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Mode I and II R-curves characterization of the Maritime Pine and Spruce under the same geometry

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ABSTRACT

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This study provides detailed data on the quasi-brittle behaviour of two softwood species, the Maritime Pine (*Pinus pinaster.*) and the Spruce (*Picea abies.*), by using R-curves. The methodology of characterization of the R-curve used in this article is an improvement of the method presented by Lespine [25]. This method is automated and based on the measurement error made on $G_R(\Delta a_u)$. And the same geometry to characterize the crack propagation in mode I and in mode II. The influence of the crack plane orientation and the density on the fracture mechanisms is investigated for both species and shows a dependency of the R-curve parameters of the Maritime Pine in mode I to the crack plane orientation. The differences between Spruce and Maritime Pine are highlighted. This demonstrate the importance to characterize the quasi-brittle behaviour of each wood species in order to provide a useful data base for engineers.

21 Nomenclature

- a_0 initial crack length
- b width of the specimen
- f_t tensile strength
- G_{exp} experimental crack propagation resistance
- G_f cohesive fracture energy
- 27 $G_{f_{L}}$ crack bridging energy
- 28 $G_{f_{i}}^{j_{b}}$ microcracking energy
- G_R crack propagation resistance
- $G_{R_{a}}$ critical crack propagation resistance
- 31 $G_{R_{ci}}$ critical crack propagation resistance for mode i
- 32 K_{ini} initial stiffness
- 33 P load applied to specimen
- P_u ultimate load
- 35 P.U ultimate load
- β factor which described the damage regime of the R-curve
- ³⁷ $β_i$ value of β for mode i
- $\Delta a \operatorname{crack} \operatorname{length}$
- 39 Δa_{c_i} critical crack length for mode i
- 40 Δa_{eq}^{i} equivalent crack length
- 41 γ annual growth ring orientation
- 42 $\kappa_{G_{R_{err}}}$ measurement error on the experimentam crack propagation resistance
- 43 λ compliance of the specimen
- 44 CZM Cohesive Zone Model
- 45 FPZ Fracture Process Zone
- 46 LTENF Linear Tapered End Notched Flexure
- 47 MC% Moisture Content

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48 1. Context

In order to improve the design methods for structural timber elements subjected to splitting, the damage and crack propagation mechanisms should be taken into account accurately by using specified mechanical properties. For this, in 1988, Gustafsson [18] provides a method to predict the load-bearing capacity of wooden notched beams based on the linear elastic fracture mechanics. This method was then implemented in Eurocode 5 [16] with parameters determined according to the current knowledge mainly based on the Spruce (*Picea abies.*) properties [1]. Nowadays, the knowledge focused on failure mechanism of wood has improved significantly, but the standards and design rules still remain the same.

However, recent studies demonstrate some limits of the current design method. First of all, in these methods the wood is considered as a perfectly brittle material with no transition phase between its elastic behaviour and the crack propagation, because only the critical crack propagation resistance is considered to estimate the ultimate load of the structural elements. But, some experiments reveal that a transition phase does exists with no plasticity and demonstrate the quasi-brittle behaviour of wood [28]. In this case, considering wood as a perfectly brittle material can lead to an overestimation of the load-bearing capacity ; because the progressive damaging of the element before the auto-similar crack propagation is not taken into account [7].

Secondly, the crack propagation in mode I is the most investigated one, whereas recent studies based on digital
 correlation measurements of strain fields clearly show the influence of the crack propagation in mode II for notched
 beams [20, 34]. The mode I is an opening of the crack due to tensile stress. The mode II is the propagation of the crack
 causes by in plane shear stress and the mode III is the one due to out of plane shear stress also named tearing.

Finally, the parameters used in the Eurocode 5 were calibrated mainly on the Spruce properties by assuming a correlation between the wood density and the fracture properties. But the literature shows that the failure properties are very different from a species to another one [15], and a correlation between the wood density and the fracture properties is not always found and accurate [21].

