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Ceria nanoparticles as promoters of  $CO_2$  electroreduction on Ni/YSZ: an efficient preparation strategy and insights into the catalytic promotion mechanism

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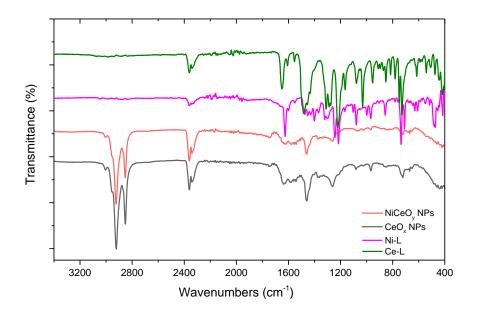
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# Supporting information 1: FTIR analysis



**Figure S 1**. FTIR spectrum of Ce (III)-L complex, Ni (II)-L complex, and as-synthesized NiCeO $_{\rm y}$  and CeO $_{\rm x}$  NPs.

# Supporting information 2: XRD patterns

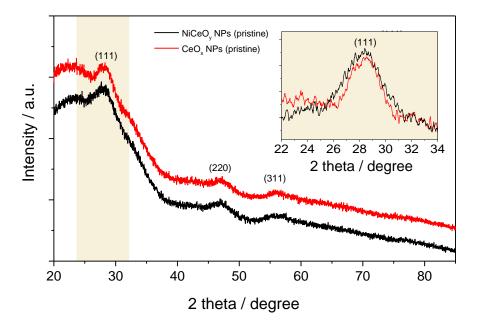
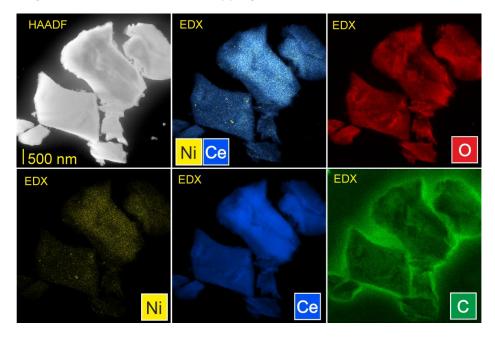


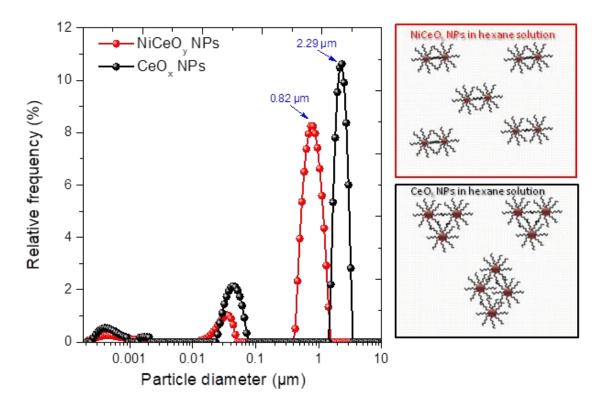
Figure S 2. XRD patterns of pristine  $CeO_x$  and  $NiCeO_y$  NPs.

# Supporting information 3: TEM-EDX mapping



**Figure S 3.** HAADF-TEM and TEM-EDS elemental mapping of NiCeO $_y$  NPs samples after calcination in  $O_2$  at  $400^{\circ}$ C.

# Supporting information 4: DLS results



**Figure S 4**(a) The particle size distributions of NiCeO<sub>y</sub> and CeO<sub>x</sub> NPs in their organic solutions in DLS measurement. (b) Schematic representation of NiCeO<sub>y</sub> and CeO<sub>x</sub> NPs arrangement in their organic solutions.

# Supporting information 5: Estimation of potential curves between NiCeO<sub>y</sub> NPs and CeO<sub>x</sub> NPs in hexane

According to DLVO theory [1–3] the total interaction forces between two particles coated by an organic protective layer  $V_{Total}$  is the sum of van der Waals attraction  $V_{vdW}$ , and total steric repulsive forces (osmotic  $V_{osm}$  and elastic  $V_{elas}$ ) [1–3].

$$V_{Total} = V_{vdW} + V_{osm} + V_{elas}$$
 (eq. 1)

The osmotic repulsion results from the energetic balance between solvent-ligand tail and tail-tail interactions, and the elastic repulsion originates from the entropy loss that occurs upon compression of the stabilizing ligands. The dispersion stability is essentially controlled by the osmotic term since it becomes effective as soon as the ligands start to overlap, while the elastic term does not contribute significantly to  $V_{Total}$  until the ligands are forced to compress.

