

Observation of Bulk Plasticity in a Polycrystalline Titanium Alloy by Diffraction Contrast Tomography and Topotomography

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Abstract

The mechanical properties of polycrystalline metals are governed by the interaction of defects that are generated by deformation within the 3D microstructure. In materials that deform by slip, the plasticity is usually highly heterogeneous within the microstructure. Many experimental tools can be used to observe the results of slip events at the free surface of a sample; however, there are only a few methods for imaging these events in the bulk. In this article, the imaging of bulk slip events within the 3D microstructure is enabled by the combined use of X-ray diffraction contrast tomography and topotomography. Correlative measurements between high-resolution digital image correlation, X-ray diffraction contrast tomography, topotomography and electron backscattered diffraction are performed during deformation of Ti-7Al to investigate the sensitivity of the X-ray topotomography method for the observation of slip events in the bulk. High-resolution digital image correlation and electron backscattered diffraction measurements uniquely provide slip and lattice rotation activities at the surface of the specimen during deformation and are used to corroborate the diffraction contrast tomography and topotomography measurements. Much larger neighborhoods of grains were mapped by topotomography measurements than in previous studies, enabling quantitative measurements of slip transmission. Significant differences were observed between surface and bulk grains, indicating the need for 3D observations of plasticity to better understand deformation in polycrystalline materials.

Keywords: Diffraction Microstructure Imaging, X-ray Topotomography, Diffraction Contrast Tomography, High Resolution Digital Image Correlation, Slip, Plastic localization, Slip Transmission, Titanium Ti-7Al

1. Introduction & Background

During plastic deformation, metallic materials may experience localized slip on glide planes over numerous inter-atomic distances [1]. As a consequence of the formation of slip bands within a material, steps form where the slip bands intersect the surface of the sample and can be observed using optical [2], confocal [3], scanning electron [4], and atomic force microscopy [5, 6]. Information about the deformation processes that occur during monotonic and cyclic loading is obtained by analyzing the slip traces and their characteristics at the surface in relation to the microstructure. Such information has provided a deep understanding of deformation processes that occur in the plastic regime. For instance, the analysis of surface slip traces in the micro-plastic regime (plasticity prior to macroscopic yield) [7] has provided insights on dislocation interactions, avalanche processes and incipient slip and their effect on mechanical properties [8]. The development of high-resolution digital image correlation (HR-DIC) inside the scanning electron microscope (SEM) provided opportunities to statistically investigate incipient slip in metallic materials [9], and to

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relate the intensity of surface slip events to fatigue properties [10]. The analysis of slip traces by SEM, focused ion beam SEMs (FIB-SEMs) and/or atomic force microscopy for specimens deformed cyclically in the macroscopic plastic regime has led to the identification of the micro-scale characteristics of fatigue structures such as persistent slip bands or deformation bands [11–13]. These analyses have provided new insights into fatigue crack nucleation mechanisms, and in the high strain regime further understanding of the strain hardening behavior [14, 15]. The experimental observation/analysis of slip traces is also extensively used to study the transmission of slip across interfaces [16]. Most transmission models are based on geometric or orientation-driven relationships with incorporated mechanical loading models [17]. For example, the m' factor [18] successfully describes surface slip transmission in titanium alloys [19, 20]. These models have been validated using 2D experimental surface observations of slip transmission, which may be influenced by free surface effects. Slip transmission may significantly differ in the bulk and has not been investigated until now.

While there exist many experimental tools to observe surface or near sub-surface slip events, there are few methods for gathering this information in the bulk. Techniques where thin foils or lamellae are imaged in a electron transmission configuration, including the transmission electron microscope (TEM) [21] and transmission SEM (known as t-SEM or SEM STEM) [22] can probe slip events in a sample extracted from the interior of a sample. However, the sample lamellae are extremely small and thin (typically $10\mu\text{m} \times 10\mu\text{m} \times 100\text{nm}$), which prevents the statistical analysis of slip over large regions containing many grains with varying orientations, or during complex loading conditions.

Titanium is a versatile material due to the wide range of microstructural conditions that are available through processing [23]. Furthermore, the processing specific microstructural conditions result in material properties that can be tuned for specific component applications. Increasing the concentration of Al and interstitials such as O, which both are highly soluble in Ti, stabilize the hcp α phase. Deformation twinning is an important deformation mode in pure α Ti. However, while the α -phase is stabilized, twinning is also suppressed by increasing the amount of solute while planar slip is promoted, leading to dislocation pile-up at boundaries [24]. Ordered, coherent Ti_3Al (α_2) precipitates form when the Al content is increased above approximately 5% [25]. These α_2 precipitates are sheared by dislocations during deformation, which can lead to dislocation pile-up at boundaries and visible contrast by diffraction imaging, making Ti-7Al a good material for studying with X-rays. For instance, grain-scale plasticity in titanium has been monitored across a wide temperature range with the far-field X-ray techniques, providing accurate measures of slip system strengths, which are critical and difficult to obtain mechanical modeling parameters [26]. Furthermore, due to the large amount of α -stabilizers in Ti-7Al, the material contains no- β phase at room temperature [27] making it ideal for studying the mechanical properties of the α phase.

Over the past two decades, synchrotron X-ray techniques have been developed that enable the imaging of grains [28–33] and sub-grain structure [34–36], stress states within grains [37–40], and recently dislocations or slip events within grains using dark field X-ray microscopy and topotomography [41, 42]. The recent increase in spatial resolution and detector sensitivity due to upgrades to synchrotron light source flux (ESRF-EBS upgrade [43, 44]) and advances in detector technology (i.e. single photon counting pixel detectors and back-illuminated sCMOS cameras) provide the opportunity to directly image plastic deformation events throughout the interior of mm^3 -scaled samples. The combination of X-ray diffraction contrast tomography (DCT), topotomography and phase contrast tomography (PCT) has been particularly effective for imaging slip events in the bulk polycrystalline structure, in situ during deformation and then relating these events to the 3D grain structure [42]. However, the mechanisms giving rise to image contrast associated with slip events in the X-ray topographs are not well understood and the sensitivity of the technique for capturing all slip events was not demonstrated.

In this research, quantitative correlative measurements between HR-DIC, X-ray DCT and topotomography are performed on a Ti-7Al titanium alloy in order to investigate the sensitivity of the X-ray topotomography method to probe slip events both at the surface and in the bulk. The opportunities provided by the X-ray topotomography method to investigate plastic localization and transmission in metallic materials are demonstrated and discussed.

2. Methods

2.1. Ti-7Al Structure and Sample Preparation

Ti-7Al was selected as a model material in the present work due to its relatively large equiaxed grain structure and extensive study in the literature [27, 45–47]. This material was extruded and well-annealed to produce large

($\sim 100\ \mu\text{m}$), fully recrystallized grains with minimal intragranular substructure [45]. The Ti-7Al material contains α_2 -Ti₃Al precipitates that are roughly 2-3 nm in size within the α hexagonal close packed matrix phase. More details on the Ti-7Al alloy structure can be found elsewhere [47].

The specimen geometry used for correlative measurements is presented in Figure 1(a) and was prepared by wire electrical discharge machining to minimize residual stresses and plastic deformation from machining. The gauge section of the sample is rectangular with a cross sectional area of 0.6 mm x 0.6 mm. The flat sections of the gauge surface orthogonal to the loading direction were mechanically mirror polished using abrasive papers and diamond suspension, then chemo-mechanically polished using a suspension of 0.04 μm colloidal silica particles. Prior to deformation, a speckle pattern was obtained by chemical etching by immersion in Kroll's reagent for 20 s. This speckle patterning method allows for HR-DIC measurements, while not affecting the collection of EBSD measurements [4].

2.2. High Resolution Digital Image Correlation and microstructure measurements

Tensile tests were performed using a custom in situ ± 5000 N stage within a Thermo Fisher Scientific Versa3D microscope with a field-emission electron emitter on the flat dogbone-shaped specimens described in subsection 2.1. Tensile tests were interrupted at a macroscopic strain level near 0.8% (just past the 0.2% offset yield strength) and then HR-DIC measurements were performed. Macroscopic strain was measured in situ using both a strain gauge and fiducial markers located at both ends of the gauge length.

SEM image sets were acquired before loading and while under load following the guidelines of Kammers and Daly [48, 49] and Stinville et al [4]. A National InstrumentsTM scan controller and acquisition system (DAQ) was used to control electron beam scanning in the microscope. This custom beam scanner removes the SEM beam defects associated with some microscope scan generators [4, 50]. Tiles of 8×4 images before and after deformation with an image overlap of 15% were used. HR-DIC calculations were performed on these series of images and the results merged using a pixel resolution merging procedure, which is described in detail elsewhere [9]. HR-DIC measurements were performed on the entire gauge section surface, an area of about $2\ \text{mm} \times 600\ \mu\text{m}$. Typical subset size values of 31×31 pixels ($1044\ \text{nm} \times 1044\ \text{nm}$) with a step size of 3 pixels (101 nm) were used for the DIC measurements. Digital image correlation was performed using the Heaviside-DIC method [51, 52]. The sample preparation, imaging conditions and Heaviside-DIC parameters enable the detection of slip events at the surface of the specimen with a discontinuous displacement resolution between 0.2 and 0.3 pixels (7 nm and 10 nm respectively) [51, 52]. The strain map of the investigated specimen deformed at 0.27% plastic deformation is displayed in Figure 1(c and d) for the entire gauge length and for a reduced region of interest (ROI), respectively. The axial loading direction is vertically oriented in all strain maps. Bands of concentrated strain identify the locations of slip events at the surface of the specimen. It is worth noting that enhanced plasticity occurred near the edges of the gauge section of the specimen due to slight misalignment during mechanical loading.

Microstructure characterization on the sample gauge surface was performed by electron backscatter diffraction (EBSD) measurements with an EDAX OIM-Hikari XM4 EBSD detector using step size of 0.5 μm . Diffraction patterns were acquired using an accelerating voltage of 30 kV, a 4×4 binning mode and a beam current of 0.2 nA. EBSD maps were acquired before and after deformation. The data registration between the gauge surface EBSD and HR-DIC measurements is done by aligning control points using a polynomial distortion. About 50 pairs of control points were picked over the area of interest. The procedure is detailed elsewhere [53].

