

Coherent Spin Dynamics of Electrons and Holes in CsPbBr 3 Colloidal Nanocrystals

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Letter

¹ Coherent Spin Dynamics of Electrons and Holes in CsPbBr₃ Colloidal ² Nanocrystals

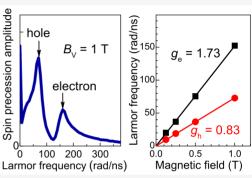
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4 **ABSTRACT:** The spin dynamics in CsPbBr₃ lead halide perovskite nanocrysts als are studied by picosecond pump-probe Faraday rotation in an external 6 magnetic field. Coherent Larmor precession of electrons and holes with spin 7 dephasing times of ~600 ps is detected in a transversal magnetic field. The 8 longitudinal spin relaxation time in weak magnetic fields reaches 80 ns at a 9 temperature of 5 K. In this regime, the carrier spin dynamics is governed by 10 nuclear spin fluctuations characterized by an effective hyperfine field strength of 11 25 mT. The Landé factors determining the carrier Zeeman splittings are $g_e =$ 12 +1.73 for electrons and $g_h = +0.83$ for holes. A comparison with a CsPbBr₃ 13 polycrystalline film and bulk single crystals evidences that the spatial 14 confinement of electrons and holes in the nanocrystals only slightly affects 15 their g factors and spin dynamics.



16 KEYWORDS: Perovskite nanocrystals, CsPbBr₃, coherent spin dynamics, electron and hole g-factors,
 17 pump-probe time-resolved Faraday rotation

ead halide perovskite nanocrystals (NCs) have only quite 18 recently joined the rich family of semiconductor NCs 19 $_{20}$ grown by colloidal synthesis.¹⁻⁶ They show a remarkable 21 quantum yield up to 90% even for bare NCs, as surface states 22 do not act detrimentally on the exciton emission efficiency. In 23 these materials, the energies of surface states and of defects and 24 impurities are not located within the band gap, which strongly 25 suppresses channels for the nonradiative recombination of 26 optical excitations in the vicinity of the band gap. This known 27 phenomenon of the perovskites is often summed up as a 28 defect-tolerant band structure. The inorganic lead halide 29 perovskites C_{sPbX_3} (X = I, Br, or Cl) allow one to tune the 30 band gap across a wide spectral range from the infrared up the 31 ultraviolet by mixing the halogen composition. Further 32 flexibility in NC band-gap tuning arises from the quantum 33 confinement of the charge carriers, whereby details of the 34 exciton fine structure are governed by the shape and size of the 35 NC.^{7–10} Simple fabrication, a high quantum yield, and tunable 36 optical properties also make lead halide perovskite NCs 37 promising materials for spintronics applications. Still, only a 38 few studies have been performed so far in this field. In 39 particular, neutral and charged excitons in single NCs were 40 identified by their Zeeman splitting in magnetic fields,¹¹ and 41 the importance of the Rashba effect was elaborated.¹² Strong 42 magnetic fields of 30 T allowed one to identify the influence of 43 negatively charged excitons (trions) and dark excitons on the 44 optical properties of NC ensemble.¹³ Anisotropic exciton 45 Zeeman splitting in magnetic field was observed,¹⁴ the 46 coherent spin dynamics of holes were explored in CsPbBr₃

NCs, 15 and the picosecond spin dynamics of carriers were $_{47}$ reported in CsPbI₃ NCs. 16 48

To understand and interpret spin-dependent phenomena, ⁴⁹ precise knowledge of the values and signs of the charge-carrier ⁵⁰ g-factors is of key importance. Also, the characteristic time ⁵¹ scales of the spin dynamics (longitudinal spin relaxation time, ⁵² T_1 , spin coherence time, T_2 , and ensemble spin dephasing ⁵³ time, T_2^*) are needed. It is important that these parameters are ⁵⁴ measured directly and preferably on the same samples. Even ⁵⁵ for bulk lead halide perovskites, the published information is ⁵⁶ still fragmentary, often providing controversial conclu-⁵⁷ sons.¹⁷⁻²⁵

