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Temperature dependence of optical properties of InAs/InP quantum rod-nanowires grown on Si substrate

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ABSTRACT

The emergence of semiconductor nanowires (NWs) as a new class of functional materials has generated a great interest in the scientific community in the fields of electronics, photonics and energy. In this work, we report on the optical properties of telecom-band emitting InAs/InP quantum rod-nanowires (QR-NWs) grown on silicon substrates by gold catalyst assisted molecular beam epitaxy (MBE). The energies of A and B band transitions in wurtzite InAs QRs are numerically evaluated by finite element method (FEM) as a function of the QR geometry and strain and compared with the experimental results obtained from photoluminescence (PL). Temperature-dependent optical properties of the QR-NWs are studied revealing that the integrated PL intensity keeps up to 30% of its value at 14 K which testify a high stability of the PL intensity. Furthermore, the investigated nanostructure shows a room temperature emission wavelength at 1.55 μm . These results demonstrate a great promise for telecom-band III-V nanoemitters monolithically grown on silicon.

1. Introduction

III-V nanowires (NWs), such as InAs, GaAs or InP, are ideal for photonic devices due to their direct band gap and superior electrical properties [1]. These NWs have been proposed for a wide variety of applications ranging from interconnects to functional device elements in electronic and optoelectronic applications [2–6]. Such structures are typically fabricated by a bottom-up approach using the Vapor-Liquid-Solid (VLS) method [7]. The NW growth using VLS method is well adapted to grow III-V semiconductors on silicon (Si) substrate which enables III-V NW optoelectronic devices to be integrated with the established Si microelectronics technology [8]. In contrast to the stable zinc blende (ZB) phase in bulk III-V materials, the Au catalyst growth of III-V NWs occurs preferentially in the hexagonal wurtzite (WZ) phase [9–11] in which the valence band splits in 3 bands, the so called A, B and C bands [12]. Furthermore, NWs with built-in homojunctions or heterojunctions, realized by variation of doping and composition along their length, constitute a new class of promising one dimensional semi-

conductor heterostructures [13,14]. The insertion of low dimensional quantum heterostructures in NWs adds new degrees of design freedom for optimization of electrical and optical performances [15]. Generally, the VLS approach allows for the growth of quantum rods of any desired height without the limits inherent to the growth of InAs quantum dots using the Stranski-Krastanov (SK) growth mode [16]. Accordingly, InP NWs with InAs QRs are particularly interesting for fiber optic communication applications through the ability to reach the 1.55 μm emission wavelength at room temperature [9]. The difference on the crystal lattice between the InAs QR and the InP NW produces an elastic strain which affects the optoelectronic properties of such nanostructures [17]. To ensure efficient optoelectronic devices operating at room temperature, a clear understanding of the effect of temperature on the optical properties of such structure is also highly required. In this context, we report on the optoelectronic properties of InAs/InP quantum rod-nanowires (QR-NWs) grown on Si substrates using the finite element method (FEM) taking into account the strain effect and we investigate their temperature dependent photoluminescence (TDPL) properties.

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2. Materials and methods

The InAs/InP QR-NWs are grown on Si(001) substrates by solid-source molecular beam epitaxy (ssMBE) and VLS method using Au-In droplets obtained by dewetting of a Au-In film at 700 °C [18,19]. The NWs are grown with a V/III beam equivalent pressure (BEP) ratio of 19. The growth begins with 2 min of InP at 460 °C. A growth interruption of 5 s is realized by closing the indium shutter. During this interruption, the phosphorus flux is switched to arsenic flux. Then, the growth of the InAs insertion is done by opening the indium shutter for a growth time of 20 s at 500 °C. After a 5 s growth interruption for the arsenic/phosphorus flux switching, the final InP axial growth is done by opening the indium shutter for a growth duration of 2 min at 460 °C. The structure is completed with 10 min of InP radial growth at 340 °C. For the photoluminescence (PL) measurements, the samples are mounted in a closed-cycle temperature-controlled He cryostat, where the temperature can be varied from 14 to 300 K, and excited by a continuous-wave off-resonant excitation. The spectra are collected by a liquid nitrogen-cooled InGaAs detector with a cutoff wavelength at 2.2 μm using a conventional lock-in technique.

