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DFT Modeling of the Oxidation Mechanism of Co(II) by Birnessite

Alain Manceau\*a and Stephan N. Steinmannb

<sup>a</sup> Univ. Grenoble Alpes, Univ. Savoie Mont Blanc, CNRS, IRD, ISTerre, 38000 Grenoble, France

<sup>b</sup> ENS de Lyon, CNRS, Laboratoire de Chimie, 69342 Lyon, France

Corresponding Author:

alain.manceau@univ-grenoble-alpes.fr

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**ABSTRACT** 

Phyllomanganates of the birnessite family are the most abundant manganese oxides on Earth and the strongest inorganic oxidants in the environment. Birnessite controls the oxidative scavenging of cobalt in soils, lake and marine sediments, and ferromanganese crusts and nodules, leading to enrichments of the order of one billion times the concentration in solution. However, a detailed mechanistic understanding of the enrichment processes is lacking. Here, we perform density functional theory (DFT) calculations to explore the mechanisms of Co(II) to Co(III) oxidation on the layer edge and surface of birnessite nanoparticles. We show that Co(II) sorption on a layer edge is an unlikely oxidation pathway. In contrast, Co(II) sorbed on a Mn(IV) vacancy site exposed on the layer surface as an octahedral triple-corner sharing (TCS) complex enters the vacancy where it is oxidized to Co(III) by a layer Mn(IV) cation, which is reduced to Mn(III). The stepwise reaction proceeds as follows. The octahedral TCS complex is transformed to a smaller tetrahedral TCS complex, allowing Co(II) to cross the surface oxygen layer and to fill the empty octahedral Mn(IV) site. When in the octahedral vacancy, Co(II) is converted from the high-spin  $(t_{2g}^5 e_g^2)$  to the low-spin  $(t_{2g}^6 e_g^1)$  state and the Co(II) octahedron becomes strongly distorted by the Jahn-Teller effect. Afterward, the electron exchange reaction between Mn(IV)  $(t_{2g}^3 e_g^0)$  and Co(II)  $(t_{2g}^6 e_g^1)$  takes place, resulting in the formation of a regular lowspin Co(III)  $(t_{2g}^{6}e_{g}^{0})$  octahedron and a Jahn-Teller distorted high-spin Mn(III)  $(t_{2g}^{3}e_{g}^{1})$  octahedron. These findings refine previously proposed mechanisms of Co(II) oxidation by birnessite and fill gaps in our understanding of global Co sequestration in natural systems.

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## 1. INTRODUCTION

Manganese dioxides (MnO<sub>2</sub>) dominate the geochemistry of cobalt (Co) in terrestrial and marine environments.<sup>1–22</sup> For example, the partitioning of Co between seawater and marine ferromanganese crusts is as high as 10<sup>9</sup>, and the second highest of all chemical elements after lead (Pb).<sup>23</sup> Evidences for the strong uptake of Co by MnO<sub>2</sub> has aroused the interest of scientists for decades and led to multiple sorption and coprecipitation studies aimed to understand the underlying chemical reaction on abiotic and biogenic MnO<sub>2</sub>,<sup>24–29</sup> and the structure, reactivity, and stability of Co-containing MnO<sub>2</sub>.<sup>30–33</sup>

Early observations of the Co-MnO<sub>2</sub> geochemical association in the 1970s invoked the oxidation of soluble Co(II) to insoluble Co(III) by Mn oxide minerals,  $^{34-36}$  and first direct evidences were obtained in 1979-1983 by Murray, Dillard, and Crowther using XPS.  $^{37-39}$  Mn(IV) was identified as the oxidizing agent at pH 6.5 based on the detection of Mn(III) in the birnessite phyllomanganate reacted with Co(II) under anaerobic condition. The XPS analysis was revisited by Yin et al.  $^{28}$  in 2011. This time, the authors measured the fractions of Mn(IV), Mn(III), and Mn(II) before and after Co(II) sorption on birnessite at pH 5, and observed an increase of both Mn(III) and Mn(II) on the solid phase. They confirmed that Mn(IV) was the primary oxidant and explained the occurrence of Mn(II) by the disproportionation of two reduced Mn(III) cations into Mn(IV) + Mn(II).

Burns<sup>35</sup> conjectured in 1976 that the edge-shared [MnO<sub>6</sub>] octahedral layers of birnessite contained Mn(IV) vacancies like chalcophanite, <sup>40,41</sup> and that Co(III) filled the vacated Mn(IV) sites on the basis of crystal field stabilization energies (CFSE) and steric considerations. Progress on the oxidation mechanism of Co(II) remained at a standstill for the following two decades due to the lack of robust structural model for birnessite. In 1997, Drits et al. <sup>42,43</sup> synthesized and determined the structure of well-crystallized hexagonal birnessite (HBi) using electron and X-ray diffraction and chemical analysis. Ever since, nanoparticulate hexagonal birnessite (δ-MnO<sub>2</sub>) has been identified in soils, lake and marine sediments, and mineral surface coatings and rock varnish, and is the main MnO<sub>2</sub> species produced by bacteria and fungi. <sup>44-51</sup> It is recognized to be the most abundant MnO<sub>2</sub> species at the Earth's surface. <sup>6,9,11,20,52-73</sup>

The elemental composition of HBi is non-stoichiometric and has for generic formula  $H_x^+Mn_y^{3+}Mn_z^{2+}(Mn_u^{4+}Mn_v^{3+}\square_w)O_2$ ,  $^{74}$  in which  $\square$  denotes an empty Mn(IV) position. The -4 charge deficit created by a vacancy is balanced by two H<sup>+</sup> on each side of the MnO<sub>2</sub> layer. Using EXAFS spectroscopy, Silvester et al.  $^{43}$  found in 1997 that the two protons on one side of a vacancy could be displaced by Zn(II) at pH 4 to form an interlayer triple-corner-sharing (TCS) inner-sphere complex at the surface of the MnO<sub>2</sub> layers (Figure 1a). The same year, Manceau et al.  $^{75}$  described the incorporation

at pH 4 of Co(III) into the octahedral vacancies (incorporated INC complex), consecutively to the oxidation of a  $Co_{TCS}^{2+}$  complex (Figure 1b). Mn(III) was inferred to be the electron acceptor because Mn(III) is considered to be a stronger oxidant than Mn(IV).<sup>75–77</sup>

Two reaction mechanisms were proposed based on X-ray diffraction, and powder and polarized EXAFS spectroscopy. The first mechanism can be written

$$Co_{Sol}^{2+}+\Box_{1}+\ Mn_{Layer}^{3+} \longrightarrow Co_{TCS}^{2+}+\Box_{1}+\ Mn_{Layer}^{3+} \longrightarrow Co_{TCS}^{3+}+\Box_{1}+\ Mn_{Layer}^{2+} \longrightarrow Co_{INC}^{3+}+\Box_{2}+\ Mn_{TCS/Sol}^{2+}\ (1)$$

where a solution (sol) Co(II) cation is sorbed on an existing vacancy, is oxidized by a layer Mn(III) cation, and the reduced Mn(II) cation either migrates to the interlayer forming a  $Mn_{TCS}^{2+}$  complex (TCS), or to solution creating a new vacancy. This mechanism can lead to high Co(III) enrichment because the density of vacancies remains constant. The second mechanism can be written

$$Co_{Sol}^{2+} + Mn_{TCS}^{3+} + \square \rightarrow Mn_{Sol}^{2+} + Co_{TCS}^{3+} + \square \rightarrow Mn_{Sol}^{2+} + Co_{INC}^{3+}$$
 (2)

where a Co(II) cation undergoes oxidation by an interlayer Mn(III) cation, which is released to solution. This reaction mechanism decreases the density of vacancies, and therefore diminishes the surface reactivity of birnessite over time.

In 2012, Yu et al.<sup>78</sup> sorbed Co(II) on abiotic and biogenic birnessite of different crystallinity and with various proportions of interlayer Mn(III). They found using EXAFS spectroscopy and elemental analysis that Co(II) was oxidized by Mn<sub>TCS</sub>, which was reduced to Mn(II) in nearly equimolar Co(III):Mn(II) ratio. In 2015, Simanova and Peña<sup>26</sup> used quick XANES and EXAFS spectroscopy to compare the reactivity of the basal and lateral surface sites of δ-MnO<sub>2</sub>. <sup>79,80</sup> Two main Co-Mn EXAFS distances were measured, 2.83-2.86 Å attributed to a Co(III)-Mn edge-sharing (ES) octahedral linkage, and 3.47-3.50 Å attributed to a Co(II)-Mn corner-sharing (CS) octahedral linkage. When the ES [CoO<sub>6</sub>] octahedron is located inside the MnO<sub>2</sub> layer (INC complex), the number of nearest Mn neighbors is six, and when it is located at the nanoparticle edges, it is two for a double-edge sharing complex (DES) and three for a triple-edge sharing complex (TES, Figures 1b-d). A CS [CoO<sub>6</sub>] octahedron has six nearest Mn neighbors in TCS position (Figure 1a), and two in double-corner-sharing (DCS) position at the nanoparticle edges (Figure 1e). EXAFS coordination numbers (CNs) are the weighted mean of the coordination numbers of all atoms bonded to  $\delta$ -MnO<sub>2</sub>. Therefore, the distribution of Co between the INC and DES sites can be inferred in principle from the CNs of the nearest Co-Mn<sub>1</sub> shell at 2.83-2.86 Å (ES linkage), and the distribution of Co between the DCS and TCS sites can be inferred from the CNs of the next-nearest Co-Mn<sub>2</sub> shell at 3.47-3.50 Å (CS linkage). Low CNs will reflect preferential

adsorption on lateral surfaces as DES and DCS complex, whereas high CNs will reflect preferential adsorption on basal surfaces as TCS and INC complex.

Simanova and Peña<sup>26</sup> observed in 2015 that CN(ES) increased from  $3.12 \pm 0.49$  after 7 min to  $4.13 \pm 0.62$  after 12 h reaction time at low surface coverage (Co/Mn = 0.05). CN(CS) increased in parallel from  $2.71 \pm 0.43$  to  $5.39 \pm 1.76$ . This evolution was interpreted to indicate that Co(II) is oxidized more rapidly on lateral than on basal surfaces. The fast oxidation on lateral surfaces was described as follows:

$$\text{Co}_{\text{Sol}}^{2+} + \text{Mn}_{\text{Layer}}^{3+} \rightarrow \text{Co}_{\text{DCS}}^{2+} + \text{Mn}_{\text{Layer}}^{3+} \rightarrow \text{Co}_{\text{Layer}}^{3+} + \text{Mn}_{\text{Sol}}^{2+}$$
 (3)

where a solution Co(II) sorbs as a DCS complex, is oxidized to Co(III) by a  $Mn_{Layer}^{3+}$  cation, and takes its position on the layer edge while Mn(III) is reduced to  $Mn_{Sol}^{2+}$ . TES complexation of Co(II) was not considered, and is unlikely to occur, because a TES surface site would have a high interfacial excess energy and would be immediately filled with a Mn(IV) cation to minimize the Gibbs free energy of the  $MnO_2$  layer. The slow oxidation reaction was attributed to reaction 2 of Manceau et al. The slow kinetics was explained by the lack of bridging ligand between the hexahydrated  $Co_{sol}^{2+}$  ion and the  $Mn_{TCS}^{3+}$  octahedron, and the outer-sphere mechanism of the electron transfer thereof. Wang et al. Pursued in 2018 the study of Simanova and Peña and concluded that the kinetics of oxidation on vacancy sites is also limited by diffusion process in the interlayer space of the  $\delta$ - $MnO_2$  nanosheets.