Time-resolved pump–probe Faraday/Kerr rotation is an 59 established technique for fulfilling the previously defined $_{60}$ conditions for comprehensive spin physics studies of semi- $_{61}$ conductors, 26,27 as it provides direct access to *g*-factors and the $_{62}$ spin relaxation times. For example, it was successfully applied $_{63}$ to colloidal II–VI semiconductor NCs. $^{28-31}$ Furthermore, its $_{64}$ potential was recently demonstrated for CsPbBr₃ single $_{65}$ crystals³² and NCs¹⁵ and for CH₃NH₃PbCl_xI_{3-x} polycrystalline $_{66}$ films.³³

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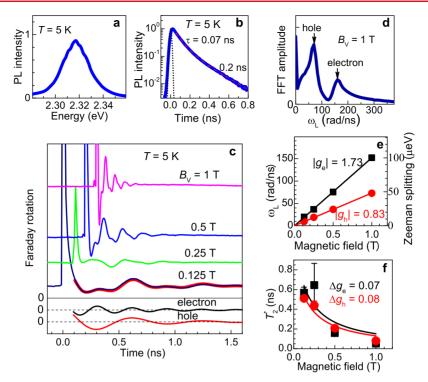


Figure 1. (a) Photoluminescence spectrum of CsPbBr₃ NCs. (b) Time-resolved recombination dynamics measured at the maximum of the PL line. The black dotted line shows the instrumental response function, whereas the red dashed line shows the double-exponential fit. (c) Coherent spin dynamics measured at 2.330 eV in different magnetic fields applied in Voigt geometry. The curves are shifted vertically and horizontally for clarity. The thick red line is a fit to the data at $B_V = 0.125$ T with eq 1 and $S_e/S_h = 0.5$. In the bottom, the black and red lines are the two components of the fit describing the electron and hole spin precession. (d) Fast Fourier transform (FFT) spectrum of the Faraday rotation dynamics at $B_V = 1$ T. (e) Magnetic field dependencies of the Larmor precession frequencies (corresponding Zeeman splittings are shown on the right axis) for electrons and holes. The error bars are within the data points. Solid lines are linear fits, giving $|g_e| = 1.73$ and $|g_h| = 0.83$. (f) Magnetic field dependencies of the spin dephasing times for electrons and holes. Lines are fits according to eq 2, with parameters given in the text.

In this Letter, we investigate the coherent spin dynamics of 9 electrons and holes in CsPbBr₃ NCs using time-resolved 70 pump-probe Faraday rotation (ppFR). We measure the 71 electron and hole *g*-factors, their spread within the NCs 72 ensemble, and the temperature dependence of the transverse 73 and longitudinal spin relaxation times. The carrier interaction 74 with the nuclear spins and its role in the spin dynamics are 75 studied. The spin dynamic parameters in CsPbBr₃ NCs are 76 compared with those in a polycrystalline film and a bulk single 77 crystal.

78 OPTICAL PROPERTIES OF NANOCRYSTALS

79 We study CsPbBr₃ NCs dispersed in polymethyl methacrylate 80 (PMMA) and spin-coated on a glass coverslip. (For synthesis 81 details, see Section S2 in the Supporting Information.) The 82 NCs have a cubic shape with a typical size of 10 nm, which 83 slightly exceeds the exciton Bohr diameter of 7 nm. Therefore, 84 the excitons experience a rather weak size quantization. 85 Transmission electron microscopy images and optical spectra 86 of similar CsPbBr₃ NCs can be found in ref 13. It should be 87 noted that whereas CsPbBr₃ NCs are known to crystallize in 88 the cubic phase,³ a recent report shows that they can exhibit 89 tetragonal and orthorhombic phases at cryogenic temper-90 atures.¹¹ Lower crystal symmetry will likely result in an 91 anisotropy of the charge-carrier *g*-factors.