The  $V_{vdW}$  can be calculated based on eq. 2:

$$V_{vdW} = -\frac{A}{6} \left[ \frac{2r^2}{d^2 - 4r^2} + \frac{2r^2}{d^2} + ln\left(\frac{d^2 - 4r^2}{d^2}\right) \right]$$
 (eq. 2)

A is Hamaker constant of cerium nanoparticle in the hexane (3.5×10<sup>-20</sup>J) [4], r is nanoparticles radius, d is center-to-center separation distance between two nanoparticle cores.

According to Vincent et al. [1,2,5],  $V_{osm}$  results from the energetic balance between solvent-ligand tail and tail-tail interactions, which was estimated as following:

$$\begin{split} V_{osm} &= \frac{4\pi r k_B T}{v_{hexane}} \emptyset^2 \left(\frac{1}{2} - \chi\right) \left(l - \frac{d-2r}{2}\right)^2 \quad (l < d - 2r < 2l) \text{ (eq.3)} \\ V_{osm} &= \frac{4\pi r k_B T}{v_{hexane}} \emptyset^2 \left(\frac{1}{2} - \chi\right) \left[l^2 \left(\frac{d-2r}{2l} - \frac{1}{4} - ln \left(\frac{d-2r}{2}\right)\right)\right] \quad (d-2r < l) \text{ (eq.4)} \end{split}$$

 $k_B$  is Boltzmann's constant,  $v_{nexane}$  is the molecular volume of the solvent hexane,  $\emptyset$  is the volume fraction profile of the organic species surrounded nanoparticle (oleylamine) ( $\emptyset=1-\frac{r^3}{(r+l)^3}$ ), l is oleylamine length ~2 nm, and  $\chi$  is the Flory-Huggins interaction parameter, which is written as the follows [3]:

$$\chi = \frac{v_{hexane}}{RT} (\delta_{hexane} - \delta_{oleylamine})^2 + 0.34 \text{ (eq.5)}$$

Here, R is gas constant, T is the temperature (298 K),  $\delta_{hexane}$  (14.9MPa<sup>1/2</sup>) and  $\delta_{oleylamine}$  (16.6MPa<sup>1/2</sup>) are the solubility parameters of hexane and oleylamine, respectively.  $v_{hexane}$  is calculated to be 2.17x10<sup>-28</sup> m<sup>3</sup>. By using eq.5  $\chi$  parameter is calculated to be 0.49.

The elastic repulsion  $V_{elas}$  originates from the entropy loss that occurs upon compression of the organic species oleylamine, namely, in the range of d-2r < l.

$$V_{elas} = \frac{2\pi r k_B T l^2 \phi \rho_{oleylamine}}{M_{oleylamine}} \left\{ x ln \left[ x \left( \frac{3-x}{2} \right)^2 \right] - 6 ln \left( \frac{3-x}{2} \right) + 3(1-x) \right\} \quad (d-2r < l)$$
(eq.6)

$$x = \frac{d-2r}{l} \quad (eq.7)$$

where,  $ho_{oleylamine}$  and  $M_{oleylamine}$  represent the oleylamine density and molecular weight, respectively.

**Table S 1.** The parameters for the calculation of the potential curves of  $NiCeO_y$  and  $CeO_x$  NPs in organic solution.

	NiCeO <sub>y</sub> NPs	CeO <sub>x</sub> NPs
$r$ (nm) $^{a}$	1.91	2.41
<i>l</i> (nm)	2	2
Ø	0.88	0.84

 $<sup>^{\</sup>mathbf{a}}$ Nanoparticle radius values r are taken from XRD results of pristine cerium-based nanoparticles

Using the above values the  $V_{Total}$ ,  $V_{vdW}$ ,  $V_{osm}$ , and  $V_{elas}$  can be calculated based on the eq. 1-7 as shown in Figure S4. Actually, when ligands are forced to compress (the inter-particle distance  $d_s < l$ , with  $d_s = d - 2r$ ), according to the calculated result presented in the Figure S4, the repulsions including osmotic term and elastic term have small contribution, so their effect can be neglected.

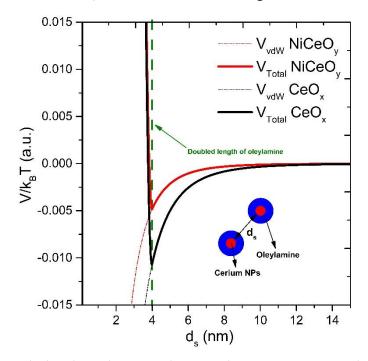


Figure S 5. The calculated total potential curves between NiCeO $_{\rm y}$  and CeO $_{\rm x}$  NPs. The inter-particle distance  $d_{\rm s}(\boldsymbol{d_s}=\boldsymbol{d-2r})$  was defined as surface-to-surface separation distance between two nanoparticles. As shown in the figure, the repulsive forces arise at the distance, where oleylamine surrounded ceria-based nanoparticles contacts each other.

