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Sombel Diaham, Z. Valdez-Nava, T. T Le, L. Lévêque, Lionel Laudebat, et al.. Field Grading Composites Tailored by Electrophoresis-Part 3: Application to Power Electronics Modules Encapsulation. IEEE Transactions on Dielectrics and Electrical Insulation, 2021, 28 (2), pp.348-354. 10.1109/TDEI.2020.009032 . hal-03266268

HAL Id: hal-03266268 https://hal.science/hal-03266268

Submitted on 21 Jun 2021

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Field Grading Composites Tailored by Electrophoresis – Part 3: Application to Power Electronics Modules Encapsulation

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ABSTRACT

A series of three articles presents an innovative way to build advanced functionally graded materials (FGM) based on polymer/ceramic (epoxy/SrTiO₃) composites tailored by electrophoresis for field grading in power electronics. In this Part 3, this method is applied in the context of power modules for DBC substrate encapsulation. An evaluation of the FGM performances is reported based on electrostatic simulations and breakdown voltage measurements on encapsulated DBC substrates. The results show a significant mitigation of the electric fringe field at the triple point while breakdown is largely increased by a factor 2 for FGM composites compared to neat epoxy. The process enables to use the electric field reinforcements of HV electrical systems (e.g. tips coming from the design), and thus potential weak points, to locally 'self-heal' them in-situ. Such an electrophoresis process used to build FGM composites paves the way of the next generation of functionalized polymer composites used in high voltage power applications for improving the electrical aging of insulating materials and power system reliability.

Index Terms — functionally graded materials (FGM), composite, electrophoresis, insulation, encapsulation, high voltage, field grading, power electronics modules

1 INTRODUCTION

IN power electronics, the recent and still ongoing development of high power semiconductor devices such as insulated gate bipolar transistor (IGBT) in Si [1] or metal oxide semiconductor field effect transistor (MOSFET) in SiC [2, 3] needs to be supported by higher performances packaging materials to enable higher voltage power module emergence. Among them, the encapsulation materials need to have superior electrical insulation properties (e.g. higher dielectric breakdown and partial discharge inception voltage) in order to increase the working voltage of power modules and/or to enhance their reliability. Usually, the encapsulation function is achieved using silicone gel or epoxy resin, depending on their topology. However, during high voltage (HV) operation of power modules, partial discharges and electrical treeing can occur at the triple point [4, 5], namely at the junction between the ceramic substrate, the metallization and the encapsulation, due to the very high electric field developed at that point, as shown in Figure 1. Treeing can propagate within the encapsulation or along the interface with the DBC substrate towards the grounded counter electrode [6, 7]. This triple point acts as a fringing electrical field region [8, 9] where discharges can incept, then progressively electrically degrade the encapsulation material, up to finally fail the insulation in power modules.

The electric field divergence around the sharp edges of an HV metallization has been a well-known problem in high voltage electrical engineering for decades. A particular way to address it, from the material point of view, has been to involve *field grading materials* in the triple point region in order to benefit of the electrical properties of the insulator itself to reduce efficiently the field reinforcement. Even though such an approach is widely used for HV cable connectors [10], HV electrical machine and other HV apparatus [11-13], it has been rarely tried in HV power electronics modules [14-16] with a very few studies that involve composites as field grading materials.

To figure out more efficiently the electric field divergence issues, some researchers have developed field grading material concept by structuring them based on a non-homogeneous distributed composite: *functionally graded materials* (FGM). Among them, Okubo's group has pioneered FGM composites with internal particle distribution gradient processed by centrifugation force for GIS applications [17, 18].

Manuscript received on 29 may 2020, in final form 30 November 2020, accepted xx Month 20yy. Corresponding author: S. Diaham

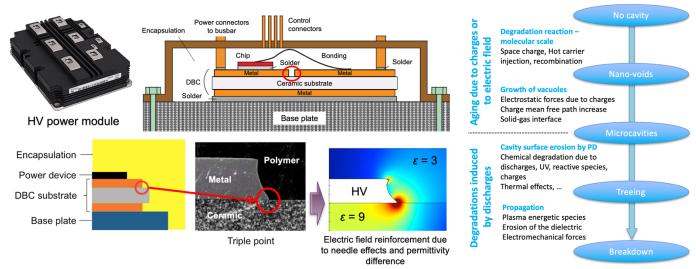


Figure 1. (a) Cross-section view of a HV power electronics module. The red circle shows the triple point location where the electric field is strongly enhanced (see FEM modelling). (b) General electrical degradation chart leading to the dielectric breakdown.

