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Fabrication of Si-based dielectric resonators: Combining etaloning with Mie resonances

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Abstract: We use low-resolution optical lithography and plasma etching joined with solid state dewetting of crystalline, ultra-thin silicon on insulator (c-UT-SOI) to form monocrystalline, atomically-smooth, silicon-based Mie resonators in well controlled large periodic arrays. The dewetted islands have a typical size in the 100 nm range, about one order of magnitude smaller than the etching resolution. Exploiting a 2 μm thick SiO_2 layer separating the islands and the underlying bulk, silicon wafer, we combine the resonant modes of the antennas with the etalon effect. This approach sets the resonance spectral position, improves the structural colorization and the contrast between scattering maxima and minima of individual resonant antennas. Our results demonstrate that templated dewetting enables to form defect-free, faceted islands that are much smaller than the nominal etching resolution and that an appropriate engineering of the substrate improves their scattering properties. These results are relevant to applications in spectral filtering, structural color and beam steering with all-dielectric photonic devices.

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1. Introduction

Sub-micrometric, dielectric objects featuring high permittivity and reduced absorption losses enable for efficient light management, potentially enhancing and extending the performances of opto-electronic devices [1]. The resonant scattering supported by individual dielectric antennas is generally rather broad [2] (50-100 nm for the fundamental Mie resonances at visible and near-infrared frequency) and are strongly influenced by the coupling with the underlying substrate [3]. In addition, for individual resonators, the intensity contrast between maxima and minima is generally poor.

Common strategies for overcoming these limitations rely in coupling the nano-antennas together, forming complex oligomers and meta-surfaces, that can provide advanced functionalities and

sharp resonances [1, 4]. However, these approaches require for advanced fabrication methods to precisely set size, shape and relative position of the monomers (e.g. forming narrow gaps in between them). Another powerful approach, that has been recently used to improve the use of dielectric Mie resonators as anti-reflection coatings, is the coupling of Fabry-Perot modes (formed within the effective medium containing Si-based nano-pillars) and Mie resonances (formed within the Si nano-pillars) in a single device [5].

Here we exploit the photonic modulation of the dielectric thick layers to extract more defined structural colors from dielectric Mie resonators. We fabricate monocrystalline Si-based islands via low-resolution optical lithography and plasma etching followed by solid state dewetting [6–12]. We show that the far-field scattering intensity of the Si-based islands sitting atop a 2 μm thick SiO_2 layer on bulk Si, can be efficiently coupled with etalon modes [3]. The Mie scattering efficiently out-couples the light interfering in the etalon according to the Si island size and re-directs the light at smaller angles with respect to the incident beam. From this combination spring structural colours covering the full visible spectrum and resonances with a high intensity contrast between maxima and minima. FDTD simulations account for this coupling showing that is a purely far-field effect with no modification of the local density of optical states.

2. Methods

2.1. Fabrication

The mono-crystalline silicon on insulator (SOI) is a single-crystal (001) film, 125 nm thick atop 2 μm thick SiO_2 layer (buried oxide, BOX) on a bulk Si (001) wafer (Fig. 2.1 a)). It is thinned to 20 nm SOI by rapid thermal oxidation (RTO) at 950 °C in O_2 atmosphere for 3 hours in order to transform the top part of the SOI in SiO_2 . By dipping the oxidised samples in a HF solution (10%) and deionized water (90%) the top SiO_2 layer is removed thus exposing the remaining bottom SOI.

Island fabrication is performed in two steps: 1) patterning by photolithography and 2) solid state dewetting via high temperature annealing by following well established methods [9, 11–13]. The photolithographic patterns are obtained by spin-coating a positive, photosensitive resist exposed to a near-UV laser (375 nm wavelength) at a fluency of 2.67 mW on a spot of about 2 μm (Dilase, from Chloe). The exposed portions of the resist are removed with a developer. Finally, a plasma etching step (70 seconds in CF_4 plasma, obtained in plasma-enhanced chemical vapor deposition machine from Oxford) attacks the exposed parts of the SOI, leaving intact the protected parts. By changing the pitch between vertical and horizontal lines (e.g. from 3 to 4 μm) we define squares patches having variable size (Fig. 2.1).

