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Gate Tunable Vertical Geometry Phototransistor based on Infrared HgTe Nanocrystals

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Abstract: Infrared nanocrystals are promising building blocks for the design of low-cost infrared sensors. Vertical geometry diode is, among possible geometries, the one that lead to the best performance so far. However, this geometry suffers from a lack of tunability after its fabrication, slowing down possible improvements. Here, we demonstrate gate control on a vertical diode in which the active layer is made of HgTe NCs absorbing in the extended short-wave infrared (2.5 μm). To reach this goal, we take advantage of the electrostatic transparency of graphene, combined with the high capacitance LaF₃ ionic glass to design a gate tunable photodiode. The latter behaves as a work-function-tunable electrode which lets the gate-induced electric field tune the carrier density in the nanocrystal film. In particular, we show that the gate allows to tune the band profile leading to more efficient charge extraction and thus an enhanced-photoresponse (x4 compared to the device with floating gate). This work also demonstrates that photoelectron extraction can still be improved in the existing diode, by better controlling the doping profile of the diode.

Keywords: HgTe nanocrystal, infrared, photodiode, gate effect, mixed dimensionalities device.

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HgTe nanocrystals (NCs) combining the semimetal nature of the bulk^{1,2} and quantum confinement³⁻⁵, presents a unique spectral tunability in the infrared range.⁶ Beyond material absorption, HgTe NCs have been used for their IR light emitting properties⁷⁻¹² as well as for their short^{13,14} and mid wave^{15,16} IR photoconduction properties.¹⁷ This has led to the demonstration of focal plane arrays operating in the short-^{18,19} and mid-wave infrared.²⁰

Among possible device geometries, the vertical diode is the one that leads to the highest performance so far. The in-built electric field allows zero bias operation reducing the dark current as well as the noise. However, the number of structures that have been proposed to design a vertical geometry diode using HgTe NCs remains quite limited,²¹⁻²⁴ and does not seem fully optimized. In particular, for an optical band gap in the 500 to 750 meV range, the open circuit voltage is only a few tens of millivolts. This is far smaller than half of the band gap value commonly achieved for PbS quantum dots. Even though a part of this loss may directly be related to the traps in the material,^{25,26} it also suggests that the currently proposed diode stack remains sub-optimal.

The design of such diode is nevertheless challenging. Part of the open circuit voltage loss can directly come from the poor doping profile within the diode. Controlling the intrinsic doping in the NC film remains a difficult task and thus it will be of utmost interest to control the film carrier density using electrostatic gates. Field effect transistors based on HgTe films have been reported with various gate technologies such as dielectrics,²⁷⁻²⁹ electrolytes³⁰ and even ionic glasses.³¹ However, all are based on planar geometry. Implementation of a gate in a vertical geometry appears far more challenging and remains unreported for HgTe NC based diodes.

Our goal is to develop electrodes for the design of infrared NC-based diodes with the dual role of collecting carriers on one hand and controlling the carrier density within the film on the other hand. The development of such electrode faces a second challenge relating to its transparency. In most NC-based diodes, transparent conductive oxides (TCOs) are used. However, in the infrared TCOs present a strong absorption, requiring a decrease in their thicknesses at the cost of an increased contact resistance. Thus, alternative materials should be preferred. Here we explore graphene coupled to LaF₃ ionic glass used as substrate and infrared transparent³² bottom electrode which can also be used to control the carrier density of the NC film. Compared to the works from Yao and coworkers,^{33,34} the switch from a conventional dielectric to the high capacitance gate allows us to expand the design of the doping landscape towards higher carrier density.

Coupling infrared NCs^{35,36} including HgTe^{37,38} to graphene and 2D materials³⁹ is an approach that stimulated a growing activity over the past few years. So far, the most investigated geometry relies on a planar configuration^{13,40} where the graphene is used as a channel of a field effect transistor which is light sensitized by the NCs.⁴¹ The large mobility of graphene is used to generate gain and achieve very large responsivities (i.e. light to current conversion). However, the lack of band gap in graphene comes with a large dark conductance partly balancing the gain benefit. Here, we go further while exploring the use of the graphene large quantum capacitance. Contrary to metal or TCO electrodes, graphene does not screen the electric field induced by the LaF₃ ionic glass. As a result, the gate below the 2D material can be used both to control the work function of the electrodes^{13,42,43} and to tune the carrier density within the NC film over its surface.