3. Results and discussion

The morphology and structure of the QR-NWs are studied by transmission electron microscopy (TEM) imaging. Fig. 1(a) shows a schematic of the investigated InAs/InP QR-NWs grown on the Si substrate. The NW diameter and length are in the 80–100 nm and 1.5–2 μm range, respectively. The length and diameter of the InAs QRs are in the range of 45–135 nm and 8–12 nm, respectively. These NWs have a WZ structure with the [0001] axis (c-axis) along the NW growth direction for both InP and InAs materials (Fig. 1(e)). Due to the high V/III BEP ratio, no stacking faults or cubic segments were observed in the QR-NWs (Fig. 1(b–e)). Similar structures are studied by SEM and show the typical four <111> directions emerging from the (001) Si [20,21]: four directions are forming a 35° angle with the Si surface and a 90° angle from each other, and this result was also observed by our group [22]. In order to improve our understanding of the optical properties of such InAs/InP QR-NWs, low temperature PL spectroscopy experiments under different excitation powers were performed (Fig. 2).

At low excitation power, the PL spectrum shows a single peak around 0.87 eV (peak A) attributed to the ground state emission of the InAs QRs. The full width at half maximum (FWHM) is about 60 meV for this peak. The low temperature inhomogeneous broadening of the spectra is associated with the InAs QR size dispersion, which is consistent with the TEM results. When the excitation power is increased, a second peak (peak B) appears at higher energy (~0.95 eV). According to our TEM studies, the QR diameter D is 4–14 times shorter than the QR length. The direct consequence of this specific geometry is that the quantum confinement of the first electron and hole states (called ground states) are strongly related to the QR diameter. In QRs, the energy spacing between the ground states and the first excited states is small when $L \gg D$. This is confirmed by the work of different groups which have already investigated the electron and hole states in such QRs (which are also called quantum posts or columnar QDs) [23–25]. Moreover, a 70 meV energy spacing has been predicted for a cylindrical InAs/InP QR with a diameter of 9.6 nm and a height of 18 nm [26]. As a consequence, the 80 meV energy spacing between the peak A and peak B cannot be related to the energy spacing between the ground and first excited states in our $D \sim 10$ nm and $L > 45$ nm QRs, and another explanation has to be found.

In order to get additional information on the origin of this peak, the WZ InAs/InP QR-NWs have been numerically investigated by finite element method (FEM) by solving the single band effective mass Schrödinger equation in cylindrical coordinates taking into account the strain effects [27,28]. The nth transition energy can be obtained as follow:

$$E_n = e_n + E_{A,B}^{InAs} + h_n \quad (1)$$

where e_n and h_n , are the electron and hole confined energies, respectively. $E_{A,B}^{InAs}$ is the A or B strained InAs band gap energy. The strain dependence of E_A^{InAs} is given by Ref. [29–31]:

$$E_A^{InAs} = E_A^{0 InAs} + (a_{cz} - D_1 - D_3) \varepsilon_{zz} + (a_{ct} - D_2 - D_4) \varepsilon_{\perp} \quad (2)$$

where $E_A^{0 InAs}$ is the A unstrained band gap energy and a_{cz} , a_{ct} and D_i are the deformation potentials of WZ InAs, $\varepsilon_{\perp} = (\varepsilon_{rr} + \varepsilon_{\theta\theta})$ and ε_{zz} are the strain components parallel and perpendicular to the c-axis, respectively. If the QR is approximated by a cylinder, the strain components can be described by the Eshelby model [17,32,33]:

$$\begin{cases} \varepsilon_{rr} = \varepsilon_{\theta\theta} = f_{cs} - f_{shear} \phi\left(\frac{L}{D}\right) \\ \varepsilon_{zz} = f_{cs} + 2f_{shear} \left[\phi\left(\frac{L}{D}\right) - 1 \right] \end{cases} \quad (3)$$

where f_{cs} is the lattice mismatch perpendicular to the NW axis defined by $f_{cs} = \frac{a_{InP} - a_{InAs}}{a_{InAs}}$ where a_{InP} and a_{InAs} are the lattice constant of InP and InAs, respectively. The quantity f_{shear} is defined by: $f_{shear} = \frac{1}{2} f_{cs} \frac{3\lambda + 2\mu}{\lambda + 2\mu}$, where λ and μ are the Lamé coefficients. For an isotropic crystal, $\mu = \frac{1}{2} (c_{11} - c_{12})$ and $\lambda = c_{12} \cdot \phi\left(\frac{L}{D}\right)$ represents the shape factor where D is the diameter and L the length of the QR. $\phi(x)$ is given by the following analytical formula [34]:

$$\begin{cases} \phi(x) = 1 + \frac{1 - \frac{x}{\sqrt{x^2 - 1}} \cosh^{-1} x}{x^2 - 1}, \text{ when } x > 1 \\ \phi(x) = 1 - \frac{1 - \frac{x}{\sqrt{1 - x^2}} \cos^{-1} x}{1 - x^2}, \text{ when } x < 1 \end{cases} \quad (4)$$

