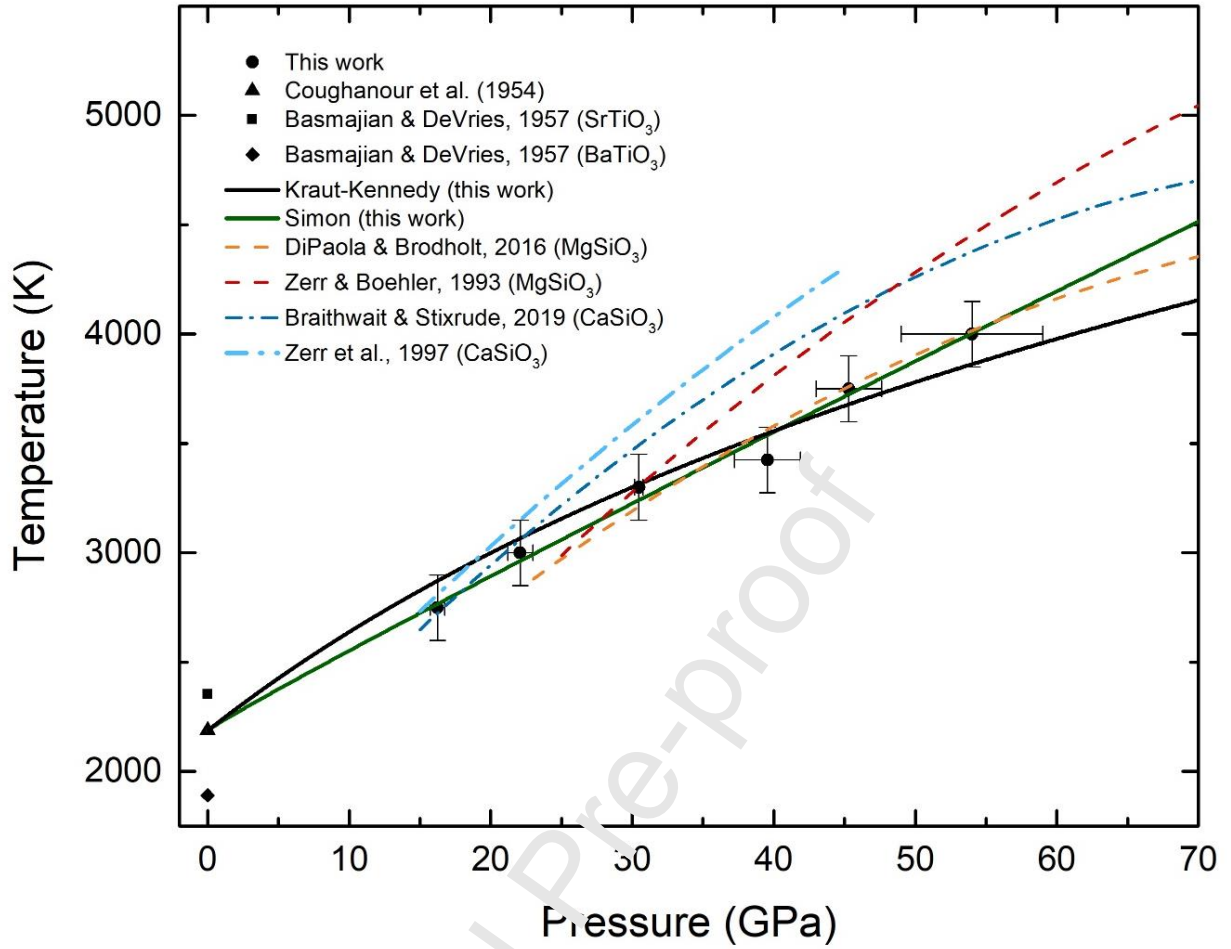


**Figure 2:** a) Room temperature X-ray diffraction patterns of  $\text{CaTiO}_3$  perovskite as a function of pressure with peak positions indicated in orange; the one pressure medium peak is marked in green and main peaks are masked so as to evidence details among lower intensity reflections; b) Room-temperature  $\text{CaTiO}_3$  cell axis evolution as a function of pressure. Cell parameters are recalculated within a pseudo-cubic cell from the  $Pbnm$  structure where  $a_{pc}=a_{pv}/\sqrt{2}$ ,  $b_{pc}=b_{pv}/\sqrt{2}$ ,  $c_{pc}=c_{pv}/2$  (pc: pseudo-cubic; pv: perovskite). The straight dotted lines are guide to the eye to evidence the change in the compression behavior. We observe a change in the slope at 90 GPa par axis a and b, and a change at 60 GPa for axis c.

