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1 **Unconventional spin pumping and magnetic damping in an insulating**
2 **compensated ferrimagnet**

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23 Keywords: spin pumping, magnetic damping, coherent and incoherent spin currents,
24 handedness, insulating compensated ferrimagnet

26 **Abstract**

27 Recently, the interest in spin pumping has escalated from ferromagnets into
28 antiferromagnetic systems, potentially enabling fundamental physics and magnonic
29 applications. Compensated ferrimagnets are considered alternative platforms for bridging ferro-
30 and antiferromagnets, but their spin pumping and the associated magnetic damping have been
31 largely overlooked so far despite their seminal importance for magnonics. Herein, we report an
32 unconventional spin pumping together with magnetic damping in an insulating compensated
33 ferrimagnet $\text{Gd}_3\text{Fe}_5\text{O}_{12}$. Remarkably, we unambiguously identified the divergence of the
34 nonlocal effective magnetic damping induced by spin pumping close to the compensation
35 temperature in $\text{Gd}_3\text{Fe}_5\text{O}_{12}/\text{Cu}/\text{Pt}$ heterostructures. Furthermore, the coherent and incoherent spin
36 currents, generated by spin pumping and spin Seebeck effect respectively, undergo a distinct
37 direction change with the variation of temperature. The physical mechanisms underlying these
38 observations are self-consistently clarified by the ferrimagnetic counterpart of spin pumping and
39 the handedness-related spin-wave spectra. Our findings broaden the conventional paradigm of
40 the ferromagnetic spin pumping model and open new opportunities for exploring the
41 ferrimagnetic magnonic devices.

42

43 **Introduction**

44 Magnonics, a research field centered on spin waves/currents, has enabled the conception and
45 the exploitation of low-dissipation and high-performance magnetic storage and computational
46 devices^[1-3]. Spin pumping (SP) induced by ferromagnetic resonance (FMR) provides an
47 efficient route for generating and driving spin currents into a nonmagnetic material from a
48 ferromagnetic spin reservoir while exerting a magnetic damping-related torque on
49 magnetization^[4-7]. Recently, the interest in SP in antiferromagnets has surged with the rapid
50 progress of the antiferromagnetic spintronics/magnonics^[8-10]. In contrast to ferromagnets only
51 hosting the right-handed mode, the coexisting left- and right-handed precession modes in
52 antiferromagnets add a new paradigm in manipulating chiral magnons and further encoding
53 information through polarization, amplitude, and phase^[11, 12]. On the other side of the coin, the
54 vanishing net magnetization in antiferromagnets, as well as the resonance frequency up to the
55 terahertz regime, raise hurdles in the excitation of magnetization dynamics and the detection of
56 SP^[13, 14]. Alternatively, compensated ferrimagnets, which host two inequivalent magnetic
57 sublattices with antiferromagnetic interaction, exhibit quasi-antiferromagnetic behavior close to
58 the compensation points and quasi-ferromagnetic characteristics far away from the
59 compensation points^[15-17]. Thus, compensated ferrimagnets stand out as a fertile platform to
60 combine the benefits of both ferro- and antiferromagnets in potential device applications^[18-20].
61 However, thus far, SP from compensated ferrimagnets and the associated magnetic damping
62 have been largely overlooked in spite of their seminal importance for magnonic device
63 development.

64 Yttrium iron garnet, $\text{Y}_3\text{Fe}_5\text{O}_{12}$ (YIG), is commonly regarded as a ferromagnet because of the
65 strong antiferromagnetic coupling between the a -Fe and d -Fe sublattices^[21]. By replacing the
66 nonmagnetic element Y with the magnetic rare-earth element Gd, an insulating compensated
67 ferrimagnet, $\text{Gd}_3\text{Fe}_5\text{O}_{12}$ (GdIG), is constructed with finite net magnetization. The antiparallel
68 aligning moments of Gd and Fe cancel each other at the magnetic compensation temperature
69 (T_M), resulting in the vanishing of the net magnetization^[22]. Like antiferromagnets, spin
70 dynamics in GdIG provide the left- and right-handed precession modes, which evolve with the
71 temperature variation^[23-25]. In addition, spin angular momentum can be transferred by the
72 magnon spin current without involving charge current between the GdIG films and the adjacent

73 materials, although GdIG is electrically insulating^[26]. Such an insulating compensated
74 ferrimagnet provides a promising platform for exploring ferrimagnetic spin dynamics and low-
75 dissipation magnonic devices. Here, we report an unconventional SP and the associated
76 magnetic damping in GdIG-based heterostructures. We established that the bulk magnetic
77 damping, as well as the SP-related nonlocal magnetic damping, both dramatically increased
78 when the temperature approached T_M . Furthermore, the different variations in the properties of
79 coherent and incoherent spin currents with temperature were examined by comparing the results
80 of the SP and spin Seebeck effect (SSE) experiments. The underlying mechanisms are well
81 understood by the ferrimagnetic SP model and the modification of the handedness-related spin-
82 wave spectra when changing the temperature.