2.3. Diffraction Contrast Tomography

X-ray characterization was carried out at the ESRF materials science beam line ID11 in Grenoble, France. The beam energy was set to 38 keV with a relative bandwidth of 3×10^{-3} . The experimental setup includes a diffractometer designed for 3D diffraction experiments, including DCT and topotomography imaging modalities [54, 55] (see Section 2.4).

The Ti-7Al specimen was mounted vertically on the rotation stage. The microstructure in the gauge length was first characterized by two partially overlapping DCT scans, each composed of 3600 equally spaced projections over 360° . Images were recorded with an exposure of 0.25 s on a high-resolution detector composed of a scintillator placed 6.5 mm after the sample that is optically coupled by a 10 \times objective to a 2048×2048 pixel ESRF Frelon camera, giving an effective pixel size of 1.4 μm . The total scan time was 30 minutes.

Each scan was reconstructed using the DCT software [56] to produce grain maps of the bulk region subsurface below the HR-DIC measurements that also contained the topotomography measurements. The transmission of the

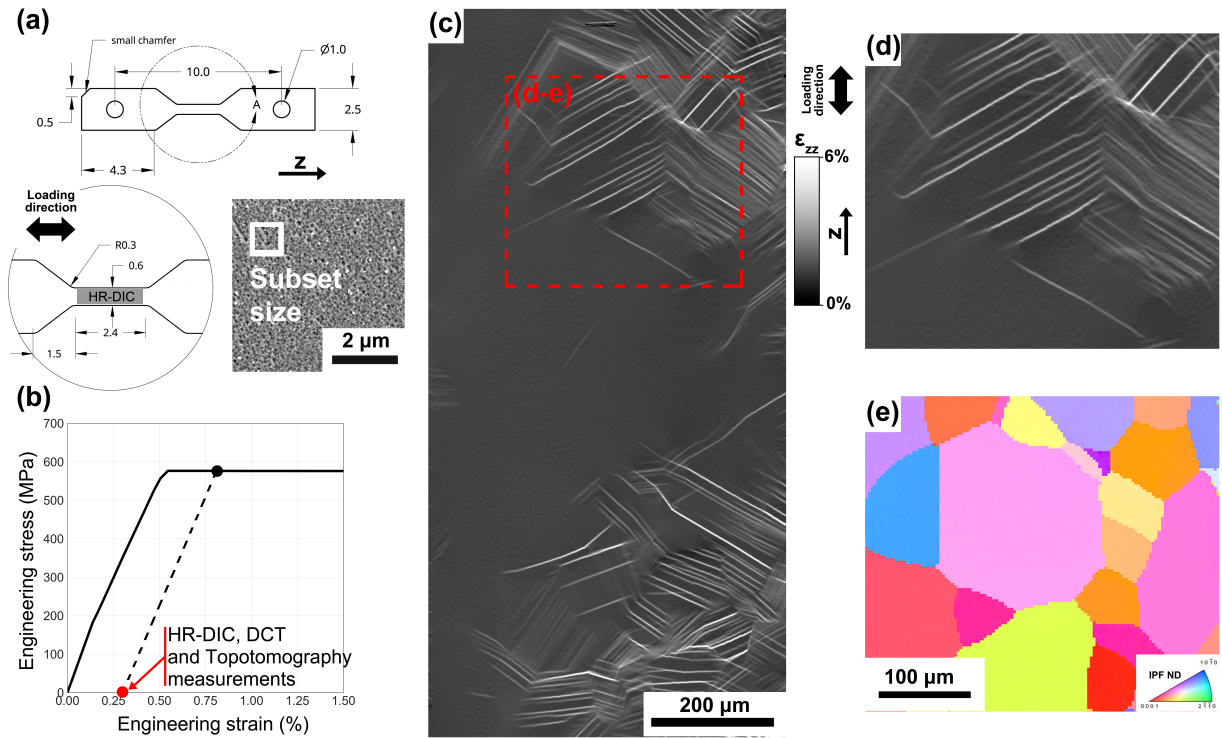


Figure 1: (a) Specimen geometry used for correlative in situ HR-DIC, DCT and topotomography measurements. SEM image of the speckle pattern and subset size (white box) used for HR-DIC. (b) Engineering stress-strain curve of the Ti-7Al material. (c) HR-DIC ϵ_{zz} strain map obtained at the surface of the specimen after about 0.8% macroscopic deformation. (d-e) Reduced region of the ϵ_{zz} strain map with its associated grain structure obtained by electron backscatter diffraction measurement.

direct beam was used to reconstruct the exact shape of the illuminated region using a classical filtered back projection. The two DCT grain volumes were fused together into one dataset using Pymicro [57] (including absorption data to produce a mask for the specimen geometry) after matching grains in the overlapping region and minimizing the correlation error between the two DCT volumes present in the overlapped region. Isotropic dilation was also carried out, but limited to the grains within the absorption mask, in order to fill all remaining void space within the DCT reconstruction. This resulted in a high fidelity grain map of the entire ROI of the sample ($0.6\text{ mm} \times 0.6\text{ mm} \times 1.1\text{ mm}$), containing 1055 grains. The average grain size in the reconstructed DCT volume, including surface grains, is $68\text{ }\mu\text{m}$. As a side note, besides the front face that was mirror polished for EBSD analysis, the other 3 lateral faces were left as received from the EDM machining. This resulted in a $10\text{ }\mu\text{m}$ thick layer of material with high defect density and low diffracted beam intensity, and therefore is absent from the final reconstruction.

2.4. Topotomography setup

The reconstructed DCT grain map from Section 2.3 was processed to select grains for topotomography inspection. The EBSD data from the exterior of the sample was matched to the identical surface of the reconstructed DCT grain map. Two grains from different regions of the sample surface with slip traces identified by HR-DIC were selected for topotomography, these grains are labeled 63 and 567 in Figure 11. More grains were selected based on the following criteria: (i) the grain is present in the neighborhood of the two selected surface grains (ii) at least one low index reflection can be aligned within the range of the sample goniometer. An automated script searched through all the indexed grains and selected 55 different grains for topotomography acquisition. Although it did not happen many times due to the limited travel range of the goniometer, for some of these grains, several reflections (up to 3) could be reached and were also collected. In total, 75 topotomography scans were collected for two grain clusters in the gauge length of the sample.

For topotomographic scan acquisition a particular scattering vector of a given grain is aligned parallel to the tomographic rotation axis [34]. The tomographic rotation axis itself is inclined by the Bragg angle Θ (diffractometer base tilt) in order to maintain Bragg condition throughout the entire 360° rotation of the sample. To account for lattice rotations and dispersion effects within the grain, the base tilt is scanned over the width of the crystal reflection curve using an alternating, continuous scan motion with a step size of 0.05° . A second camera with a higher magnification was used to record these projection topographs with a pixel size of $0.7\text{ }\mu\text{m}$. In the current experiment, topotomography scans were comprised of 90 equispaced series of projection topographs (one every 4° of sample rotation) and require between 10 minutes to a couple of hours to collect, depending on the level of mosaicity (the Θ integration range must be increased for more deformed grains, a typical value was used for this sample with a range of 1.6° and 64 images). The topotomography experimental setup is presented in the Supplementary Material, including a high-speed video ($20\times$ speed) of the sample goniometer stages moving during data collection. A snapshot of the video is provided in Figure 2 to describe the experimental setup. An example of a set of topographs is provided in Supplementary Material for a grain that is subject to plastic deformation.

X-ray topographs contain orientation contrast from the diffraction of a crystal. Defects present in the volume locally affect the Bragg condition and will give rise to contrasts evolving as a function of the propagation distance. To select the best detector distance to acquire topotomography scans, grain 567 was first aligned using reflection $(1\ 1\ \bar{2}\ 0)$ and integrated projection topographs were recorded while increasing the sample detector distance from 7 mm to 41 mm in steps of 2 mm. The contrast related to slip events evolves from low contrast intensity at short propagation distance to high contrast intensity at large propagation distance, whereas the outline of the grain evolves from a nearly geometric projection into a distorted projection with ill defined boundaries at large distances, as shown in Figure 3. For the rest of the experiment the detector distance was kept at 12 mm, allowing enough space to rotate the tension specimen freely without risking a collision between the goniometer stage and the detector. Note that in situ testing (not reported here) requires an optimized design of the load frame [42] and of the detector head in order to maintain such short propagation distances. Also note that for future experiments, acquisitions with different propagation distances could be used to provide additional constraints for improving the convergence of the iterative optimization algorithms used for reconstructing the local orientation field inside a grain [58].

2.5. Detecting slip system activation

The 75 topotomography scans were each post-processed to detect slip activity. The grain shape, position, and the 4 goniometer rotation angles (the base tilt Θ , the tomography rotation ω and the two eucentric tilt motors to