Currently, the literature presents some experimental investigations carried out to determine the mechanical prop-71 erties governing the damage and failure of wood in mode II [1, 13, 35, 37], but these works are much scarcer in 72 comparison with those in mode I. Some studies were also conducted to determine the fracture properties of wood 73 in mixed-mode [12, 19, 26, 36] but these works are even scarcer than the one in mode II. However, recent stud-74 ies [20, 23, 33, 34] demonstrate that the splitting of wood in structural elements is mostly governed by the combination 75 of mode I and II, with a non negligible contribution of the mode II. So, as a revision of the Eurocode 5 is conducted [14], 76 the design methods of the structural elements submitted to splitting should be investigated again in order to consider 77 the latest knowledge relative to the crack propagation in wood. This implies experimental campaigns rigorously chosen 78 in order to characterise the degree of anisotropy of the failure properties. So, this study presents a methodology and 79 the corresponding results to supply additional knowledge relative to the failure mechanisms of wood. 80

The quasi-brittle behaviour of the wood is due to the presence of a damaged area in front of the "visible" crack 81 tip [28]. This area is named Fracture Process Zone (FPZ) and presents micro-cracking and crack bridging. The FPZ 82 may measure several centimetres. This local damage should be well considered because it produces a progressive 83 decrease of the local stress transfer which can lead to the sudden failure of the structural element. In this case, a design 84 method based on the brittleness assumption might overestimate the load-bearing capacity of the structural element. 85 This particularity of the wood has been studied since several years [10, 11, 28]. However the phenomenon is still 86 not currently considered in the design methods. Some studies considered the quasi-brittleness of the wood to design 87 structural element [5, 8, 17] but they used non-linear models which are too specific and too greedy in numerical 88 resources, limiting their uses by engineers. Moreover, the fracture properties implemented in those models are not 89 based on dedicated experimental campaigns, while size effects and boundary conditions may influence the results [4]. 90

In order to consider the quasi-brittle behaviour of wood without the inconvenient of the non-linear models, some 91 studies [24, 33] used Equivalent Linear Elastic Fracture Mechanics (eqLEFM) proposed by Bazant [2, 3] and applied 92 to wood by Morel [28]. This method considers the damage of the material by using an equivalent crack length (Δa_{ea}) 93 94 able to reproduce the FPZ development, and the auto-similar crack propagation according to the compliance evolution of the specimen. This leads to the determination of the crack resistance propagation function (G_R) according to the 95 equivalent crack length, which is an R-curve. This one depends on the material, and is regarded as a fracture property of 96 97 the material. So, with the eaLEFM, the progressive failure of the specimen during the damage phase can be considered by using R-curves which are more elaborated and complete than the fracture energy (G_f) [18] only. 98

⁹⁹ However, the R-Curve characterization implies to obtain experimentally a stable phenomenon from damage initi-



Figure 1: Schema of the specimen geometry (LTENF) with dimensions in mm

ation to the auto-similar propagation of the crack. But, in the literature, several specimen geometries have too short 100 ligament (CT, SENB for instance) to reach the critical energy release rate corresponding to the fracture energy [6]. And, 101 although the R-curve is a material property, its characterization is dependent on the specimen geometry [3, 27, 30]. This 102 dependence is supposed to be linked to the average compliance of the specimen. This can explain the differences in the 103 fracture properties proposed by the literature [15, 25, 28, 32, 35], which complicates the comparison between studies. 104 All those specificities imply that the choice of the specimen geometry used to characterize the fracture properties is cru-105 cial for quasi-brittle material. Moreover as the wood is an orthotropic material with three principal orientation (radial 106 (R), tangential (T) and longitudinal (L)) the crack plane orientation can have an influence one the fracture properties. 107 Two crack planes orientation (RL) for Radial/Longitudinal and (TL) for Tangential/Longitudinal were investigated. In 108 those cases the crack propagates along the longitudinal direction in parallel (RL) or in perpendicular (TL) to annual 109 growth rings as shown by figure 2. 110

The R-curves describe the FPZ development which is governed by the geometry, the elastic properties and the mode of crack propagation. So, to predict the failure of structural elements subjected to splitting, where cracks are developing due to mode I and II [9, 20, 34], the R-curves of each mode must be characterized.

The main goal of the current study is to provide fracture properties of Spruce(*Picea abies.*) and Maritime Pine (*Pinus pinaster.*)for mode I and mode II using the same specimen geometry. The interest is to avoid geometry and size effect dependence, of fracture properties. In a second time, the study aims at providing data of the fracture properties of Maritime Pine and Spruce for mode I and II crack propagation according to the orthotropy of the wood. Thus, correlations between R-Curves and the wood density or the annual growth rings orientation (from RL to TL) are proposed.

