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# Enhancement of both optical and catalytic activity of copper-decorated porous silicon micro-particles

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**Abstract.** To the best of our knowledge, this study is the first to investigate the effect of chemical vapour etching (CVE) combined with copper decoration on both the optical and catalytic activities of silicon micro-particles (Si $\mu$ Ps). After exposure to acid vapours emanating from a hot solution of hydrogen fluoride/nitric acid (HF/HNO<sub>3</sub>), scanning electron microscope images of the treated powder show the formation of a porous, sponge-like structure on the sidewalls of Si $\mu$ Ps. Fourier transmission infra-red analysis shows the appearance of hydride bonds related to the formation of the porous structure. X-ray diffraction measurements and Raman spectroscopy show the good crystallinity of the samples. The strong photoluminescence properties of the obtained porous Si $\mu$ Ps (pSi $\mu$ Ps) reveal that the vapour etching process generated silicon nanocrystals within these particles. In this work, we have investigated the catalytic activity of copper nanoparticles (CuNPs) loaded on the surface of pSi $\mu$ Ps, in order to reduce the toxic compound 4-nitrophenol to 4-aminophenol. The results show excellent catalytic performance in very short times (less than 1 min).

## 1 Introduction

Owing to its unique and interesting optical and electrical properties, porous silicon (abbreviated as pSi) is one of the most extensively studied semiconductors [1]. In the 1990s, the discovery of visible photoluminescence (PL) [2,3] from pSi at room temperature has motivated extensive studies. In recent years, a large effort has been made to study both the light emission origin and engineering technology of silicon micro-particles (abbreviated as Si $\mu$ Ps). In particular, pSi micro-particles (abbreviated as pSi $\mu$ Ps) were explored in diverse domains, such as in photovoltaic applications as a gettering layer, which traps undesirable impurities from metallurgical grade silicon by using different methods [4]. Chemically-modified pSi $\mu$ Ps have been used for sensitive and rapid humidity detection [5]. Furthermore, this porous material was used for broader applications in chemical [6] and biological sensors [7], drug delivery [8] and biomedical applications [9,10]. Moreover, many reports show that Si $\mu$ Ps extract hydrogen, bonded to pSi, for efficient hydrogen storage and production [11,12]. In addition, owing to its high theoretical capacity, it has been recognized as the most attractive and promising anode material for lithium-ion batteries, (more than

11 times theoretical capacity of graphite) [13,14]. It has also been reported by Arafa et al. [15] that polyaniline immobilized onto metallurgical pSi $\mu$ Ps is highly promising for the development of an inexpensive and flexible photovoltaic system.

These types of porous micro-particles were prepared via numerous methods, such as electrochemical etching [16,17] (the most familiar method), stain-etching in a hydrofluoric acid-based solution [18,19] and by a metal-assisted chemical etching method [20,21]. Other techniques have been used to form pSi by thermal annealing and etching of silicon gels and silica nanoparticles, at 900 °C under an argon atmosphere [22,23].

However, these methods have some critical drawbacks, such as high reaction temperatures, high cost of materials, toxic reactants and low yield, limiting their large scale application [24]. So far, an inexpensive and scalable technique to produce pSi $\mu$ Ps represents a big challenge. The morphology of this nano-sponge material is influenced by many factors, such as the doping level of the starting material, concentration of the oxidant in the mixture, etching duration and etching temperature.

Therefore, in this paper, metallurgical grade Si $\mu$ Ps were selected as the feedstock, reduces the price of the raw material by nearly ten times [25]. We have studied the formation of pSi $\mu$ Ps via a simple and low-cost method. It is one of the cheapest approaches which acts without contacting the acid solution, since only vapour-phase

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etching is applied. This study was performed for the first time by Saadoun and co-workers [26], wherein pSi was produced by exposing flat silicon wafers to HF/HNO<sub>3</sub> acid vapours. In this case, almost all procedural steps can be performed in a chemical laboratory with low-cost equipment. Recently, metal deposition onto pSi has attracted great attention in the field of catalysis. Coating of nanoparticles is an important tool that can be used to obtain different physical and chemical properties. In particular CuNPs, which are several orders of magnitude cheaper than other noble metals, were deposited on pSi and acted as a catalyst for catalytic reactions [27]. More recently, Abu Bakar et al. [28] have reported the use of pSi prepared by a stain etching method, using HF/HNO<sub>3</sub>/H<sub>2</sub>O with a ratio of (1:5:10 v/v). This pSi was used for the reduction of p-nitrophenol to p-aminophenol, which occurs within 20 min. Concerning the stability and the reusability of the catalyst, they have shown that the percentage of reduced p-nitrophenol decreased to 68% after the 4th cycle. Afterwards, they suggest the deposition of metal on pSi to enhance catalytic activity.

In this context, the surface of pSi $\mu$ P, prepared in our work using a CVE method, was loaded with CuNPs by a galvanic displacement reaction. Then, the catalytic activity of the as-prepared sample was investigated in reduction of toxic 4-nitrophenol to 4-aminophenol, using sodium borohydride as a reducing agent. Structural and morphological characterisations of the etched and unetched samples have been performed using scanning electron microscopy/energy dispersive x-ray analysis (SEM/EDX) and X-ray diffraction (XRD) measurements, Raman, Fourier transform infra-red (FTIR) and PL spectroscopy, and UV-Vis spectrometry.

## 2 Experimental procedure

### 2.1 Chemicals

All cleaning and etching reagents were of VLSI grade. Nitric acid (HNO<sub>3</sub>, 65%) was purchased from Carlo Erba and hydrofluoric acid (HF, 48%) was supplied by BASF. Unless stated otherwise, all chemicals were reagent grade or higher. Copper (II) chloride (CuCl<sub>2</sub>), acetone, ethanol (99.5%), 4-nitrophenol and sodium borohydride (NaBH<sub>4</sub>) were purchased from Sigma-Aldrich and used as-received. Purified water (18.2 M $\Omega$ cm<sup>-1</sup> resistivity) was used for all preparations and for all rinses.