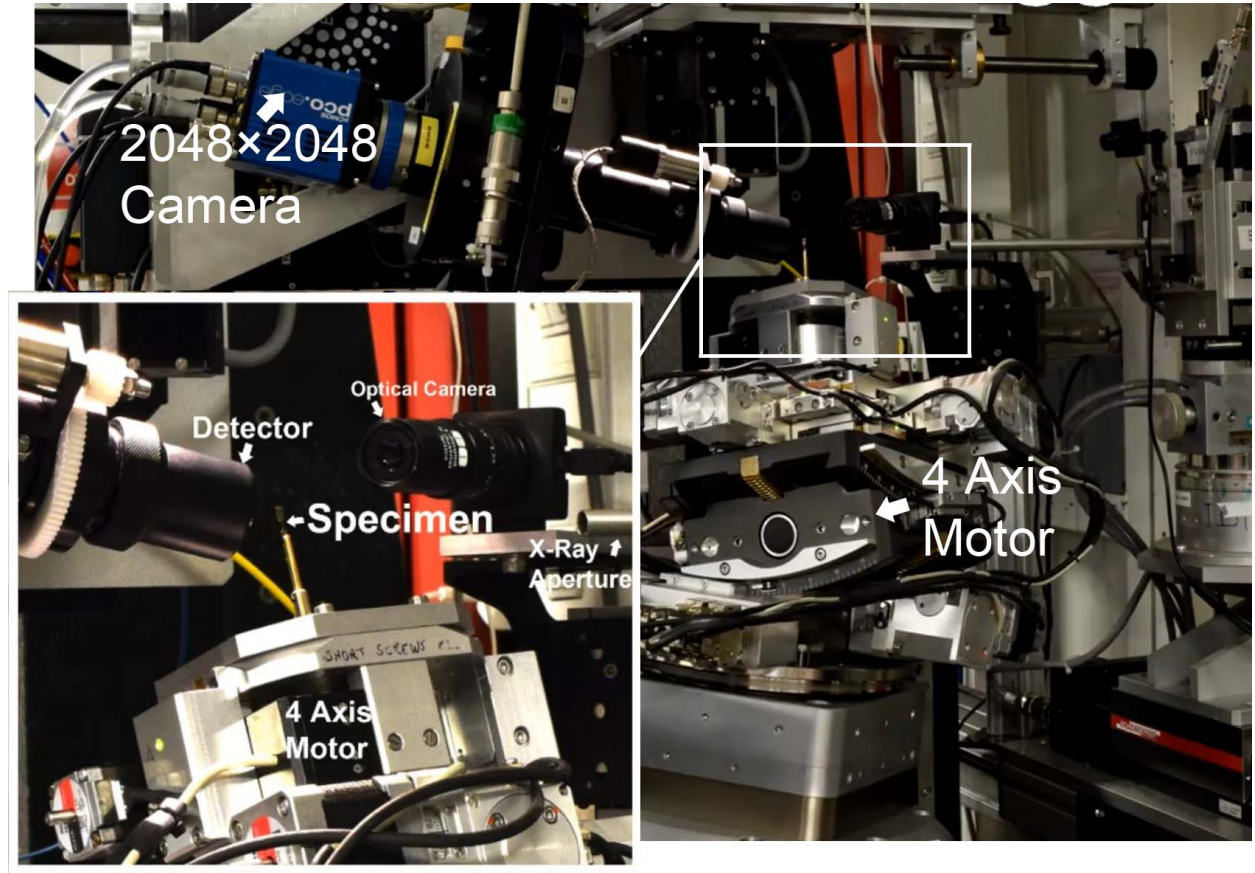


Figure 2: Experimental setup for topotomography measurement. The complete setup is rotated by the base tilt Θ while maintaining Bragg condition visibility for a specific grain throughout the entire 360° rotation around the tomographic rotation axis. A high resolution detector with a pixel size of $0.7\ \mu\text{m}$ is used to record series of integrated projection topographs every 4° . The reader is invited to consult the Supplemental Material about the experimental setup of the topography measurement.

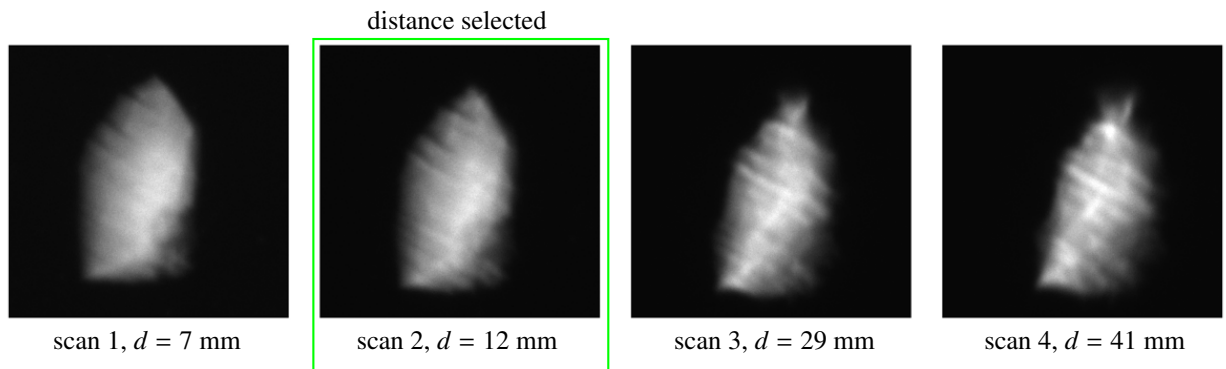


Figure 3: Topographs of grain 567 extracted from the topotomography scans ($\omega=44^\circ$) taken at different detector distances. The second distance of 12 mm was selected for our observations. A movie showing the evolution of contrast as a function of the detector distance is available as Supplemental Material.

align the scattering vector with the axis of ω , refer to [34, 42] for all the details) were used to produce a simple forward simulation (parallel projection of the 3D grain volume) in the topotomography configuration. Using the grain orientation, slip planes are added in the volume of the grain by manually selecting the best match with the orientation contrast visible in the topographs over the 360° . If necessary, the active slip system was then identified using the highest Schmid factor within the plane. This is a tedious but efficient way of identifying slip planes that have produced plasticity in all the grains [42]. Figure 4 depicts two topographs close to edge-on configuration (i.e. diffracted beam co-planar to slip plane) with their slip system identified and instantiated in the simulated projection images on the right figure panels. This process was repeated for the whole set of 75 topotomography scans comprising 55 grains.

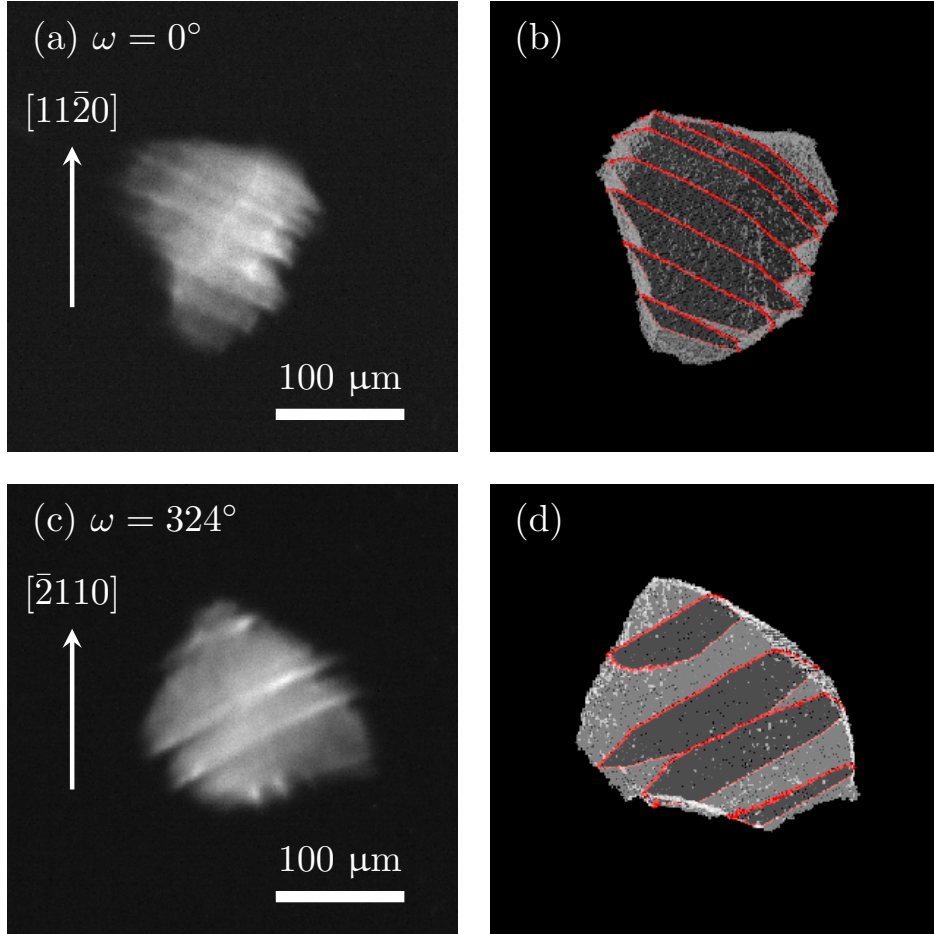


Figure 4: Examples of slip planes identified by comparison between the recorded topographs (a and c for grains 572 and 579 respectively) and the forward modeling of the diffraction for the corresponding grains (b and d), the slip plane intersections with the grain boundary are marked in color. The white arrows indicate the topotomography scattering vector used to image each grain, a more detailed description of this is in Section 2.4.

2.6. Orientation spread plots from topotomography

Whereas the projection topographs shown in Figure 4 are integrated over angular range covered by the base tilt Θ , the individual images taken during this motion (termed a rocking scan) constitute a rocking curve for each particular angle ω . The change observed in the Bragg condition here is largely dominated by the lattice rotation (compared to elastic lattice distortions) and can be directly attributed to the orientation spread around the base tilt axis. To quantify

this, the full width of the rocking curve (simply integrating the intensity collected by the detector at this angle) at 10% of the peak is measured for each position of ω (every 4°) [42]. Compared to the full width at half maximum (FWHM), this was preferred as the shape of the peaks can be quite far from Gaussian (with long tails) and because of that, FWHM may underestimate the lattice rotation magnitude for the low plastic strain investigated here. To compute this value reliably, a mask for the grain being imaged needs to be created, because grains with similar Bragg conditions can diffract and be captured on the detector during the rotation. This phenomenon happens frequently enough to prevent a simple sum of the intensity on the detector, even with a proper ROI applied. To work around this problem, a forward simulation routine of the diffraction imaging was developed with Pymicro [57] to create a mask of the grain projected onto the detector using the DCT reconstruction and the value of all motor positions (see Figure 5a). The mask was dilated by 10 pixels to allow for small grain deformations (and hence image distortions), which are not accounted for in the 3D DCT grain reconstruction used as input to create the mask. This processing combines the two rotations Θ and ω to analyse the lattice rotation distribution in a given grain. For a given value of ω , the evolution of the intensity as a function of the base tilt angle $I_\omega(\Theta)$ is analyzed and the width of the rocking curve is extracted. Then this width is plotted as a function of ω , as illustrated in Figure 5a.