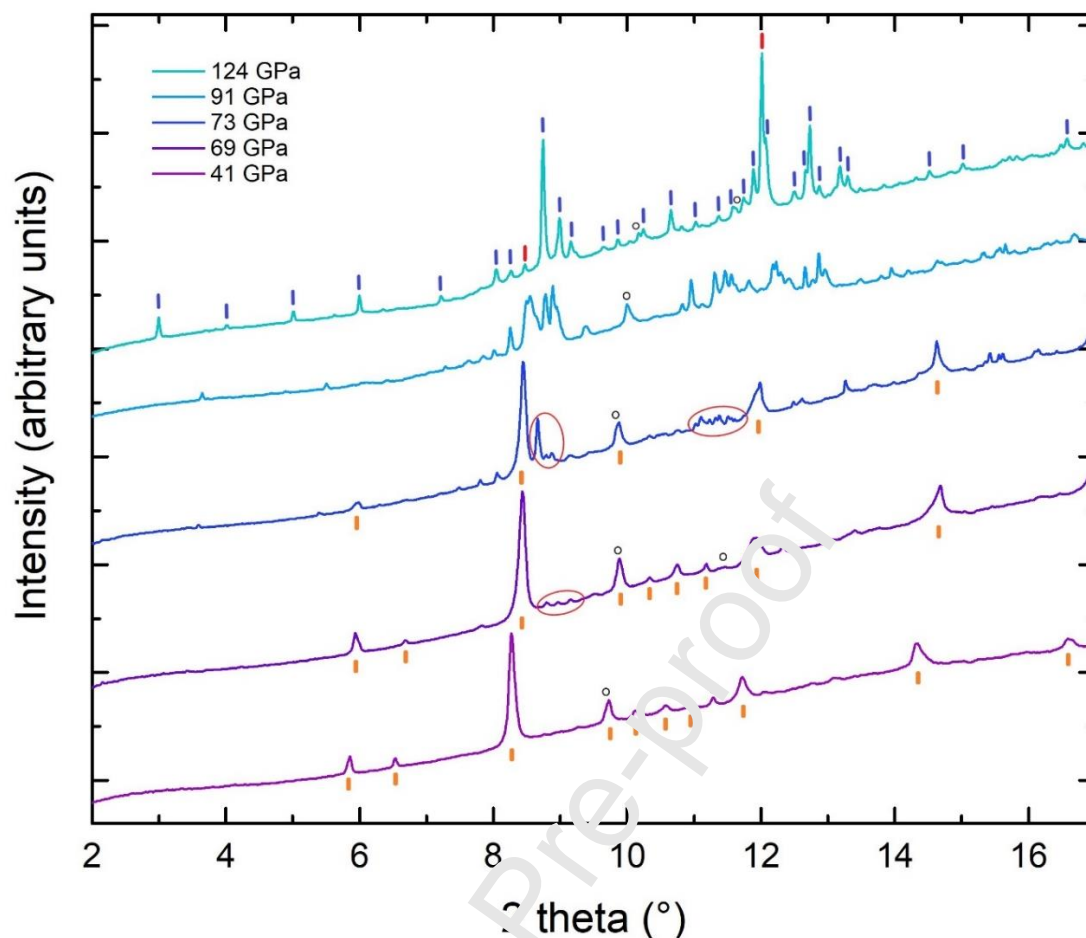


**Figure 3:** Orthorhombic perovskite to cubic structure transition. At the bottom, the perovskite (blue ticks) diffractogram with open circles corresponding to the platinum reflections. On top, the diffractogram of the cubic phase (orange ticks) with open triangles corresponding to the KCl reflections.

