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Rémi<br/> Rafael, Etienne Puyoo, Christophe Malhaire. Piezo-tunneling strain sensors integrated on plastic<br/> by combining vacuum thin film coatings and 3D printing technologies. Microsystem Technologies,<br/> 2020, 10.1007/s00542-020-04832-7. hal-03344465

# HAL Id: hal-03344465 https://hal.science/hal-03344465

Submitted on 15 Sep 2021

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# Piezo-tunneling strain sensors integrated on plastic by combining vacuum thin film coatings and 3D printing technologies

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Keywords: flexible strain sensors, piezo-tunnel effect, Al<sub>2</sub>O<sub>3</sub>, atomic layer deposition, 3D printing, Fused Deposit Modeling

Abstract: There is a growing demand for the integration of sensor functions on flexible substrates for wearable electronics, robotics or medical monitoring. For this, it is necessary to develop strain gauges both sensitive and integrable at low cost with a low thermal budget. The gauge factor of metal/insulator/metal piezo-tunneling strain sensors is first measured as a function of applied current and polarity, for different electrode materials (Al, Pt or Pd) and insulator (Al<sub>2</sub>O<sub>3</sub>) thicknesses. A maximum gauge factor of 90 is obtained with an Al/Al<sub>2</sub>O<sub>3</sub>(10nm)/Al junction and top electrode injection. Results are discussed based on the Fowler-Nordheim model and it is shown that the electron effective mass in Al<sub>2</sub>O<sub>3</sub> most likely plays a major role in the observed mechano-sensitivity. Next, the feasibility of a low-pressure sensor demonstrator based on a 3D-printing process on a polymer substrate is shown with a sensitivity of 0.19 bar<sup>-1</sup> in the 0 – 450 mbar range.

#### **1. Introduction**

Piezoresistive sensors are of major interest for the development of both wearable and conformable electronics industry sectors (Amjadi 2016). Flexible strain sensors integrating piezoresistive elements can indeed be used to develop tactile sensing functions (Park 2016), human motion and medical monitoring (Tolvanen 2018; Puyoo 2018; Kang 2014), and multirecognition functions (Su 2016). They can also be used in the field of robotics, aerospace engineering and transportation to develop lightweight plastic Micro-Electro-Mechanical Systems (MEMS) that can be conformably integrated on non-planar surfaces (Xiao 2008). Among the different approaches which have been proposed in literature these past few years, piezo-tunneling strain sensors give rise to gauge factors in the range of 10 to 300 which compete with that of monocrystalline silicon (standard for piezoresisitive MEMS) (Herrmann 2007; Schlike 2016; Jiang 2015; Puyoo 2017; Zhu 2015; Rafael 2017). The principle of piezo-

tunneling strain sensor is based on the fact that the electrical resistance of a tunnel junction exponentially varies with its parameters (junction length, electron effective mass, barrier height). For example, when submitted to mechanical stress, the variation of the tunnel junction length implies an exponential variation of the tunnel resistance which gives rise to these high gauge factors. This concept has been first exploited with metallic nanoparticle assemblies in which the current is flowing by tunnel effect from nanoparticle to nanoparticle (Herrmann 2007; Schlike 2016; Jiang 2015; Puyoo 2017). This concept has been later used to advantage with single MOS (Metal Oxide Semiconductor) junctions (Zhu 2015) and more recently by our group with MOM (Metal Oxide Metal) junctions (Rafael 2017). The key advantage of MOM junctions as compared to MOS junctions is attributed to their elaboration process which requires a low thermal budget and which can be applied on a variety of substrates (e.g. flexible polymer substrates, back-end-of-line integration). In this work, piezo-tunneling MOM strain gauges based on Al<sub>2</sub>O<sub>3</sub> oxide material are first studied and optimized on silicon platform with conventional vacuum thin film deposition techniques: e-beam evaporation and atomic layer deposition (ALD). Then, the optimized process is transferred to flexible polyimide substrates and combined with 3D printing technology to develop a pressure sensor on plastic.