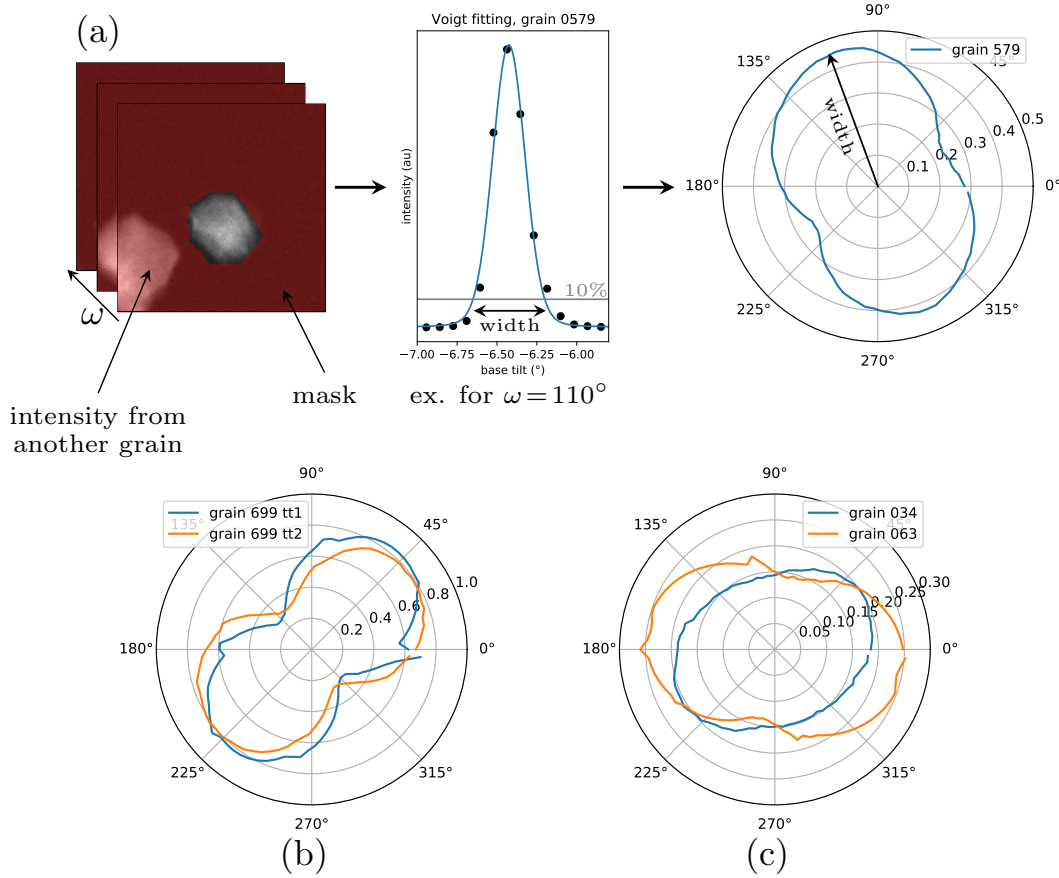


Figure 5: Orientation spread plots showing some examples of the measured width of the rocking curve plotted as a function of ω . (a) For each ω angle the individual topographs are masked and the resulting integrated intensity $I_\omega(\Theta)$ is fitted to extract the width at 10% of the peak; the evolution of this value with ω depicts a typical dumbbell shaped curve here shown for grain 579 located in the bulk. Both the maximum width of this curve and its orientation with respect to ω are linked to the lattice curvature distribution in this particular grain. (b) Comparison of the effective misorientation measured on the same grain with two different reflections. (c) Effective misorientation plots for surface grains 34 and 63. Note that small spikes can sometimes appear on the curves when spurious diffracting grains are located within the actual grain mask.

A representative set of orientation spread curves are shown in Figure 5. They almost all depict a characteristic dumbbell shape as already observed in a previous topotomography experiment [42]. This shape is the signature of a crystal bent perpendicularly to the elongation direction (in ω) and can be related to dislocation activity and the presence of geometrically necessary dislocations (GNDs) introducing lattice rotations in the grains [59]. This is consistent with a single slip dominated deformation state as observed in most of the grains. An in depth look at the data processing shows that most of the $I_{\omega}(\Theta)$ rocking curves display regular single peak shapes which are the signature of rather homogeneous rotation field while some other display double peak shapes characteristic of more heterogeneous distribution of the lattice rotation or even formation of a subgrain (although this specimen was scanned ex situ so the initial state in the bulk is unknown). In this experiment some of the grains have been measured in topotomography for the first time using several reflections. For instance Figure 5b shows the orientation spread plots for grain 699 obtained with reflections $(10\bar{1}1)$ and $(11\bar{2}2)$ and one can see that the results are in very good agreement. Using several reflections will be useful in the future when extracting the orientation field from the measurements [58] (not done in this paper).

3. Results

The combination of X-ray diffraction contrast tomography and topotomography allow for the spatially resolved imaging of slip events within grains and with regard to the full 3D grain structure, as shown in Figure 6. Reconstruction of the entire specimen gauge section by DCT is shown in Figure 6(a). When compared to the EBSD map, the DCT reconstruction appears to be very accurate, including the shape of the surface grains. A detailed analysis showed that all of the 143 surface grains captured by DCT are also mapped at the surface by EBSD, except for 6 very small and presumable shallow grains. On the other hand, the grains located close to the edges of the lateral faces that are difficult to capture in EBSD (slightly rounded specimen shape due to the polishing) are correctly reconstructed by DCT. The grains investigated by topotomography are displayed in Figure 6(b). Two regions of surface grains were selected from HR-DIC measurements, with many grains exhibiting slip events after tensile deformation. The grains selected for topotomography measurements from within the bulk were chosen to be in contact with the selected surface grains and to have at least one low index lattice plane within reach by the diffractometer tilt motions (see Section 2.4). Analysis of the topographs and the spatial correlation of DCT and topotomography measurements allows slip event detection and the determination of their location in the 3D grain structure, as shown in Figure 6(c), for all the grains investigated by topotomography.

3.1. Comparison between HR-DIC and Topotomography measurements

Most of the grains investigated by topotomography, at both the surface and in the bulk, show the same slip events. Further, surface grains examined by HR-DIC exhibit the same slip events as the topotomography grains. In addition, most of these grains appear to be in a single slip condition, where only one slip system is activated.

The HR-DIC strain measurements from the surface grains were analyzed completely independently from the topotomography measurements of the same grains, which contain full 3D grain information, in order to minimize potential bias and to validate the datasets with each other. Among the 55 investigated grains, 19 are surface grains. The HR-DIC and X-ray measurements are compared for all of these surface grains - containing 109 HR-DIC measured slip events. Specifically, the location and average in-plane slip amplitude of individual slip events (HR-DIC measurements) were extracted. An example is given in Figure 7(a and b) for a single grain with 7 intense surface slip events measured by HR-DIC, while only 5 slip events are detected by topotomography. The slip locations detected by topotomography in all the surface grains correlate accurately with slip events detected by HR-DIC measurement. For all the investigated surface grains, 67 slip events were detected by both techniques, while 41 were not detected by topotomography. The distributions of these detected and undetected events by topotomography are reported in Figure 7(c) as a function of their average in-plane slip amplitude, as obtained from the HR-DIC measurements. It must be noted that slip events with an in-plane slip amplitude below 20 nm can not be differentiated using HR-DIC due to the resolution limit of measuring in-plane slip amplitude with this technique. The slip events that were undetected by topotomography, but detected by HR-DIC, are mainly low intensity slip events of in-plane slip amplitude below 50 nm (or roughly lower than 170 emitted dislocations [10]). Further inspection indicates that the most intense undetected slip events by topotomography are located near the edges of the grains as displayed at the red arrows in the grain in Figure 7(a

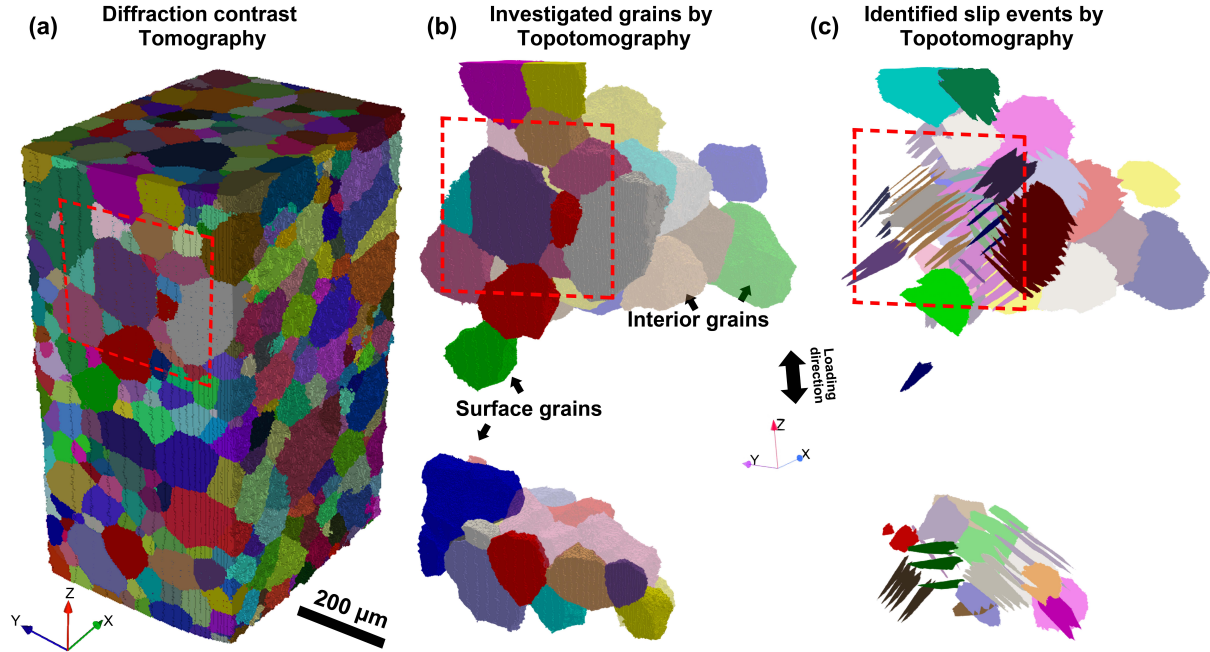


Figure 6: (a) DCT of the Ti-7Al specimen (random colors are used to differentiate grains), the front face corresponds to the free surface where the HR-DIC measurement was performed. (b) DCT of the grain investigated by topotomography. (c) Identification of surface and bulk slip events by topotomography. The red dashed box indicates the region highlighted in Figure 1(d-e).

and b). The detection of slip events at grain boundaries is not favored using topotomography for two reasons: first, the diffracting grain volume at the periphery of a grain may be small and tend to have a lower diffraction signal to noise ratio. Second, the edges of the grains are more likely to accumulate high lattice rotations during deformation, which could obscure the contrast resulting from well defined slip plane localization and prevents their detection on the topographs.

