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Characterization and performance within the WGS reaction of Cu catalysts obtained from hydrotalcites

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Abstract

In large-scale industry, the WGS reaction is the most widely used method for the purification of hydrogen obtained from natural gas. This is a process that hinges on factors such as temperature, pressure, type of catalyst, etc. Experimentation with copper-based catalysts synthesized from HTLc-type solids can lead to enhancements in the hydrogen production processes, due to the thermal and mechanical properties of the solids that can be obtained by the calcination of such materials. In this work, copper-based solids were obtained from hydrotalcites, which were characterized through various techniques and evaluated within the WGS reaction at different temperatures. Hydrotalcite-type precursors displayed thermal behaviors that depend on their composition, whereby higher temperatures become necessary for the collapse of the HTLc structure in the solids with higher zinc and aluminum content. The catalysts were active in the reaction under study with $C_{0.50}Z_{0.25}A_{0.25}$ being the most active material due to the higher content of metallic copper on the surface. The most effective material in terms of metallic copper content present on the surface was $C_{0.33}Z_{0.33}A_{0.33}$, in which the occurrence of CuO was not evinced after its use, in addition to displaying the smallest particle size of Cu⁰ as well as, possibly, the existence of Cu-Zn interactions on the surface.

Introduction

Catalytic reactions for hydrogen production are of great importance for the industrial synthesis of ammonia, hydrogenation of vegetable oils, and petrochemical processes as well as in the production of methanol, among others [1,2]. Currently, hydrogen offers a high potential for the generation of energy in fuel cells, mainly used in the transport sector [3-9]. Most of the industrial hydrogen is obtained from methane reform, and, to a lesser extent, in alternative or complementary processes such as partial oxidation, dry and autothermal methane reforms, methanol reform and biomass reform [10-17]. During the generation of hydrogen at an industrial level, a gaseous mixture is produced, which contains, among other compounds, carbon monoxide that can act as a poison for catalysts used in subsequent reactions [2]. Given this circumstance, it became necessary that the water gas shift (WGS) reaction be used as part of the hydrogen generation process from methane, which, in addition to eliminating carbon monoxide, increases hydrogen production. Before the use of reform and WGS reactions, hydrogen was produced through the addition of strong acids to metals or by means of water electrolysis [15,18]; however, these methods were difficult to control and failed to meet the growing demand for hydrogen. This need led to the development of research aimed at obtaining hydrogen through heterogeneous catalysis processes, which led to the use of nickel-based catalysts for the methane reform, and of iron-chromium for the WGS reaction. [2].

Presently, the WGS reaction is carried out in two stages: high temperature shift (HTS) and low temperature shift (LTS) in temperatures ranging between 350-450 °C and 150-250 °C, respectively [2,19-22]. The catalysts used in HTS are made from Fe_2O_4 , which contains, in a lower proportion, Cr_2O_3 that acts as a textural promoter increasing both mechanical and thermal resistance, besides avoiding the sintering of the catalyst [20,23,24]. The LTS process is necessary to obtain a gaseous mixture in which the remaining carbon monoxide concentration is negligible [2,24]. In this stage, copper-based catalysts supported on Al_2O_3 are utilized to improve physical properties and to increase the dispersion of copper on the surface of the material. In addition to these catalysts, ZnO is incorporated, which acts as a structural promoter leading to increased activity and selectivity of the catalyst and performing additionally as an inhibitor of poisoning processes. [24-26].

The WGS reaction shows thermodynamic and kinetic limitations that must be considered to ensure high conversions and selectivity; therefore, the use of sintering-resistant active solids is required. This has led to the study of new materials based on gold [3,26], cerium [24,26], manganese [27], iron [28-31], nickel [32], as well as the evaluation of catalysts containing molybdenum or yttrium as dopants [33, 34], among others. By the same token, catalysts that show activity at intermediate temperatures (MTS) have been studied [35-41]. Among the materials most studied in recent years in heterogeneous

catalysis are the precursors of the hydrotalcite (HTLc) or layered double hydroxides (LDH) type, which can be made up of various metal cations and anions that confer different properties. HTLc by calcination processes can generate a wide variety of solids whose characteristics depending on their composition, presenting properties such as homogeneous particle distribution, as well as thermal and mechanical stability of solids [42,43].

In this work, catalysts of the Cu/ZnO/Al₂O₃ type prepared from hydrotalcites were studied. The materials obtained were characterized by using different instrumental techniques and evaluated in the WGS reaction at temperatures between 150 and 350 °C.

Experimental

The catalysts were obtained from the calcination of hydrotalcites prepared by the low saturation method at a constant pH. For the synthesis of hydrotalcites, saline solutions based on copper nitrate were used. [Cu(NO₃)₂·6H₂O], zinc nitrate [Zn(NO₃)₂·6H₂O] and aluminum [Al(NO₃)₃·9H₂O], as well as alkaline solutions containing potassium hydroxide (KOH) and potassium carbonate (K₂CO₃) [28,30]. The molar ratios of the Cu²⁺ and Zn²⁺ cations, as well as (CO₃)²⁻ and OH⁻ anions were adjusted based on the fixed use of 0.05 moles of the Al³⁺ cation in order to synthesize the Cu_{0.50}Zn_{0.25}Al_{0.25}(OH)₂(CO₃)_{0.125}, Cu_{0.37}Zn_{0.37}Al_{0.25}(OH)₂(CO₃)_{0.125}, Cu_{0.44}Zn_{0.22}Al_{0.33}(OH)₂(CO₃)_{0.125} and Cu_{0.33}Zn_{0.33}Al_{0.33}(OH)₂(CO₃)_{0.125} materials, in which the subscripts indicate the molar ratios of the species present in each material. During the synthesis, the saline and alkaline solutions were mixed in a controlled manner by maintaining the pH at 8.3-8.4 throughout the process. The solids formed were filtered, washed with deionized water, and dried at 60 °C for 24 h. Subsequently, the materials were calcined at 500 °C under an air flow of 100 ml / min for 4 hours. The solids obtained were characterized by thermogravimetric analysis (TGA-DTA), X-ray diffraction (XRD), Fourier transform infrared spectroscopy (FTIR), specific surface area measurement by the BET method as well as porosity by the BJH method, reduction at a set temperature (RST), copper dispersion measurements and X-ray photoelectron spectroscopy (XPS).

The TGA and DTA analyzes were performed on the precursors of the hydrotalcite type, using a TA Instruments SDT 2670. In these experiments 15 mg of samples were used, which were heated at a rate of 10 °C·min⁻¹ from 30 to 800 °C, under air flow of 50 mL·min⁻¹.

The X-ray diffraction (XRD) patterns of hydrotalcites were recorded in a Bruker D8 Advance with CuKα radiation (λ= 1.54059 Å) generated at 40 kV and 40 mA in the 5-80° (2θ) range. The X-ray

diffraction patterns of the calcined (10-80°, 2 θ range) and spent catalysts (20-80°, 2 θ range) were recorded in Rigaku Multiflex, with CuK α radiation (λ = 1.54059 Å), generated at 40 kV and 40 mA. The specific surface area and porosity of calcined solids were determined by BET (Brunauer-Emmett-Teller) and BJH (Barrett-Joyner-Halenda) methods respectively. These parameters were measured through adsorption/desorption of nitrogen at 196.15 °C, using a Micromeritics ASAP 2010 instrument. Prior to analysis, 0.2 g of samples were heated up to 200 °C under vacuum (2 mm Hg) to remove water and volatile matter.