In this work, we demonstrate that exploiting the electrostatic transparency across mixed-dimensional hetero-interfaces allows to use the back gate as a knob to control the current flow within the diode. The gate bias enables us to modify the band profile in the NC junction, that can now be switched from *n-p* configuration to a *n⁺-n-p* configuration which is more favorable for charge extraction. This work confirms that HgTe NC-based diode performance can be further improved by optimizing the carrier density profile, in particular, on the *n*-side, as well as pinpoints this issue as a key element that future work will need to address.

We start by growing HgTe NCs absorbing in the extended short wave infrared using Keuleyan's procedure.⁴⁴ The particles (later called HgTe 4k) present a clear excitonic feature at 4000 cm^{-1} ($\approx 0.5\text{ eV}$ or $2.5\text{ }\mu\text{m}$), see **Figure 1a**. The nanocrystals have a planar tripod shape according to transmission electron microscopy (see **Figure 1b**). The nature of the majority carriers is also a central parameter for the following material integration and this can be probed using field effect transistor measurements. The latter is designed from electrodes onto a LaF_3 substrate which is an ionic glass.⁴⁵ LaF_3 provides a high capacitance³¹ ($0.4\text{ }\mu\text{F}\cdot\text{cm}^{-2}$) while being all solid-state (i.e. without the constraints of electrolyte). Under such condition, the channel of HgTe NCs shows an ambipolar behavior (**Figure 1c**) with both hole and electron conduction. The minimum of conductance occurs for a small negative gate bias ($V_{\text{GS}}=-2\text{ V}$) meaning that there is a small inherent n doping in this material. This picture is consistent with photoemission measurements on this material, which demonstrate that the Fermi level is close to the middle of the band gap and lies slightly closer to the conduction band.²²

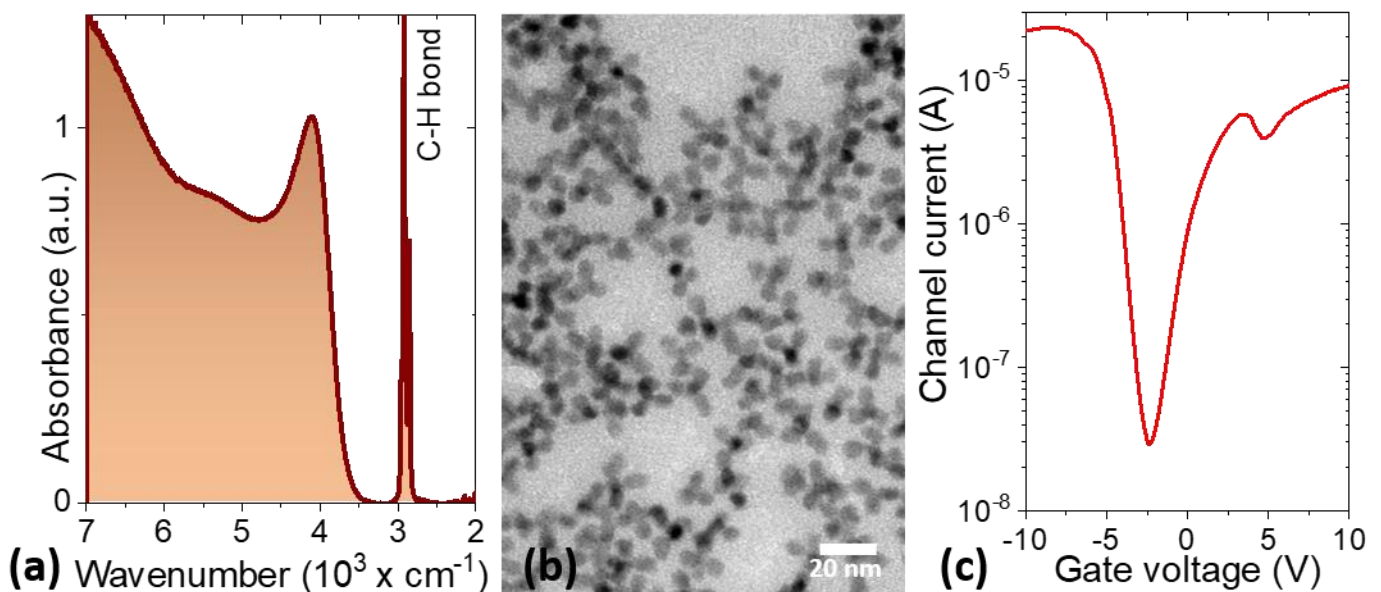


Figure 1 HgTe infrared nanocrystals. *a. Absorption spectrum of the HgTe NCs. b. Transmission electronic microscopy image of HgTe NCs associated with the spectrum of part a. c. Transfer curve (drain current as a function of the gate bias) for a thin film of HgTe NCs used as the channel of a field-effect transistor which gate is made of LaF_3 ionic glass.*