3.2. Contrast at slip events from topotomography measurement

While HR-DIC measurements provide the location and quantitative measurement of the slip events (in-plane slip amplitude given in nm), topotomography measurements provide the slip location and an associated contrast on the topographs with contributions from the entire volume of the grain. The contrast detected in the topographs associated with slip events in the HR-DIC measured surface grains were classified into 4 groups: no contrast (no slip event detected by topotomography measurement but a slip event detected by HR-DIC), low contrast (slip event detected by topotomography with weak contrast on the topographs), medium contrast (slip events detected by topotomography with average contrast on the topographs), high contrast (slip events detected by topotomography with intense contrast on the topographs). The groups of contrast have been defined through the difference between the average gray values across the entire grains and the high or low gray values within the bands. High and low contrast refer to difference values that are superior to 100 and inferior to 30 on the 255 gray scale, respectively. The distributions of the in-plane amplitude as measured from HR-DIC measurements of slip events for these 4 groups are displayed in Figure 8.

As a general trend, the visibility of the topographic image contrast associated with a slip event is related to the in-plane slip amplitude of the slip event. High in-plane slip amplitude slip events tend to have high visibility in the topographs. Conversely, low intensity slip events tend to display low contrast in the topographs.

3.3. Comparison between EBSD and Topotomography measurements

Measurement of the lattice rotation magnitude induced during deformation has been performed for surface grains by conventional EBSD measurements and topotomography using grain orientation spread (GOS). The spread is defined as the average deviation between the orientation of each point in the grain and the average orientation of the

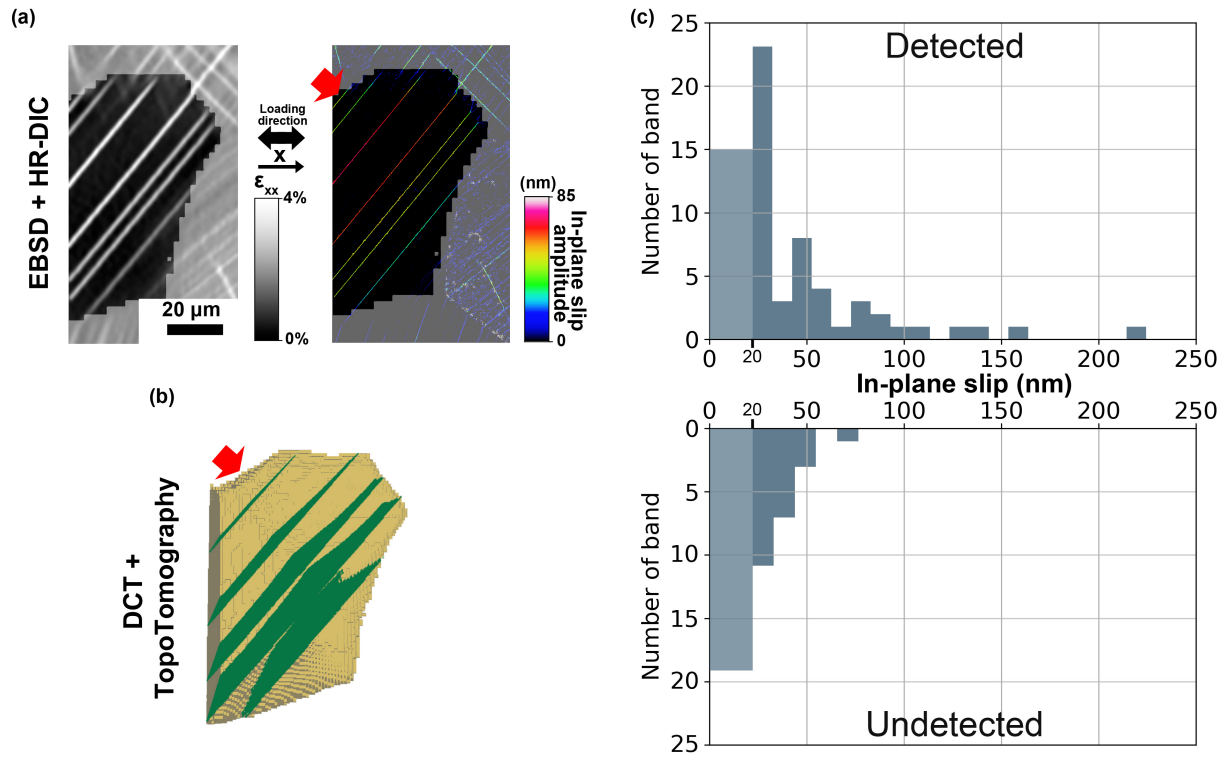


Figure 7: Comparisons between HR-DIC and topotomography measurements. (a) Strain map after deformation of a Ti-7Al specimen at about 0.8% and the associated in-plane slip amplitude map for a grain of interest. The in-plane slip amplitude corresponds to the physical in-plane displacement produced by a slip event onto the specimen free surface. (b) Recorded topographs and reconstructed grain shape and slip events for the grain of interest. (c) Distribution of slip events that are detected (top) and undetected (bottom) by topotomography measurements. The strain maps are used as reference to determine the location and number of slip events. The red arrows indicate two slip events at the edge of the grain of interest that are not detected by topotomography measurements.

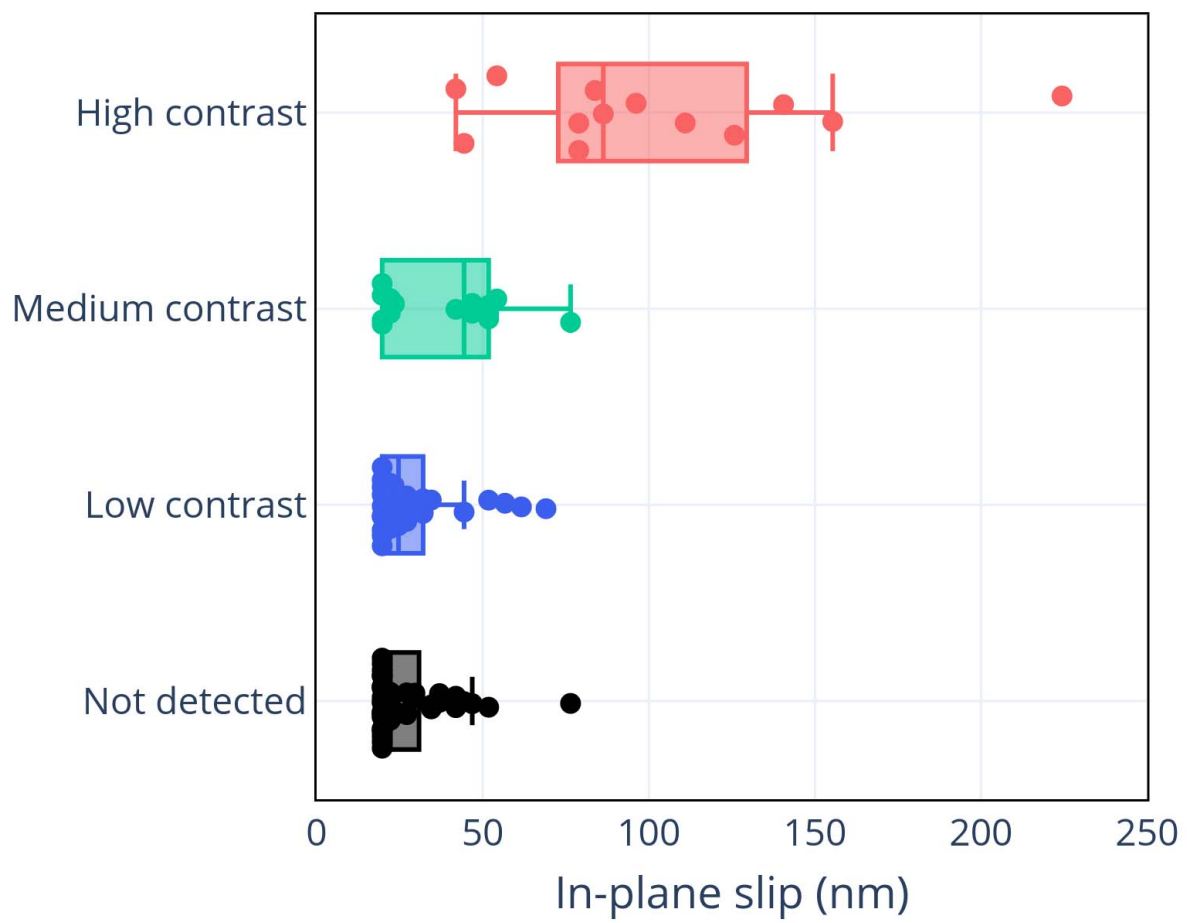


Figure 8: Contrast level of slip bands measurements by topotomography as a function of the HR-DIC surface measured in-plane slip intensity of the identical bands. Only slip events at the surface, or within the grains at the surface, are considered in the present analysis. The HR-DIC strain maps are used as a reference to determine the location and number of slip events. The slip bands that are not detected by topotomography measurements, but are present in the HR-DIC strain maps are labeled as “Not detected”.