83 **Results and discussions**

84 35 nm-thick $\text{Gd}_3\text{Fe}_5\text{O}_{12}$ (GdIG) films were epitaxially grown on (111)-oriented substituted
85 gadolinium gallium garnet substrates (sGGG) by pulsed laser deposition (PLD). The crystal
86 quality was characterized using scanning transmission electron microscopy (STEM). In Figure
87 1a, the sharp interface between the GdIG film and the sGGG substrate can be clearly observed
88 from the STEM cross-sectional image in the high-angle annular dark-field mode. There was no
89 elemental intermixing at the interface, as determined by the energy-dispersive X-ray
90 spectroscopy (EDX) mapping of the Fe atoms in the same area, as shown in Figure 1b. The
91 atomic-resolution STEM image in Figure 1c exhibits a good epitaxial relationship between the
92 substrate and the GdIG film with an atomically sharp interface. A set of diffraction patterns
93 corresponding to Figure 1c is shown in Figure 1d.

94 Figure 1f presents hysteresis loops with the in-plane magnetic field measured at different
95 temperatures after subtracting the paramagnetic background of the substrate. The saturation
96 magnetization, M_s , extracted from the loops, is plotted in Figure 1g as a function of
97 temperature. The net magnetization of GdIG almost vanishes at the T_M of ~ 200 K. In reality,
98 GdIG consists of three magnetic sublattices: an octahedral Fe sublattice (a sites), a tetrahedral
99 Fe sublattice (d sites) and a dodecahedral Gd sublattice (c sites), as shown in Figure 1e. The two
100 Fe sublattices can be treated as one effective Fe magnetic sublattice because of their robust
101 antiferromagnetic superexchange. As illustrated in the insets of Figure 1g, the Gd sublattice is
102 antiferromagnetically coupled to the effective Fe sublattice, and the effective Fe (Gd) sublattice

dominates the net magnetization above (below) T_M . The high-quality single-crystalline GdIG film with a magnetic compensation temperature offers an excellent platform for studying the magnetic damping and spin transport across the interface in ferrimagnets.

To investigate the spin dynamics in GdIG films, we conducted in-plane FMR measurements^[27]. Here the low-frequency uniform mode, regarded as the ferromagnetic counterpart^[28], was triggered by the absorption of radiofrequency/microwave radiation, as illustrated in Figure 2a. Figure 2b shows the typical FMR spectra recorded as a function of the in-plane magnetic field (H) at various frequencies at 300 K, from which the resonant magnetic field (H_r) and linewidth (ΔH) can be obtained. The magnetic damping manifests itself in the resonant linewidth in the FMR experiments. Figure 2c shows plots of the extracted resonant frequency (f) versus H_r for the GdIG film. The dispersion relations between f and H_r at various temperatures display different characteristics, which can be described by the Kittel formula^[29]:

$$f = \frac{g\mu_0\mu_B}{h} \sqrt{H_r(H_r + M_{eff})}. \quad (1)$$

Here, h , μ_0 , μ_B , g , and M_{eff} are the Plank constant, vacuum permeability, Bohr magneton, Landé g-factor and effective magnetization, respectively. As plotted in Figure 2d, the extracted value of ΔH from the FMR spectra depends linearly on the frequency with different slopes analyzed for the different temperatures. This suggests that the magnetic relaxation of GdIG films is properly represented by Gilbert damping. The Gilbert damping can be decomposed into two contributions: a bulk contribution involving magnon - magnon and magnon - phonon scattering, and an interfacial contribution arising from the SP and the magnetic proximity effect (MPE) at the interface^[6, 30-32]. Furthermore, the interfacial damping from the SP is generally termed as nonlocal magnetic damping^[4, 33]. Accordingly, the values of the Gilbert damping (α) can be extracted by fitting the data of ΔH and f using the following equation^[34],

$$\Delta H = \Delta H_0 + \frac{2h\alpha}{g\mu_0\mu_B} f, \quad (2)$$

where ΔH_0 is the inhomogeneous linewidth broadening. Eq. 2 was employed to obtain the Gilbert damping for all the samples.

To explore the interfacial magnetic damping, a Pt layer with a thickness of 10 nm, which is larger than the spin diffusion length of Pt $\sim 1-8$ nm^[35], was deposited on the GdIG (thickness: 35 nm) film. Thereafter, the GdIG/Pt sample was subjected to FMR measurements. As shown in Figure 2e, the linewidth of the FMR spectra at 6 GHz for GdIG/Pt is noticeably enhanced

133 compared to that for GdIG. The inset in Figure 2e displays f versus H_r at 375 K. The extracted
 134 ΔH as a function of H_r is shown in Figure 2f for the GdIG and GdIG/Pt films. Evidently, the
 135 slope of the linear ΔH versus f plot for the GdIG/Pt sample is larger than that for the GdIG
 136 sample. The data shown in Figures 2e and 2f indicate that the Pt layer introduces additional
 137 damping.