The bottom electrode of the diode made of graphene was also characterized independently (*ie* without the capping HgTe NC layer). The graphene is a CVD-grown graphene which has been transferred onto the LaF_3 substrate. As expected for graphene, we also observe an ambipolar behavior (**Figure 2a**) for this material under gate bias application, with a minimum of conductance corresponding to the Dirac point observed for $V_{\text{GS}} = -4\text{ V}$. The graphene is then patterned as stripes, all connected to a gold contact, see **Figure 2b**. Using Raman mapping of the 2D mode of graphene (**Figure 2c**), we check that stripes are well defined. The processing of graphene electrode under stripes allows simultaneous charge collection, while leaving part of the film in direct contact with the gate to further enhance the carrier density tunability within the NC film.

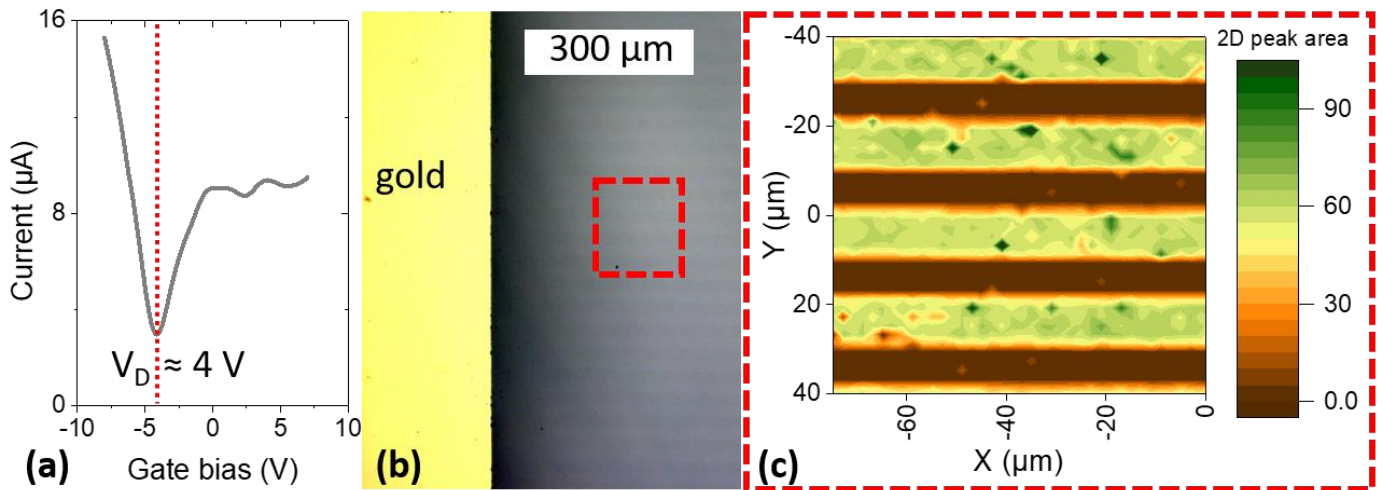


Figure 2 Graphene as infrared transparent, work function tunable electrode. *a.* Transfer curve (drain current as a function of the gate bias) for a graphene layer used as a channel of a field-effect transistor-which gate is made of LaF_3 ionic glass. *b.* Image of the patterned graphene deposited onto a LaF_3 substrate and connected to a gold contact. *c.* Map of the intensity of the graphene in 2D Raman mode of graphene for the electrode deposited on LaF_3 . On this map the substrate is brown while the graphene is yellow green

We then design the diode according to the scheme of **Figure 3a**. A thin (35 nm) ITO layer is deposited on one side of a double side polished LaF_3 substrate. The latter will be used as IR transparent gate electrode. The LaF_3 substrate provides two functionalities: an IR transparent bottom substrate and a high capacitance gate medium. In LaF_3 , the gating results from F^- vacancies displacement, optimal operating condition is in the 200-250 K range (220 K is used as operating in the following of the paper). Above 250 K the leakage in the gate are high and compete with the channel flow, while below 180 K, the vacancies freeze and the gate tunability is lost. On the other side of the substrate, the CVD-grown graphene is transferred and patterned as stripes, see **Figure 2b** and **c**. The absorbing layer is made of a 300 nm film of HgTe 4k (*i.e.* presenting a band gap at $2.5 \mu\text{m}$). On top of the HgTe 4k film, a thin layer of HgTe 7k (*i.e.* with a band-edge at 7000 cm^{-1} or 0.87 eV) is deposited. Finally, a top gold contact is used as a hole extractor.

