



**HAL**  
open science

## Light Management in Perovskite Photovoltaic Solar Cells: A Perspective

Florian Berry, Raphaël Mermet-Lyauoz, Jose Maria Cuevas Davila, Djihad Amina Djemmah, Hai Son Nguyen, Christian Seassal, Erwann Fourmond, Céline Chevalier, Mohamed Amara, Emmanuel Drouard

► **To cite this version:**

Florian Berry, Raphaël Mermet-Lyauoz, Jose Maria Cuevas Davila, Djihad Amina Djemmah, Hai Son Nguyen, et al.. Light Management in Perovskite Photovoltaic Solar Cells: A Perspective. *Advanced Energy Materials*, 2022, 12 (20), pp.2200505. 10.1002/aenm.202200505 . hal-03708280

**HAL Id: hal-03708280**

**<https://hal.science/hal-03708280v1>**

Submitted on 10 Feb 2024

**HAL** is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

# Light Management in Perovskite Photovoltaic Solar Cells: a perspective

Florian Berry<sup>1</sup>, Raphaël Mermet-Lyaudoz<sup>1</sup>, Jose Maria Cuevas Davila<sup>1</sup>, Jihad Amina Djemmah<sup>1</sup>, Hai Son Nguyen<sup>1,2,\*</sup>, Christian Seassal<sup>1</sup>, Erwann Fourmond<sup>1</sup>, Céline Chevalier<sup>1</sup>, Mohamed Amara<sup>1</sup>, and Emmanuel Drouard<sup>1†</sup>

<sup>1</sup>*Univ Lyon, Ecole Centrale de Lyon, CNRS, INSA Lyon,  
Université Claude Bernard Lyon 1, CPE Lyon, INL, UMR5270, 69130 Ecully, France and*

<sup>2</sup>*Institut Universitaire de France (IUF)*

(Dated: March 29, 2022)

Light Management (LM) is essential for metal-halide perovskite solar cells in their race for record performance. In this review, criteria on materials, processes and photonic engineering are established such as to enhance mainly the short circuit current density, towards high energy yields. These criteria are used to analyse a large panel of solutions envisaged in the literature for single junction cells. Moreover, a perspective based on rigorous electromagnetic simulations performed on various comparable structures is proposed in order to clarify the conclusions, and to pave the way to further performance enhancement in the case of all-perovskite, two-terminal tandem cells.

KYEWORDS: light trapping, single junction perovskite cell, all perovskite tandem cells

arXiv:2203.13891v1 [physics.optics] 25 Mar 2022

---

\* hai\_son.nguyen@ec-lyon.fr

† emmanuel.drouard@ec-lyon.fr

## I. Introduction

Perovskite photovoltaic solar cells (PPSC) have reached efficiency comparable to those of the best crystalline silicon (c-Si) devices within only a few years [1], thanks to the tremendous electrical and optical properties of the metal-halide hybrid perovskites materials [2–4]. For single junction PPSC, the best performance yet demonstrated reach or even overpass a short-circuit current density  $J_{sc}$  of  $25.8 \text{ mA.cm}^{-2}$ , an open circuit voltage  $V_{oc}$  of  $1.179 \text{ V}$ , and a fill factor  $FF$  of  $0.846$ , leading to power conversion efficiency  $PCE$  of  $25.7\%$  [1]. These achievements are already quite close to the theoretically expected maximum values, especially for the  $J_{sc}$ , which is typically of  $27.83 \text{ mA.cm}^{-2}$  under AM1.5G illumination, given that the band gap energy is estimated in their studies to  $1.53 \text{ eV}$ . Furthermore, tandem PPSC made of at least two different perovskite materials appears promising but also challenging and less mature. In the case of the most common 2 terminal (2T) configuration, a net performance increase could have been recently demonstrated compared to single junction PPSC [5].

As for every optoelectronic device, light management (LM) in PPSC is of primary importance to reach the best performance. As schematized in Figure 1, the LM is expected to impact three key performance parameters. First, thanks to the LM, impinging sunlight can be efficiently collected and then absorbed leading to a high  $J_{sc}$ . Then the LM can help to enhance the  $V_{oc}$  thanks to the so-called photon recycling (PR)[6], provided a strong external luminescence [7] and considering the semiconductor exhibits weak non radiative recombinations. Finally, as recently proposed, the energy yield (EY) tends to substitute to the two previous criteria usually considered under standard test conditions (STC). For the latter, requested studies, however, remain complex since they intend to couple illumination models and angular-dependant optical, thermal and electrical models of the cell for one year [8]. However, simply considering the incidence angle in the evaluation of absorption and thus  $J_{sc}$  is a first step towards EY improvement.

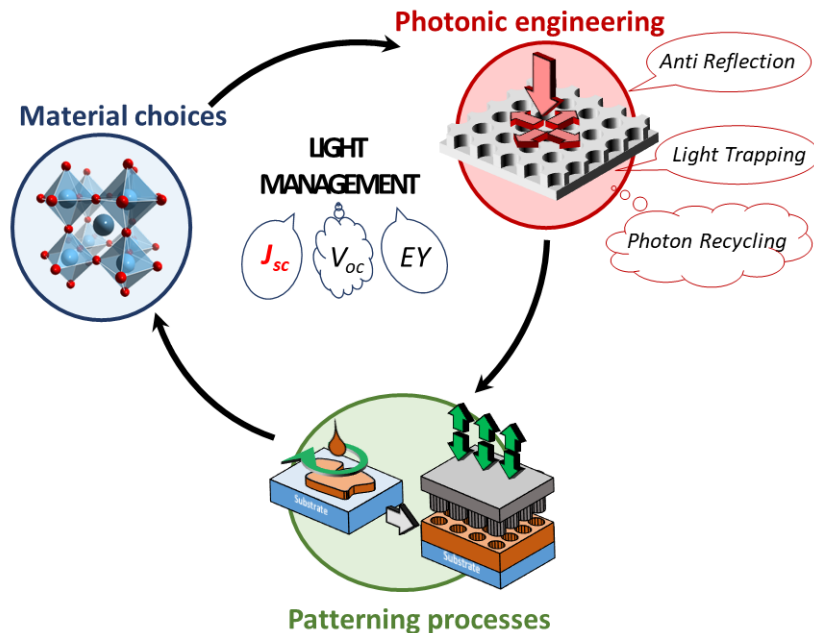


FIG. 1: The three main correlated aspects that governed Light Management.

As illustrated in Figure 1, the LM in PPSC results from three interdependent considerations: the photonic engineering (PE) at the wavelength scale, the material choices, and the nanofabrication processes. Practically, the last two directly define the PPSC architecture.

In brief, still regardless of the envisaged semiconductor, the LM can invoke several kinds of PE. Most frequently, it results in anti-reflection (AR) over the whole absorbed spectral range, leading to a larger  $J_{sc}$  as well as possibly larger EY if this the AR effect also occurs over a wide angular domain. Moreover, beyond the AR effects, light trapping (LT) mechanisms can be involved at the band edge of the perovskite material, also leading to a larger  $J_{sc}$  [9, 10], whereas enhanced PR thanks to absorption suppression just above the band gap could enhance  $V_{oc}$  [11], still provided previously mentioned conditions. This well established PE is reminded in more details in section IV B. the LM could even be extended to the infrared domain to possibly limit the parasitic sub band gap absorption and enhance the thermal radiation, so as to decrease the operating temperature [12]. Finally, for the specific case of 2T tandem PPSC, the LM is required to ensure at the same time absorption optimization and current matching between the two cells.

In the specific case of metal-halide perovskites, intrinsic properties make these materials particularly relevant for LT, compared to the standard case of c-Si solar cells. First, metal-halide perovskites are direct band gap semiconductors, the light is then more easily absorbed in a submicron thick layer, where LT and possibly PR can occur at their band edge, as detailed later. Second, their refractive indices are lower than that of c-Si, thus, AR effect is easier to obtain over the spectral range of interest. Lastly, the ability to recycle photons has been observed/shown in the specific case of lead iodide perovskite material [13].

Other key properties of metal-halide perovskites, mainly electrical and with regards to aging, have direct consequences on the typical architecture of the PPSC. The diffusion lengths of the carriers in the perovskites imply both contacts should be in the vicinity of the photocarrier generation, thereby covering both sides of the cell [14]. Additionally, one of these contacts should obviously be transparent to the sunlight. The double heterojunction architecture is widespread, leading to the use of additional intermediate layers, the electron/hole transport layers (ETL/HTL). Then, the back metallic contact, ideally made of noble metal, can thus act as a back mirror. In any case, a careful optical design is required to keep low parasitic absorption in these surrounding layers. Moreover, an effective encapsulation should be added to prevent from the degradation of the perovskite. It should be carefully chosen considering the impact of its optical properties.

Let us finally envisage the wide panel of low-cost elaboration techniques and strategies enabling the structuration of the perovskite thin film at various scales, from the sub-micron to the over-micron one. First PPSC mainly relied on a mesoporous architecture, in which the perovskite infiltrated a scaffold. Then, progress in the elaboration techniques of the perovskite in liquid or vapor phases basically results in a better microcrystallization, and thus larger diffusion lengths. Simple planar architectures have therefore replaced mesoporous ones [15–17]; thanks to these far larger diffusion lengths, current planar PPSC overpass the performances of the best mesoporous PPSC.

Following these huge modifications, the LM has evolved: if the mesoporous architecture could have been optimized for efficient LM thanks to the scaffold (typically made of  $\text{TiO}_2$ ) to enhance the  $J_{sc}$ , most of the planar cells demonstrated in the last few years (see e.g. supporting information of ref [18]) exhibit performances approaching the limits, up to about 90% of the maximum achievable  $J_{sc}$ , mostly by simply taking care of AR.

However, refined the LM still enables additional improvements of the performances towards the limits. For example, reduced optical losses thanks to LT especially at the band edge helps to gain the last missing  $\text{mA}/\text{cm}^2$ , as detailed in the following. In addition, the  $V_{oc}$  could be improved thanks to an accurate treatment of the photocarrier recombination mechanisms and the associated rates.

A few review papers discussed on various demonstrations of LM for PPSC at the time of their writing [19, 20]. In this paper, we focus on PE mainly in corrugated dielectric, without any additional metallic material inside or at the vicinity of the perovskite material. Indeed, if using metal can induce LT due to plasmonics effect, other phenomena such parasitic absorption can frequently balance this effect [21].

After a brief presentation of the main PPSC architectures and material choices that impact LM, and summarizing the PE, we propose to further analyze various published structures relying on a wave optics approach. Thus, by unraveling the various mechanisms involved, we are able to compare the various proposed strategies to deduce guidelines for further optimization. These guidelines are finally illustrated thanks to simulations of the optical properties of a few case studies.

## II. Key PPSC materials and architectures

The LM in the PPSC directly derive from their architecture, i.e. a stack of layers of various materials and thus optical indices, each layer having a thickness of the order of a few tenth of wavelengths, and with possible structurations from small to large scales compared to the wavelength. The main properties of the perovskite materials impacting LM have already been discussed. However, comparisons will appear difficult and need to be treated carefully, due to discrepancies between the structures investigated, as detailed hereafter.

### A. Patterning of the PPSC

Let us present an overview of the main patterning processes and deriving architectures that have been proposed in the literature. Due to the low diffusion lengths in the first fabricated metal-halide hybrid perovskites, first PPSC used mesoporous architectures to limit recombination. Indeed,  $\text{TiO}_2$  scaffolds were infiltrated with the perovskite material, enabling a more efficient collection of the carriers, and, in some cases, LM [22–24]. Nevertheless, as already mentioned in the introduction, most promising approaches relies nowadays in planar architectures.

In the following, layers will be considered as almost flat provided their roughness is at a sub-wavelength, nanometre scale, especially thanks to a good crystallinity in the case of the perovskite.



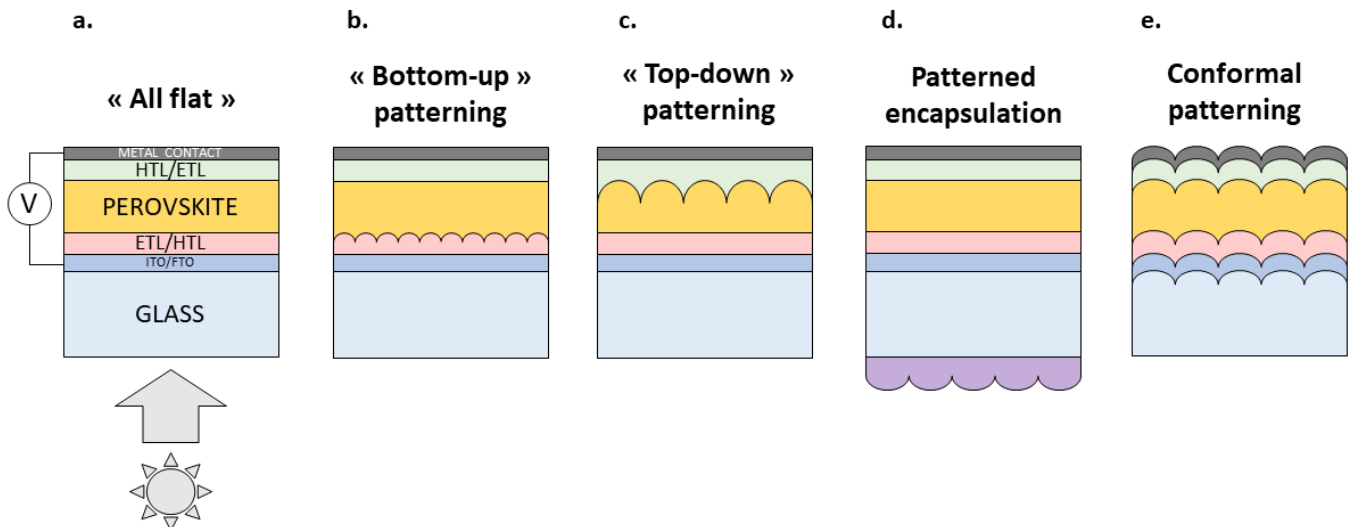


FIG. 2: Various typical architectures of complete PPSC obtained using nanofabrication. a. Almost flat (possible nanoscale roughness) b. patterned perovskite using "bottom-up" approach, i.e. structuration of the ETL/HTL, then overcoating with perovskite; c. patterned perovskite using "top down" approach, i.e. perovskite coating then structuration and overcoating. d. Flat perovskite but patterned surrounding layers and e. Patterned substrate coated with conformal stack. All envisaged patterned can be either periodic or aperiodic.

However, a 2D structuration can be introduced in the PPSC stack, notably for single junction PPSC. In this frame, the perovskite layer itself can be one of the patterned layers, or only surrounding layers are structured, and the perovskite can be considered as flat. The various envisaged technical solutions within each of the two strategies are sketched in Figure 2 b-e. Patterning of the perovskite layer can be achieved by various approaches. Such a structuration can first be generated by depositing this active layer on top of patterned layers, for example the ETL/HTL standing underneath ("bottom-up" like approach, Figure 2 b.); the pattern can be either random [18, 25, 26], or periodic [27–30]. Alternatively, the perovskite layer itself can be directly patterned *after its deposition* ("top-down" like approach, Figure 2 c.), again either in a random [31, 32] or periodic way [33, 34].

Another approach is to realize flat perovskite layers and to corrugate other adjacent layers (Figure 2 d.), such as the encapsulation layer [35–37], transport layers [38–40], possibly associated with the metal back contact [41, 42], or only the front contacts [43, 44], and finally a glass substrate covering layer [45–50].

Finally, the front substrate or the back contact layer can be patterned within a broad range of sizes, from the hundreds of nanometers to almost the millimeter; then the other layers of the stack are conformally deposited, leading to a fully structured cell [39, 51–58] (Figure 2 e.). It is noticeable that among these numerous papers, only two [44, 57] are dealing with experimental results. Indeed, spin coating of perovskite on a patterned substrate remains challenging. The former mesoporous architecture has recently been revisited under the form of opal like structuring [59].

## B. Influence of the choice of Materials

### 1. Perovskite medium

The choice of the absorbing material for PPSC results from multiple constraints. It should first exhibit an optimal band gap energy, in the range 1.1 - 1.4 eV [60], as well as optimal electrical properties and stability. Then, various deposition processes can be envisaged for a given metal halide perovskite. It may mainly lead to various grain sizes of the material during the crystallization process. For sizes of the order of the wavelength, this can affect LM.

Methylammonium lead iodide perovskite material ( $\text{CH}_3\text{NH}_3\text{PbI}_3$ , MAPI) has been widely considered in the early stages of PPSC development, thanks to its rather simple chemical composition, and despite  $E_g$  slightly above the optimal range. It is even noticeable that different  $E_g$  are reported for MAPI typically from 1.55 to 1.6 eV (see e.g. [61]). This implies a more than 7% uncertainty on the maximum achievable  $J_{sc}$ , that is already of the order of the achievable gain using LM as reported in the following. Thus, the quantitative comparison between different studies,

even implying the same material within the meaning of the chemical composition, is rather delicate. Differences of aging of the perovskite could also impact the comparison. Finally, various dispersion models can be used to fit the dielectric function [62], leading to differences that also impact the optical indices that are used for simulations.

For all these reasons, it has to be emphasized that, in the following, only the relative impact of LM could be put into perspective, but the various given performances cannot be accurately compared. Finally, if MAPI can still be used as a typical case study, other materials have been developed in the past few years. They are more stable and exhibit a slightly lower  $E_g$ , mainly thanks to the substitution of MA by FA (Formamidinium,  $(\text{NH}_2)_2\text{CH}$ ), possibly with Cs, as can be seen later.

## 2. Transport and contact layer materials

For the transport layers, mainly polymer materials are used, as well as thin layers of inorganic materials such as  $\text{TiO}_2$ . It is noticeable than materials such as those based on  $\text{C}_{60}$  fullerene, acting as ETL, exhibit significant absorption at the shortest wavelengths of the solar spectrum, whereas polymers used as HTL mainly absorb at larger wavelengths, close to and above the band gap of the perovskite. Then, there is often a compromise to be found between suitable band levels, high electrical conductivity, and barrier to possible migrations, and low parasitic absorption, even if the thicknesses of these transport layers remain limited.

## III. Photonic engineering

Let us now summarize the various kinds of PE, mostly at the wavelength scale, that can be envisaged to improve LM in both single-junction and 2T multi-junction PPSC, and their impact on geometrical requirements in the PPSC. They are sketched in Figure 3. Most of the strategies derive from those already developed for other kinds of direct or even indirect band gap material used in thin film [63].