Fig. 3(a) shows the values of ε_{rr} and ε_{zz} as a function of the QR aspect ratio (L/D). Finally, the energy difference between A and B band energies, Δ_{AB} is deduced from the following equation [30,31]:

$$\begin{aligned} \Delta_{AB} &= \frac{\Delta_{cr}}{2} + \frac{\Delta_{so}}{2} + \frac{D_3}{2} \varepsilon_{zz} + \frac{D_4}{2} \varepsilon_{\perp} \\ &- \sqrt{\left[\frac{\Delta_{cr} - \frac{1}{3} \Delta_{so} + D_3 \varepsilon_{zz} + D_4 \varepsilon_{\perp}}{2} \right]^2 + \frac{2}{9} \Delta_{so}^2} \end{aligned} \quad (5)$$

where Δ_{cr} , Δ_{so} are the crystal field and spin orbit splittings. The value of the deformation potentials, crystal field, spin orbit splitting and unstrained band gap energies used in the calculations are listed in Table 1. All these values are theoretical values with the exception of the WZ InAs band gap being experimentally measured in Refs. [35].

Fig. 3(b) shows the QR aspect ratio dependence of the A and B band gap energies. For infinite QR ($\frac{L}{D} \gg 1$), the energy difference between A and B band gap energies remains constant. In the other hand, for flat QR ($\frac{L}{D} < 1$), the A and B band gap energies are strongly affected by the strain effect. In our case, the calculations are based on WZ InP NWs with infinite length and diameter. Fig. 4(a) shows the schematic structure of the band alignment of the WZ InAs QR inserted in a WZ InP NW used in our calculations.

The parameters used in the energy level calculations are given in Table 2, where m_e^* , m_A^* and m_B^* are the effective masses for the electron, A-band hole and B-band hole in m_0 units (free electron mass), respectively. According to the selection rule in NWs with WZ structure, the emission from recombination of electrons in the conduction band with holes from A-band is expected to be polarized perpendicularly to the NW axis, while for the recombination with holes from B-band, both perpendicular and horizontal polarizations are allowed [12,18]. In that