However, for power electronics systems (as power modules) with more complex electrode design, the centrifugation process is not suitable any longer to tailor locally a property gradient exactly where the electric field is the most concentrated.

Previously, in Parts 1 and 2 of this three-articles series [19, 20], an innovative electrophoresis process was reported to build a polymer composite with a permittivity gradient (see Figure 2) that redistribute the electric field at the electrode sharp edge thanks to the self-organization of the high-*k* particles. In this present and last Part 3, such a concept of FGM design is applied to the encapsulation of power module substrates to evaluate its impact on the field grading efficiency and on the breakdown voltage.

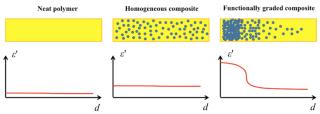


Figure 2. Impact of high permittivity particles arrangement in a polymer composite on the permittivity gradient along a distance: (a) neat polymer, (b) particle homogenous distribution, (c) particle gradient distribution (FGM with gradient).

2 EXPERIMENTAL

2.1 MATERIALS AND COMPOSITE COMPOUNDS

The polymer used as the host matrix is a diglycidyl ether of bisphenol F (DGEBF) thermosetting epoxy resin. It is crosslinked using an amine-type hardener with a ratio of 10:1 (resin:hardener). The viscosity of the epoxy resin with harder is of 3000 mPa·s at room temperature and its density after curing is 1.23 g/cm³.

The high permittivity fillers used are strontium titanate (SrTiO₃) particles with a density of 5.11 g/cm³ and a BET specific surface of 1.83 m²/g. They are polyhedral-shaped

and have an average diameter of 600 nm. All the details can be found in ref. [19].

The composite mixture was elaborated by a direct blending process of the $SrTiO_3$ particles (10 vol%) into the liquid epoxy resin containing the hardener. The particles were then homogeneously dispersed within the matrix using a planetary mixer rotating at 3000 rpm for 8 minutes. The mixture was then degassed in primary vacuum (<1.5 kPa) before being poured on the electrophoretic test structures.

2.2 ELECTROPHORESIS PROCESS ON DBC SUBSTRATES

To evaluate the FGM insulation properties in the context of power electronics module's encapsulation, the material has been applied onto direct-bounded copper (DBC) substrates with an alumina ceramic.

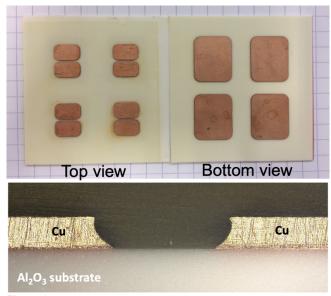


Figure 3. DBC substrates used during this study: top, bottom and crosssection views.

These substrates are typically used in power electronics modules for intermediate power with a 635 µm-thick Al₂O₃ ceramic and 300 um-thick copper metallization. In order to apply the electrophoresis process, lateral Cu tracks were designed and chemically etched on the top side of the DBC test structure, while a unique wider electrode was patterned, as the bottom electrode (see Figure 3). Lateral gap distances for the top electrodes, ranging from 300 µm up to 1 mm, were designed. The etching profiles of the Cu layers is representative of those found in power modules where a concave shape and sharp tip edges of the Cu tracks is usually observed, as shown on the cross-section view in Figure 3. A set of four individual test structures on each DBC substrate was patterned and several of them have been produced to perform a breakdown statistical analysis for each type of encapsulation.

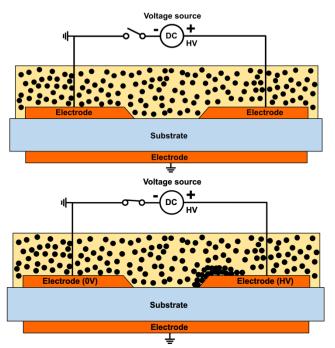


Figure 4. Electrophoresis process applied on DBC substrates to accumulate SrTiO₃ particles in a divergent electric field.

These DBC test structures were then encapsulated with different materials: a neat epoxy (as a reference), an homogeneous (or randomly-dispersed) epoxy/SrTiO₃ composite and a FGM-type epoxy/SrTiO₃ composite for which an electrophoresis process was applied. To achieve this latter, a DC voltage set at 500 V for 15 min between the two top tracks (through soldered wires) was applied. Figure 4 shows the general principle of the particles accumulation under electrophoresis in DC towards the HV electrode. Further details about this process under a uniform and non-uniform electric fields can be found in previous papers [19-23]. After switching the voltage off, the DBC test structures were put into an oven at 150 °C for 20 min in air to crosslink the epoxy resin and thus to 'freeze' the particles within the matrix.