In this frame, LM results from various light-structure interaction mechanisms. These mechanisms occur mainly either in the vertical direction (i.e. orthogonal to the stack) or in the directions of the patterns. Depending on the scale of the structure with respect to the wavelength, three main models can be used: first, at sub-wavelength scales,

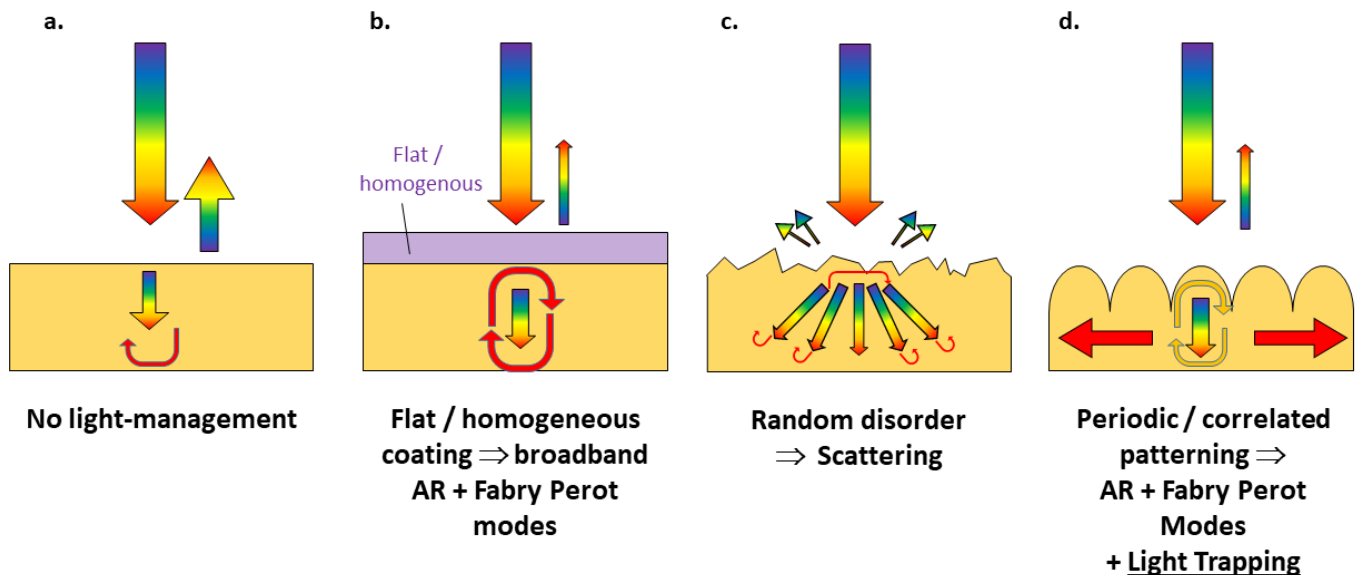


FIG. 3: Main Light Management strategies a. Without peculiar LM, reflectance can be important. b. Using a flat structure, or a patterned that can be considered as homogeneous (e.g. given its subwavelength dimensions), LM can lead to a rather broadband AR effect as well as to Fabry Perot modes that can enhance the absorption. c. Using a random texturation, LM can results from scattering, possibly Lambertian, that can also enhance the absorption. d. Using a periodic or strongly correlated patterned, in addition to AR and Fabry Perot effects, some Light Trapping can occur, drastically enhancing the absorption especially at band edge of perovskite.

an effective medium; then, at wavelength scale, wave optics, or, at larger scales, geometrical optics.

### A. Photonic engineering in multilayer unpatterned stack

It is well known that geometrical optics are mainly suitable when the dimensions are far larger than the wavelength. Thus, it cannot accurately describe the interference effects that occur within the thickness of the numerous thin films that compose the cell. In contrast, the optical indices of medium textured at subwavelength scales can be homogenized in a approximated way using Effective Medium Theories such as the Bruggeman model. Then, descriptions involving wave optics are the most rigorous, and wave optics mechanisms might also be the most promising ones.

Let us first briefly recall the two basic mechanisms able to enhance light harvesting in a flat stack, i.e. without the assistance of any patterning. In a rather high index, highly absorbing material, a properly designed stack enables to drastically reduce the impedance mismatch with the surrounding environment, whereas in a finite thickness absorbing layer, especially with a low absorption, it is rather a Fabry Perot like approach that can be used, leading to highly enhanced but narrow band absorption.

Within a typical PPSC stack, having a metallic contact that can act as a back mirror, these two approaches can be invoked. Schematically, at short wavelength, the large extinction coefficient of the perovskite enables the single pass absorption in the perovskite film, but AR effect is required. At larger wavelengths close to band gap, where the absorption drops, multiple passes are required, the second approach is well suited (Figure 3 b.).

However, the exact nature of the layers, and thus their optical indices and thicknesses are restricted by other constraints such as energy band levels, electrical resistance, risk of shunting, or even diffusion barriers for some species. Consequently, parasitic optical absorption has to be carefully studied. Moreover, broadband enhancement of the absorption under normal incidence is unlikely to be obtained using such simple architectures.

### B. Photonic engineering in multilayer, patterned stack

In addition to the continuum of propagative waves, light can be confined into the discrete set of in-plane, transverse guided modes existing in the stack. It is noticeable that the in-plane wave vector of any of these guided modes is larger than those of the free space modes. Thus these modes cannot be simply coupled from the free space in a perfectly flat stack, without any periodic or aperiodic corrugation. Then, the various kinds of in-plane structurations - having dimensions from the wavelength scale up to two orders larger - all induce diffraction. Whereas the impinging light lies around the normal incidence and has thus negligible in-plane wave vector  $k_{in||}$ , the light inside the device is diffracted. Indeed, the electromagnetic field can be expanded over a set of plane waves thanks to the in-plane spatial frequencies induced in the medium by the structuration. The reflected waves also experience diffraction accordingly. Considering the various kinds of structurations and thus of the resulting spatial frequencies, three main cases can be envisaged:

- i) A random structuration tends to induce isotropic diffraction, so Lambertian scattering. Moreover, this scattering does not depend on the wavelength within the absorbed spectral range. It can typically lead to broadband AR effect (Figure 3 c.). However the absorption enhancement remains limited. Indeed, given the already large absorption (except at the band edge) of the perovskite, and the back reflection induced by the metallic contact, it will be at most of the order of  $4n^2$  with an index lower than other materials, in particular inorganic ones [64].
- ii) A correlated disordered structuration that enhances a specific set of spatial frequencies, leads to a spectral dependent absorption enhancement. A careful choice of the sizes of the patterns is required to enhance the absorption in the desired spectral range [65].
- iii) As a particular case from the previous case, periodic, so-called photonic crystal (PC) structurations can be envisaged. Most of them consists in a square or triangular lattice a simple pattern such as pillars or holes with a vertical profile, or smoother pyramidal, or even parabolic profiles. They mainly lead to discrete modal properties. A few complex patterns arranged in a periodic way have been explored [66–70], enabling an absorption enhancement in a targeted spectral range thanks to a larger local density of modes.

In any case, the diffraction efficiency, i.e. the amount of light that is effectively diffracted, and that is thus intended to be absorbed, strongly depends on the pattern of the structuration, including the optical index contrast, as well as on the diffraction order  $p$ . Whereas, as already mentioned, for scattering, the diffraction efficiency hardly depends on the direction, for diffraction by periodic structures, it is generally larger for the first diffraction order.

In this frame, it is noticeable that a LT phenomenon rigorously occurs when the impinging light is coupled thanks to, at least, strongly correlated structure, ideally periodic patterns, having respectively a correlation length or a period  $\Lambda$ , within at least one guided mode of the stack (Figure 3 d.). According to a perturbation approach [71], in a 1D case for sake of illustration without losing generality, a phase matching condition can be written under the form:

$$\beta_m = k_{in\parallel} + p \frac{2\pi}{\Lambda} \quad (1)$$

Where  $\beta_m$  is the in-plane wave number of the  $m^{th}$  guided mode of the stack. Due to the time reversal symmetry, the light coupled into the guided mode, if not fully absorbed, can be decoupled after a certain length. These modes are thus so-called "quasi-guided mode" (QGM).

Such modes can drastically enhance the light path inside the waveguide layer. Thus, if the in-coupled QGM is mainly confined in the perovskite layer rather than in the surrounding layers, the useful absorption is enhanced. Moreover, it also depends on the absorption; more precisely, the absorption is optimized at critical coupling, i.e. when an equilibrium is reached between diffraction efficiency and absorption. Consequently, the spectral bandwidth of the LT is set since the quality factor of the mode at the optimal absorption is of the order of  $n/2\kappa$ ,  $n, \kappa$  being respectively the real and imaginary parts of the complex optical index [72].

Such a resonant LT appears promising to enhance the low absorption close to the band gap of the perovskite, provided it is not reduced elsewhere at shorter wavelengths. More precisely, within the width of the resonance, the absorption can be significantly larger than the previously mentioned broadband limit [73]. It could also be used in tandem PPSC mainly to optimize the  $J_{sc}$ , especially provided LT occurs in a guided mode specifically confined in one of the perovskite layers. As already discussed, for the other spatial frequencies that do not lead to LT, patterning can still result in a rather broadband AR effect, in addition to the one resulting from the design of the stack. Further analysis of all these effects can be found elsewhere [74, 75].

In addition to these PE approaches that contribute at first order to enhance the absorption and so the  $J_{sc}$ , other strategies could be used to enhance the  $V_{oc}$ , using a PR mechanism. In a simplified way, PR is supposed to induce at the  $V_{oc}$  operating point a high density of photons into the absorbing materials, and so to give a chance to electrons-holes pairs to recombine radiatively rather than in a non radiative way. Typically, this requires at first order to inhibit the radiation, and consequently the absorption, in the luminescence spectral domain of the semiconductor. Thus, enhancing simultaneously the  $J_{sc}$  using LT and the  $V_{oc}$  using PR is a compromise [11].

In any case, it has to be emphasized that all these effects should ideally target the perovskite layer only and not enhance the parasitic absorption in the other layers.

### C. Main geometrical requirements on patterns for Light Trapping

As mentioned previously, 2D in-plane structurations are able to diffract the impinging sunlight under a quasi-normal incidence into the large number of modes of the PPSC, possibly including the guided modes. Indeed, the architectures of both single and multi-junction PPSC reviewed in the section II A can be schematized as one or several sub-micron thick high index perovskite layers lying between two lower indices ETL and HTL and possibly also lower index layers in the multi-junction case. The whole is lying on medium index TCO on a low index glass substrate and coated on top with metal. Such structures exhibit several guided modes in both TE and TM polarization states. Let us first make the assumption of weak corrugation that doesn't change significantly the effective indices of the stack of active layers lying on a substrate. The largest effective index modes - typically 2.2 - are mainly confined in the perovskite, whereas the other modes, whose effective indices are below 1.9, can be confined in the whole stack of layers. In this frame, for normal incident light ( $k_{in\parallel} = 0$ ) and for the highest efficient coupling at order  $p = 1$ , equation (1) can be rewritten under the form:

$$\beta_m = \frac{2\pi N_{eff,m}}{\lambda} = \frac{2\pi}{\Lambda} \quad (2)$$

where  $\Lambda$  is the period, and could even be a characteristic length in correlated patterns. This 1D case can easily be extended to 2D patterns.

Thus, to be able to efficiently couple into the fundamental guided mode, which is mainly confined into the perovskite, at wavelengths corresponding to its band edge, e.g. 750 nm,  $\Lambda$  should be around 340 nm, whereas it should be larger than 400 nm for higher order guided modes. This results in a pseudo-guided mode, as discussed at the end of section III B. At shorter wavelengths, such structurations can also couple light into the various low effective index modes (including Fabry Perot like modes) of the stack, that may also result in a broadband AR effect. This typically implies that all the micron scale or even larger patterns in any of the layers of the stack are rather unlikely to couple the

impinging light into guided modes in the low absorption domain, where such a coupling is of main interest. Such patterns only induce coupling into the Fabry Perot like modes of the stack and/or diffraction at high orders with a limited efficiency.

#### IV. Examples of reported light management strategies in PPSC

We report here on some studies to illustrate concrete LM strategies. For any case, the chosen perovskite material and its thickness are of primary importance. First of all, planar PPSC as envisaged in the section II A for their architecture and in section III A for the corresponding PE are discussed, putting mainly into evidence the importance of the thicknesses on LM. Then, results on in-plane patterned PPSC are reviewed, with focuses on the involved PE.

##### A. Optimization of the thicknesses in planar PPSC

Most studies related to thickness optimization only focus on that of the perovskite layer, and report mainly on the derived useful absorption, as detailed just below. However, the whole stack has been considered in a few papers, with a view to investigate the parasitic absorption, as discussed later.

###### 1. Choice of the perovskite thickness in single junction PPSC

As already mentioned, the optimal thickness of the sole perovskite layer is chosen as a compromise between maximizing optical absorption and keeping low bulk recombination of the carriers. In addition to the analysis of a few seminal cases of single junction PPSC, we present a synthesis of several studies, mainly experimental, that report on the effect of the perovskite layer thickness of planar single junction PPSC, first and mainly made of MAPI, then of other lead-halide perovskites. The detailed performances of the representative cells are summarized in Table I and, when available, the EQE are plotted in Figure 4 in a spectral range limited from 400 to 800 nm for sake of simple comparison.

###### a. MAPI

One of the first high efficiency planar PPSC [15] which is made of an about 300 nm thick MAPICl layer exhibited a  $J_{sc}$  of  $21.5 \text{ mA.cm}^{-2}$ . In this seminal paper, M. Liu *et al.* already mentioned the possibility of an optimum thickness due to the balance between absorption and recombination. It was shown later by Y. Da *et al.* [76] that this cell suffered from optical losses that represented 11.3% of the energy losses, and especially more than 20% of the achievable  $J_{sc}$ . This was mainly related to reflectance and parasitic absorption in the FTO. This thus emphasizes the importance of minimizing these effects by a careful design of the stack.

Then, the first high-efficiency PPSC, solution processed at room temperature, proposed by D. Liu *et al.* [16] was made of an about 300 nm thick MAPI layer. A  $J_{sc}$  of  $20.4 \text{ mA.cm}^{-2}$  was obtained, so about  $5 \text{ mA.cm}^{-2}$  below the maximum achievable  $J_{sc}$  given a typical  $E_g$  of MAPI. The EQE (see Figure 4) was rather flat and remained lower than 0.8. The deficit could be due to both optical losses (too low absorption in the perovskite) and electrical losses (too high recombination).

C. Momblona *et al.* [77] conducted an experimental study on PPSC made of MAPI deposited by thermal evaporation, especially by increasing its thickness from 160 to 900 nm. If the  $J_{sc}$  increased rather monotonously with the perovskite layer thickness, it never exceeded  $20.4 \text{ mA.cm}^{-2}$ . The main parameter that decreased with increasing perovskite layer thickness was the fill factor. As a result, the cell owning optimal performance was obtained for a perovskite layer thickness around 300 nm. More precisely, whereas the  $J_{sc}$  slightly increases, both the  $V_{oc}$  and the  $FF$  significantly decreased at thicknesses larger than 300 nm. At these early ages of PPSC, the corresponding optimal PCE was 12.7%, with a  $J_{sc}$  of  $18.8 \text{ mA.cm}^{-2}$ . Then, leading further investigations on the  $J_{sc}$  deficit, it could be derived from the EQE (see Figure 4) that optical losses in almost 300 nm thick perovskite were important at both short wavelengths and from 600 to 700 nm, compared to the 900 nm thick perovskite layer that fully absorbs. The enhanced absorption in the range 700 to 750 nm is likely due to a cavity effect. Thus, even if such a thickness of about 300 nm appears to be optimal under the PCE point of view, it remains too thin to avoid optical losses. In addition, even with such thin absorbing layers, such cells still suffers from electrical losses.

In a similar way, D. Liu *et al.* [78] also then studied PPSC consisting in a ITO/ZnO/MAPI/P3HT/Ag stack, in which MAPI was obtained using thermal evaporation or solution processing. Again, in spite of a larger absorption and limited short-circuit for thicker films, a limited diffusion length led to an optimum thickness of the MAPI layer of about 330 nm for the solution processed PPSC. It is noticeable that the corresponding EQE (see Figure 4) was rather low and flat. This could be related to scattering due to sub-micron crystallinity that mitigates cavity effects



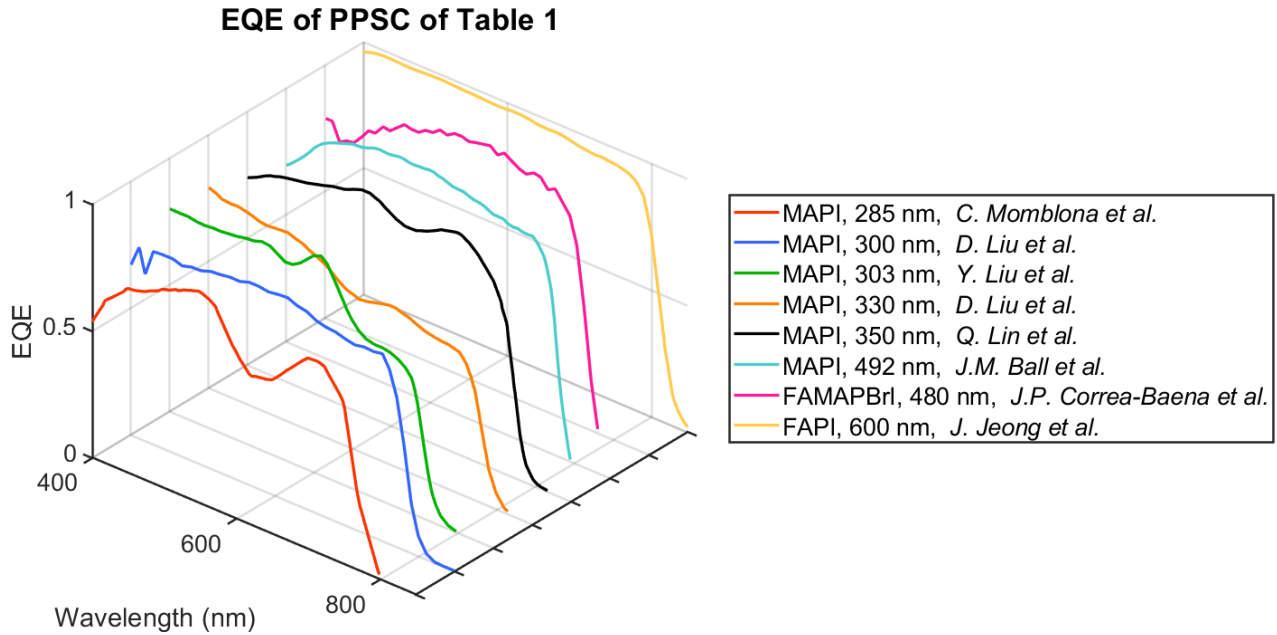


FIG. 4: EQE of various PPSC also reported in Table I [16, 77–83].

at larger wavelengths.

More recently, Y. Liu *et al.* [79], still using MAPI, similarly concluded that there were more power loss in PPSC when the thickness of the MAPI layer was either less or more than about 300 nm. They noticed that the grain size increased with the thickness, in favor of a larger thickness. However, it was noticeable that, at wavelengths larger than 650 nm, the EQE of the 303 nm thick MAPI PPSC (see Figure 4) was lower than the one of the 564 nm thick MAPI PPSC. This is related to too low absorption in this spectral range.

#### b. Other lead-halide perovskites

As an alternative, J.P. Correa-Baena *et al.* [80] used multi-cations perovskites materials; these were recently under the spotlight, especially considering their better stability [84]. Using  $\text{FA}_{0.83}\text{MA}_{0.17}\text{Pb}(\text{I}_{0.83}\text{Br}_{0.17})_3$  together with mesoporous  $\text{TiO}_2$ , the thickness that maximises the PCE was found to be at least 480 nm. The corresponding  $J_{sc}$  was almost  $24.4 \text{ mA}\cdot\text{cm}^{-2}$ . Considering the band gap of this material, reported to be about 1.63 eV [85], this is already the maximum achievable value. The EQE (see Figure 4) is noticeably high (also thanks to the an IQE of almost 100%) and flat at large wavelength. This could be related to the mesoporous  $\text{TiO}_2$  that induces some scattering effect.