⁹² To characterize the optical properties of the studied ⁹³ CsPbBr₃ NCs, we measured the photoluminescence (PL) ⁹⁴ spectrum and recombination dynamics at a temperature of T =⁹⁵ 5 K; see Figure 1a,b. The results are typical for CsPbBr₃ NCs,

as reported in literature; see, for example, ref 13 and references 96 therein. The PL line has maximum at 2.317 eV and a full width 97 at half-maximum of 29 meV. It shows a fast recombination, 98 two-component dynamics with a decay of 0.07 and 0.2 ns 99 (Figure 1b), in line with the 0.2 ns decay measured in single 100 NC experiments.¹¹ We have checked that the application of 101 the magnetic field up to 6 T only slightly elongates the PL 102 decay. These decay times are considerably shorter than the 103 exciton decay time of 0.9 ns in CsPbBr₃ bulk crystals.³² In 104 general, the emission of an NC ensemble can be contributed 105 by neutral and charged NCs. Therefore, the PL can contain the 106 emission of neutral, negatively charged, and positively charged 107 excitons. The ratio of their populations depends on the 108 synthesis protocol, the sample preparation, and the optical $_{109}$ excitation conditions. In our previous study, 13 we measured $_{110}$ polarized PL and recombination dynamics in high magnetic 111 fields up to 30 T of similar CsPbBr₃ NCs and concluded that at 112 liquid helium temperatures, the emission is mainly contributed 113 by negatively charged excitons with some contribution from 114 neutral excitons. The present study using the ppFR 115 technique²⁷ allows us to gain more information on the NC ¹¹⁶ charging,³⁰ as will be discussed as follows. ¹¹⁷

COHERENT SPIN DYNAMICS OF ELECTRONS AND 118 HOLES IN NANOCRYSTALS 119

To address the carrier spin dynamics, we use ppFR; see the 120 Methods in the Supporting Information. Here circularly 121 polarized pump pulses optically induce a carrier spin 122

123 polarization, while the linearly polarized probe pulses detect its 124 time evolution by varying their delay relative to the pump.

Figure 1c shows the spin dynamics measured for different magnetic fields applied in the Voigt geometry (field orientation perpendicular to the collinear pump and probe beams). The Faraday rotation amplitude oscillates with time, reflecting the coherent spin precession at the Larmor frequency about the magnetic field. The precession frequency increases with magnetic field strength, B_{V} , growing from 0.125 to 1 T. Simultaneously, the signal decay accelerates. The modulations in the oscillatory signal arise from the contribution of two components with different precession frequencies, as confirmed in the fast Fourier transform (FFT) spectrum of the dynamics in Figure 1d. Indeed, the Faraday rotation dynamics rot be well-fitted by the sum of two exponentially decaying socillations

$$A_{\rm FR} = S_{\rm e} \cos(\omega_{\rm L,e}t) \exp(-t/T_{2,e}^*) + S_{\rm h} \cos(\omega_{\rm L,h}t)$$
139
$$\exp(-t/T_{2,h}^*)$$
(1)

140 Here $S_{e(h)}$ is the amplitude of the electron (hole) spin signal, 141 $\omega_{L,e(h)}$ is the Larmor precession frequency of electrons (holes), 142 and $T_{2,e(h)}^*$ is the electron (hole) spin dephasing time. The 143 contributions of the two dynamics components at $B_V = 0.125$ 144 T are shown separately in the bottom of Figure 1c. The fit 145 gives $T_{2,e}^* \approx 0.6$ ns and $T_{2,h}^* \approx 0.5$ ns with comparable 146 amplitudes of the electron and hole signals, $S_e/S_h \approx 0.5$. The 147 Larmor precession frequencies of both components increase 148 linearly with magnetic field (Figure 1e). By fitting them with 149 $\omega_{L,e(h)} = |g_{e(h)}| \mu_B B_V / \hbar$, where μ_B is the Bohr magneton and $g_{e(h)}$ 150 is the electron (hole) g factor, we evaluate $|g_e| = 1.73$ and $|g_h| =$ 151 0.83.