#### 2. Optimization of MOM strain gauges

Here, MOM vertical stacks were first elaborated in a cross-bar geometry (surface of 0.25mm<sup>2</sup>) on oxidized silicon substrates (thermal oxide thickness of 500nm). As shown in Figure 1a, bottom and top electrodes were patterned at room temperature by electron beam assisted evaporation through solid shadow masks. In this study different bottom electrode materials were tested (Al, Pt and Pd) whereas the top electrode was kept in Al. The oxide thin film was an Al<sub>2</sub>O<sub>3</sub> layer (10nm or 20nm thick) deposited by ALD at 200°C from TMA (Trimethyl Aluminum) and H<sub>2</sub>O precursors (pulse and purge durations of 60ms and 10s, respectively, for both precursors). Figure 1b presents a typical J(E) characteristic (i.e. current density as a

function of the applied electric field) obtained on a Pt/Al<sub>2</sub>O<sub>3</sub>/Al stack. As previously shown for Al/Al<sub>2</sub>O<sub>3</sub>/Al junctions (Rafael 2017), electron transport in the MOM structure can be attributed to Fowler-Nordheim (FN) tunneling mechanism. Note that this FN mechanism is also observed on Pd/Al<sub>2</sub>O<sub>3</sub>/Al stacks. The strain sensitivity of the MOM junctions is tested using a cantilever deflection method according to the experimental procedure described in reference (Rafael 2017). A 725µm thick Si cantilever (4cm long by 1cm wide) integrating 3 MOM cross-bar structures is clamped in a vice at one end while a variable deflection is applied at the other end by a motorized actuator. A constant voltage bias (typically between 3 and 10V) is applied on the MOM device while the current is measured as a function of time for different deflection levels. The gauge factor GF is defined here as the relative change of current divided by the longitudinal strain  $\varepsilon$ . It is calculated assuming that the strain  $\varepsilon$  for a deflection level  $\delta$  at a distance L from the clamping position is given by (Rafael 2017):

$$\varepsilon = \frac{3t_{si}(L-\mathbf{x})}{2L^3}\delta\tag{1}$$

where  $t_{Si}$  is the thickness of the Si cantilever and x the position of the MOM junction on the beam.

In a previous paper (Rafael 2017), we have demonstrated that the strain sensitivity in Al/Al<sub>2</sub>O<sub>3</sub>/Al junctions not simply originates from pure geometrical effects but also from the variation of physical parameters under strain (i.e. barrier height and/or electron effective mass in Al<sub>2</sub>O<sub>3</sub>). In order to ascertain the precise influence of barrier height and effective mass on the mechano-sensitivity, we have performed a parametric study which is summarized in Figure 2. The gauge factors of 6 junctions with different bottom electrodes (Al, Pt or Pd) and different tunnel oxide thicknesses (10 and 20nm) were characterized as a function of the no-load current level. For each MOM device, two different gauge factors were extracted by injecting electrons respectively from bottom electrode (positive voltage bias with bottom electrode grounded) and top electrode (negative voltage bias with bottom electrode grounded). Gauge factors ranging