### 2.2 Sample preparation

#### 2.2.1 Chemical engineering of Si $\mu$ P

Figure 1 represents the schematic illustration of the experimental chemical vapour etching setup used in this work. Indeed, commercial metallurgical-grade Si $\mu$ P were used in this work as a starting material. In the first step, 0.5 g of the dark-grey Si $\mu$ P were weighed, and ultrasonically cleaned in acetone, ethanol and ultrapure water for 10 min in sequence. The cleaned micro-particles were then dried at 60 °C overnight. In the second step, a 100 ml polypropylene container was filled with two-thirds of its volume with HF/HNO<sub>3</sub> solution. Then, the micro-particles

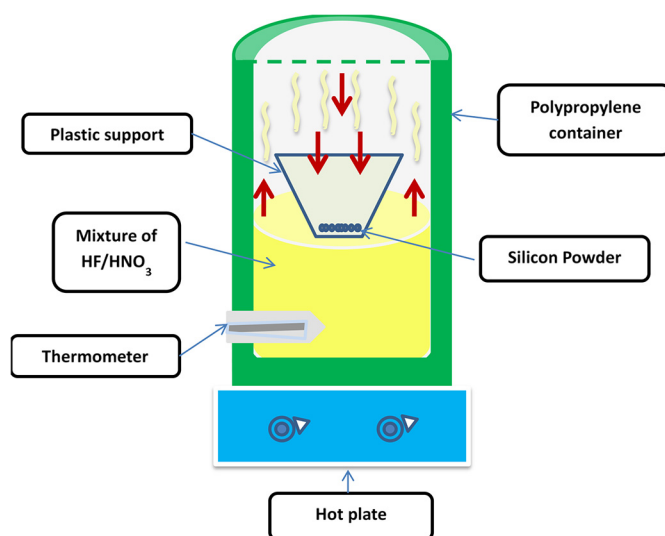


Fig. 1. Schematic illustration of the experimental CVE setup.

were dispersed in the bottom of a plastic holder. This holder was transferred into the container filled with the acid solution HF/HNO<sub>3</sub>, and placed on the top of this mixture. Finally, the container was closed and placed on a hot plate.

In this work, we used a mixture of acids containing a 48% of Hydrofluoric acid (HF) and 65% of Nitric acid (HNO<sub>3</sub>). The HF/HNO<sub>3</sub> volume ratio was fixed at 3/1, based on the previous work of Amri et al. [29].

In order to obtain more vapor inside the container, the acid mixture was heated at 80 °C. The etching duration was 30 min (sample named CVE<sub>30</sub>). Subsequently, the etched silicon micro-particles were rinsed in DI water and dried in 60 °C oven overnight.

#### 2.2.2 Deposition of copper nanoparticles on pSi $\mu$ P

CuNPs were deposited on the surface of pSi $\mu$ P using a galvanic reaction. This was performed by dipping the Si $\mu$ P into an aqueous solution of CuCl<sub>2</sub> (0.015 M)/HF (1.8 M), for 5 min at ambient conditions. Afterwards, the as-prepared samples were centrifuged and washed repeatedly with DI water. Finally, the resulting micro-particles were dried at 55 °C in an oven.

### 2.3 Catalytic reduction of 4-nitrophenol

4-nitrophenol, a model toxic substance, was prepared in DI water at 0.1 M as a stock solution. Then, NaBH<sub>4</sub> (200  $\mu$ L, 0.1 M) was mixed with freshly-prepared aqueous solution of the hazardous compound (0.1 mM). For the reduction step, the CuNPs decorated pSi $\mu$ P catalyst was added to the previous mixture. After addition, the solution was carefully mixed by shaking it gently. At the end of the reaction, the solution was taken out and filtered to remove the catalyst. Therefore, 2 cm<sup>3</sup> quartz cuvette containing the filtered solution of 4-nitrophenol was transferred into the UV-Vis spectrophotometer to measure its absorption. To monitor the reduction of this compound, the absorption band

centred at 400 and 298 nm corresponding to 4-nitrophenolate ions and 4-aminophenol, respectively was followed.

## 2.4 Sample characterisation

### 2.4.1 SEM

The morphology of Si $\mu$ P $s$  before and after acid vapour etching was characterised using an electron microscope (JEOL JSM-5400, JEOL Ltd., Akishima-shi, Japan).

### 2.4.2 FTIR spectroscopy

FTIR spectra were taken using a Nicolet MAGNA-IR 560 Spectrometer in the infrared region 500–4000 cm<sup>-1</sup>.

### 2.4.3 XRD measurements

XRD patterns were collected using a Bruker X-ray diffractometer, with Cu-K $\alpha$  radiation at room temperature. X-rays were generated at 40 kV and 30 mA, with a scan speed of 1.2 min<sup>-1</sup>. The X-ray wavelength was 0.15406 nm.

### 2.4.4 PL spectroscopy

Photoluminescence spectra were collected using a (HR250) Jobin-Yvon spectrometer at ambient temperature, with a 4765 Å laser.

### 2.4.5 Raman spectroscopy

Raman spectra were measured at room temperature using a laser excitation line of 488 nm (laser power 5 mW).

### 2.4.6 UV–Vis spectrometry

UV–vis absorption measurements were recorded using a Perkin Elmer Lambda UV–vis 950 spectrophotometer, with a 1.0 cm quartz cuvette.

## 3 Results and discussion

### 3.1 Grain size distribution of Si $\mu$ P $s$

Before any treatment, the particle size distribution of the initial micro-particles was measured using a Malvern Instruments Mastersizer 2000 (Malvern Instruments, Malvern, UK), using the laser scattering method. As represented in Figure 2, the measurements show that the size distribution of the starting Si $\mu$ P $s$  ranges from 1 to 140  $\mu$ m. It is clear that the distribution is not uniform, and can be divided into three distinct populations belonging to three areas: the first population ranges from 1 to 15  $\mu$ m, the second is between 15 and 40  $\mu$ m and the third is from 40 to 140  $\mu$ m. We note that the powder is composed of micro-particles, with most of them belonging in the third area, ranging from 40 to 140  $\mu$ m.