# Supporting information 6: Photographs during the drying treatment

Figure S6 shows photographs of two Ni/YSZ electrodes impregnated with 50  $\mu$ L of NiCeO<sub>y</sub> or CeO<sub>x</sub> solution, taken at different drying stages. In the images the organic solution appears darker than the Ni/YSZ support which has a light grey color. Just after drop-casting, the support infiltrated by the NiCeO<sub>y</sub> shows less dark areas as compared to CeO<sub>x</sub>, indicating better spreading and pore-filling in the former case. After drying for 30 min at 80 °C, the surface of 50NiCeO<sub>y</sub>@Ni/YSZ seems homogenous in color, while in case of 50CeO<sub>x</sub>@Ni/YSZ spots with dark and light contrast remains even after 90 min at 80 °C. Since hexane evaporates fast at 80 °C (boiling point 68 °C) the homogenous surface color of NiCeO<sub>y</sub> infiltrated electrode indicates that the solution was successfully entered into the pores of the Ni/YSZ electrode. On the contrary for 50CeO<sub>x</sub>@Ni/YSZ the appearance of dark spots on the surface suggests that part of the solution remains on top of the electrode, allowing CeO<sub>x</sub> NPs to form a layer on the surface of Ni/YSZ rather than penetrating into its pores.

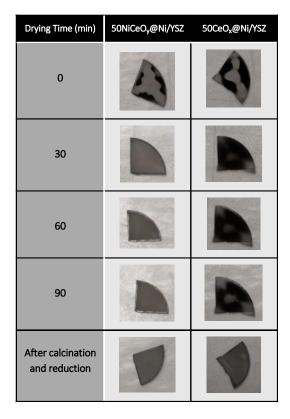


Figure S 6. Photographs of  $50NiCeO_y@Ni/YSZ$  and  $50CeO_x@Ni/YSZ$  samples during the drying treatment at 80 °C, and after calcination and reduction for 30 min in  $O_2$  and  $H_2$ .

# Supporting information 7: Laboratory XPS spectra of modified electrodes and NiCeO $_y$ NPs

The Ni 2p and Ce 3d XPS spectra of the Ni/YSZ electrode after infiltration with NiCeO $_{\rm y}$  or CeO $_{\rm x}$  solutions and subsequent drying and calcination/reduction treatments (the spectra correspond to the samples at the bottom of figure S6). The spectrum of the reduced Ni/YSZ electrode before infiltration is also included for comparison. After calcination and reduction the Ni 2p signal increases for both electrodes, but in case of 50NiCeO $_{\rm y}$ @Ni/YSZ it is notably higher (Figure S7). Since nickel signal derives principally from Ni/YSZ while Ce 3d from ceria NPs, this difference implies that less cerium remains on the surface of 50NiCeO $_{\rm y}$ @Ni/YSZ as compared to 50CeO $_{\rm z}$ @Ni/YSZ. Taking into account that the two impregnation solutions contain the same amount of NPs, this is indirect evidence that NiCeO $_{\rm y}$  NPs are efficiently infiltrated inside the pores of Ni/YSZ.

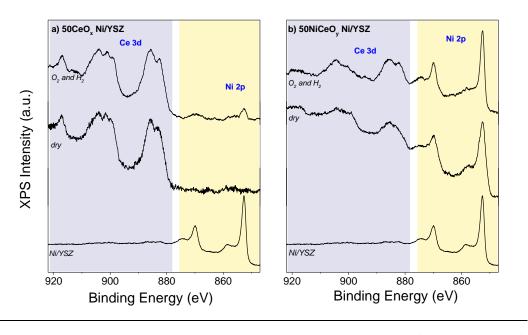


Figure S 7. The Ni 2p and Ce 3d core level XPS spectra of  $50NiCeO_y@Ni/YSZ$ ,  $50CeO_x@Ni/YSZ$  as well as pristine Ni/YSZ electrode a) after drying treatment and b) after calcination and reduction treatment in  $O_2$  at 400 °C and  $H_2$  at 550 °C.