Density functional theory (DFT) modeling of Ni(II), Cu(II), and Zn(II) uptake by birnessite predicts that sorption as DES complex is favored over sorption as DCS and TCS complex,  $^{81}$  which is consistent with free edge-shared surface sites being the crystal growth sites of two-dimensional compounds. Therefore, formation of a  $Co_{DCS}^{2+}$  complex (reaction 3) challenges current understanding of the surface reactivity of birnessite.

Here, a DFT modeling of the oxidation mechanism of Co(II) on birnessite was conducted to explore the oxidation pathways of Co(II) on the lateral and basal surface sites, and whether the electron acceptor is Mn(IV) or Mn(III). One difficulty of modeling interfacial electron transfer from adsorbed Co(II) to structural Mn(III) is to construct realistic non-stoichiometric MnO<sub>2</sub> layers. Elzinga<sup>82–84</sup> showed that Mn(III) is labile, using <sup>54</sup>Mn radiotracers. Adsorption of Mn<sup>2+</sup><sub>sol</sub> on a Mn(IV) surface site produces transient Mn(III)-Mn(III) pairs by comproportionation, which subsequently undergo disproportionation to regenerate Mn<sup>2+</sup><sub>sol</sub> and Mn(IV).<sup>82–87</sup> Therefore, comproportionation needs to be considered in the generation of the structural models. This study is divided in two main parts. The first

part focuses on the oxidation of Co(II) in TCS position, and the second part on the oxidation of Co(II) in DCS and DES position on layer edges. Each part is divided in two sections. The first section deals with the construction of the MnO<sub>2</sub> layers with and without Mn(III), and the second section deals with the Gibbs free energy change ( $\Delta G$ ) for the complexation and oxidation of Co(II) on the surface sites of the MnO<sub>2</sub> models.

## 2. COMPUTATIONAL METHODS

Calculations were performed on cluster models with ORCA  $5.0.2^{88}$  using a computational methodology detailed previously for the complexation energies of Ni(II), Cu(II), Zn(II), and Pb(II) on birnessite. <sup>81</sup> Briefly, the DFT hybrid functional PBE0<sup>89,90</sup> together with the atom-pairwise dispersion correction D3BJ<sup>91</sup> has been used. The electronic configurations of all atoms were treated with the all-electron polarized def2-TZVP basis sets of triple  $\zeta$  quality, <sup>92</sup> in combination with the Coulomb fitting auxiliary def2 /J basis sets. <sup>93</sup> Frequency calculations were carried out at 1 atm and 298.15 K on cluster models geometrically optimized with a convergence criterium of  $10^{-8}$  Eh between two SCF cycles (TightSCF). The water solvent was modeled with the continuum solvent model SMD, <sup>94</sup> and all energies reported are free energies in solution (i.e., sometimes denoted as  $\Delta G^*$ ) and in kcal/mol. The proton solvation energy was taken to be -264.0 kcal/mol. <sup>95</sup> The potential energy surface of Co(II) involved along the TCS  $\rightarrow$  INC reaction was calculated by the stepwise (n = 18) reduction of the dihedral angle of Co(II) with three layer Mn atoms. The energetics of the Co(II)  $\rightarrow$  Co(III) electron transfer was modeled using the minimum energy crossing point (MECP) procedure proposed by Harvey et al. <sup>96</sup>, as implemented with the SurfCrossOpt keyword in ORCA.

The accuracy of prediction of  $\Delta G$  has been benchmarked previously<sup>81</sup> with the first hydrolysis constants of the hexahydrated Mn<sup>2+</sup>, Ni<sup>2+</sup>, and Zn<sup>2+</sup> ions.<sup>97</sup> It was shown that  $\Delta G$  values for these ions are calculated reliably within chemical accuracy (1–2 kcal/mol). For example, the experimental and calculated p $K_a$  values for Mn<sup>2+</sup> are 10.6 and 11.3, respectively. The method accuracy was further validated here with Co<sup>2+</sup>: p $K_a$ (exp) = 9.7 ± 1<sup>98</sup>, corresponding to  $\Delta G$  = 13.2 kcal/mol, and p $K_a$ (calc.) = 11.5, corresponding to  $\Delta G$  = 15.65 kcal/mol. Therefore, the computational procedure probably tends to overestimate  $\Delta G$  values by approximately 2.5 kcal/mol. The total electronic charges of the cluster models, and their Mulliken atomic spin densities and multiplicities are listed in Table S1.

## 3. RESULTS

## 3.1. Co(II) oxidation on basal surfaces

3.1.1. Cluster models. The active sites on basal surfaces are Mn(IV) vacancies. Four birnessite nanolayers composed of 10 octahedral sites in the layer plane were generated (Figure 2). Nanolayers 1 and 2 will be used to test reactions 1 and 2, respectively. The first nanolayer has one Mn(III) in the layer plane (Mn $_{Layer}^{3+}$ ) and a fully protonated vacancy. It is denoted as  $Mn_{Layer}^{3+}Mn_8^{4+}V$  and has for composition Mn<sub>9</sub>O<sub>32</sub>H<sub>26</sub>. The second nanolayer has only Mn(IV) in the layer plance (Mn $_{Layer}^{4+}$ ) and a vacancy capped by a Mn(III) cation on one side (Mn $_{TCS}^{3+}$ ) and protonated on the other side. It is denoted as  $Mn_{TCS}^{3+}Mn_9^{4+}V$  and has for composition Mn<sub>10</sub>O<sub>35</sub>H<sub>30</sub>. The third nanolayer also has only Mn $_{Layer}^{4+}$  in the layer plane, but no  $Mn_{TCS}^{3+}$ . It is denoted as  $Mn_{9}^{4+}V$  and has for composition Mn<sub>9</sub>O<sub>32</sub>H<sub>26</sub>. The fourth nanolayer differs from nanolayer 2 by the replacement of  $Mn_{TCS}^{3+}$  with  $Mn_{TCS}^{2+}$ , and has for composition Mn<sub>10</sub>O<sub>35</sub>H<sub>30</sub>. This configuration was unstable, Mn(II) underwent comproportionation with a  $Mn_{Layer}^{4+}$  to form a  $Mn_{TCS}^{3+}$  —  $Mn_{Layer}^{3+}$  pair. The total energy decreased by 1 kcal/mol when the Mn(III) cations were antiferromagnetically coupled. The antiferromagnetic model is denoted as  $Mn_{TCS}^{3+}Mn_{Layer}^{3+}Mn_{R}^{4+}V$ .

3.1.2. Energetics of the oxidation pathways. All oxidation pathways that were explored are listed in Table 1 with their  $\Delta G$  value. According to the mass action law, a reaction with  $\Delta G > 0$  can be favorable above a given pH when protons are released. In this case, the ratio of the concentrations of the reactant (R) and product (P) at pH 4 and 7 was calculated from [P]/[R] =  $K_{eq}$ /[H<sup>+</sup>]<sup>n</sup>, where  $K_{eq}$  is the equilibrium constant of the reaction (ln  $K_{eq} = -\Delta G/RT$ ).

In reaction 1,<sup>75</sup> Co<sub>Sol</sub><sup>2+</sup> sorbs on a free vacancy bordered by five  $Mn_{Layer}^{4+}$  and one  $Mn_{Layer}^{3+}$ . Co<sub>Sol</sub><sup>2+</sup> is oxidized by  $Mn_{Layer}^{3+}$ , and  $Co_{TCS}^{3+}$  migrates into the vacancy (Co<sub>INC</sub><sup>3+</sup>) while  $Mn_{Layer}^{2+}$  either moves in TCS position ( $Mn_{TCS}^{2+}$ , pathway 1a) or is released in solution ( $Mn_{sol}^{2+}$ , pathway 1b) (Figure 3). Pathway 1a has  $\Delta G = 4.4$  kcal/mol and a net-zero H<sup>+</sup> balance, whereas pathway 1b has  $\Delta G = 4.2$  kcal/mol and consumes two H<sup>+</sup>. Therefore, reaction 1 is thermodynamically unfavorable. As an aside, we note that the  $Mn_{TCS}^{2+}$ -Co<sub>INC</sub><sup>3+</sup> pair formed in pathway 1a was stable;  $Mn_{TCS}^{2+}$  did not comproportionate with a  $Mn_{Layer}^{4+}$  in this configuration.

In reaction 2a,  $^{75}$   $Co_{Sol}^{2+}$  is oxidized to  $Co_{TCS}^{3+}$  by  $Mn_{TCS}^{3+}$ , which is reduced to  $Mn_{Sol}^{2+}$  (Figure 4). Afterward,  $Co_{TCS}^{3+}$  migrates into the vacancy ( $Co_{INC}^{3+}$ ) releasing the two H<sup>+</sup> located on the other side of the vacancy (pathway 2b). Pathway 2a has a highly positive  $\Delta G$  value of 17.1 kcal/mol, and pathway 2b has  $\Delta G = -17.7$  kcal/mol and  $[Co_{INC}^{3+}]/[Co_{TCS}^{3+}] = 9.4 \times 10^{26}$  at pH 7. Pathway 2a is thermodynamically prohibited, wheareas pathway 2b is spontaneous. If a Co(III) occurs on a vacancy, it will immediately fill it. Reaction  $3^{26}$  occurs on lateral surfaces and will be examined in the following section.