¹⁵² We attribute the high-frequency component with the *g*-¹⁵³ factor of 1.73 to the electron and the low-frequency ¹⁵⁴ component with the *g*-factor of 0.83 to the hole by analogy ¹⁵⁵ to bulk CsPbBr₃ perovskites.³² Note that our experimental ¹⁵⁶ technique is not sensitive to the *g*-factor sign, which should be ¹⁵⁷ determined using an alternate method. We use here the ¹⁵⁸ knowledge of the exciton *g*-factor, $g_X = +2.4$, that we measured ¹⁵⁹ in CsPbBr₃ NCs,¹³ and the fact that in lead halide perovskites, ¹⁶⁰ $g_X = g_e + g_h$. This allows us to conclude that in the studied ¹⁶¹ NCs, both electron and hole *g*-factors are positive, similar to ¹⁶² CsPbBr₃ bulk crystals;³² for details see Section S4 in the ¹⁶³ Supporting Information.

Let us clarify now the important question of whether neutral 164 165 or singly charged NCs are responsible for the ppFR signal. In 166 the case of neutral NCs, the signal should originate from the 167 photogenerated electrons and holes bound to neutral excitons. 168 There are two arguments that let us exclude this neutral 169 exciton scenario. First, the measured spin dynamics should 170 then be limited by the exciton recombination time, which is 0.2 171 ns for the studied NCs (Figure 1b). The measured spin 172 dephasing times of electrons (0.6 ns) and holes (0.5 ns) are 173 much longer than the exciton recombination time. The second 174 argument is related to the exciton fine structure. Magneto-175 optical studies of emission from single CsPbBr₃ NCs show that 176 the zero-field exciton exchange splitting of neutral excitons is $177 \sim 0.6 \text{ meV}^{11}$ and that most NCs at low temperatures have 178 either tetragonal or orthorhombic crystal symmetry. Note that 179 in rather similar FAPbBr₃ NCs the splitting is found to be in 180 the range from 0.3 to 1.7 meV.¹⁰ The exciton exchange 181 interaction couples electron and hole spin so that the spins of the electrons and holes forming the exciton do not precess 182 independently. This becomes possible only in strong magnetic 183 fields where the carrier Zeeman splitting exceeds the exchange 184 energy and the spin coupling is broken. One can see from the 185 right scale of Figure 1e that in the studied NCs, we detect 186 electron and hole spin precession in the low field regime with 187 Zeeman splittings of ~0.01 meV, that is, much smaller than the 188 neutral exciton exchange splitting. We conclude that neutral 189 NCs (i.e., neutral excitons) cannot contribute notably to the 190 coherent spin dynamics measured in the ppFR experiment. 191 Note that this conclusion contradicts the one of ref 15 that the 192 measured hole spin beats can be associated with a hole bound 193 to an exciton. 194

In NCs charged by either a single electron or a single hole, 195 the ppFR signal is contributed by the spin dynamics of these 196 resident carriers, limited only by their spin relaxation times. 197 The spin coherence of the resident carriers is generated via 198 photoexcitation of a charge exciton (trion) state, as studied in 199 great detail for (In,Ga)As/GaAs epitaxially grown quantum 200 dots.^{27,34} The exchange splitting is canceled in the trion and 201 obviously also cannot occur for the resident carrier left in an 202 NC after trion recombination. Therefore, the magnetic field 203 dependence of the Larmor precession frequency is expected to 204 be linear with zero energy offset at zero magnetic field. This is 205 exactly what we find in the experiment in Figure 1e for both 206 electrons and holes. On the basis of these arguments, we can 207 safely conclude that the ppFR signals originate from singly 208 charged NCs, both positively and negatively charged. From $S_e/_{209}$ $S_{\rm h} \approx 0.5$, we suggest that the fractions of negatively and 210 positively charged NCs are comparable, with some preference 211 for the positively charged NCs. It was shown that in CsPbBr₃ 212 and FAPbBr₃ NCs photocharging results in negative and 213 positive loading of the NCs.^{35,36} This is consistent with the 214 Fermi level being close to the midpoint of the gap in Cs and 215 FA lead halide NCs, as demonstrated by photoemission 216 spectroscopy.37,38 217

The spin dephasing times, of both electrons and holes, show 218 strong magnetic field dependences; see Figure 1f. The 219 dynamics shorten from 0.6 to 0.05 ns for electrons and from 220 0.5 to 0.08 ns for holes with increasing field up to 1 T. This 221 decrease is provided by the spin dephasing in the ensembles 222 with a dispersion of the *g*-factors, which translates to a spread 223 of the Larmor precession frequencies. It is described by the 224 following equation 225

$$\frac{1}{T_{2,e(h)}^{*}(B_{V})} = \frac{1}{T_{2,e(h)}^{*}(0)} + \frac{\Delta g_{e(h)} \mu_{B} B_{V}}{\hbar}$$
(2) 226