Unlike the previous papers, M. Rai *et al.* [86] focused on the  $V_{oc}$  deficit related to non-radiative recombination as a function of the thickness, expressed in terms of molar concentration of the precursor solution. This time, the studied perovskite was  $\text{Cs}_{0.2}\text{FA}_{0.8}\text{Pb}(\text{I}_{0.85}\text{Br}_{0.15})_3$ , whose band gap has been found to be 1.62 eV, so close to the one of MAPI. They noticed as well that the  $V_{oc}$  decreases with the thickness, whereas the  $J_{sc}$  increases. As for the previously mentioned paper, the grain size increases with the thickness; the molar concentration of the precursor solution that maximizes PCE correspond to thickness of the perovskite layer of about 400 nm, so 30% larger than the one usually obtained using MAPI. This might be due to a larger diffusion length of the photocarriers for this perovskite.

K.B. Nine *et al.* [87] simulated optically and electrically the effect of the thickness of several FACsPI layers, leading to a larger optimal thickness, of almost 600 nm, so twice the thickness usually found using MAPI. Indeed, better electrical properties of this perovskite material, especially carrier mobility, allow such a thicker layer.

#### c. Synthesis

It is noticeable that these studies tend to an optimal thickness of about 300 nm when using MAPI. This thickness could lead to state-of-the-art PCE of J. Li *et al.* [88], with noticeably high  $J_{sc}$  of  $24.1 \text{ mA}\cdot\text{cm}^{-2}$ , mainly thanks to an ETL including graphdiyne, which improves electrical properties. Then, FAPI is the perovskite used for the best up-to-date single junction PPSC [81] which performance were mentioned in the introduction, with a thickness of about 600 nm, even if, among numerous refinements of the architecture, the rough, about 50 nm thick  $\text{TiO}_2$  layer

Material	Thickness (nm)	$J_{sc}(mA.cm^{-2})$	$V_{oc}(V)$	$FF$	PCE (%)	Ref
MAPICl	330	21.5	1.07	0.67	15.4	[15]
MAPI	300	20.4	1.03	0.749	15.7	[16]
MAPI	285	18.8	1.07	0.63	12.7	[77]
MAPI (solution process)	330	17	0.94	0.62	11.8	[78]
MAPI	303	21.3	1.07	0.715	18.4	[79]
FAMAPBrI	480	24	1.14	0.75	20.8	[80]
FAPI	600	26.35	1.189	0.817	25.59	[81]
MAPI	492	21.56	nr	nr	nr	[82]
MAPI	350	21.9	1.05	0.72	16.5	[83]

TABLE I: Experimental performances of planar PPSC with optimized perovskite thickness (nr: not reported)

already may help to trap the light.

It appears that choosing a thickness smaller than the one maximizing the  $J_{sc}$  could help to keep low resistance, so high FF and PCE. Then, with such a  $J_{sc}$  around 90% of the maximum achievable  $J_{sc}$ , LM can optimize the absorption in the perovskite, especially at photon energies close to the band gap, where the absorption starts to decrease.

In any case, it is of high interest to understand the reasons why large variations of the internal perovskite absorption spectrum have been obtained for similar thicknesses of the same perovskite material (MAPI), as can be noticed through the cases 2 to 5 of Table I and the corresponding EQE plotted in Figure 4. They might be due to differences in the surroundings layers (thicknesses, indices) or in the perovskite itself, like microcrystallinity that can induce scattering. These phenomena will be described in the following, starting with the analysis of the effect of the surroundings layers in a unpatterned stack.

## 2. Single junction unpatterned PPSC stack analysis

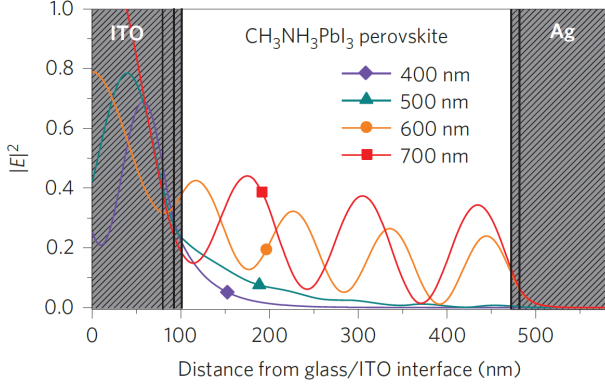
If there can be an optimal thickness for the perovskite layer, the effect of the other layers, especially on the absorption and reflectance, should also be analyzed. Here, we review a selected set of publications focusing on this issue.

J.M. Ball *et al.* [82] studied planar PPSC using an optical model based on the transfer-matrix formalism with experimentally determined complex refractive index data. They focused on a typical stack made of FTO/TiO<sub>2</sub>/MAPI/SPIRO-OMeTAD/Au. Under the assumption of  $E_g = 1.56eV$ , the detailed analysis revealed that for a calculated  $J_{sc}$  of 21.56  $mA.cm^{-2}$ , parasitic absorption induces a lack of about 1.72  $mA.cm^{-2}$ , and reflection losses corresponding to a  $J_{sc}$  decrease of 1.36  $mA.cm^{-2}$ , so a total of more than 10% of the collected current. On the other hand, the IQE losses can be estimated to another 10%. Moreover, most of the optical losses take place in the 420 nm thick FTO. This underlines the importance of taking care of all the layers of the stack.

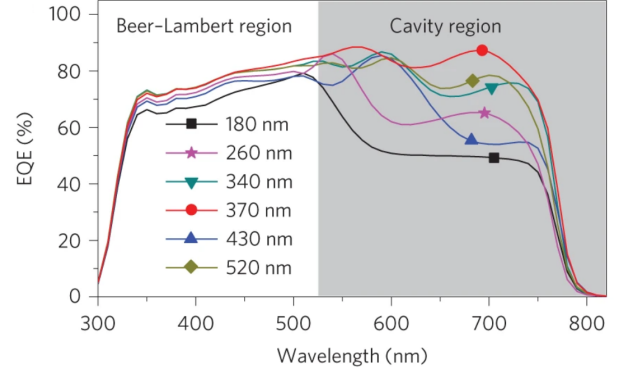
Q. Lin *et al.* [83] deeply investigated the properties of PPSC made of MAPI. Thanks to their measurement of the optical indices of the MAPI and of the other materials, they simulated the properties of PPSC, assuming an IQE of 100%. They identified the two absorption regimes of a flat PPSC, previously discussed in section III A, namely the single pass and the cavity regime. Thus, thanks to optical cavity effects at wavelengths larger than 500 nm, the noticeably high EQE of their cell (see Figure 4) appeared to be optimized for a thickness of about 370 nm (see Figure 5). The experimental data noticeably confirmed the simulation study, since a cell having a 350 nm thick MAPI layer led to the best properties,  $J_{sc} = 21.9mA.cm^{-2}$ ,  $V_{oc} = 1.05V$ ,  $FF = 0.72$  and  $PCE = 16.5\%$ . Given the resulting limited thickness of the MAPI, the obtained  $J_{sc}$  was 87% of the maximum achievable value. These remarkable results were also obtained thanks to the optimization of the p and n type interlayers that were used to optimize the heterojunctions. It was especially shown that the PCDTBT p type interlayer has to be as thin as possible ( $\sim 5nm$ ), because of its absorption, even if it enabled a nice crystallinity of the MAPI. Otherwise, given the almost 100% IQE, most of the 4 to 5  $mA.cm^{-2}$  decrease of the photocurrent (depending on the exact value of the band gap energy of the considered MAPI) corresponds to optical losses, which could be either parasitic absorption or reflectance.

M. Van Eerden *et al.* [89] analyzed the optical losses of 350 nm thick MAPI layers PPSC. Within the considered cell, most of the losses are related to reflectance (equivalent to 4.2  $mA.cm^{-2}$ ), whereas the parasitic absorption remains limited to the half, and despite a subwavelength roughness (typically from 10 to 50 nm RMS) at both ITO/TiO<sub>2</sub> and MAPI/SPIRO-OMeTAD interfaces.





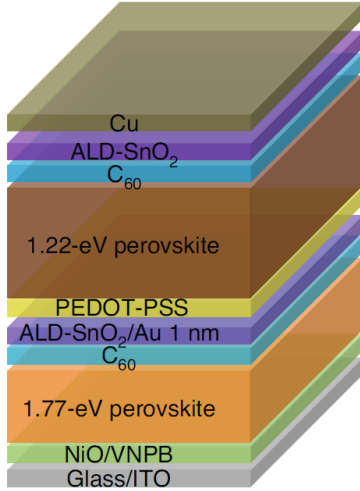
(a) Optical-field distribution for four wavelengths: for  $\lambda < 500nm$  the optical-field distribution follows the Beer–Lambert law and no optical field reaches to the back electrode as a result of the high absorption coefficient. In such cases the absorption is saturated and no optical interference occurs. For  $\lambda > 500nm$  the optical field is governed by low-finesse cavity interference.



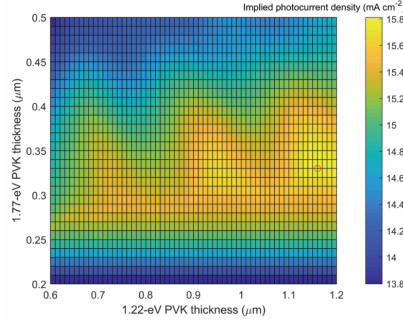
(b) EQE spectra of devices with different MAPI-layer thicknesses. For  $\lambda < 500nm$  there is minimal influence of the film thickness, but for  $\lambda > 500nm$  the EQE is strongly thickness dependent because of the optical interference (Fabry Perot like modes)

FIG. 5: PPSC properties as a function of the MAPI layer thickness [83]. Reproduced with permission

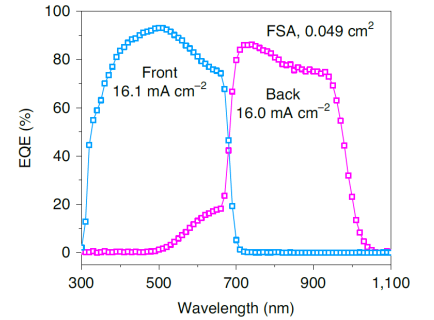
3. Analysis of unpatterned, up-to-date record 2T tandem PPSC



(a) Stack of materials



(b) Simulated  $J_{sc}$  as function of wide-band gap and narrow-band gap perovskite layer thicknesses.



(c) EQE of both subcells.

FIG. 6: Best up-to-date 2T tandem, all perovskite PPSC [90]. Reproduced with permission.

When compared to other tandem architectures, the 2T case leads to possibly limited optical losses, but such cells require an equilibrium of the current densities delivered by the two subcells. Among recent results [91], K. Xiao *et al.* [90] have deeply investigated 2T tandem PPSC (see Figure 6a) both numerically and experimentally. They have shown using optical simulations that the reachable  $J_{sc}$ , which is indeed the smallest  $J_{sc}$  delivered by each of the subcell, strongly depends on the thicknesses of each perovskite layer (see Figure 6b), provided that thicknesses of the other layers have been set taking into account both electrical and optical properties (low parasitic absorption). Indeed, it appears that the top perovskite has to be thick enough to absorb enough light, but that a too thick layer also absorbs too much light, and finally that the overall  $J_{sc}$  is limited by the bottom sub cell. The thickness of the perovskite bottom sub cell has to be large enough, more than  $1.1 \mu m$ . When fabricated on a small surface, such a cell

exhibits even a slightly larger  $J_{sc}$  than simulated, likely due to the too pessimistic predicted absorption. Even if this result remains a record at the time of its publication, its EQE (see Figure 6c) reveals some non-idealities. Indeed, in addition to the relatively low values of the EQE plateaus, there is an overlap of the absorption domains of the two perovskite materials between 500 and 700 nm. This means that some undesirable thermalization still occurs below 700 nm in the bottom sub cell.

## B. In-plane structuration for light management in single junction PPSC

At this stage, it appears that even the best reported cells exhibit an EQE that can still be improved using PE at the wavelength scale, even if most of them already benefit from the roughness of the microcrystalline perovskite.

In this section, we will highlight important criteria that should be met to ensure a fair demonstration of LM. The report on selected studies is organized as a function of the various architectures as envisaged in section II A.

### 1. General criteria for fair LM demonstration

The impact of LM on the performance of a solar cell can be demonstrated by comparing a patterned device with an unpatterned reference. In case of superficial structurations or with the addition of specific LM layers, the absorption improvement appears generally obvious compared to the same unpatterned stack, or, even more favorable, to the unpatterned stack without the additional flat LM layer. Moreover, integrating LM layer could lead to a better charge collection, by decreasing the carrier path; the specific LM effect would then be difficult to distinguish. Therefore, great care should be taken in the definition of a reference structure or device. In particular, the volume of perovskite materials should be as close as possible in both structures. Additionally, the net enhancement of EQE, efficiency or even yield value can be accurately estimated only if the unpatterned reference has been optimized first. These conditions should be met in order to discuss LM for PPSC performance optimization.

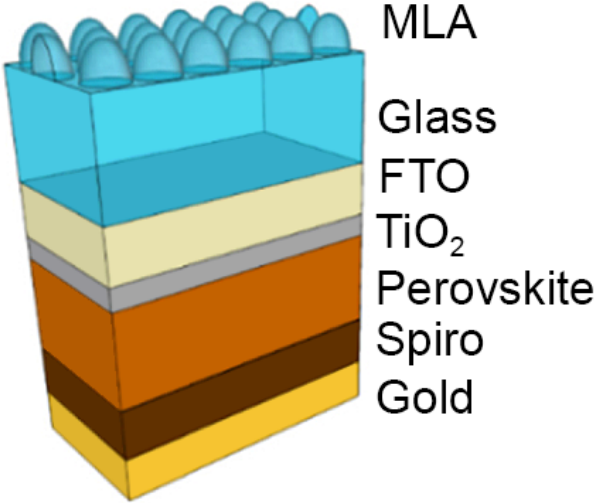
### 2. Periodic patterning for resonant LT

#### a. Superficial structuration and light management layers

A. Peer *et al.* [48] simulated planar PPSC with a stack including a 400 nm thick MAPI layer. The cell lay on top of a 700  $\mu\text{m}$  thick glass substrate. A micro lens array was then added on the top face of the glass substrate, by imprinting a polymer such as polystyrene (see Figure 7a). Each micro lens of the triangular lattice had a smooth profile close to truncated pyramids. For an aspect ratio period/height of the micro lenses close to 1, and a period of 700 nm, optimized gain of 6.3% of the  $J_{sc}$  was reported compared to the flat reference, whose EQE noticeably shown dips likely due to cavity effects in the cell (see Figure 7b). Then, according to the EQE of the cell coupled to the micro lens array, the overall resulting broadband enhancement of the absorption is due at lower wavelength to AR effect, and to LT at the band edge. The optimal period of the pattern is rather large, in agreement with a low effective index of the coupled guided mode. It means that LT occurs in the mode that is hardly guided in the perovskite, but that has a nonnegligible overlap with the pattern. This overlap remains anyway limited, as revealed by the high quality factor of the resonance. Therefore, the absorption reaches almost 1, so is close to critical coupling conditions.

J. Wei *et al.* [41] fabricated bioinspired back electrodes by imprinting the HTL made of PCBM before the conformal deposition of the Bphen/Ag back contact. Patterns were either a periodic sinusoidal 1D grating or a uniform 2D moth-eye structure (see Figure 8a), with a typical pitch of about 600 nm. It has then been experimentally shown (see Figure 8b) that the EQE of the 240 nm thick MAPICl PPSC in increased by 10% up to 40% at the band edge, mainly thanks to the 2D moth eye pattern. The overall  $J_{sc}$  increased by 14.3%, mainly due to absorption increase, but a lower series resistance was also measured for patterned cells, which led to a slight increase of the IQE. As shown by there FDTD simulations of periodic structures, this is due light diffraction into the guided modes, rather than plasmonic effects. It can be confirmed by Fourier Analysis of the top view of the moth eye pattern that a significant part of the spatial frequencies lies in the optimal range for LT (see Figure SI-S1).

H. Kim *et al.* [37] simulated nanosphere arrays of TiO<sub>2</sub> conformally coated with silica on a standard MAPI PPSC stack, with a MAPI thickness limited to 100 nm. Provided suitable spacing and nanosphere diameter, the  $J_{sc}$  could jump from 14 to 18.7  $\text{mA}\cdot\text{cm}^{-2}$ . Authors invoked the Mie scattering effect of the array to explain the enhancement, but it appears that the FEM simulation has likely been done using periodic boundary conditions, and thus that a LT effect occurs at several wavelengths, such as 705 or 790 nm for those at the band edge of MAPI. This is in line with the larger EQE enhancement close to the band edge, as can be seen in Figure 9. It remains that the overall  $J_{sc}$  is anyway limited due to the unusually low thickness.



(a) Schematic of the PPSC.

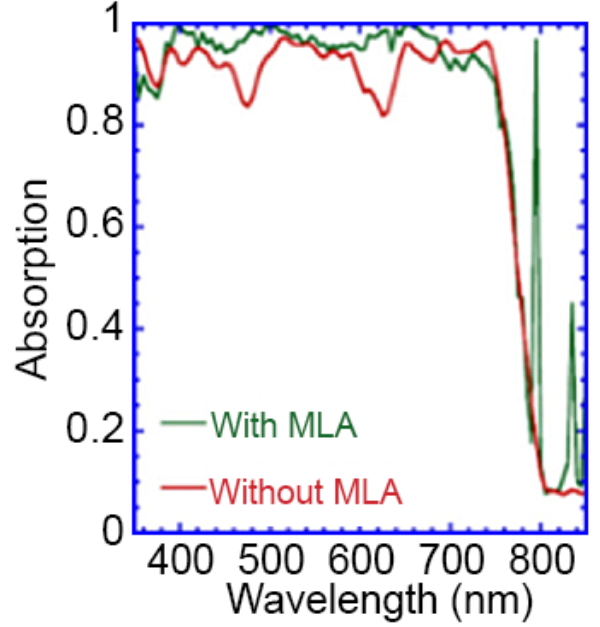
(b) Simulated absorption spectrum for microlens array of period  $700\text{nm}$  and height  $800\text{nm}$ . The absorption of flat solar cell is overlaid for comparison.

FIG. 7: PPSC with microlens array on air-glass side [48]. Reproduced with permission.

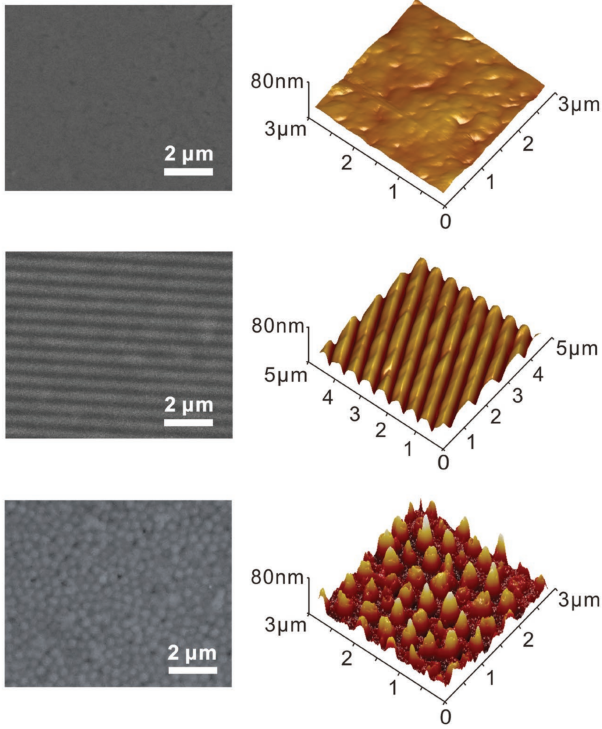
### b. Structuration of the perovskite

R. Schmager *et al.* [33] simulated a classical MAPI PPSC (see Figure 10a). Compared to the planar configuration, etching of the MAPI layer was envisaged, thanks to a square lattice of holes, filled with Spiro-OMeTAD, with various etching depths. To avoid any short circuit, an optimized partial etching of  $120\text{ nm}$  led to a 5.6% increase of the  $J_{sc}$ , compared to the initially flat  $300\text{ nm}$  thick MAPI layer, provided an equivalent volume of perovskite. It results from a LT at the band edge, but without any sharp resonance, as can be observed on the spectral response of other patterns (see Figure 10b).