In reaction 4,  $Co_{Sol}^{2+}$  sorbs on a free vacancy and  $Co_{TCS}^{2+}$  is oxidized to  $Co_{INC}^{3+}$  by a  $Mn_{Layer}^{4+}$ , which is reduced to  $Mn_{Layer}^{3+}$ . The net proton balance of the oxidation reaction is -2 if the two H<sup>+</sup> located on the other side of the vacancy are released in solution (pathway 4a), or zero if the H<sup>+</sup> release is balanced by the adsorption of two H<sup>+</sup> on the undersaturated oxygens shared by the  $[Co^{3+}O_6]$  and  $[Mn^{3+}O_6]$  octahedra (pathway 4b, Figure 5). Pathway 4a has  $\Delta G = 16.9$  kcal/mol and  $[Co_{INC}^{3+}]/[Co_{TCS}^{2+}] = 40.9$  at pH 7, and pathway 4b has  $\Delta G = -5.2$  kcal/mol and  $[Co_{INC}^{3+}]/[Co_{TCS}^{2+}] = 6.5 \times 10^3$ , regardless of pH. Correcting the two  $\Delta G$  values by the estimated 2.5 kcal/mol overestimation of the calculation (section 2) yields  $\Delta G = 14.4$  kcal/mol and  $[Co_{INC}^{3+}]/[Co_{TCS}^{2+}] = 2.8 \times 10^3$  at pH 7 for pathway 4a, and  $\Delta G = -7.7$  kcal/mol and  $[Co_{INC}^{3+}]/[Co_{TCS}^{2+}] = 3.1 \times 10^5$  for pathway 4b. This Co(II) oxidation pathway is thermodynamically favorable.

In reaction 5,  $Co_{Sol}^{2+}$  replaces  $Mn_{TCS}^{3+\downarrow}$  in the antiferromagnetic  $Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow}$  pair, forming a  $Co_{TCS}^{2+}$  complex, while  $Mn_{TCS}^{3+\downarrow}$  is reduced to  $Mn_{Sol}^{2+}$  by  $Mn_{Layer}^{3+\uparrow}$ , which is oxidized to  $Mn_{Layer}^{4+}$  (Figure 5). Afterward,  $Co_{TCS}^{2+}$  is oxidized following the reaction 4 pathway. In this mechanism, adsorption of  $Co_{Sol}^{2+}$  reverses the  $Mn_{TCS}^{2+}Mn_{Layer}^{4+} \rightarrow Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow}$  comproportionation reaction, and  $Co_{Sol}^{2+}$  substitutes for  $Mn_{TCS}^{2+}$  according to the  $Co_{Sol}^{2+}Mn_{TCS}^{2+} \leftrightarrow Co_{TCS}^{2+}Mn_{Sol}^{2+}$  exchange reaction. Reaction 5 is thermodynamically favorable with  $\Delta G = -6.7$  kcal/mol. We verified that  $Co_{Sol}^{2+}$  is not oxidized by  $Mn_{TCS}^{3+\downarrow}$  in this configuration, as is the case for pathway 2a when  $Mn_{TCS}^{3+}$  is not paired with a  $Mn_{Layer}^{3+}$ . Here,  $\Delta G$  of  $Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow} \rightarrow Co_{TCS}^{3+}Mn_{Layer}^{3+}$  is 22.0 kcal/mol compared to 17.1 kcal/mol for pathway

2a. Oxidation of Co(II) on Mn(III)-capped vacancies will be slower than on vacancy sites freed from interlayer Mn(III).

Because the nanolayers are treated as molecules in DFT geometry-optimization, small cluster models with a vacancy can experience too much relaxation that would not occur to mineral (nano)particles, and therefore would lead to wrong interpretation. This cautionary tale was addressed by increasing to 23 [MnO<sub>6</sub>] octahedra the size of the nanolayer and by calculating the  $\Delta G$  values of reaction 5 (Mn $_{TCS}^{3+\downarrow}$ Mn $_{Layer}^{3+\uparrow}$ Mn $_{22}^{4+}$ V  $\rightarrow$  Co $_{TCS}^{2+}$ Mn $_{23}^{4+}$ V) (Figure S1).  $\Delta G(5) = -6.3$  kcal/mol for the Mn<sub>23</sub>V model compared to  $\Delta G(5) = -6.7$  kcal/mol for the Mn<sub>9</sub>V model (Table 1).

3.1.3. Oxidation mechanism of Co(II) on vacancies. To be examined now is how Co(II) enters the vacancy site and is oxidized by Mn(IV) during reaction 4  $(Co_{TCS}^{2+}Mn_{Layer}^{4+} \rightarrow Co_{INC}^{3+}Mn_{Layer}^{3+})$ . We interrogated first if the electron transfer occurred between  $Co_{TCS}^{2+}$  and  $Mn_{Layer}^{4+}$  before the entry into the vacancy site. This was performed by optimizing the geometry of  $Co_{TCS}^{3+}Mn_{Layer}^{4+}Mn_8^{4+}V$ , with the  $d^6Co^{3+}$  ion in low-spin state  $(t_{2g}^6e_{g}^0,S=1)$ , and the  $d^4Mn^{3+}$  and  $d^3Mn^{4+}$  ions in high-spin state  $(t_{2g}^3e_{g}^1,S=5;t_{2g}^3e_{g}^0,S=4$ , respectively,  $S_{tot}=29$ ). On analysis of the spin electronic densities  $(\rho)$  obtained through the Mulliken population analysis, Co had  $\rho=0.94$   $(S\sim2)$  and all the Mn atoms had  $\rho\sim3$   $(S\sim4)$  in the optimized structure. Thus, the thermodynamically most stable complex is  $Co_{TCS}^{2+LS}Mn_9^{4+}V$  when the total spin is 29, with the  $d^7Co^{2+}$  ion in low-spin state  $(t_{2g}^6e_{g}^1,S=2)$ , instead of its usual high-spin state  $(t_{2g}^5e_{g}^2,S=4)$ . The unpaired electron in the  $e_g$  orbitals of low-spin Co(II) leads to a Jahn-Teller distortion of the  $[Co_{TCS}^{2+}O_3(H_2O)_3]$  octahedron with  $d(Co-O_{2Mn})=1.96$  Å, 1.99 Å, 2.14 Å and  $d(Co-H_2O)=2.04$  Å, 2.05 Å, 2.22 Å. Still, a stable  $Co_{TCS}^{3+}Mn_{Layer}^{3+}Mn_8^{4+}V$  complex could be obtained by enforcing a Mn(III) octahedron in the layer, but the total electronic energy of the complex was 12 kcal/mol higher than that of  $Co_{TCS}^{2+LS}Mn_9^{4+}V$ .

This finding suggests that the  $Co_{TCS}^{2+,LS}Mn_9^{4+}V$  complex may be a stable intermediate state, in which Co(II) now possesses a single unpaired electron, which is transferred to  $Mn_{Layer}^{4+}$  yielding the  $Co_{INC}^{3+}Mn_{S}^{4+}V$  reaction product. The  $Co_{TCS}^{2+,LS}Mn_9^{4+}V$  complex is  $\Delta G=19.2$  kcal/mol higher in energy than the  $Co_{TCS}^{2+}Mn_9^{4+}V$  reactant complex. The  $Co_{TCS}^{3+}Mn_{S}^{4+}V$  complex is further higher with

 $\Delta G = 28.7$  kcal/mol. We show below with a PES scan, that the hypothesis of a  $\mathrm{Co_{TCS}^{2+}Mn_{Layer}^{4+}} o \mathrm{Co_{TCS}^{2+,LS}Mn_{Layer}^{4+}} o \mathrm{Co_{INC}^{3+}Mn_{Layer}^{3+}}$  reaction pathway is not supported.

The reactant complex (R) of the PES scan was  $Co_{TCS}^{2+}Mn_9^{4+}V_d^{}$  (Stot = 31) and the product complex (P) was  $Co_{INC}^{2+}Mn_9^{4+}$  with the two protons from the opposite side of the vacancy removed to avoid obstructing the migration of Co (V<sub>d</sub> denotes a deprotonated vacancy). Note that the spin state cannot be changed during a PES scan. The PES has a double-hump profile, characteristic of the occurrence of an intermediate product (IP1) bracketed by two transition states (TS1, TS2) (Figure 6). The IP1 state is a tetrahedral complex (  $^{IV}Co_{TCS}^{2+}Mn_9^{4+}V_d$  ), in which Co(II) is bonded to the three surface  $O_{2M}$  atoms at 1.92-1.95 Å and one water molecule at 2.06 Å, and resides in the high-spin state (Figure 7). The transition from the octahedral to the tetrahedral coordination occurs via the loss of a first water molecule on the way to TS1. In the actual transition state (Video S1), a second coordination bond between a water molecule and  $Co^{2+}$  is broken ( $v = 98 i \text{ cm}^{-1}$ , single negative frequency calculated at PBE0/tzvp-D3BJ level), providing a tetrahedral complex (IP1) behind. In IP1, the two expelled water molecules are hydrogen bonded to the oxygen atom of the remaining water molecule coordinated to Co<sup>2+</sup>. The penetration of  $^{\text{IV}}\text{Co}_{\text{TCS}}^{2+}$  in the vacancy is shown in Video S2, which represents the imaginary frequency of TS2 ( $v = 1308 i \text{ cm}^{-1}$ , single negative frequency calculated at PBE0/tzvp-D3BJ level). The three surface  $O_{2M}$  atoms move away from Co (d(Co- $O_{2M}$ ) = 2.01-2.10 Å) to enlarge the effective size of the trigonal aperture, while the three basal  $O_{2M}$  atoms move upward to  $d(Co-O_{2M}) = 2.3-2.5 \text{ Å}$  to bond the entering Co atom. <sup>IV</sup>Co(II) looses its water molecule when the distance to the basal O<sub>2Mn</sub> atoms is 2.2 Å. At this step, Co(II) is located slightly below the surface oxygen plane of the octahedral cavity with  $d(\text{Co-O}_{2\text{Mn}}) = 2.0 \text{ Å. Animations of the PES scans for the } \text{Co}_{\text{TCS}}^{2+} \text{Mn}_9^{4+} \text{V}_d \rightarrow \text{Co}_{\text{INC}}^{2+} \text{Mn}_9^{4+} \text{ and the } \text{Mn}_9^{4+} \text{V}_d \rightarrow \text{Co}_{\text{INC}}^{2+} \text{Mn}_9^{4+} \text{ and the } \text{Mn}_9^{4+} \text{Mn}_9^{4+} \text{ and the } \text{Mn}_9^{4+} \text{Mn}_9^{4+} \text{ and the } \text{Mn}_9^{4+} \text{Mn}_9^{4+$  $\text{Co}^{2+}_{\text{TCS}}\text{Mn}^{4+}_{23}\text{V}_{\text{d}} \rightarrow \text{Co}^{2+}_{\text{INC}}\text{Mn}^{4+}_{23}$  pathways can be viewed in Videos S3, and S4, respectively.

The first potential energy barrier is  $\Delta E(\text{TS1-R})^{\ddagger} = 7.2$  kcal/mol relative to R, and the second is  $\Delta E(\text{TS2-IP1})^{\ddagger} = 3.0$  kcal/mol relative to IP1 (Figure 6). The Gibbs free energy differences  $\Delta G^{\ddagger}(\text{TS1-R})$ ,  $\Delta G^{\ddagger}(\text{TS2-IP1})$ , and  $\Delta G(\text{IP1-R})$  were calculated to be 4.4 kcal/mol, 2.4 kcal/mol, and -0.4 kcal/mol, respectively (reaction 7a, Table 1, Figures 7 and S2).