The temperature-programmed reduction experiments were performed in a Zeton Altamira AMI 90 instrument, equipped with TCD detector. In this analysis, 0.05 g of samples were heated up to 300 °C to remove water and impurities and subsequently cooled down to 25 °C. During analysis, the samples were heated from 25 to 500 °C under 5% H₂/Ar flow of 50 mL·min⁻¹.

X-ray photoelectron spectroscopy (XPS) analyses were performed on samples C_{0.44}Z_{0.22}A_{0.33} and C_{0.33}Z_{0.33}A_{0.33}, in a Kratos Axis Ultra DLD spectrometer, using monochromated AlK α rays as the excitation source. Survey scans were performed using a pass energy of 160 e, and the specific regions (O1s, C1s, Cu2p, Zn2p, Al2p) were recorded at a pass energy of 20 eV to obtain a better peak resolution. In XPS calibration of the energies requires information on the origin of the energy scale (0 eV) that is provided (on conducting materials) by recording the valence band and measuring the Fermi level of the analyzer [44,45]. As it is unreliable to try to measure the Fermi energy level of the analyzer with poorly conducting materials such as ours we can minimize measurement inaccuracies of the binding energies (BEs) by calibrating all peaks with respect to an internal reference to the material. This reference can be either a building block of the material [46,47] or adventitious contamination carbon [48,49]; in our case we would ideally consider the Al2p peak but unfortunately there is a contribution of copper (Cu3p) near Al2p peak which renders the precise positioning of this peak rather hazardous. We have thus chosen the second solution and referenced the BEs to the principal component of the peak-fitted C1s envelope, arbitrarily assigned the often used [50,51] and consistently independently measured with our spectrometer [49] value of 284.6 eV, characteristic of C-(C,H) bonds. Shirley background subtraction and peak fitting to relevant Gaussian-Lorentzian functions were realized on each acquired spectrum prior to any measurements.

For dispersion and metallic area of copper measurements, the solids were previously reduced using a 5% H₂/N₂ flow, from 25 to 350 °C and heating rate of 10 C min⁻¹. Once this step was done, the samples were maintained for 1 h at 350 °C, registering the hydrogen consumption. These conditions were used to simulate those used during the copper catalysts reduction prior to the catalytic evaluation. Subsequently the solids were then cooled to 90 °C and oxidized with nitrous oxide in a

flow of $30 \text{ mL} \cdot \text{min}^{-1}$, for 80 min. The ratio of hydrogen consumption before and after oxidation nitrous oxide was used to calculate the copper dispersion. The metallic area of copper of the solids was determined by the number of copper atoms per unit surface area ($1.47 \times 10^{19} \text{ atom Cu/m}^2$).

The turnover frequency (TOF) was calculated using the number of moles of carbon monoxide converted as a function of the number of moles of Cu^0 obtained from the metallic area by the time unit.

For the activation of the copper catalysts, 100 mg of the solids were reduced using a $100 \text{ mL} \cdot \text{min}^{-1}$ flow of 10 % H_2/N_2 mixture at 350°C for 1 hr. After this step, the solids were evaluated in the WGS reaction of 150 to 350°C range and 1 atm of pressure. In catalytic evaluation, a flow of water vapor/ CO =10 molar ratio was used. This mixture was obtained by passing a 10 % CO/N_2 flow of $60 \text{ mL} \cdot \text{min}^{-1}$ for water vapor generated through of the dosage of water liquid using a pump in a vaporizer at 303°C . The quantification of gases used and generated during the reaction, was performed using a gas chromatograph Varian Model 3800, equipped with a Porapak column and molecular sieve and two TCD detectors.

Results and Discussion

The hydrotalcite type structure (JCPDS card 89-0460) was confirmed in all solids through the peaks recorded in the X-ray diffraction patterns shown in Figure 1. The solids with the $\text{Cu} / \text{Zn} = 1$ molar ratio presented more defined peaks than those present in materials with higher copper content, where the peak corresponding to the crystallographic plane was observed in the ($\text{Cu}_{0.37}\text{Zn}_{0.37}\text{Al}_{0.25}$ and $\text{Cu}_{0.33}\text{Zn}_{0.33}\text{Al}_{0.33}$) solids (110). In the $\text{Cu}_{0.50}\text{Zn}_{0.25}\text{Al}_{0.25}$ material, a peak attributable to the $\text{Cu}(\text{OH})_2$ phase was recorded, which is probably due to the Jahn-Teller effect seen in high copper hydrotalcites [52,53].

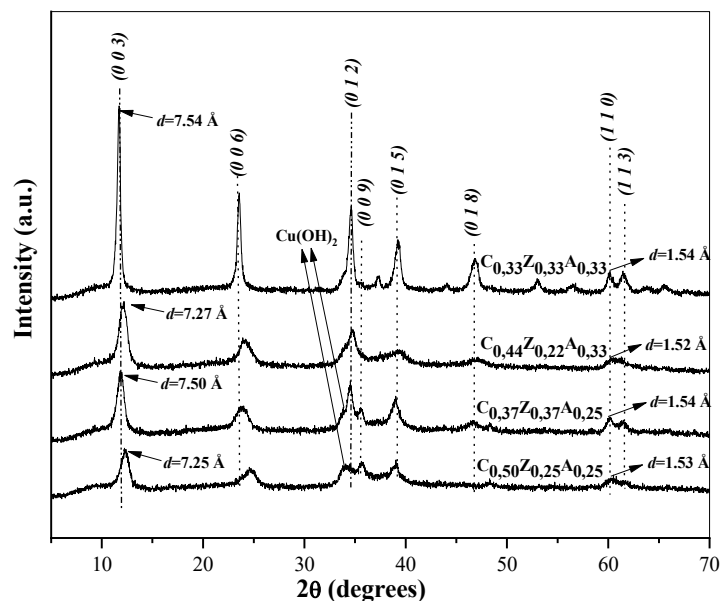


Figure 1. X-ray diffractograms of solids before calcination process.

In the mass loss curves of the hydrotalcite type precursors, shown in Figure 2, four mass loss zones can be observed, which are associated with; (I) the exit of water molecules located on the surface of the materials and in the interlaminar regions, (II) the beginning of the decomposition of interlaminar anions and with the collapse of the hydrotalcite type structure, (III) the decomposition of hydroxide groups and formation of divalent metal oxides (CuO , ZnO) at low temperature and trivalent ones at higher temperature (Al_2O_3 , $\geq 450^\circ\text{C}$), and (IV) the continuation of aluminum oxide formation and the exit of water and carbon dioxide that were trapped in the pores due to the simultaneous processes of collapse of the HTLc-type structure and the decomposition of the OH and CO_3^{2-} anions by effect of temperature.