138 The temperature dependence of the Gilbert damping ($\alpha_{GdIG}(T)$) for the GdIG film is
 139 presented in Figure 3a. The damping values were calculated to be 3.3×10^{-3} and 5.8×10^{-3}
 140 at the lowest temperature (10 K) and the highest temperature (375 K) in our experiments,
 141 respectively. The rather low damping values on the order of 10^{-3} , observed at low (10 K) and
 142 high (375 K) temperatures can be attributed to the fact that the magnon–magnon and magnon–
 143 phonon scattering processes dominate the magnetic relaxation process, whereas the magnon–
 144 electron scattering is suppressed because of the absence of conduction electrons in the insulating
 145 GdIG film^[31]. Although these two values are quite similar and small, considerably larger
 146 damping values (one order of magnitude higher than those observed at 10 K and 375 K) were
 147 obtained when the temperature approached T_M from both directions. Evidently, the Gilbert
 148 damping significantly increases when the temperature approaches T_M . This divergent
 149 temperature dependence of the damping close to T_M is in stark contrast with that in ferromagnets,
 150 whose damping commonly decreases upon cooling^[36]. As derived in Supporting Information
 151 (SI), the effective Gilbert damping (α_{eff}) in ferrimagnets can be expressed as $\alpha_{eff} =$
 152 $\frac{\alpha_1 L_1 + \alpha_2 L_2 + \delta}{|L_1 - L_2|}$. Here, δ , L_i , and α_i represent the off-diagonal term of the damping, angular
 153 momentum and Gilbert damping of the magnetic sublattice (i [$i = 1, 2 = \text{Fe, Gd}$]), respectively.
 154 The equation indicates that the effective Gilbert damping diverges at the compensation
 155 temperature (T_A) of a ferrimagnet, where two angular momenta are equal ($L_{Fe} = L_{Gd}$).
 156 Consequently, because T_A is close to T_M in GdIG^[37], a divergent damping was reported close to
 157 T_M in our FMR measurements.

158 The temperature dependence of the damping in GdIG/Pt is very similar to that of GdIG, as
 159 shown in Figure 3a. Noticeably, compared to the damping of the GdIG film (α_{GdIG}), an
 160 enhancement of the effective Gilbert damping, $\Delta\alpha(\alpha_{GdIG/Pt} - \alpha_{GdIG}) > 0$, is clearly observed
 161 for the GdIG/Pt film over the entire temperature range, suggesting that the GdIG/Pt interface
 162 substantially contributes to the overall damping. In analogy with the damping enhancement in

163 ferromagnet/normal metal multilayers, the damping enhancement in ferrimagnet/Pt films may
 164 be derived from SP and MPE^[30]. Pt can have a spontaneous magnetic moment at the interface
 165 with a ferromagnet owing to its large Stoner enhancement^[38,39]. The polarized Pt layer possesses
 166 a large damping owing to the strong spin-orbit interaction of Pt. Thus, the effective damping in
 167 the effective ferromagnet, composed of the polarized Pt layer and GdIG film, is larger than that
 168 in a single GdIG film. This is the so-called enhancement of the damping from MPE. Another
 169 route for the damping enhancement is the nonlocal damping induced by SP. The magnetization
 170 precession of GdIG triggered by microwave results in a nonequilibrium chemical potential
 171 imbalance of the spins at the GdIG/Pt interface, causing an angular momentum flow (spin
 172 currents) from GdIG to Pt. The nonlocal absorption of the pumped spin current by the Pt layer,
 173 owing to the Elliott – Yafet and Dyakonov – Perel spin relaxation, results in the enhancement
 174 of the Gilbert damping^[6,40].

175 To exclude the contribution from MPE, a 5 nm-thick Cu spacer with a negligible spin-orbit
 176 interaction and a long spin diffusion length was inserted to obstruct the MPE between the GdIG
 177 and the Pt layers^[30,41,42]. The temperature-dependent damping in the GdIG/Cu/Pt sample is also
 178 plotted in Figure 3a. A diverging damping versus temperature curve is observed for GdIG/Cu/Pt.
 179 Note that the transmission of spin currents across the interfaces can be affected in some extent
 180 when the polarized Pt layer is replaced with the Cu layer^[43-45]. The effective Gilbert damping
 181 contributed by the MPE is approximatively estimated by $\Delta\alpha_{MPE} = \alpha_{GdIG/Pt} - \alpha_{GdIG/Cu/Pt}$
 182 [see SI]^[30,42,46]. As shown in Figure 3b, $\Delta\alpha_{MPE}(T)$ is nearly independent of temperature with
 183 very small values. The nonlocal damping contributed by SP is evaluated by $\Delta\alpha_{SP} =$
 184 $\alpha_{GdIG/Cu/Pt} - \alpha_{GdIG}$. Different from the behavior of $\Delta\alpha_{MPE}(T)$, $\Delta\alpha_{SP}(T)$ increases as the
 185 temperature approaches T_M from both sides, as shown in Figure 3b. Moreover, $\Delta\alpha_{MPE} =$
 186 1.2×10^{-3} is twice smaller than $\Delta\alpha_{SP} = 2.7 \times 10^{-3}$, as exemplified by the magnetic damping
 187 at 300 K. Furthermore, the values of $\Delta\alpha_{MPE}(T)$ are almost negligible below T_M . Therefore, SP
 188 plays a dominant role in the enhancement of the effective Gilbert damping in GdIG/Pt,
 189 particularly below T_M . In ferrimagnet with two magnetic sublattices, the spin current, I_S ,
 190 generated by SP is given by $I_S = \frac{\hbar}{e} \sum_{i,j=1,2} G_{ij} \mathbf{m}_i \times \dot{\mathbf{m}}_j$, where G_{ij} and \mathbf{m}_i denote the spin-
 191 mixing conductance matrix and the unit vector of the magnetic moment of sublattice i ,

192 respectively^[15, 47]. That is, the spin currents can be treated as the sum of two independent SP
 193 processes in the two magnetic sublattices and the inter-sublattice contributions, implying that
 194 the Gilbert-type terms (damping-like torque) should be added to the Landau - Lifshitz - Gilbert
 195 equation. Note that the magnetic moment of Gd originates from the inner 4*f* shells, which are
 196 more localized compared to the 3*d* shells of Fe. Consequently, the angular momentum transfer
 197 between Pt and GdIG is dominated by the effective Fe sublattice *via s-d* exchange. Therefore,
 198 the SP-driven damping ($\Delta\alpha_{SP}$) can be simplified as $\Delta\alpha_{SP} \approx \frac{\Delta\alpha_{Fe}L_{Fe}}{|L_{Fe}-L_{Gd}|}$. Here, $\Delta\alpha_{Fe}$ denotes the
 199 damping enhancement induced by SP in the effective Fe sublattice. This suggests that $\Delta\alpha_{SP}$
 200 diverges at temperature close to T_A , which is consistent with our experimental results shown in
 201 Figure 3b.