The release of the two basal protons located on the other side of the Mn(IV) vacancy during the migration of Co(II) is challenging to model, and therefore was omitted. A sense of how much they modify the energetic of the  $Co_{TCS}^{2+} \rightarrow {}^{IV}Co_{TCS}^{2+}$  reaction was obtained by calculating the PES scan with

two protons bonded to the basal  $O_{2Mn}$  atoms. The energy difference between IP1 and R increased from -0.4 kcal/mol to 4.2 kcal (reaction 7b, Table 1, and Figure S2), and d(Co- $O_{2Mn}$ ) increased from 1.92-1.95 Å to 1.94-1.96 Å. In real situation where the vacancy is protonated, the two basal protons are likely released during the migration of the tetrahedral complex into the octahedral cavity, and they raise the TS1 barrier from  $\Delta E^{\ddagger}(TS1-R) = 7.2$  kcal/mol to  $\Delta E^{\ddagger}(TS1-R) = 19.4$  kcal/mol, but not the TS2 barrier relative to IP1 ( $\Delta E^{\ddagger}(TS2-IP1) = 3.0$  kcal/mol).

Modeling of the oxidation state switching of Co(II) and Mn(IV) after the MECP procedure of Harvey et al.  $^{96}$  shows that Co(II) undergoes a spin flip inside the vacancy allowing for interconversion between the quartet (S = 4)  $t_{2g}^{5}e_{g}^{2}$  spin state and the doublet (S = 2)  $t_{2g}^{6}e_{g}^{1}$  spin state (Figure 7). The Gibbs free energy of the  $Co_{TCS}^{2+}Mn_{Layer}^{4+} \rightarrow Co_{INC}^{2+,LS}Mn_{Layer}^{4+}$  reaction is  $\Delta G = 0.1$  kcal/mol when protons are omitted (reaction 8, Table 1, and Figure S2). Finally,  $Co_{INC}^{3+}$  is formed through the energetically favorable reaction  $Co_{INC}^{2+,LS}Mn_{INC}^{4+} \rightarrow Co_{INC}^{3+,LS}Mn_{INC}^{3+}$  ( $\Delta G = -17.7$  kcal/mol, Figure 7).

## 3.2. Co(II) oxidation on lateral surfaces

3.2.1. Cluster models. Two birnessite nanolayers composed of 7 octahedral sites were generated (Figure 8). Nanolayer 1 has one  $Mn_{Layer}^{3+}$  at the surface and will be used to test reaction 3, and nanolayer 2 has only  $Mn_{Layer}^{4+}$  and will be used to test the oxidation of Co(II) by a surface Mn(IV). Nanolayer 1 is denoted as  $Mn_{Layer}^{3+}Mn_6^{4+}$ , nanolayer 2 is denoted as  $Mn_7^{4+}$ , and they both have for composition  $Mn_7O_{24}H_{18}$ . The [MnO6] octahedra exposed to the surface share two (DES linkage) or three (TES linkage) edges with the nearest octahedra, and therefore a surface Mn(III) cation may occupy two positions in nanolayer 1. This cation results most likely from the comproportionation of a sorbed Mn(II) cation on the DES or TES surface sites and a layer Mn(IV) cation. Therefore, the linkage of  $Mn_{Layer}^{3+}$  in nanolayer 1 was assessed by optimizing the structure of a  $Mn_{DES}^{2+}Mn_7^{4+}$  and a  $Mn_{TES}^{2+}Mn_6^{4+}$  nanolayer (Figure S3) A  $Mn_{DES}^{2+}$  complex is stable, whereas a  $Mn_{TES}^{2+}$  complex undergoes comproportionation with a  $Mn_{Layer}^{4+}$  to form a  $Mn_{TES}^{3+}Mn_{Layer}^{3+}$  pair. Therefore, the surface Mn(III) cation was placed in TES position in nanolayer 1 (Figure 8a).

3.2.2. Energetics of surface complexation and oxidation pathways. Co(II) can form a DCS or a DES complex on nanolayers 1 and 2 (Figure 9). The two complexes are bonded to one  $Mn_{Layer}^{3+}$  and to one  $Mn_{Layer}^{4+}$  in nanolayer 1 ( $Mn_{Layer}^{3+}Mn_6^{4+}$ ), and to two  $Mn_{Layer}^{4+}$  in nanolayer 2 ( $Mn_7^{4+}$ ) (Figures 9 and S4). The Gibbs free energies of complexation of the four configurations are:  $\Delta G(Co_{DCS}^{2+} - Mn_{Layer}^{3+}Mn_6^{4+}) = -23.2$  kcal/mol;  $\Delta G(Co_{DCS}^{2+} - Mn_7^{4+}) = -16.5$  kcal/mol;  $\Delta G(Co_{DES}^{2+} - Mn_6^{4+}) = -12.1$  kcal/mol;  $\Delta G(Co_{DES}^{2+} - Mn_7^{4+}) = -18.1$  kcal/mol (Figure S4). Co(II) has a higher affinity for Mn(III) than for Mn(IV) in DCS position, and a higher affinity for Mn(IV) than for Mn(III) in DES position. Complexation on a DES site releases one proton, and no proton on a DCS site. At pH 7, the calculated  $[Co_{DCS}^{2+} - Mn_{1ayer}^{3+}Mn_6^{4+}]/[Mn_{Sol}^{2+}]$  ratio is  $1.0 \times 10^{17}$ , and the  $[Co_{DES}^{2+} - Mn_7^{4+}]/Mn_{Sol}^{2+}]$  ratio is  $1.8 \times 10^{20}$ . Therefore, formation of a DES complex is preferred over that of a DCS complex, as is the case for the complexation of Ni(II), Cu(II), and Zn(II).

In the oxidation reaction 3, solvated Co(II) forms a  $Co_{DCS}^{2+}$  complex on a  $Mn_{Layer}^{3+}Mn_6^{4+}$  surface site, transfers an electron to  $Mn_{Layer}^{3+}$ , which is reduced to  $Mn_{Sol}^{2+}$ , and the oxidized  $Co_{DCS}^{3+}$  species fills the vacant  $Mn_{Layer}^{3+}$  site (Figure 9a).<sup>26</sup> This pathway is thermodynamically unfavorable with  $\Delta G = 34.1$  kcal/mol (Table 1).  $Mn_{Layer}^{4+}$  is the oxidant in reactions 9 and 10, and the reduced  $Mn_{Layer}^{3+}$  cation remains in the structure forming a  $Co_{Layer}^{3+}$  -  $Mn_{Layer}^{3+}$  pair (Figures 9b,c). Reaction 9 involves a  $Co_{DCS}^{2+}$  complex and has  $\Delta G = 19.5$  kcal/mol and reaction 10 involves a  $Co_{DES}^{2+}$  complex and has  $\Delta G = 21.1$  kcal/mol. The two pathways are also thermodynamically unfavorable. Other pathways were tested, but we could not find any which was energetically favorable on lateral surfaces.

## 4. DISCUSSION

Computational chemistry reveals an evolving new view of the oxidation mechanism of Co(II) on hexagonal birnessite. According to the predictive thermodynamic models explored in this study, Co(II) is not oxidized on  $MnO_2$  layer edges but on vacancy sites on basal planes. Vacancy sites are known to form strong TCS complexes with divalent metals, and to be involved in electron exchange reaction with Mn(III) and Mn(IV). They facilitate photoconductivity and photoreductive dissolution of the  $MnO_2$  layer. Although Mn(III) has a strong electron affinity

and is usually regarded to mediate the oxidative activity of birnessite, whether it is bonded over a vacancy ( $Mn_{TCS}^{3+}$ ) or is incorporated in the lattice ( $Mn_{Layer}^{3+}$ ),  $^{82-84,125-131}$  our results are at odds with that interpretation for the oxidation of Co(II) to Co(III). We found that Co(II) sorbed on a vacancy site ( $Co_{TCS}^{2+}$ ) is oxidized to  $Co_{INC}^{3+}$  by a layer Mn(IV) in the vacancy. This molecular mechanism is consistent with the increase in Mn(III) content measured by XPS.  $^{39,28}$ 

Co(II) undergoes successively a coordination change from octahedral (VICo<sub>TCS</sub>Mn<sub>Laver</sub>) to tetrahedral  $(^{IV}Co_{TCS}^{2+}Mn_{Layer}^{4+})$  to enter the vacancy site followed by a quartet  $(t_{2g}{}^5e_g{}^2,\,S=4)$  to doublet  $(t_{2g}{}^6e_g{}^1,\,S=4)$ 2) to singlet  $(t_{2g}^{6}e_{g}^{0}, S = 1)$  spin transition during its oxidation to Co(III). The potential energy barriers for the octahedral to tetrahedral conversion ( $Co_{TCS}^{2+}Mn_{Laver}^{4+} \rightarrow {}^{IV}Co_{TCS}^{2+}Mn_{Laver}^{4+}$ , TS1) and for the penetration of the tetrahedral complex into the vacancy (TS2) are as low as  $\Delta E^{\ddagger} = 7.2$  kcal/mol and 3.0 kcal/mol, respectively. This can be understood in light of the d<sup>7</sup> electronic configuration of Co(II) with regard to that of Ni(II) (d<sup>8</sup>), which can also enter the vacancy sites. 54,57,132 The TS1 and TS2 barriers of Ni are 14.4 kcal/mol and 3.0 kcal/mol, respectively (Figure 6). The 7.2 kcal/mol difference in height of the TS1 state for Co and Ni can be explained with reference to their difference of crystal field stabilization energy (CFSE) in octahedral and tetrahedral coordination. An electron gains -0.4  $\Delta$  in one of the orbitals of the  $t_{2g}$  levels and looses  $+0.6 \Delta$  in one of the orbitals of the  $e_g$  levels, where  $\Delta$  is the crystal field splitting parameter. <sup>133</sup> A high-spin Co<sup>2+</sup> ion acquires a CFSE of -0.8 Δ<sub>oct</sub> in octahedral coordination and -1.2  $\Delta_{\text{tet}}$  in tetrahedral configuration, and a high-spin Ni<sup>2+</sup> ion acquires, respectively, a CFSE of 1.2  $\Delta_{oct}$  and -0.8  $\Delta_{tet}$ . Taking  $\Delta_{tet} = 4/9 \Delta_{o}$ , we can calculate the difference in CFSE between octahedral and tetrahedral geometries by referencing everything in terms of  $\Delta_{oct}$ . The net difference is 0.27  $\Delta_{\rm oct}$  for Co(II) and 0.85  $\Delta_{\rm oct}$  for Ni(II), and therefore the gap between the two coordinations and height of the TS1 state are smaller for Co(II) than for Ni(II).