It was observed that materials with higher copper content collapse at lower temperatures (142°C for $\text{C}_{0.50}\text{Z}_{0.25}\text{A}_{0.25}$ and 157°C for $\text{C}_{0.44}\text{Z}_{0.22}\text{A}_{0.33}$). This may be due to the presence of $\text{Cu}(\text{OH})_2$ in these materials or to the lower interplanar distance reached by these materials through the values of the c parameter (Table 1). For materials with lower copper content ($\text{C}_{0.37}\text{Z}_{0.37}\text{A}_{0.25}$ and $\text{C}_{0.33}\text{Z}_{0.33}\text{A}_{0.33}$), there is wider separation between units of the Cu^{2+} type within the hydrotalcite structure, thus decreasing the Jahn-Teller effect and increasing the thermal stability of these materials. The solids with the highest aluminum content achieved the greatest mass loss in zones I, II and IV, a fact that may be related to the high-water adsorption capacity of aluminum oxyhydroxides. All the materials reached an abrupt loss of mass at temperatures above 570°C , probably due to the exit of carbon dioxide and water molecules that were trapped in closed spaces within the solids due to the collapse of the

structure at low temperatures as well as through the formation of metal oxides from the decomposition of the hydroxide groups present in the initial structure [54].

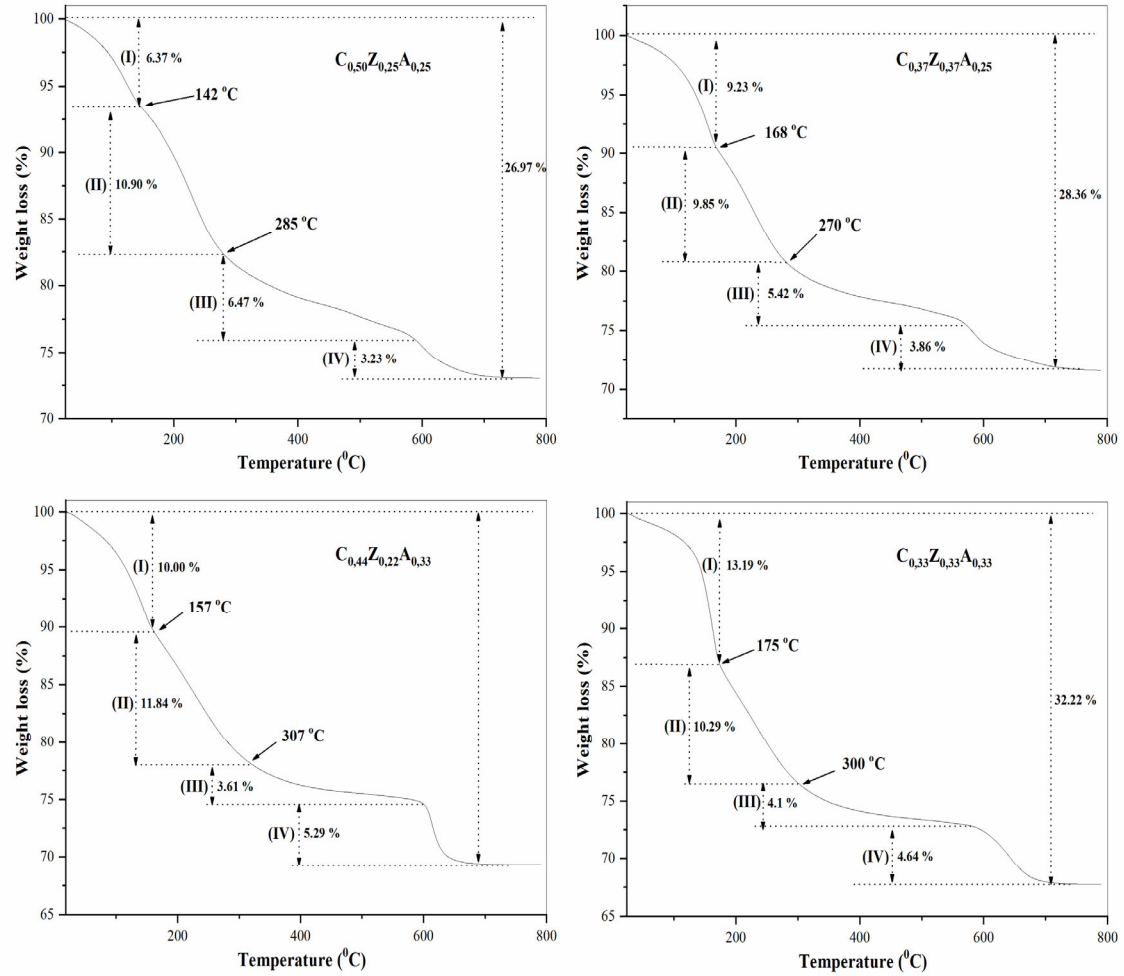


Figure 2. Thermogravimetry curves of hydrotalcite-type precursors.

Table 1. Values of interplanar distance (ID), collapse temperature (CT), and mass loss (ML) associated with events recorded in the TGA curve of hydrotalcite-type precursors.

Sample	ID (Å)*	CT (°C)	ML Step I	ML Step II	ML Step III	ML Step IV
$C_{0.50}Z_{0.25}A_{0.25}$	21.76	142.0	6.37 %	17.27 %	23.74 %	26.97 %
$C_{0.37}Z_{0.37}A_{0.25}$	22.48	168.0	9.23 %	19.08 %	24.50 %	28.36 %
$C_{0.44}Z_{0.22}A_{0.33}$	21.81	155.9	10.01 %	21.84 %	25.45 %	30.74 %
$C_{0.33}Z_{0.33}A_{0.33}$	22.63	175.0	13.19 %	23.48 %	27.58 %	32.22 %

* The values of the interplanar distance (ID) were calculated through the peak attributed to the plane (0 0 3) in the X-ray diffractograms of the precursors with hydrotalcite type structure.

Figure 3 contains the X-ray diffractograms of the solids calcined at 500 °C. The presence of the CuO and ZnO phases, resulting from the decomposition of hydrotalcites, was mainly perceived in materials with lower aluminum content. In materials in which the aluminum content is higher, diffraction patterns shown broad and poorly defined peaks, which may be produced by the increase in the amorphous character of the solids due to the presence of transitional alumina.

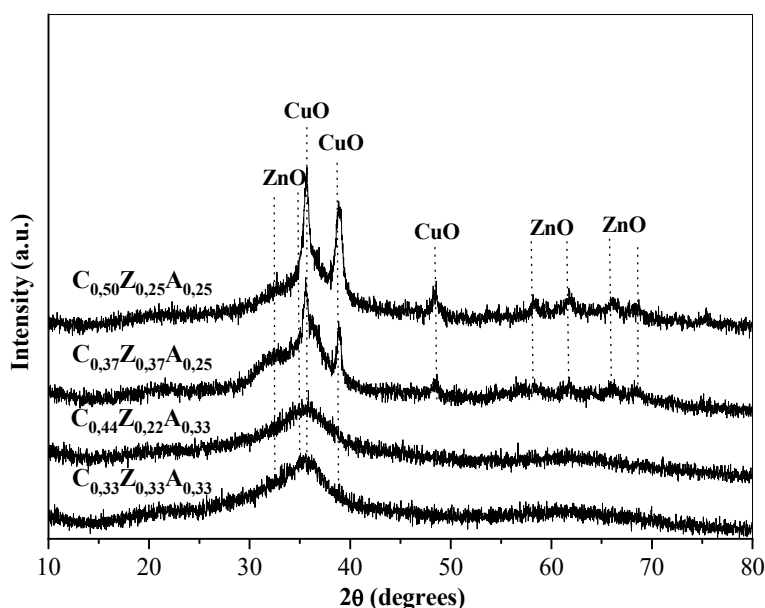


Figure 3. X-ray diffractograms of solids obtained from calcination at 500 °C of the hydrotalcites.