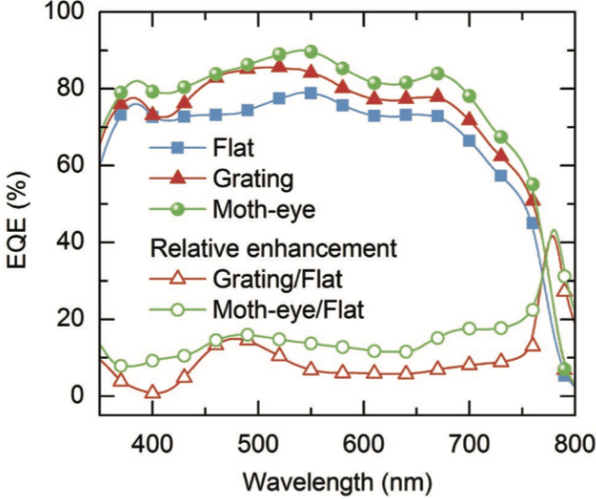
The same group [34] also realized nanoimprinted PPSC made of  $\text{Cs}_{0.1}(\text{FA}_{0.83}\text{MA}_{0.17})_{0.9}\text{Pb}(\text{I}_{0.83}\text{Br}_{0.17})_3$ . Their experimental results show a relative improvement of 2% of the PCE compared to the planar reference for the complete, gold coated solar cell. This was obtained thanks to an increase of the  $J_{sc}$  from  $19.1$  to  $19.4\text{ mA.cm}^{-2}$  and noticeably identical  $V_{oc}$  and  $FF$ . The EQE of the patterned cell was improved at wavelengths larger than  $680\text{ nm}$ . A coupling to a quasi-guided mode is observed in the perovskite band edge (see Figure 11), leading to a significant EQE enhancement (see Figure 12).

### c. Textured substrate

Q. G. Du *et al.* [52] simulated a strongly corrugated MAPI layer. It was obtained thanks to cones realized in ITO coated glass substrate with an intermediate conformal PEDOT layer lying on ITO (see Figure 13a). Thanks to optimized period and radius, a 15-17% improvement was expected in the PCE compared to the best flat cells of equivalent volume (see Figure 13b). The optimal period of about  $400\text{ nm}$  fits well the required period for LT into the guided modes, given the fact that the effective index of the modes is lower than our examples, due to the strong corrugation of the MAPI filled with low index material. AR and resonant light trapping are thus simultaneously obtained.



(a) SEM (left) and AFM (right) images of the spin-coated layer (top) of grating patterned layer (middle) and moth-eye patterned layer (bottom).



(b) External quantum efficiency (EQE) spectra of PPSC, and relative enhancement obtained by dividing the spectra of grating and moth-eye patterned devices by that of flat one.

FIG. 8: PPSC with bio inspired PCBM HTL [41]. Reproduced with permission.

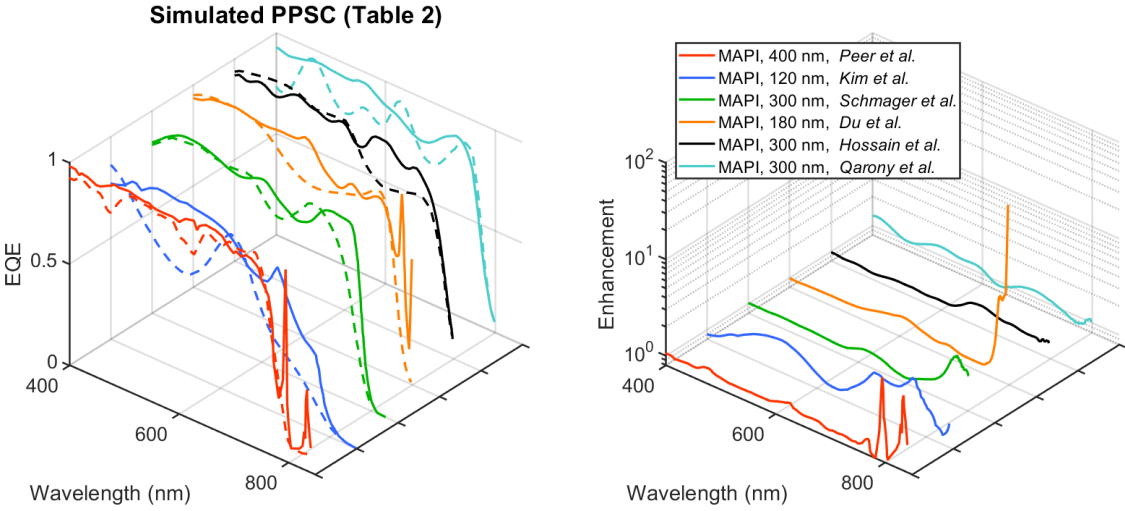
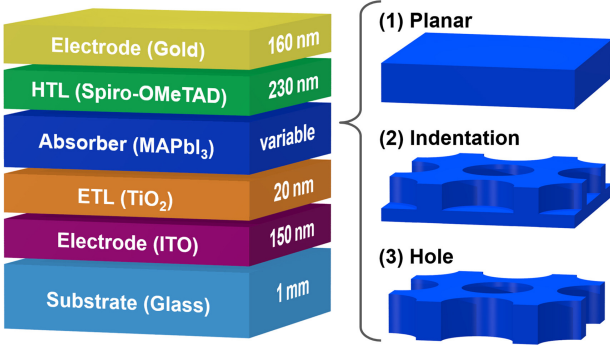
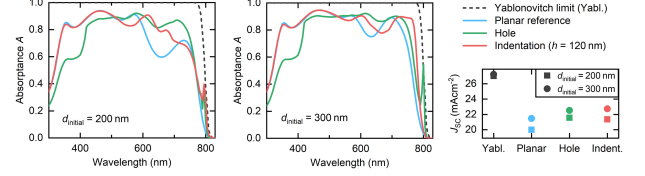


FIG. 9: EQE of the patterned PPSC (solid line) and of their planar references (dashed line) (left) and EQE enhancement ((right) of reported simulated  $J_{sc}$  enhancements of Table II, mainly resulting from Light Management, for MAPI single junction PPSC, with any kind of in-plane pattern (additional layers, structuration of the perovskite or conformal perovskite) and whatever the main identified photonic concept [33, 37, 39, 41, 43, 48, 52]



(a) Layer stack of the simulated patterned PPSC, along with associated layer thicknesses. Three configurations of the active layer are simulated (1) a planar reference, (2) a cylindrical indentation into the perovskite layer of variable depth; and (3) a hole geometry which corresponds to the maximum cylindrical indentation.



(b) Absorbance in the perovskite layer of the hole pattern and the 120 nm deep indentation compared to the corresponding planar reference and the theoretical Yablonovitch limit for two different initial perovskite layer thicknesses of (left) 200 nm and (middle) 300 nm.  $J_{sc}$  for all data are compared (right). The displayed nanophotonic patterns employ a geometrical  $ff = 0.4$  and a period of 380 nm.

FIG. 10: Various 2D in plane patterns of the MAPI layer in a PPSC [33]. Reproduced with permission

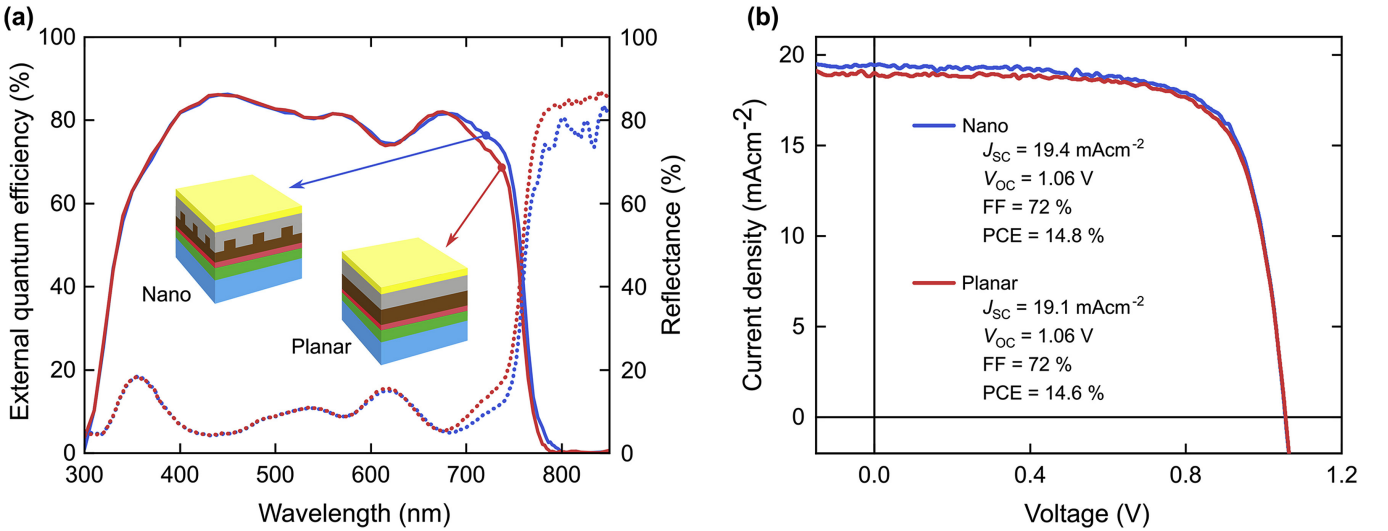


FIG. 11: External quantum efficiency and reflectance spectra of the planar and nanoimprinted perovskite solar cells. The nanoimprinted perovskite layer has a period of 480 nm. The nanoimprinted perovskite solar cell shows enhanced absorption and current generation close to the band gap. For energies below the band gap (wavelength larger than 790 nm) discrete peaks are visible in the reflectance measurement. [34]. Reproduced with permission.

### 3. Periodic structures for non resonant LM: anti-reflection and scattering

#### a. Superficial structuration and light management layers

M. I. Hossain *et al.* [43] proposed to coat the MAPBI like perovskite, having a thickness from 100 to 400 nm, with zinc oxide. More precisely, a first thick, lightly doped ZnO ensured the top contact, then it could be covered by texturations patterned either as pyramids or as non-resonant metasurfaces. The period was set to 800 nm. Accordingly to the mechanism typically occurring using such a period, no sharp resonance enhances the absorption at the band-edge, only an AR like effect occurs. Both kinds of patterns lead to simulated equivalent  $J_{sc}$  enhancements of about  $3 \text{ mA.cm}^{-2}$  whatever the thickness of the perovskite between 100 and 400 nm, so a relative enhancement roughly from 25% to 12% within the same range, but only 10% for the thickness 300 nm.

P. Tockhorn *et al.* [44] simulated and fabricated a 550 nm thick PPSC made of a mixed cation, mixed halide  $\text{Cs}_{0.05}(\text{FA}_{0.83}\text{MA}_{0.17})_{0.95}\text{Pb}(\text{I}_{0.83}\text{Br}_{0.17})$  perovskite material on various periodically patterned glass substrates, in p-i-n configuration. The top side of the glass substrates was coated with low index, NaF thin film. Compared to the



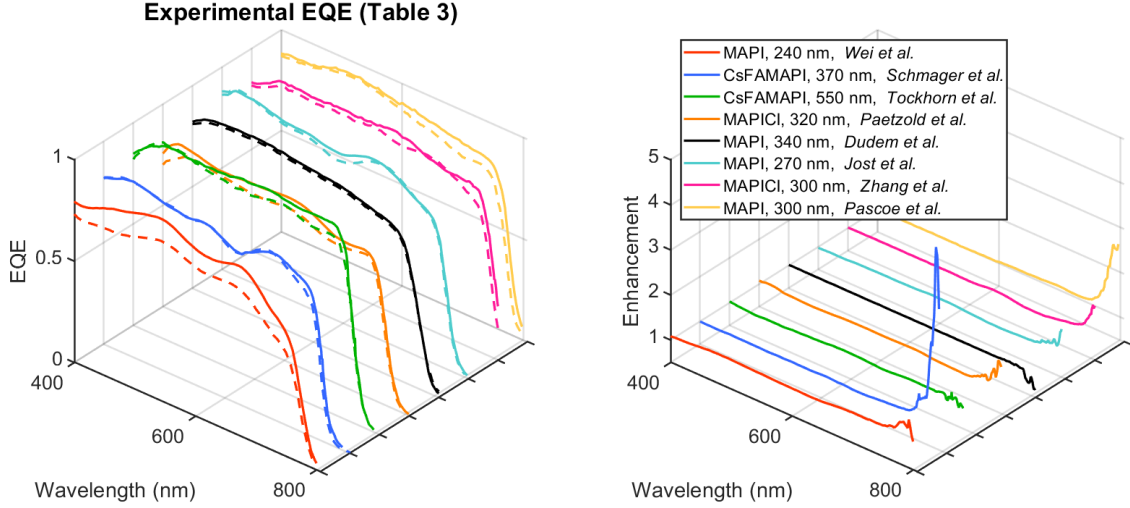


FIG. 12: EQE of the patterned PPSC (solid line) and of their planar references (dashed line) (left) and EQE enhancement ((right) of Table III, mainly resulting from Light Management, for single junction PPSC, with any kind of in-plane pattern (additional layers, structuration of the perovskite or conformal perovskite) and whatever the main identified photonic concept [26, 27, 34, 42, 44, 46, 47]

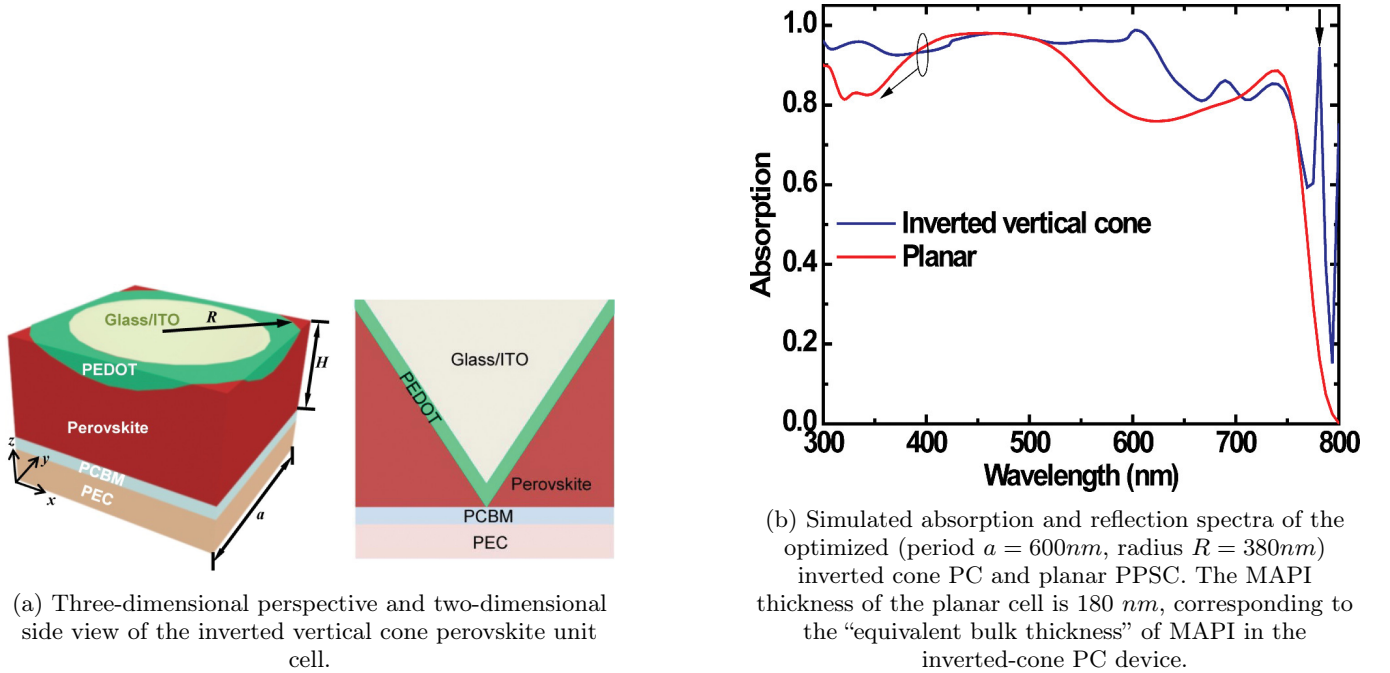


FIG. 13: MAPI PPSC patterned with inverted cones[52]. Reproduced with permission.

planar reference, patterned structures could exhibit a  $J_{sc}$  up to  $1 \text{ mA/cm}^2$  larger. This results from a broadband enhancement of the EQE, including at the band edge, mainly attributed to AR effect, as verified by simulations where the volume of perovskite has been kept constant. This is in line with the flat EQE enhancement, as can be seen in Figure 12 The resulting PCE reaches 19.7%, 1% absolute above the one of the planar reference.

*b. Structuration of the perovskite*

U. W. Paetzold *et al.* [27] proposed to pattern the front ITO electrode with a square lattice of pillars. The ETL was then corrugated, as well as the about  $320 \text{ nm}$  thick MAPICl layer that planarized the stack, before the HTL

and Al back contact deposition. They noticed increased absorption and EQE as the lattice period decreased, up to the smallest value envisaged, i.e.  $500\text{ nm}$ . This led to an increase of the  $J_{sc}$  by 5%. Broadband enhancements were observed, at wavelengths shorter than the band edge of the material, as well as a limited LT at the band edge, according to the EQE enhancement plotted in Figure 12. This is likely due to the lack of large enough spatial frequencies to couple the quasi-guided modes of the structure, given a period slightly above the optimal range. In addition, there might be a slight change in the volume of perovskite between the patterned cells compared to the flat reference.

### c. Textured substrate

W. Qarony *et al.* [39] calculated the EQE for three various configurations of PPSC using the same volume of perovskite and including moth eye periodical patterns with a period of about  $150\text{ nm}$ , typically leading to scattering. The first one had only a pattern at the top air / ZnO interface (thus is more comparable to former structures), but the two others considered a conformally patterned stack, either on a patterned Al substrate, or on a patterned NiO/ITO layer. The patterned Al substrate clearly led to a lower EQE due to additional parasitic absorption. Moreover, a slightly lower  $J_{sc}$  was obtained with the conformal stack on the flat Al substrate compared to the top patterned stack. Even this last case exhibits a low EQE enhancement (see Figure 12).

