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## Temperature sensors based on europium polyoxometalate and mesoporous terbium metal-organic framework

Cédric Viravaux,<sup>a</sup> Olivier Oms,<sup>a</sup> Anne Dolbecq,<sup>a</sup> Emma Nassar,<sup>a</sup> Lucas Busson,<sup>a</sup> Caroline Mellot-Draznieks,<sup>b</sup> Rémi Dessapt,<sup>c</sup> Hélène Serier-Brault<sup>c\*</sup> and Pierre Mialane<sup>a\*</sup>

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The first example of ratiometric luminescent thermometers based on lanthanide complexes encapsulated in a lanthanide metal-organic framework (MOF) is reported. In EuW<sub>10</sub>@Tb-TATB, the [EuW<sub>10</sub>O<sub>36</sub>]<sup>9-</sup> polyoxometalate (POM) is incorporated into the cavities of the mesoporous terbium MOF Tb-TATB with two different POM loadings (9.1 and 19.5 wt%). Eu<sup>3+</sup> and Tb<sup>3+</sup> ions of the POM and MOF, respectively, act as dual emitters. Optical measurements evidence efficient Tb<sup>3+</sup>-to-Eu<sup>3+</sup> energy transfer, suggesting that the EuW<sub>10</sub> units are close to terbium centers. These observations are supported by computational investigations whereby a favored localisation of the EuW<sub>10</sub> POM in the MOF's pores is found close to Tb-centers evidencing hydrogen bond type interactions between terminal oxygen atoms of the POM and Tb-coordinated water molecules. The reported materials act as temperature sensors in the physiological domain, exhibiting high relative thermal sensitivities of 2.68 %K<sup>-1</sup> and 2.37 %K<sup>-1</sup> at 300 K for the 9.1 wt% and 19.5 wt% composites, respectively, with a thermal uncertainty of 0.09 K.

### Introduction

The field of luminescent thermometers is in expansion due to the numerous advantages of non-contact devices which can find applications in domains ranging from biomedicine to electronics.<sup>1–3</sup> Best thermometric performances are obtained with ratiometric sensors where two discriminable emission peaks are used, preventing effects of optoelectronic drifts of excitation source or detectors. Many reported ratiometric temperature sensors are based on various lanthanide materials such as fluorides<sup>4,5</sup> or, more recently, coordination polymers and metal-organic frameworks.<sup>6</sup> The latter indeed represent excellent platforms for engineering light-emitting centers and tuning photoluminescence properties. Not surprisingly, lanthanide-bearing MOFs (denoted as LnMOFs) have been the most studied ones.<sup>6</sup> The first LnMOFs ratiometric thermometer, based on two emissions of Tb<sup>3+</sup> at 545 nm and Eu<sup>3+</sup> at 613 nm, have been reported by Cui *et al.* in 2012.<sup>7</sup> Since then, several Eu/Tb mixed LnMOFs have been prepared, with maximal sensitivities reached at temperature ranging from cryogenic to 450 K.<sup>8–15</sup>