The adsorption-desorption isotherms of the materials displayed similar profiles, which reflect the presence of lumped particles in the form of sheets that originate pores in the form of notches, which conform to the type II isotherms as per the IUPAC classification (Figure 4a) [55].

The narrow hysteresis loops that were registered by the adsorption-desorption graphs point out to the presence of mesopores, which is confirmed by the pore distribution shown in Figure 4b. All materials contain pores between 2 and 75 nm, and a mean radius that slightly decreases in samples with higher aluminum content. The values in Table 2 indicate that the $C_{0.50}Z_{0.25}A_{0.25}$ material has the largest pore diameter and volume, as well as the highest specific surface area ($69 \text{ m}^2/\text{g}$); this is probably due to the fact that, in this solid, the number or length of the pores is greater than that accomplished by the other samples, which leads to an increase in the surface area.

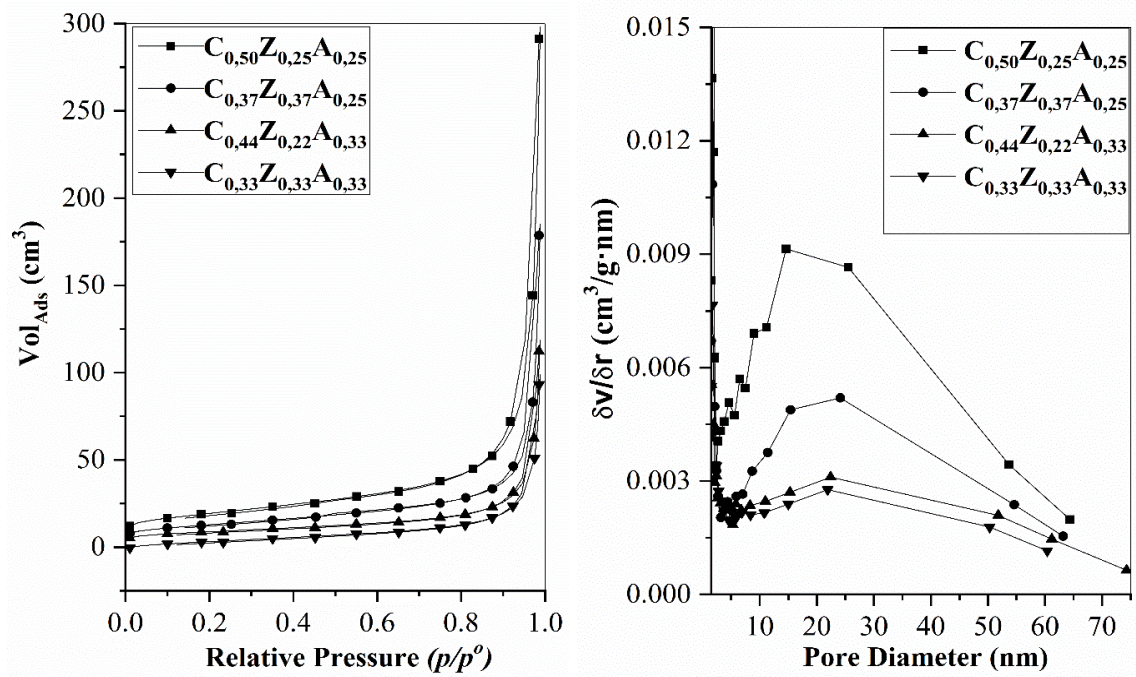


Figure 4. Isotherms of adsorption-desorption of nitrogen at 77 K and pore distribution of the calcined solids.

The TPR graphs, shown in Figure 5, display a single reduction peak, which is attributable to the CuO present in all the calcined solids [56,57]. It was observed that the material with the highest copper content ($C_{0.50}Z_{0.25}A_{0.25}$) achieved the lowest reduction temperature, with a peak centered at 225 °C and extending to 270 °C. In the case of the other materials, the temperature required to reduce CuO to Cu^0 was higher, which indicates that the hike in aluminum content makes the copper reduction process difficult, in a way similar to that observed in the case of other metal cations [57,58]. Such results suggest that the $C_{0.50}Z_{0.25}A_{0.25}$ material contains the highest content of CuO on the external surface, which facilitates its reduction at low temperatures, whereas, in solids with higher aluminum content, CuO is located mainly in the internal volume of the materials, which makes the prior diffusion of the hydrogen necessary to react with CuO. This fact is confirmed by the metallic dispersion values shown in Table 2, in which it can be seen that the solids with the lowest aluminum proportion have the highest copper dispersion on the surface, which favors the reduction reaction at low temperatures. The $C_{0.50}Z_{0.25}A_{0.25}$ material showed the highest metallic copper area, with a value close to the specific surface area reached by this material, which confirms a higher proportion of CuO on the surface for this material, when compared to the other solids studied.

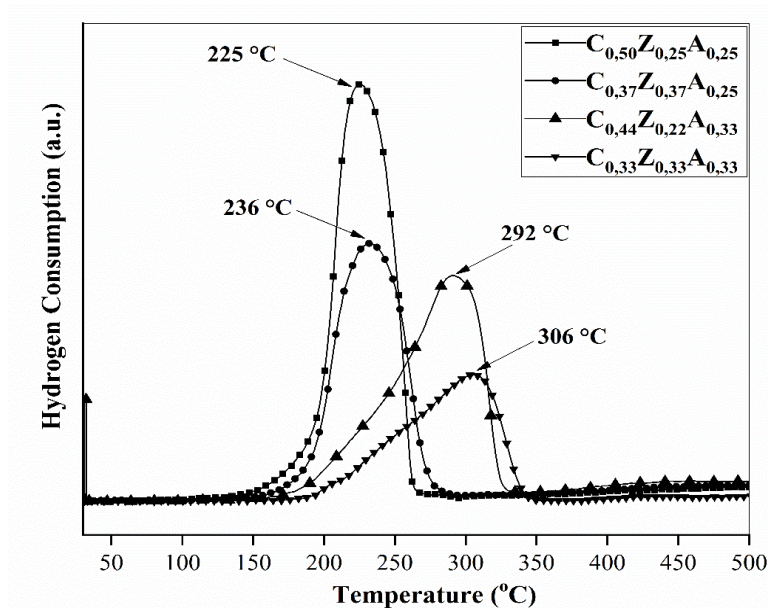


Figure 5. Curves of temperature programmed reduction of solids after calcination.

Table 2. Values of the specific surface area (Sg), average pore radius and pore volume of calcined solid; dispersion and metallic area of copper reduced using conditions similar to those used before the catalytic evaluation.

Sample	Sg (m ² /g)	Pore diameter (nm)	Pore volume (cm ³ /g)	Metallic area (m ² /g)	Particle size (nm)	Dispersion (%)
C _{0.50} Z _{0.25} A _{0.25}	69	13.35	0.46	60.7	12.9	39.2
C _{0.37} Z _{0.37} A _{0.25}	46	11.84	0.29	36.6	13.4	40.0
C _{0.44} Z _{0.22} A _{0.33}	36	10.48	0.18	25.2	4.6	25.0
C _{0.33} Z _{0.33} A _{0.33}	33	10.36	0.16	14.9	4.4	22.4

* Specific surface area (Sg), average pore radius and pore volume were obtained by adsorption-desorption of nitrogen at 77 K; dispersion and metallic area of copper were obtained by reduction with hydrogen, previous surface oxidation with nitrous oxide of CuO present in the calcined solids; particle size of the metallic copper were calculated using the Scherrer equation and the plane (1 1 1) of the Cu⁰ crystal lattice obtained from the XRDs of the catalysts used.