202 To further elucidate the difference in SP between ferrimagnets and ferromagnets, we
 203 characterized the efficiency of the interfacial spin transport parameterized by the spin-mixing
 204 conductance. Assuming that the SP-related damping obeys the classical spin pumping in
 205 ferromagnetic systems, the effective spin-mixing conductance is determined by^[6] $\Delta\alpha_{SP} =$
 206 $\frac{g\mu_B}{4\pi M_s t_{GdIG}} G_{FM}^{\uparrow\downarrow}$. Here, t_{GdIG} and M_s are the thickness and saturation magnetization of the GdIG
 207 film, respectively. The cyan curve in Figure 3b displays temperature-dependent $G_{FM}^{\uparrow\downarrow}$. The
 208 obtained $G_{FM}^{\uparrow\downarrow}$ is $0.9 \times 10^{18} \text{ m}^{-2}$ at 375 K. As the temperature decreases, $G_{FM}^{\uparrow\downarrow}$ almost
 209 monotonically increases and reaches up to $3.4 \times 10^{19} \text{ m}^{-2}$ at 25 K. Note that at 25 K, $G_{FM}^{\uparrow\downarrow}$
 210 sharply increases by ~ 40 times relative to the value at 375 K. Such a dramatic increase in $G_{FM}^{\uparrow\downarrow}$
 211 is elusive and even unreasonable, particularly considering the weak SP efficiency between the
 212 Gd sublattice and the Pt layer, as well as the Gd sublattice dominating the net magnetization of
 213 GdIG at low temperatures. Essentially, the angular momentum transfer predominantly occurs
 214 between the effective Fe sublattice and Pt. Therefore, the enhanced damping ($\Delta\alpha_{Fe}$) and the
 215 magnetization (M_{Fe}) of the Fe sublattices should replace $\Delta\alpha_{SP}$ and M_s in Eq. 6, respectively.
 216 According to the SP model in ferrimagnets [see SI], the spin mixing conductance ($G_{FiM}^{\uparrow\downarrow}$) is
 217 estimated by $\Delta\alpha_{Fe} = \frac{g\mu_B}{4\pi M_{Fe} t_{GdIG}} G_{FiM}^{\uparrow\downarrow}$. As shown in Figure 3b, the values of $G_{FiM}^{\uparrow\downarrow}$ are
 218 $(2-9) \times 10^{18} \text{ m}^{-2}$ over the temperature range, which are comparable with the values for
 219 YIG/Pt^[48-50]. Consequently, the significant difference between $G_{FM}^{\uparrow\downarrow}$ and $G_{FiM}^{\uparrow\downarrow}$ further verifies

that the ferrimagnetic counterpart of SP should be carefully considered in ferrimagnetic systems, particularly in compensated rare earth-transition element ferrimagnets.

To gain further insights on the SP from compensated ferrimagnets, we measured the spin current under the FMR condition for the GdIG/Pt film. The uniform magnetization precession of GdIG excites coherent spin waves that inject spin angular momentum (spin currents, \mathbf{j}_s) into the adjacent Pt layer, in which a charge current (\mathbf{j}_c) can be generated *via* the inverse spin Hall effect (ISHE) due to the strong spin-orbit coupling in Pt^[51], as schematically drawn in Figure 4a. Figure 4b displays the SP-driven ISHE voltage signals ($V_{SP}(H)$) nearby FMR condition at sweeping positive (black) and negative (red) magnetic fields at a excitation frequency of 6 GHz at 375 K. The V_{SP} versus H curves exhibit a resonant Lorentzian line shape, and its sign reverses as the polarity of the sweeping magnetic field changes, confirming the ISHE origin following the relationship $\mathbf{j}_c \propto \mathbf{j}_s \times \boldsymbol{\sigma}$ with the spin polarization $\boldsymbol{\sigma}$. Figure 4c shows the microwave power dependence of $V_{SP}(H)$ at $f = 6$ GHz. As shown in the inset of Figure 4c, the extracted amplitudes of the $V_{SP}(H)$ scale linearly with the microwave power (P), indicating that the observed SP-induced ISHE is in the linear regime. The ISHE spectra were also measured at varying temperatures, and all exhibit a resonant Lorentzian line shape with a negative amplitude in the sweeping positive magnetic field, as shown in Figure 4d.