Before high-temperature annealing and dewetting the SOI are first chemically cleaned with acetone, ethanol, NO_2 plasma and finally in a HF solution at 5% in de-ionised water under N_2 atmosphere in a glove box. Thus, the SOI samples are transferred to the ultra-high vacuum ($\sim 10^{-10}$ Torr) of the MBE reactor (Riber Gen-HDDHHD). The annealing process is carried out in two stages, the first one is an *in-situ* cleaning where the temperature is risen at about 650 °C for 30 min in order to remove any residual native oxide from the surface of the sample. The second annealing step induces the solid-state dewetting of the SOI when the temperature is increased at about 750 °C for 30 minutes.

2.2. Characterization

Dark field (DF) images and spectra were collected by using an optical microscope (ZEISS Axio Observer) mounting a 100X magnification objective lens (numerical aperture $\text{NA} = 0.9$) working both in bright and dark field configurations, coupled with a spectrometer and Si-based CCD linear array (Flame-T-VIS-NIR by Ocean Optics). In order to investigate individual islands, the scattered light was collected using an optical fiber (Ocean Optics multimode fiber, VIS-NIR,

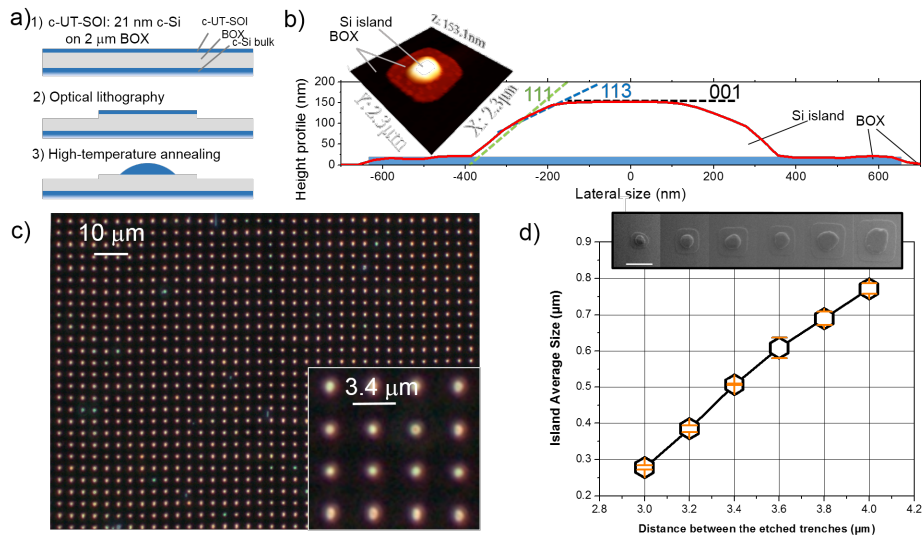


Fig. 1. a) Scheme of the main fabrication steps. 1) Crystalline, ultra-thin silicon on insulator (c-UT-SOI). 2) Optical lithography and plasma etching. 3) High temperature annealing in ultra-high vacuum. b) Atomic force microscope profile of an individual island. The main crystal facets are highlighted. The inset shows a 3D representation of the island. c) Average island size as a function between the etching pitch as obtained by scanning electron micrographs (SEM). The error bar represents the standard deviation. The line is a guide for the eyes. The top inset show a typical SEM for each etched pitch.

core diameter 200 mm) defining a lateral resolution of about $2 \mu\text{m}$.

The Si islands were imaged via scanning electron microscopy (SEM) performed with a FEI Helios 600 Nano-Lab. Micrographs were acquired using a through-the-lens detector secondary electron detector (5 kV acceleration voltage), probe current of 0.17 nA and working distance of 4.2 mm. Atomic force microscopy (AFM) was performed in non-contact mode using a CGDG DG AFM.