Also, the porosity of LnMOFs with a single emitting Ln<sup>3+</sup> ion

can be used to design ratiometric thermometer where the guest molecule acts as a second emitter. However, the incorporation of organic dyes into porous LnMOFs has been scarcely reported so far. A dual-emitting perylene@EuMOF acting as ratiometric temperature sensor in the physiological temperature range has been synthesized *via* an impregnation process using the ZJU-88 MOF ( $S_r = 1.28\%$  at 293 K).<sup>16</sup> In addition, a coumarin derivative C460 has been encapsulated in a TbMOF, exhibiting a high thermal sensitivity of 4.48% at 300 K,<sup>17</sup> while very recently Coumarin 7 was immobilized in a trimesate EuMOF with moderate thermometric properties ( $S_r = 0.5\%$  at 363 K).<sup>18</sup> Nevertheless, to the best of our knowledge, the incorporation of lanthanide complexes into LnMOFs for thermometry remains unexplored. Polyoxometalates (POMs) incorporating lanthanides<sup>19,20</sup> appear as ideal candidates, as they combine excellent luminescence properties (narrow emission bands, large Stokes shift and long lifetime), high stability and rigidity. Furthermore, the incorporation of POMs into MOFs has been extensively studied, in particular for catalytic purposes.<sup>21</sup> It is worth mentioning that the non-luminescent Ln-free POM [SiMo<sub>12</sub>O<sub>40</sub>]<sup>4-</sup> has been also used for templating the formation of Eu/Tb mixed LnMOF for temperature sensing purposes.<sup>22</sup> In addition, we recently reported the incorporation of the highly luminescent europium-containing POM [EuW<sub>10</sub>O<sub>36</sub>]<sup>9-</sup> (EuW<sub>10</sub>) into the UiO-67 MOF, where Eu<sup>3+</sup> ions and the organic linker of the framework act as dual emitters. However, its relative thermal sensitivity was quite low (up to 0.26% K<sup>-1</sup> at 303 K) as a result of the very limited luminescent performances of the organic emitter.<sup>23</sup>

Herein, we report the encapsulation of the EuW<sub>10</sub> POM into the Tb-TATB MOF built of terbium tetranuclear units

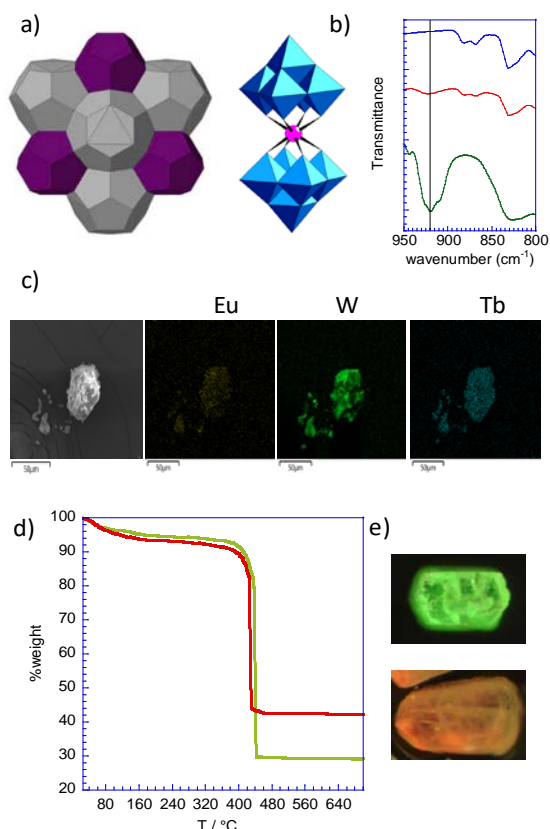
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connected by TATB ligands ( $\text{H}_3\text{TATB}$  = triazine-1,3,5-tribenzoic acid) (Fig. 1a).<sup>24</sup> This mesoporous MOF contains large nanoscopic distorted spherical cages of 3.9 and 4.7 nm diameters accessible through pentagonal and hexagonal windows of 1.3 and 1.7 nm aperture, respectively. This MOF has already been used as host for enzymes via impregnation in water<sup>25</sup> and is well-suited for the incorporation of  $\text{EuW}_{10}$ , whose dimensions are 0.9x1.7 nm. It is demonstrated here that this first dual-luminescent Ln complex@LnMOF composite behaves as a highly sensitive luminescent thermometer in the physiological domain.