3 RESULTS AND DISCUSSION

3.1 FIELD ATTENUATION LAYER WITH PERMITTIVITY DISTRIBUTION

Figure 4 shows a cross-section optical microscope views of a cured FGM-type epoxy/SrTiO₃ composites after electrophoresis. As detailed in ref. [20], the sharp electrode configuration has as main consequence to concentrate the high-k particles around the highest electric field regions (i.e. around the tip regions of the HV Cu tracks). Consequently, the FGM composite is made up of an accumulated particle layer of higher permittivity (field grading attenuation layer) around the HV electrode and of a slightly depleted homogenously-dispersed bulk.

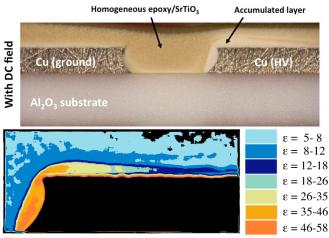


Figure 5. Cross-section optical microscope image of a cured FGM-type epoxy/SrTiO₃ composite after the electrophoresis process (top image). 2D numerical reconstruction of the permittivity distribution within and around the accumulated particle layer after [20] (bottom image).

A method, based on numerical image treatments of the particle concentration [20], had enabled to determine the permittivity distribution within the accumulated particle layer. As show in Figure 5b, accumulation follows the gradient of particle concentration and noteworthy that the permittivity has been self-structured in the form of a gradient. It follows the electric fringe field with higher permittivity values around the electrode sharp corners (ϵ >35) while it decreases all along the FGM layer on the topside of the metallization until progressively reaching down ϵ ~6 (as for the randomly-dispersed composites) where the field is the lowest.

3.2 FEM MODELLING OF THE FIELD GRADING EFFICIENCY

To finally evaluate the impact of the FGM layer structuration on the electric field distribution, FEM simulations (COMSOL Multiphysics) were performed by implementing the layer into the DBC structure model. An electrostatic analysis has been performed by solving the Poisson's equation. The exact shape of the FGM particle layer has been designed under the simulation software. The new DBC structure, including the field grading layer, was then meshed.

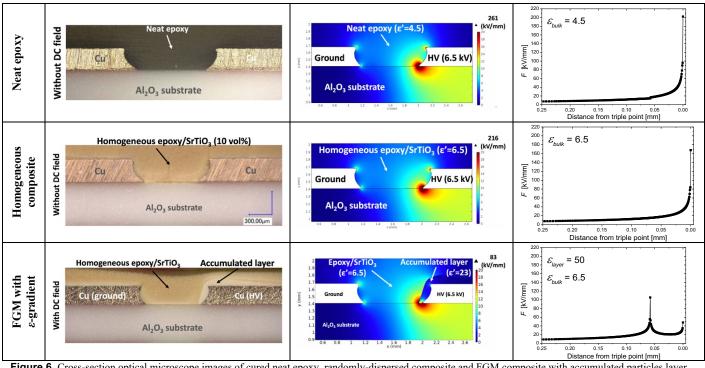


Figure 6. Cross-section optical microscope images of cured neat epoxy, randomly-dispersed composite and FGM composite with accumulated particles layer. FEM simulations of the electric field under 6.5 kV: 2D cartography and 1D profile along the interface from the triple point.

For a more accurate analysis near the curved lines, the mesh was refined in these zones with a variable quadratic tetrahedral element for a more precise representation of the corner's shape and tips [24]. Around the triple point zone, the mesh was refined to allow the calculation of the electric field every 0.4 μ m. The mesh was kept constant for all simulations and permittivity was adjusted for the different cases (more details can be found in [20]). The applied potential on the metallization covered by the FGM layer was 6.5 kV while the non-covered and counter electrodes were both grounded. For convenient reasons, the FGM layer was defined by a constant permittivity and this latter was varied from low to high values for the entire layer.