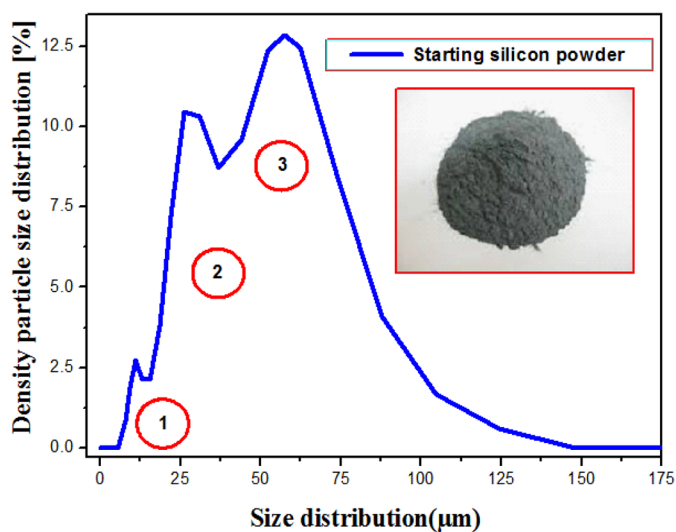


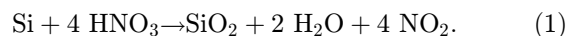
Fig. 2. Particle size distribution of Si $\mu$ P $s$  (insert shows a photograph of the starting micro-particles).

### 3.2 Morphological characterisation

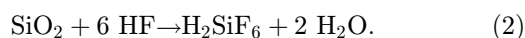
As depicted in the SEM image in Figure 3, the Si $\mu$ P $s$  have a large size distribution, confirming the previous results from the grain size distribution analysis (Sect. 3.1). The insert corresponds to the SEM image of the micro-particles at high magnification, and shows a smooth surface without pores.

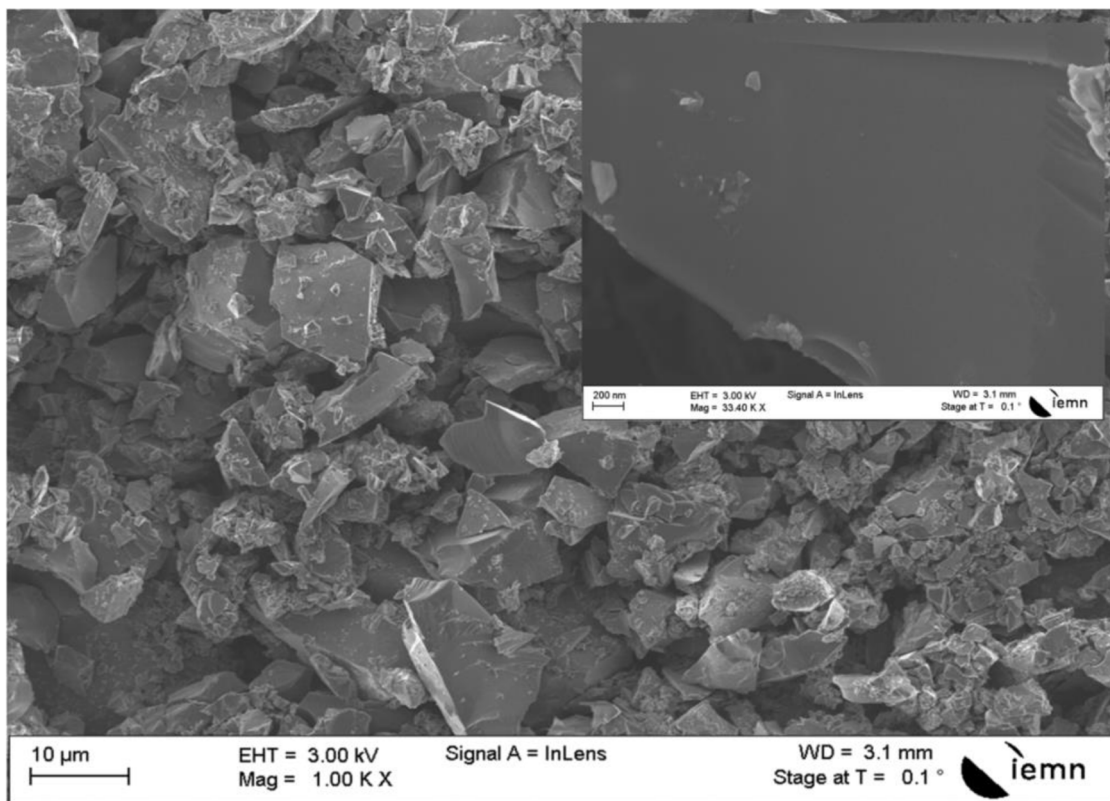
After CVE for 30 min, the formation of a porous thin layer distributed over the entire surface of the micro-particles can be clearly observed (Fig. 4). It can also be observed that CVE of the particles occurs homogeneously on all facets. The image in the insert is a magnification of the image in Figure 4, which clearly shows that the pSi $\mu$ P $s$  have a pore diameter below 50 nm. These micro-particles retain their polyhedral morphology, and contain a clear sponge-like structure. The formation of a porous layer on the micro-particle sidewalls may be rapidly confirmed by the change in colour, from dark-grey to black, and by the orange emission under UV illumination.

The chemical vapour etching method is used to generate porous structure. Indeed, the etching mechanism occurs via two steps [30]. Briefly, in the first step, the surface of silicon microparticles is oxidized after exposure to nitric acid (HNO<sub>3</sub>), which produce a thin silicon-oxide layer (SiO<sub>2</sub>) as described by the following chemical reaction:

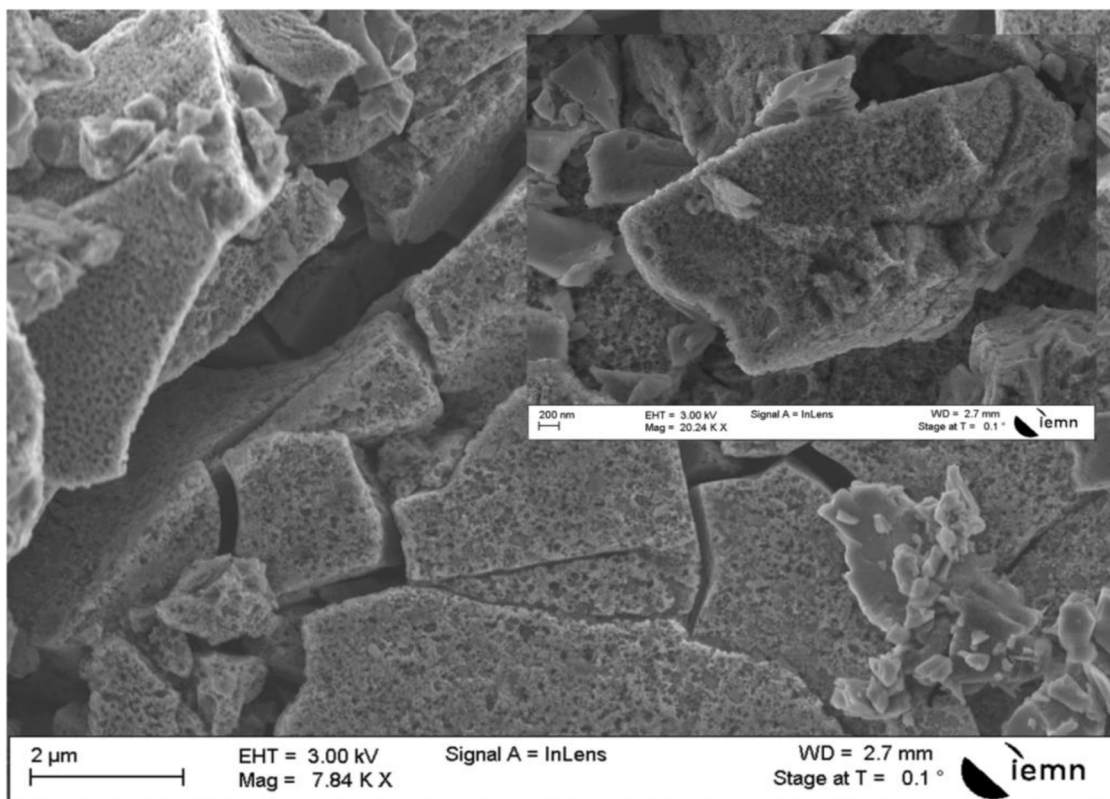


The second step is the dissolution of the formed SiO<sub>2</sub> layer by HF vapors. This reaction produces a highly water-soluble H<sub>2</sub>SiF<sub>6</sub> product, leading to a porous structure as described by the following reaction:

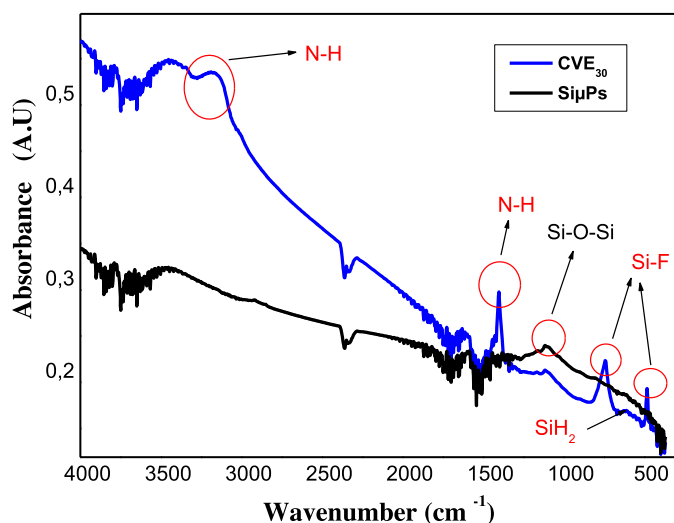




**Fig. 3.** SEM image of untreated Si $\mu$ Ps (insert image is a high-magnification view).



**Fig. 4.** SEM image of the pSi $\mu$ Ps (inserted image is a high-magnification view).



**Fig. 5.** FTIR absorption spectra of Si $\mu$ Ps before and after CVE treatment.

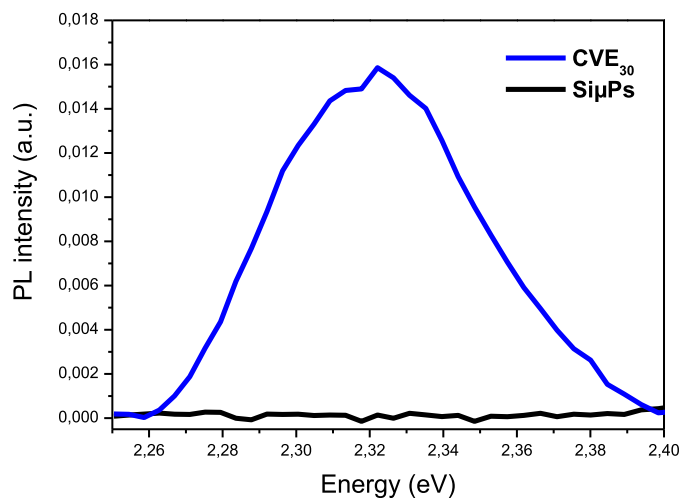
The chemical products resulting from these two reactions are confirmed by FTIR analysis as shown in the next section.

### 3.3 FTIR analysis

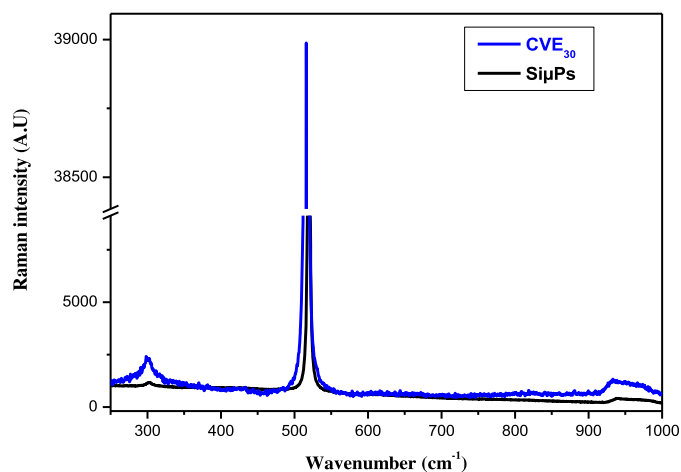
Figure 5 shows typical FTIR absorption spectra corresponding to Si $\mu$ Ps before and after CVE for 30 min. For both samples, the presence of large peak at 1090  $\text{cm}^{-1}$  is noted, which corresponds to the Si-O-Si asymmetric stretching mode [31]. After CVE treatment, a considerable surface chemical modification was observed. Noteworthy are the appearance of bands which correspond to the three main products resulting from the etching reaction between the silicon surface and the acid vapour:  $\text{N-H}_4^+$ , N-H and  $\text{SiF}_6^{2-}$  species, localised at 1410, 3170  $\text{cm}^{-1}$  and 747; 485  $\text{cm}^{-1}$ , respectively. In the low-energy region, the appearance of Si-H<sub>2</sub> in bending mode was detected, localised at 614  $\text{cm}^{-1}$ . These bonds are characteristic of the CVE method performed with HF/HNO<sub>3</sub> solution, confirming the porosification of the powder.