**Fig. 1** a) left: truncated supertetrahedra forming the Tb-TATB MOF (from ref 26); right: the  $[\text{EuW}_{10}\text{O}_{36}]^{9-}$  POM. Blue octahedra,  $\text{WO}_6$ , pink sphere, Eu; b) IR spectra of Tb-TATB (blue line),  $\text{EuW}_{10}$  (dark green line) and  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  (red line). The vertical black line indicates the  $\text{W}=\text{O}$  vibrations position; c) SEM image of  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  and EDS elemental mapping of the Eu, W and Tb elements, respectively; d) TGA curves of Tb-TATB (green line) and  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  (red line); e) Images of Tb-TATB (top) and  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  (bottom) under 365 nm irradiation.

## Results and discussion

### Synthesis and characterizations of the composites

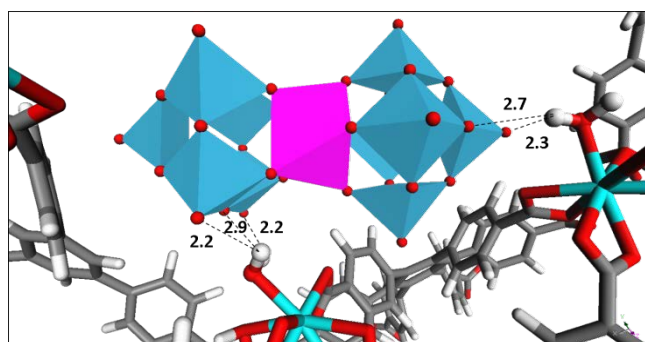
$[\text{Tb}_{16}(\text{TATB})_{16}(\text{DMA})_{24}] \cdot 91\text{DMA} \cdot 108\text{H}_2\text{O}$  (DMA = dimethylacetamide) (Tb-TATB) has been prepared as previously described.<sup>24</sup> The  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  composite was synthesized by mixing  $\text{Na}_9[\text{EuW}_{10}\text{O}_{36}]$  and Tb-TATB in aqueous medium, the resulting white material being thoroughly washed with water in order to remove any POM potentially deposited at the surface of the MOF particles. The

infrared spectrum of  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  mainly presents the feature of the MOF (Fig. S1, ESI<sup>†</sup>). However, a band located at  $920\text{ cm}^{-1}$  and attributed to  $\text{W}=\text{O}$  vibrations (Fig. S1, ESI<sup>†</sup> and 1b) strongly suggests the presence of the POM. The latter is confirmed by SEM-EDS elemental mapping experiments, which show a uniform distribution of Eu, W and Tb elements throughout the material (Fig. 1c). In addition, the Tb:W:Eu ratio of 16:9:1.2 determined by EDS measurements indicates a ratio of 0.9:1 POM: $\{\text{Tb}_{16}\}$  unit while no significant amount of sodium counterions was detected. This implies, as usually observed for POM@MOF materials,<sup>27</sup> that the negative charges introduced by the POMs may be compensated by linker deficiencies. Thermogravimetric analysis (TGA) measurements show mass losses at  $430^\circ\text{C}$  of 70.5% for Tb-TATB and 57.5% for  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  (Fig. 1d). These weight losses are assigned to water removal in a first step and then to the linker's decomposition along with the formation of inorganic oxides. The lower weight loss of  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  with respect to the pristine Tb-TATB solid is indeed in line with the presence of POMs in the POM@MOF material. The following chemical formula  $[\text{Tb}_{16}(\text{TATB})_{13.3}(\text{H}_2\text{O})_{24}](\text{EuW}_{10}\text{O}_{36})_{0.9}(\text{H}_2\text{O})_{40}$  is thus proposed for  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$ , in accordance with both the total weight loss observed by TGA (exp. 57.5%, calc. 56.3%) and the water loss (exp. 9.7%, calc. 8.2%) and corresponding to a  $\text{Tb}_{0.95}\text{Eu}_{0.05}$  molar composition. Besides, we can note that while the nature of the obtained large single-crystals of the Tb-TATB precursor has been checked by single-crystal X-Ray diffraction, affording the expected cell parameters,<sup>24</sup> only low resolution powder X-Ray diffraction patterns were obtained for both the MOF and POM@MOF compounds (Fig. S2, ESI<sup>†</sup>). Finally, Fig. 1e and S3, ESI<sup>†</sup> show pictures of Tb-TATB and  $19.5\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$  crystals under visible and UV-light irradiation. It can be clearly seen that the green luminescence of Tb-TATB turns orange after  $\text{EuW}_{10}$  impregnation. With the aim to assess the influence of the POM loading on the luminescent performances,  $9.1\text{wt}\%\text{EuW}_{10}@Tb\text{-TATB}$ , corresponding to the  $[\text{Tb}_{16}(\text{TATB})_{14.8}(\text{H}_2\text{O})_{24}](\text{EuW}_{10}\text{O}_{36})_{0.4}(\text{H}_2\text{O})_{40}$  composite was also prepared following a similar synthetic procedure but working at a lower POM:Tb-TATB ratio (see experimental section in the ESI for details).