In HgTe, the strong asymmetry of effective mass between the valence and the conduction bands makes that most of the quantum confinement occurs in the conduction band. As the result of the heavy mass of the hole, the valence band is poorly affected by the confinement (*i.e.* the valence band weakly shifted when confinement is introduced) and it makes that various sizes of HgTe NCs present quasi resonant valence band.²² The layer of HgTe 7k NCs with a larger band gap (7000 cm^{-1} or 0.87 eV) on top of HgTe 4k is used as a unipolar barrier²² preventing electron diffusion towards the top electrode, while leaving the photo-holes flow.

To confirm the effect of the gate, we measure the current flow through the diode while the gate bias is tuned (the equivalent of a transfer curve in a field-effect transistor), see **Figure 3b**. Under a gate bias application, we observe a clear modulation of the current, typically by one order of magnitude. This confirms that the gate electrode can be used to tune the carrier density in the vicinity of the graphene electrode.

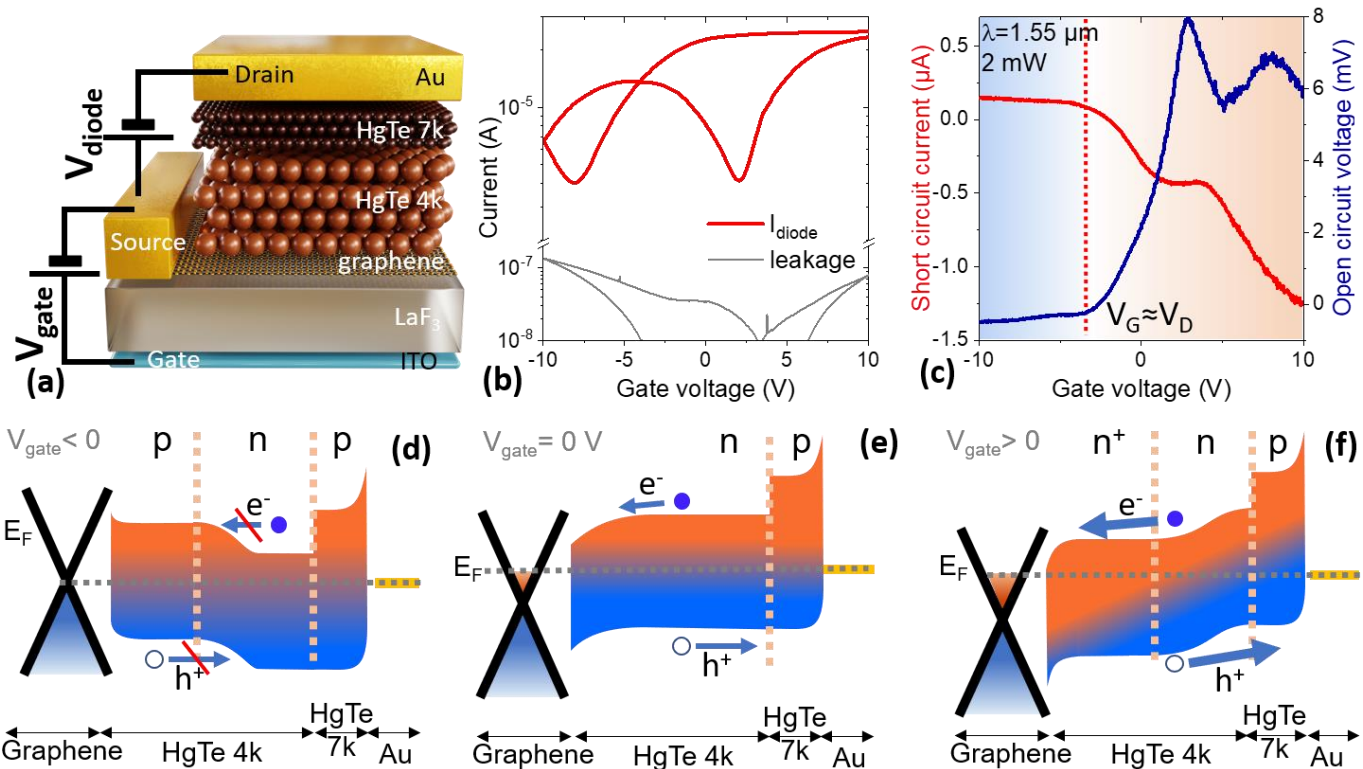


Figure 3 Gate tunable vertical geometry infrared diode. *a.* Sketch of the gate tunable diode based on the ITO/LaF₃/graphene/HgTe4k/HgTe7k/gold stack. *b.* Current through the diode while the gate bias is tuned ($V_{diode}=0.1$ V). *c.* Short circuit current and open circuit voltage for the diode under illumination ($\lambda=1.55 \mu\text{m}$ – $P=2$ mW) as a function of gate bias. Band structure of the diode under negative gate bias (part *d.*), 0 V bias (part *e.*) and positive gate bias (part *f.*). Measurements are conducted at 220 K.

The band profile associated with this diode structure at 0 V is given in **Figure 3e**. The slightly smaller work function of graphene (4.4-4.5 eV) compared to HgTe 4k^{19,22} (4.6-4.7 eV) induces a downward band bending for HgTe 4k at its interface with graphene, while the trend is opposite at the HgTe7k/gold interface. Under positive gate bias application, the *n*-type nature of the HgTe 4k NC thin film is reinforced creating a *n⁺np* band profile which is expected to ease hole and electron extraction, see **Figure 3f**. On the other hand, when a negative gate bias is applied, holes are injected and a *pn*p band profile is generated, see **Figure 3d**. In this case a very poor photo-charge extraction is anticipated.