2.3. Simulations

We performed numerical calculations with a finite-difference time domain method (FDTD) to explain the influence of the $2 \mu\text{m}$ thick BOX layer on the island scattering spectra. We consider a hemispherical Si island with 200 nm base diameter and a refractive index $n = 4$, neglecting absorption. The Si island is excited with a broadband point-like dipole (emitting at $\lambda = 600 \text{ nm}$, with a full width at half maximum FWHM = 300 nm) positioned in the middle of the island, with polarization parallel to the BOX surface. The flux of the Poynting vector across a box sensor, a cube of side 300 nm, positioned around the exciting dipole gives us the local density of states (LDOS) of the island at the dipole position. With planar sensors $1 \mu\text{m}$ above the island positioned parallel to the BOX surface and located to different angles θ we analyze the amount of light leaving the resonators in different directions. By monitoring flux of Poynting vector at different angle θ we simulate the collect far field spectra under different acquisition angles (Fig. 4 a)).

3. Results

3.1. Templated dewetting of Si-based Mie resonators on thick BOX

At the end of the dewetting process we obtain monocrystalline and atomically smooth islands featuring the typical facets of the equilibrium shape of silicon (Fig. 2.1 b)) [7–10, 12, 14]. Large islands arrays (Fig. 2.1 c)) can be fabricated with controlled size (base diameter from about 280 to 780 nm) and regular organization [8, 11, 14, 15] (Fig. 2.1 d)). The homogeneous scattering color account for a good homogeneity, as also confirmed by the small fluctuation obtained by measuring the base size of several islands for each pitch (error bars in Fig. 2.1 d)). Islands height are in the 90 to 150 nm range, depending on the initial pitch size (see an example of large island in the AFM image in Fig. 2.1 b)). Smaller island size (e.g. featuring bright scattering at visible frequency) can be obtained by reducing the pitch of the lithographic step [8, 11, 14, 15] (not shown).

3.2. Dark-field spectroscopy of individual Mie resonators

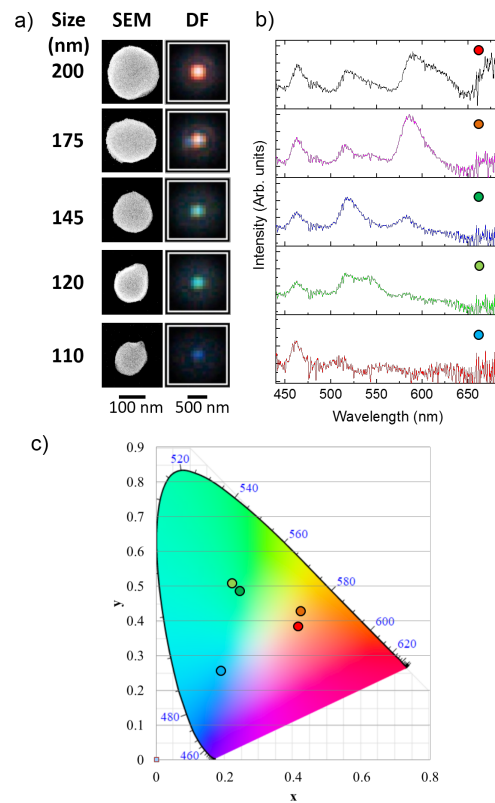


Fig. 2. a) Left column: SEM images of c-Si dewetted islands. Their corresponding size is highlighted at their left. Right panel: dark-field optical microscope images of the islands shown in the left column. b) Dark-field scattering spectrum corresponding to the islands shown in a). c) CIE chromaticity gamut for the islands shown in a) and b).

For optical spectroscopy at visible frequency (owing to the limits of our silicon-based detector) we chose a set of dewetted islands with increasing base diameter, from about 100 to 200 nm (Fig. 2 a), left column) [8, 9, 11, 14]. Slight shape asymmetries observed in some island are ascribed to a non-complete dewetting that did not lead to the final equilibrium shape of silicon

crystals. In principle, these defects could be improved with a longer annealing time or a higher temperature.