from 30 to 90 are obtained over three decades of no-load current levels (from  $10^{-10}$ A to  $10^{-7}$ A). From figure 2, the first main observation is that there is no significant influence of the nature of the injecting electrode (Al, Pd or Pt) on the measured gauge factor. Therefore, we believe that the mechano-sensitivity of the MOM junction mainly relies on the intrinsic properties of Al<sub>2</sub>O<sub>3</sub> and not on the properties of the metal from which electrons are injected. The barrier height of the MOM junction corresponds to the difference between the work function (WF) of the injecting metal and the electronic affinity of Al<sub>2</sub>O<sub>3</sub>. According to Hölzl (1979), Al, Pd and Pt have significantly different WFs. As a result, the barrier heights associated to these three electrode materials should be different and should as well vary differently under strain. From this assumption, we conclude that the barrier height has not a significant influence on the mechano-sensitivity and that the electron effective mass in Al<sub>2</sub>O<sub>3</sub> most likely plays a major role. The second main observation concerns the influence of the tunnel current direction on the gauge factor. Whatever the electrode material and no-load current level, the gauge factor is kept almost twice higher when electrons are injected from the top electrode as compared to an injection from the bottom electrode. In FN tunneling regime, the triangular barrier seen by the electrons has an effective thickness that is smaller than the actual thickness of the junction. In other words, the electrons do not tunnel through the same Al<sub>2</sub>O<sub>3</sub> effective layer when the polarity of the junction is reversed. The variation under strain of the effective mass at the interfaces could be affected by the chemical composition of  $Al_2O_3$ . A composition gradient of  $Al_2O_3$  due to interdiffusion phenomena or electrode deposition processes could explain these different gauge factors observed with the two polarities. The last observation from figure 2 concerns the slight decrease of the gauge factor with respect to the no-load current level. This gauge factor variation is observed for all the junctions whatever the polarity. We believe that this tendency comes from the non-linearity of the FN tunneling current  $I_{FN}$  model which is given by the following expression (Rafael 2017):

$$I_{FN} = \frac{Se^2 m_m^*}{8\pi m^* h \varphi} E^2 \exp(-\frac{8\pi \sqrt{2m^* e} \varphi^{3/2}}{3hE})$$
(2)

where S is the surface of the MOM junction, e the elementary charge,  $m_m^*$  the electron effective mass in the injecting metal,  $m^*$  the electron effective mass in Al<sub>2</sub>O<sub>3</sub>, *h* the Planck constant,  $\varphi$ the barrier height and E the total electric field. From equation (2), we have calculated theoretical gauge factors by considering that the effective mass  $m^*$  in Al<sub>2</sub>O<sub>3</sub> varies under strain  $\varepsilon$  while the barrier height  $\varphi$  is kept constant. For this calculation, we have also considered that the thickness *t* and the surface *S* of the MOM junction both vary linearly with strain. The relative changes of thickness *t* and surface *S* with respect to the strain  $\varepsilon$  are given by the two following equations:

$$\frac{\Delta t}{t} = -\varepsilon \, v_{Al_2O_3} \qquad (3)$$
$$\frac{\Delta S}{S} = (1 - v_{Si}) \, \varepsilon \qquad (4)$$

where  $v_{Al_2O_3} = 0.22$  is the Poisson coefficient of Al<sub>2</sub>O<sub>3</sub> and  $v_{Si} = 0.0625$  is the Poisson coefficient of monocrystalline Si on the <110> axis.

A good agreement between theory and experiment is observed in Figure 2c for relative changes  $\Delta m^*/_{m^*}$  of -0.17% and -0.28% per 0.1% strain for bottom and top injections, respectively. This agreement confirms that the slight decrease of gauge factors with respect to the no-load current can be attributed to the non-linearity of the FN tunnel current model.

#### **3.** Pressure sensor integrated on plastic

Based on the piezo-tunneling strain gauges previously presented, an original approach for the realization of a pressure sensor on a polyimide (Kapton) substrate, with a 3D printed polylactic acid (PLA) rigid frame, is shown. A circular-shape diaphragm, 5mm in diameter and 75µm thick (that of the Kapton substrate), was choosen as mechanical transducer. The strain gauges were first made on the front side of the substrate. Two laser-cut polysilicon shadow masks were used to define the layout of the bottom (same diameter as the future membrane) and top

electrodes (Figure 3a and 3b). In fact, a total of seven redundant top electrodes were present (Figure 3c) but only one was measured at a time. The surface of each MIM junction was 0.25mm<sup>2</sup>. Process steps are summarized in Figure 3d. A 100nm thick palladium bottom electrode was evaporated through a first mask level. Then, a 10nm thick Al<sub>2</sub>O<sub>3</sub> film was deposited on the whole substrate by ALD. Finally, 100nm thick aluminum top electrodes were deposited through a second mask level. Both metals were deposited by electron beam evaporation.