These predictions are experimentally confirmed by measuring the photocurrent spectrum, see **Figure 4a**. We clearly observe that the gate bias drives the magnitude of the photocurrent while the bias over the diode is kept constant. No photoresponse is observed while the diode is operated in the *pn*p configuration. We also clearly observe an enhancement by a factor ≈ 4 of the photocurrent magnitude as positive gate bias is applied.

The gate effect is quantitatively better monitored by measuring the short circuit current (I_{sc}) and the open circuit voltage (V_{oc}), see **Figure 3c** and IV curves in Figure S5. The *pn*p regime ($V_G < 0$) is characterized by very low performance: $V_{oc} < 1$ mV and weak photocurrent corresponding to responsivity of the order of $50 \mu\text{A}\cdot\text{W}^{-1}$ @ $1.55 \mu\text{m}$, which is weak compared to best diodes based on graphene/HgTe NCs.³² As the gate bias is tuned, we observe a change of sign for the V_{oc} and I_{sc} . This confirms the change of nature for the majority carriers from holes to electrons as gate bias is increased. The gate bias where the open circuit change of sign is -4 V matching well with the Dirac point of graphene. For higher gate biases (*n⁺np* regime), the open circuit voltage reaches 6 mV, while the current is 10 times higher ($R \approx 0.6 \text{ mA}\cdot\text{W}^{-1}$ @ $1.55 \mu\text{m}$) than for negative gate biases, see

Figure 3c. Time response of the diode has been measured and found to be fast (<100 ms, setup limited), see Figure 4b.

Our results demonstrate that increasing the doping level of the NC absorbing layer enables an improvement of the diode performance. It also confirms that in current NC diodes the electron extraction is not as favorable as it could be if more strongly n-doped HgTe NCs might be synthesized while keeping the band gap constant.