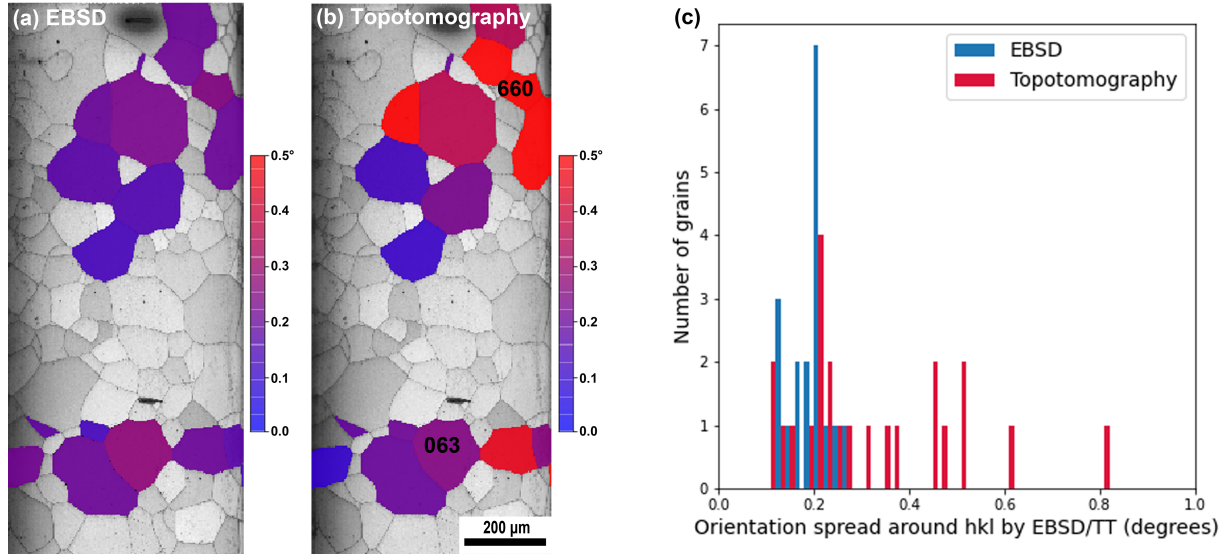


Figure 9: Grain orientation spread (GOS) around the scattering vector for the investigated surface grains from (a) conventional 2D EBSD measurements and (b) full 3D topotomography measurements. The distribution of the GOS is displayed in (c).

grain. The conventional EBSD measurements provide information from the surface and near-surface of the grains, while topotomography measurements contain a measure of GOS (projections of the grain orientation distribution along the selected scattering vector) for the entire volume of the grain. Note that lattice rotations around the scattering vector are not captured by single rocking curve analysis presented in the current work. For this reason, the orientation spread by EBSD is also computed with respect to the topotomography scattering vector of each grain. This hides the rotation part around the scattering vector that is not visible with topotomography and allow a more rigorous comparison (although the effect is small). In some grains, the orientation spread measured by topotomography was compared with the spread computed from the orientation field reconstructed by 6D-DCT (not reported here); the values were in very good agreement, suggesting the orientation spread from topotomography is an accurate descriptor of the grain wise amount of lattice rotation.

A GOS map (around the scattering vector) is shown in Figure 9(a) from EBSD alongside measurements of the orientation spread made by topotomography. Figure 9(b) was constructed by overlaying the map of the orientation spread measured by topotomography described in Section 2.6 onto the EBSD image quality signal - to provide the grain boundary locations. We note that this measure of orientation spread by topotomography captures the orientation deviations from within the entire 3D grain structure. The distribution of the orientation spread obtained from both techniques is compared in Figure 9(c).

The analysis from EBSD and topotomography of the orientation spread in surface grains shows overall similar values, indicating that the measurement of orientation spread by topotomography is sensitive enough to capture the evolution of lattice rotations during plastic deformation. However, the grain-to-grain GOS variation in EBSD (surface measurement) compared to topotomography (entire 3D grain) is (systematically smaller), indicating the significance of the bulk measurement information, and the amount of information lost by only capturing a cross section of the grain orientations at the surface by EBSD. The observed differences can be attributed partly to the lower angular resolution of EBSD, but more likely due to the nature of bulk versus surface measurements. Furthermore, this difference appears to vary as a function of the orientation of the active slip direction with respect to the sample surface, this is discussed in more detail in Section 4.3.

4. Discussion

Plastic deformation does not occur uniformly during the loading of polycrystalline metallic materials, but instead it occurs initially by localized slip in regions where dislocations first overcome obstacles to deformation. This manifests

at the surface of the specimen by the formation of surface steps [1]. With continued straining, numerous occurrences of these slip events result in a continuous network of plastic flow across the entire cross section of the specimen, corresponding to macroscopic yielding of the specimen [60].

Plasticity and its transmission across grains has been extensively investigated at the surface of the specimen during loading. However, very little experimental data is available on the localization and transmission of plasticity in the material bulk as a function of the 3D grain structure. For example, it has recently been observed that the 3D grain structure is especially critical to understanding the origin of incipient localization in nickel-based superalloys [61]. The combination of DCT and topotomography [34, 42] provides the unique capability to detect slip events as a function of the grain structure. This technique has tremendous opportunity for investigation of the bulk slip localization, however the sensitivity of the technique with regard to slip detection has not yet been quantified and parameters other than the slip amplitude may influence the visibility of slip localization in topographs.

4.1. X-ray Topotomography slip event detection and slip event contrast

Compared to a previous study in an AlLi alloy [42], the level of deformation is higher in the presently investigated Ti-7Al sample (0.8% vs 0.3%), as seen in the integrated topographs in Figure 4 and Figure 10(a). Due to the higher level of deformation, the grain projections formed on the topographs are more deformed - making it more challenging to identify the slip planes. Nevertheless, the vast majority of the grains depict strong contrast localization in the form of bands, just as in the AlLi experiment [42]. Using the grain orientations from DCT, these bands can be correlated with active slip planes in the bulk of the grain. Another difference in the Ti-7Al material is that the orientation contrast localized around slip bands remains visible on a much larger angular range than in the previous AlLi study (typically 50° vs 10°). In summary, using this slip system identification technique, it appears that we approach the upper limit (i.e. 1 or 2% plastic strain) for the topotomography technique. At increased levels of plasticity, the projection contrast becomes too convoluted to reliably identify the slip planes. This limitation restricts the use of topotomography for investigation of the slip activity at relatively low levels of plastic deformation. However, we note that for fatigue properties - typically much of the activity is dictated by the first few cycles [10] at relatively low strain - potentially making topotomography an extremely useful technique.

The results in Section 3.2 demonstrates that slip events with low intensity as detected by HR-DIC measurements were not all detected by topotomography, indicating that low intensity slip events do not produce enough contrast to be integrated on topotomography images. Note that in this paper we have used integrated images to detect slip events while one could also use individual images (typically 64 per topograph here) to look at contrast variations. Individual image analysis is hardly tractable with a manual approach as pursued here, but could potentially increase sensitivity if it was automated. Topographic image contrast can be roughly divided into two categories: (i) contrasts related to dynamical diffraction and extinction effects and (ii) orientation contrast linked to local variations of the effective misorientation (i.e. the combined effect of rotation and elastic distortions of the crystal lattice) [62]. Whereas the former can be predominant in nearly perfect single crystals, the latter are more relevant in the case of metal grains, typically containing a high density of dislocations. Topographic image contrast of slip localization events can be attributed to local gradients in lattice rotation induced by high dislocation densities generated by the avalanche, pile-up, and dislocation accumulation processes associated with slip events. These processes are well investigated in the literature [63–66]. The interaction between dislocation pile-ups, dislocation accumulation, and grain boundaries gives rise to heterogeneous stress distributions. Such stress heterogeneity leads to very local gradients in effective misorientation of the crystal lattice [64, 65, 67] that are captured by topotomography measurements. Other effects, like local reduction of the scattered intensity due to the disorder in the crystal lattice in proximity of slip localizations are likely to contribute to the intensity variations observed in the topographs.

To further investigate the characteristics of the contrast at slip events in the topograph, a specific grain in the specimen bulk with significant topograph contrast intensity is explored in Figure 10. The topograph from the tilt series with the highest contrast intensity is presented in Figure 10(a). Four slip events within the investigated grain are observed and depicted by the purple planes in the reconstruction of the grain (blue colored) and slip events in Figure 10(b). The contrast intensity varies along the plane of the slip events and is especially high on the left and right edges, as indicated by green arrows. Interestingly, intense contrast is also observed at the bottom edge of the grain at the red arrows in Figure 10(a), which correspond to the impingement of two slip bands that developed in the neighboring grain (yellow grain in Figure 10(b)).

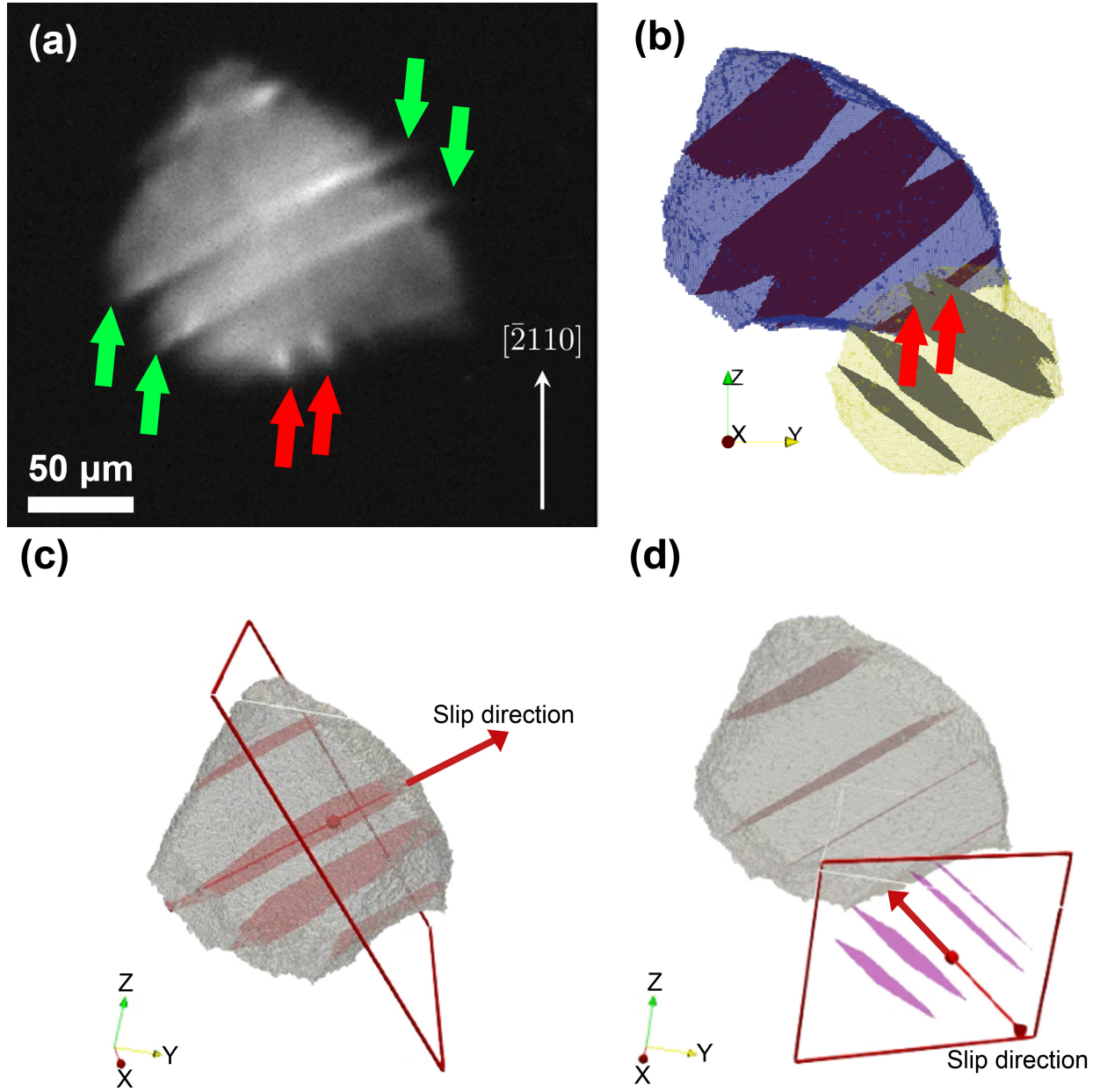


Figure 10: (a) Topograph of an investigated grain with intense contrast at slip events (green arrows) and impingement of slip events at the boundary from a neighboring grain (red arrows). (b) 3D DCT reconstruction of the investigated grain (blue) and a neighboring grain (yellow) and the overlaid slip event planes, as detected by topotomography. (c-d) Slip directions of the active basal slip systems for the investigated grain and neighboring grain.