### Computational investigations

We further probed the likely positions of the  $\text{EuW}_{10}$  POMs within the porous Tb-TATB through a computational approach using simulated annealing (SA) calculations (see ESI for details) as in previously reported procedures.<sup>28</sup> Typically, the POMs are found located in the vicinity of Tb sites accessible at the MOF's internal surface, whereby they are stabilized by hydrogen bond interactions with water molecules coordinated to the Tb centers. Fig. 2 illustrates one such position of the POM as extracted from the lowest energy ones computed in SA calculations. The  $\text{EuW}_{10}$  POM is shown here to benefit from multiple H-bond interactions with two neighboring Tb centers, with  $\text{W}=\text{O}\cdots\text{H}(\text{H}_2\text{O})$  distances in the 2.2–2.8 Å range, allowing here a  $\text{Tb}\cdots\text{Eu}$  distance of 6.3 Å. Interestingly, the simulations

show that the  $\text{EuW}_{10}$  POMs tend to locate preferably in the vicinity of Tb centers possessing water molecules in their coordination sphere, leaving Tb centers coordinated exclusively to TATB linkers unvisited.



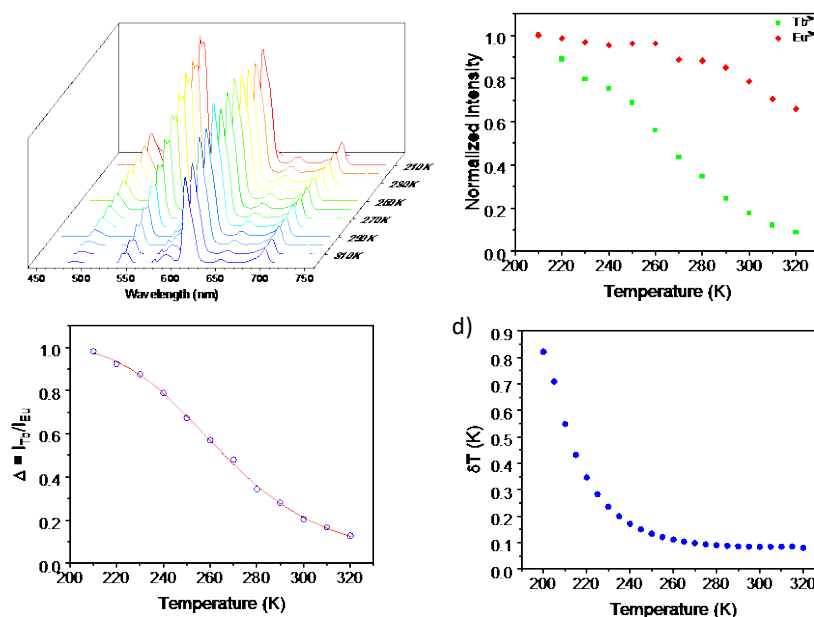
**Fig. 2** Representation of  $\text{EuW}_{10}@Tb\text{-TATB}$  where the position of the POM is obtained from computations (see text).  $\text{WO}_6$ , blue polyhedra;  $\text{EuO}_8$ , pink polyhedron; O, red spheres; C, gray sticks; Tb, turquoise sticks; H, white sticks.