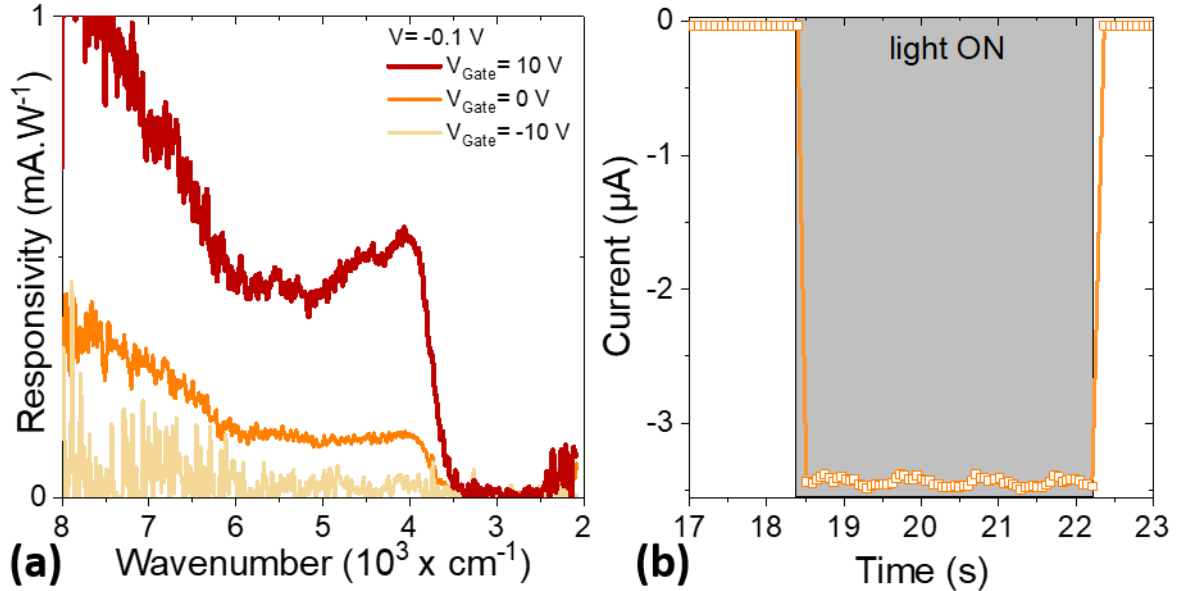


Figure 4 Infrared detection using the gate tunable diode. a. Spectral response of the gate tunable diode operated under three gate biases, while the diode bias is set to -0.1 V. Curve have been normalized to take into account the amplifier gain. b. Photocurrent under 0 V diode bias and 10 V gate bias while is the illumination ($\lambda=1.55 \mu\text{m} - P=2 \text{ mW}$) is turned on and off. Measurements are conducted at 220K

To summarize, gate control is commonly integrated in planar geometries for field effect transistors, but the use of gates in HgTe NC-based photodiodes remains unexplored. Here, we demonstrate the fabrication of a gate tunable photodiode architecture based on HgTe NCs and a mixed-dimensional vertical stack. The combination of high capacitance LaF_3 ionic glass and graphene is used as an infrared transparent electrode that allows us to adjust electrostatically the band profile of the NCs absorbing layer after fabrication. We demonstrate that in the floating condition (*i.e.* without gate), the band profile does not lead to the optimal condition for charge extraction. This work suggests that additional research will be needed to design better *n*-doped HgTe NCs to obtain optimized photoelectron extraction. Moreover, as demonstrated here on a first device generation, 2D-0D vertical heterostructures reveal a promising system to introduce electrostatic engineering of the diode absorbing layer.

See the supplementary material for information about nanocrystal growth, material characterization, device fabrication and device characterization.

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The authors declare no competing interests.

The data that supports the findings of this study are available within the article and its supplementary material.