### 3.4 PL analysis

Figure 6 exhibits the PL spectra of untreated and treated Si $\mu$ Ps, recorded in the range of 2.25–2.40 eV at room temperature. Untreated Si $\mu$ Ps do not show any PL emission, likely due to the absence of a confined structure on their sidewalls. After CVE treatment, a significant gaussian PL peak localised at 2.32 eV is detected. L.T. Canham et al. [32] have shown in their quantum confinement model that the appearance of a PL peak is attributed to the formation of silicon nanocrystallites (SiNCs), smaller than the free exciton box size. On the other hand, FTIR analysis shows that the porous micro-particle surface is partially passivated by Si-H<sub>2</sub> and Si-O-Si bands. These two bands can create new electronic states in the band gap of the SiNCs, acting as localised states of the



**Fig. 6.** Room-temperature PL spectra of Si $\mu$ Ps before and after CVE treatment.



**Fig. 7.** Raman spectra of Si $\mu$ Ps before and after CVE treatment.

excited electron hole pairs. In this case, recombination in SiNCs proceeds through carriers trapped in oxygen-related localised states which are stabilized by the widening of the gap.

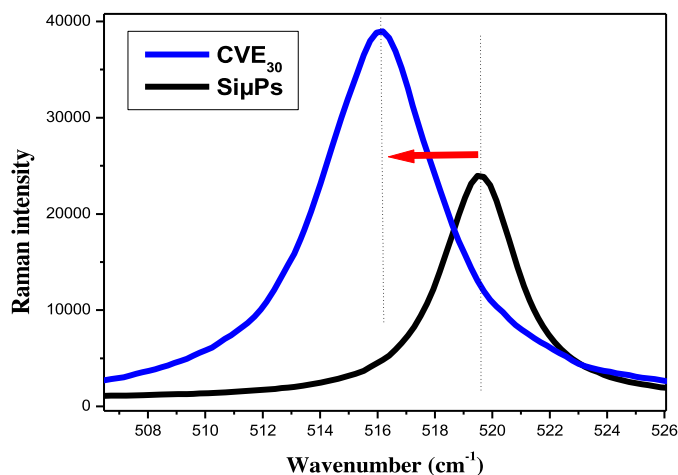
The crystallite size may be evaluated from the PL peak position of the treated Si $\mu$ Ps, according to the following equation:

$$E = E_0 + \frac{88.34}{d^{1.37}} \quad (3)$$

where  $E_0 = 1.17$  eV (band gap of bulk Si),  $E$  corresponds to the position of the peak in the PL spectrum and  $d$  is the size of the nanocrystallite [30]. Using this empirical formula, the crystallite size is 2.3 nm.

### 3.5 Raman analysis

The spectrum of Si $\mu$ Ps exposed to acid vapour for 30 min is similar to that of untreated silicon, as shown in Figure 7. Three major peaks are clearly observed in the 250–1000  $\text{cm}^{-1}$



**Fig. 8.** First-order Stokes Raman spectra of Si $\mu$ Ps before and after CVE treatment.

range. The most intense central peak corresponds to the first-order phonon mode, situated around  $519.6\text{ cm}^{-1}$ . On either side of the central peak, we observe two less intense peaks related to the second-order Raman spectra, implicating two phonons. The wide peak from  $920\text{ to }990\text{ cm}^{-1}$  originates from the second-order Raman band of crystalline Si-Si vibrations [33].

After CVE treatment, the principal peak exhibits an enhancement in Raman intensity. Compared with the untreated Si $\mu$ Ps, Raman measurements on porous micro-particles exhibit an enhancement of the first-order longitudinal optical (LO) mode peak intensities, as shown in Figure 8. This is mainly attributed to the generation of the porous layer. Raman intensity may be affected by the porosity, SiNC density or photon penetration length in the material. In addition, a shift of this mode to lower energy, from  $519.8\text{ to }516.2\text{ cm}^{-1}$ , was observed. This red shift is indicative of the phonon confinement effect, due to the formation of confined nanocrystallites on the Si $\mu$ Ps' sidewalls.

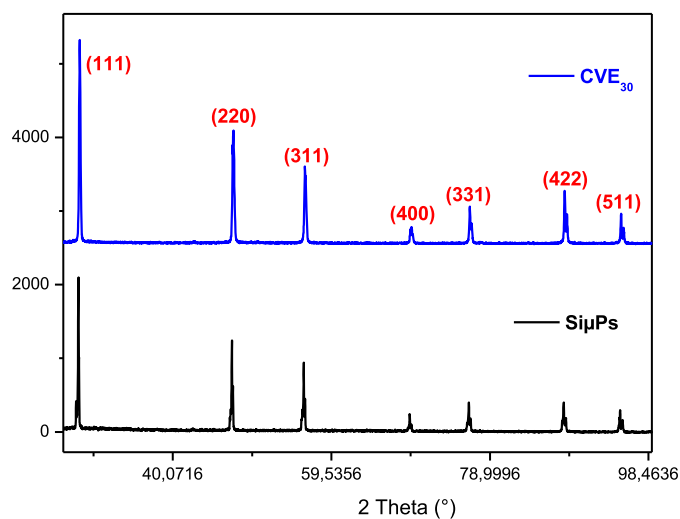
The size of SiNC can be estimated by this shift in the Raman peak, using the following analytical equation [34]:

$$\Delta w(D) = -A\left(\frac{a}{D}\right)^\gamma \quad (4)$$

where  $\Delta w(D)$  is the Raman peak shift compared to monocrystalline silicon, and  $D$  is the crystallite size.  $a$  is the lattice constant of silicon ( $0.543\text{ nm}$ ),  $A = 47.41\text{ cm}^{-1}$  and  $\gamma = 1.44$ , all of which are fitting parameters that explain the phonon confinement in nanometric spheres. The crystallite size estimated from this method was  $3.25\text{ nm}$ .