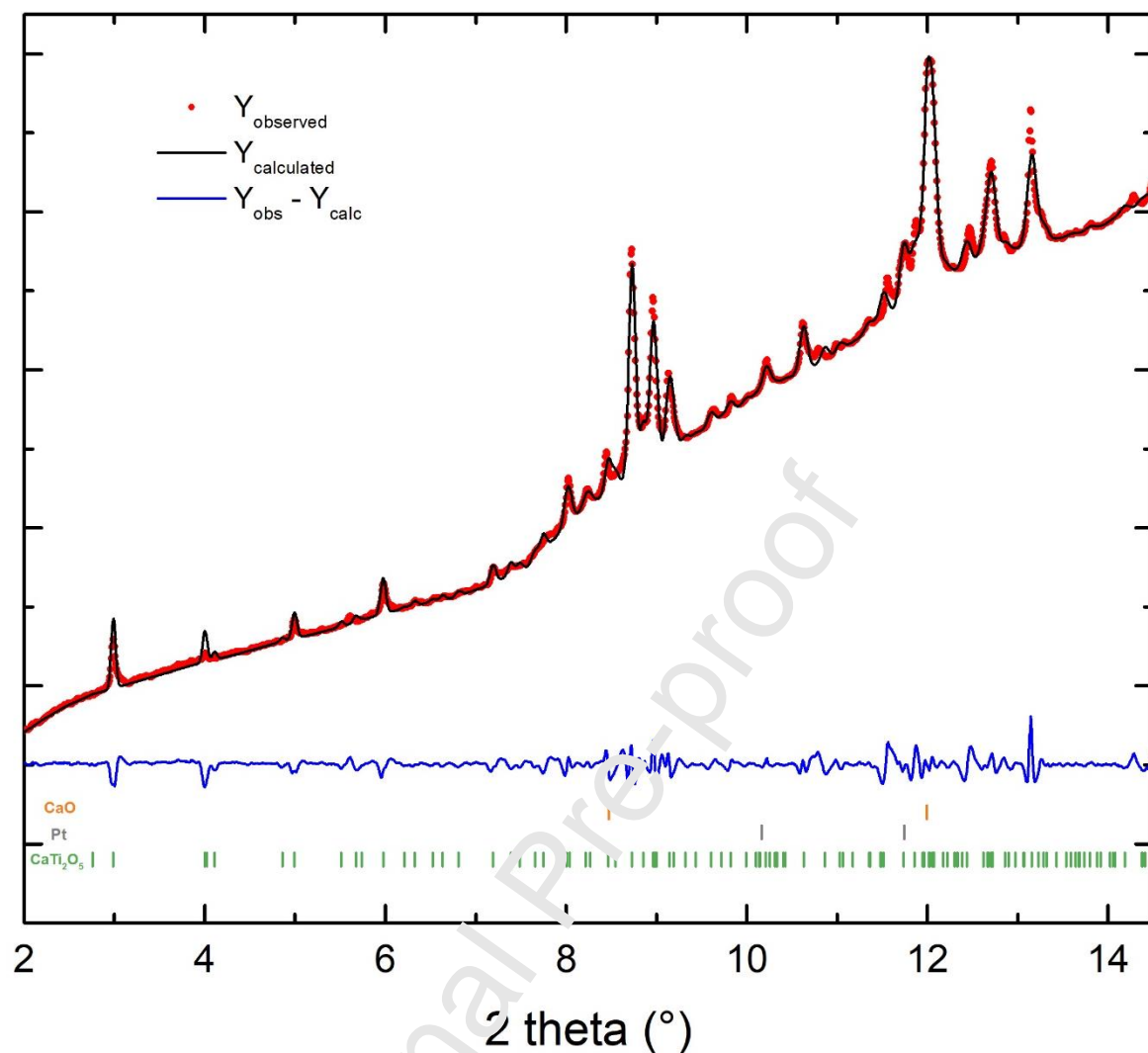




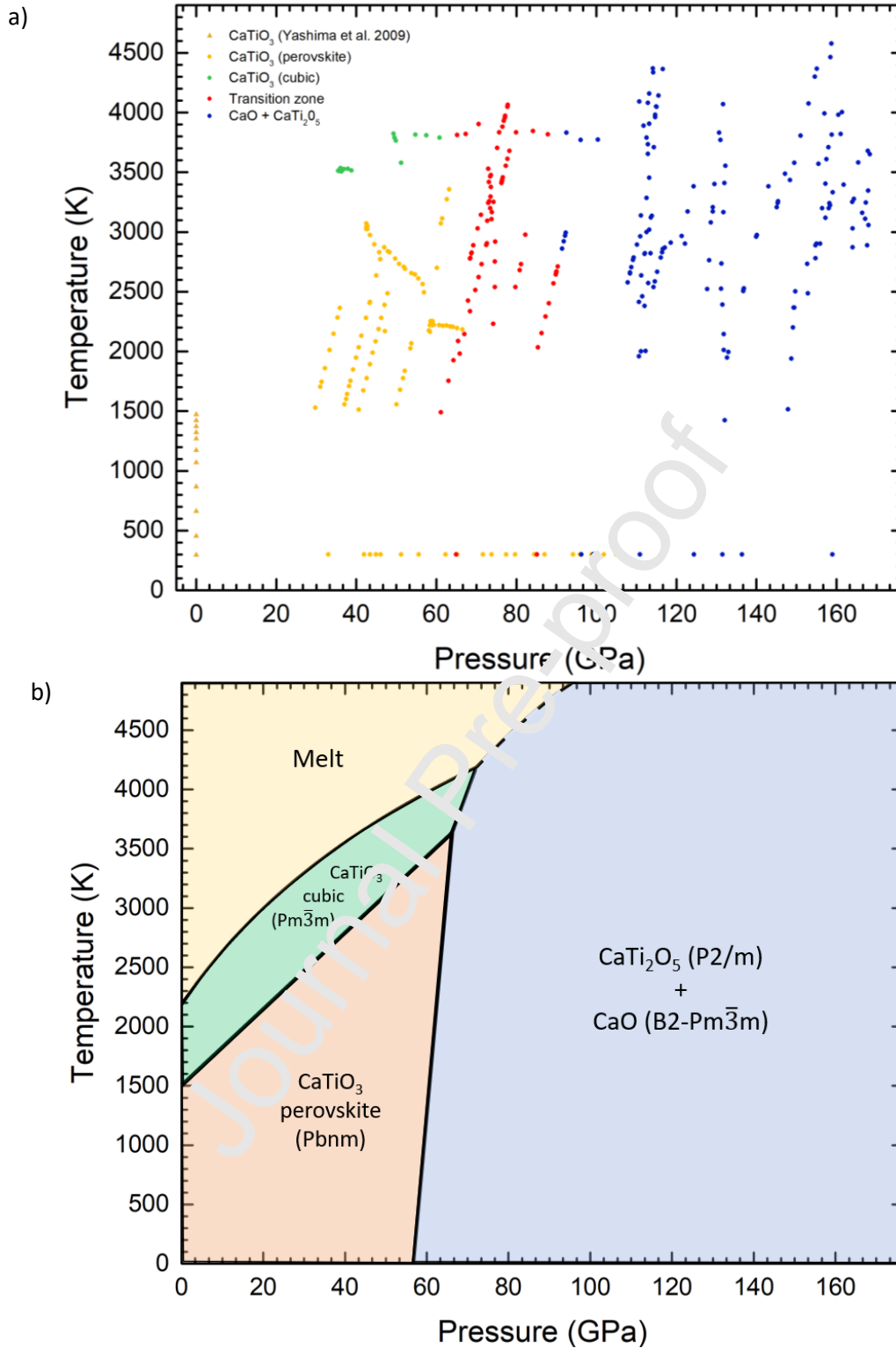
**Figure 4:** Our melting curve of  $\text{CaTiO}_3$  plotted along with melting curves of  $\text{CaSiO}_3$  and  $\text{MgSiO}_3$  for comparison. The study from Zerr & Bohler (1993) and Zerr et al. (1997) have been also performed with DAC experiments, while the study from DiPaola & Brodholt (2016) and Braithwaite & Stixrude (2019), are from atomistic simulations. The black circular dots are our experimental data and the green and black lines are Simon's and Kraut-Kennedy's fits of our data points. The data at ambient pressure come from Coughanour et al. (1954) for  $\text{CaTiO}_3$  and from Basmajian & DeVries (1957) for  $\text{SrTiO}_3$  and  $\text{BaTiO}_3$ .