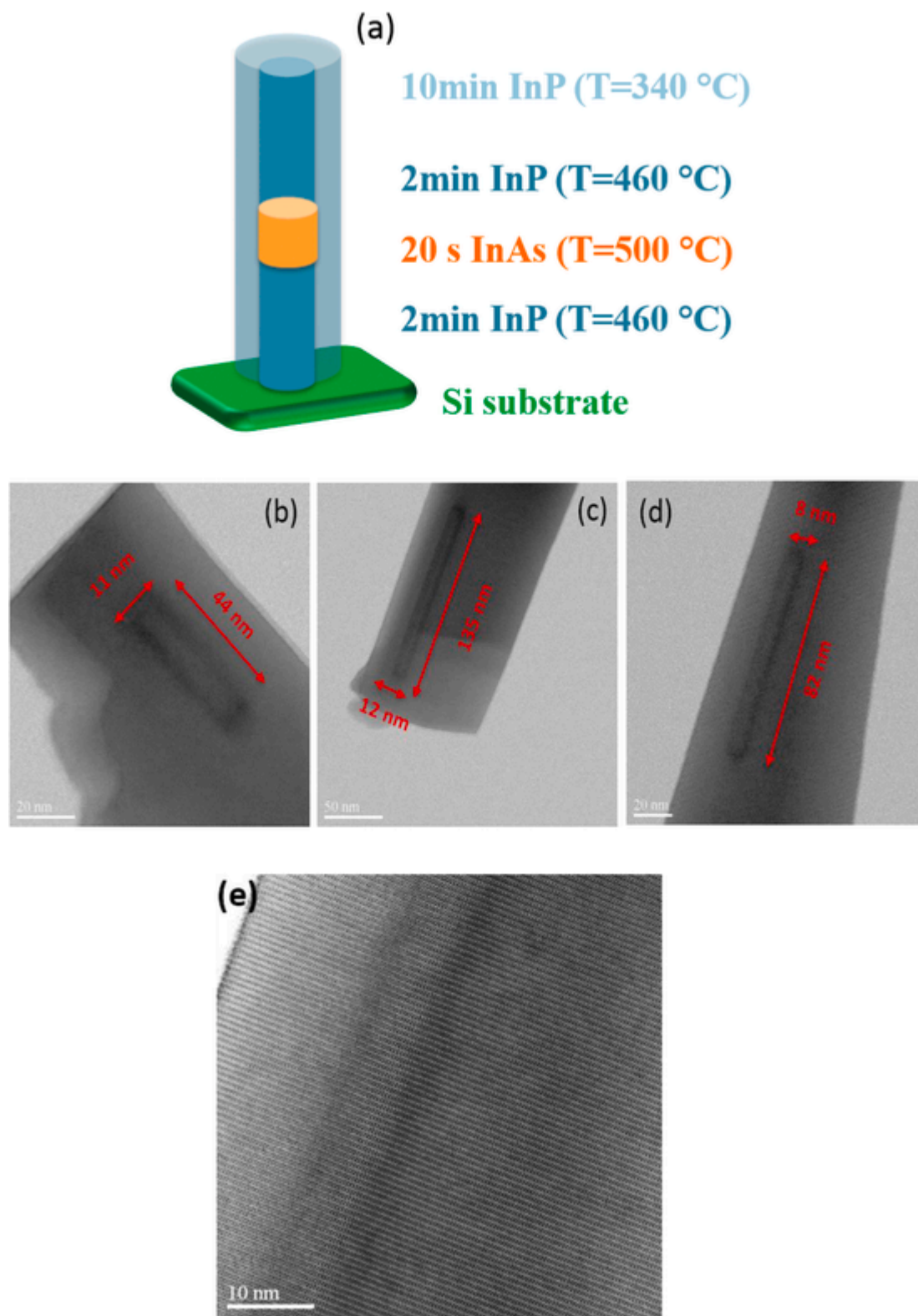


Fig. 1. (a) Schematic of the investigated sample. (b–d) TEM images showing InP NWs with different dimensions of InAs QRs. (e) A close-up HAADF-TEM image of the InAs/InP interface.

case, the effective masses perpendicular to the NW axis are used in our calculations.

The ground state energies of A and B transitions as a function of the InAs QR diameter and length are represented in Fig. 4(b).

The transition energy is found to exhibit a strong variation up to a QR length around 15 nm. However, no major contribution is observed for longer QRs. When the QR length increases from 2 nm to 60 nm, the energy spacing between the A and B transition energies decreases from ≈ 114 meV to ≈ 23 meV. Whereas a transition from heavy hole-like to

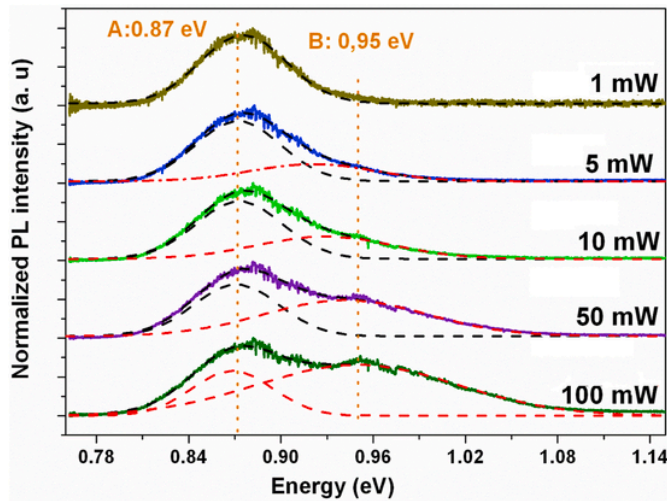


Fig. 2. Normalized PL spectra at 14 K of InAs/InP QR-NWs as a function of the excitation power. The solid lines are the experimental data and the dotted lines are the Gaussian fit of the PL spectra.