Turning to the height of the TS2 state, the close correspondance for Co(II) and Ni(II) can be explained now with reference to the similarity of their effective ionic radii in tetrahedral coordination with  $r(^{IV}Co^{2+} = 0.58 \text{ Å})$  and  $r(^{IV}Ni^{2+} = 0.55 \text{ Å}).^{134}$  The decrease in size of Co(II) and Ni(II) resulting from the octahedral to tetrahedral coordination change facilitates the passage through the small triangular-shaped aperture delimited by the three  $O_{2Mn}$  atoms at the surface of the MnO<sub>2</sub> layer.

The  $d^7$  electronic structure of Co(II) also confers special characteristics on Co(II) systems regarding the interconversion between the high-spin ( $t_{2g}^5e_g^2$ , S = 4) and the low-spin ( $t_{2g}^6e_g^1$ , S = 2) states. Unpaired electrons can easily undergo spin inversion and both spin states configurations are known in

six-coordinate Co(II) complexes. <sup>135</sup> The single electron in the  $e_g$  orbitals induces a strong Jahn-Teller effect inside the vacancy site with predicted  $d(\text{Co-O}_{eq}) = 1.88\text{-}1.90\,\text{ Å}$  and  $d(\text{Co-O}_{ax}) = 2.27\,\text{ Å}$ . A significant result is that the intermediate  $Co_{\text{INC}}^{2+\text{LS}}Mn_9^{4+}V$  complex is thermodynamically as stable as the initial  $Co_{\text{TCS}}^{2+}Mn_9^{4+}V$  complex ( $\Delta G = 0.1\,\text{kcal/mol}$ ). Formation of the  $Co_{\text{INC}}^{2+\text{LS}}Mn_9^{4+}V$  complex is followed by an electron transfer from a low-spin  $Co^{2+}$  ion to a high-spin  $Mn^{4+}$  ion ( $t_{2g}^{3}e_{g}^{0}$ ), which is reduced to high-spin  $Mn^{3+}$  ( $t_{2g}^{3}e_{g}^{1}$ ) while  $Co^{2+}$  is oxidized to low-spin  $Co^{3+}$  ( $t_{2g}^{3}e_{g}^{0}$ ). The  $Co^{3+}$ -O distances of the resulting [CoO<sub>6</sub>] octahedron are regular ( $d(\text{Co-O}) = 1.90\text{-}193\,\text{Å}$ ), whereas the  $Mn^{3+}$ -O distances are distinctive of a Jahn-Teller distorsion with  $d(Mn\text{-O}_{eq}) = 1.90\text{-}1.92\,\text{Å}$  and  $d(Mn\text{-O}_{ax}) = 2.19\text{-}2.21\,\text{Å}$ . The close match between the low-spin  $Co^{2+}$ -O and high-spin  $Mn^{3+}$ -O distances is consistent with the similarity of the ionic radii for the two metal ions (low-spin  $Co^{2+} = 0.65\,\text{Å}$ , high-spin  $Mn^{3+} = 0.645\,\text{Å}$ ).  $l_{134}^{3+}$  Formation of the low-spin  $l_{2g}^{2}e_{g}^{1}$  Co(II) configuration makes it easier for an  $l_{2g}^{2}e_{g}^{1}$  Mn(III) configuration. The oxidation state switching is predicted to be highly exergonic ( $\Delta G = -17.6\,\text{kcal/mol}$ ). The question remains as to which reaction step is rate limiting in the stepwise  $l_{134}^{2+}$  and  $l_{134}^{2+}$  are  $l_{134}^{2+}$  and  $l_{134}^{2+}$  and  $l_{134}^{2+}$  are  $l_{134}^{2+}$  and  $l_{134}^{2+}$  and  $l_{134}^{2+}$  are  $l_{134}^{2+$ 

The release in solution of Mn(II) that accompanies the oxidation of Co(II) is interpreted usually as evidence for Mn(III) acting as the electron acceptor. In the new model, aqueous Mn(II) measured by chemical analysis during the oxidation of Co(II) results, at least partly, from the stepwise disproportionation of  $Mn_{TCS}^{3+}Mn_{Layer}^{3+}$  into  $Mn_{TCS}^{2+}Mn_{Layer}^{4+}$  followed by the substitution of  $Co_{TCS}^{2+}$  for  $Mn_{TCS}^{2+}$  and release of Mn(II) to solution. The chemical formula of well-crystallized hexagonal birnessite is typically  $H_{0.33}^{+}Mn_{0.111}^{3+}Mn_{0.055}^{2+}(Mn_{0.722}^{4+}Mn_{0.11}^{3+}\Box_{0.167})O_2 \cdot (H_2O)_{0.50}$  at pH 5,<sup>74</sup> and the chemical formula of Na-exchanged  $\delta$ -MnO2 is typically  $H_{0.16}^{+}Na_{0.18}^{+}Mn_{0.15}^{3+}(Mn_{0.72}^{4+}Mn_{0.11}^{3+}\Box_{0.17})O_2 \cdot (H_2O)_{0.75}$  at pH 6.<sup>80</sup> The [  $Mn_{TCS}^{3+}$ ] ratio is close to one, which suggests that the majority of the Mn(III) cations are formed by comproportionation of a  $Mn_{TCS}^{2+}$  and  $Mn_{Layer}^{4+}$ , yielding discrete  $Mn_{TCS}^{3+}$  -  $Mn_{Layer}^{3+}$  pairs in the structure. These pairs are metastable, <sup>83</sup> and the comproportionation reaction is reverted in the presence of Co(II) ( $\Delta G = -6.7$  kcal/mol), reaction 5, Table 1). DFT calculations show that  $Mn_{TCS}^{2+}$  is also replaced with Ni(II) ( $\Delta G = -6.3$  kcal/mol), Cu(II) ( $\Delta G = -6.9$  kcal/mol), and Zn(II) ( $\Delta G = -7.5$  kcal/mol). These metals inhibit formation of Mn(III) through comproportionation. <sup>136,137</sup>

Aqueous Mn(II) may also result from the disproportionation of two  $Mn_{Layer}^{3+}$ . According to the proposed oxidation mechanism,  $Mn_{Layer}^{3+}$  build up with time in the MnO<sub>2</sub> layer, and therefore could be subject to  $Mn_{Layer}^{3+}Mn_{Layer}^{3+} \rightarrow Mn_{Layer}^{4+} + \Box + Mn_{sol}^{2+}$  disproportionation, as occurs in the conversion of buserite (10 Å birnessite) to hexagonal birnessite. This mechanism may explain the decrease of the amount of Jahn-Teller distorted Mn(III) octahedra in Co(II)-reacted birnessite at high Co/(Co+Mn) molar ratios. Should this be the case, the disproportionation reaction would create  $Mn_{Layer}^{4+}$ -vacancy pairs capable of oxidizing new  $Co^{2+}$  ions. The reaction can be written

$$Co_{Sol}^{2+} + \square_{1} + \ Mn_{Layer}^{3+}Mn_{Layer}^{4+} \longrightarrow \ Co_{INC}^{3+}Mn_{Layer}^{3+}Mn_{Layer}^{3+} \longrightarrow \ Co_{INC}^{3+}Mn_{Layer}^{4+} + \square_{2} + \ Mn_{Sol}^{2+}Mn_{Layer}^{3+} \longrightarrow \ Co_{INC}^{3+}Mn_{Layer}^{4+} + \square_{2} + \ Mn_{Sol}^{2+}Mn_{Layer}^{3+}Mn_{Layer}^{4+} \longrightarrow \ Co_{INC}^{3+}Mn_{Layer}^{4+} + \square_{2} + \ Mn_{Sol}^{2+}Mn_{Layer}^{4+} \longrightarrow \ Co_{INC}^{3+}Mn_{Layer}^{4+} \longrightarrow \ Co_{INC}^{3+}Mn_{Layer}^{4+}Mn_{Layer}^{4+} \longrightarrow \ Co_{INC}^{3+}Mn_{Layer}^{4+} \longrightarrow \ Co_{INC}^$$

and is equivalent to the balanced reaction 1. A similar mechanism with production of new vacancy sites was proposed for the oxidation of Cr(III) to Cr(VI) by birnessite. <sup>103</sup>

Existence of an intermediate low-spin Co(II) species in the oxidation pathway on vacancy sites provides a hint for the lack of oxidation on the edge sites. Co(II) switches easily between the high- and low-spin states in inorganic complexes depending on the molecular environment. In the case of birnessite, the  $t_{2g}^{5}e_{g}^{2} \rightarrow t_{2g}^{6}e_{g}^{1}$  spin transition is likely driven by the dimension of the empty Mn(IV) layer site. This site can accommodate a Mn<sup>3+</sup> ion having a size of 0.645 Å compared to 0.53 Å for Mn<sup>4+</sup>. With a size of 0.65 Å, a low-spin Co<sup>2+</sup> ion is a better match to the dimension of the vacancy than a high-spin Co<sup>2+</sup> ion (r = 0.745 Å). The  $t_{2g}^{6}e_{g}^{1}$  electronic configuration of low-spin Co(II) enforced by steric constraints facilitates the electron transfer to the unoccupied  $e_{g}$  orbitals of Mn(IV). Steric constraints do not exist on edge sites, and high-spin Co(II) is the stable conformation (reaction 10, Figure 9c, Table 1). Therefore, the  $t_{2g}^{6}e_{g}^{1}$  electronic configuration of low-spin Co(II) appears to be key condition for the transfer of an electron between the  $e_{g}$  orbitals of Co(II) and Mn(IV), a condition which is not fulfilled on the edge sites.