**Figure 5:** Examples of diffractograms at pressures from 41 to 124 GPa and temperature between 1500 K and 2500 K showing the evolution from pure  $\text{CaTiO}_3$  at low pressures (orange ticks) to  $\text{CaTi}_2\text{O}_5$  (blue ticks) +  $\text{CaO}$  (red ticks) mixture at the highest pressures. Open black circles represent the platinum. Open red circles show regions where new reflections are first observed, until we observe the full crystallisation of  $\text{CaTi}_2\text{O}_5$  mixed with  $\text{CaO}$  at pressures above 100 GPa. The diffractogram at 76 GPa shows a coexistence between the  $\text{CaTiO}_3$  perovskite phase and the appearance of what we assume are the reflections from the  $\text{CaTi}_2\text{O}_5$  +  $\text{CaO}$  phase. The diffractogram at 91 GPa shows the reflections of the  $\text{CaTi}_2\text{O}_5$  +  $\text{CaO}$  phase in formation but not fully formed due to kinetic reasons.



**Figure 6:** Full profile LeBail fit of a high-pressure diffraction pattern recorded at 122 GPa and 2900 K (series #2, Table 1). The green ticks correspond to  $P2/m$ - $\text{CaTi}_2\text{O}_5$  structure ( $a=7.78$  Å,  $b=5.36$  Å,  $c=7.18$  Å and  $\beta=91.24^\circ$ ), the orange ticks to  $B2$ - $\text{CaO}$  ( $a=2.53$  Å) and the grey ticks correspond to the platinum ( $a=3.65$  Å).  $\chi^2 = 0.055$ . Formula unit  $Z=4$ . Data available here : [10.17632/rmkzfc75.3](https://doi.org/10.17632/rmkzfc75.3)



**Figure 7: Summary of the  $\text{CaTiO}_3$  phases observed at different pressures and temperatures.**

- a) Summary of XRD data collected at different pressures and temperatures. Yellow dots:  $\text{CaTiO}_3$  orthorhombic ( $\text{Pbnm}$ ) perovskite phase. Yellow triangles: neutron diffraction data from Yashima et al. (2009) at ambient pressure and up to 1500K. Green dots:  $\text{CaTiO}_3$  cubic perovskite phase  $\text{Pm}\bar{3}\text{m}$ . Red symbols: mixed phase regime of kinetic

origin containing  $\text{CaTiO}_3$  perovskite (Pbnm) together with a higher-pressure assemblage marked as blue dots when devoid of perovskite signal. This high-pressure assemblage consists of CaO (B2) and  $\text{CaTi}_2\text{O}_5$  P2/m. Each pressure condition was measured either with KCl, platinum or gold pressure standards.

- b) Tentative phase diagram of  $\text{CaTiO}_3$ , according to X-ray diffraction data collected in this study at different pressure-temperature and to previous literature data. The hypothetical phase boundaries have been drawn without fit, using data of figure 7a. The melting curve of  $\text{CaTiO}_3$  perovskite was fitted with a Simon model. The tetragonal phase intermediate between orthorhombic and cubic perovskite structures was not represented.

## Tables

Series #	CaTiO <sub>3</sub> type	Origin	Pressure transmitting medium	Pressure scale	Pressure domain (GPa)	Temperature domain (K)	Diamond culets
1	Powdered	Synthetic	CaTiO <sub>3</sub> + Pt	Pt	30 - 78	1500 - 4000	150/300
2	Powdered	Synthetic	CaTiO <sub>3</sub> + Pt	Pt	83 - 120	2000 - 3000	100/300
3	Powdered	Synthetic	CaTiO <sub>3</sub> + Pt	Pt	121 - 155	2000 - 4400	250
4	Polycrystalline	Natural	KCl	KCl	35 - 75	2100 - 3500	200
5	Polycrystalline	Natural	KCl	KCl	47 - 115	2100 - 4400	100/300
6	Polycrystalline	Natural	KCl	KCl	127 - 161	1400 - 4000	100/300
7	Polycrystalline	Natural	KCl	KCl	154 - 168	2800 - 4500	150/300
8	Polycrystalline	Natural	Ne	Au	28 - 109	300	150/300

