Feasibility and Limitations of High-Voltage Lithium-Iron-Manganese Spinels

Supporting information

Anna Windmüller^{1,z}, Tatiana Renzi^{2,3,4}, Hans Kungl¹, Svitlana Taranenko¹, Emmanuelle Suard⁴, François Fauth⁵, Mathieu Duttine², Chih-Long Tsai¹, Ruoheng Sun¹, Yasin Emre Durmus¹, Hermann Tempel¹, Peter Jakes¹,

Christian Masquelier^{3,6}, Rüdiger-A. Eichel^{1,7}, Laurence Croguennec^{2,6}, Helmut Ehrenberg⁸

- Institute of Energy and Climate Research (IEK-9), Forschungszentrum Jülich, 52425 Jülich, Germany
- 2 Univ. Bordeaux, CNRS, Bordeaux INP, ICMCB UMR 5026, F-33600, Pessac, France.
- 3 Laboratoire de Réactivité et de Chimie des Solides, Université de Picardie Jules Verne, NRS-UMR 7314, F-80039 Amiens Cedex 1, France
- 4 Institut Laue-Langevin, 71 avenue des Martyrs, 38042 Grenoble, France
- 5 CELLS-ALBA synchrotron, 08290 Cerdanyola del Vallès, Barcelona (Spain)
- RS2E, Réseau Français sur le Stockage Electrochimique de l'Energie, FR CNRS 3459, F-80039 Amiens Cedex 1, France.
- 7 Institut für Materialien und Prozesse für elektrochemische Energiespeicher- und wandler, RWTH Aachen University, D-52074 Aachen, Germany
- 8 Institute for Applied Materials (IAM), Karlsruhe Institute of Technology (KIT), Eggenstein-Leopoldshafen D-76344, Germany
- z Corresponding author, Dr. Anna Windmüller, E-Mail: a.windmueller@fz-juelich.de

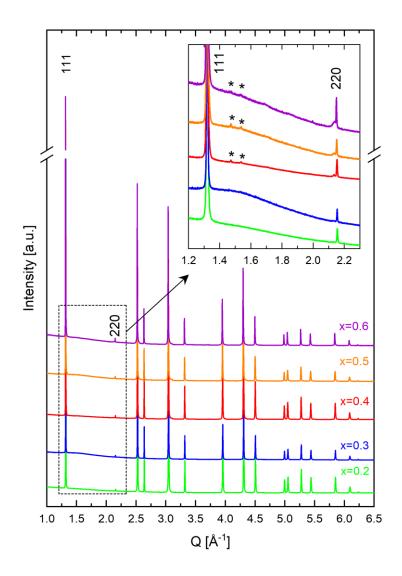


Figure S1: Synchrotron powder diffraction patterns for LiFe_xMn_{2-x}O₄ ($0.2 \le x \le 0.6$). Secondary phase reflections are highlighted with the symbol * for Li₂MnO₃

Table S1: Calculated weight fractions of phases identified from SRPD data. Considering Li₂MnO₃ fractions is critical for the refinement of the spinel parameters, as main reflections of Li₂MnO₃ overlap with the spinel main reflections. Given the small amount of free standing and low intense Li₂MnO₃ reflections the listed statistical errors might be underestimated.

x in LiFe _x Mn _{2-x} O ₄	LiFe ₅ O ₈ wt%	Li ₂ MnO ₃ wt%	Spinel wt%
0.4	< 1	1.0 ± 0.1	98.2 ± 0.2
0.5	< 1	2.0 ± 0.1	97.5 ± 0.2

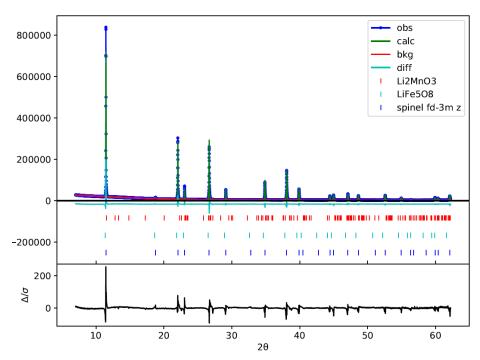


Figure S2: Rietveld analysis from SRPD: Experimental (Iobs) in blue, calculated (Icalc) in green and difference (Iobs – Icalc) in light blue for x = 0.6 to point out intensity residuals in the difference plot for the applied model.

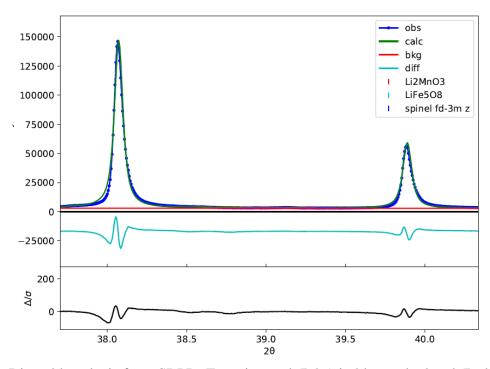


Figure S3: Rietveld analysis from SRPD: Experimental (Iobs) in blue, calculated (Icalc) in green and difference (Iobs – Icalc) in light blue for x = 0.6 for two individual reflections to point out the difficulties in describing the observed asymmetric peak shapes by the applied model.