120 2. Methods

121 2.1. Specimens characteristics

The selected fracture specimen geometry (named LTENF) (Figure 1) was taken from a previous experimental 122 campaign carried out by Aicher *et al.* [1]. The P line represents the loading line for mode I test and the center of the 123 support for the mode II test, the value of a_0 is given from this line. It was mainly used for mode II characterization 124 of wood as a brittle material. This specific geometry was chosen because it ensure a stable crack propagation near the 125 initial crack (a_0) and during all the microcraking process, which can be of many centimetres for some wood species, 126 like the Maritime Pine. This stability is crucial to characterize the fracture properties of quasi-brittle material. In order 127 to take into account the geometry specificities and to avoid uncertainties on the calculation of the energy release rate, 128 the evolution of the compliance of the specimen was computed using a finite element model. Others geometries in the 129 literature, like de ENF or DCB, do not provide stable cracking for the mode I and the mode II unless the mechanic press 130 is very specially monitored. Which is more difficult and expensive in raw materials than using a finite element model 131 in place of the beam theory. A geometry presenting a similar slenderness ratio was used by Phan [31] to characterize 132 the mode I R-curve of Maritime Pine. 133

In order to be representative of the wood variability 92 tests were carried out for Maritime Pine and 90 for Spruce. The crack propagation plane orientation [1] goes from RL to TL with a quiet homogeneous repartition of the tests. This orientation was determined by image analysis for every specimen tested (Figure 2). The picture was taken in order to be perpendicular with the crack propagation direction. The annual growth ring orientation was determined as the

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Figure 2: Measurement of the annual growth ring orientation (γ) of a test specimen



Figure 3: Opening test for mode I

mean value of three annual growth ring orientation present on (γ_1) , below (γ_2) and under (γ_3) the crack path.

Free-defect specimens were chosen to avoid the heterogeneities on the supposed crack path and to privilege a straight wood grain. The wooden pieces were previously conditioned to a temperature of 20°C and a relative humidity of 65% to obtain a moisture content close to 12%. The average density at a moisture content of 12% is equal to 470 $kg/m^{/3}$ for the Spruce and 570 kg/m^3 for the Maritime Pine.

143 2.2. Test protocol

The mechanical characterizations were carried out on a 50 kN MTS criterion electro-mechanic press (MTS Systems). Experiments in mode I consist in an opening test (Figure 3) and the corresponding mode II is a 3 points flexion test (Figure 4).

For each test the same protocol was strictly followed. First, the initial notch was realised at the earliest 48 hours 147 before the test. Then a little knife cut was performed just before the test beginning (1 mm long). The controlled 148 displacement of the test was determined according to the wood species in order to ensure a short time protocol between 149 3 and 7 minutes. After each failure of the specimens, the moisture content (MC%) was measured as well as the local 150 density on the observed crack path at MC% = 12% and MC% = 0% (anhydrous state). The moisture measurements were 151 carried out according to the ISO-13061-1 standard, this standard is not official for research, and the weighing material 152 was calibrated. The objective of these tests is to measure the compliance variation due to the damage development and 153 the potential crack extension of the specimen, which can be done by using the load-displacement curve (Figure 5(a)) if 154 the displacement is measured on the loading line. For mode I, the compliance relative to the fracture specimen opening 155 at the load support is measured by image tracking. For mode II, the displacement is measured using an LVDT fixed 156 to a system avoiding the crushing at the supports (Figure 4). 157



Figure 4: 3-points bending test for mode II



Figure 5: Example of a Load-Displacement curve (a) used to determine the characteristic R-curve (b) of a specimen (Mode II)