Here $\Delta g_{e(h)}$ is the dispersion of electron (hole) *g*-factors. 227 $T_{2,e(h)}^*(0)$ is the spin dephasing time at zero magnetic field, 228 which is limited by the spin coherence time, $T_{2,e(h)}$, of the 229 individual carriers and the spin relaxation mechanisms not 230 related to the *g*-factor dispersion, for example, the hyperfine 231 interaction with nuclear spin fluctuations. This equation gives 232 reasonable fits to the experimental dependencies of $T_{2,e(h)}^*(B_V)_{233}$ with $T_{2,e}^*(0) = 1.2$ ns and $T_{2,h}^*(0) = 1.0$ ns and *g*-factor 234 spreads of $\Delta g_e = 0.07$ and $\Delta g_h = 0.08$. Note that the measured 235 *g*-factor spreads are about two times larger than those in bulk 236 CsPbBr₃ single crystals.³² This may be related to the spread of 237 NC sizes and shapes, which determines the *g* factor and its 238 anisotropy. f2

Oscillating signals in the Faraday rotation dynamics are
 241 detectable up to temperatures of 35 K, as shown in Figure 2.

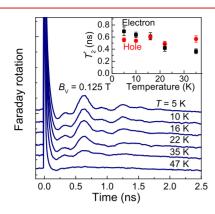


Figure 2. Dynamics of Faraday rotation signal in CsPbBr₃ NCs measured at 2.330 eV for different temperatures in $B_V = 0.125$ T. Data are shifted vertically for clarity. The inset shows the temperature dependence of the spin dephasing time for electrons and holes.

242 The amplitudes of both the electron and hole spin signals 243 gradually decrease with increasing temperature. That means 244 that the decrease in the signal amplitude is provided by a 245 decreasing efficiency of the optical spin polarization. On the 246 contrary, the spin dephasing times, shown in the inset of Figure 247 2, change only weakly in this temperature range. The hole dephasing time remains almost constant at $T_{2,h}^* = 0.5$ ns, whereas the electron time, $T_{2,e}^*$, slightly decreases from 0.7 to 249 250 0.4 ns. This is in contrast with the strong temperature $_{251}$ dependence of $T_{2,e}^*$ in bulk CsPbBr₃,³² where temperature 252 induces the delocalization of carriers and the activation of spin 253 relaxation mechanisms related to the spin-orbit coupling. On 254 the contrary, in NCs, carriers are localized at any temperature. 255 The other difference in NCs from bulk CsPbBr₃, presumably 256 related to the localization, is the much more homogeneous 257 spin dynamics for NCs (dephasing times and ratio between 258 electron and hole contributions) for different positions on the 259 sample.

LONGITUDINAL SPIN RELAXATION OF CARRIERS IN NANOCRYSTALS

262 Let us turn to experiments in the Faraday geometry where the 263 external magnetic field, $B_{\rm F}$, is applied along the light wave 264 vectors. Consequently, the optically induced spin polarization of carriers along the field direction does not show Larmor 265 precession. This allows one to measure the longitudinal spin 266 relaxation time, T_1 , and gather information on the nuclear spin 267 fluctuations. 268