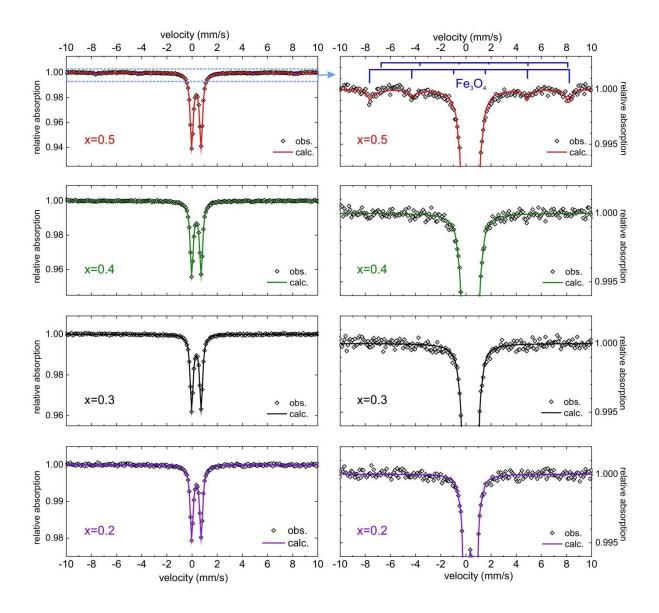


Figure S4: Room temperature ⁵⁷Fe Mössbauer spectra recorded for the series of LiFe_xMn_{2-x}O₄ spinel materials $(0.2 \le x \le 0.5)$ with a high-velocity scale. See Table 2 for the refined hyperfine parameters of the different sub-spectra.

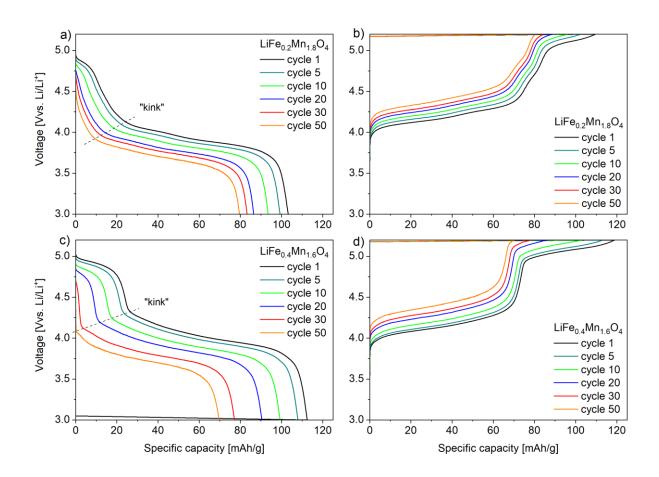


Figure S5: Charge and discharge curves of selected cycles for LiFe_xMn_{2-x}O₄ cycled at a C/5 rate with an upper cut-off voltage at 5.2 V and a lower cut-off voltage at 3.0 V vs. Li⁺/Li. a) discharge LiFe_{0.2}Mn_{1.8}O₄ b) charge LiFe_{0.2}Mn_{1.8}O₄ c) discharge LiFe_{0.4}Mn_{1.6}O₄ d) charge LiFe_{0.4}Mn_{1.6}O₄. Repeated cycling to 5.2 V vs. Li⁺/Li leads to a shift of charge "plateaus" in the high voltage range towards higher voltages and - for the discharge "plateau" - towards lower voltages, from cycle to cycle. Yet, the same behavior can be observed for the "plateaus" in the 4 V range. This behavior needs to be interpreted as an increase in overvoltages.

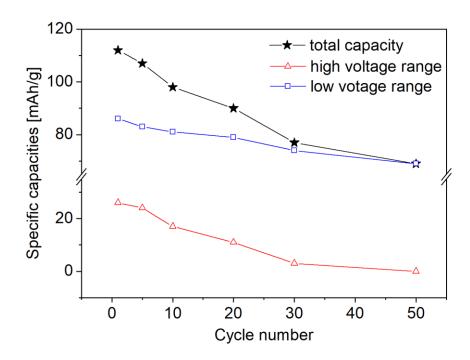


Figure S6: Specific discharge capacities for selected cycles for LiFe_{0.4}Mn_{1.6}O₄ cycled at C/5 rate with upper cut-off voltage at 5.2 V and lower at 3.0 V vs. Li/Li⁺ above and below the "kink" in the discharge curve (Figure S5), which indicates the change from the Mn^{4+/3+} reaction to the Fe^{4+/3+} reaction. Lines are drawn to guide the eye. The figure illustrates that most of the total capacity degradation is caused by degradation in the high voltage region.