Spin current injection by incoherent spin waves can be triggered by applying a thermal gradient across the interface, *via* an effect called longitudinal SSE^[52,53]. As schematically shown in Figure 4e, an insulating SiO₂ layer was heated by a Ti/Pt strip with a charge current, giving rise to a temperature gradient (∇T) along the z axis. The incoherent spin waves induced by the ∇T *via* the SSE carry spin angular momentum into the adjacent Pt layer. The voltage signals [$V_{SSE}(H)$] can be detected in the Pt layer *via* ISHE by sweeping the applied magnetic field along the y axis. Figure 4f shows the $V_{SSE}(H)$ as a function of the applied magnetic field at various temperatures. The sign of $V_{SSE}(H)$ changes as the magnetic field polarity reverses, which agrees with the M - H hysteresis loops in Figure 1f. This confirms that the measured $V_{SSE}(H)$ originates from the spin current induced by ∇T for GdIG/Pt. Interestingly, the sign of the amplitude of $V_{SSE}(H)$ at the positive field at 300 K is opposite to that at 125 K, but identical to that at 10 K.

The amplitudes of the SP- and SSE-driven ISHE voltages as a function of temperature are shown in Figure 5a. It was established that the sign of $V_{SSE} = (V_{SSE}(+H) - V_{SSE}(-H))/2$ reverses

at ~ 72 K and reverses again at the magnetic compensation temperature (~ 200 K)^[54]. In contrast, the sign inversion was not observed in the SP experiment of the GdIG/Pt systems; the sign of V_{SP} remains negative over the measured temperature range. It appears counterintuitive that the sign of V_{SP} does not change at T_M , at which point the moments of the Fe and Gd sublattices reverse [see the schematic illustration in Figure 1g]. Moreover, the amplitude of V_{SP} decreases with temperature approaching to T_M . These results raise the question why the direction of the coherent and incoherent spin currents varies differently upon changing temperature, particularly when the temperature crosses the T_M .

To unravel the origin of the discrepancy in the sign variation in the SSE and SP experiments, we calculated the spin-wave spectra of GdIG at different temperatures, as shown in Figure 5b-d, in which the orange and navy lines represent the α modes (left-handed) and β modes (right-handed), respectively. The α and β modes host opposite spin polarizations. Noticeably, the spin-wave spectrum ($T < T_M$) of GdIG above the β_1 branch, dominated by the Fe sublattices, is very similar to those of YIG^[55], and the additional α_1 modes below the β_1 branch mainly originate from the contribution of the Gd sublattices. In the SSE experiments, the temperature-dependent V_{SSE} predominantly depends on the thermal occupation of the α and β modes. At very low temperatures, the thermal magnons from the β modes are frozen out owing to the large magnon gap. Consequently, below ~ 72 K, the contributions of the α modes dominate the SSE signals, while those of the β modes are almost negligible. Increasing temperature above 72 K, the magnon gap is gradually suppressed; therefore, the thermal magnons of the β modes are continuously activated^[25]. Due to the stronger exchange coupling between the Fe 3d-electrons and the Pt conduction electrons, compared to that for the local Gd 4f-electrons, the contribution from the β modes become dominant in the SSE signals, causing the change of the V_{SSE} sign at ~ 72 K^[54]. In contrast, the competition between the α and β modes is absent in the SP experiments because only the lowest α_1 mode in the GHz frequency range is uniformly excited under the FMR condition. Consequently, the sign of V_{SP} remains unchanged across ~ 72 K.

As the moment of Gd rapidly decreases with further increasing temperature, the net magnetization nearly vanishes with temperature up to $\sim T_M$. Meanwhile, V_{SP} decreases owing to the small net magnetization and the large damping. With increasing the temperature further

above T_M , the net magnetization is dominated by the Fe sublattices, and aligned by the external magnetic field, resulting in the polarization inversion of the spin-wave spectra across T_M ^[56]. Thus, the sign change of V_{SSE} reverses at ~ 200 K. It appears that V_{SP} should reverse as a consequence of the α modes being converted into the β modes above T_M . Nevertheless, the red shift of the β modes occurs with increasing temperature [see Figure 5b-d]. Above T_M , the frequency of the lowest β modes is lower than that of the α modes [see Figure 5d]. Upon applying an external magnetic field, the polarizations of all the magnon bands flip. As a result, the lowest β mode behaves similarly to the lowest α mode below T_M under the FMR condition. Therefore, the sign of V_{SP} remains unchanged across T_M .

Conclusion

In conclusion, we systematically investigated the SP and magnetic damping in insulating compensated ferrimagnetic GdIG films. In addition to the divergent bulk damping in GdIG films close to the compensation temperature, we observed the divergence of the nonlocal damping induced by SP in GdIG-based heterostructures. Furthermore, different from the case of the incoherent spin currents generated by SSE, the direction of the coherent spin currents generated by SP does not change when varying the sample temperatures. The physical reason behind these observations was explained by the ferrimagnetic SP model and the evolution of the handedness-related spin-wave spectra as the temperature increased across T_M . The spin-mixing conductance is self-consistently determined in the theoretical framework of ferrimagnetic SP. Our work offers significant fundamental insight into the SP and magnetic damping in compensated ferrimagnets, which will facilitate their applications in magnonics.

Experimental Section

Gd₃Fe₅O₁₂ (GdIG, thickness: 35 nm) films were grown on (111)-oriented Gd_{2.6}Ca_{0.4}Ga_{4.1}Mg_{0.25}Zr_{0.65}O₁₂ (sGGG) substrates by PLD at a laser repetition frequency of 5 Hz. During deposition, the substrate temperature was set to 640 K, and the oxygen pressure was 85 mTorr. After deposition, the samples were cooled to room temperature at a rate of 5 K/min. Thereafter, they were transferred to a magnetron sputter system, where a 10 nm-thick Pt layer was deposited at room temperature. Magnetization measurements were performed using a

309 magnetic property measurement system (MPMS3, Quantum Design). The structures were
310 characterized using STEM (Titan 80-300, FEI) after the sample was prepared using a focused
311 ion beam (Helios 450, FEI).