All the catalysts were active for the WGS reaction, though the formation of species other than hydrogen and carbon dioxide was not observed. Figure 6a comes to show that the conversion percentage increases with temperature, being C_{0.50}Z_{0.25}A_{0.25} the most active solid up to 250 °C, while, at higher temperatures, the C_{0.37}Z_{0.37}A_{0.25} solid led to higher percentages of monoxide to carbon dioxide conversion through the WGS reaction, reaching values close to 90% conversion. In the case of catalysts with higher aluminum content, the conversion was lower, which may be due to the lower specific surface of such materials. It was observed that, in materials with the same aluminum content, catalysts with a Cu/Zn molar ratio equal to 1 are more active, which can be attributed to the fact that zinc also contributes to the WGS reaction [59]. This effect is more evident in the turnover frequency

curves (TOF, Figure 6b), in which it is verified that the catalysts with this Cu / Zn molar ratio are more efficient, even when the metallic area for such solids is less than that shown for the $C_{0,50}Zn_{0,25}Al_{0,25}$ catalyst.

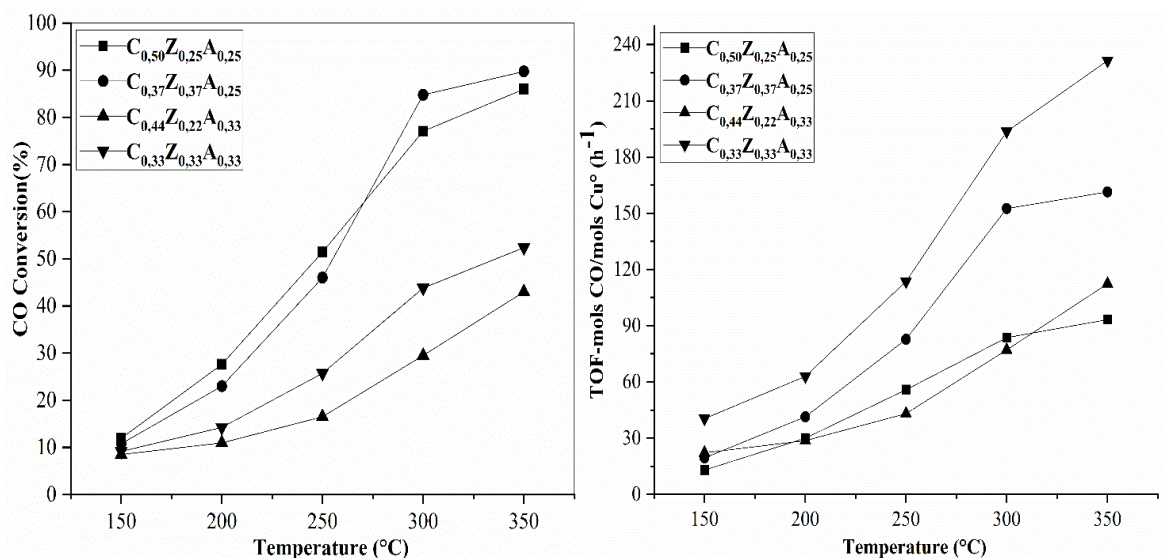


Figure 6. Conversion and turnover frequency (TOF) of monoxide to carbon dioxide through the WGS reaction using calcined catalysts, after reduction with hydrogen at 350 °C.

XPS was performed $C_{0,44}Zn_{0,22}Al_{0,33}$ and $C_{0,33}Zn_{0,33}Al_{0,33}$ materials after the process of reduction with hydrogen. In the literature, XPS studies on (Cu,Zn,Al) hydrotalcite [50], on mixed Cu-Zn-Al mixed oxides [60-63] or even on copper aluminates [51,64-66] generally concentrate the information given on Cu and Zn (often the Cu2p and Zn2p BE regions) and only seldom give information on Al (Al2p BE region); indeed only a few measurements of the Al2p BE are given in the case of copper aluminates [51,66]. In the case of copper aluminates, either [64] or [66] are often referred to as references for data on $CuAl_2O_4$. Occasionally Al2p BE is rather used as energy reference instead of the BE of C-(C,H) bonds in adventitious carbon; the assigned value can vary from 74,0 eV [64] to 74.5 eV [60]. Actually the latter value is also derived from data on $CuAl_2O_4$ where the reference energy corresponding to BE of C-(C,H) bonds in adventitious carbon was taken to be 285.1 eV [66] instead of the more often used 284.6 eV [49-51]. This is important when comparing data on the BE measured for the different components of the material since it introduces a 0.5 eV shift in this data. In the following we will take this into account and will explicitly inform when the data was corrected for this shift.

The spectra obtained for both samples for aluminum (Al2p), copper (Cu2p) and zinc (Zn2p) are shown in Figure 6. Data extracted from these spectra is available in Tables 3 (BEs), 4 (surface

composition) and 5 (relative content of the components used for fitting the experimental peaks). We can notice that in binding energy (BE) region of the Al 2p peak (Figure 6a) there is a high BE contribution arising from the presence of the Cu3p peak within the same BE region. Although this may be hazardous if one wants to use the BE of Al2p as the reference energy, it does not (strongly) affect the determination of the BE of Al2p which was measured to be 73.8 eV for both samples. This is in good agreement with Al2p BE found either in copper aluminates or CuAlZn hydrotalcites. As mentioned above, only a few measurements of the Al2p BE are given in the case of copper aluminates: for CuAlO₂ the value given is 74±0.5 eV [51] (in this reference, variation in BE is due to Eu cation doping) and for CuAl₂O₄ the value given is 74.5 eV [66]; the latter value corrected for the shift in the reference energy mentioned above (-0.5 eV) is thus also around 74 eV.

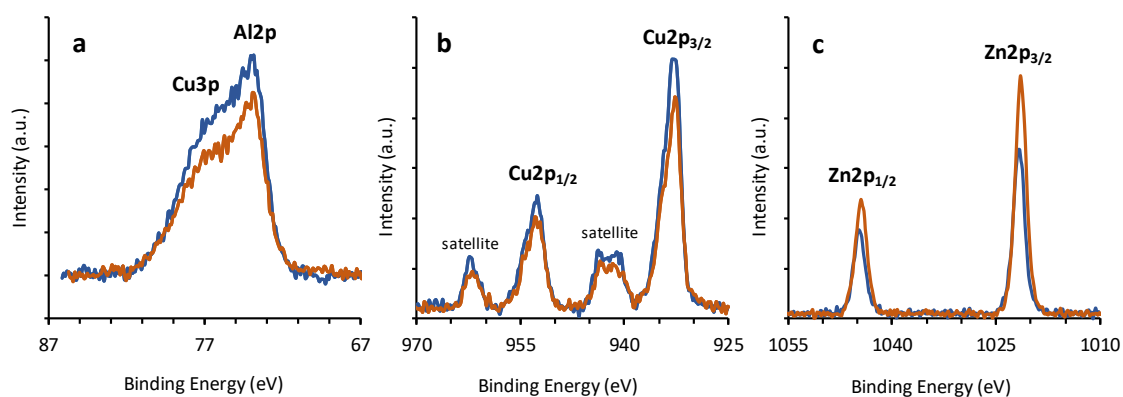


Figure 6. XPS of the reduced C_{0.44}Z_{0.22}A_{0.33} (orange spectra) and C_{0.33}Z_{0.33}A_{0.33} (blue spectra) catalysts: (a) Cu3p, Al2p BE region; (b) Cu2p BE region and (c) Zn2p BE region.