Figure 6 shows the 2D electric field cartography and 1D cutline field profile for neat epoxy, homogeneous composite and FGM epoxy/SrTiO₃ composite with a layer permittivity value of ε_{layer} =50 (set as constant in the entire layer for more convenience). In the cases of neat epoxy (ε_{bulk} =4.5) and homogeneous epoxy/SrTiO₃ composite (ε_{bulk} =6.5), one electric field reinforcement peak appears at the triple point and another at the top corner of the metallization. The cutline profiles along the ceramic/encapsulation interface (at 1 µm above) show the fringing electric field magnitudes at the triple point. The homogeneous composite with higher permittivity enables to decrease the electric field peak by 20% compared to epoxy. On the other hand, in the presence of the FGM layer with higher permittivity than that of the bulk ($\varepsilon_{layer} > \varepsilon_{bulk}$), field mitigation occurs. Firstly, the top corner field peak almost disappears due to the layer permittivity. Secondly, the triple point fringe field is now split into two peaks each of lower magnitude.

Figure 7 summarizes the maximum electric field dependency versus the FGM layer permittivity variation. As said above, the homogeneous composite already participates to attenuate the electric field peak magnitude. However, the maximum electric field can be further mitigated in the case of the FGM composites. Particularly, an optimal permittivity range for the FGM layer has been identify for $15 \le \epsilon_{layer} \le 50$ wherein the electric field is reduced by $\ge 50\%$ with a minimum for $25 \le \epsilon_{layer} \le 30$.

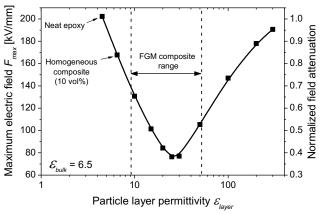


Figure 7. Maximum electric field and normalized field attenuation as a function of the FGM layer permittivity (voltage: 6.5 kV).

Noteworthy that a too high $\varepsilon_{layer} \ge 100$ induces an undesired field grading effect by re-increasing the peak field magnitude at the FGM layer/bulk interface.

In the case of the epoxy/SrTiO₃ FGM layer presenting an internal permittivity gradient $12 \leq \varepsilon_{layer} \leq 50$ (as shown in Figure 5), such values fall in the permittivity range that is required around the triple point in order to minimize the nonuniform electric field of the DBC substrate under concern.

3.3 AC BREAKDOWN VOLTAGE

To experimentally confirm the field grading efficiency at the triple point with the FGM structuration, AC breakdown measurements between the two lateral Cu electrodes of a sample population of encapsulated DBC substrates were performed. Thus, various DBC substrates, as those shown in Figure 3, have been processed with neat epoxy, homogeneous composite and the FGM electrodeposited structuration with permittivity gradient. An AC 50 Hz voltage with a rising ramp of 1 kV_{rms}/s was applied on each DBC test structure until reaching the breakdown. All the tests were done at room temperature by immersing the DBC structures into an insulating oil (Galden HT55, Solvay) to avoid surface flashover.

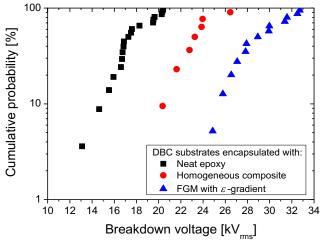


Figure 8. Weibull probability for the AC breakdown voltage of encapsulated DBC substrates between Cu tracks separated by 400 μ m. Comparison of the type of encapsulation: neat epoxy, homogeneous composite and FGM with ϵ -gradient.

Figure 8 presents the Weibull probability for the AC breakdown voltage of the encapsulated DBC substrates between Cu tracks separated by a 400 μ m gap distance and comparing the three different types of encapsulation (neat epoxy, homogeneous composite and FGM with ϵ -gradient).

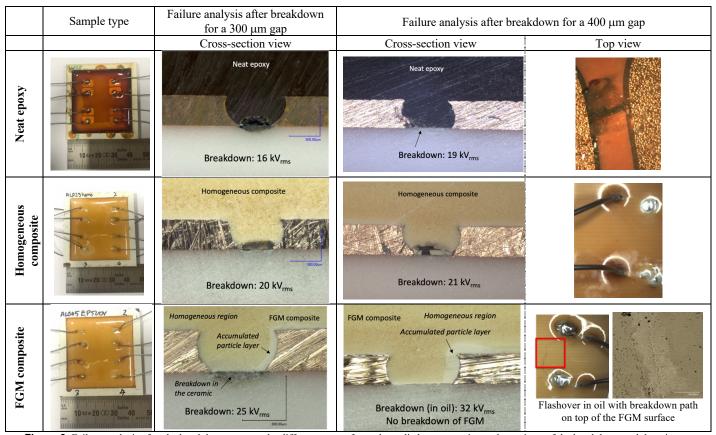


Figure 9. Failure analysis after the breakdown tests on the different types of sample studied: cross-section and top views of the breakdown path location.