## 4. Aperiodic patterning for LM

### a. Superficial structuration and light management layers

B. Dudem *et al.* [46] proposed a multifunctional inverted micro-structured pyramidal Polydimethylsiloxane (PDMS) AR layer for enhancing the device efficiency, through AR and self cleaning. The MAPI layer was  $340\text{ nm}$  thick. The sizes of the pyramid were in the range from  $1$  to  $10\text{ }\mu\text{m}$ , therefore too large for efficient diffraction. Compared to flat PDMS, the  $J_{sc}$  increases by  $0.38\text{ mA.cm}^{-2}$  up to  $21.25\text{ mA.cm}^{-2}$ , corresponding to a limited AR effect, as confirmed by the flat EQE enhancement (see Figure 12).

Rather similarly, M. Jošt *et al.* [47] fabricated a so-called light management foil on the glass substrate of a planar  $270\text{ nm}$  thick MAPI cell, which led to a limited EQE enhancement (see Figure 12). At first, the thick glass substrate prevents from a strong overlap between any guided mode into the perovskite and the foil. Moreover, whatever the glass thickness, the lack of LT is also related to the far too low spatial frequencies resulting from the texturation. Indeed, thanks to available top view of the foil, a Fourier transform analysis reveals (see Figure SI-S2) that most of Fourier's components lie below  $7\text{ }\mu\text{m}^{-1}$ , whereas the optimum  $\beta_m$  should be in the range  $18 - 34\text{ }\mu\text{m}^{-1}$  at first order of diffraction, and even larger for other orders.

On the other side of stack, H. Zhang *et al.* [42] measured and modelled the impact of the roughness of the back mirror on the back scattering of MAPICl PPSC. The reference was made of a perovskite layer with a significant roughness after crystallization, coated with a thick Spiro layer that planarizes, then a flat gold mirror. With a thinner HTL, the gold mirror replicated the roughness of the perovskite layer leading to light back scattering. The PCE increased from 19.3% to 19.8%, mainly thanks to the  $J_{sc}$  increase from  $22.7$  to  $23.6\text{ mA.cm}^{-2}$ . It is noticeable that the obtained EQE enhancement is the largest at the band edge. This could be related to the grain size that appears to be in the range  $200 - 500\text{ nm}$ , so including the optimal range for efficient LT.

### b. Structuration of the perovskite

A. R. Pascoe *et al.* [26] proposed textured MAPI at the scale of several hundreds of nanometers thanks to gas crystallization. It enhanced the EQE at wavelengths larger than  $550\text{ nm}$  (see Figure 12), thanks to an induced so-called scattering. Accordingly, the averaged  $J_{sc}$  increased from  $21.3$  for planar references having an about  $300\text{ nm}$  thick MAPI layer to  $22.1\text{ mA.cm}^{-2}$  for patterned samples of comparable volume. As previously, the noticeably large EQE enhancement close to the band edge can be related to the typical grain size of the order of  $500\text{ nm}$ .

## 5. Synthesis

In the various previously described studies reporting on  $J_{sc}$  enhancements thanks to LM, it can be noticed that most of the possible architectures envisaged in the section II A have been considered, using all the PE described in section III. Tables II and III summarize the  $J_{sc}$  enhancements reported in the various previously described studies.

From Table II, focusing on simulations of periodic patterning of MAPI based single junction PPSC, it comes out that LT can indeed significantly increase the  $J_{sc}$  for very thin perovskite layers. The enhancement is more limited at a thickness of about  $300\text{ nm}$ . It remains that an accurate comparison of the  $J_{sc}$  enhancements is not possible, since performance enhancement strongly depends on the choice of the unpatterned reference, and especially its optimization



in terms of LM as discussed previously. However, the EQE enhancements (Figure 9) can clearly reach larger values thanks to LT than for AR.

From Table III, focusing on experimental results, using various perovskite materials, the  $J_{sc}$  enhancements are of the same order as the simulated ones, again with the same precautions as above. As for the simulated PPSC, the EQE enhancements (Figure 12) also reach larger values using patterns at the scale of the wavelength in the material. Moreover, the periodic case lead to the largest enhancement.

TABLE II: Reported simulated  $J_{sc}$  enhancements mainly resulting from Light Management, for MAPI single junction PPSC, with any kind of in-plane pattern (additional layers, structuration of the perovskite or conformal perovskite) and whatever the main identified PE

Perov.thick. (nm)	$J_{sc}$ enhanc. (%)	Pattern. type	Photonic concept	Ref
400	6.3	periodic add. patterned layer	Broadband AR and LT	[48]
120	31.7	periodic add. patterned layer	Broadband AR and LT	[37]
300	5.6	periodic struct. of the perovskite	Broadband AR and LT	[33]
180	17	periodic conformal perovskite	Broadband AR and LT	[52]
300	10	periodic add. patterned layer	Broadband AR	[43]
300	10	periodic conformal perovskite	Broadband AR	[39]

TABLE III: Reported experimental  $J_{sc}$  enhancements mainly resulting from Light Management, for single junction PPSC, with any kind of in-plane pattern, and whatever the main identified PE

Material	Perov.thick. (nm)	$J_{sc}$ enhanc. (%)	Pattern. type	Photonic concept	Ref
MAPI	240	14.3	periodic add. patterned layer	Broadband AR and LT	[41]
CsFAMAPBI	370	2	periodic struct. of the perovskite	Broadband AR and LT	[34]
CsFAMAPBI	550	6.3	periodic, struct. of the ETL	Broadband AR	[44]
MAPICl	320	5	periodic struct. of the perovskite	Broadband AR	[27]
MAPI	340	1.8	aperiodic add. patterned layer	Broadband AR	[46]
MAPI	270	4.8	aperiodic add. patterned layer	Broadband AR	[47]
MAPICl	300	4	aperiodic add. patterned layer	Broadband AR	[42]
MAPI	300	3	aperiodic substrate corrugation	Broadband AR	[26]

### C. LM for PR

LM for  $V_{oc}$  enhancement thanks to PR in PPSC is discussed in a limited number of publications.

S. Nanz *et al.* [92] investigated mainly theoretically the effect of various kinds of LM strategies on the PR, for multilayer stacks that are part of PPSC, mainly without the HTL and metallic contact. They were thus able to derive an upper limit  $\Delta V_{oc}$  for each case, under the assumption of pure radiative recombination. According to the summarized principle reminded previously, the  $\Delta V_{oc}$  resulting from PR in a patterned multilayer was in between the values obtained with the Lambertian multilayer and the rigorously planar multilayer. Indeed, the Lambertian multilayer, as it consists in a better absorber, also radiates the luminescence, whereas, for targeted thicknesses, luminescence can be partly guided, leading to recycling. Then, the quasi-guided mode of patterned multilayer led to enhanced PR compared to the Lambertian multilayer, and also an enhanced  $J_{sc}$  compared to the flat multilayer. However, A. Bowman *et al.* [93] showed that using a more realistic model including recombination, PR was rather unlikely to occur at maximum peak power even if the cell only interacts with a limited solid angle. In this context, it thus appears more promising to increase the absorption and thus the extraction, to the detriment of recycling.

### V. Simulations and Perspective

As shown in the previous section, a limited number of studies, mainly focused on  $J_{sc}$  enhancements, evidenced a LM effect in PPSC. This synthesis also illustrates that due to a lack of common references, different kinds of PE can hardly be compared, so as the most promising architectures for AR and light trapping. Therefore, we propose

to simulate some of the promising patterns, to be able to compare their performances using typical materials and architecture of a PPSC.

### A. Methodology

The Rigorous Coupled Wave Analysis [94] suits well for the simulation of the stacks periodically patterned under plane wave illumination. We have used the  $S^4$  code [95] available in the Solcore package [96]. The derived  $J_{sc}$  are obtained using AM1.5G spectrum [97] provided an IQE of 1.

Optical indices used are from S. Manzoor *et al.* [98] for MAPI, from K.R. McIntosh *et al.* [99] for PMMA, and from J.M. Ball *et al.* [82] for all other materials: Sodalime Glass as a substrate, ITO,  $\text{TiO}_2$ , doped Spiro-OMeTAD (as HTL material), and gold. Thicknesses are set to 100 nm for top contact, 20 nm for ETL, and 300 nm for Au.

The ETL is supposed to be a thin  $\text{TiO}_2$  layer, dense and flat, to avoid scattering, and to limit parasitic absorption compared to other envisaged organic materials.

### B. Optimization of the key layers thicknesses in various planar single-junction PPSC

Let us first consider a planar PPSC on an infinitely thick substrate, illuminated through this substrate under normal incidence (see Figure 14). The thicknesses of the MAPI layer  $th_{MAPI}$  and of the HTL  $th_{HTL}$  are supposed to vary in the ranges 300 - 700 nm and 200 - 400 nm respectively. As can be seen in Figure SI-S3, the  $J_{sc}$  of such a cell increases with  $th_{MAPI}$ . Moreover, for a given  $th_{MAPI}$ , the  $th_{HTL}$  has a non-negligible influence in the  $J_{sc}$ ; e.g. for the smallest MAPI thickness of the considered range, the  $J_{sc}$  can be increased by more than  $0.5 \text{ mA.cm}^{-2}$ , up to about  $21.74 \text{ mA.cm}^{-2}$ .

It remains that the low  $J_{sc}$  for the thinnest considered MAPI is due to lower absorption at long wavelengths, as can be seen in Figure SI- S5, for a given thickness of HTL of 240 nm.

Given the possible identified advantages of using a thinner MAPI layer, mainly for electrical properties, its thickness will be set to 300 nm in the following. An AR PMMA layer is then coated on the glass substrate, which thickness is set to 1 mm (see Figure 14), with a negligible roughness. For the chosen  $th_{MAPI}$ , the coupled influence of PMMA thickness,  $th_{PMMA}$ , and  $th_{HTL}$  is studied. According to Figure SI- S6, a significantly increased  $J_{sc}$ , up to about  $21.92 \text{ mA.cm}^{-2}$ , can be obtained for  $th_{PMMA} = 360 \text{ nm}$  together with  $th_{HTL} = 250 \text{ nm}$ . This last structure will be then used as a planar but optimized reference for fair estimation of the impact of PE in the following. The corresponding spectrum is drawn in Figure 15.

### C. Introduction of various 2D PC to enhance the current density

As already discussed, to further increase the absorption and thus the  $J_{sc}$ , the most efficient strategies should be to couple the impinging light into the guided modes thanks to properly designed patterns. In the following, we envisage 2D square lattices of cylindrical patterns. Each resulting 2D PC owns typically three parameters that can be optimized: i) its period  $P$ , ii) its filling fraction ( $ff$ ) which is the ratio between the hole surface and the period surface, and iii) its thickness  $t$ . These 2D PC can be located either (see Figure 14):

- in the MAPI layer, made of holes in the MAPI layer filled with HTL material; for a fair comparison, the volume of MAPI material is the same as the planar PPSC, so its total thickness changes. Moreover,  $t < th_{MAPI}$  to prevent from short circuits between HTL and ETL. A  $t_{HTL} = 250\text{nm}$  thick slab of HTL is kept for planarization;
- in the top PMMA layer, patterned in a PC of air holes, with  $t = th_{HTL}$ , in favor of the diffraction efficiency, given the low index of the PMMA;
- simultaneously at the two previous locations, but each PC has its own set of parameters.

In the following studies, all the PC parameters are scanned over realistic ranges. The step for  $P$  and  $t$  is 5 nm, whereas only 3  $ff$  have been envisaged: 0.3, 0.4 and 0.5; these appear to be the most realistic values compatible with a large area patterning at a reasonable cost.

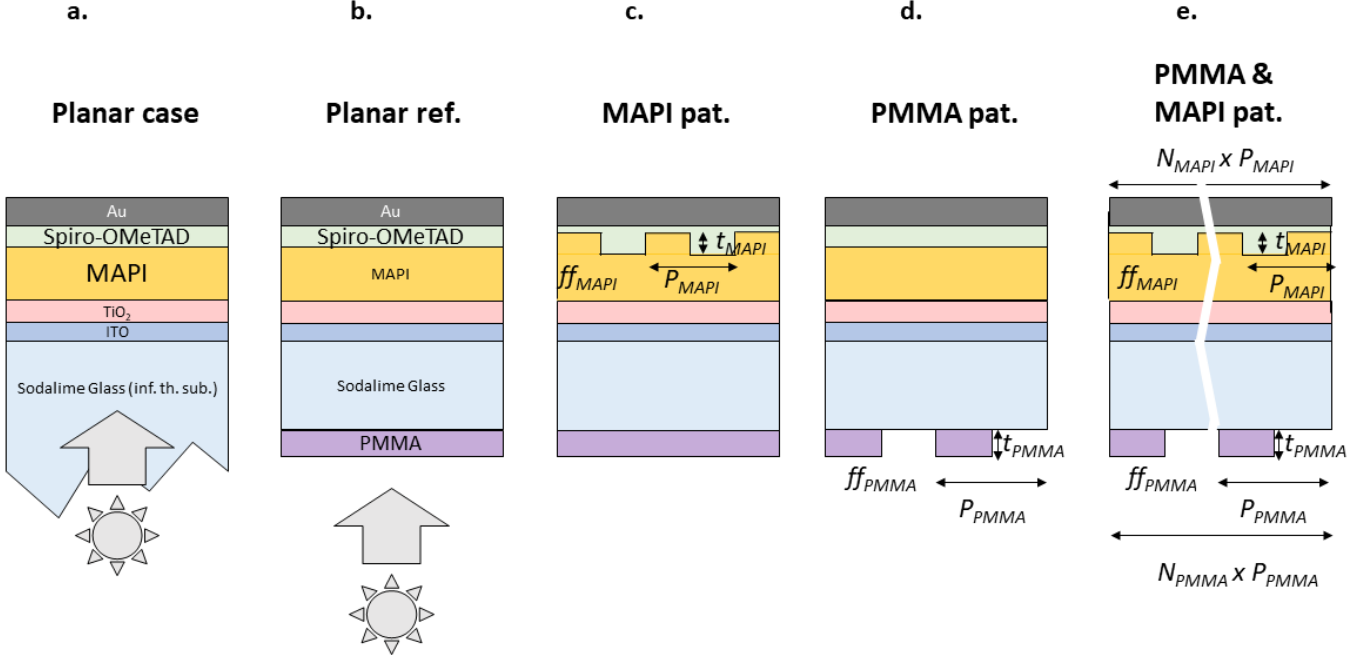


FIG. 14: Various configurations of single junction PPSC simulated in the perspective. (a) planer PPSC on an infinite glass substrate, to study the effect of MAPI and Spiro-OmeTAD thicknesses (see results in Figure SI-S3), (b) Planar reference having a 300 nm thick MAPI layer, and various PMMA and Spiro-Ometad thicknesses to maximise absorption (see results in Figure SI-S6), (c) Cross section of a 2D square lattice of cylindrical holes in MAPI, to study the effect of the corresponding PC parameters (see results in Figures SI-S7 and 15), (d) Cross section of a 2D square lattice of cylindrical holes in PMMA, to study the effect of the corresponding PC parameters (see results in Figures SI-S8 and 15), (e) Cross section of two 2D distinct square lattices of cylindrical holes in PMMA and MAPI, to study the effect of PC parameters (see results in Figures SI-S9 and 15)

### 1. Single junction: 2D PC in the perovskite layer

With respect to the best planar PPSC coated with PMMA, an increase of almost  $1 \text{ mA.cm}^{-2}$ , leading to a  $J_{sc}$  about  $22.88 \text{ mA.cm}^{-2}$  is obtained thanks to a PC into the MAPI (see Figure SI-S7), with  $P = 400 \text{ nm}$ ,  $ff = 0.4$  and  $t = 110 \text{ nm}$ . It has been checked that  $ff$  of 0.3 and 0.5 lead to lower current densities than the optimal one, but still larger than the reference. The corresponding spectra in Figure 15 reveal that the improvement is mainly due to a larger band edge absorption, so LT, as confirmed by the absorption enhancement in the same Figure.

It is noticeable that other periods, smaller than  $550 \text{ nm}$ , can lead to more limited  $J_{sc}$  enhancements. However, it has been checked that using periods around  $5$  and  $10 \mu\text{m} \pm 0.5$  lead to a  $J_{sc}$  lower than  $21.85 \text{ mA.cm}^{-2}$ . It confirms that such periods, far larger than the sub-micron optimal one, do not enable an efficient diffraction and thus do not lead to any  $J_{sc}$  enhancement, when compared to an optimized planar reference.

### 2. Single junction: 2D PC in the PMMA covering layer

With respect to the best planar PPSC coated with PMMA, an increase of almost  $0.9 \text{ mA.cm}^{-2}$ , leading to a  $J_{sc}$  about  $22.75 \text{ mA.cm}^{-2}$ , is obtained thanks to a PC in the PMMA layer. The 2D PC parameters are  $P = 665 \text{ nm}$ ,  $ff = 0.4$  and  $t = 615 \text{ nm}$  (see Figure SI-S8). It has been checked that  $ff$  of 0.3 and 0.5 lead to lower current densities. It can be noticed on the corresponding spectra in Figure 15 that the improvement is due to both a AR effect and limited LT at band edge absorption, since it implies low effective index guided modes that are the only able to interact, weakly, with the pattern on top of the thick substrate. The enhancement remains lower than the one induced by the 2D PC in the perovskite layer. Again, it has been checked that periods far larger than the optimal one, typically around  $5$  and  $10 \mu\text{m} \pm 0.5$  lead to a  $J_{sc}$  lower than  $22 \text{ mA.cm}^{-2}$ , so to a limited enhancement, because of a reduced diffraction efficiency.

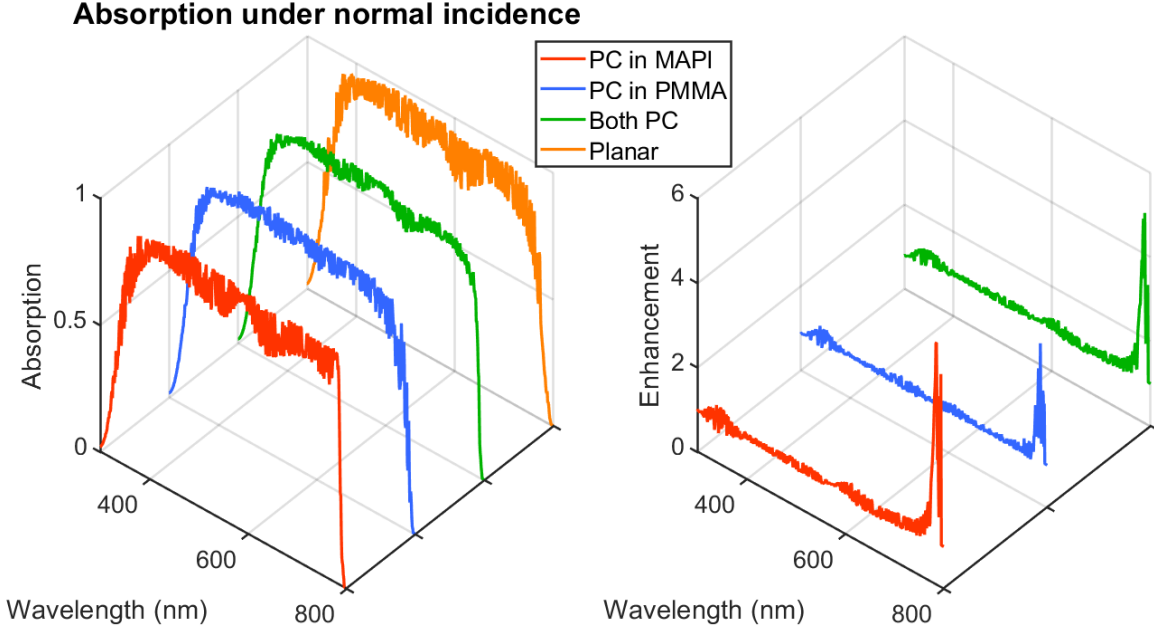


FIG. 15: Simulated absorption spectra for various 1 mm thick glass substrate PPSC with maximized short circuit current density: the planar PPSC (Figure 14 (b)), the PPSC with a 2D PC into the MAPI (Figure 14 (c)), the PPSC with a 2D PC in the PMMA (Figure 14 (d)), and the PPSC with both 2D PC (Figure 14 (e)). Small amplitude, high spectral resolution Fabry Perot resonances take place in the 1mm thick glass substrate.