XPS spectra of the Cu2p and Zn2p BE regions are presented in Figures 6b and 6c and respectively; more detailed data for these regions are available in Figure 7 where decompositions of the Cu2p_{3/2}-Cu2p_{1/2} (Figure 7a) and Zn2p_{3/2}-Zn2p_{1/2} (Figure 7b) are provided. The main features are similar for both samples. Spectra of Cu2p present Cu2p_{3/2} and Cu2p_{1/2} peaks with a main component and one or two high BE energy shoulders; a key feature is the presence of satellites characteristic of Cu²⁺ in CuO [67-72], Cu(OH)₂ [67,69-71,73], CuAl₂O₄ [61,65,66] and (Cu,Zn,Al)HT [50] and which is extremely faint (even non-existent) in Cu₂O [69-72,74] and Cu [69-71,75]. Thus the presence of strong satellites rules out the possibility of the presence of Cu₂O in the sample. A close view of satellite present just after the Cu2p_{3/2} peak exhibit two shoulders (black and grey arrows in Figure 7a) which are characteristic of CuO [68-72] and CuAl₂O₄ [61,65,66]; these shoulders are absent for Cu(OH)₂ [69-71,73], and (Cu,Zn,Al)HT [50] for which only a broad peak exists. Also, despite the noise in the

spectra we can also advance that for $C_{0.33}Z_{0.33}A_{0.33}$ catalyst the low BE shoulder (grey arrow in Figure 7a) in this satellite is more marked than for $C_{0.44}Z_{0.22}A_{0.33}$ catalyst indicating that the ratio between $CuAl_2O_4$ and CuO is certainly lower for the latter catalyst since the prominence of this low BE shoulder in this satellite is more marked in $CuAl_2O_4$ than in CuO . This qualitative observation indicates thus the presence of CuO and $CuAl_2O_4$ in the samples with variations of the relative amounts of these two components.

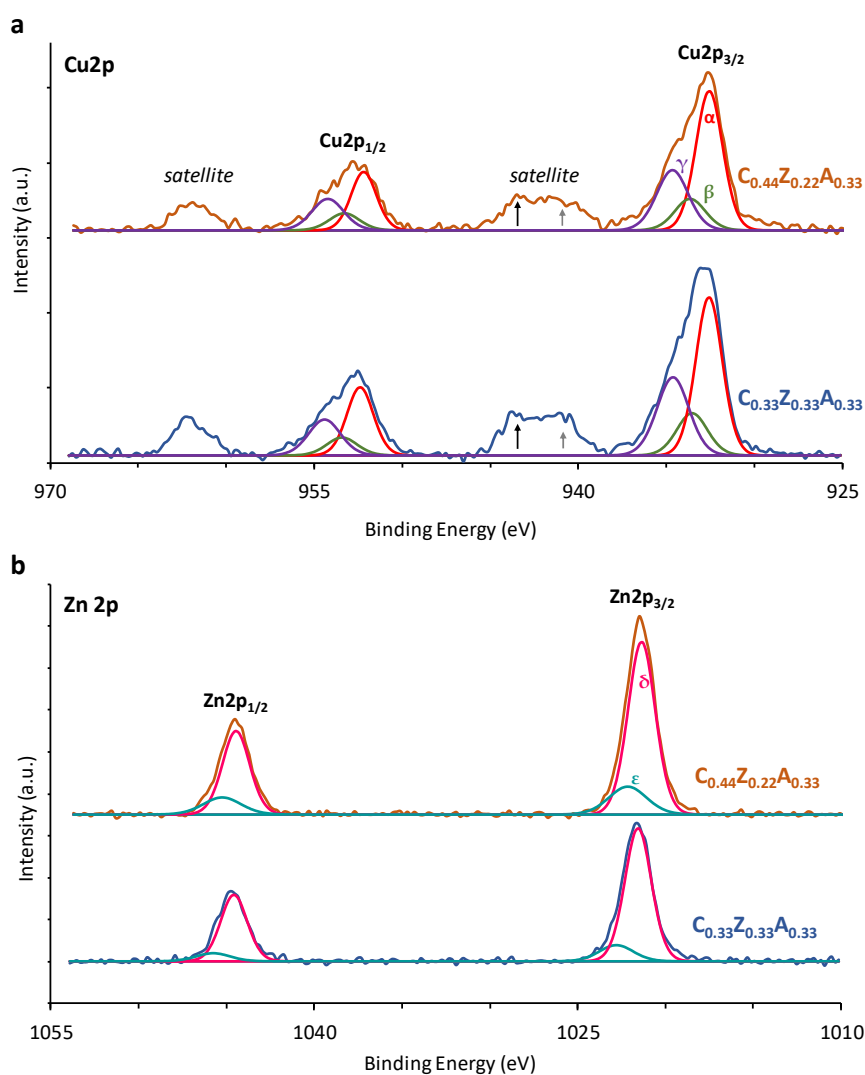


Figure 7. XPS of the reduced $C_{0.44}Z_{0.22}A_{0.33}$ (orange spectra) and $C_{0.33}Z_{0.33}A_{0.33}$ (blue spectra) catalysts: (a) $Cu2p_{3/2}$ and $Cu2p_{1/2}$ and corresponding decomposition peaks (α , β , γ – see text for details), arrows in the low BE satellite are commented in the text; (b) $Zn2p_{3/2}$ and $Zn2p_{1/2}$ and corresponding decomposition peaks (δ , ϵ – see text for details).

Table 3. Experimental BE of the principal peaks (P) and shoulders (high energy S_{HE} , low energy S_{LE}) and BE components used for the best fit for $Cu2p_{3/2}$, $Zn2p_{3/2}$ and O1s (see text and Figures 7 and 8 for details) of the two reduced catalysts: $C_{0.33}Z_{0.33}A_{0.33}$ and $C_{0.44}Z_{0.22}A_{0.33}$. For Al2p the experimental BE value is given in the text.