312 The FMR and SP-induced ISHE measurements were performed using a coplanar waveguide-
313 based setup in a physical property measurement system (PPMS, Quantum Design). Microwave
314 excitation was generated using a microwave signal generator (R&S®SMB100A). By sweeping
315 the magnetic field, the FMR transmission signals modulated by a Helmholtz coil were read by
316 the Stanford Research SR830 lock-in amplifiers based on a phase-sensitive technique. The SP-
317 driven ISHE signals were read using a Keithley 2182 meter. The details of the FMR and SP-
318 induced ISHE measurements are provided in the SI.

319 The devices for the SSE measurements were fabricated using a standard photolithography
320 technique and Ar ion milling. The GdIG/Pt film was first patterned into Hall bars with a width
321 of 50 μm , after which an 80 nm-thick SiO_2 film was deposited on it as an insulating layer.
322 The temperature gradient was applied using a heater strip [Ti(5 nm)/Pt(40 nm)] deposited on
323 the SiO_2 layer. Ti(10 nm)/Au(190 nm) stacks were deposited as electrical contacts. The SSE
324 measurements were performed in a PPMS chamber. A temperature gradient along the z axis
325 was built by a heater strip with a charge current of 10 mA (Keithley 6221 meter). The
326 temperature gradient induced SSE-generated and driven inverse spin Hall voltages in the
327 sweeping magnetic field along the y axis, which were detected using the Keithley 2182 meter.

328 **Supporting Information**

329 Supporting Information is available from the Wiley Online Library or from the author.

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335 **Conflict of Interest**

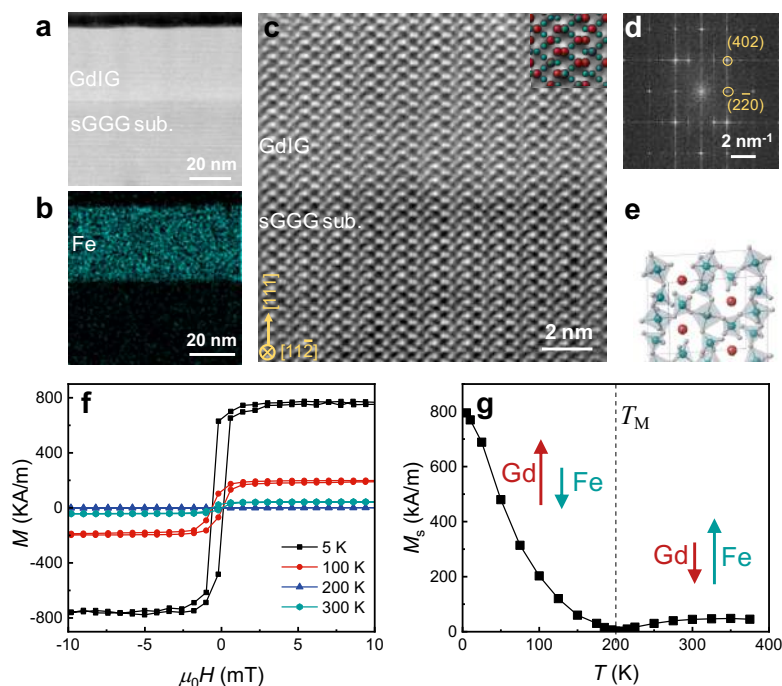
336 The authors declare no competing interests.

337

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- 427

428 **Figure**

429

430 **Figure 1.** Structure and magnetic properties of the GdIG film. a) STEM cross-sectional image.

431 b) EDX color mapping of Fe. c) Atomic-resolution STEM image. The enlarged part shows the

432 atomic model overlaid on the real atoms. The green and red balls represent the Fe and Gd

433 atoms. d) Corresponding FFT patterns of GdIG. e) Schematic of the crystal structure of GdIG.

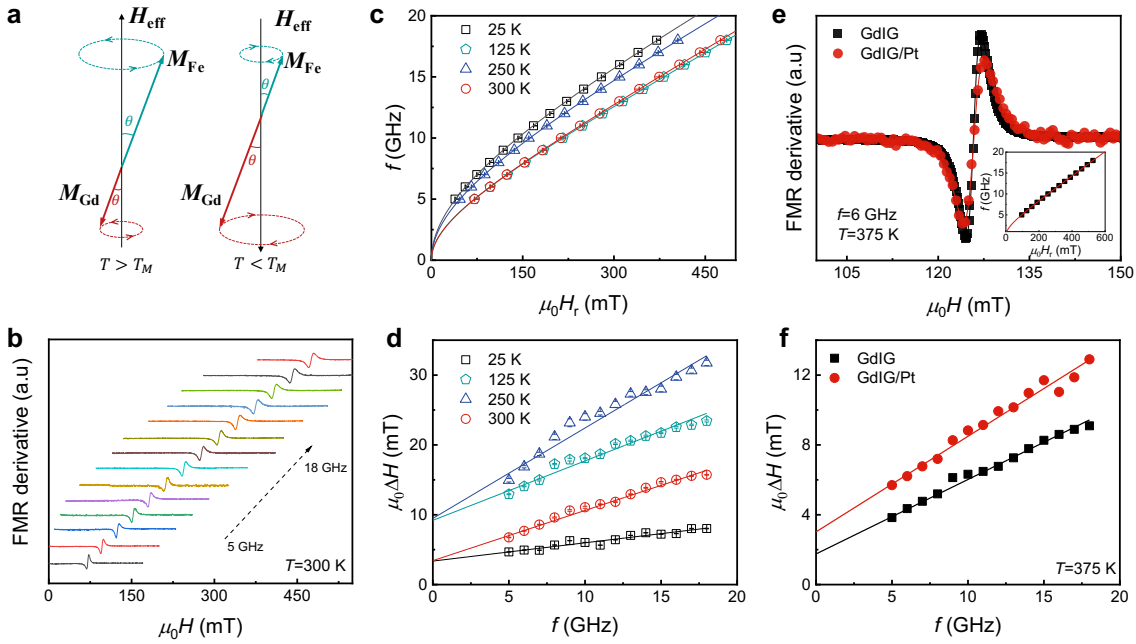
434 The green, red, and gray balls represent the Fe, Gd, and O atoms, respectively. f) Magnetic