158 2.3. Experiments

In the literature, the cohesive zone models (CZM) are presented as the most efficient method to characterize the 159 quasi-brittle behaviour of wood [6, 7, 29]. In order to characterize its corresponding parameters from the R-curves, 160 which are determined from the experimental load-displacement curves, a step by step correspondence linking the soft-161 162 ening function to $G_R(\Delta a)$ was established by Morel et al. [29]. So, the plateau value (G_{Rc}) of the R-curve corresponds to the cohesive fracture energy (G_f), which is the total energy required to completely separate the material in a pure 163 mode of failure. The critical equivalent length (Δa_c) is representative of the FPZ length which is governed by ultimate 164 displacement in the material. The parameter β is influenced by the energy distribution between the microcracking en-165 ergy $(G_{f\mu})$ and the crack-bridging energy (G_{fb}) , the value of β increases when the crack-bridging energy increases [6]. 166 Finally, the initial part of the R-curve is determined by the tensile strength (f_t) of the material which governs the devi-167 ation from the linearity part of the load-displacement curve. So, by this correspondence, the R-curve parameters can 168 be linked to the characteristics fracture mechanisms of quasi-brittle materials. 169

In order to determine the experimental crack resistance propagation of a specimen, the relation between the equivalent crack length (Δa_{eq}) and the compliance (λ) must be known. This relation is determined in post-treatment using a finite element model. For both tests a numerical model is realised with two calculations each, for each wood species. The crack propagation is simulated by the decohesion of the nodes along the crack path. This provides a numerical relation between the equivalent crack length and the numerical compliance ($\lambda_n \left(\Delta a_{eq_n} \right)$). Then the experimental relation between λ and Δa_{eq} can be determined by a dichotomic approach. However, in order to take into account the variability of the elastic properties of wood the numercial relation $\lambda_n \left(\Delta a_{eq_n} \right)$ must be corrected by a correction coefficient $(c_{n/e})$. This coefficient is determined by the ratio between the experimental and numerical initial toughness (K_{ini}) which are only influenced by the elastic properties and the initial state of the test (equation 1).

$$c_{n/e} = \frac{K_{ini_e}}{K_{ini_n}} \tag{1}$$

This method was presented by Morel *et al.* [29] and also used by Lespine [25] and Phan [31]. As the relation between the experimental compliance (λ) and the equivalent crack length, noted Δa in the rest of this article, is known, the evolution of the experimental crack resistance propagation $(G_R(\Delta a))$ (Figure 5(b)) can be calculated using the equation 2 [28]($P(\Delta a)$ the load and b the width of the specimen):

$$G_R(\Delta a) = \frac{P^2(\Delta a)}{2b} \lambda'(\Delta a) \tag{2}$$

Equation 2 is only valid in the non-elastic domain. The end of the elastic domain is determined by choosing a load
 value corresponding to the deviation of the load-displacement curve from the initial regression as shown on figure 5(a).
 The end of the elastic domain corresponds to the passage from the black line to the red line.

Morel [28] proposed to idealize experimental R-Curves by a 3-parameters power law (Equation 3). Nevertheless,
 this equation can have multiple solutions. So, a strategy of determination is proposed in this study to extract different
 parameters, and detailed below.

$$G_{R}(\Delta a) = \begin{cases} G_{Rc} \left[\frac{\Delta a_{c}}{\Delta a} \right]^{\beta} & if \ \Delta a < \Delta a_{c} \\ G_{Rc} & if \ \Delta a \ge \Delta a_{c} \end{cases}$$
(3)

The critical energy release rate or plateau value of the R-curve (G_{Rc}) which conditioned the values of the two others parameters must be determined at first. Indeed, the critical crack length (Δa_c) and the damage regime described by (β) are dependent of the transition between two energetic states. The undamaged state $G_R = 0$ and the auto-similar crack propagation state $G_R = G_{Rc}$, the R-curve ensure the continuity between those states, which govern the process zone (FPZ) development. So, the determination of the plateau value on the experimental R-curve leads to the value of G_{Rc} .

In order to connect this value to the experimental reality, it was chosen to consider the measurement errors ($\kappa_{G_{Rerr}}$) relative to G_R as a critical parameter of determination. So the plateau of $G_R(\Delta a)$ was determined by dragging an interval from the minimum to the maximum value of the experimental energy release rate (G_R) (Figure 6(a)). G_{Rc} is the value on which the interval containing the maximum of experimental points is centred. For this experimental campaign, $\kappa_{G_{Rerr}}$ was equal respectively to 3 J/m^2 for the mode I and to 30 J/m^2 for the mode II, according to the experimental device.