The magnetic field dependence of the Faraday rotation 269 signal measured at a negative time delay of -1 ns is shown in 270 Figure 3a. The finite signal amplitude implies that the spin 271 f3 relaxation time of carriers exceeds the repetition period of the 272 laser pulses, $T_{\rm R}$ = 13.1 ns. In fact, as we will show, it exceeds 273 several $T_{\rm R}$ periods and corresponds to a cumulative effect due 274 to repeated pulse application. In this experiment, the laser 275 helicity is kept constant, but the laser intensity is modulated 276 with frequencies in the f = 0.3-6 MHz range. One can see that 277 at the low modulation frequency of 0.3 MHz (black line), the 278 signal amplitude has a minimum at zero magnetic field and 279 increases with the field increase. It thus corresponds to a so- 280 called polarization recovery curve (PRC).³⁹ The half width at 281 half-maximum of the PRC equals 25 mT, which is the 282 characteristic field of nuclear spin fluctuations acting on the 283 carriers via the hyperfine interaction. In our recent study of 284 CsPbBr₃ crystals, we showed experimentally and theoretically 285 that in lead halide perovskites, the holes in the valence band 286 have a considerably stronger hyperfine interaction compared 287 with the conduction band electrons.³² Therefore, we suggest 288 that the PRC signal in CsPbBr3 NCs is mainly contributed by 289 the hole-nuclear hyperfine interaction. 2.90

When the pump modulation frequency is increased to 6 291 MHz, the PRC amplitude decreases; see Figure 3a. This is 292 evidence that the modulation period, 1/f, is comparable to T_1 293 in this case, and the spin accumulation loses its efficiency. 294 Therefore, we used the spin inertia method to measure the 295 spin relaxation time, T_1 , in the longitudinal magnetic 296 field.^{32,39,40} In addition, the detected signal is retarded by the 297 phase, ϕ , which we also measure; see the Methods in the 298 Supporting Information for details. Figure 3b shows the PRC 299 amplitude and tan ϕ as a function of the modulation frequency, 300 which gives us two ways to evaluate T_1 . Fitting the frequency 301 dependence of the PRC amplitude with the spin inertia 302 equation (Section S3 in the Supporting Information), we 303 obtain $T_1 = 83 \pm 9$ ns. The fit of the tan ϕ dependence on f 304 with a linear equation (Section S4 in the Supporting 305 Information) gives $T_1 = 56 \pm 5$ ns. This difference in T_1 is 306 caused by the nonmonoexponential spin relaxation dynamics: 307 The amplitude of the spin polarization is more sensitive to the 308 slow component in the dynamics, whereas the retardation 309 phase is more dependent on the fast component. We note that 310

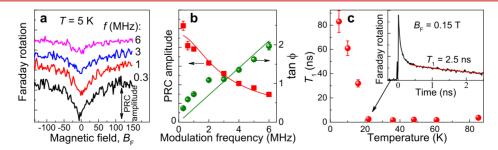


Figure 3. Spin dynamics of CsPbBr₃ NCs measured in longitudinal magnetic fields (Faraday geometry). (a) Polarization recovery curves measured at different pump modulation frequencies. Data are shifted vertically for clarity. T = 5 K. (b) PRC amplitude (red squares) and tangent of the retardation phase (green circles) as a function of the modulation frequency. Lines show fits with eqs S3 and S4 in the Supporting Information to the experimental data. T = 5 K. (c) Temperature dependence of the longitudinal spin relaxation time, T_1 . Inset shows the Faraday rotation signal dynamics in the longitudinal magnetic field, $B_F = 0.15$ T, at T = 22 K.

t1

f4

Table 1. Spin Parameters Measured for Dif	fferent CsPbBr ₃ Perovskite Samples"
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	nanocrystals	polycrystalline film	bulk ³²
electron g-factor, g _e	+1.73	+1.78	+1.96
hole g-factor, g _h	+0.83	+0.77	+0.75
exciton g-factor, g_X	$+2.4^{13}$		+2.35
electron spin dephasing time, $T^*_{2,e}$	0.6-0.7 ns	0.25 ns	0.9-5.2 ns
hole spin dephasing time, $T^*_{2,\mathrm{h}}$	0.5 ns	0.4 ns	0.7-1.9 ns
longitudinal spin relaxation time, T_1	56-83 ns		32–53 ns

^{*a*}Results for bulk CsPbBr₃ (single crystal) are taken from ref 32. T_2^* values in NCs and bulk are given for $B_V = 0.125$ T, and those in film are given for $B_{\rm V} = 0.25$ T.

311 the evaluated T_1 values for NCs are a few tens of nanoseconds 312 longer than T_1 for the corresponding bulk perovskites;³² see 313 Table 1.