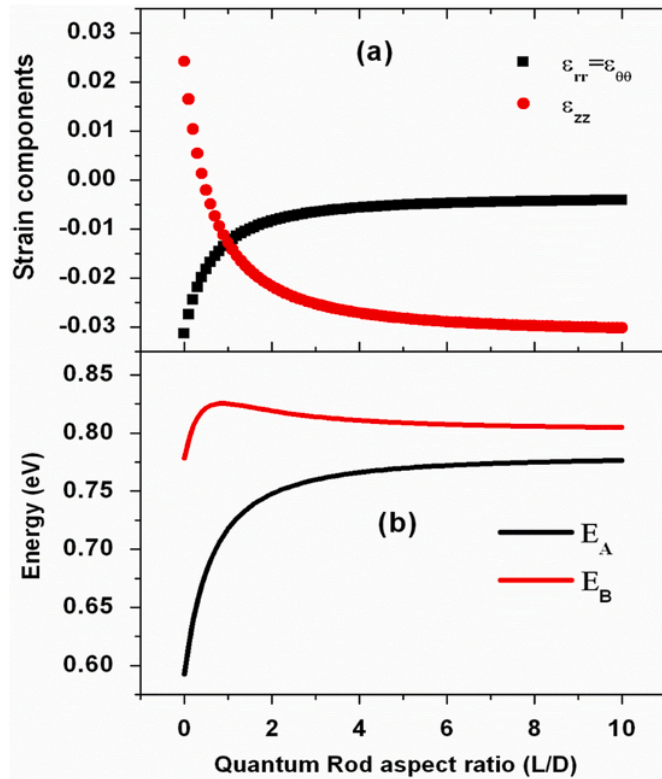


Fig. 3. (a) Strain components and (b) Energies of the strained A and B band gap energies, as a function of the QR aspect ratio (L/D).

light hole-like ground state is predicted in ZB InAs QRs [16], in WZ structures the ground state emission is always related to A-band holes. By comparing the calculated and the experimental peak energies, a QR diameter of about 10–14 nm and an average QR length of more than

15 nm can be deduced. In that case, InP NWs with larger QR diameter greater than 10 nm (Fig. 1(b)-1(c)) dominate the PL spectra. As far as the theoretical results are concerned, several phenomena could explain the discrepancy: the piezoelectric field [19], the strain dependence of the effective masses [37] and the need to take into account the exciton binding energy. Furthermore, a discrepancy is observed in the literature on the theoretical values of some fundamental parameters of WZ InAs [29,35,36,38].

To ensure efficient optoelectronic device operating at room temperature, a clear understanding of the effect of temperature on the optical properties of such structure is highly required. Fig. 5 shows the PL spectra of the sample from 14 K to 300 K using low laser excitation power.

The PL emission peak is shifted toward higher wavelength, up to 1.55 μm at room temperature with a PL intensity keeping as much as 30% of its value at 14 K. This figure is close to the best values (50–67%) reported for ZB InAs QDs or quantum dashes grown in an InP matrix [39,40]. As the temperature increases, the QR-NW emission peak due to the thermal escape of carriers from A-band to B-band. A multi-Gaussian fitting is performed on each PL spectrum to take into account the fact that two peaks are observed on the spectrum at room temperature with an average energy spacing of 70 ± 5 meV which is close to the average energy splitting observed between the A- and B-bands at low temperature.

The temperature dependence of the A exciton integrated PL intensity is shown in Fig. 6(a). A fit of the experimental data has been performed using the following multi-channel Arrhenius equation [41]:

$$I = \frac{I_0}{1 + \sum_i c_i \exp\left(-\frac{E_i}{k_b T}\right)} \quad (6)$$

where i represents the number of involved channels for carrier thermal escape, k_b is the Boltzmann constant, E_i is the thermal activation energy, I_0 is the PL intensity at 0 K and c_i is a constant. We found that the temperature dependence of the integrated PL intensity can be well fitted by considering two escape channels ($i = 2$). The best fit gives the values of E_1 and E_2 around $10 (\pm 2)$ meV and $125 (\pm 10)$ meV, respectively. These values are comparable with the findings of Wu et al. [42] where 14.3 and 73.4 meV activation energies were obtained for GaAs/GaAsP NW quantum dots and associated with the GaAs surface states, which limits the optical properties of the NWs, and to thermal excitation of photocarriers from the GaAs QDs into the GaAsP barriers, respectively. Alonso-Álvarez et al. [43] also reported 11 and 170 meV activation energies for InAs/GaAsP quantum posts (QPs). The lower activation energy is associated to carrier recombination through defects in the quantum structures or their interfaces while the higher activation energy is compatible with the unipolar escape of carriers from the nanostructures to the matrix [43]. If we assume an isotropic dielectric constant, the WZ InAs exciton binding energy can be estimated using the following equation:

$$E_b = \frac{13.58 \text{ eV}}{\varepsilon(0)^2} \left[\frac{2}{3} \left(\frac{1}{m_c^\perp} + \frac{1}{m_A^\perp} \right) + \frac{1}{3} \left(\frac{1}{m_c^\parallel} + \frac{1}{m_A^\parallel} \right) \right]^{-1} \quad (7)$$

where $\varepsilon(0)$ is the dielectric constant (15.15 for ZB InAs [44]), m_c and m_A are the electron and A-band hole effective masses that takes in account the existing anisotropy in the WZ InAs effective masses. Using the parameters given in Table 2, we thus obtain $E_b = 2$ meV. This value does not differ from that of the free exciton in bulk ZB InAs

Table 1

The deformation potentials a_{cz} , a_{cr} and D_i , the crystal field splitting Δ_{cr} , spin orbit splitting Δ_{so} and the unstrained band gap energy $E_A^{0 \text{ InAs}}$ (eV) used in calculations.