### 3.6 XRD measurements

XRD was used to inspect the crystallographic structure of the treated and untreated Si $\mu$ Ps. As depicted in Figure 9, the samples show similar and well-defined peaks, indicating



**Fig. 9.** XRD patterns of treated and untreated Si $\mu$ Ps.

the good crystallinity of the material. It is noted that the micro-particles are polycrystalline, and presented facets with different crystallographic orientations [35]. The main surface facets present in the micro-particles were (111), (200), (220) and (311) planes. From X-ray analysis, it can clearly be seen that the crystallites have preferential orientation along the (111) plane. This strong peak at  $2\theta = 28.65^\circ$  indicates the higher population of silicon (111) [25]. This peak is broader than the same peak for untreated silicon, confirming the formation of pores on the silicon surface. Therefore, this slight broadening may be due to the formation of SiNCs after the etching process, as supported by the results of PL and Raman analysis.

### 3.7 Reduction of 4-nitrophenol to 4-aminophenol

Firstly, copper nanoparticles were loaded on pSi $\mu$ Ps. As can be seen in Figure 10, the CuNPs were successfully deposited homogeneously on all facets of the porous micro-particles. The nanoparticles are deposited uniformly and are not agglomerated.

The EDX spectrum in Figure 11 showed the presence of Si, Cu, O and F. This result is consistent with the presence of copper on the surface of pSi $\mu$ Ps. To further confirm deposition of the CuNPs, EDX mapping analysis was performed (Fig. 12). The mapping image of Si-K, Cu-L and O-K was well-defined with sharp contrast, confirming that the CuNPs have been successfully deposited on the surface of the micro-particles.

We investigated the catalytic activity of the prepared CuNPs/pSi $\mu$ Ps toward the reduction of 4-nitrophenol to 4-aminophenol, in the presence of  $\text{NaBH}_4$ . The catalytic reduction was followed by UV-Vis spectroscopy. After addition of  $\text{NaBH}_4$  to 4-nitrophenol solution, it was observed that the absorption peak is shifted to  $400\text{ nm}$ . Moreover, the color changes from light yellow to yellow-green. This displacement and this color change are due to the formation of nitrophenolate ion [36], while

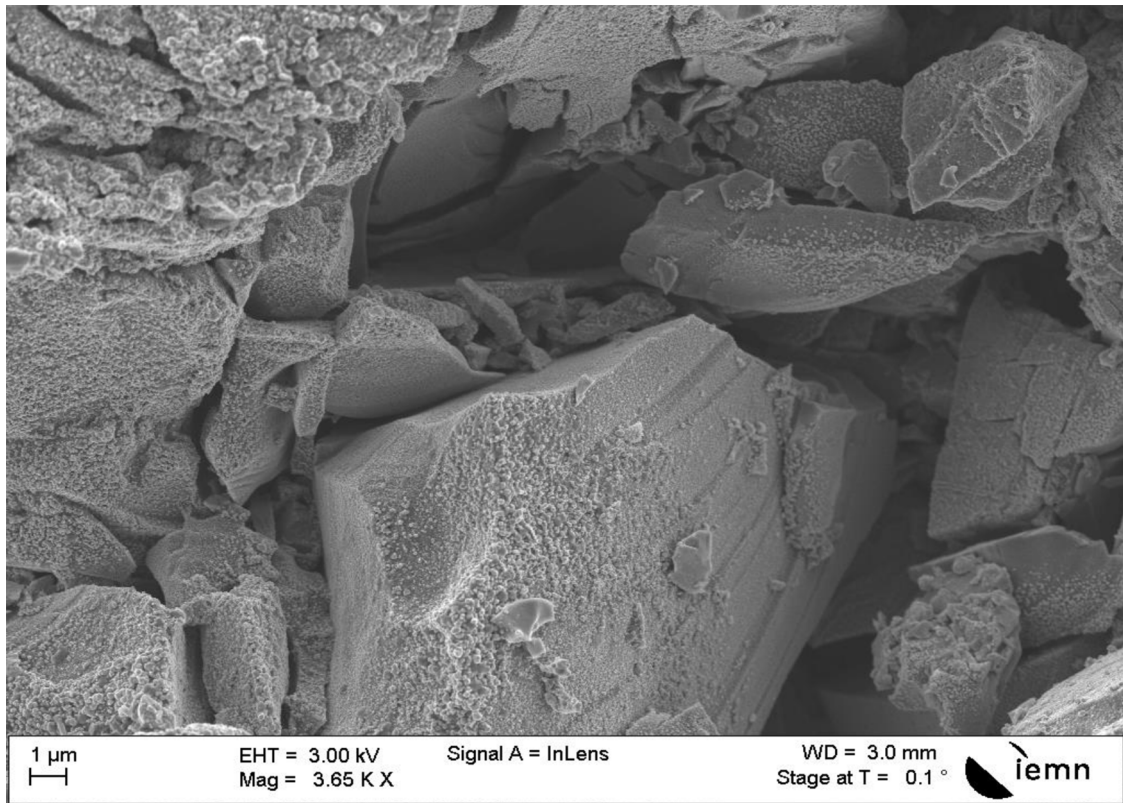


Fig. 10. SEM image of CuNPs-decorated pSiμPs.