References

- ¹ A. Delin and T. Klüner, *Phys. Rev. B* **66**, 035117 (2002).
- ² A. Svane, N.E. Christensen, M. Cardona, A.N. Chantis, M. van Schilfgaarde, and T. Kotani, *Phys. Rev. B* **84**, 205205 (2011).
- ³ E. Lhuillier, S. Keuleyan, and P. Guyot-Sionnest, *Nanotechnology* **23**, 175705 (2012).
- ⁴ N. Moghaddam, C. Gréboval, J. Qu, A. Chu, C. Livache, S.-S. Chee, A. Khalili, X.Z. Xu, B. Baptiste, S. Klotz, G. Fishman, F. Capitani, S. Ithurria, S. Sauvage, and E. Lhuillier, *J. Phys Chem C* **123**, 24360 (2020).
- ⁵ G. Allan and C. Delerue, *Phys. Rev. B - Condens. Matter Mater. Phys.* **86**, 165437 (2012).
- ⁶ N. Goubet, A. Jagtap, C. Livache, B. Martinez, H. Portalès, X.Z. Xu, R.P.S.M. Lobo, B. Dubertret, and E. Lhuillier, *J. Am. Chem. Soc.* **140**, 5033 (2018).
- ⁷ A. Rogach, S.V. Kershaw, M. Burt, M.T. Harrison, A. Kornowski, A. Eychmüller, and H. Weller, *Adv. Mater.* **11**, 552 (1999).
- ⁸ A.A. Sergeev, D.V. Pavlov, A.A. Kuchmizhak, M.V. Lapine, W.K. Yiu, Y. Dong, N. Ke, S. Juodkazis, N. Zhao, S.V. Kershaw, and A.L. Rogach, *Light Sci. Appl.* **9**, 16 (2020).
- ⁹ P. Geiregat, A.J. Houtepen, L.K. Sagar, I. Infante, F. Zapata, V. Grigel, G. Allan, C. Delerue, D. Van Thourhout, and Z. Hens, *Nat. Mater.* **17**, 35 (2018).
- ¹⁰ D.S. Koktysh, N. Gaponik, M. Reufer, J. Crewett, U. Scherf, A. Eychmüller, J.M. Lupton, A.L. Rogach, and J. Feldmann, *ChemPhysChem* **5**, 1435 (2004).
- ¹¹ J. Qu, P. Rastogi, C. Gréboval, D. Lagarde, A. Chu, C. Dabard, A. Khalili, H. Cruguel, C. Robert, X.Z. Xu, S. Ithurria, M.G. Silly, S. Ferré, X. Marie, and E. Lhuillier, *Nano Lett.* **20**, 6185 (2020).
- ¹² É. O'Connor, A. O'Riordan, H. Doyle, S. Moynihan, A. Cuddihy, and G. Redmond, *Appl. Phys. Lett.* **86**, 201114 (2005).
- ¹³ U.N. Noubé, C. Gréboval, C. Livache, A. Chu, H. Majjad, L.E. Parra López, L.D.N. Mouafo, B. Doudin, S. Berciaud, J. Chaste, A. Ouerghi, E. Lhuillier, and J.-F. Dayen, *ACS Nano* **14**, 4567 (2020).
- ¹⁴ Y. Dong, M. Chen, W.K. Yiu, Q. Zhu, G. Zhou, S.V. Kershaw, N. Ke, C.P. Wong, A.L. Rogach, and N. Zhao, *Adv. Sci.* **7**, 2000068 (2020).
- ¹⁵ M.E. Cryer and J.E. Halpert, *ACS Photonics* **5**, 3009 (2018).
- ¹⁶ S. Keuleyan, E. Lhuillier, V. Brajuskovic, and P. Guyot-Sionnest, *Nat. Photonics* **5**, 489 (2011).
- ¹⁷ E. Lhuillier and P. Guyot-Sionnest, *IEEE J. Sel. Top. Quantum Electron.* **23**, 6000208 (2017).
- ¹⁸ C. Gréboval, S. Ferre, V. Noguier, A. Chu, J. Qu, S.-S. Chee, G. Vincent, and E. Lhuillier, *ArXiv200111554 Cond-Mat Physicsphysics* (2020). <https://arxiv.org/abs/2001.11554>
- ¹⁹ A. Chu, B. Martinez, S. Ferré, V. Noguier, C. Gréboval, C. Livache, J. Qu, Y. Prado, N. Casaretto, N. Goubet, H. Cruguel, L. Dudy, M.G. Silly, G. Vincent, and E. Lhuillier, *ACS Appl. Mater. Interfaces* **11**, 33116 (2019).
- ²⁰ C. Buurma, R.E. Pimpinella, A.J. Ciani, J.S. Feldman, C.H. Grein, and P. Guyot-Sionnest, in *Opt. Sens. Imaging Photon Count. Nanostructured Devices Appl.* 993303 (2016).
- ²¹ M.M. Ackerman, X. Tang, and P. Guyot-Sionnest, *ACS Nano* **12**, 7264 (2018).
- ²² A.M. Jagtap, B. Martinez, N. Goubet, A. Chu, C. Livache, C. Gréboval, J. Ramade, D. Amelot, P. Trousset, A. Triboulin, S. Ithurria, M.G. Silly, B. Dubertret, E. Lhuillier, *ACS Photonics* **5**, 4569 (2018).