Catalyst	Cu2p _{3/2} BE (eV)						Zn2p _{3/2} BE (eV)			O1s BE (eV)				
	Exp			Fit			Exp	Fit		Exp		Fit		
	P	S_{HE-I}	S_{HE-II}	α	β	γ'	P	δ	ϵ	P	S_{LE}	η	θ	ν
C_{0.33}Z_{0.33}A_{0.33}	932.8	933.7	934.7	932.6	933.6	934.6	1021.5	1021.6	1022.8	531.3	529.6	529.6	530.5	531.4
C_{0.44}Z_{0.22}A_{0.33}	932.7	933.8	934.8	932.6	933.6	934.6	1021.7	1021.4	1022.1	531.4	529.6	529.6	530.5	531.4

Since we have a better signal/noise ratio for the $Cu2p_{3/2}$ peak we will concentrate on this peak for the numerical data (see Tables 3, 4 and 5) extracted from the spectra and its discussion; we have verified that these measurements are consistent with the data for $Cu2p_{1/2}$. For the two samples we have similar characteristics for the $Cu2p_{3/2}$ peak: a main peak with two high BE shoulders indicating the presence of at least three components. In the experimental spectra the main peak is centred at 932.7 eV for $C_{0.33}Z_{0.33}A_{0.33}$ and at 932.8 eV for $C_{0.44}Z_{0.22}A_{0.33}$. The two high BE energy shoulders are at around 933.8 eV and 934 eV for $C_{0.33}Z_{0.33}A_{0.33}$ and at around 933.7 eV and 934.7 eV for $C_{0.44}Z_{0.22}A_{0.33}$. These values correlate well with values from literature for Cu(0) at around 932.6 ± 0.2 eV [69,71,72], CuO at around 933.6 ± 0.4 eV [69,71,72] and for either $CuAl_2O_4$ [61,65,66] or $(Cu,Zn,Al)HT$ [50] at around 934.6 eV (value corrected by -0.5eV when the energy reference was 285.1eV for C1s and consequently 74.5 eV for Al2p). Since from the qualitative analysis of the satellites we determined that the most probable attribution is $CuAl_2O_4$ rather than $(Cu,Zn,Al)HT$ we will only consider the former in the following.

Table 4. Surface composition* determined by XPS for reduced catalysts $C_{0.33}Z_{0.33}A_{0.33}$ and $C_{0.44}Z_{0.22}A_{0.33}$.

Catalyst	Cu2p (%)	Zn2p (%)	Al2p (%)
C_{0.33}Z_{0.33}A_{0.33}	24.2	35.6	40.1
C_{0.44}Z_{0.22}A_{0.33}	32.1	26.0	41.9

* In order to directly compare the surface composition (contents in Cu, Zn and Al) of the catalysts with the bulk composition (contents in Cu, Zn and Al) we have exclude O from the quantification despite it being a component of the catalyst; the value found was quite similar ($50.0 \pm 0.5\%$) for both samples We have also excluded C from the quantification since it is due to pollution and as such it is not part of the catalyst; the value found was also quite similar ($12.5 \pm 1.0\%$) for both samples.

Table 5. Quantification on the different components of $Cu2p_{3/2}$, $Zn2p_{1/2}$ and O1s for the two reduced catalysts $C_{0.33}Z_{0.33}A_{0.33}$ and $C_{0.44}Z_{0.22}A_{0.33}$.

Catalyst	Cu2p _{3/2}	Cu2p _{3/2}	Cu2p _{3/2}		Zn2p _{3/2}	Zn2p _{3/2}		O1s	O1s	O1s
	α (%)	β (%)	γ (%)		δ (%)	ϵ (%)		η (%)	θ (%)	ν (%)
C_{0.33}Z_{0.33}A_{0.33}	52.6	16.1	31.3		86.2	13.8		18.7	2.2	79.1
C_{0.44}Z_{0.22}A_{0.33}	56.4	14.6	29.0		81.8	18.2		8.8	2.4	88.8

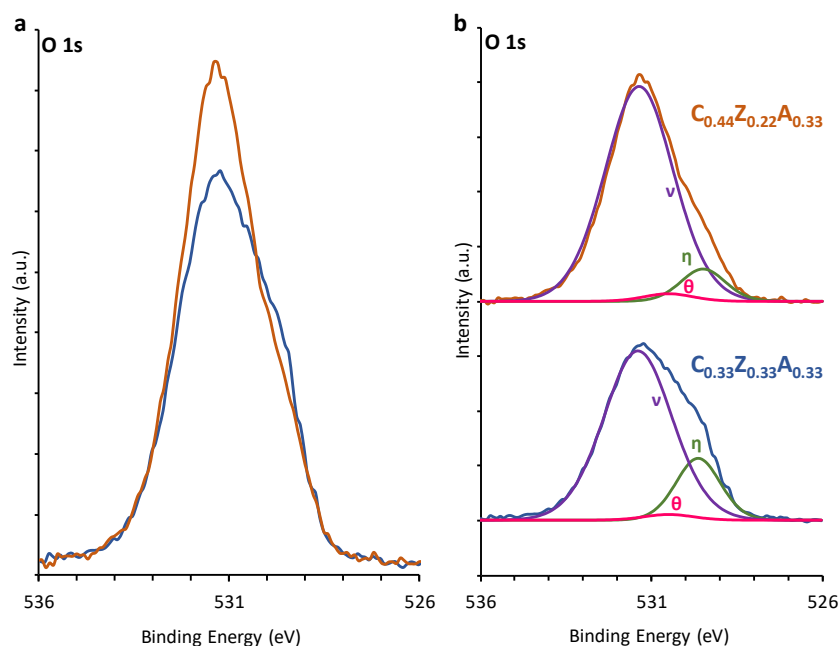


Figure 8. XPS of the reduced $C_{0.44}Z_{0.22}A_{0.33}$ (orange spectra) and $C_{0.33}Z_{0.33}A_{0.33}$ (blue spectra) catalysts: (a) O1s BE region; (b) O1s and corresponding decomposition peaks (η , θ , ν – see text for details).

We have thus used these literature values to determine the BE of the three components (α , β , γ) and realize a consistent fit of peaks $Cu2p_{3/2}$ and $Cu2p_{1/2}$ (we did not fit the satellites, for which we made a qualitative comment above, since there was no quantitative data to be extracted and to simplify Figure 7). Since we have used fixed BEs for the three components (α , β , γ) for both catalysts, we can see that the experimental BE (P , S_{HE-I} , S_{HE-II}) are slightly higher (see Table 3), although within the expected variations from literature. This very small shift, as expected from metal-metal interaction, may also hint at Zn-Cu interaction within the three different components. This is particularly the case for the α component attributed to Cu(0) which is also detected by XRD. After reduction, we have thus the presence of metallic Cu in both catalysts with a slightly higher relative content for the catalysts richer in Cu (Table 5); this Cu domains might be slightly modified by the presence of Zn leading to the formation of intermetallics or at least C-Zn bonds. The β component, which is the minor contribution for both catalysts, was attributed to Cu(II) in CuO and the γ component to Cu(II) in $CuAl_2O_4$. Crystalline CuO was detected for $C_{0.44}Z_{0.22}A_{0.33}$ but not for $C_{0.33}Z_{0.33}A_{0.33}$ and $CuAl_2O_4$ was not detected for both by XRD. These components are either in a poorly crystalline form or are only

characteristic of the surface of the catalysts. Furthermore, we can see that the contribution from CuAl_2O_4 is larger than that of CuO which is consistent with both the evolution of the (Cu,Zn,Al)HT initial material during temperature treatments and the consecutive reduction process undergone by the catalysts. We have also to consider the presence of Zn in substitution of Cu within those components leading to both mixed oxides ($\text{Cu}_{1-x}\text{Zn}_x\text{O}$) or mixed aluminates ($\text{Cu}_{1-x}\text{Zn}_x\text{Al}_2\text{O}_4$) indicating a close interaction between Cu and Zn. This seems favoured by the increasing content of Zn in the catalysts and thus more pronounced in $\text{C}_{0.33}\text{Z}_{0.33}\text{A}_{0.33}$ than in $\text{C}_{0.44}\text{Z}_{0.22}\text{A}_{0.33}$.