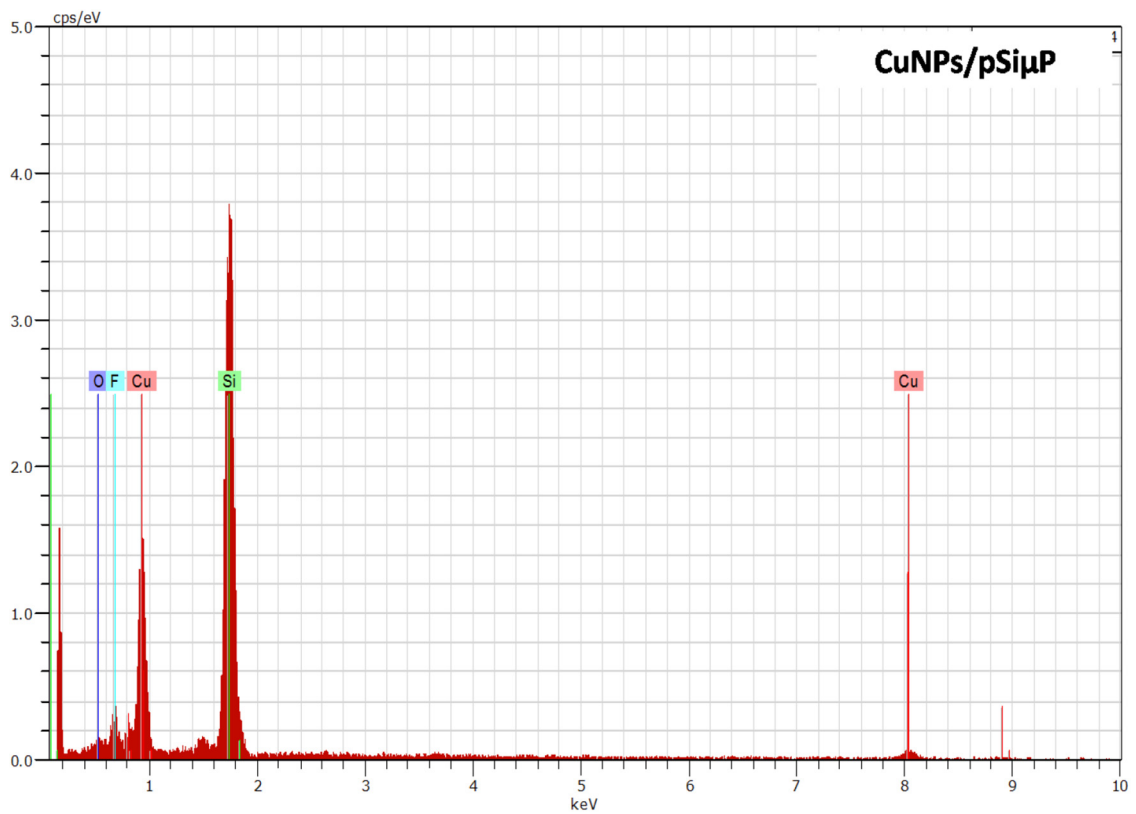
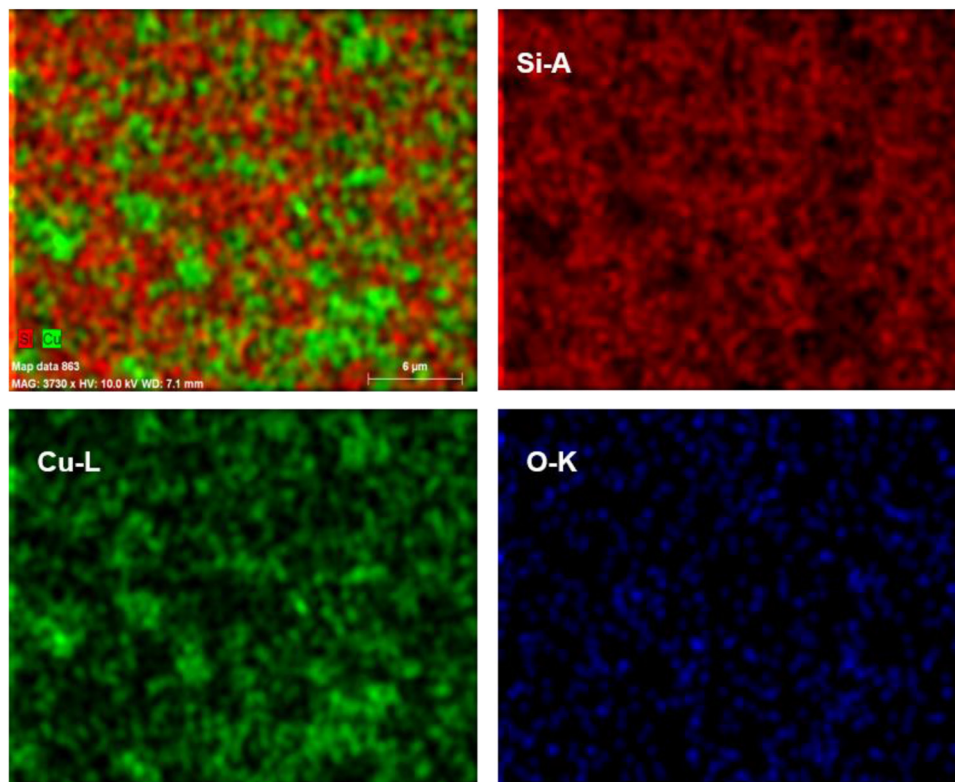


Fig. 11. EDX spectrum of CuNPs-decorated pSiμPs.





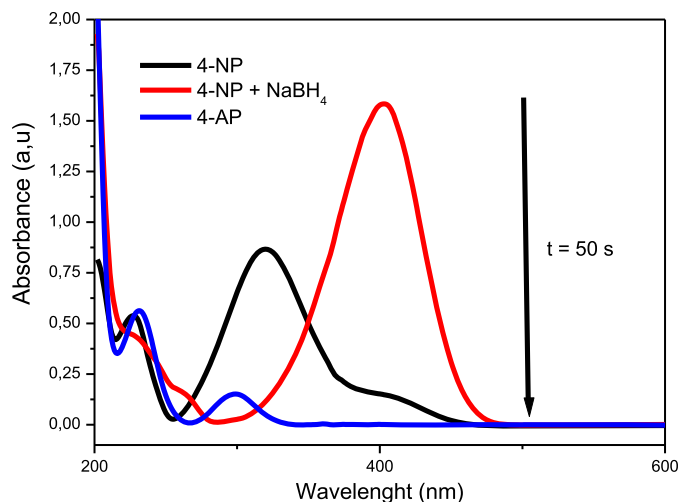
**Fig. 12.** EDX mapping of CuNP-decorated pSi $\mu$ Ps.

4-nitrophenol absorbs at 315 nm (Fig. 13). After addition of CuNPs/pSi $\mu$ P catalyst to the solution, the color changes from yellow-green to colourless. The peak at 400 nm disappears within 50 s, with the appearance of a new peak at 298 nm due to the formation of 4-aminophenol.

In the literature, the catalytic reduction of this nitroaromatic compound can be described by general mechanism of heterogeneous catalysis [37,38]. In this work, this reduction occurs on the copper surface. Therefore, a plausible schematic for the reaction mechanism is represented in Figure 14. Adding NaBH<sub>4</sub> to 4-nitrophenol increases the pH of the solution, resulting in the deprotonation of nitrophenols to nitrophenolate ions. The first step of this process is the diffusion and adsorption of hydride species (BH<sub>4</sub><sup>-</sup>) and 4-nitrophenol on the catalyst surface via nitro group (-O-N=O). In the second step, the electron transfer from BH<sub>4</sub><sup>-</sup> (donor) to 4-nitrophenol (acceptor) occurred which allow the desorption of 4-aminophenol from the surface.