### 3. Single junction: Combination of the two 2D PC

Given the previous enhancements, a structure that combines the two 2D PC, one in the PMMA and the second at the MAPI/HTL interface, can be envisaged. It is noticeable that RCWA method implies that the period of such a combined architecture is a integer  $N_{PMMA}$  times the period of the 2D PC in the PMMA ( $P_{PMMA}$ ), and another integer  $N_{MAPI}$  times the pitch of the 2D PC in the MAPI ( $P_{MAPI}$ ); other PC parameters ( $ff_{PMMA}$ ,  $t_{PMMA}$  on the one hand,  $ff_{MAPI}$ ,  $t_{MAPI}$  on the other hand) can differ (see Figure 14). For the sake of illustration of a possible further  $J_{sc}$  enhancement, it has been chosen to set the parameters of the 2D PC at the MAPI/HTL as for the optimized cell with a flat PMMA layer, i.e.  $P_{MAPI} = 400 \text{ nm}$ ,  $ff_{MAPI} = 0.4$ ,  $t_{MAPI} = 110 \text{ nm}$ , as well as  $ff_{PMMA} = 0.5$  and  $t_{PMMA} = 405 \text{ nm}$ , among the thinnest most favorable values of the 2D PC in PMMA associated with planar MAPI. Then,  $N_{PMMA}$  is scanned from 5 to 8 and  $N_{MAPI}$  from 9 to 15, given the fact that  $P_{PMMA}$  is typically larger  $P_{MAPI}$  according to the previous studies. The simulated  $J_{sc}$  displayed in Figure SI-S9 shows that a limited increase, of about  $0.26 \text{ mA.cm}^{-2}$ , up to  $23.14 \text{ mA.cm}^{-2}$ , is possible provided  $P_{PMMA} = 530 \text{ nm}$ . It can be seen in Figure 15 that such  $J_{sc}$  enhancement results from both LT and AR effects compared to the planar reference. If the full space of various possible PC parameters has not been scanned, and thus the previous parameters not fully optimized, the interest of such a combination is yet demonstrated.

## D. 2T tandem

A 2T tandem PPSC generally exhibits thermalization, as the one previously described in section IV A 3. Thus, we will consider this example as a case study for possible improvement. In the following simulations, the optical indices they provided for the considered materials are used. The planar stack (see Figure 16), considered as a reference, is close to the one shown in Figure 6a. Oppositely to our previous studies, the glass substrate is again supposed to be infinitely thick to avoid the additional interferences in the substrate. Within the stack, the layer thicknesses (except perovskites) have been set to realistic values such as  $t_{ITO} = 100 \text{ nm}$ ,  $t_{NiO} = t_{NVPB} = 10 \text{ nm}$  (simplified version of a mixed material ETL),  $t_{C60} = 10 \text{ nm}$  for both HTL, as well as  $t_{PEDOT-PSS} = 10 \text{ nm}$  and  $t_{Cu} = 100 \text{ nm}$ . As justified later, the  $\text{SnO}_2$  layer, with  $t_{\text{SnO}_2} = 100 \text{ nm}$ , acts as an optical spacer (the 1 nm thick Au layer has thus been neglect).

Setting the thicknesses of both perovskite layers to  $t_{1.77 \text{ eV } PK} = 400 \text{ nm}$  (also the maximum in the considered

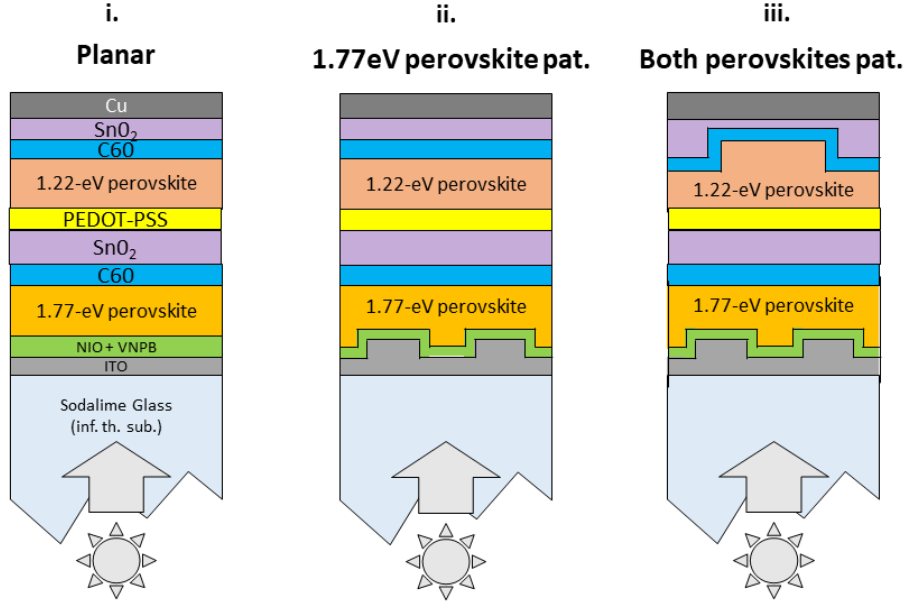


FIG. 16: Various configurations of 2T tandem PPSC simulated in the perspective. (i) Planar ref on an infinite glass substrate with varying perovskite materials thicknesses, to reach and equilibrium of  $J_{sc}$ . (ii) Cross section of a 2D square lattice of ITO pillars in 1.77-eV perovskite, to enhance  $J_{sc,1.77eV-perovskite}$  (iii) Cross section of a 2D square lattice of ITO pillars in the 1.77-eV perovskite, and of dots of 1.22-eV perovskite in HTL. Inspired by the configuration proposed by K. Xiao *et al.* [90]

range) and  $t_{1.22\text{ eV } PK} = 880\text{ nm}$  (in the range 800 - 1200 nm) leads to the highest  $J_{sc}$  of  $16.25\text{ mA.cm}^{-2}$ , that appears to be limited by the 1.77 eV perovskite subcell. This value is slightly larger than the one obtained by K. Xiao *et al.* Moreover, it is obtained in our case for different perovskite thicknesses, due to a mismatch between the thicknesses of the charge transport layers we have chosen and author's choices. However, our derived absorbance spectra for both sub cells (see Figure 17) still exhibits a thermalization effect. Simply increasing  $t_{1.77\text{ eV } PK}$  could be at the expense of the charges collection.

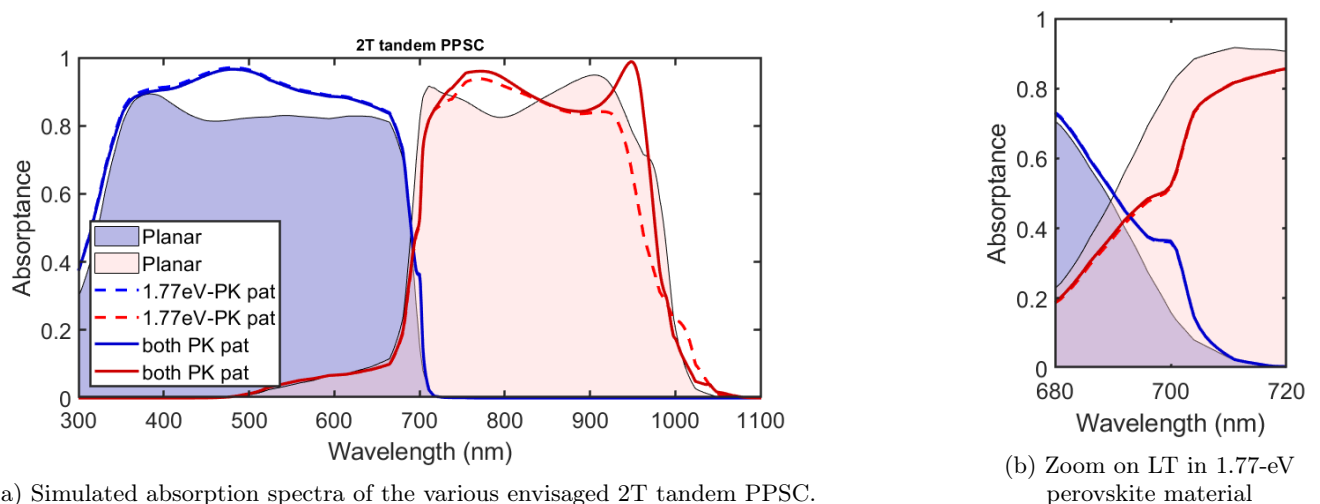
In this frame, the possible enhancement of the  $J_{sc,1.77\text{ eV } PK}$  thanks to a 2D PC at the ITO / 1.77 eV perovskite layer interface is studied. The 2D PC consists of a square lattice of ITO pillars, coated with conformal ETL and then with a 1.77 eV perovskite layer that planarizes the corresponding subcell, while keeping an equivalent volume of perovskite as in the planar tandem cell.

A significant enhancement of  $J_{sc,1.77\text{ eV } PK}$ , up to  $17.92\text{ mA.cm}^{-2}$ , has been found for  $P = 325\text{ nm}$ ,  $ff = 0.5$  (here defined as the ITO filling fraction) and  $t = 160\text{ nm}$ . The reason is twofold (see spectra in Figure 17): a broadband AR as well as a limited LT at the band edge of the 1.77 eV perovskite material that slightly reduces the thermalization. LT specifically occurs in one of the absorbing materials, to the detriment of the other one. It results from a guided mode mainly confined in the perovskite of interest, with a limited overlapping in the other perovskite, especially thanks to the rather large, 100 nm  $\text{SnO}_2$  optical spacer. Indeed, this layer can prevent evanescent coupling between quasi guided modes of the perovskites. Additionally, it could protect the 1.77 eV perovskite material during the fabrication, but the junction between the two subcells might then also be less efficient.

Moreover, this first step leads to a strong  $J_{sc}$  disequilibrium between the two sub cells, since  $J_{sc,1.22\text{ eV } PK}$  even slightly decreases to  $15.7\text{ mA.cm}^{-2}$ , compared to the planar reference. To increase in a second step  $J_{sc,1.22\text{ eV } PK}$ , this low-Eg perovskite layer can also be patterned rather than simply increasing its already large thickness.

Starting from the last structure, a second 2D PC is introduced at the 1.22 eV perovskite material / HTL interface. To target a LT at larger wavelength, close to the band edge of this perovskite, its period has to be increased. However, given the constraint induced by the boundary conditions of the simulation, we simply chose a supercell having a period twice the one of the single PC device, with one pattern in the 1.22 eV perovskite material, and two patterns in the 1.77 eV perovskite material (see Figure 16), while keeping  $ff$  and thicknesses constant. Moreover, again, the volume of this perovskite is kept constant, HTL acting as a planariser, with a minimum thickness of 10 nm (with a larger volume as in the planar alternative).

It appears that  $J_{sc,1.77\text{ eV } PK}$  remains unchanged, whereas  $J_{sc,1.22\text{ eV } PK}$  is increased up to  $16.2\text{ mA.cm}^{-2}$ , reducing, but not cancelling the disequilibrium. As expected, this results from LT at the band edge in the 1.22 eV perovskite



(a) Simulated absorption spectra of the various envisaged 2T tandem PPSC.

(b) Zoom on LT in 1.77-eV perovskite material

FIG. 17: 2T tandem PPSC simulated in the perspective, Inspired by the configuration proposed by K. Xiao *et al.* [90].

material (see Figure 17). This last result appears as a proof of concept to integrate two PC in order to induce LT, as well as possibly AR, in two different absorbing layers of a given stack, provided each one exhibits guided modes without overlapping. If not yet optimized, it is noticeable that, providing the first subcell is planarized, the two PC could practically have independent parameters, offering additional degrees of freedom to reduce disequilibrium. Given the required spacer, such a concept could be also applied to other kind of three or four terminal multijunctions cells.

## VI. Conclusion - Perspectives

Light management for PV solar cells was mainly intended to increase the  $J_{sc}$ , simply thanks to a larger absorption. When using metal-halide perovskite materials, LM is shown to result from an interaction between the various material choices, the patterning processes and the PE. Moreover, demonstrating sole LM needs to meet some rigorous criteria to get rid of mixed electrical, material and photonic effects.

The investigations presented in this review have demonstrated that the easiest  $J_{sc}$  enhancement is indeed the most frequently envisaged effect, especially for single junction cells, compared to the still highly challenging  $V_{oc}$  enhancement, and the immature studies of the EY due to the lack of studies at the module scale. As demonstrated by several authors, even thicknesses optimization within flat PPSC can enhance absorption thanks to Fabry Perot effects while taking care of the electrical constraints. To further increase the absorption at the perovskite band edge, an in-plane pattern, at various scales, can be introduced in one of the layers. However, patterns at the wavelength scale appeared as the most efficient. In any case, the  $J_{sc}$  enhancement compared to already optimized flat reference remained limited to a few percent. As regards the photonic regimes, the broadband AR effect was the most frequently observed, but only LT was able to significantly enhance the absorption at the band edge, as confirmed by our own simulations. In addition, we confirmed that direct patterning of the perovskite leads to more efficient LT than an additional layer on top of the substrate. Finally, in 2T tandem cells, we showed that one PC in each subcell can enhance each of the  $J_{sc}$  separately.

Beyond, some of the strategies developed for these opaque, single junction PPSC can be adjusted for other applications of hybrid perovskites for sun light harvesting. Indeed, accurately tuned spectral absorption is required in semitransparent single junction. Moreover, the same concepts can be tweaked to tailor reflectivity spectrum for perovskite-based color printing devices [100–103] and engineering absorption management in full-color perovskite detectors [104, 105].

This work is even part of far larger context. Indeed, other studies are ongoing concerning materials, which need to be more stable, and processes. In this frame, it can be noticed that record cells might not share the same encapsulation strategies as more realistic, large surface cells and modules [106], both for aging and safety reasons [107, 108]. All perovskite tandem cells are even more challenging on the fabrication point of view [109], but are also very promising since they combine all the potential of perovskite based cells and modules with high yields.

Finally, according to the reciprocity relation between absorption and emission [110], light extraction strategies in



perovskite LED [100, 111, 112], as well as resonances of high quality factor in perovskite-based laser [113–115] can use similar concepts to those derived in this work. Even LM in 2T tandem cells could be mimicked in white LEDs obtained by stacking several emitting materials. For LEDs, a high light extraction efficiency is a even more the key of good efficiency.

*Acknowledgement:*

RCWA simulations were performed on the Newton computer cluster facilities operated by PMCS2I at Ecole Centrale de Lyon and on the CNRS/IN2P3 Computing Center in Lyon. R. M. L. acknowledges project EMIPERO (ANR-18-CE24-0016).

- 
- [1] “Best research-cell efficiency chart,” .
- [2] H. J. Snaith, *Journal of Physical Chemistry Letters* **4**, 3623 (2013).
- [3] N. J. Jeon, J. H. Noh, W. S. Yang, Y. C. Kim, S. Ryu, J. Seo, and S. I. Seok, *Nature* **517**, 476 (2015).
- [4] W. Tress, *Advanced Energy Materials* **7**, 1602358 (2017).
- [5] R. Lin, J. Xu, M. Wei, Y. Wang, Z. Qin, Z. Liu, J. Wu, K. Xiao, B. Chen, S. M. Park, G. Chen, H. R. Atapattu, K. R. Graham, J. Xu, J. Zhu, L. Li, C. Zhang, E. H. Sargent, and H. Tan, *Nature* **603**, 73 (2022), number: 7899 Publisher: Nature Publishing Group.
- [6] O. D. Miller, E. Yablonovitch, and S. R. Kurtz, *Photovoltaics*, *IEEE Journal of* **2**, 303 (2012).
- [7] R. T. Ross, *The Journal of Chemical Physics* **46**, 4590 (1967).
- [8] I. M. Peters and T. Buonassisi, *Joule* **2**, 1160 (2018).
- [9] P. Bermel, C. Luo, L. Zeng, L. C. Kimerling, and J. D. Joannopoulos, *Optics Express* **15**, 16986 (2007).
- [10] D. Zhou and R. Biswas, *Journal of Applied Physics* **103**, 093102 (2008).
- [11] S. Sandhu, Z. Yu, and S. Fan, *Optics Express* **21**, 1209 (2013).
- [12] J. Dumoulin, E. Drouard, and M. Amara, *Sustainable Energy & Fuels* **5**, 2085 (2021).
- [13] L. M. Pazos-Outón, M. Szumilo, R. Lamboll, J. M. Richter, M. Crespo-Quesada, M. Abdi-Jalebi, H. J. Beeson, M. Vručinić, M. Alsari, H. J. Snaith, B. Ehrler, R. H. Friend, and F. Deschler, *Science* **351**, 1430 (2016).
- [14] Z. Yang, Z. Yu, H. Wei, X. Xiao, Z. Ni, B. Chen, Y. Deng, S. N. Habisreutinger, X. Chen, K. Wang, J. Zhao, P. N. Rudd, J. J. Berry, M. C. Beard, and J. Huang, *Nature Communications* **10**, 4498 (2019), number: 1 Publisher: Nature Publishing Group.
- [15] M. Liu, M. B. Johnston, and H. J. Snaith, *Nature* **501**, 395 (2013).
- [16] D. Liu and T. L. Kelly, *Nature Photonics* **8**, 133 (2014).
- [17] Y. Zhao, Q. Ye, Z. Chu, F. Gao, X. Zhang, and J. You, *Energy & Environmental Materials* **2**, 93 (2019).
- [18] C. Y. Xu, W. Hu, G. Wang, L. Niu, A. M. Elseman, L. Liao, Y. Yao, G. Xu, L. Luo, D. Liu, G. Zhou, P. Li, and Q. Song, *ACS Nano* **14**, 196 (2020).
- [19] H. Zhang and J. Toudert, *Science and Technology of Advanced Materials* **19**, 411 (2018).
- [20] Y. Zhang, C. K. Lim, Z. Dai, G. Yu, J. W. Haus, H. Zhang, and P. N. Prasad, *Physics Reports* **795**, 1 (2019).
- [21] R. Siavash Moakhar, S. Gholipour, S. Masudy-Panah, A. Seza, A. Mehdikhani, N. Riahi-Noori, S. Tafazoli, N. Timasi, Y. Lim, and M. Saliba, *Advanced Science* **7**, 1902448 (2020).
- [22] J. W. Lee, S. H. Lee, H. S. Ko, J. Kwon, J. H. Park, S. M. Kang, N. Ahn, M. Choi, J. K. Kim, and N. G. Park, *Journal of Materials Chemistry A* **3**, 9179 (2015).
- [23] S. H. Lin, Y. H. Su, H. W. Cho, P. Y. Kung, W. P. Liao, and J. J. Wu, *Journal of Materials Chemistry A* **4**, 1119 (2016).
- [24] F. J. Ramos, M. Oliva-Ramírez, M. K. Nazeeruddin, M. Graetzel, A. R. González-Elipe, and S. Ahmad, *Journal of Materials Chemistry A* **4**, 4962 (2016).
- [25] Y. Huang, J. Zhu, Y. Ding, S. Chen, C. Zhang, and S. Dai, *ACS Applied Materials and Interfaces* **8**, 8162 (2016).
- [26] A. R. Pascoe, S. Meyer, W. Huang, W. Li, I. Benesperi, N. W. Duffy, L. Spiccia, U. Bach, and Y.-B. Cheng, *Advanced Functional Materials* **26**, 1278 (2016).
- [27] U. W. Paetzold, W. Qiu, F. Finger, J. Poortmans, and D. Cheyns, *Applied Physics Letters* **106** (2015), 10.1063/1.4918751.
- [28] O. A. Abdelraouf and N. K. Allam, *Solar Energy* **137**, 364 (2016).
- [29] Y. Wang, P. Wang, X. Zhou, C. Li, H. Li, X. Hu, F. Li, X. Liu, M. Li, and Y. Song, *Advanced Energy Materials* **8**, 1702960 (2018).
- [30] D. I. Kim, J. W. Lee, R. H. Jeong, J.-H. Yu, J. W. Yang, S.-H. Nam, and J.-H. Boo, *Applied Surface Science* **487**, 409 (2019).
- [31] Y. Wang, P. Wang, X. Zhou, C. Li, H. Li, X. Hu, F. Li, X. Liu, M. Li, and Y. Song, *Advanced Energy Materials* **8**, 1702960 (2018).
- [32] Y. Wang, M. Li, X. Zhou, P. Li, X. Hu, and Y. Song, *Nano Energy* **51**, 556 (2018).
- [33] R. Schmager, G. Gomard, B. S. Richards, and U. W. Paetzold, *Solar Energy Materials and Solar Cells* **192**, 65 (2019).
- [34] R. Schmager, I. Hossain, F. Schackmar, B. Richards, G. Gomard, and U. Paetzold, *Solar Energy Materials and Solar Cells* **201**, 110080 (2019).
- [35] T. K. Nguyen, P. T. Dang, and K. Q. Le, *Journal of Optics (United Kingdom)* **18**, 125901 (2016).
- [36] K. Li, S. Zhang, Y. Ruan, D. Li, T. Zhang, and H. Zhen, *Optics Express* **27**, A1004 (2019).
- [37] H. Kim, H. Kwak, I. Jung, M. S. Kim, J. Kim, H. J. Park, and K.-T. Lee, *Optics Express* **29**, 35366 (2021).