**Table 1:** Samples used for the different experimental runs.

$V_0$ (cm <sup>3</sup> /mol/f.u.)	33.622 (fixed)
$K_0$ (GPa)	180.6 (4)
$K'_0$	4 (fixed)
$dK/dT$ (GPa.K <sup>-1</sup> )	-0.022 (1)
$\alpha_0$ (x10 <sup>-5</sup> K <sup>-1</sup> )	3.25 (5)
$\alpha_1$ (x10 <sup>-8</sup> K <sup>-2</sup> )	1.3 (1)

**Table 2:** Parameters of the thermal equation of state of CaTiO<sub>3</sub> ( $\chi^2 = 1.03$ ), with  $V_0$  fixed and set to the value determined by Guennou et al. (2010) and  $K'_0$  fixed to 4.

	$V_0$ (cm <sup>3</sup> /mol/f.u.)	$K_0$ (GPa)	$K'_0$	$dK/dT$ (GPa K <sup>-1</sup> )	$\alpha_0$ (10 <sup>-5</sup> K <sup>-1</sup> )	$\alpha_1$ (10 <sup>-8</sup> K <sup>-2</sup> )
<b>CaTiO<sub>3</sub></b>						
This study	33.62 (7) <sup>a</sup>	180.6 (4)	4 (fixed)	-0.022 (1)	3.25 (5)	1.3 (1)
Guennou (2010)	33.62 (7)	181.0 (6) 172.1	4 (fixed)	-	-	-
Webb (1999)	33.82	(10)	6.09 <sup>b</sup>	-0.0225 (7)	3.87	-
Kung (1997)	34.17 (2)	175	5.78	-	-	-
Fischer (1993)	33.87	177 (3)	5.1 (8)	-	-	-
<b>SrTiO<sub>3</sub><sup>*</sup></b>						
Guennou (2010)	35.93 (1)	165 (3)	6.4 (8)	-	-	-
Webb (1999)	35.89	172.0 (1) 178.8	5.36 <sup>b</sup>	-0.031 (1)	3.23 (2) <sup>c</sup>	-
Fischer (1993)	35.87	(46)	4.31 (1)	-	-	-
<b>BaTiO<sub>3</sub><sup>**</sup></b>						
Pruzan (2002)	38.72	135 139.2	6.4	-	-	-
Fischer (1993)	38.74	(31)	9 <sup>d</sup>	-	-	-
<b>MgSiO<sub>3</sub></b>						
Tange (2012)	24.45 (fixed)	256.7 (15)	4.09 (6)	-	-	-
Ricolleau (2009)	24.69 (fixed)	245 (1)	4 (fixed)	-0.036 (1)	3.19 (17)	0.88 (16)
Fiquet (2000)	24.43 (1)	259.5 (9)	3.69 (4)	-0.017 (2)	2.18 (12)	0.11 (8)
<b>CaSiO<sub>3</sub></b>						
Ricolleau (2009)	27.46 (fixed)	244 (1)	4 (fixed)	-0.035 (2)	3.06 (19)	0.87 (18)
Shim (2000)	27.45 (3)	236 (4)	3.9 (2)	-0.028 (11)	2.2 (3)	-

<sup>a</sup> From Guennou et al. (2010) – Room temperature DAC with x-ray diffraction.

<sup>b</sup> From Fischer et al. (1993) – Extrapolated from 500 MPa. Piston cylinder apparatus and ultrasonic interferometric technique.

<sup>c</sup> From de Ligny & Richet (1996) – Calorimetry and energy-dispersive x-ray diffraction.

<sup>d</sup> From Fischer & Polian (1987) – Brillouin scattering in DAC.

