Supporting Information

Synergistic Effect of Chemical Substitution and Insertion on the Thermoelectric Performance of Cu₂₆V₂Ge₆S₃₂ Colusite

Yuta Shimizu¹, Koichiro Suekuni^{1,2}*, Hikaru Saito³, Pierric Lemoine⁴, Emmanuel Guilmeau⁵, Bernard Raveau⁵, Raju Chetty⁶, Michihiro Ohta⁶, Toshiro Takabatake⁷, Michitaka Ohtaki^{1,2}

¹Interdisciplinary Graduate School of Engineering Sciences, Kyushu University, Kasuga, Fukuoka 816–8580, Japan ²Transdisciplinary Research and Education Center for Green Technologies, Kyushu

University, Kasuga, Fukuoka 816-8580, Japan

³Institute for Materials Chemistry and Engineering, Kyushu University, Kasuga, Fukuoka 816-8580, Japan

⁴Université de Rennes 1, CNRS, ISCR-UMR 6226, F-35000 Rennes, France

⁵CRISMAT, CNRS, Normandie Université, ENSICAEN, UNICAEN, 14000 Caen, France

⁶Global Zero Emission Research Center, National Institute of Advanced Industrial Science and Technology (AIST), Tsukuba, Ibaraki 305-8569, Japan

⁷Graduate School of Advanced Science and Engineering, Hiroshima University, Higashi-Hiroshima 739-8530, Japan

 $Cu_{26+y}V_2Ge_{6-x}Sb_xS_{32} \ samples.$ x = 0*x* = 1 *x* = 2 *x* = 3 $d_{\rm s}$ $d_{\rm t}$ $d_{\rm s}/d_{\rm t}$ $d_{\rm s}$ $d_{\rm s}/d_{\rm t}$ $d_{\rm s}$ $d_{\rm t}$ $d_{\rm s}/d_{\rm t}$ $d_{\rm s}/d_{\rm t}$ $d_{\rm t}$ $d_{\rm s}$ $d_{\rm t}$ 4.40 4.52 0.97 4.55 0.98 4.41 4.58 0.96 4.46 4.60 0.97 y = 04.44 4.43 4.61 0.96 4.41 4.63 0.95 y = 1y = 24.54 4.67 0.97 4.50 4.68 0.96 4.73 0.97 4.57 4.75 0.96 y = 34.60

Table S1 Sample density d_s [g cm⁻³], theoretical density d_t [g cm⁻³], and the ratio d_s/d_t for the



Figure S1. Specific heat C_p in units of J g^{-1} K⁻¹ and J mol⁻¹ K⁻¹ for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$, $Cu_{26+y}V_2Ge_6S_{32}$, and $Cu_{26+y}V_2Ge_{6-x}Sb_xS_{32}$ (x = 1) samples.



Figure S2. Powder X-ray diffraction patterns for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$ samples. Arrows for x = 0 and x = 4 denote peaks from Cu_2S and an unidentified phase, respectively. The simulated pattern based on the colusite $(Cu_{26}V_2Ge_6S_{32})$ structure is shown at the bottom. The right panel shows expanded views of 622 peaks.



Figure S3. Powder X-ray diffraction patterns for $Cu_{26+y}V_2Ge_6S_{32}$ samples. Arrows denote peaks from Cu_2S . The simulated pattern based on the colusite ($Cu_{26}V_2Ge_6S_{32}$) structure is shown at the bottom. The right panel shows expanded views of 622 peaks.



Figure S4. Powder X-ray diffraction patterns for $Cu_{26+y}V_2Ge_{6-x}Sb_xS_{32}$ with x = 1 samples. Arrows denote peaks from Cu_2S . The simulated pattern based on the colusite ($Cu_{26}V_2Ge_6S_{32}$) structure is shown at the bottom. The right panel shows expanded views of 622 peaks.



Figure S5. Rietveld refinement of power X-ray diffraction patterns for Cu₂₆V₂Ge₆S₃₂.



Figure S6. Rietveld refinement of power X-ray diffraction patterns for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$ with x = 1.



Figure S7. Rietveld refinement of power X-ray diffraction patterns for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$ with x = 2.



Figure S8. Rietveld refinement of power X-ray diffraction patterns for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$ with x = 3.



Figure S9. Rietveld refinement of power X-ray diffraction patterns for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$ with x = 4.

$Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$	x = 0	x = 1	x = 2	<i>x</i> = 3	x = 4
<i>a</i> (Å)	10.5684(2)	10.6026(2)	10.6334(2)	10.6671(2)	10.6988(2)
SOF_Ge(6c)	1.00	0.87(2)	0.69(2)	0.50(2)	0.38(2)
SOF_Sb(6 <i>c</i>)	-	0.13(2)	0.31(2)	0.50(2)	0.62(2)
<i>x</i> _Cu(8 <i>e</i>)	0.248(1)	0.248(1)	0.247(1)	0.245(1)	0.246(1)
<i>x</i> _Cu(12 <i>f</i>)	0.260(1)	0.260(1)	0.257(1)	0.257(1)	0.256(1)
<i>x</i> _S(8 <i>e</i>)	0.126(1)	0.127(1)	0.125(1)	0.123(1)	0.124(1)
$x_S(24i)$	0.371(1)	0.371(1)	0.372(1)	0.375(1)	0.376(1)
$y_S(24i)$	0.373(1)	0.373(1)	0.372(1)	0.370(1)	0.368(1)
$z_S(24i)$	0.121(1)	0.122(1)	0.122(1)	0.126(1)	0.128(1)
χ^2	0.678	0.641	0.575	0.560	0.680
$R_{ m wp}$; $R_{ m exp}$	4.09 ; 4.96	4.19 ; 5.23	3.98 ; 5.25	4.11 ; 5.49	4.62 ; 5.60
$R_{ m Bragg}$; $R_{ m F}$	2.62 ; 7.54	3.05 ; 6.45	2.20 ; 5.37	2.72;5.17	3.03 ; 6.29
Distances (Å)					
$V(2a)$ - $S(8e) \times 4$	2.308	2.325	2.297	2.267	2.296
$V(2a)-Cu(12f)\times 6$	2.751	2.755	2.735	2.741	2.740
M(6c)-S(24 i) × 4	2.254	2.269	2.291	2.349	2.383
$Cu(6d)$ - $S(24i) \times 4$	2.276	2.291	2.281	2.286	2.286
$Cu(8e)$ - $S(8e) \times 1$	2.237	2.220	2.245	2.265	2.255
$Cu(8e)$ - $S(24i) \times 3$	2.286	2.289	2.303	2.297	2.293
$Cu(12f)$ - $S(8e) \times 2$	2.359	2.366	2.346	2.341	2.348
$Cu(12f)-S(24i) \times 2$	2.287	2.291	2.313	2.297	2.300
Cu-S (average)	2.291	2.297	2.300	2.298	2.298

Table S2. Results from the Rietveld refinements of power X-ray diffraction patterns for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$ samples (space group $P\overline{4}3n$).



Figure S10. Annular dark-field scanning transmission electron microscopy images along the (a) 100 and (b) 111 directions for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$ (x = 2) sample.



Figure S11. Line profiles of intensity for atomic columns along the (a) 100 and (b) 111 directions depicted in Figure S10 and the corresponding view of the crystal structure.



Figure S12. Annular dark-field scanning transmission electron microscopy images along the (a) 100 and (b) 111 directions for $Cu_{26+y}V_2Ge_6S_{32}$ (y = 3) sample.



Figure S13. Line profiles of intensity for atomic columns along the (a) 100 and (b) 111 directions depicted in Figure S12 and the corresponding view of the crystal structure.



Figure S14. Rietveld refinement of power X-ray diffraction patterns for $Cu_{26+y}V_2Ge_6S_{32}$ with y = 3.

Table S3. Results from Rietveld refinements of powder X-ray diffraction patterns for y = 0 and y = 3 samples of Cu_{26+y}V₂Ge₆S₃₂ (space group $P\overline{4}3n$).

$Cu_{26+y}V_2Ge_6S_{32}$	y = 0	<i>y</i> = 3	<i>y</i> = 3
<i>a</i> (Å)	10.5684(2)	10.6149(2)	10.6158(2)
$SOF_Cu(24i)^*$	—	—	0.042(1)
<i>x</i> _Cu(8 <i>e</i>)	0.248(1)	0.247(1)	0.246(1)
<i>x</i> _Cu(12 <i>f</i>)	0.260(1)	0.260(1)	0.260(1)
x_S(8 <i>e</i>)	0.126(1)	0.127(1)	0.127(1)
$x_S(24i)$	0.371(1)	0.372(1)	0.372(1)
$y_S(24i)$	0.373(1)	0.372(1)	0.373(1)
<i>z</i> _S(24 <i>i</i>)	0.121(1)	0.121(1)	0.120(1)
χ^2	0.678	0.760	0.651
$R_{\rm wp}$; $R_{\rm exp}$	4.09 ; 4.96	4.21 ; 4.82	3.88 ; 4.81
$R_{ m Bragg}$; $R_{ m F}$	2.62;7.54	7.21;10.5	4.11 ; 9.01

* Cu atoms in interstitial position (0.236, 0.235, 0.007)



Figure S15. Scanning electron microscopy images for $Cu_{26}V_2Ge_{6-x}Sb_xS_{32}$, $Cu_{26+y}V_2Ge_6S_{32}$, and $Cu_{26+y}V_2Ge_{6-x}Sb_xS_{32}$ with x = 1 samples.

Table S4. Hole carrier concentration *n* and Hall mobility $\mu_{\rm H}$ for ${\rm Cu}_{26}{\rm V}_2{\rm Ge}_{6-x}{\rm Sb}_x{\rm S}_{32}$, ${\rm Cu}_{26+y}{\rm V}_2{\rm Ge}_6{\rm S}_{32}$, and ${\rm Cu}_{26+y}{\rm V}_2{\rm Ge}_{6-x}{\rm Sb}_x{\rm S}_{32}$ with x = 1 samples.

	$n / 10^{21} \mathrm{cm}^{-3}$	$\mu_{ m H}$ / cm ² V ⁻¹ s ⁻¹
x = y = 0	6.8	5.5
x = 1, y = 0	5.5	4.0
x = 2, y = 0	4.0	2.8
x = 3, y = 0	1.9	1.9
x = 0, y = 2	5.1	2.6
x = 0, y = 3	4.1	1.6
x = 1, y = 3	1.7	1.0



Figure S16. Lattice thermal conductivity κ_{lat} for $\text{Cu}_{26}\text{V}_2\text{Ge}_{6-x}\text{Sb}_x\text{S}_{32}$, $\text{Cu}_{26+y}\text{V}_2\text{Ge}_6\text{S}_{32}$, and $\text{Cu}_{26+y}\text{V}_2\text{Ge}_{6-x}\text{Sb}_x\text{S}_{32}$ with x = 1 samples.