Then the critical crack length (Δa_c) is determined on the curve $log(G_R)/log(\Delta a)$ as the intercept between the linear regression of the experimental curve on the interval $[log(G_R(0)); log(G_R(G_{Rc}))]$ with the straight equal to $log(G_R(G_{Rc}))$ (Figure 6(b)). With this method, β is determined from the slope of the linear regression. Nevertheless, there is often some noise at the beginning of curve, so, to minimize this influence, the interval of the linear regression is reduced to ensure the best regression possible. It is important to notice that the critical crack length is not always equal to the ultimate load crack length (Δa_u) which corresponds to the crack length where the ultimate load (P_u) is reached.

A reverse analysis is then performed by comparing the load-displacement curve generated from the 3-parameters R-curve, using equation 4, to the experimental load-displacement curve (Figure 7(a)). The blue dash curve allows checking that equation 4 can be used to determine a load-displacement curve from a given R-curve. If the experimental load-displacement curve and the blue dash one are equal then equation 4 can be used. The red dash curve represents the load-displacement curve obtained by using equation 4 on the 3-parameters R-curve which fitted the experimental curve (red dash curve on figure 7(b)). This approach aims at checking if the theoretical R-curve (Figure 7(b)) determined by



Figure 6: Determination of the plateau value (G_{Rc}) (a) and linear regression (b) on the $log(G_R(log(\Delta a)))$ function in order to obtain β and Δa_c for the characteristic R-Curve (Mode II)



Figure 7: Comparison between the non-linear party of the Load-Displacement curves (a) corresponding to the experimental and the fitted R-Curves (b) (Mode II)

this method can be used to characterize the fracture properties of the wood.

$$P(\Delta a) = \sqrt{\frac{2bG_R(\Delta a)}{\lambda'(\Delta a)}}$$
(4)

214 $P(\Delta a)$ is the load for a given crack length Δa . *b* is the specimen widt. G_R is the evolution of the crack resistance 215 according to the equivalent crack length Δa . And $\lambda'(\Delta a)$ is the derivative of the compliance.

216 3. Results

Some tests were rejected from the final results because no plateau was observed. This can be due to the residual presence of knots along the crack path, or a too highly deviation of the wood grain (Figure 8(a)). In the case of mode II, this can also be due to a too short ligament of the specimen (Figure 8(b)). Finally, this led to a rejected rate of 15% for mode I tests and 30% for mode II.



Figure 8: Example of an R-Curve with an energy leap (a) (Mode I) and another where the energy release rate constantly increases without reaching a plateau value before failure (b) (Mode II)

Table 1

Experimental results for Spruce ($\rho_{12\%} = 470 kg/m^3$)

Parameters	G_{Rc_I}	$G_{Rc_{II}}$	Δa_{c_I}	$\Delta a_{c_{II}}$	β_I	β_{II}
	J/m^2	J/m^2	(<i>mm</i>)	<i>(mm)</i>	(-)	(-)
Mean value	190	800	12	22	0.14	0.54
Standard deviation	60	180	8	8	0.06	0.17
CoV	32%	23%	65%	36%	45%	31%

Table 2

Experimental results for Maritime Pine ($\rho_{12\%} = 570 kg/m^3$)

Parameters	G_{Rc_I}	$G_{Rc_{II}}$	Δa_{c_I}	$\Delta a_{c_{II}}$	β_I	β_{II}
	J/m^2	J/m^2	(<i>mm</i>)	(<i>mm</i>)	(-)	(-)
Mean value	500	1190	24	25	0.25	0.63
Standard deviation	210	275	15	6	0.10	0.10
CoV	43%	23%	61%	22%	41%	16%

221 3.1. R-Curve parameters values

Table 1 and 2 present the average values of the R-curve parameters in mode I and II for Spruce (*Picea abies.*) and Maritime Pine (*Pinus pinaster.*).

For Spruce, the value obtained for G_{Rc_I} (190 J/m^2) and $G_{Rc_{II}}$ (800 J/m^2) are similar to the ones of the literature [15, 25, 32]. But, the critical crack length Δa_{c_I} and Δa_{cII} are two times lower than those of the literature [25, 32], and the value of β_{II} is much higher than β_I which is different from the results of Phan et al. [32].