The spectra below show the Ni 2p and Ce 3d XPS peak of NiCeO $_{\rm V}$  NPs measured by drop casting the hexane solution onto an Au foil. Evidently on the pristine sample the signal is highly attenuated due to the 2 nm oleylamine layer covering the NPs surface. After calcination the Ni 2p and Ce 3d peaks can be clearly distinguished however the intensity of Ni 2p is around 5 % of the overall. This indicates that when NiCeO $_{\rm V}$  NPs precipitate on the surface of Ni/YSZ the contribution of the Ni 2p signal deriving from NiCeO $_{\rm V}$  should be minor.

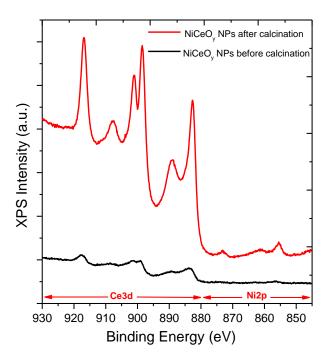


Figure S 8. Ni2p Ce3d core level XPS spectra of NiCeO<sub>v</sub> NPs before and after calcination.

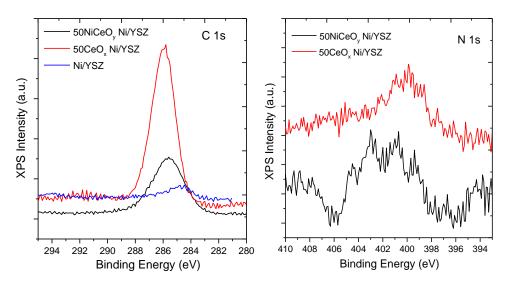


Figure S 9. C 1s and N 1s core level XPS spectra of  $50NiCeO_y@Ni/YSZ$ ,  $50CeO_x@Ni/YSZ$  and Ni/YSZ after drying treatment.

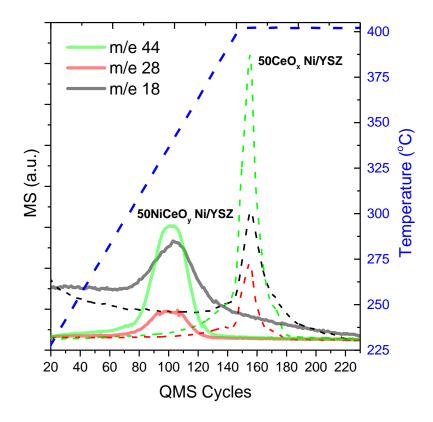
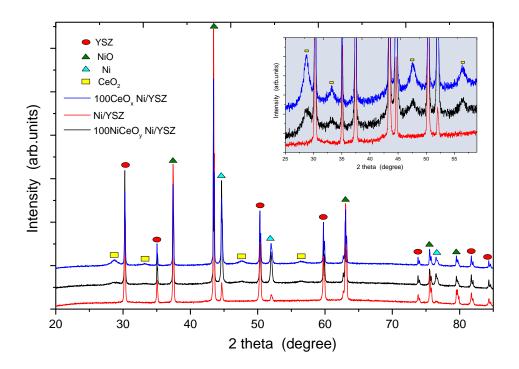


Figure S 10. Mass spectrum signal of  $50NiCeO_y$  Ni/YSZ (full lines) and  $50CeO_x$  Ni/YSZ (dashed lines) during calcination treatment in  $O_2$ .

9

### Supporting information 9: XRD of pristine and infiltrated Ni/YSZ electrodes



**Figure S 11.** XRD patterns of 100 NiCeO<sub>y</sub> Ni/YSZ, 100CeO<sub>x</sub> Ni/YSZ as well as Ni/YSZ after two infiltration/co-firing steps (100  $\mu$ l NiCeO<sub>y</sub> or CeO<sub>x</sub> solution was dropped into Ni-YSZ through two times infiltration/co-firing steps, each time for 50 $\mu$ l; Prior to the each infiltration step the electrode was reduced in H<sub>2</sub>).

**Table S 2.** Crystallite size results of NiCeO $_y$  and CeO $_x$  NPs measured directly on the 100 NiCeO $_y$  Ni/YSZ and 100 CeO $_x$  Ni/YSZ samples respectively, after two infiltration/co-firing steps.