The longitudinal spin relaxation time, T_1 , determined from 314 315 the frequency dependence of the PRC amplitude is shown in 316 Figure 3c as a function of temperature. It rapidly decreases with temperature and saturates at ~ 2 ns above 20 K. At 317 318 temperatures higher than 20 K, T_1 was measured by fitting the 319 decay of the Faraday rotation signal with a single exponential 320 function; see the inset of Figure 3c.

COHERENT SPIN DYNAMICS IN 321 POLYCRYSTALLINE FILM 322

323 We also studied the coherent spin dynamics in a CsPbBr₃ 324 polycrystalline film. Its PL spectrum (Figure 4a) has an

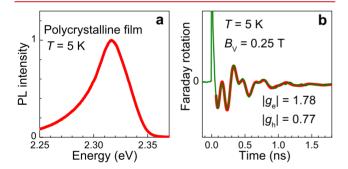


Figure 4. (a) Photoluminescence spectrum of the CsPbBr₃ polycrystalline film. (b) Coherent spin dynamics measured in the film at 2.330 eV for $B_{\rm V}$ = 0.25 T. The panel gives experimental data (green line) and fit to the data (red line) with eq 1 and $S_e/S_h = 0.75$.

325 asymmetric shape with a low-energy tail, similar to the low-326 temperature emission of bulk CsPbBr₃.³² The spin precession 327 monitored by ppFR is given in Figure 4b at $B_{\rm V} = 0.25$ T and T 328 = 5 K. Similar to the NCs, it contains two Larmor precession 329 frequencies, where a fit with eq 1 gives the following $_{330}$ parameters: $|g_e| = 1.78$, $|g_h| = 0.77$, $T^*_{2,e} = 0.25$ ns, and $_{331} T^*_{2,h} = 0.4$ ns. It is reasonable to suggest that similar to 332 CsPbBr₃ bulk and NCs, $g_e > 0$ and $g_h > 0$ in the film.

SUMMARY OF *q*-FACTORS AND SPIN 333 **RELAXATION TIMES IN CsPbBr₃ PEROVSKITE** 334 **STRUCTURES** 335

336 It is instructive to compare the g-factor values for CsPbBr₃ 337 perovskite NCs, polycrystalline film, and bulk crystals (Table 338 1). We note a small but monotonic variation of the g-factors 339 when going from NCs to film and further to bulk. The rather 340 close g-factor values are related to the weak quantum 341 confinement in the studied NCs, which is also confirmed by

their close energy positions of the PL emission. The behavior is 342 in line with the well-known relation between the optical 343 transition energy and the electron g-factor for semiconduc- 344 tors.⁴¹⁻⁴³ The small effect of confinement on the spin 345 properties of the compared samples is further confirmed by 346 the similar range of spin relaxation times in NCs and bulk 347 perovskites.³² 348

In conclusion, we performed a detailed study of the electron 349 and hole spin dynamics in CsPbBr3 perovskite NCs. Electron 350 and hole g-factors of +1.73 and +0.83, respectively, have been 351 measured from spin beats in the transverse magnetic field. The 352 spin dephasing times T_2^* of ~ 1 ns considerably exceed the $_{353}$ exciton recombination time, which has allowed us to attribute 354 the measured spin dynamics to NCs singly charged with an 355 electron or a hole. In the longitudinal magnetic field, a spin 356 relaxation time T_1 exceeding 80 ns has been measured. The 357 influence of the carrier-nuclear hyperfine interaction, which 358 controls the carrier spin dynamics at zero and weak magnetic 359 fields, has been identified. Our study underlines that perovskite 360 NCs have promising spin-dependent properties, very much 361 comparable to the comprehensively studied epitaxially grown 362 III-V quantum dots and colloidally synthesized II-VI NCs. 363

ASSOCIATED CONTENT 364

Supporting Information

365

The Supporting Information is available free of charge at 366 https://pubs.acs.org/doi/10.1021/acs.nanolett.1c03292. 367

Details of experimental methods, synthesis and charac- 368 terization of CsPbBr3 NCs and polycrystalline film, and 369 Zeeman Hamiltonian and g-factors of charge carriers and 370 excitons (PDF) 371

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398 Notes

399 The authors declare no competing financial interest.

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