For the neat epoxy encapsulated DBC structures, the results show a breakdown voltage ranging between 13 to around 20 kVrms. The homogeneous epoxy/SrTiO3 composite encapsulated DBC structures present a first improvement of the breakdown voltage with values from 20 to 26 kV_{rms}. Finally, the ε -graded FGM composite structure type presents the highest enhancement of the breakdown voltage with values from 25 to 33 kV_{rms}. These results clearly highlight the large enhancement of the breakdown values in the case of the FGM with permittivity gradient with 70% and 30% of improvement compared to neat epoxy and the homogeneous composite, respectively. Such improvements are definitively a confirmation of the field grading effect that spread the equipotential lines and mitigate the electric field around the triple point region, as shown in Figures 6 and 7 by FEM modeling.

3.4 FAILURE ANALYSIS

Post-breakdown analyses of the failure location were done on each type of samples and for two different tested Cu gap distances (300 and 400 µm). Figure 9 shows the crosssection and top-view optical microscopy images after the breakdown tests for the different types of samples. In the reference case of the neat epoxy, one can observe both in cross-section and top-view that the failure has clearly occurred between the Cu electrodes showing an electrical weakness starting from the triple point. In the homogeneous composite case, due to opacity issues the top-view postbreakdown analysis does not bring any information. However, the cross-section images clearly show a failure location between the electrodes within the composite whatever the gap distance. Finally, the most interesting case is that of the FGM with permittivity gradient. In this last situation, two different behavior were observed. Firstly, for the smallest gap distance of 300 µm, a failure location mode between the electrode but within the ceramic has been clearly seen. It shows that the spread electric field from the triple point becomes then stronger within the alumina ceramic leading to its failure before the FGM composite one. Secondly, for larger electrode gap distances as 400 µm, neither the FGM composite nor the Al₂O₃ ceramic substrate has failed. Indeed, in this configuration the surrounding insulating oil was the electrical weak medium and a surface flashover did occur at the surface with the encapsulation (as shown with the surface treeing formed between the soldered bonds). Thus, the breakdown data for the FGM composites in Figure 8 mainly corresponds to surface flashover in the oil and so to inconsistent values: the genuine intrinsic breakdown data should occur at higher voltage. Here also, this proves the efficiency of the developed field grading material to increase the breakdown voltage.

Consequently, such a local self-adaptation of the dielectric properties (in this case with a higher permittivity region) in the vicinity of the geometric 'tip effect' zones of an electronic system, like a power module, appears as a great novelty in the design of tailored insulation system. One of the main reasons and motivations is that it opens new ways to smartly structure advanced insulating materials able to locally relax the constraints and to selectively mitigate the

electric field geometrical defects related to HV electrical system designs.

4 CONCLUSION

This third and last paper of this three-articles series reports on an original method to tailor innovative functionally graded materials (FGM) by local handling of high permittivity ceramic particles in a polymer encapsulation using electrophoresis. This new way of tailoring composites allows to build directly in-situ FGM with permittivity gradient between the high voltage and the ground electrodes of a power systems in order to reduce the electrical stress. A particle layer is so self-arranged conformally around the highest electric field regions developed at the triple point and wherever the field is high (i.e. tip regions). This layer presents locally a permittivity gradient where the highest values are located around the electric field peaks. This structuration participates to selfmitigate the fringe field region by an efficient spreading effect. A clear demonstration of the FGM performances is reported based on electrostatic field simulations and breakdown voltage measurements on encapsulated DBC substrates used in power electronics modules. The results have shown a significant mitigation of the field reinforcement at the triple point while the breakdown voltage has been largely increased. It is so possible to use the electric field weaknesses of any HV electrical system (i.e. the tips coming from the design) to selectively adapt the permittivity gradient in the areas that need it. Complementary studies will have to be carried out to look at the impact of FGMs on mechanical and thermal properties, but also on potentially interesting multilayer FGMs like those developed for GIS spacers [25]. However, such an electrophoresis process used to build FGM composites paves the way of what could be the next generation of functionalized polymer composites used in high voltage power applications for improving the electrical aging of insulating materials and power system reliability.

ACKNOWLEDGMENT

The authors would like to thank the French Research National Agency (ANR) for funding this work in the frame of the ELECTRON project N°ANR-13-JS09-0007-01 (ANR JCJC 2013-2017).

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