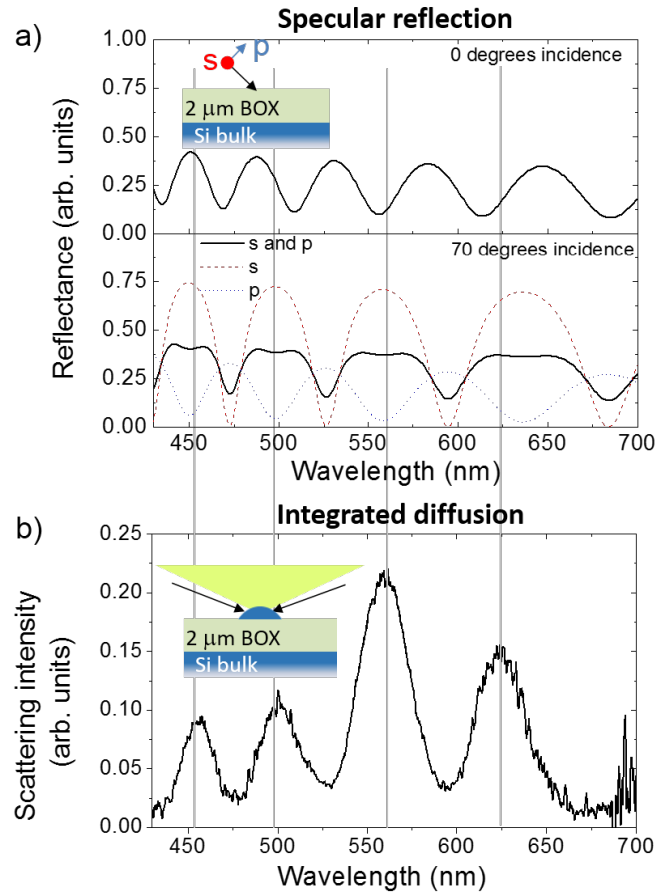


Fig. 3. a) Simulated intensity of the specular reflection for s and p polarization from a 2 μm thick SiO₂ layer (BOX) atop Si bulk. Top panel: normal incidence. Bottom panel: 70 degrees incidence. The inset highlight the geometry considered in the model. b) Dark-field scattering spectrum of a dewetted island. The inset highlights excitation and collection conditions (respectively ~70 degrees incidence and within a 64 degrees aperture cone).

Dark-field images reveal a net colorization from blue to red, when increasing particle size (Fig. 2 a), right column). The corresponding scattering spectra (normalised by the white lamp used for illumination) confirm this tendency, showing several sharp bands that increase in number at long wavelength for larger islands (Figure 2 b)) [9, 11]. These resonances share a similar spectral position and spacing between them, although some differences (up to 25-30 nm) are observed (see also Fig. 3.2) [3]. These differences can be accounted for by imperfections in the etching (locally thicker or thinner BOX), incomplete dewetting leaving some pristine c-UT-SOI nearby the islands, residual Si particles on the BOX nearby the islands, intermixing of Si and SiO₂ [16–18], and asymmetries in the dewetted islands [9].

A more precise assessment of the colourization of the scattering can be obtained by computing the CIE chromaticity gamut [11] (Fig. 2 c)). We can roughly separate the islands in three main groups: those with blue, green and orange-red colorization corresponding, respectively to an

increasing number of peaks and with increasing intensity in the red side.

3.3. Comparison with etalon reflection.

We compare the dark field scattering spectrum of a large silicon island with that of a 2 μm thick, flat SiO_2 layer atop bulk Si that mimics the BOX of the c-UT-SOI (Fig. 3.2). The specular reflection at normal incidence (0 degrees) features 5 broad peaks in the investigated spectral range having a contrast of about 3 (Fig. 3.2 a) top panel). For 70 degrees incidence (that is similar to the incidence angle of the light in dark-field configuration) the overall picture is similar but the peaks are only four, with a pronounced flattening at their top and with a lower contrast of roughly 2 (Fig. 3.2 a) bottom panel). These latter features are understood by observing the s and p polarized components of the reflection: they present maxima and minima in opposition of phase thus providing a more complex behavior with respect to the normal incidence case, where the two components are identical. Note that at this large incident angle, the s polarization is dominant with an intensity that is about double with respect to that one of the p counterpart.