However, the comparison between mode I and mode II parameters exhibits the degree of anisotropy of the material. The critical energy release rate (G_{R_c}) and β are 4 times higher in mode II. A higher β in mode II indicates a significant crack-bridging in the development of the FPZ. The critical crack length (Δa_c) is 2 times higher in mode II, so the failure is more quasi-brittle in mode II than in mode I. With such important differences the mode II crack propagation should not be ignored for the prediction of structural elements in wood subjected to splitting.

For the Maritime Pine the value obtained for G_{Rc_1} (500 J/m^2) is in accordance with the value obtained by Phan [31]



Figure 9: Evolution of G_{Rc_I} (a), Δa_{c_I} (b) and β_I (c) according to the crack orientation plane for both species

but higher than the one found by Dourado et al.[15]. The value of Δa_{c_r} is 30% higher than Phan [31] when β_I is similar. 233 These differences reveal the importance to propose a retrofitting method to extract the R-curve. The comparison 234 between the R-curve parameters in mode I and II shows that $G_{Rc_{II}}$ and β_I are 2.5 times higher in mode II when Δa_{c_I} 235 and Δa_{cII} are similar. The lack of data in the literature is significant for the Maritime Pine contrary to the Spruce. This 236 highlighted the interest of the present study to supply data on this species. Indeed, the critical energy release rate G_R 237 of the Maritime Pine is 2.5 higher in mode I and 1.5 higher in mode II than the one of the Spruce. So, the Maritime 238 Pine is globally more resistant to the crack propagation than the Spruce. Moreover, unlike Spruce, the critical crack 239 length are equal in mode I and II for the Maritime Pine. β_I is 2 times higher for the Maritime Pine, but β_{II} is almost 240 the same for both species. So, the Maritime Pine is less brittle in mode I, but the mode II crack propagation of both 241 species are similar in terms of behaviour. 242

243 3.2. Influence of the crack plane orientation

In this part, the influence of the crack plane orientation according to the annual growth ring orientation on the Rcurve is investigated for both species. The influence on the mode I R-curve parameters is presented on figures 9(a), 9(b)and 9(c) and on the figures 10(a), 10(b) and 10(c) for the mode II. When it was relevant the equation of the correlation straight line was added to the graphic.

Those graphics exhibit an influence on G_{Rc_I} and β_I for the Maritime Pine, whereas it is not the case for Spruce. This leads to values of G_{Rc_I} and β_I really similar to Spruce for RL ($\gamma = 0^\circ$) orientation, with a Δa_{c_I} 2 times higher. And for the TL ($\gamma = 90^\circ$) orientation, G_{Rc_I} and β_I are 4 times higher for the Maritime Pine. So, according to the crack plane orientation in mode I, the Maritime Pine can have a significantly higher fracture energy and a less brittle



Figure 10: Evolution of $G_{Rc_{II}}$ (a), $\Delta a_{c_{II}}$ (b) and β_{II} (c) according to the crack orientation plane for both species

252 behaviour than the Spruce.

On another hand, the figures 10(a),10(b) and 10(c) that illustrate the influence of the crack plane orientation on the R-curve parameters in mode II do not show any significant correlation for both. Moreover, $\Delta a_{c_{II}}$ and β_{II} are almost equal for both species. The differences between those two species are less important in mode II than in mode I, and are only obtained for the critical energy release rate ($G_{Rc_{II}}$). Some graphics seem to show some correlation, but as the R^2 is inferior to 0.5 no conclusion can be made.

258 3.3. Influence of the wood density

The influence of the wood density on the R-curve parameters is investigated for both species in mode I (Figures 11(a),11(b) and 11(c)) and II (Figures 12(a),12(b) and 12(c)). The density of each specimen was measured on a cube of approximately 20 mm side taken close to the crack tip after each test. The results are given for a moisture content of 12%.

For G_{Rc_I} (Figure 11(a))the values obtained for the Maritime Pine seem to decrease for the highest values of wood density, but there is not a clear correlation between these two parameters ($R^2 < 0.2$). The variability is very high and does not allow to have a good correlation. It can be a consequence of the crack plane orientation dependence of the R-curves parameters in mode I. On another hand, the value obtained for the Spruce, which are not influenced by the crack plane orientation, does not show any correlation between G_{Rc_I} and the wood density too. When both species are considered together, the R^2 is inferior to 0.2, which is to low to exhibit any influence of the density on G_{Rc_I} . The influence of the density cannot be decoupled from the influence of the species. For Δa_{c_I} (Figure 11(b)) and β_I



Figure 11: Evolution of G_{Rc_I} (a), Δa_{c_I} (b) and β_I (c) according to the wood density for both species

(Figure 11(c)), the variability of the values obtained for both species do not allow to observe any correlation. So, for
the mode I, the results of the experimental campaign presented in this article does not show any clear influence of the
wood density on the R-curve parameters for the Spruce and the Maritime Pine.