Δ_{so} (eV)	Δ_{cr} (eV)	$E_A^{0 \text{ InAs}}$ (eV)	D_1 (eV)	D_3 (eV)	$a_{cr} - D_2$ (eV)	$a_{cz} - D_1$ (eV)	InAs (WZ)
0.379 [36]	0.195 [36]	0.477 [35]	-3.17 [29]	5.85 [29]	-8.33 [29]	-2.69 [29]	

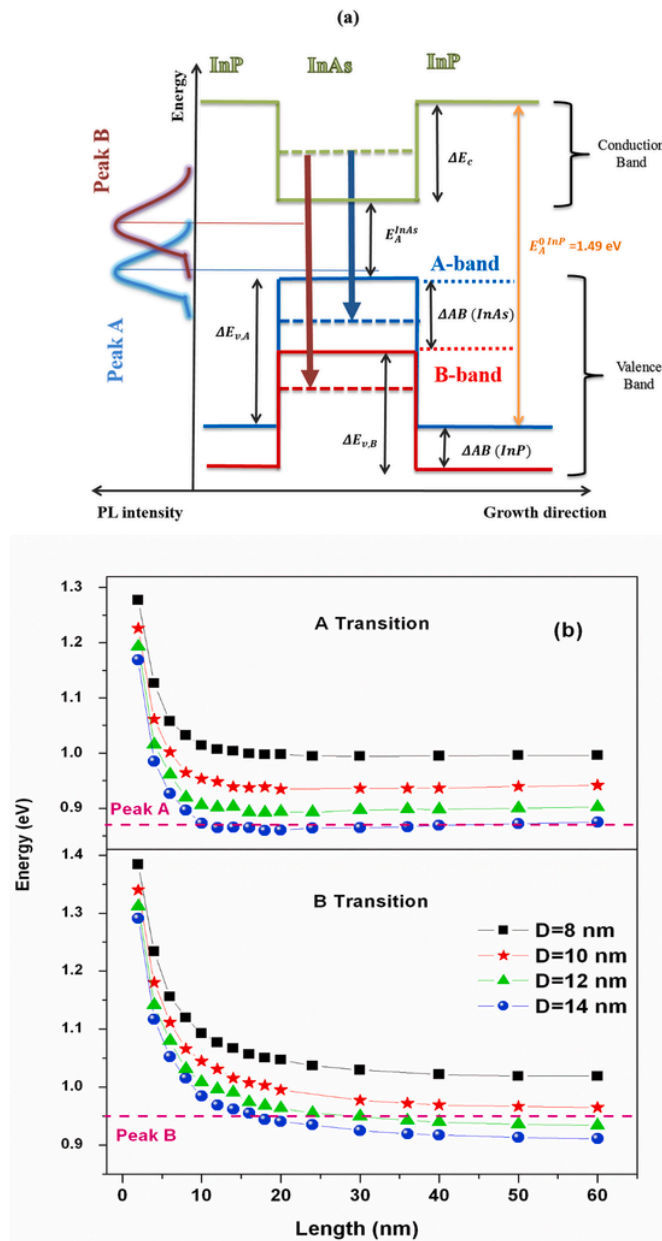


Fig. 4. (a) Schematic representation of the band alignment for WZ InAs/WZ InP used in our calculation. (b) Simulated A and B transition energies of WZ InAs/InP QR-NWs as a function of the QR diameter (D) and length (L) (from 2 nm to 60 nm). Horizontal dashed lines represent the energies of A and B peaks observed by PL.