Sample	Crystallite size (nm)
100CeO <sub>x</sub> Ni/YSZ	9.6
100NiCeO <sub>y</sub> Ni/YSZ	6.8

# Supporting information 10: Effect of the number of infiltration/co-firing steps

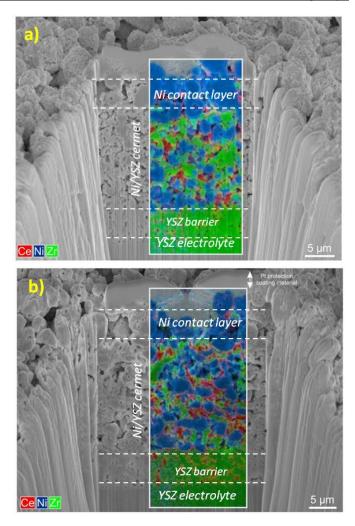
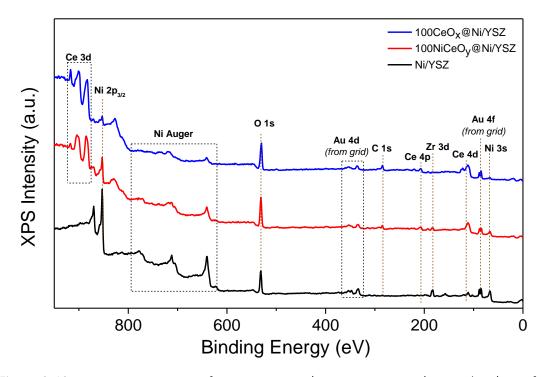
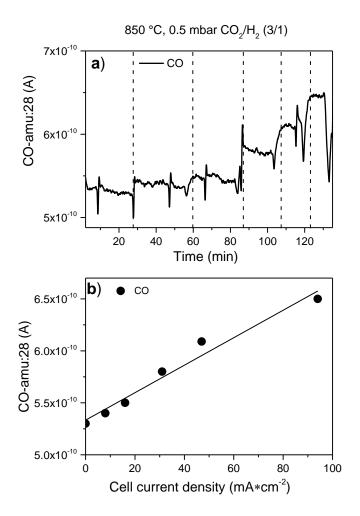


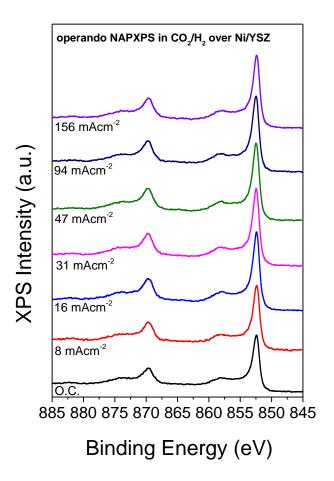
Figure S 12. Cross-sectional FIB-SEM images and the corresponding EDS elemental mapping of a)  $50 \text{NiCeO}_{\gamma} @ \text{Ni/YSZ}$  prepared in 1 infiltration/co-firing step and b)  $200 \text{NiCeO}_{\gamma} @ \text{Ni/YSZ}$  prepared in 4 infiltration/co-firing steps (4x50  $\mu$ l each). After infiltration/co-firing step(s) the samples were reduced in 1 mbar  $H_2$  at 550 °C for 30 min. The at.% of Ce as calculated from the EDS was  $7.4\pm0.6$  for  $50 \text{NiCeO}_{\gamma} @ \text{Ni/YSZ}$  and  $12.4\pm0.6$  for  $200 \text{NiCeO}_{\gamma} @ \text{Ni/YSZ}$ .



**Figure S 13.** XPS survey spectra of  $100\text{NiCeO}_y@\text{Ni/YSZ}$ ,  $100\text{CeO}_x@\text{Ni/YSZ}$  and Ni/YSZ after  $\text{CO}_2/\text{H}_2$  (96/4) electrochemical tests shown in Figure 9.



**Figure S 14.** The evolution CO gas phase signal as a function of **a**) time and **b**) the applied current density. The experiment was performed in the NAP-XPS chamber at  $850^{\circ}$ C in 0.5 mbar  $CO_2/H_2$  (75/25).



**Figure S 15**. NAP-XPS Ni 2p spectra of the pristine Ni/YSZ cathode recorded at 850  $^{\circ}$ C in CO<sub>2</sub>/H<sub>2</sub> 75/25 (0.5 mbar) under galvanostatic operation at several current densities. The spectra were recorded with hv= 1065 eV photon energy.

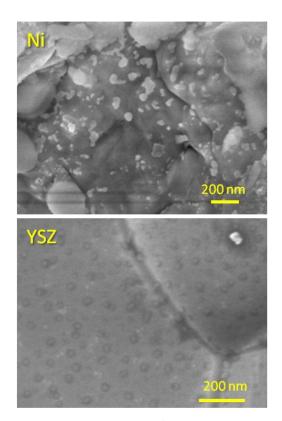


Figure S 16. SEM images of the  $100 \text{NiCeO}_{\gamma} @ \text{Ni/YSZ}$  electrodes, focusing in specific areas dominated by Ni (top) and YSZ (bottom) particles.

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