435 hysteresis loop of the GdIG film. g) Saturation magnetization (M_s) as a function of temperature.

436 The arrows show the magnetic states above and below the magnetic compensation temperature,

437 respectively.

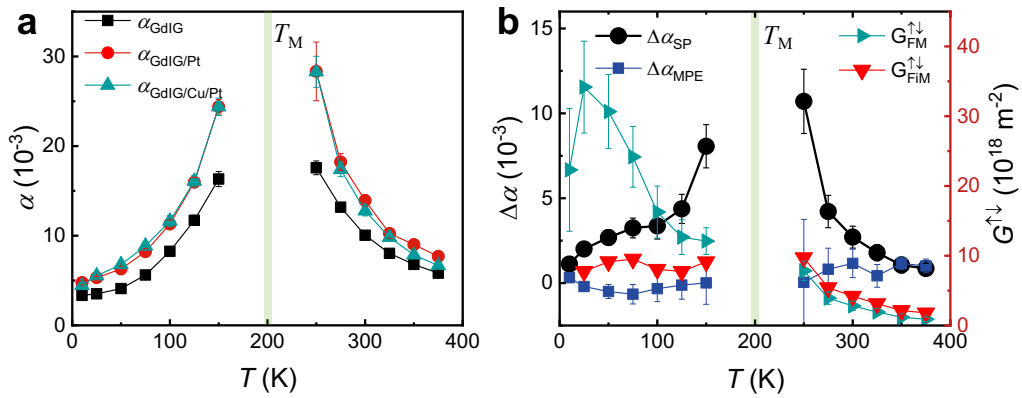
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440 **Figure 2.** FMR measurements of the GdIG and GdIG/Pt films. a) Low-frequency uniform
 441 modes in the FMR measurements. b) FMR spectra of GdIG (thickness: 35 nm) at 300 K. c)
 442 Frequency dependence of the resonant magnetic field (H_r) at various temperatures in GdIG. d)
 443 Resonant linewidth as a function of frequency at various temperatures in GdIG. e) Comparison
 444 of the FMR spectra for GdIG (35 nm)/Pt (10 nm) and GdIG (35 nm) at $f = 6$ GHz at 375 K. The
 445 inset shows H_r versus f for GdIG/Pt. f) Comparison of the linewidth versus frequency for
 446 GdIG/Pt and GdIG at 375 K.

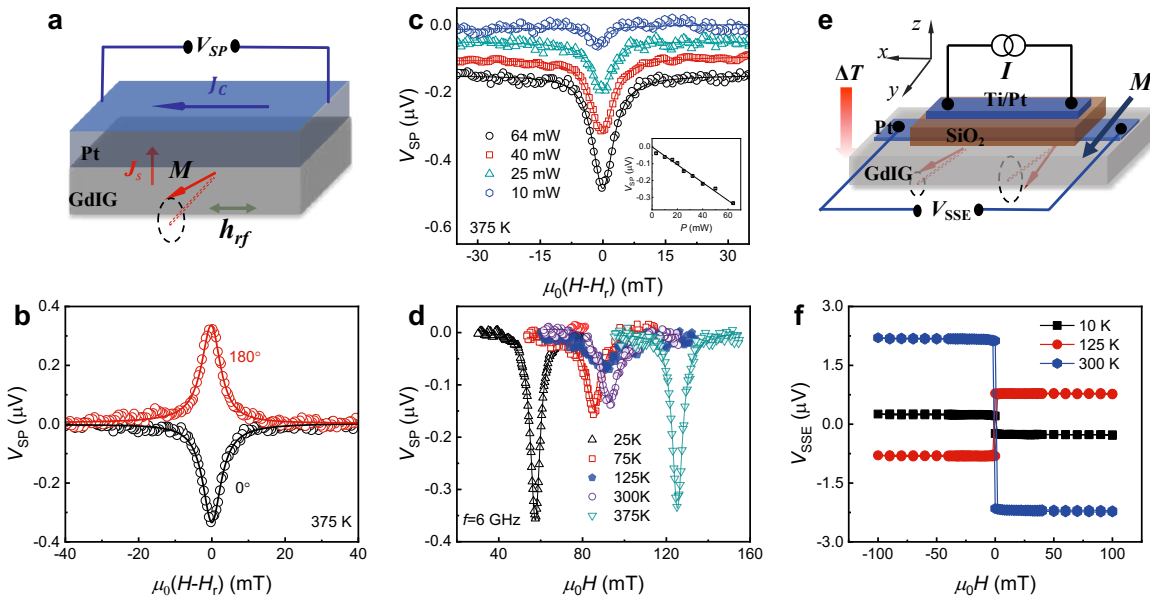
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149 **Figure 3.** Magnetic damping at various temperatures. a) Values of the effective Gilbert damping
 150 as a function of the temperature for the GdIG, GdIG/Pt and GdIG/Cu/Pt samples. b) Damping
 151 contributions from the SP effect and the MPE, and the evaluated spin-mixing conductance based
 152 on ferromagnetic and ferrimagnetic SP models. The color region corresponds to the magnetic
 153 compensation temperature T_M .