- ²³ B. Martinez, J. Ramade, C. Livache, N. Goubet, A. Chu, C. Gréboval, J. Qu, W.L. Watkins, L. Becerra, E. Dandeu, J.L. Fave, C. Méthivier, E. Lacaze, and E. Lhuillier, *Adv. Opt. Mater.* **7**, 1900348 (2019).
- ²⁴ M.M. Ackerman, M. Chen, and P. Guyot-Sionnest, *Appl. Phys. Lett.* **116**, 083502 (2020).
- ²⁵ C.-H.M. Chuang, A. Maurano, R.E. Brandt, G.W. Hwang, J. Jean, T. Buonassisi, V. Bulović, and M.G. Bawendi, *Nano Lett.* **15**, 3286 (2015).
- ²⁶ S. Pradhan, M. Dalmases, and G. Konstantatos, *J. Phys. Chem. Lett.* **10**, 3029 (2019).
- ²⁷ E. Lhuillier, S. Keuleyan, P. Zolotavin, and P. Guyot-Sionnest, *Adv. Mater.* **25**, 137 (2013).
- ²⁸ X. Lan, M. Chen, M.H. Hudson, V. Kamysbayev, Y. Wang, P. Guyot-Sionnest, and D.V. Talapin, *Nat. Mater.* **19**, 323 (2020).
- ²⁹ J. Jang, K. Cho, S.H. Lee, and S. Kim, *Nanotechnology* **19**, 015204 (2007).
- ³⁰ C. Livache, E. Izquierdo, B. Martinez, M. Dufour, D. Pierucci, S. Keuleyan, H. Cruguel, L. Becerra, J.L. Fave, H. Aubin, A. Ouerghi, E. Lacaze, M.G. Silly, B. Dubertret, S. Ithurria, and E. Lhuillier, *Nano Lett.* **17**, 4067 (2017).
- ³¹ C. Gréboval, U. Noumbe, N. Goubet, C. Livache, J. Ramade, J. Qu, A. Chu, B. Martinez, Y. Prado, S. Ithurria, A. Ouerghi, H. Aubin, J.-F. Dayen, and E. Lhuillier, *Nano Lett.* **19**, 3981 (2019).
- ³² X. Tang, M. Chen, A. Kamath, M.M. Ackerman, and P. Guyot-Sionnest, *ACS Photonics* **7**, 1117 (2020).
- ³³ Y. Che, Y. Zhang, X. Cao, X. Song, H. Zhang, M. Cao, H. Dai, J. Yang, G. Zhang, and J. Yao, *Appl. Phys. Lett.* **109**, 263101 (2016).
- ³⁴ X. Song, Y. Zhang, H. Zhang, Y. Yu, M. Cao, Y. Che, H. Dai, J. Yang, X. Ding, and J. Yao, *Nanotechnology* **28**, 145201 (2017).
- ³⁵ G. Konstantatos, M. Badioli, L. Gaudreau, J. Osmond, M. Bernechea, F.P.G. de Arquer, F. Gatti, and F.H.L. Koppens, *Nat. Nanotechnol.* **7**, 363 (2012).
- ³⁶ Y. Zhang, X. Song, R. Wang, M. Cao, H. Wang, Y. Che, X. Ding, and J. Yao, *Nanotechnology* **26**, 335201 (2015).
- ³⁷ M. Grotevent, C.U. Hail, S. Yakunin, D. Bachmann, M. Calame, D. Poulidakos, M.V. Kovalenko, and I. Shorubalko, (2020). <https://doi.org/10.26434/chemrxiv.12871331.v1>
- ³⁸ X. Tang and K.W.C. Lai, *ACS Appl. Nano Mater.* **2**, 6701 (2019).
- ³⁹ N. Huo, S. Gupta, and G. Konstantatos, *Adv. Mater.* **29**, 1606576 (2017).
- ⁴⁰ I. Nikitskiy, S. Goossens, D. Kufer, T. Lasanta, G. Navickaite, F.H.L. Koppens, and G. Konstantatos, *Nat. Commun.* **7**, 11954 (2016).
- ⁴¹ D. Spirito, S. Kudera, V. Miseikis, C. Giansante, C. Coletti, and R. Krahne, *J. Phys. Chem. C* **119**, 23859 (2015).
- ⁴² Y. Che, Y. Zhang, X. Cao, H. Zhang, X. Song, M. Cao, Y. Yu, H. Dai, J. Yang, G. Zhang, and J. Yao, *ACS Appl. Mater. Interfaces* **9**, 32001 (2017).
- ⁴³ U.N. Noubé, C. Gréboval, C. Livache, T. Brule, B. Doudin, A. Ouerghi, E. Lhuillier, and J.-F. Dayen, *Adv. Funct. Mater.* **29**, 1907462 (2019).
- ⁴⁴ S. Keuleyan, E. Lhuillier, and P. Guyot-Sionnest, *J. Am. Chem. Soc.* **133**, 16422 (2011).
- ⁴⁵ C.L. Wu, H. Yuan, Y. Li, Y. Gong, H.Y. Hwang, and Y. Cui, *Nano Lett.* **18**, 2387 (2018).