### Optical properties

The photophysical properties of both **19.5wt% $\text{EuW}_{10}@Tb\text{-TATB}$**  and **9.1wt% $\text{EuW}_{10}@Tb\text{-TATB}$**  were thoroughly

investigated at room temperature. First, their emission spectra (Fig. S4, ESI<sup>†</sup>) display the characteristic green and red luminescence of  $\text{Tb}^{3+}$  and  $\text{Eu}^{3+}$  ions, respectively, with the typical lines at 490, 545, 581, and 619 nm attributed to the  $\text{Tb}^{3+} \ ^5D_4 \rightarrow \ ^7F_{6-3}$  transitions, and at 587, 611, 652, and 695 nm for the  $\text{Eu}^{3+} \ ^5D_0 \rightarrow \ ^7F_{1-4}$  transitions. As already observed for  $\text{EuW}_{10}@UiO\text{-}67$ ,<sup>23</sup> the intensity of the  $^5D_0 \rightarrow \ ^7F_2$  transition with respect to that of the  $^5D_0 \rightarrow \ ^7F_1$  magnetic dipole transition is significantly increased in the  $\text{EuW}_{10}@Tb\text{-TATB}$  composites when compared to the same transitions observed for the sodium salt of  $\text{EuW}_{10}$ , which confirms that the site symmetry of the  $\text{Eu}^{3+}$  ion in  $\text{EuW}_{10}$  decreases when the POM is introduced into the MOF pores.

The photoluminescence excitation (PLE) spectrum of **19.5wt% $\text{EuW}_{10}@Tb\text{-TATB}$**  monitored within the  $^5D_4 \rightarrow \ ^7F_5$   $\text{Tb}^{3+}$  transition (Fig. S5, ESI<sup>†</sup>) displays mainly a broad band between 300 and 370 nm, which is due to the  $\pi\text{-}\pi^*$  transition of the TATB ligand, the f-f lines being very weak. When monitored within the  $^5D_0 \rightarrow \ ^7F_2$   $\text{Eu}^{3+}$  transition, the PLE spectrum is composed of a broad band centered at 280 nm, corresponding to the O $\rightarrow$ W ligand-to-metal charge transfer (LMCT) in the  $\text{EuW}_{10}$  moiety, and of the MOF's ligand broad band together with the f-f lines of  $\text{Eu}^{3+}$ . Thus, the presence of the LMCT band confirms



**Fig. 3** a) Emission spectra of **9.1wt% $\text{EuW}_{10}@Tb\text{-TATB}$**  in the 200–320 K range with the excitation fixed at 330 nm; b) Corresponding temperature dependence of the  $I_{Tb}$  (green) and  $I_{Eu}$  (red) parameters; c) Corresponding temperature dependence of  $\Delta$  in the 200–320 K range and Mott-Seitz fit (red line,  $R^2 = 0.999$ ); d) Corresponding temperature uncertainty ( $\delta T = 1/S_e \times \delta\Delta/\Delta$ ).  $\delta\Delta/\Delta$  was estimated by the average relative errors in the same temperature range (0.2%).

the integrity of the POM  $\text{EuW}_{10}$  in the POM@MOF composite, while the presence of the excitation bands of the MOF evidences an intermolecular MOF-to- $\text{Eu}^{3+}$  energy transfer. Hereafter, the excitation wavelength will be fixed at 330 nm, which corresponds to the intersection between the PLE spectra monitored at 541 nm ( $\text{Tb}^{3+}$  transition) and 617 nm