The variability of the R-curve parameters values in mode II is almost the same for both species. This observation is in accordance with the assumption that the variability observed, for the Maritime Pine, on the R-curve parameters in mode I is due to the influence of the crack plane orientation. For $G_{Rc_{II}}$ (Figure 12(a)) and $\Delta a_{c_{II}}$ (Figure 12(b)) there is no correlation with the wood density for both species. For β_{II} (Figure 12(c)), the values obtained for the Spruce seem to increase according to the wood density, but the value of R^2 is inferior to 0.2, so there is no correlation between those two parameters. So, even more clearly than for the mode I, there is no influence of the wood density on the R-curve parameters in mode II.

These results are in accordance with the observation made by Jockwer et al. [21] who question the relation proposed
by Larsen and Gustafsson [22] which link the fracture energy to the wood density.

Figure 13(a) and 13(b) show no correlation between the annual growth ring orientation and the wood density. So there is no hidden influence between those two parameters.

284 4. Discussions and Conclusions

This article proposes a new methodology to characterize the quasi-brittle properties of wood, by using R-curves, based on the method presented by Lespine [25]. The main improvement of this methodology compared to Lespine's



Figure 12: Evolution of $G_{Rc_{II}}$ (a), $\Delta a_{c_{II}}$ (b) and β_{II} (c) according to the wood density for both species

one is to automatically determined the plateau value G_{Rc} by using the measurement error made on $G_R(\Delta a_u)$. This 287 improves the determination of G_{Rc} , but also the one of β and Δa_c , which are both strongly linked, to the plateau value 288 by preventing mostly the statistical variations in the results induced by the subjectivity of the observer. This method is 289 an important step for the characterization of the quasi-brittle properties of wood, but may still be improved. Indeed, the 290 rejected rate of 15% in mode I and 30% in mode II may be reduced by increasing the ligament length and by eliminating 291 every knot on the crack path. But this second point would drastically increase the needs in raw materials in order to 292 obtain specimens of clear wood long enough for the chosen geometry. And, in order to be in accordance with the on site 293 construction challenges, may be the knots should be taken into account into the statistic studies. Nevertheless, by using 294 this methodology, the R-curves parameters of two wood species (Maritime Pine and Spruce) have been determined 295 for two crack propagation mode (mode I and II). And, the differences of behaviour between those two species were 296 297 highlighted, with a more brittle behaviour for Spruce than for Maritime Pine. Moreover, contrary to the Spruce, the Maritime Pine crack propagation in mode I showed a dependency to the crack orientation plane. The differences 298 between the mode I and the mode II crack propagations were also highlighted by this work, with a G_{Rc} 2 to 4 times 299 higher in mode II than in mode I according to the wood species. And, the mode II of propagation appears more stable 300 than the mode I and less influenced by the other properties of the wood (density and crack plane orientation). 301

This study does not show a clear correlation between the wood density and the R-curves parameters. But, in the case of the Maritime Pine, the dependency of the R-curve parameters to the crack plane orientation may create an artificial variability which could hide a possible correlation. In order to eliminate any possible correlation between the wood density and the R-curve parameters in mode I, an experimental campaign which investigates the influence



Figure 13: Correlations between the wood density and the crack orientation plane for both species for the mode I (a) and the mode II (b)

of the wood density for specific crack plane orientations should be carried out. Such experimental campaign should also be carried out on other wood species in order to increase the density range investigated. Moreover, the differences observed between only two wood species, which are both softwood, show the necessity to establish a data base of the R-curves parameters for many wood species and more particularly for hardwood, by using a rigorous and systematic methodology. To conclude, this study is another step to improve the methodology to characterize the quasi-brittle behaviour of wood. In the future, in order to improve the characterization of the R-curve parameters, the focus should be given to the determination of the plateau value G_{Rc} .

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