Results of this study call into question the proposal by Simanova and Peña<sup>26</sup> that oxidation occurs on both the MnO<sub>2</sub> edges and layer vacancies. The number of Mn atoms measured by EXAFS spectroscopy in the nearest (CN(ES)) and next-nearest (CN(CS)) shells increased from CN(ES) =  $3.12 \pm 0.49$  and CN(CS) =  $2.71 \pm 0.43$  after 7 min reaction time to CN(ES) =  $4.13 \pm 0.62$  and CN(CS) =  $5.39 \pm 1.76$  after 12 h reaction time at low surface coverage (Co/Mn = 0.05). Because EXAFS coordination numbers are averaged over all Co atoms present and are normalized on a per Co basis, CN(ES) =  $3.12 \pm 0.49$  and CN(CS) =  $2.71 \pm 0.43$  can be interpreted as about 50% Co atoms in TCS position (CN(CS) = 6) and 50% in INC position (CN(ES) = 6) within uncertainty. CN values obtained

after 12 h reaction time are problematic. The sum of Mn neighbors (4.13 + 5.39 = 9.52) exceeded the structural maximum of 6, even after consideration of uncertainty ((4.13-0.62) + (5.39-1.76) = 7.14). In addition, the majority of the Debye-Waller parameters  $(\sigma)$ , which account for the distribution of interatomic distances, were lower for the Co-Mn shells than for the Co-O shells. This is physically unrealistic because structural order decreases (i.e.,  $\sigma$  increases) with radial distance in solid matter. Furthermore, the Co-CS distance decreased from 3.48-3.50 Å after 66 min reaction time to 3.43 Å after 12 h. The shortening was attributed to  $\text{Co}_{\text{TCS}}^{3+}\text{-Mn}_{\text{Layer}}^{3+/4+}$  atomic pairs resulting from the oxidation of Co(II) by interlayer Mn(III) with formation of a Co(III)-TCS complex and release of  $\text{Mn}_{\text{Sol}}^{2+}$  (reaction 2a). This pathway is thermodynamically unfavorable ( $\Delta G = 17.1 \text{ kcal/mol}$ ). The Co-Mn distance of 3.43 Å can be attributed instead to  $\text{Co}_{\text{INC}}^{3+}\text{-Mn}_{\text{TCS}}^{3+}$  atomic pairs formed at longer reaction times by the readsorption and comproportionation of  $\text{Mn}_{\text{Sol}}^{2+}$  on vacancy sites adjacent to  $\text{Co}_{\text{INC}}^{3+}$ . Therefore, in our opinion the DFT results do not necessarily conflict with the EXAFS data of Simanova and Peña.  $^{26}$ 

In summary, extensive quantum mechanical calculations of the oxidative scavenging of Co(II) by birnessite nanolayers led us to identify two new intermediate products. One IP is a tetrahedral  $\,{\rm Co}^{2+}_{TCS}$ complex, which facilitates the crossing of the surface oxygen layer to reach the underlying Mn(IV) cavity in reducing the effective ionic radius of Co(II). The same coordination change occurs for Ni(II). The other IP is a low-spin  $Co_{INC}^{2+}$  complex, which facilitates the electron transfer between the  $e_g$  orbitals of Co(II) and Mn(IV), producing Co(III) and Mn(III). This study also provides new insight into the origin of Mn(II) measured in solution during the oxidation reaction. DFT calculations show that the Mn<sub>TCS</sub> complex is unstable and undergoes comproportionation with a layer Mn(IV) cation, forming  $Mn_{TCS}^{3+} - Mn_{Layer}^{3+}$  pairs. In the presence of Co(II), the opposite disproportionation reaction occurs, releasing Mn(II) in solution and reforming a layer Mn(IV), which subsequently may oxidize the  $Co_{TCS}^{2+}$ complex. The layer Mn(III) cations resulting from the oxidation of Co(II) may also disproportionate, producing Mn(IV) and soluble Mn(II) and creating a new vacancy site. This site may be prone to Co(II) sorption, thereby maintaining the oxidative activity of the MnO<sub>2</sub> layer. Results of this study will help understanding the oxidation of Co(II) to Co(III) in MnO<sub>2</sub>-based batteries and catalysts. 139-148 Experimental studies are needed to test the effect of oxygen vacancies 149,150 on Co(II) oxidation and to allow consider this effect in DFT modeling.

#### ASSOCIATED CONTENT

# **Supporting Information**

The Supporting Information is available free of charge on the ACS Publications website at DOI:

Schematic pathways and Gibbs free-energies of reactions 5, 6, 7, and 8, and of complexation of Co(II) as DCS and DES complex, geometry optimized ferromagnetic and antiferromagnetic Mn(II)-DES complex and ferromagnetic Mn(II)-TES complex, and Cartesian coordinates of the  $Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow}Mn_8^{4+}V\,,\quad Co_{TCS}^{2+}Mn_9^{4+}V\,,\quad Co_{INC}^{3+}Mn_8^{4+}V\,,\quad VCo_{TCS}^{2+}Mn_9^{4+}V\,,\quad VCo_{TCS}^{2+}Mn_9^{4+}V\,,\quad VCo_{INC}^{2+}Mn_9^{4+}V\,,\quad VCo_{INC}^{2+}Mn_9^{4+}$ 

Potential energy surfaces and transition states TS1 and TS2 (AVI, AVI, AVI, AVI)

## **AUTHOR INFORMATION**

# **Corresponding Author**

E-mail: alain.manceau@univ-grenoble-alpes.fr

## **ORCID**

Alain Manceau: 0000-0003-0845-611X

Stephan N. Steinmann: 0000-0002-2777-356X

## **Notes**

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## FIGURE CAPTION

**Figure. 1.** Polyhedral representation of Co-birnessite surface complexes. (a) triple-corner-sharing (TCS) complex wherein a  $[Co(II)O_6]$  octahedron binds to the three doubly-coordinated oxygen atoms  $(O_{2Mn})$  surrounding a vacancy site of the MnO<sub>2</sub> layer. (b) Incorporation (INC) of Co(III) into a vacancy. (c) Double-edge-sharing (DES) complex wherein a  $[Co(II)O_6]$  octahedron binds to two singly- and one doubly-coordinated oxygen atoms at the layer edge. (d) Triple-edge-sharing (TES) complex wherein a  $[Co(II)O_6]$  octahedron binds to two singly- and two doubly-coordinated oxygen atoms at the layer edge. (e) Double-corner-sharing (DCS) complex wherein a  $[Co(II)O_6]$  octahedron binds to two singly-coordinated oxygen atoms at the layer edge. Mn(IV) octahedron = gray; Co(II) octahedron = cyan; Co(III) octahedron = blue; oxygen = red, proton = gray.

**Figure 2.** Polyhedral representation of the four Mn<sub>9</sub>V-type nanolayers used to model the oxidation of Co(II) on basal surfaces. (a) Nanolayer 1,  $Mn_{Layer}^{3+}Mn_8V$ . (b) Nanolayer 2,  $Mn_{TCS}^{3+}Mn_9V$ . (c) Nanolayer 3, Mn<sub>9</sub>V. (d) Nanolayer 4,  $Mn_{TCS}^{3+\downarrow}Mn_8V$ . Mn(IV) octahedron = gray; Mn(III) octahedron = purple; oxygen = red, proton = gray.

**Figure 3.** Schematic pathways of reaction 1. Co(II) adsorbs as a TCS complex at a vacancy site, and is oxidized as an INC complex by an adjacent Mn(III) in the layer. Reduced Mn(II) forms a TCS complex in pathway 1a, and is released to solution in pathway 1b. Mn(IV) octahedron = gray; Mn(III) octahedron = purple; Mn(II) octahedron = dark red; Co(II) octahedron = cyan; Co(III) octahedron = blue; oxygen = red, proton = gray.

**Figure 4.** Schematic pathways of reaction 2. Mn(III) from a Mn(III)-TCS complex reacts with a hexahydrated  $Co^{2+}$  ion (a). Co(II) is oxidized by Mn(III), which is reduced to hexahydrated  $Mn^{2+}$  (pathway 2a) (b). Afterward, Co(III) fills the vacancy site (Pathway 2b) (c). Mn(IV) octahedron = gray; Mn(III) octahedron = purple; Co(III) octahedron = blue; oxygen = red, proton = gray.

**Figure 5.** Schematic pathways of reactions 4 and 5. In reaction 4, Co(II) adsorbs as a TCS complex at a vacancy site (a), and is oxidized as an INC complex by an adjacent Mn(IV) in the layer. The two undersaturated oxygens bonded to Mn(IV), Mn(III) and Co(III) are deprotonated in pathway 4a (b), and protonated in pathway 4b (c). Reaction 5 involves the disproportionation of a  $Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow}$  pair (d) by a Co(II)-TCS complex (a), which is subsequently oxidized to a Co(III)-INC complex through reaction 4. Mn(IV) octahedron = gray; Mn(III) octahedron = purple; Co(II) octahedron = cyan; Co(III) octahedron = blue; oxygen = red, proton = gray.

**Figure 6.** Releaxed potential energy surface (PES) scan of the migration of Co(II) and Ni(II) into a deprotonated Mn(IV) vacancy site. The torsion angle is defined as a dihedral angle between Co/Ni and three Mn atoms. The two reactants (R) are an octahedral TCS complex  $(Me_{TCS}^{2+}Mn_9^{4+}V_d)$  over a fully deprotonated vacancy  $(V_d)$  and the two products (P) are an INC complex  $(Me_{INC}^{2+}Mn_9^{4+})$ . The intermediate product (IP1) is a tetrahedral complex over the vacancy site  $(^{IV}Me_{TCS}^{2+}Mn_9^{4+})$ . The oxidation and spin states of Co cannot be changed during a PES scan, and therefore the variation of the electronic energy  $(\Delta E = -11.2 \text{ kcal/mol})$  is higher than when Co(II) is oxidized to Co(III).

**Figure 7.** Energetic span model of the oxidation of Co(II) to Co(III) on deprotonated Mn(IV) vacancy sites in birnessite. Co initially forms an octahedral  $Co_{TCS}^{2+}Mn_9^{4+}V_d$  complex (reactant, R) and goes through transition state TS1 to reach the intermediate product (IP1), in which Co(II) is tetrahedrally

coordinated ( $^{IV}\text{Co}_{TCS}^{2+}\text{Mn}_9^{4+}\text{V}_d$  complex). The potential electronic energy of TS1 is  $\Delta E^{\ddagger} = 7.2$  kcal/mol higher than that of R ( $\Delta G^{\ddagger} = 4.4$  kcal/mol). IP1 is as stable as R ( $\Delta E = 1.5$  kcal/mol,  $\Delta G = -0.4$  kcal/mol, reaction 7a, Table 1). When the vacancy is protonated,  $\Delta E(IP1) = 5.5$  kcal/mol and  $\Delta G(IP1) = 4.2$  kcal/mol relative to R (reaction 7b). Then, IP1 surmounts a potential activation barrier of 3.0 kcal/mol ( $\Delta G^{\ddagger} = 2.4$  kcal/mol) to form a second intermediate product (IP2), in which Co(II) is in low-spin state ( $t_{2g}{}^6e_g{}^1$ ) and occupies the center of the vacancy ( $^{VI}\text{Co}_{INC}^{2+,LS}\text{Mn}_9^{4+}$  complex). IP2 is also as stable as R ( $\Delta E = 1.7$  kcal/mol,  $\Delta G = 0.1$  kcal/mol , reaction 8). Finally, the  $e_g$  electron of Co(II) is transferred to a Mn<sup>4+</sup> ion to form the reaction product (P,  $\text{Co}_{INC}^{3+}\text{Mn}_{Layer}^{3+}\text{Mn}_8^{4+}$  complex,  $\Delta E = -16.4$  kcal/mol and  $\Delta G = -17.6$  kcal/mol relative to R, reaction 6, Figure S2). When the vacancy of the reactant is protonated, protons are likely released in the course of the migration of the  $^{IV}\text{Co}^{2+}$  ion into the empty Mn<sup>4+</sup> site. Mn(IV) octahedron = gray; Mn(III) octahedron = purple; high-spin Co(II) octahedron = cyan; low-spin Co(II) octahedron = light blue; Co(III) octahedron = blue; oxygen = red, proton = gray.