(1 meV) [44]. Moreover, the exciton binding energy is in the 5–25 meV range for SK InAs/InP QDs [45,46]. In that case, the smaller activation energy $E_1 = 10$ meV can be associated to the exciton binding energy in WZ InAs. However, the high temperature (160–300 K temperature range) activation energy E_2 is about five times less than the difference (≈ 620 meV) between the energy of the WZ InP barrier band gap (1.49 eV) and the energy of the PL emission of the A transition

Table 2

WZ InAs and WZ InP effective masses and lattice parameters used for the calculations. $m_{A,B}^\perp$ and $m_{A,B}^\parallel$ are the effective masses perpendicular and parallel to the c-axis, respectively.

	c (Å)	a (Å)	m_c (m_0)	m_A^\perp (m_0)	m_A^\parallel (m_0)	m_B^\perp (m_0)	m_B^\parallel (m_0)
InP (WZ)	6.77 [29]	4.15 [29]	0.088 [36]	0.158 [36]	1.273 [36]	0.169 [36]	0.839 [36]
InAs (WZ)	7 [29]	4.283 [29]	0.042 [36]	0.084 [36]	1.7 [36]	0.113 [28]	0.101 [36]

(0.87 eV). In the other hand, this value is close in magnitude to the energy spacing between the A- and B-bands (80 meV). In that case, E_2 can be associated to the thermally populated high-energy states in the B-band. This thermal population is confirmed by the relative increase of B transition PL intensity along with the temperature as observed for WZ InP NWs [47].

Fig. 6(b) shows the FWHM of the A peak as a function of temperature. The FWHM almost remains constant for temperature below 100 K. With further increasing temperature to 160 K, the FWHM was slightly increased by about 5–6 meV and then remains constant at higher temperature range. This behavior has been observed for SK InAs/GaAs QDs and attributed to the thermal population of excited hole states [48]. As far as our sample is concerned, this behaviour might be attributed to the thermal population of the B-band. This is in a good agreement with studies on the temperature dependence of the A transition integrated PL intensity.

The temperature dependence of the emission energy of the A peak is plotted in Fig. 6(c). Generally, this behaviour is discussed in terms of the empirical equation of Varshni [49] and Bose-Einstein formula [50] given by:

$$E_g(T) = E_g(T) + \frac{\alpha T^2}{T + \beta} \quad (8)$$

$$E(T) = E_B - a_{BE} \left[1 + \frac{2}{\exp\left(\frac{\theta_{BE}}{T}\right) - 1} \right] \quad (9)$$

where α and β parameters are the Varshni coefficients, a_{BE} represents the strength of the electron-phonon interaction and θ_{BE} is the average phonon temperature. Solid and dashed lines in Fig. 6(c) are the fitting curves of the experimental data based on equations (8) and (9), respectively. The best fitting parameters for the two models are summarized in Table 3 along with parameters from the literature. These parameters are closed to the values reported in the literature for ZB InAs which indicates that ZB and WZ InAs have similar thermal parameters as observed for InP [47,54] and GaAs [55] (even if there is some discrepancy for this semiconductor material [56]). The low temperature data points (< 100 K) are systematically below the Varshni curves by about 5 meV. This value is close in magnitude to the smaller activation energy E_1 calculated from the fitting of the temperature dependence of the integrated PL intensity by the Arrhenius equation and attributed to the formation of excitons at low temperatures. When the temperature is raised further, the energy shift follows the Varshni law. This constitutes an additional proof of the good quality of our InAs/InP QR-NWs grown on silicon.

4. Conclusions

In summary, we have grown high quality InAs/InP QR-NWs on Si(001) substrates by VLS assisted MBE. Optical studies performed at cryogenic temperature reveal an InAs QR emission in the telecom band with two peaks. FEM calculations taking into account strain effects in the InAs QR were developed to explain these experimental results. Temperature-dependent optical properties revealed a high stability of the PL intensity in the 14–300 K temperature range and showed an emission at the 1.55 μm wavelength at room temperature. These results confirm a great promise for InAs-InP QR-NWs for telecom-band na-