( $\text{Eu}^{3+}$  transition), where both emitters are equally sensitized. Moreover, a  $\text{Tb}^{3+}$ -to- $\text{Eu}^{3+}$  energy transfer is discerned at room-temperature, as shown by the presence of the  $^7F_6 \rightarrow \ ^5D_4$  (at 483 nm)  $\text{Tb}^{3+}$  transition in the excitation spectrum monitored within the  $^5D_0 \rightarrow \ ^7F_2$   $\text{Eu}^{3+}$  transition (Fig. S5 insert, ESI<sup>†</sup>). Emission lifetimes were determined at room-temperature for

both  $\text{EuW}_{10}@\text{Tb-TATB}$  compounds and for  $\text{Tb-TATB}$ . For all samples, a monoexponential behavior was observed for the  $^5\text{D}_4$  decay curve monitored at 541 nm ( $\text{Tb}^{3+}$  transition) and for the  $^5\text{D}_0$  decay curve monitored at 617 nm ( $\text{Eu}^{3+}$  transition), the different lifetimes being given in Table S1. Thus, the  $\text{Tb}^{3+}$ -to- $\text{Eu}^{3+}$  energy transfer efficiency was determined from the equation (1):

$$\eta_{ET} = 1 - \frac{\tau}{\tau_0} \quad (1)$$

with  $\tau$  and  $\tau_0$  corresponding to the lifetimes of the donor in the presence and absence of the acceptor, respectively. Thus, the  $\text{Tb}^{3+}$ -to- $\text{Eu}^{3+}$  energy transfer amounts to 40% and 37% for **9.1wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  and **19.5wt% $\text{EuW}_{10}@\text{Tb-TATB}$** , respectively. Finally, as expected by computational investigations one may assume that the  $\text{EuW}_{10}$  entities are close enough to the  $\text{Tb}^{3+}$  ions of the MOF to lead to a non-negligible  $\text{Tb}^{3+}$ -to- $\text{Eu}^{3+}$  energy transfer. This observation is indeed corroborated by the above SA calculations which point towards the vicinity of Tb centers in the POM's environment.

The application of the  $\text{EuW}_{10}@\text{Tb-TATB}$  composites as ratiometric luminescent thermometers was characterized using the emission spectra in the 200–320 K range. As an illustrative example, Fig. 3a presents the temperature dependent emission spectra of **9.1wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  in the 200–320 K range (similar data for the **19.5wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  material is reported in Fig. S6a, ESI<sup>†</sup>). Five consecutive emission spectra were recorded at each temperature. The temperature dependence of the  $I_{\text{Tb}}$  (green) and  $I_{\text{Eu}}$  (red) parameters is represented in Fig. 3b. The integrated areas of the  $^5\text{D}_4 \rightarrow ^7\text{F}_5$  ( $I_{\text{Tb}}$ ) and of the  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  ( $I_{\text{Eu}}$ ) emissions were used to define the thermometric parameter as  $\Delta = I_{\text{Tb}}/I_{\text{Eu}}$ ,  $I_{\text{Tb}}$  and  $I_{\text{Eu}}$  being obtained using the emission spectra in the 530–560 nm and 602–633 nm intervals, respectively.

The thermometric parameter  $\Delta$  of both samples (Fig. 3c and Fig. S6c, ESI<sup>†</sup>) is depicted according to the temperature in the 200–320 K range. It may be described by the phenomenological Mott-Seitz expression involving a single deactivation channel assuming a temperature dependence of the  $\text{Eu}^{3+}$  emission smaller than that of the  $\text{Tb}^{3+}$  emission (Fig. 3b and Fig. S6b, ESI<sup>†</sup>) according to the equation (2):