- [38] S. M. Kang, S. Jang, J.-K. Lee, J. Yoon, D.-E. Yoo, J.-W. Lee, M. Choi, and N.-G. Park, *Small* **12**, 2443 (2016).
- [39] W. Qarony, M. I. Hossain, R. Dewan, S. Fischer, V. B. Meyer-Rochow, A. Salleo, D. Knipp, and Y. H. Tsang, *Advanced Theory and Simulations* **1**, 1800030 (2018).
- [40] F. Wang, X. Li, J. Du, H. Duan, H. Wang, Y. Gou, L. Yang, L. Fan, J. Yang, and F. Rosei, *Nano Energy*, 106765 (2021).
- [41] J. Wei, R.-P. Xu, Y.-Q. Li, C. Li, J.-D. Chen, X.-D. Zhao, Z.-Z. Xie, C.-S. Lee, W.-J. Zhang, and J.-X. Tang, *Advanced Energy Materials* **7**, 1700492 (2017).
- [42] H. Zhang, M. Kramarenko, J. Osmond, J. Toudert, and J. Martorell, *ACS Photonics* **5**, 2243 (2018).
- [43] M. I. Hossain, N. Yumnam, W. Qarony, A. Salleo, V. Wagner, D. Knipp, and Y. H. Tsang, *Solar Energy* **198**, 570 (2020).
- [44] P. Tockhorn, J. Sutter, R. Colom, L. Kegelmann, A. Al-Ashouri, M. Roß, K. Jäger, T. Unold, S. Burger, S. Albrecht, and C. Becker, *ACS Photonics* **7**, 2589 (2020).
- [45] M. M. Tavakoli, K. H. Tsui, Q. Zhang, J. He, Y. Yao, D. Li, and Z. Fan, *ACS Nano* **9**, 10287 (2015).
- [46] B. Dudem, J. H. Heo, J. W. Leem, J. S. Yu, and S. H. Im, *Journal of Materials Chemistry A* **4**, 7573 (2016).
- [47] M. Jošt, S. Albrecht, L. Kegelmann, C. M. Wolff, F. Lang, B. Lipovšek, J. Krč, L. Korte, D. Neher, B. Rech, and M. Topič, *ACS Photonics* **4**, 1232 (2017).
- [48] A. Peer, R. Biswas, J.-M. Park, R. Shinar, and J. Shinar, *Optics Express* **25**, 10704 (2017).
- [49] G. Martinez-Denegri, S. Colodrero, Q. Liu, J. Toudert, G. Kozyreff, and J. Martorell, *Advanced Optical Materials* **0**, 1900018 (2019).
- [50] N. R. Thangavel, G. W. Adhyaksa, H. A. Dewi, L. Tjahjana, A. Bruno, M. D. Birowosuto, H. Wang, N. Mathews, and S. Mhaisalkar, *ACS Photonics* **7**, 1971 (2020).
- [51] D.-L. Wang, H.-J. Cui, G.-J. Hou, Z.-G. Zhu, Q.-B. Yan, and G. Su, *Scientific Reports* **6**, 18922 (2016).
- [52] Q. G. Du, G. Shen, and S. John, *AIP Advances* **6**, 065002 (2016).
- [53] B. Shi, B. Liu, J. Luo, Y. Li, C. Zheng, X. Yao, L. Fan, J. Liang, Y. Ding, C. Wei, D. Zhang, Y. Zhao, and X. Zhang, *Solar Energy Materials and Solar Cells* **168**, 214 (2017).
- [54] M. Soldera and K. Taretto, *Physica status solidi (a)* **215**, 1700906 (2018).
- [55] A. Tooghi, D. Fathi, and M. Eskandari, *Scientific Reports* **10** (2020), 10.1038/s41598-020-75630-4.
- [56] F. Wang, Y. Zhang, M. Yang, L. Fan, L. Yang, Y. Sui, J. Yang, and X. Zhang, *Nano Energy* **60**, 198 (2019).
- [57] M. Soldera, Q. Wang, F. Soldera, V. Lang, A. Abate, and A. F. Lasagni, *Advanced Engineering Materials* **22**, 1901217 (2020).
- [58] S. Haque, M. Alexandre, M. J. Mendes, H. Águas, E. Fortunato, and R. Martins, *Applied Materials Today* **20**, 100720 (2020).
- [59] M. Lobet, A. Mayer, A. Maho, P. Piron, J. Dewalque, C. Henrist, and J. Loicq, *Applied Sciences* **10**, 1783 (2020).
- [60] W. Shockley and H. J. Queisser, *Journal of Applied Physics* **32**, 510 (1961).
- [61] J. Y. Jiang, A. M. Soufiani, and A. Ho-Baillie, *Article in Journal of Physical Chemistry Letters* (2015), 10.1021/acs.jpcclett.5b01865.
- [62] H. Li, C. Cui, X. Xu, S. Bian, C. Ngaojampa, P. Ruankham, and A. P. Jaroenjittchai, *Solar Energy* **212**, 48 (2020).
- [63] I. Massiot, A. Cattoni, and S. Collin, *Nature Energy* **5**, 959 (2020).
- [64] E. Yablonovitch and G. Cody, *IEEE Transactions on Electron Devices* **29**, 300 (1982).
- [65] K. Vynck, M. Buresi, F. Riboli, and D. S. Wiersma, *Nature materials* **11**, 1017 (2012).
- [66] E. R. Martins, J. Li, Y. Liu, V. Depauw, Z. Chen, J. Zhou, and T. F. Krauss, *Nature communications* **4**, 1 (2013).
- [67] A. Oskooi, M. De Zoysa, K. Ishizaki, and S. Noda, *ACS Photonics* **1**, 304 (2014), arXiv:1311.5410.
- [68] A. Bozzola, M. Liscidini, and L. C. Andreani, *Progress in Photovoltaics: Research and Applications* **22**, 1237 (2014).
- [69] H. Ding, L. Lalouat, B. Gonzalez-Acevedo, R. Orobtcouk, C. Seassal, and E. Drouard, *Optics Express* **24**, A650 (2016).
- [70] K. Li, S. Haque, A. Martins, E. Fortunato, R. Martins, M. J. Mendes, and C. Schuster, *Optica* **7**, 1377 (2020).
- [71] E. Camarillo Abad, H. J. Joyce, and L. C. Hirst, *Optics Express* **28**, 39093 (2020).
- [72] Y. Park, E. Drouard, O. Daif, X. Letartre, P. Viktorovitch, A. Fave, A. Kaminski, M. Lemitte, and C. Seassal, *Optics Express* **17** (2009), 10.1364/OE.17.014312.
- [73] Z. Yu, A. Raman, and S. Fan, *Proceedings of the National Academy of Sciences of the United States of America* **107**, 17491 (2010).
- [74] S. Zanotto, M. Liscidini, and L. C. Andreani, *Optics express* **18**, 4260 (2010).
- [75] M. L. Brongersma, Y. Cui, and S. Fan, *Nature Materials* **13**, 451 (2014).
- [76] Y. Da, Y. Xuan, and Q. Li, *Solar Energy Materials and Solar Cells* **174**, 206 (2018).
- [77] C. Momblona, O. Malinkiewicz, C. Roldán-Carmona, A. Soriano, L. Gil-Escrig, E. Bandiello, M. Scheepers, E. Edri, and H. J. Bolink, *APL Materials* **2**, 081504 (2014).
- [78] D. Liu, M. K. Gangishetty, and T. L. Kelly, *Journal of Materials Chemistry A* **2**, 19873 (2014).
- [79] Y. Liu, B. Jin, H. Zhang, Y. Zhang, Y. Kim, C. Wang, S. Wen, B. Xu, C. Im, and W. Tian, *ACS Applied Materials and Interfaces* **11** (2019), 10.1021/acsami.9b01952.
- [80] J.-P. Correa-Baena, M. Anaya, G. Lozano, W. Tress, K. Domanski, M. Saliba, T. Matsui, T. J. Jacobsson, M. E. Calvo, A. Abate, M. Grätzel, H. Míguez, and A. Hagfeldt, *Advanced Materials* **28**, 5031 (2016), eprint: <https://onlinelibrary.wiley.com/doi/pdf/10.1002/adma.201600624>.
- [81] J. Jeong, M. Kim, J. Seo, H. Lu, P. Ahlawat, A. Mishra, Y. Yang, M. A. Hope, F. T. Eickemeyer, M. Kim, Y. J. Yoon, I. W. Choi, B. P. Darwich, S. J. Choi, Y. Jo, J. H. Lee, B. Walker, S. M. Zakeeruddin, L. Emsley, U. Rothlisberger, A. Hagfeldt, D. S. Kim, M. Grätzel, and J. Y. Kim, *Nature* **592**, 381 (2021).
- [82] J. M. Ball, S. D. Stranks, M. T. Hörantner, S. Hüttner, W. Zhang, E. J. Crossland, I. Ramirez, M. Riede, M. B. Johnston,

- R. H. Friend, and H. J. Snaith, [Energy and Environmental Science](#) **8**, 602 (2015).
- [83] Q. Lin, A. Armin, R. C. R. Nagiri, P. L. Burn, and P. Meredith, [Nature Photonics](#) **9**, 106 (2015).
- [84] N. J. Jeon, J. H. Noh, W. S. Yang, Y. C. Kim, S. Ryu, J. Seo, and S. I. Seok, [Nature](#) **517**, 476 (2015).
- [85] T. J. Jacobsson, J.-P. Correa-Baena, M. Pazoki, M. Saliba, K. Schenk, M. Grätzel, and A. Hagfeldt, [Energy & Environmental Science](#) **9**, 1706 (2016).
- [86] M. Rai, L. H. Wong, and L. Etgar, [The Journal of Physical Chemistry Letters](#) **11**, 8189 (2020).
- [87] K. B. Nine, M. N. H. Shazon, M. N. H. Shazon, and S. A. Mahmood, [JOSA B](#) **37**, 2996 (2020), publisher: Optical Society of America.
- [88] J. Li, T. Jiu, C. Duan, Y. Wang, H. Zhang, H. Jian, Y. Zhao, N. Wang, C. Huang, and Y. Li, [Nano Energy](#) **46**, 331 (2018).
- [89] M. van Eerden, M. Jaysankar, A. Hadipour, T. Merckx, J. J. Schermer, T. Aernouts, J. Poortmans, and U. W. Paetzold, [Advanced Optical Materials](#) **5**, 1700151 (2017).
- [90] K. Xiao, R. Lin, Q. Han, Y. Hou, Z. Qin, H. T. Nguyen, J. Wen, M. Wei, V. Yeddu, M. I. Saidaminov, Y. Gao, X. Luo, Y. Wang, H. Gao, C. Zhang, J. Xu, J. Zhu, E. H. Sargent, and H. Tan, [Nature Energy](#) **5**, 870 (2020).
- [91] R. Wang, T. Huang, J. Xue, J. Tong, K. Zhu, and Y. Yang, [Nature Photonics](#) **15**, 411 (2021).
- [92] S. Nanz, R. Schmagier, M. G. Abebe, C. Willig, A. Wickberg, A. Abass, G. Gomard, M. Wegener, U. W. Paetzold, and C. Rockstuhl, [APL Photonics](#) **4**, 76104 (2019).
- [93] A. R. Bowman, M. Anaya, N. C. Greenham, and S. D. Stranks, [Physical review letters](#) **125**, 067401 (2020).
- [94] M. G. Moharam and T. K. Gaylord, [JOSA](#) **71**, 811 (1981).
- [95] V. Liu and S. Fan, [Computer Physics Communications](#) **183**, 2233 (2012).
- [96] D. Alonso-Álvarez, T. Wilson, P. Pearce, M. Führer, D. Farrell, and N. Ekins-Daukes, [Journal of Computational Electronics](#) **17**, 1099 (2018).
- [97] C. A. Gueymard, Report No. FSEC-PF-270-95 (1995).
- [98] S. Manzoor, J. Häusele, K. A. Bush, A. F. Palmstrom, J. Carpenter, Z. J. Yu, S. F. Bent, M. D. Mcgehee, and Z. C. Holman, [Optics Express](#) **26**, 27441 (2018).
- [99] K. R. McIntosh, G. Lau, J. N. Cotsell, K. Hanton, D. L. Bätzner, F. Bettiol, and B. S. Richards, [Progress in Photovoltaics: Research and Applications](#) **17**, 191 (2009).
- [100] B. Gholipour, G. Adamo, D. Cortecchia, H. N. S. Krishnamoorthy, M. D. Birowosuto, N. I. Zheludev, and C. Soci, [Advanced Materials](#) **29**, 1604268 (2017).
- [101] Y. Gao, C. Huang, C. Hao, S. Sun, L. Zhang, C. Zhang, Z. Duan, K. Wang, Z. Jin, N. Zhang, A. V. Kildishev, C.-W. Qiu, Q. Song, and S. Xiao, [ACS Nano](#) **12**, 8847 (2018).
- [102] Y. Fan, Y. Wang, N. Zhang, W. Sun, Y. Gao, C.-W. Qiu, Q. Song, and S. Xiao, [Nature Communications](#) **10**, 2085 (2019).
- [103] J. J. Yoo, G. Seo, M. R. Chua, T. G. Park, Y. Lu, F. Rotermund, Y. K. Kim, C. S. Moon, N. J. Jeon, J. P. Correa-Baena, V. Bulović, S. S. Shin, M. G. Bawendi, and J. Seo, [Nature](#) **590**, 587 (2021).
- [104] M. I. Hossain, H. A. Khan, M. Kozawa, W. Qarony, A. Salleo, J. Y. Hardeberg, H. Fujiwara, Y. H. Tsang, and D. Knipp, [ACS Applied Materials & Interfaces](#) **12**, 47831 (2020).
- [105] W. Qarony, M. Kozawa, H. A. Khan, M. I. Hossain, A. Salleo, Y. H. Tsang, J. Y. Hardeberg, H. Fujiwara, and D. Knipp, [Advanced Materials Interfaces](#) **7**, 2000459 (2020).
- [106] S. Wang, Z. Zhang, Z. Tang, C. Su, W. Huang, Y. Li, and G. Xing, [Nano Energy](#) **82**, 105712 (2021).
- [107] S. Wu, Z. Chen, H.-L. Yip, and A. K. Y. Jen, [Matter](#) **4**, 3814 (2021).
- [108] T. Wu, Z. Qin, Y. Wang, Y. Wu, W. Chen, S. Zhang, M. Cai, S. Dai, J. Zhang, J. Liu, Z. Zhou, X. Liu, H. Segawa, H. Tan, Q. Tang, J. Fang, Y. Li, L. Ding, Z. Ning, Y. Qi, Y. Zhang, and L. Han, [Nano-Micro Letters](#) **13**, 152 (2021).
- [109] X. Zheng, A. Y. Alsalloum, Y. Hou, E. H. Sargent, and O. M. Bakr, [Accounts of Materials Research](#) **1**, 63 (2020).
- [110] U. Rau, [Physical Review B - Condensed Matter and Materials Physics](#) **76**, 1 (2007).
- [111] H. Wang, S.-C. Liu, B. Balachandran, J. Moon, R. Haroldson, Z. Li, A. Ishteev, Q. Gu, W. Zhou, A. Zakhidov, and W. Hu, [Opt. Express](#) **25**, A1162 (2017).
- [112] E. Tiguntseva, A. Chebykin, A. Ishteev, R. Haroldson, B. Balachandran, E. Ushakova, F. Komissarenko, H. Wang, V. Milichko, A. Tsypkin, D. Zuev, W. Hu, S. Makarov, and A. Zakhidov, [Nanoscale](#) **9**, 12486 (2017).
- [113] S. Chen, K. Roh, J. Lee, W. K. Chong, Y. Lu, N. Mathews, T. C. Sum, and A. Nurmikko, [ACS Nano](#) **10**, 3959 (2016).
- [114] N. Pourdavoud, S. Wang, A. Mayer, T. Hu, Y. Chen, A. Marianovich, W. Kowalsky, R. Heiderhoff, H.-C. Scheer, and T. Riedl, [Advanced Materials](#) **29**, 1605003 (2017).
- [115] C. Qin, A. S. Sandanayaka, C. Zhao, T. Matsushima, D. Zhang, T. Fujihara, and C. Adachi, [Nature](#) **585**, 53 (2020).