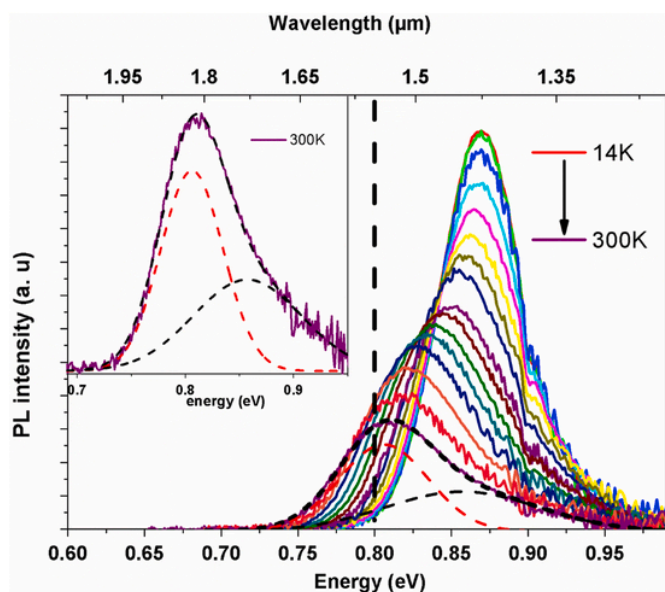


Fig. 5. Temperature induced changes in the PL spectrum for InAs/InP QR-NWs on Si(001) using low laser excitation power. The solid lines are the experimental data and the dotted lines are the multi-Gaussian fitting of the room temperature PL spectra. The vertical dashed line indicates the 1.55 μm emission wavelength.

noemitters monolithically integrated on silicon substrates for photonic chips.

Author statement

Mohamed Helmi Hadj Alouane, Nicolas Chauvin and Catherine Bru-Chevallier carried out the PL measurements on QR-NWs, data analyses, strain calculations and wrote the major part of the paper. Olfa Nasr, Mohamed Hadj Alouane, Mohamad Mahmoud Ahmad and Bouraoui Ilahi contributed in data analyses and FEM calculations. Hammadi Khmissi and Michel Gendry conducted the MBE growth. Gilles Patriarche performed TEM measurements. All authors reviewed the manuscript.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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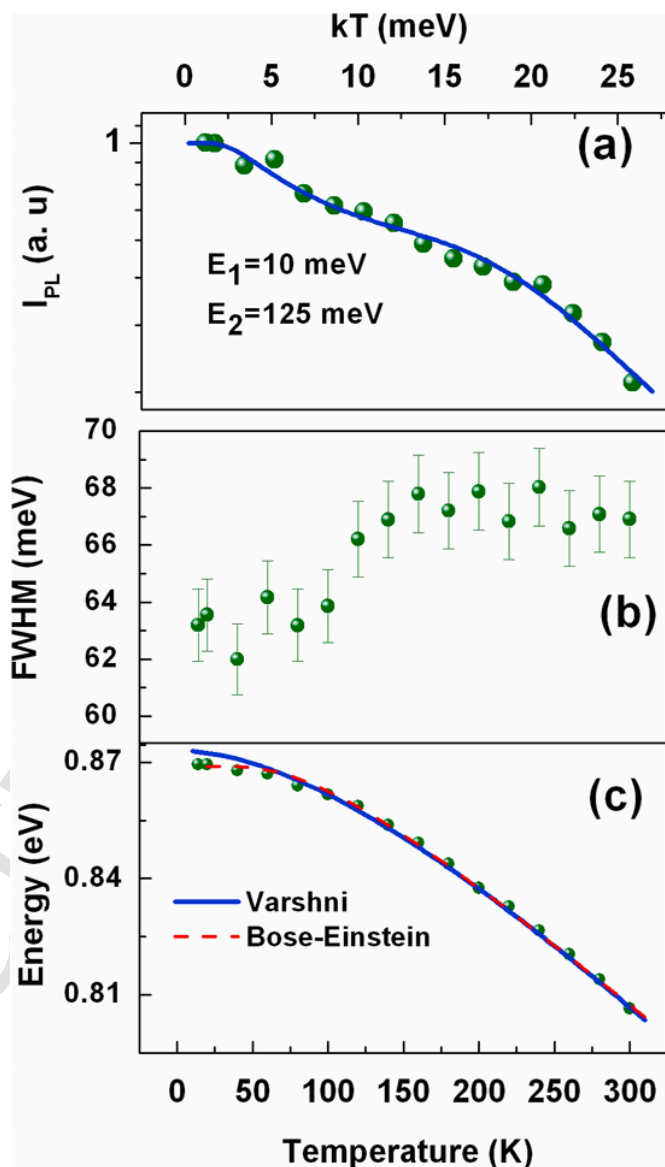


Fig. 6. Temperature dependence of the: (a) integrated PL intensity, (b) FWHM and (c) PL peak energy of A exciton (green circles). The solid curves give the fitting result.

Table 3

Varshni and Bose-Einstein parameters extracted from the fitting of temperature dependence of the band gap energy for InAs/InP QR-NWs.

θ_{BE} (K)	a_{BE} (meV)	β (K)	α (10^{-4} eV/K)	
270 ± 10	45 ± 10	270 ± 10	4.2 ± 0.5	This study
270 [52]	99 [52]	271 [51]	4.19 [51]	
247 [50]	56 [50]	93 [53]	2.76 [53]	

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