$$\Delta(T) = \frac{\Delta_0}{1 + \alpha \exp(-\Delta E/k_B T)} \quad (2)$$

where  $\Delta_0$  is the  $\Delta$  parameter at  $T \rightarrow 0$  K,  $\alpha = W_0/W_R$  is the ratio between the non-radiative and radiative decay rates in the limit of  $T \rightarrow 0$  K and  $\Delta E$  is the energy difference between the emitting level ( $^5\text{D}_4$ ) and the level responsible for its nonradiative deactivation. Fitting Equation (2) to the experimental  $\Delta(T)$  curves, we find excellent correlation coefficients attesting the adequacy of this simple model to get the calibration curve of the hybrid materials (Fig. 3c and Fig. S6c, ESI<sup>†</sup>). Finally, the  $\Delta E$  value was found equal to  $2119 \pm 70$   $\text{cm}^{-1}$  for **9.1wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  and  $1932 \pm 65$   $\text{cm}^{-1}$  for **19.5wt% $\text{EuW}_{10}@\text{Tb-TATB}$** , which can be ascribed to the back-energy transfer from the  $^5\text{D}_4$  level of  $\text{Tb}^{3+}$  ( $20500$   $\text{cm}^{-1}$ )<sup>6</sup> to the

triplet level of the TATB ligand ( $T_1 = 21380$   $\text{cm}^{-1}$ ).<sup>29</sup> This is confirmed by the thermal evolution of  $^5\text{D}_4$  lifetimes which decreases when the temperature increases whereas the  $^5\text{D}_0$  lifetime is almost constant (Fig. S7 and S8, ESI<sup>†</sup>). The same  $^5\text{D}_4$  lifetime thermal quenching was also observed on the  $\text{Tb-TATB}$  MOF (Fig. S9, ESI<sup>†</sup>), indicating that the  $\text{EuW}_{10}$  entity plays the role of reference in the ratiometric thermometers. Then, in order to evaluate the thermal performances of mixed compounds, we calculated the relative thermal sensitivity defined as  $S_r = |\partial\Delta/\partial T|/\Delta$  which is a usual figure of merit to compare the performance of distinct systems.<sup>30</sup> Both materials exhibit good thermal sensitivity in the physiological range (293–323 K) with a maximum of relative thermal sensitivity of 2.68  $\%K^{-1}$  at 300 K for **9.1wt% $\text{EuW}_{10}@\text{Tb-TATB}$** , and 2.37  $\%K^{-1}$  for **19.5wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  (Fig. 4). These values are higher than those of most previously reported luminescent thermometers operating in the physiological range. Finally, the temperature uncertainty  $\delta T = 1/S_r \times \delta\Delta/\Delta$ , which defines the smallest temperature change that may be ascertained in a given measurement, was evaluated as amounting to 0.08 K for both samples on a wide temperature range (from 280 to 320 K), which makes them very promising materials for biological applications (Fig. 3d and S6d, ESI<sup>†</sup>).

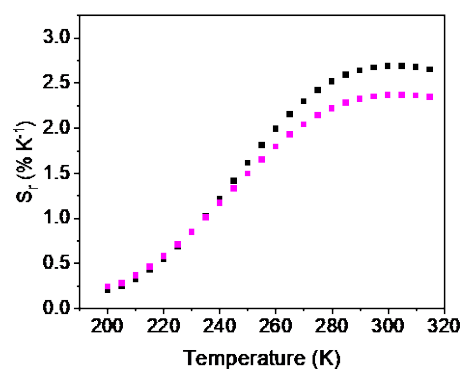


Fig. 4 Relative thermal sensitivity of **9.1wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  (black line) and **19.5wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  (pink line) in the 200–320K range.

## Conclusions

The red emitting  $\text{EuW}_{10}$  POM was successfully incorporated into the mesoporous green emitting  $\text{Tb-TATB}$  MOF. According to the POM:MOF ratio used for the impregnation step, two composites, **9.1wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  and **19.5wt% $\text{EuW}_{10}@\text{Tb-TATB}$** , were prepared and characterized. The temperature-dependence of their optical properties were investigated and undoubtedly pointed out an energy transfer from the  $\text{Tb}$  MOF network to the  $\text{Eu}^{3+}$  centers. The latter is supported by the computed positioning of the  $\text{EuW}_{10}$  within the  $\text{Tb-TATB}$  host whereby it is favorably immobilized near  $\text{Tb}$ -centers through hydrogen bond type interactions with  $\text{Tb}$ -coordinated water molecules. As a result, these composites appear as effective ratiometric luminescent thermometers, their relative thermal sensitivity values (2.68  $\%K^{-1}$  for **9.1wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  and 2.37  $\%K^{-1}$  for **19.5wt% $\text{EuW}_{10}@\text{Tb-TATB}$**  at 300 K) clearly evidencing their high optical performance in the physiological