The analysis of the $\text{Zn}2p_{3/2}$ and $\text{Zn}2p_{1/2}$ peaks decomposition is somewhat less straightforward since the chemical shift in Zn is quite small leading to BE values very similar for Zn(0) and Zn(II) in ZnO [50,61,62,67,71,76-81] or even for Zn(II) in ZnAl_2O_4 [61,62,76], (Cu,Zn,Al) mixed oxides [61-63,76, 82-84] or (Cu,Zn,Al)HT [50]. Indeed, the consulted literature gives values, for $\text{Zn}2p_{3/2}$ BE, around 1021.7 ± 0.8 eV (when necessary BE values were corrected by -0.5 eV when the energy reference was 285.1 eV for C1s and consequently 74.5 eV for Al2p) for all these compounds; and even contradictory chemical shifts ($\text{BE}_{\text{ZnO}} < \text{BE}_{\text{Zn}}$) were found [71]. It is thus very difficult to differentiate those compounds without information on the Auger parameter [85]; we could not calculate the latter since we could not gather reliable information on the LMM Zn Auger lines. However for the (Cu,Zn,Al) mixed oxides a tendency was observed where the BE would increase with increasing Al content [61,62]. Considering all this we have thus decided not to fix the BE of the components used to fit the $\text{Zn}2p_{3/2}$ peak, as we did previously for $\text{Cu}2p_{3/2}$. For both samples the $\text{Zn}2p_{3/2}$ peak is constituted by a single principal peak (P) with no apparently detectable high BE shoulders (Figure 6c and Figure 7b). However, the best fit (Figure 7b) was obtained by a combining a principal component (δ) associated with a very small high BE component (ϵ) accounting for a slight high BE asymmetry. The component δ was found to have a similar BE for both samples (see Table 3): 1021.4 eV and 1021.6 eV for $\text{C}_{0.44}\text{Z}_{0.22}\text{A}_{0.33}$ and $\text{C}_{0.33}\text{Z}_{0.33}\text{A}_{0.33}$, respectively. Conversely, the BE of component ϵ has been found to be rather different (see Table 3): 1022.1 eV and 1022.8 eV for $\text{C}_{0.44}\text{Z}_{0.22}\text{A}_{0.33}$ and $\text{C}_{0.33}\text{Z}_{0.33}\text{A}_{0.33}$, respectively. The BE of the two components are both higher (slightly in the case δ of and strongly in the case of ϵ) for the catalysts containing the higher amount of Zn ($\text{C}_{0.33}\text{Z}_{0.33}\text{A}_{0.33}$) hinting at a stronger Cu-Zn interaction for this catalyst.

The O1s peak (Figure 8 and Table 3) is composed by a principal peak (P) with a low BE shoulder (S_{LE}). However, the best fit (Figure 8b) requires three components (η , θ , ν), the component θ being very minor (less than 3%, see Table 5); the latter corresponds to the ZnO contribution with a BE around 530.5 eV in both samples consistent with literature values which are 530.4 ± 0.5 eV [61,62,82-84,86]. The low BE shoulder at around 529.6 eV for both catalyst corresponds

to a contribution of CuO probably slightly modified (by Zn) since this value is slightly higher than the one expected (529.4 eV) [68]. A main high BE component η with a BE value of 531.4 eV for both catalysts which is slightly higher but consistent with the BEs of alumina, (Cu,Zn) aluminates or (Cu, Zn, Al) mixed oxides. Indeed alumina exhibits a broad range of BEs (531.0 \pm 1.0 eV) [67]; values for ZnAl₂O₄ [61,62,76], CuAl₂O₄ [61,62,66] and (Cu,Zn,Al) mixed oxides [61,62] are 531.1 \pm 0.3 eV (as previously, BE values were corrected by -0.5 eV when the energy reference was 285.1 eV for C1s and consequently 74.5 eV for Al2p). Together with the results for Cu2p_{3/2} and Zn2p_{3/2} this clearly indicates the presence of copper aluminates and that within oxides/aluminates there is an interaction between Cu and Zn with a more pronounced effect for the C_{0.33}Zn_{0.33}Al_{0.33} catalyst; furthermore Cu(0) particles (again probably in interaction with Zn) exist for the two catalysts. The content of Cu and Zn on the surface varies according to the composition of the catalysts, which highlights the fact that the surface copper ratio between the solids studied (C_{0.33}Zn_{0.33}Al_{0.33}/C_{0.44}Zn_{0.22}Al_{0.33} = 0.75) is identical to the total copper ratio in the materials (0.75), which indicates that, in these materials, the content distribution is rather homogenous from surface to bulk.

Figure 9 reports XRD peaks that confirm the presence of metallic copper particles in all the materials used, as well as copper oxide and zinc oxide, except in the C_{0.33}Zn_{0.33}Al_{0.33} material, which failed to reach peaks assignable to CuO.

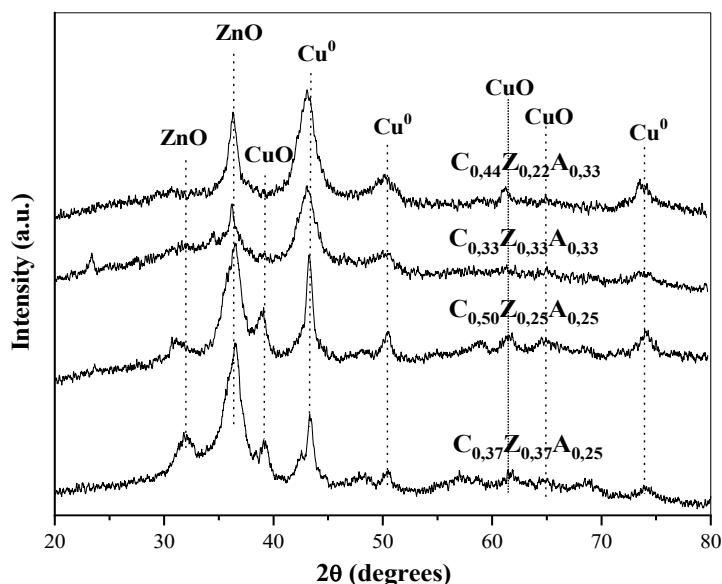


Figure 9. X-ray diffractograms of solids after of the catalytic evaluation.

The TOF indicates that this solid has a better performance, which can be ascribed to factors such as the smaller particle size of Cu⁰ and to the presence of particles in which there are Cu-Zn interactions. Furthermore, the absence of CuO after catalytic evaluation indicates that the Cu⁰ particles in this material are more resistant to oxidation, an effect that could have been induced by zinc and by the lower dispersion and metallic area displayed by this solid.

Conclusions

The calcination of the hydrotalcite-type precursors led to the formation of CuO particles in all the materials, which were more resistant to the reduction processes as the aluminum content augmented. All the catalysts displayed Cu⁰ particles after the reduction process; however, the materials also contain CuO in their composition, except for the C_{0.33}Z_{0.33}A_{0.33} material. The percentage of reaction increased with temperature, whereby the C_{0.50}Z_{0.25}A_{0.25} and C_{0.37}Z_{0.37}A_{0.25} materials were the most active with conversion rates nearing 90%. Through the TOF, it was observed that the C_{0.33}Z_{0.33}A_{0.33} catalyst is the most effective depending on the copper content, which can be attributed to the absence of CuO, to the smaller size of the Cu⁰ particles and, possibly, to the existence of Cu-Zn interactions on the surface of the solid.

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