Both slip systems in the investigated grain (blue) and neighboring grain (yellow) were identified as basal type. The slip direction for the active basal slip systems in the detected slip planes for both grains are depicted with red arrows in Figure 10(c and d). The high contrast intensity occurs near the grain edges (green arrows in Figure 10(a)) in the topograph where the slip direction is orthogonal to the imaging plane. In other words, the intensity of the contrast is maximized when the slip direction is in the plane of the page, as approximated by the red arrow in Figure 10(c). Furthermore, the two intense points of contrast at the red arrows in Figure 10(a), which are due to impingement of the intense slip bands from the neighboring grain (yellow grain in Figure 10(b)), occur at the location where the active basal slip system direction is pointing towards the grain boundary. Identical types of observations were made at several instances of grains and slip events in the topotomography Ti-7Al dataset. There is also an interplay with grain volume here that can affect the contrast. When a grain is large and the slip plane is tilted increasingly away from parallel to the viewing direction, the contrast at the front and back faces can decrease as they will overlap and interfere in the projection through the grain. This effect is lessened in the faces of the grain that are not parallel with the viewing direction, and so contrast from slip is increased in these regions.

From the previous example, it is observed that some of the high contrast regions in topographs that reveal the presence of slip events originate from the accumulation of dislocations at grain boundaries. These accumulations of dislocations either induce significant contrast in the grain where the slip events develop or in the neighboring grain where slip band impingement occurs. The lattice rotation contrast induced by dislocation accumulation have previously been investigated by HR-EBSD measurements in titanium alloys [64], and their amplitudes are on the order of the magnitude that corresponds to the contrast observed on topotomograph. It is therefore reasonable now to consider that the contrast observed in the topographs, which provides the identification of slip events, may be enhanced in the proximity of grain boundaries due to dislocation accumulation. On the other hand, the contrast also appears to come from the collective presence of dislocations and their stress fields moving within the slip band and providing edge-on contrast. This consideration can explain the propensity for the topotomography technique to tend to not detect low intensity slip events, which are associated with low dislocation densities and where the contrast created by the band within the grain is too faint. An in situ experiment would be helpful to analyse the evolution of contrast when a slip band, dislocation pile-ups and dislocation accumulation increase in intensity as the deformation progresses.

Based on the observation of the many topographs in this work, it appears that the contrast for prismatic slip bands is generally higher than for basal slip. There is a similar observation in HR-DIC experiments where basal slip tends to be distributed on many slip bands with fewer dislocations per band, resulting in lower surface discontinuities. A similar mechanism is presumably taking place here, although this was not studied in detail. Also, pyramidal slip was not observed in this experiment so no conclusions can be made on this point.

4.2. Slip transmission in the bulk

One of the interesting applications of topotomography is the investigation of slip transmission. Typically, transmission analyses are performed from surface measurements that fail to experimentally capture the inclination of the slip events and geometry of the grain boundaries in the sub-surface/bulk. While slip transmission in the bulk of the specimen will be more rigorously studied in a future study of Ti-7Al, demonstrative examples are presented in Figure 11 from regions measured by HR-DIC and topotomography, where slip events (blue arrows) are transmitting in the neighboring grains (red arrows). The incoming and transmitted slip events were detected in the topotomography measurements and their 3D representation is provided in Figure 11(c and d) for the first example and in Figure 11(g and h) for the second example. Incoming slip events (blue arrows) in both of the investigated grains were identified as the slip system with the highest Schmid factor, and the active slip directions are depicted by the long red arrows in Figure 11(d and h). The associated HR-DIC and EBSD maps are provided for both examples in Figure 11(a,b and e,f).

The first example shown in Figure 11(a-d) displays a direct transmission event. Upon consideration of the Schmid factor for grain 615, slip system $(\bar{1}010)[-1\ 2\ -1\ 0]$ has the highest Schmid factor of 0.405 and is not activated, while the slip system $(\bar{1}100)[1\ 1\ -2\ 0]$ is activated (purple planes in Figure 11(c-d)) with a lower Schmid factor of 0.373. Consequently, the transmission event triggered a slip system in grain 615 with a lower Schmid factor but with favored transmission configuration. The m' factor [18] is usually an effective indicator to describe the propensity of a incoming slip event to transmit to the neighboring grain. It is defined as $m' = (d_1 \cdot d_2)(n_1 \cdot n_2)$, with d_1 the slip direction of the incoming slip system, d_2 the slip direction of the outgoing slip system, n_1 the slip plane normal of

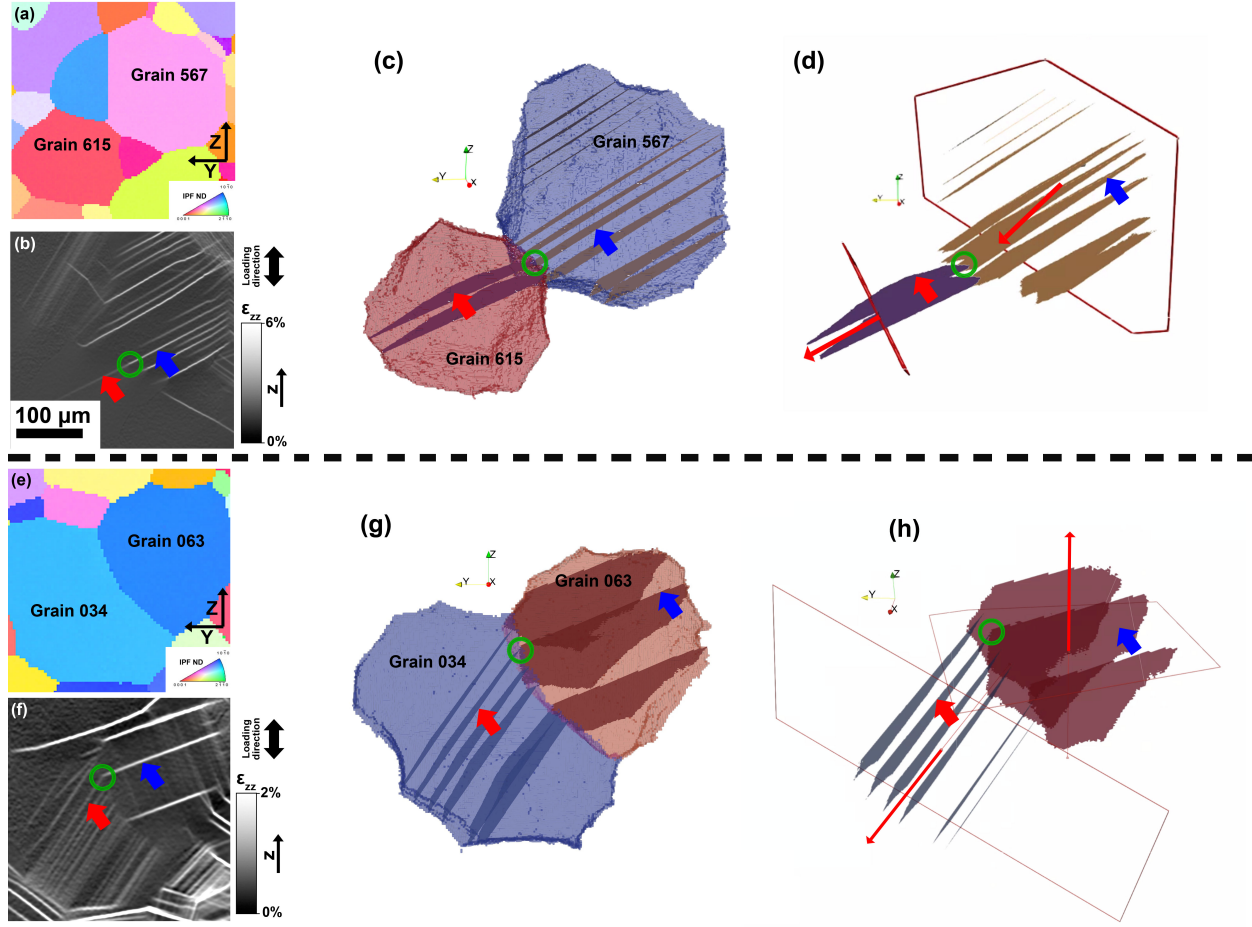


Figure 11: Two transmission events observed by DCT and topotomography. (a,e) EBSD map of the surrounding grain structure around the transmission events. (b,f) HR-DIC ϵ_{zz} strain map showing the investigated slip transmission events. (c,g) 3D reconstruction of the grain structure and slip events for the grains of interest. The incoming and transmitted slip are indicated by blue and red arrows, respectively. The green circles highlight the location of slip transmission. (d,h) 3D reconstruction of the active slip planes detected by topotomography for the grains of interest.

the incoming slip system, and n_2 the slip plane normal of the outgoing slip system. m' ranges between 0 and 1 and higher values indicate more favorable geometric compatibility between both slip systems that favor slip transmission. The m' factor between the incoming and transmitted slip event is 0.7 while the one between the incoming slip and the highest Schmid factor slip system in grain 615 is 0.1. This is evidence of plasticity behavior that is dependent on the local surrounding microstructural configuration as has previously been reported for titanium alloys [19, 20]. The topotomography measurements provide accurate spatial resolution to capture such behavior at the surface as conventional surface measurements (trace analysis from SEM, HR-DIC or AFM), but also into the specimen bulk.