Using the same sample, 5 successive cycles of the catalytic reduction of 4-nitrophenol were performed, to evaluate the stability and reusability of the CuNPs/pSi $\mu$ Ps catalyst (Fig. 15).

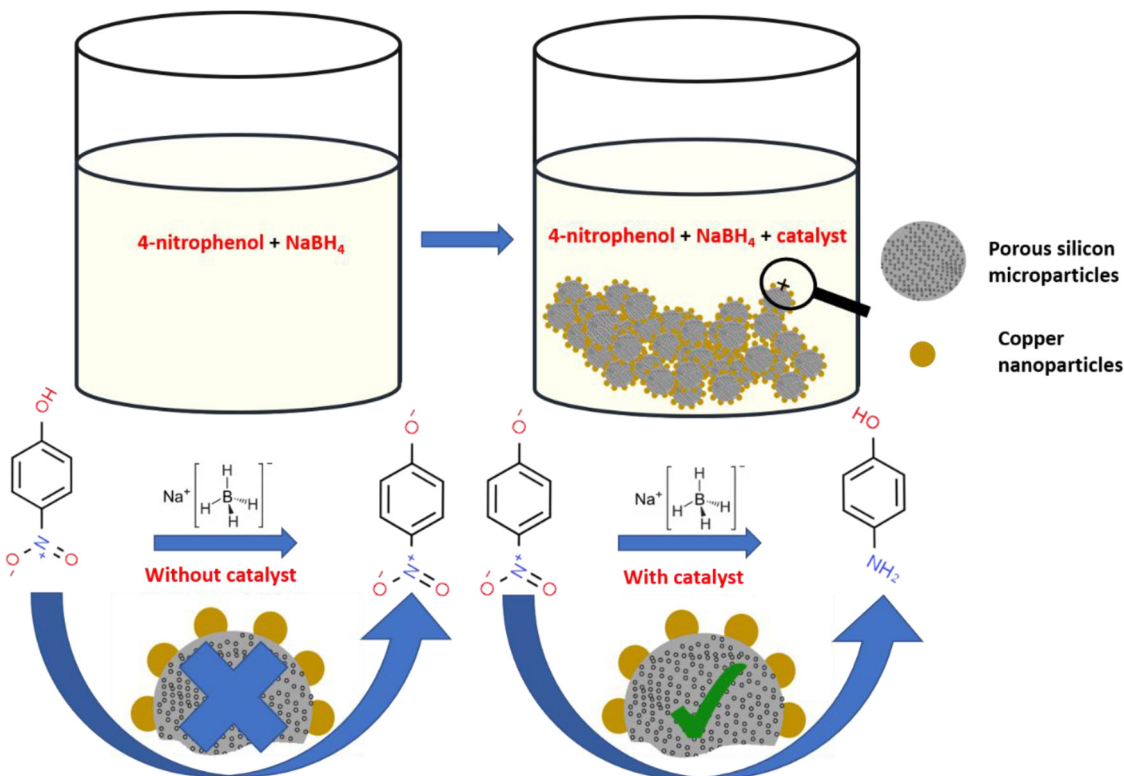
It can be clearly seen that, after each cycle, the repeated tests did not affect the reduction activity of the catalyst; it continues to have high catalytic performance. This experiment confirms the stability and the reusability of the CuNPs/pSi $\mu$ Ps catalyst, during the catalytic reduction of 4-nitrophenol.



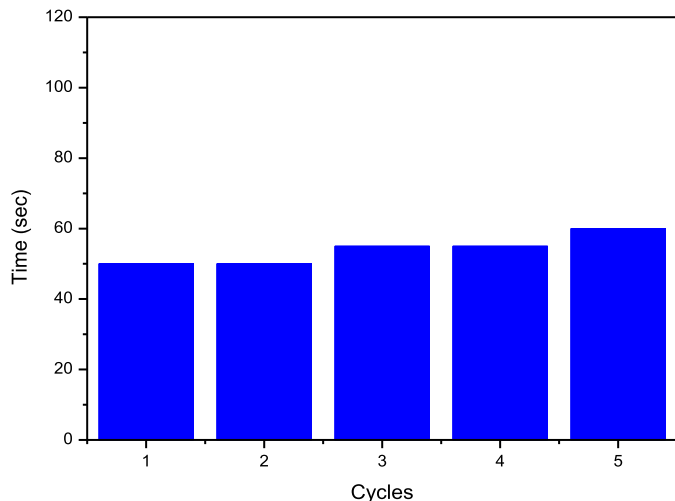
**Fig. 13.** UV-Vis absorption spectra during the reduction of 4-nitrophenol by NaBH<sub>4</sub> catalysed by CuNPs/pSi $\mu$ Ps, over a time period of 50 s.

## 4 Conclusion

In summary, metallurgical grade pSi $\mu$ Ps were synthesised via CVE technique, using a mix of HF/HNO<sub>3</sub>. SEM images showed that the surface of the micro-particles after chemical etching is porous, compared to the unetched



**Fig. 14.** The mechanism of reduction of 4-nitrophenol by  $\text{NaBH}_4$  in the presence of CuNPs/pSi $\mu$ P catalyst.



**Fig. 15.** Stability of the CuNPs/pSi $\mu$ P catalyst after 5 cycles.

surface, and the pSi $\mu$ P have a nano-sponge-like structure. XRD was selected to study the changes in the crystal structure caused by the CVE processes, and showed that Si $\mu$ P were polycrystalline, with one facet having several crystallographic orientations. Raman spectroscopy confirmed that porous micro-particles still retain crystallinity. FTIR spectroscopy analysis illustrated that the pSi $\mu$ P contained  $\text{SiF}_6^{2-}$  and  $\text{NH}_4^+$  ions. We conclude that the CVE process does not affect the material. Finally, it

was found that the as-prepared CuNPs/pSi $\mu$ P sample exhibited excellent catalytic performance in reducing 4-nitrophenol to 4-aminophenol within 50 s. We believe that this material may be an effective and stable catalyst for the reduction of a toxic pollutant.

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### Author contribution statement

All the authors have participated in the preparation of the manuscript. They have read and approved the final manuscript.

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