The scattering spectrum of the individual Si island shows 4 peaks with a rather large intensity contrast (between 3 and 6, depending on the peak taken into account, Fig. 3.2 b)). The spectral position and spacing between the scattering peaks show marked similarities with the reflection spectrum of the flat SiO_2 layer when comparing the case of s polarization at an incidence of 70 degrees (as highlighted by the vertical gray lines in (Fig.3.2).

4. Discussion

The use of solid state dewetting for the fabrication of ordered and disordered dielectric Mie resonators made of Si(Ge) has been largely addressed in the past years. [9–11, 19–21] However, obtaining ordered arrays of simple islands or complex nano-architectures required so far, e-beam lithography and reactive etching [4, 22] or direct etching via focused ion beam [8, 11, 13, 15]. Although very precise, these methods are rather costly and eventually difficult to extend to very large surfaces. Here we replaced these approaches extending the idea of templated dewetting to conventional optical lithography. This is a much less expensive technique and can, in principle, be extended to larger areas. In our case the lateral etching resolution was rather poor, only 2 μm . Although this does not allow for a large islands density (the minimal distance between island is about 2 μm), it is worth noting that large homogeneous arrays of nano-objects can be obtained. This is possible thanks to the dewetting approach, that leads to the collapse of all the c-UT-SOI present in the original patch in a single island, thus effectively gaining a factor of the order of 10 in lateral resolution [12]. These islands can be organized in large arrays with a fine size tuning and a small spread in size (size distribution is within 10% or smaller). This size distribution is improved with respect to previous reports of Si and SiGe templated-dewetting where the etching was performed with a focused ion beam [8, 11, 13, 15] and are similar to those reported for e-beam lithography and reactive ion etching [10, 12]. Aside, we also note that the use of this self-assembly method enables the formation of monocrystalline objects with atomically smooth, faceted interfaces [7–9, 12, 14, 20]. A feature that cannot be obtained with other conventional top-down methods that always induce a more or less pronounced roughness and it may be an advantage when electronic properties are concerned [12].

Let's move to the discussion of the optical properties: we obtain colours that are comparable to those obtained with $\text{Si}_{0.7}\text{Ge}_{0.7}$ islands [11] (e.g. in terms of distance between the white colour) and to those obtained in state-of-the-art Si-based Mie resonators on silica fabricated via e-beam and reactive ion etching [22]. This is possible by combining the sharper resonances springing from the etalon effect produced by the thick BOX atop the Si bulk and the broader Mie scattering, as discussed later.

The scattering data, linked to the reflection simulation of the silica etalon, lead to the conclusion that the s polarization is playing a major role in the observed structural color. In addition to the

more intense s polarization excitation associated to the etalon effect (Fig. 3.2 a)), we expect less scattering around normal incidence for p polarization due to the electric dipole angular pattern. Indeed, a marked similarity emerges when observing the spectrum of the s polarised specular reflection at 70 degrees of incidence and the dark-field resonant scattering spectrum of the Si island. This suggests that the Mie resonator scatters in the far-field some of the light coupled in the BOX etalon modes. Note that in the geometry of the dark-field experiment, such a direct reflection is not visible (the areas nearby the islands are dark), as the numerical aperture of the excitation (about 70 degrees) exceeds that one of the detection (about 64 degrees).