**Figure 8.** Polyhedral representation of the  $Mn^{3+}Mn_6^{4+}(a)$ , and  $Mn_7^{4+}(b)$ , nanolayers used to model the oxidation of Co(II) on lateral surfaces. Mn(IV) octahedron = gray; Mn(III) octahedron = purple; oxygen = red, proton = gray.

**Figure 9.** Schematic pathways of reactions 3, 9 and 10. In reaction 3, Co(II) adsorbs as a DCS complex on a nanolayer edge, is oxidized by an adjacent Mn(III) in the layer and replaces reduced Mn(II), which is released to solution (a). In reaction 9, a Co(II)-DCS complex is oxidized by a an adjacent Mn(IV) in the layer and forms a Co(III)-DES complex (b). In reaction 10, Co(II) adsorbs as a DES complex and is oxidzed to a Co(III)-DES complex by an adjacent Mn(IV) in the layer (c). Mn(IV) octahedron = gray; Mn(III) octahedron = purple; Co(II) octahedron = cyan; Co(III) octahedron = blue; oxygen = red, proton = gray.

**Table 1**. Gibbs free energy change ( $\Delta G$  in kcal/mol) and ratio of the concentrations of the Co(III) and Co(II) complexes at pH 4 and 7 for all oxidation reactions modeled by DFT.

ID	Nanolayer	Chemical reaction	$\Delta G$	[P]/[R], pH4	[P]/[R], pH7
1a	$Mn_{Layer}^{3+}Mn_{8}^{4+}V$	$\text{Co}_{\text{TCS}}^{2+} \text{Mn}_{\text{Layer}}^{3+} \rightarrow \text{Co}_{\text{INC}}^{3+} + \text{Mn}_{\text{TCS}}^{2+}$	4.4	5.9 × 10 <sup>-4</sup>	$5.9 \times 10^{-4}$
1b	$Mn_{Layer}^{3+}Mn_{8}^{4+}V$	$\text{Co}_{\text{TCS}}^{2+} \text{Mn}_{\text{Layer}}^{3+} + 2\text{H}^{+} + 3\text{H}_{2}\text{O} \rightarrow \text{Co}_{\text{INC}}^{3+} + \text{Mn}_{\text{sol}}^{2+}$	4.2	$8.3 \times 10^{-12}$	$8.3 \times 10^{-18}$
2a	$Mn_{TCS}^{3+}Mn_9^{4+}V$	$\operatorname{Co}^{2+}_{\operatorname{sol}} + \operatorname{Mn}^{3+}_{\operatorname{TCS}} \to \operatorname{Co}^{3+}_{\operatorname{TCS}} + \operatorname{Mn}^{2+}_{\operatorname{sol}}$	17.1	$2.9 \times 10^{-13}$	$2.9 \times 10^{-13}$
2b	$\mathrm{Mn_{TCS}^{3+}Mn_9^{4+}V}$	$Co_{TCS}^{3+} \rightarrow Co_{INC}^{3+} + 2H^{+} + 3H_{2}O$	-17.7	$9.4 \times 10^{20}$	$9.4 \times 10^{26}$
3	$\mathrm{Mn}_{\mathrm{Layer}}^{3+}\mathrm{Mn}_{6}^{4+}$	$Co_{DCS}^{2+}Mn_{Layer}^{3+} + 3H_2O \rightarrow Co_{TES}^{3+} + Mn_{sol}^{2+}$	34.1	$1.0 \times 10^{-25}$	$1.0 \times 10^{-25}$
4a	$\mathrm{Mn_9^{4+}V}$	$\text{Co}_{\text{TCS}}^{2+}\text{Mn}_{\text{Layer}}^{4+} \rightarrow \text{Co}_{\text{INC}}^{3+}\text{Mn}_{\text{Layer}}^{3+} + 2\text{H}^{+} + 3\text{H}_{2}\text{O}$	16.9	$4.1 \times 10^{-5}$	40.9
4b	$\mathrm{Mn_9^{4+}V}$	$Co_{TCS}^{2+}Mn_{Layer}^{4+} \rightarrow Co_{INC}^{3+}Mn_{Layer}^{3+}2H^{+} + 3H_{2}O$	-5.2	$6.5 \times 10^{3}$	$6.5 \times 10^{3}$
5	$Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow}Mn_{8}^{4+}V$	$Co_{sol}^{2+} + Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow} \rightarrow Co_{TCS}^{2+}Mn_{Layer}^{4+} + Mn_{sol}^{2+}$	-6.7	$8.1 \times 10^{4}$	$8.1 \times 10^{4}$
5	$Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow}Mn_{22}^{4+}V$	$Co_{sol}^{2+} + Mn_{TCS}^{3+\downarrow}Mn_{Layer}^{3+\uparrow} \rightarrow Co_{TCS}^{2+}Mn_{Layer}^{4+} + Mn_{sol}^{2+}$	-6.3	$4.1 \times 10^{4}$	$4.1 \times 10^4$
6	$\mathrm{Mn_9^{4+}V_d}$	$Co_{TCS}^{2+}Mn_{Layer}^{4+} \rightarrow Co_{INC}^{3+}Mn_{Layer}^{3+}$	-17.6	$8.0 \times 10^{12}$	$8.0 \times 10^{12}$
7a	$Mn_9^{4+}V_d$	$\text{Co}_{\text{TCS}}^{2+} \text{Mn}_{\text{Layer}}^{4+} \rightarrow {}^{\text{IV}} \text{Co}_{\text{TCS}}^{2+} \text{Mn}_{\text{Layer}}^{4+}$	-0.4	2.0	2.0
7b	$\mathrm{Mn_9^{4+}V}$	$\text{Co}_{\text{TCS}}^{2+} \text{Mn}_{\text{Layer}}^{4+} \rightarrow {}^{\text{IV}} \text{Co}_{\text{TCS}}^{2+} \text{Mn}_{\text{Layer}}^{4+}$	4.2	$8.3 \times 10^{-4}$	$8.3 \times 10^{-4}$
8	$\mathrm{Mn_9^{4+}V_d}$	$\text{Co}_{\text{TCS}}^{2+}\text{Mn}_{\text{Layer}}^{4+} \rightarrow \text{Co}_{\text{INC}}^{2+,\text{LS}}\text{Mn}_{\text{Layer}}^{4+}$	0.1	1.0	1.0
9	Mn <sub>7</sub> <sup>4+</sup>	$\text{Co}_{\text{DCS}}^{2+}\text{Mn}_{\text{Layer}}^{4+} \rightarrow \text{Co}_{\text{DES}}^{3+}\text{Mn}_{\text{Layer}}^{3+} + \text{H}^{+} + 3\text{H}_{2}\text{O}$	19.5	$5.1 \times 10^{-11}$	$5.1 \times 10^{-8}$
10	Mn <sub>7</sub> <sup>4+</sup>	$\text{Co}_{\text{DES}}^{2+} \text{Mn}_{\text{Layer}}^{4+} \rightarrow \text{Co}_{\text{DES}}^{3+} \text{Mn}_{\text{Layer}}^{3+}$	21.1	$3.4 \times 10^{-16}$	$3.4 \times 10^{-16}$

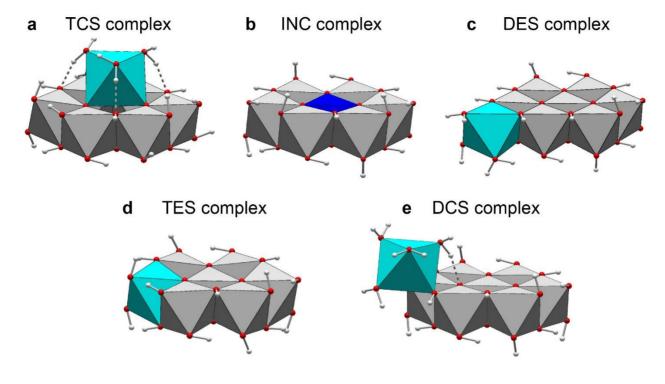


Figure 1

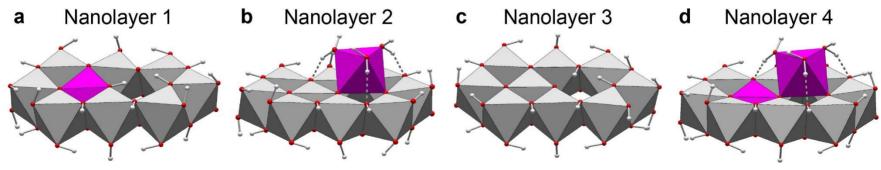


Figure 2

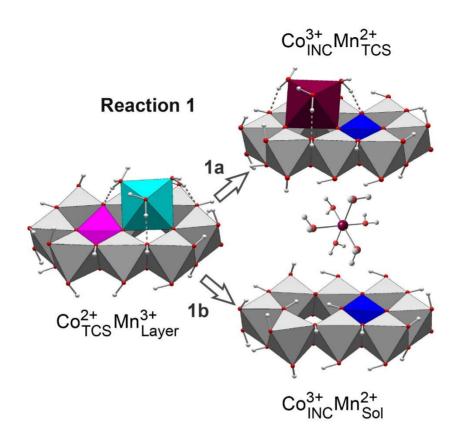
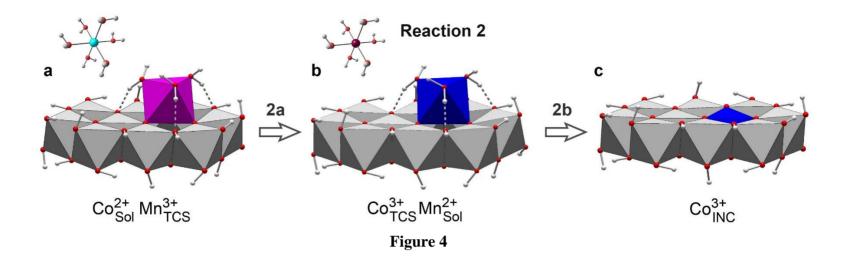