Then, a Fused Filament Fabrication (FFF) 3D printer (MakerBot Replicator 2X) was used to grow a rigid frame, directly on the back side of the Kapton film, and define the membrane geometry (Figure 4). Printing parameters were: plate temperature of 100°C; layer thickness of 0.2mm. Four PLA corners, one layer thick, were deposited on the plate. The Kapton film was placed in the middle, and attached to the plate by tape, before continuing the PLA filling. The four-pointed star shape ensured adherence of the PLA frame to the printer plate during process. In addition, small laser-drilled holes in the substrate form rivets to improve adhesion between Kapton and PLA. In order to limit residual stresses that may occur during cooling, the inside of the frame had a honeycomb structure (30% filling). Despite this, an initial 100µm downwards deformation of the membrane was measured later without any applied pressure, so the printing process could probably be improved. A front view of a finished device, with its electrical connections, is given in the inset of Figure 5. The rear pressure chamber was connected to a pump through a manometer and a large buffer tank. Differential pressures were measured by means of a Fluke 718 pressure calibrator. Figure 5 shows, for one gauge, the relative current changes as a function of pressure. The non-linearity response of the gauge simply corresponds to the nonlinear load-deflection behavior of the membrane (large-deflection regime). From a linear fit between 0 and 450mbar, a 0.19 bar<sup>-1</sup> sensitivity was obtained. Although the sensitivity could be further increased by optimizing the design of our pressure sensor (membrane thickness, size and position of the gauge, Wheatstone bridge configuration...), we believe that this

demonstrator opens the way to a new variety of conformable MEMS devices integrated on plastic. Future works will have to focus on the combination of 3D printing and LASER cutting processes in order to develop more complex MEMS structures like cantilever sensors, accelerometers or gyroscopes.

## 4. Conclusion

In summary, MOM junctions elaborated at low temperature ( $\leq 200^{\circ}$ C) by combining ALD and e-beam evaporation were first optimized on Si substrate. A maximum gauge factor of 90 was obtained with an Al/Al<sub>2</sub>O<sub>3</sub>(10nm)/Al junction and electron injection from the top electrode. This process was then transferred to polyimide substrate and combined with FFF process to develop a pressure sensor integrated on plastic showing a sensitivity of 0.19 bar<sup>-1</sup> in the 0-450mbar range.

### Acknowledgements

Authors gratefully acknowledge facilities and technological staff's cooperation from NanoLyon platform.

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Zhu L and McNamara S (2015) Low Power Tunneling Current Strain Sensor Using MOS Capacitors. J Microelectromechanical Syst 24:755. https://doi.org/10.1109/JMEMS.2014.2351778 **Figure 1.** a) Elaboration process of a Metal/Al<sub>2</sub>O<sub>3</sub>/Metal junction. b) Current density as a function of the applied electric field measured on a  $Pt/Al_2O_3/Al$  junction. Inset: schematic representation of a  $Pt/Al_2O_3/Al$  junction integrated on a thermally oxidized Si substrate.





**Figure 2.** Gauge factors measured on six different junctions as a function of the no-load current level for both top and bottom injections.

**Figure 3.** a) Laser-cut polysilicon shadow masks used to define the layout of bottom a) and top b) electrodes. c) Photograph of seven Metal/Oxyde/Metal junctions integrated on a 75 $\mu$ m thick polyimide substrate. d) Cross section view of the elaboration process.



**Figure 4.** Photographs of the Fused Filament Fabrication process realized on the backside of a polyimide film.



**Figure 5.** a) Response of the sensor for different pressure steps ranging from 50mbar to 850mbar; b) Relative variation of current with respect to the applied differential pressure. Inset: photograph of the pressure sensor integrated on plastic.