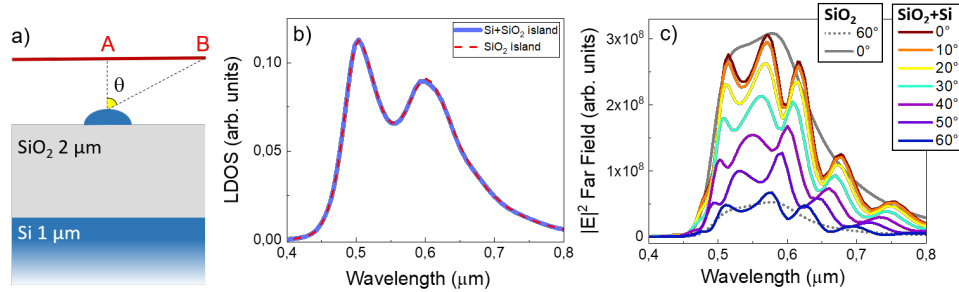


Fig. 4. a) Sketch of the simulated structure: a hemispherical Si island on a substrate formed by a $2\ \mu\text{m}$ SiO_2 layer atop $1\ \mu\text{m}$ Si layer. The red line indicates the positions of the far field sensors over which the light scattering is integrated for detection, positioned at a height such that the points A and B correspond to an acquisition angle θ of 0 degrees and 60 degrees, respectively. b) Local Density of States (LDOS) of the island sitting atop $2\ \mu\text{m}$ SiO_2 layer only (red dashed line) and on $2\ \mu\text{m}$ SiO_2 plus $1\ \mu\text{m}$ Si (blue line). c) Colored lines: far field simulated spectra acquired at different values of theta for the island atop $2\ \mu\text{m}$ SiO_2 plus $1\ \mu\text{m}$ Si substrate (from $\theta = 0$ to 60 degrees). The far field intensity from the island on $2\ \mu\text{m}$ SiO_2 layer only acquired at $\theta = 0$ 60 degrees) are reported in grey for comparison (respectively continuous and dashed lines).

In order to corroborate the picture that the scattering spectrum of the islands is strongly influenced by interference effects we compare the FDTD simulation of an hemispherical Si island placed atop of $2\ \mu\text{m}$ thick layer of SiO_2 ($n = 1.5$) plus a $1\ \mu\text{m}$ thick substrate of Si ($n = 4$) with the same island atop only $2\ \mu\text{m}$ of SiO_2 (Figure 4). The simulations conditions are different with respect to the experimental ones and their aim is to explain the underlying phenomenology. A quantitative comparison with the experimental data goes beyond the aim of this paper.

The corresponding local density of optical states (LDOS) for the two cases taken into account are dominated by two resonances at about 500 nm and 600 nm, that can be identified as the electric (ED) and magnetic dipolar (MD) modes, respectively [3, 9, 11, 19] (Figure 4 b)). The two curves, within the line thickness, are identical, suggesting that the LDOS is not significantly modified by the presence of the Si substrate. On the contrary, the far field spectra are dramatically influenced by the presence of the Si substrate. Far-field spectra collected at different points of the far field correspond to spectra detected at different acquisition angles (Figure 4 c)). The simulation without the Si substrate shows smooth spectra (solid and dashed gray lines) where is possible to identify the ED and MD modes characterized by different relative weights. The scattering intensity at 0 degrees is about six times more intense with respect to that one at 60 degrees. The far field spectra obtained by taking into account the Si substrate show a clear intensity modulation superimposed on the smooth spectra obtained without Si substrate. The spectral position of the observed fringes shifts toward the blue by increasing the detection angle. A similar effect is observed by considering the specular reflection of $2\ \mu\text{m}$ of SiO_2 on Si (not shown). These observation clearly supports the interference effect related to the scattering propagating and

interfering in the SiO₂ layer sandwiched between the Si island and the Si substrate.

5. Conclusion

In conclusion we showed that a low resolution etching method joined with solid-state dewetting can provide ordered arrays of monocrystalline Si-based islands with size control and featuring a bright Mie scattering in the visible spectral range. Coupling the scattering with etalon resonances allows to fix the position of maxima and minima and enhance the contrast between them with respect to individual dielectric antennas. In addition to this, our simple approach enables to obtain good structural colours avoiding the use of expensive and slow fabrication tools.

The photonic LDOS is not changed in our configuration, but we expect to observe effects whenever the single layer etalon would be replaced by Fabry-Perot cavity with higher finesse. Still, our design has relevant features in terms of dielectric antennas for quantum emitters [23–25]. The simulation indicates a spectral dependent emission steering of the Mie antenna when coupled with the dielectric etalon. This effect can be exploited for quantum applications, such as the selective and different directionality in exciton-biexciton cascade of a quantum emitter [26] embedded in Mie resonators.

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