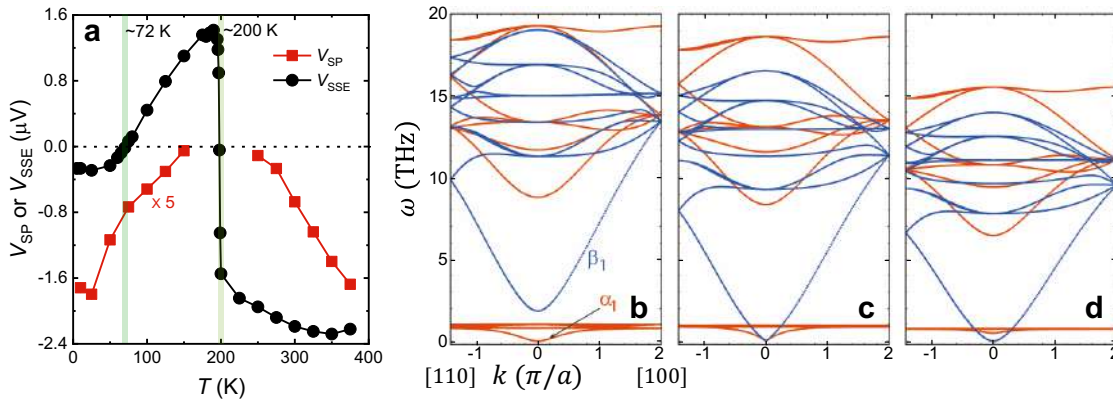
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456 **Figure 4.** Observation of the SP- and SSE-driven ISHE in GdIG/Pt. a) Schematic of the SP
 457 measurements. b) ISHE voltage (V_{SP}) nearby FMR condition ($H = H_r$) at $P = 64$ mW with
 458 sweeping positive (black) and negative (red) magnetic fields at 375 K. c) Microwave power
 459 dependence of V_{SP} at 375 K. The inset shows the extracted V_{SP} under the FMR condition. d)
 460 ISHE spectra at $P = 64$ mW in the SP experiments at various temperatures. e) Schematic of the
 461 SSE measurements. f) SSE signal (V_{SSE}) as a function of the applied magnetic field at various
 462 temperatures.

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465 **Figure 5.** SP- and SSE-driven ISHE voltages and spin-wave spectra at various temperatures. a)
 466 Comparison of the SP- and SSE-driven ISHE voltages as functions of temperature for the
 467 GdIG/Pt samples. The sign of V_{SSE} reverses at the corresponding temperatures (color regions).
 468 V_{SP} is multiplied by a factor of 5 for comparison. The calculated spin-wave spectra at b) $T < T_M$,
 469 c) $T \sim T_M$ and d) $T > T_M$. The orientations of the Gd moment are set to the same direction in all
 470 calculated spectra.

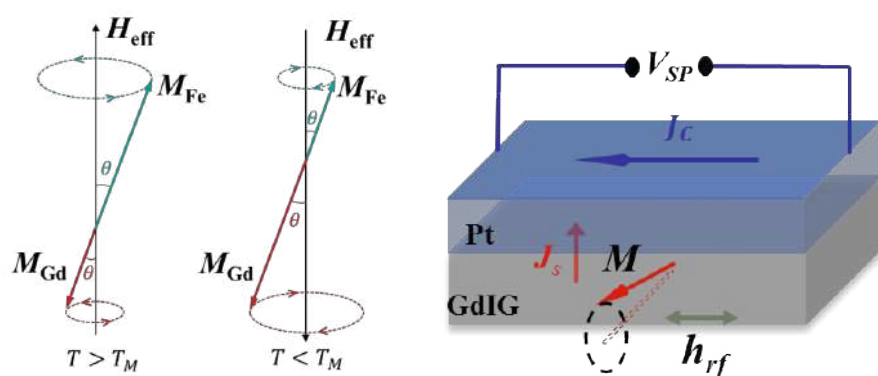
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Table of Contents

Yan Li, Dongxing Zheng, Bin Fang, Chen Liu, Chenhui Zhang, Aitian Chen, Yinchang Ma, Ka Shen*, Haoliang Liu, Aurélien Manchon, and Xixiang Zhang*

Unconventional spin pumping and magnetic damping in an insulating compensated ferrimagnet



An unconventional spin pumping together with the divergent nonlocal magnetic damping is unambiguously reported in the compensated ferrimagnetic $\text{Gd}_3\text{Fe}_5\text{O}_{12}$ -based heterostructures. The coherent and incoherent spin currents undergo a distinct direction change with the variation of temperature. The underlying physical mechanisms are self-consistently clarified by the ferrimagnetic counterpart of spin pumping and the handedness-related spin-wave spectra.