— SUPPLEMENTAL MATERIAL —

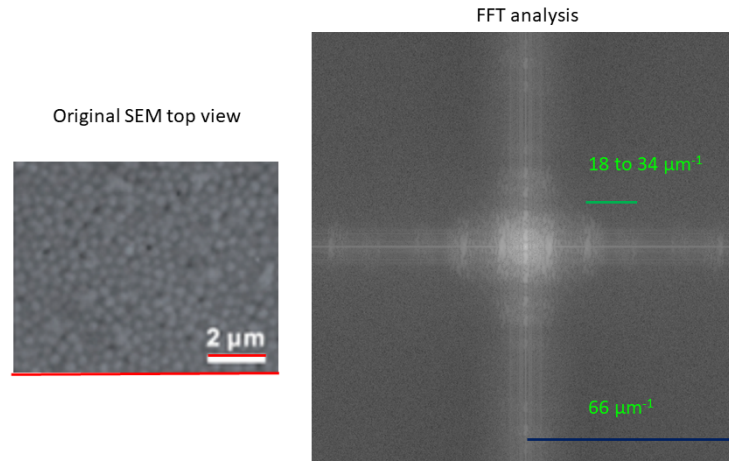


FIG. S1: Fourier space analysis of the moth-eye pattern proposed by Wei *et al* [41]. The 615 pixels of the real space view correspond to  $9.26\mu m$ ; so, the resolution is  $0.015\mu m$  per pixel. Thus, the largest spatial frequency in the Fourier space is  $66\mu m^{-1}$ . From equation 2, the spatial frequency range enabling the coupling of a plane wave under normal incidence into a guided mode having an effective index of two in the absorption domain of MAPI is  $[18 - 34\mu m^{-1}]$ . It thus appears that some significant spatial frequencies of the pattern lie in the domain of interest, especially at large wavelengths (low spatial frequencies), partially justifying LT.

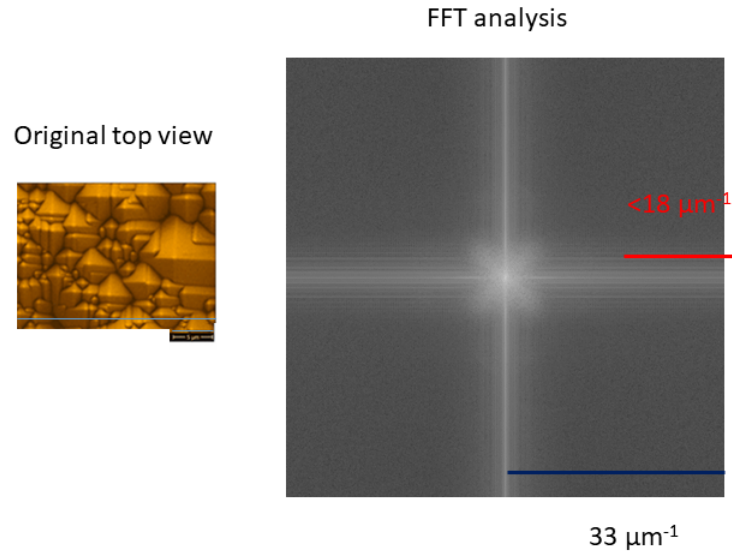


FIG. S2: Fourier space analysis of the moth-eye pattern proposed by M. Jöst *et al* [47]. The 828 pixels of the real space view correspond to  $24.55\mu m$ ; so, the resolution is  $0.03\mu m$  per pixel. Thus, the largest spatial frequency in the Fourier space is  $33\mu m^{-1}$ . From equation 2, the spatial frequency range enabling the coupling of a plane wave under normal incidence into a guided mode having an effective index of two in the absorption domain of MAPI is  $[18 - 34\mu m^{-1}]$ . Most of the significant spatial frequencies of the pattern lie outside the domain of interest. LT is thus unlikely in such a device.

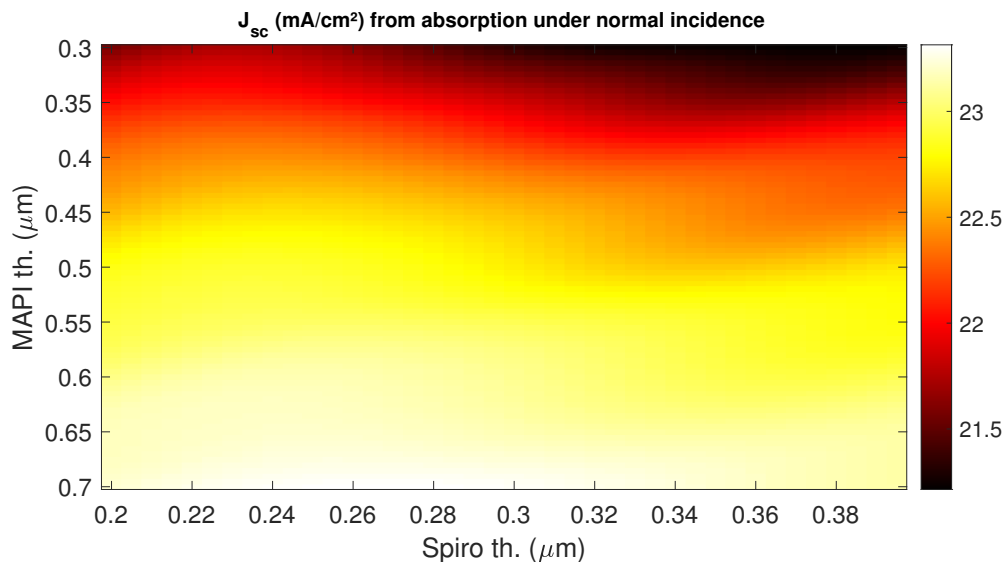


FIG. S3: Simulated short circuit current density, under AM1.5G spectrum under normal incidence provided a unitary IQE for the planar MAPI PPSC as a function of HTL and perovskite thickness.

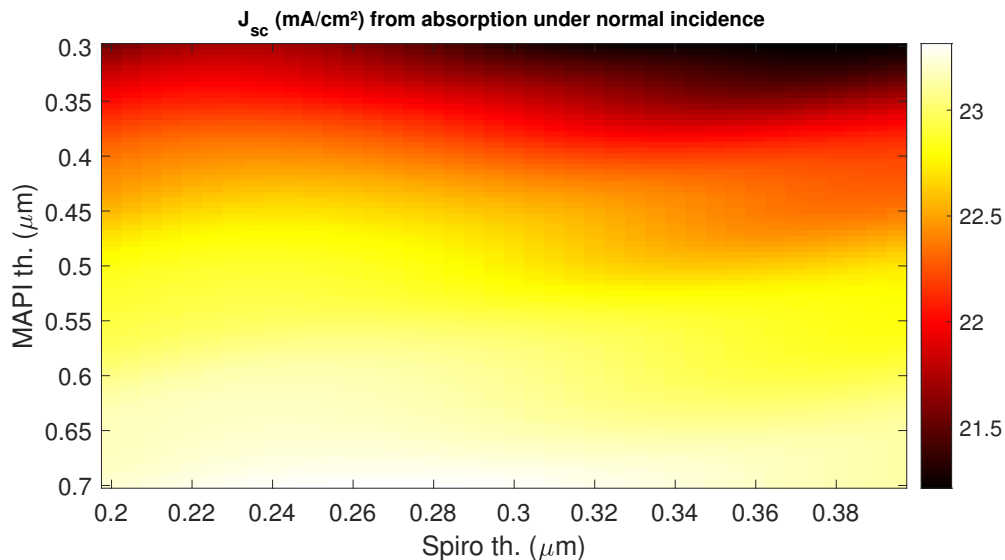


FIG. S4: Simulated short circuit current density under AM1.5G spectrum under normal incidence provided a unitary IQE for the planar MAPI PPSC as a function of HTL and perovskite thickness.

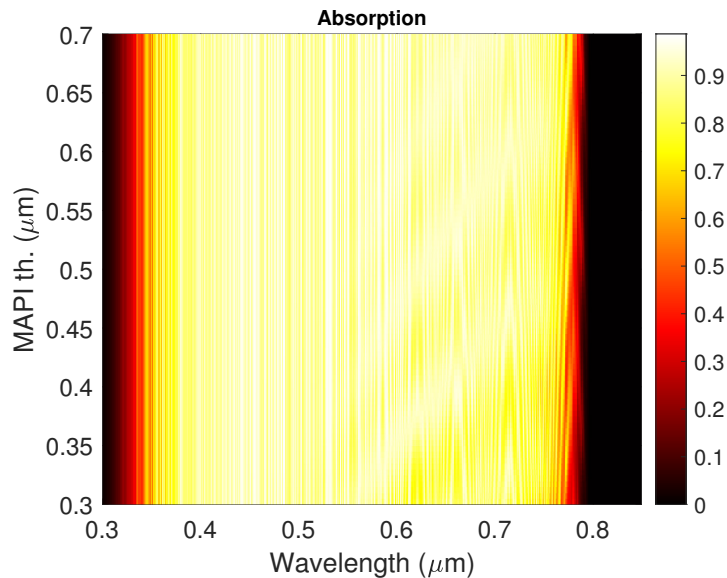


FIG. S5: Simulated absorption spectra of a planar MAPI PPSC as a function of the perovskite thickness for a  $0.24 \mu\text{m}$  thick HTL.

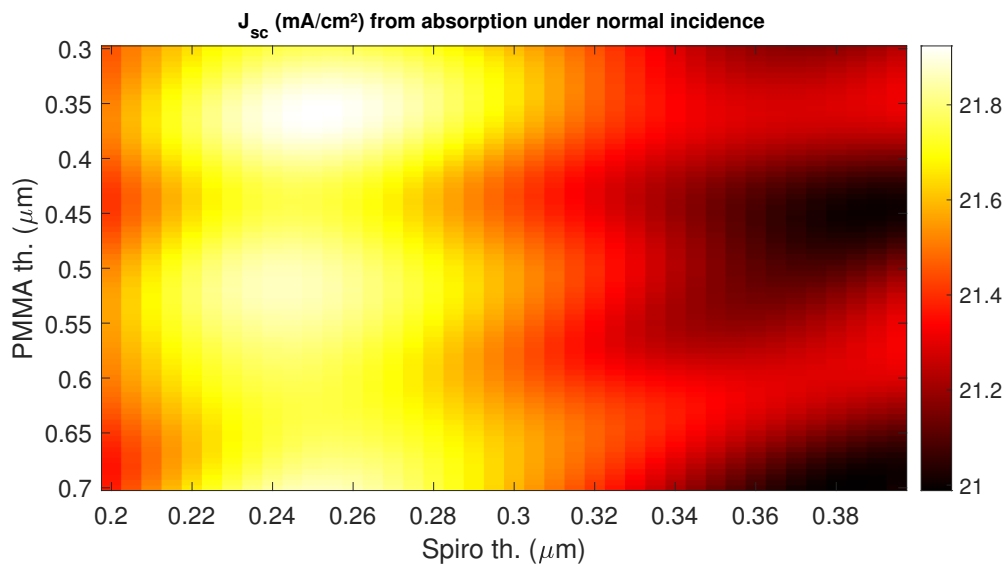


FIG. S6: Simulated short circuit current density under AM1.5G spectrum under normal incidence, provided a unitary IQE of a planar  $0.3 \mu\text{m}$  thick MAPI PPSC as a function of HTL and PMMA thickness.

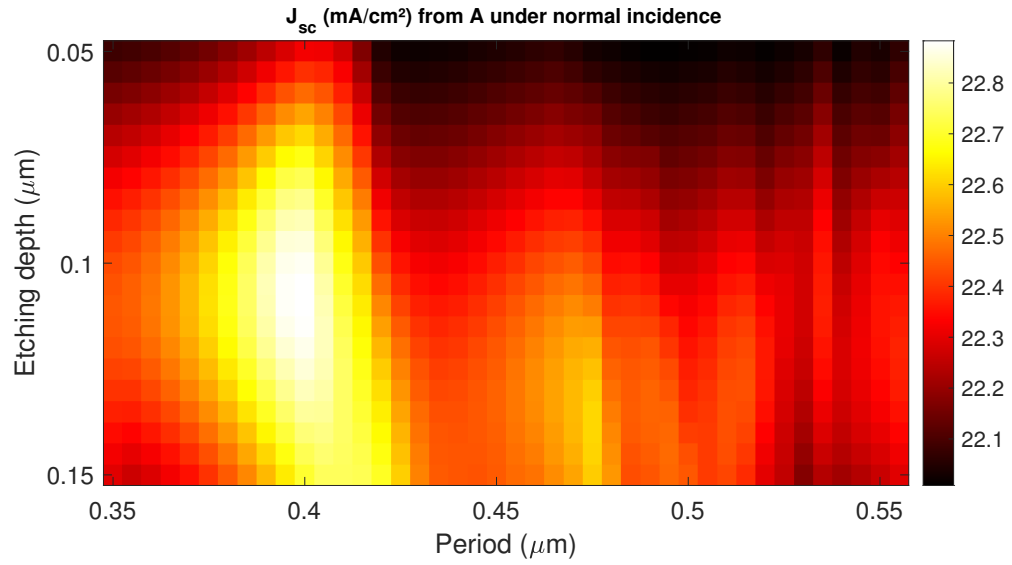


FIG. S7: Simulated short circuit current density under AM1.5G spectrum under normal incidence provided a unitary IQE of a 2D MAPI PPSC with  $0.25 \mu\text{m}$  thick flat HTL, covering  $0.32 \mu\text{m}$  thick PMMA layer as a function of the hole depth and the period for  $ff$  of 0.4. For all configurations, the volume of MAPI remained the same as that of the  $0.3 \mu\text{m}$  thick unpatterned layer previously considered.

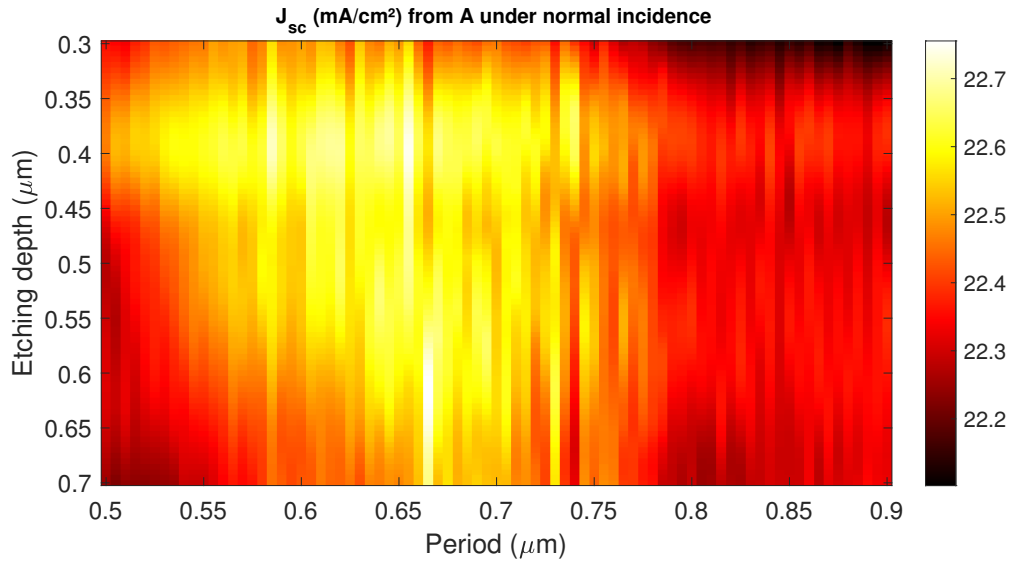


FIG. S8: Simulated short circuit current density under AM1.5G spectrum under normal incidence provided a unitary IQE, of a planar  $0.3 \mu\text{m}$  thick MAPI PPSC, with a  $0.25 \mu\text{m}$  thick flat HTL coated with 2D PC PMMA layer as a function of thickness and period, for  $ff$  of 0.4



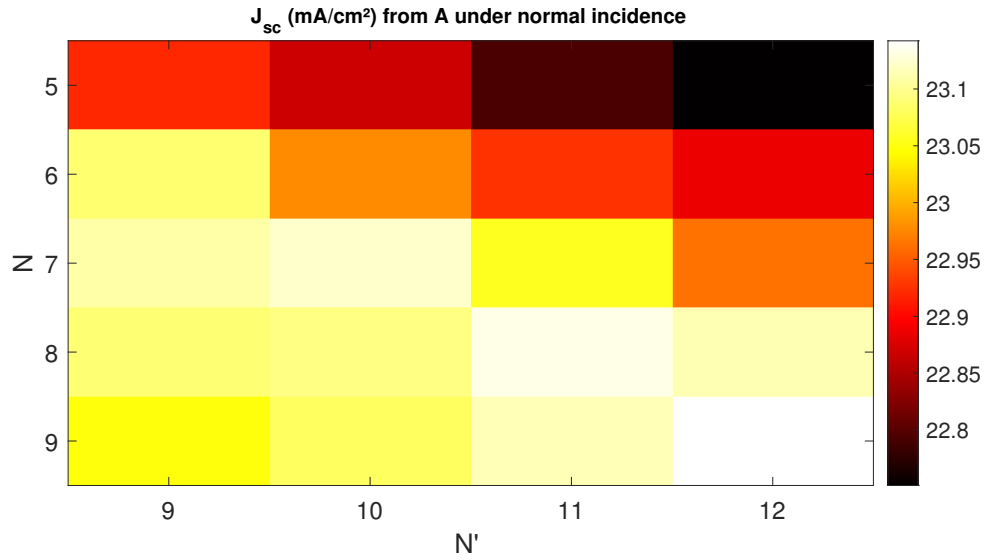


FIG. S9: Simulated short circuit current density under AM1.5G spectrum under normal incidence provided a unitary IQE of a PPSC combining two 2D PCs, one in PMMA and the second at the MAPI/HTL interface. The 2D PC in MAPI has a pitch of  $0.4 \mu\text{m}$ , thickness of  $0.11 \mu\text{m}$ , and the  $ff$  is 0.4. The  $ff$  of the 2D PC in PMMA is set to 0.5, and the thickness of the PMMA layer is set to 0.405. The simulated PPSC thus has a period which is  $N$  times the pitch of the 2D PC with PMMA and  $N'$  times the pitch of the 2D PC with MAPI with  $0.4 \mu\text{m}$

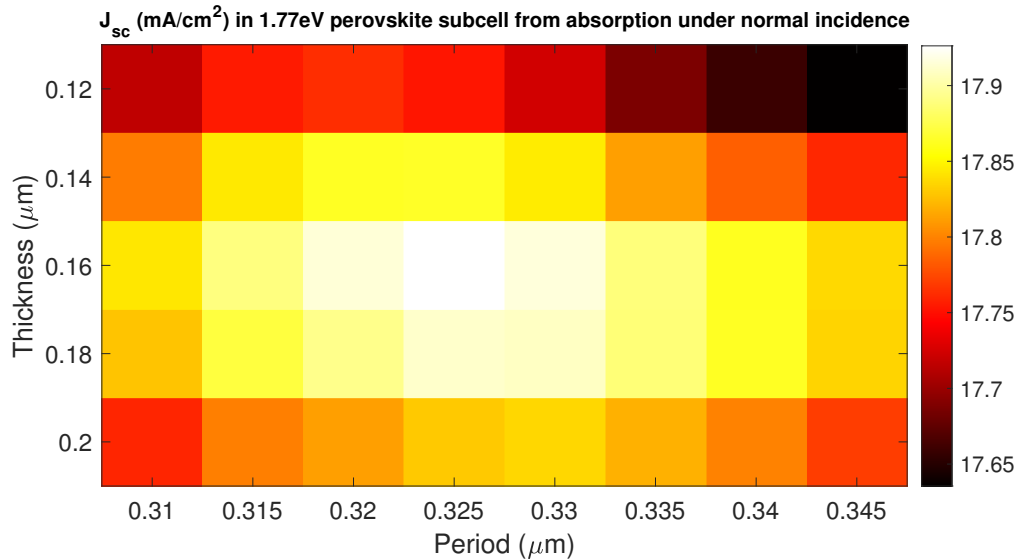


FIG. S10: Simulated short circuit current density, under AM1.5G spectrum under normal incidence provided a unitary IQE, of a 2T tandem PPSC combined with a 2D PC in the large bandgap subcell as a function of thickness and period, provided  $ff = 0.5$ .