domain. These composites constitute the first members of a new family of hybrid dual-emitting POM@MOF materials. The richness of the POMs and MOFs chemistry makes it possible to envisage a multitude of related systems, including other LnPOM@LnMOFs composites with tunable POM/MOF interactions, but also the co-incorporation of POMs with other fluorophores (quantum dots,...) in non-luminescent MOFs for the development of new POM-based luminescent thermometers. Both strategies are currently under investigation in our groups.

## Experimental section

All reagents were purchased and used without further purification except  $\text{Na}_9[\text{EuW}_{10}\text{O}_{36}] \cdot 32\text{H}_2\text{O}$  ( $\text{EuW}_{10}$ )<sup>31</sup> and  $[\text{Tb}_{16}(\text{TATB})_{16}(\text{DMA})_{24}] \cdot 91\text{DMA} \cdot 108\text{H}_2\text{O}$  ( $\text{Tb-TATB}$ )<sup>24</sup> which were synthesized according to reported procedures. Prior to the impregnation of the polyoxometalate,  $\text{Tb-TATB}$  was washed with water and ethanol and dried overnight.

**Preparation of 19.5wt%EuW<sub>10</sub>@Tb-TATB.** In a scintillation vial, 24 mg of  $\text{EuW}_{10}$  (7.3  $\mu\text{mol}$ ) were dissolved in 3 mL of water. Then, 24 mg of crushed  $\text{Tb-TATB}$  (1.1  $\mu\text{mol}$ ) was added, and the suspension stirred at 60°C for one night using a shaker. The obtained white powder was then filtrated, thoroughly washed with water, ethanol and dried overnight ( $m = 18$  mg). EDS: Tb : W : Eu calc. : 16 : 9 : 0.9, exp. : 16 : 9 : 1.2. IR ( $\text{v}/\text{cm}^{-1}$ ): 1421 (m), 1393 (s), 1377 (s), 1353 (vs), 1136 (m), 1103 (m), 1015 (m), 976 (w), 922 (w), 868 (w), 830 (m), 798 (m), 771 (s), 699 (m), 543 (m), 493 (s).

**Preparation of 9.1wt%EuW<sub>10</sub>@Tb-TATB.** This compound was prepared as described for **19.5wt%EuW<sub>10</sub>@Tb-TATB** but using 16 mg of  $\text{Tb-TATB}$  (0.7  $\mu\text{mol}$ ) and 3 mg of  $\text{EuW}_{10}$  (0.9  $\mu\text{mol}$ ). 13 mg of **9.1wt%EuW<sub>10</sub>@Tb-TATB** were finally obtained. EDS: Tb : W calc. : 16 : 4, exp.: 16 : 4. IR ( $\text{v}/\text{cm}^{-1}$ ): 1423 (m), 1399 (s), 1378 (s), 1359 (vs), 1137 (m), 1103 (m), 1015 (m), 931 (w), 868 (w), 829 (m), 796 (m), 774 (s), 702 (m), 552 (m), 496 (s).

## Author Contributions

The project was conceived and designed by PM with significant input from AD and OO. CV, EN and LB prepared the samples and performed the characterizations. CMD carried out the computational investigations. RD and HSB performed the optical measurements. The first manuscript draft was prepared by PM, HSB and CMD. All the authors reviewed the manuscript and participated actively in the discussion of the results.

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## Conflicts of interest

There are no conflicts to declare.

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