The topotomography measurements allow differentiation between slip activity that occurs at the surface and in the bulk. The m' factors were calculated for all slip events that are connected (at least one voxel in common at a grain boundary) across surface grains and from bulk grains. The distributions of the m' factor is reported in Figure 12. It is striking to notice such a difference in the distribution from bulk and surface grains. Surface grains tend to promote slip transmission and as a consequence, the m' factor is important when investigating slip activity at the surface, as reported in the literature [20]. Conversely, the distribution of the m' factor between bulk grains indicates that it may not be as relevant for the description of slip activity in the bulk. Direct transmission may not be of significant importance within the bulk, in agreement with the recent observation of the importance of the triple junction within the bulk [61]. In this work, it was observed in a deformed metallic material that a large amount of the slip events that develop originate from triple junction lines that have significant stresses. From the present analysis, the possibility should be considered that slip activity within the bulk is mainly controlled by stress heterogeneity at junction lines, with slip transmission having a lesser influence.

Interestingly, multiple peculiar cases of direct transmission were observed at the surface of the specimen as displayed by the second example in Figure 11. In this example, the incoming slip event and the transmitted slip event are connected by the unique surface point highlighted by green open dots in Figure 11(g and h). The transmission factor m' is calculated as 0.1 indicating a configuration that does not favor slip transmission since both the plane normal and slip directions of the active slip systems are highly disoriented from each other, as displayed in Figure 11(d). From the 3D observation of this slip transmission event, and considering that the slip direction of the incoming slip event is almost normal to the surface, it can be hypothesized that the intense incoming slip event developed a relatively significant surface step that triggered slip activation in the neighboring grain. The constraint of the slip extrusion at a grain boundary can generate large local stresses that then can trigger slip in the neighboring grain. This kind of transmission event was observed in several instances in the investigated surface grains and contributed to the low values of the m' factor in the distribution in Figure 12(a). It is generally accepted that direct or indirect slip transmission is controlled by dislocation transmission across grain boundaries [18]. However, the present example evidences another type of transmission related more likely to stress concentration and surface effects due to the constraint of a slip step near a grain boundary at the free surface. This is further evidence that the m' factor is not always relevant for characterization of slip transmission at the surface.

4.3. Lattice rotation from bulk measurements

Significant differences in orientation spread, induced by deformation, are observed from near-surface measurements (EBSD) and bulk measurements (topotomography). This result is expected since the interior of the grain can evolve differently during deformation than the near surface of the grain. It is important to question the relationship between slip and orientation spread from near-surface and bulk measurements.

For instance, the grain labeled 660 has an orientation spread of 0.5° , whereas grain 063 displays a spread of only 0.25° . The analysis of the active slip system Burgers vectors shows that in grain 063, the vector points out of the free surface meaning the dislocations can readily escape the volume. However, in grain 660 the vector is almost parallel to the surface meaning that the dislocations do not escape, and instead pile-up and accumulate at the grain boundary, storing GNDs and creating mosaicity as a result. This behavior appears to be general for the surface grains measured by topotomography. This relation is evidenced in Figure 12(b) where the orientation spread is displayed as a function of the angle between the slip direction and the surface vector of the active slip plane. Grains with high orientation spread tend to display a slip direction that is close to the plane of the free surface. This provides more evidence of the effect of the free surface on the plastic activity. Other factors such as the grain size, the level of plasticity, the neighboring grains and the grain shape are also certainly relevant and are required to be investigated to highlight the relation between orientation spread and plastic activity. Furthermore, the additional consideration of the Schmid

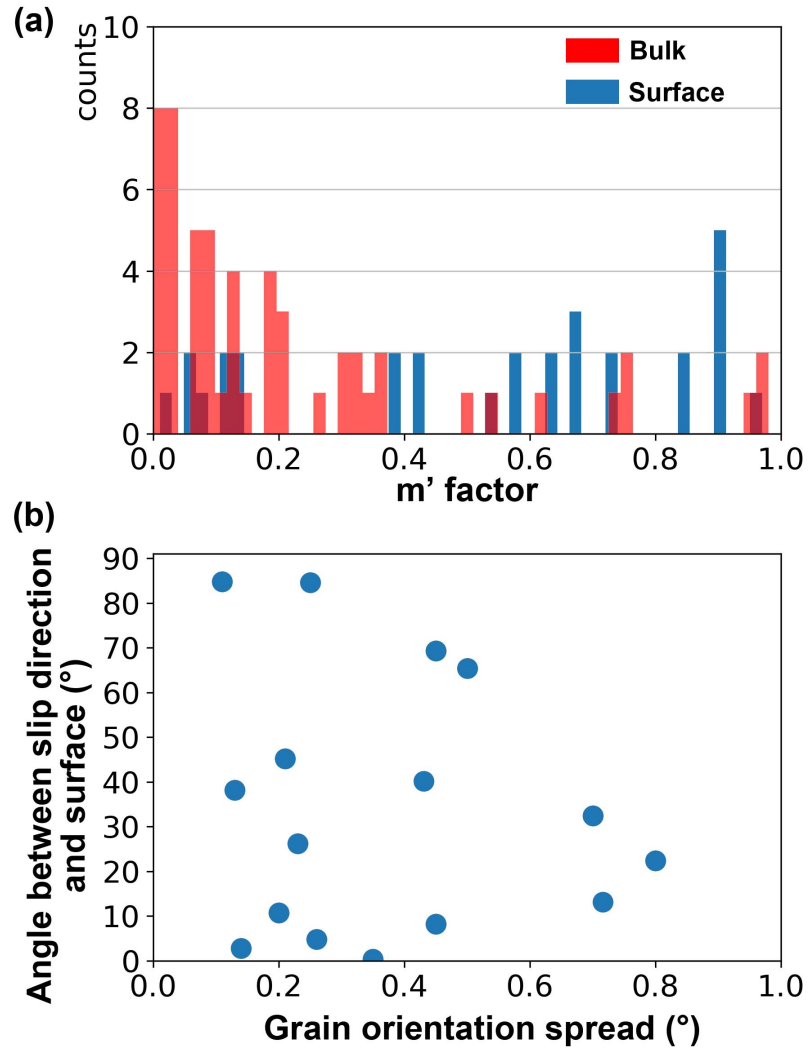


Figure 12: (a) Distribution of m' factor between surface grains and from grains in the bulk. The m' factors were calculated for all slip events detected by topotomography that are connected (at least one voxel in common at a grain boundary) across surface grains and from bulk grains. (b) Relation between GOS from topotomography measurements and the slip direction in relation to the free surface.

factor, Taylor factor, or local stress fields may provide information on the difference in behavior observed between the surface and the bulk.

The relationship observed previously between orientation spread and slip direction is not observed in orientation spread for the near-surface measurements obtained by EBSD. This indicates that EBSD and other surface measurements may not be representative of the plasticity that occurs within the entire grain, highlighting an advantage of the toptomography measurements, even for surface grains.

5. Future Directions

Topotomography is a materials characterization technique that has matured over the last 10 years, all while becoming more and more automated. It now also benefits from major synchrotron source upgrades that reduce data collection times to a few minutes per grain, as compared to several tens of minutes for the experiment reported here. This enables the systematic mapping of massive regions of grains selected from an initial DCT reconstruction of the grain volume. Scripts performing automated toptomography grain alignment and driving scan sequences (DCT, PCT and series of toptomography scans over a list of grains) have been developed and enable repeated acquisitions at increasing levels of applied strain, which could be instrumental in capturing plasticity propagation events. This framework and in particular the selection of candidate grains could possibly be guided by a digital-twin mechanical analysis conducted on the initial DCT reconstruction prior to a deformation step.

The question of the nature of the contrast mechanism(s) remains open, with likely origins being orientation contrast created by GNDs piling up at grain boundaries as evidenced in this paper, dislocation structures located within the slip bands, or a combination of both. Dislocation dynamics simulations based on experimental microstructures and coupled with forward modeling diffraction are currently being developed; they may accurately predict and indicate the potentially complicated contrast mechanisms in the near future.

Another current challenge relies on inverting the toptomography images to reconstruct the orientation field. Since toptomography cannot resolve lattice rotations around the probed scattering vector, the inversion has to include complementary grain projection data from DCT. This is in principle feasible using iterative approaches as demonstrated in [58], or by adapting the forward modeling strategy developed by [31] to the combined DCT and toptomography acquisition geometry. As the toptomography experiment geometry is more complex (it involves several rotations around the eucentric point), the sensitivity and spatial resolution of this approach may suffer from diffractometer error motion and requires an accurate calibration of the instrument geometry. Correlative experiments will be instrumental to validate the reconstructions in the future. To this end, recent updates in experimental stations at ESRF such as the nanobeam scanning 3DXRD station at ID11 or the dark field X-ray microscopy station at ID06 can be leveraged.

While the present results demonstrate different transmission behaviors in the bulk compared to at the surface, some uncertainties exist in the involved underlying mechanisms. For example, the complex role of the 3D grain structure on slip transmission still needs to be addressed. Specifically, significant efforts are proposed to elucidate the effect of triple junction lines on bulk slip transmission behavior. The combination of DCT and toptomography measurements, advanced segmentation and analysis tools and crystal plasticity calculations will be of primary importance to understanding this effect.

6. Conclusions

The correlative measurements from HR-DIC, toptomography and DCT made explicit the sensitivity and spatial resolution of the toptomography technique. Topotomography measurements were able to detect most of the intense slip events that developed during deformation of a titanium alloy. However, slip events of very low intensity or in regions of high lattice rotation were not detected by toptomography measurements, at least with the slip detection method used in this paper. Topotomography measurements provide the unique opportunity to investigate plastic activity in a large number of grains both at the surface and in the bulk. Both slip event location and orientation spread are obtained from toptomography. Several transmission events were also investigated in a titanium alloy by toptomography. The analysis of slip transfer between surface grains shows that many slip transmission events can be rationalized in terms of the geometrical alignment of activated slip systems in neighboring grains (using the m' factor for instance). However, the slip activity within the bulk of the specimen is observed to **not** be controlled by

slip transmission in the same extent as those of surface grains. In addition, an unknown transmission phenomena was observed at the surface, indicating again that completely different plastic behavior may occur in the bulk versus at free surfaces.

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