\* Data for the cubic phase (0-10 GPa)

\*\* Data for the tetragonal phase (0-2 GPa)

**Table 3:** Thermal equation-of-state parameters of different titanate, bridgmanite and calcium silicate perovskite.



Material	Pressure	geometry	a (Å)	b (Å)	c (Å)	$\beta$ (°)	V (Å <sup>3</sup> )
CaTi <sub>2</sub> O <sub>5</sub>	120 GPa	Monoclinic $P2_1/m$	7.776	5.362	7.167	91.29	298.75
	155 GPa	Monoclinic $P2_1/m$	7.662	5.268	7.049	91.422	284.43
NaMg <sub>2</sub> F <sub>5</sub> (Dutta, 2019)	88 GPa	Monoclinic $P2_1/c$	5.618	4.977	7.729	89.23	197.08
	137 GPa	Monoclinic $P2_1/c$	5.423	4.732	7.312	89.2	187.62
FeTi <sub>2</sub> O <sub>5</sub> (Akaogi, 2016)	31 GPa HT°	Orthorhombic	8.567	5.753	5.257	-	259.10
MnTi <sub>2</sub> O <sub>5</sub> (Okada, 2010)	61 GPa HT°	Orthorhombic	8.263	5.743	5.179	-	245.77
FeTi <sub>2</sub> O <sub>5</sub> (Wu, 2009)	40 GPa HT°	Monoclinic $C2/c$	10.135	4.305	10.023	141.48	272.37
SrSi <sub>2</sub> O <sub>5</sub> (Kojitani, 2005)	16 GPa HT°	Orthorhombic $Cmca$	5.239	9.280	13.441	-	653.47

**Table 4:** Comparison of the AB<sub>2</sub>X<sub>5</sub> structures and lattice parameters.

## Author statement

**Baptiste Truffet:** Validation, Formal analysis, Investigation, Writing – Original Draft, Writing – Review & Editing, Visualization.

**Guillaume Fiquet:** Conceptualization, Methodology, Validation, Investigation, Writing – Original Draft, Supervision, Project administration, Funding acquisition.

**Guillaume Morard:** Validation, Investigation.

**Marzena A. Baron:** Validation, Investigation.

**Francesca Miozzi:** Validation, Investigation

**Marion Harmand:** Validation, Investigation.

**Alessandra Ravasio:** Validation, Investigation.

**Mohamed Mezouar:** Validation, Investigation, Resources.

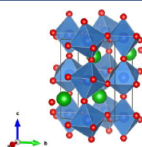
**François Guyot:** Conceptualization, Methodology, Validation, Investigation, Writing – Original Draft, Writing – Review & Editing, Supervision, Project administration.

## Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

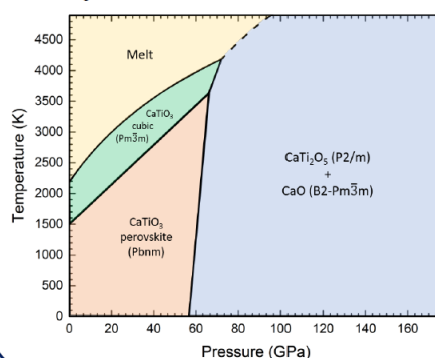
## Graphical abstract

### High pressure dissociation of $\text{CaTiO}_3$ perovskite into $\text{CaO}$ and $\text{CaTi}_2\text{O}_5$

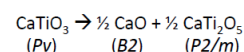


- We investigated the phase transformations of  $\text{CaTiO}_3$  perovskite using x-ray diffraction at high pressure and high temperature up to 170 GPa and 4500 K in a laser-heated diamond-anvil cell
- We report the thermal equation of state as well as the phase diagram at these extreme conditions

$\text{CaTiO}_3$  phase diagram up to 170 GPa and 4500 K



- We show evidence of the dissociation of  $\text{CaTiO}_3$  instead of transformations into denser post-perovskite phases.



- We propose that this transition may be favored by the *B1* to *B2* phase change of  $\text{CaO}$  at around 60 GPa

**Highlights:**

- Phase diagram of the calcium titanate perovskite  $\text{CaTiO}_3$  up to 170 GPa and 4500 K.
- Thermal equation of state of the orthorhombic ( $Pbnm$ ) perovskite  $\text{CaTiO}_3$ .
- Dissociation of the  $\text{CaTiO}_3$  perovskite into  $\text{CaO}$  and  $\text{CaTi}_2\text{O}_5$  without going through a post-perovskite phase.