Figure 3



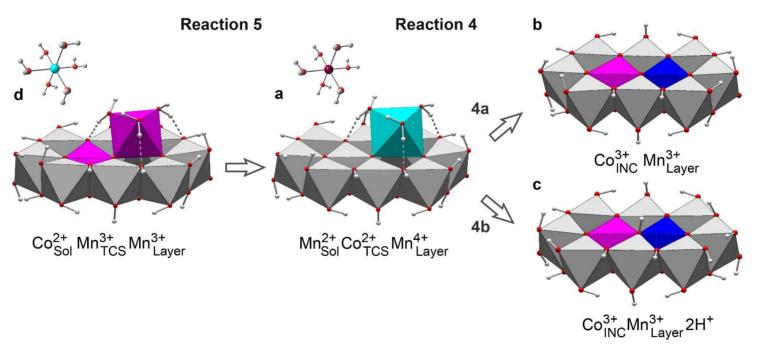
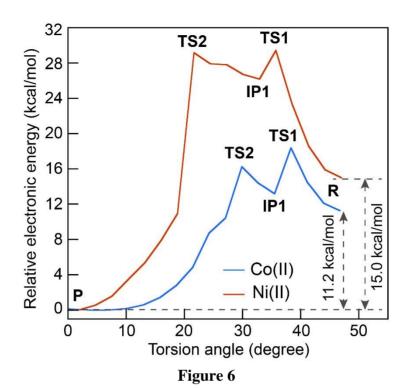


Figure 5



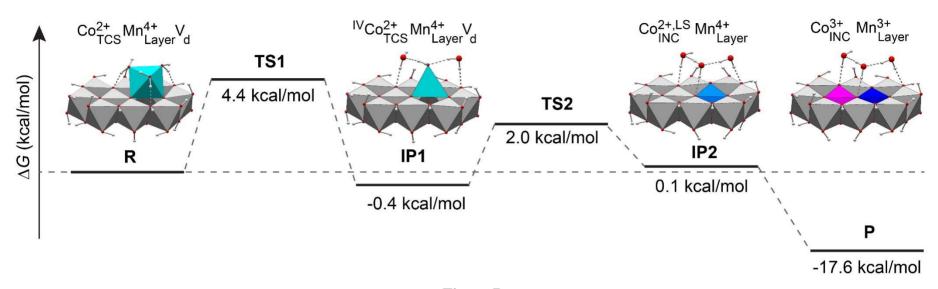
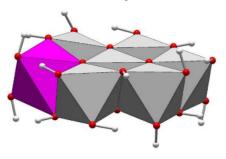


Figure 7

## a Nanolayer 1



# **b** Nanolayer 2

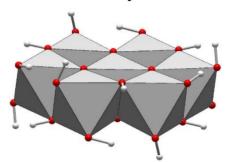
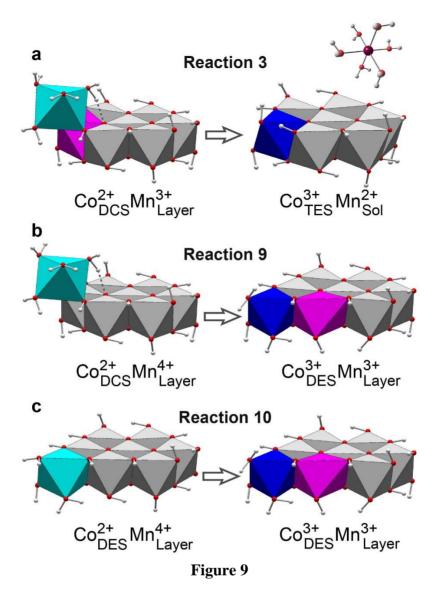


Figure 8



#### **Supplementary Information**

### Density Functional Theory Modeling of the Oxidation Mechanism of Co(II) by Birnessite

Alain Manceau\*a and Stephan N. Steinmannb

Corresponding Author:

alain.manceau@univ-grenoble-alpes.fr

#### **Supplementary Table**

**Table S1.** Total electronic charge, Mulliken atomic spin density ( $\rho$ ), and multiplicity of the complexes

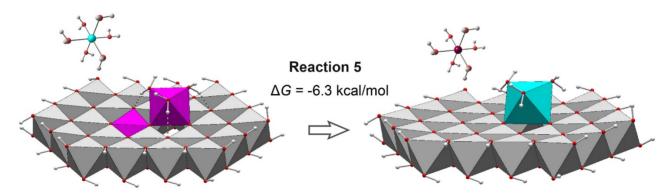
Figure	Complex	Charge	Spin density	Multiplicity
3	$Co_{TCS}^{2+}Mn_{Layer}^{3+}$	-3	Co : 2.81 Mn: 3.97	32
3	Co <sup>3+</sup> <sub>INC</sub> Mn <sup>2+</sup> <sub>TCS</sub>	-3	Co: 0.02 Mn: 4.91	30
3	Co <sup>3+</sup> <sub>INC</sub> V	-3	Co: -0.01	25
4	Mn <sup>3+</sup> <sub>TCS</sub>	-1	Mn: 4.02	32
4	Co <sub>TCS</sub> <sup>3+</sup>	-1	Co: 0.04	28
4	Co <sup>3+</sup> <sub>INC</sub>	-3	Co: 1.95	28
5	$Mn_{TCS}^{3+}Mn_{Layer}^{3+}$	-2	Mn: 3.97 Mn: 4.04	33
5, S2	$\mathrm{Co}_{\mathrm{TCS}}^{2+}\mathrm{Mn}_{\mathrm{Layer}}^{4+}$	-2	Co: 2.81	31
5	Co <sub>INC</sub> Mn <sup>3+</sup> <sub>Layer</sub>	-4	Co: 0.00 Mn: 3.93	29
5	Co <sub>INC</sub> Mn <sub>Layer</sub> <sup>3+</sup> 2H <sup>+</sup>	-2	Co: 0.00 Mn: 3.94	29
7, S2	$Co_{TCS}^{2+}Mn_{Layer}^{4+}V_{d}$	-4	Co: 2.81 31	
7, S2	$^{\rm IV}{\rm Co}^{2+}_{\rm TCS}{\rm Mn}^{4+}_{\rm Layer}{\rm V_d}$	-4	Co: 2.82	31
7, S2	$\mathrm{Co_{INC}^{2+,LS}Mn_{Layer}^{4+}}$	-4	Co: 0.96	29
7, S2	Co <sub>INC</sub> Mn <sup>3+</sup> <sub>Layer</sub>	-4	Co: 0.00 Mn: 3.95	29
9	Co <sub>DCS</sub> Mn <sub>Layer</sub> <sup>3+</sup>	-1	Co: 2.81 Mn: 4.02	26

<sup>&</sup>lt;sup>a</sup> Univ. Grenoble Alpes, CNRS, ISTerre, CS 40700, 38058 Grenoble, France

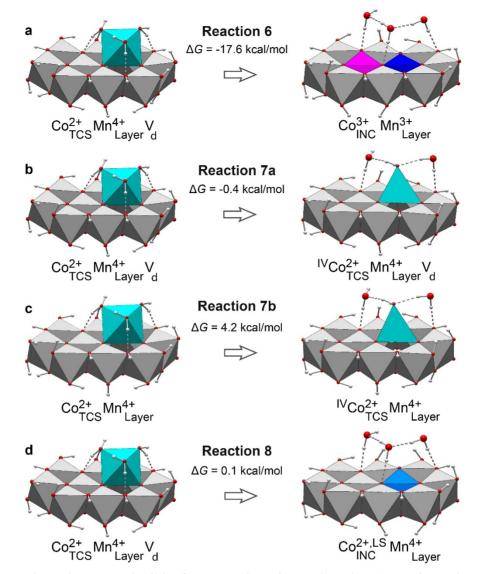
<sup>&</sup>lt;sup>b</sup> ENS de Lyon, CNRS, Laboratoire de Chimie, 69342 Lyon, France

9	Co <sub>TES</sub>	-3	Co: 0.02	19
9	$\mathrm{Co}_{\mathrm{DCS}}^{2+}\mathrm{Mn}_{\mathrm{Layer}}^{4+}$	0	Co: 2.81	25
9	Co <sub>DES</sub> Mn <sup>3+</sup> <sub>Layer</sub>	-1	Co: 0.02 Mn: 4.00	23
9	Co <sub>DES</sub> Mn <sub>Layer</sub> <sup>4+</sup>	-1	Co: 2.81	25
S2	$^{\mathrm{IV}}\mathrm{Co}_{\mathrm{TCS}}^{2+}\mathrm{Mn}_{\mathrm{Layer}}^{4+}$	-2	Co: 2.82	31

#### **Supplementary figures**



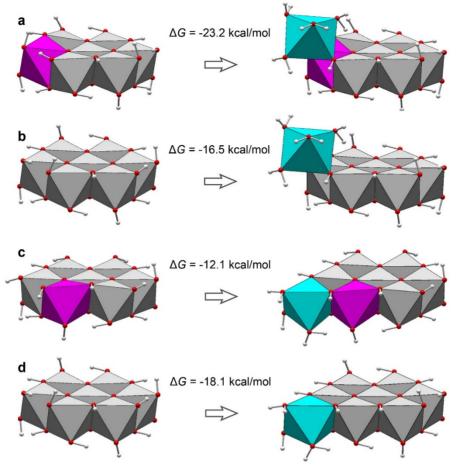
**Figure S1.** Schematic pathway and Gibbs free-energy ( $\Delta G$ ) of reaction 5 (Figure 5) for a Mn<sub>23</sub>V nanolayer. Mn(IV) octahedron = gray; Mn(III) octahedron = purple; Co(II) octahedron = cyan; Co(III) octahedron = blue; oxygen = red, proton = gray.



**Figure S2.** Schematic pathways and Gibbs free-energies of complexation ( $\Delta G$ ) of reactions 6, 7a, 7b, and 8. Mn(IV) octahedron = gray; Mn(III) octahedron = purple; high-spin Co(II) octahedron = cyan; low-spin Co(II) octahedron = light blue; Co(III) octahedron = blue; oxygen = red, proton = gray.



**Figure S3.** Geometry-optimized ferromagnetic Mn(II)-DES complex (G = -7060297.1 kcal/mol) (a), antiferromagnetic Mn(II)-DES complex (G = -7060297.4 kcal/mol) (b), and ferromagnetic Mn(II)-TES complex (c), at the layer edge. The two DES complexes are stable and the TES complex undergoes comproportionation with a layer Mn(IV). Mn(IV) octahedron = gray; Mn(III) octahedron = purple; Mn(II) octahedron = dark and bright red; oxygen = red, proton = gray.



**Figure S4.** Schematic pathways and Gibbs free-energies of complexation ( $\Delta G$ ) of hexahydrated Co(II) as DCS (a, b) and DES (c, d) complex on the Mn<sup>3+</sup>Mn<sub>6</sub><sup>4+</sup>(a, c) and Mn<sub>7</sub><sup>4+</sup>(b, d) nanolayers. Mn(IV) octahedron = gray; Mn(III) octahedron = purple; Co(II) octahedron = cyan; oxygen = red, proton = gray.

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